#### **ORIGINAL PAPER**



# **Tuning the magnetic properties of neodymium (Nd)‑doped cobalt ferrite thin flms through nebulizer spray technique**

A.M. S. Arulanantham<sup>1</sup> · K. V. Gunavathy<sup>2</sup><sup>®</sup> · M. Antony<sup>3</sup> · N. Sundaramurthy<sup>4</sup> · M. Maria Stephy<sup>1</sup> · P. Mohanraj<sup>5</sup> · **V. Ganesh<sup>6</sup>**

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### **Abstract**

A thin film of cobalt ferrite  $(CoFe<sub>2</sub>O<sub>4</sub>)$  was successfully deposited through nebulizer assisted spray pyrolysis technique. Composition of elements in the thin flm plays an important role in determining its magnetic properties. Signifcant changes in the structural and magnetic properties can be observed in the  $\text{CoFe}_2\text{O}_4$  thin films when doped with rare earth elements making it suitable for the applications in magnetic and related devices. Undoped and neodymium-doped  $\text{CoFe}_2\text{O}_4$  thin films were prepared through nebulizer assisted spray pyrolysis (NSP) method on bare glass substrates by changing the percentage of Nd doping level from 0 to 5% atomic weight. Standard characterization techniques were used to analyze the deposited flms to reveal the efect of doping. Crystallite size is estimated from x-ray difraction spectra, and is observed to be of the order of 14 nm. The size of the crystallites in the flm seems to decrease with the increase in doping concentration. The optical band gap value shows an increase and gets shifted from 1.82 to 1.95 eV on increasing the concentration of Nd doping. The saturation magnetization of the prepared CoFe<sub>2</sub>O<sub>4</sub> films obtained at 0% and 5% show a strong dependence on the Nd doping concentration. This study shows that the Nd-doped CoFe<sub>2</sub>O<sub>4</sub> films can compete with the existing materials for magnetic device applications.

**Keywords**  $\text{CoFe}_2\text{O}_4 \cdot \text{Magnetic properties} \cdot \text{Nebulizer assisted spray pyrolysis} \cdot \text{Rare earth doping} \cdot \text{Thin films}$ 

 $\boxtimes$  K. V. Gunavathy gunavathy.kec@gmail.com

- <sup>1</sup> Department of Physics, St. Joseph College of Arts and Science, Vaikalipatti 627808, India
- <sup>2</sup> Thin Film Research Centre, Kongu Engineering College, Perundurai 638060, India
- <sup>3</sup> PG and Research Department of Physics, Arul Anandar College, Karumathur 625 514, India
- <sup>4</sup> Department of Physics, Annamalai University, Chidambaram 608002, India
- <sup>5</sup> Department of Physics, Pondicherry University, Puducherry 605014, India
- <sup>6</sup> Advanced Functional Materials and Optoelectronic Laboratory (AFMOL), Department of Physics, Faculty of Science, King Khalid University, P.O. Box. 9004, Abha 61413, Saudi Arabia

# **Introduction**

The architecture of the thin flm plays a vital role in most of the electronic devices that use magnetic materials. The magnetic thin flms are commonly utilized in technological applications such as magnetic recording and in microelectromechanical applications (Garcia-Sanchez et al. [2008](#page-8-0)). Spinel ferrite materials are interesting since they exhibit ferrimagnetic and semiconductor properties that suggest their usage as magnetic media for in numerous applications such as in recording high density data (Khandekar et al. [2011](#page-8-1)), generation and detection of ultrasonic waves, electronics and telecommunication (Šutka and Mezinskis [2012\)](#page-9-0), gas sensors (Deraz [2010\)](#page-8-2), torque sensors and magnetic hyperthermia (Kim et al. [2008](#page-8-3)). Its characteristic property such as strong anisotropy, high mechanical hardness and chemical stability paved way for its usage in several industrial applications too. Thus, great efforts have been made for the synthesis and characterization of nano-sized  $\text{CoFe}_2\text{O}_4$  materials. The synthesis method decides the distribution of ions in diferent positions of the spinnel structure which determines the composition as well as microstructure of these flms (Tiwari et al. [2020](#page-9-1)). The larger surface area to volume ratio as the size of the bulk material is reduced to nano material creates distinct physical, chemical, mechanical and especially magnetic properties due to the presence of more number of atoms on the surface creating spin glass behavior and superparamagnetism (Cheng et al. [2005\)](#page-8-4).

Adeela et al. [\(2015](#page-8-5)) investigated the infuence of manganese substitution on structural and magnetic properties of  $CoFe<sub>2</sub>O<sub>4</sub>$  nanoparticles through the co-precipitation technique and found that the saturation magnetization and coercivity values of these thin flms increased up to 30% of Mn concentration. Rao et al. [\(2015\)](#page-8-6) showed that the saturation magnetization and remnant magnetization depend on particle size as well as crystallinity of the nanoparticles. Maaz et al. [\(2007](#page-8-7)) observed an increased magnetic moment for particles of smaller size. Houshiar et al. ([2014](#page-8-8)) compared the values of coercivity and saturation magnetization of  $\text{CoFe}_2\text{O}_4$ thin flms prepared through diferent routes and reported that the values are better when  $\text{CoFe}_2\text{O}_4$  was synthesized through the combustion method rather than using precipi-tation techniques. Gandha et al. [\(2015\)](#page-8-9) prepared  $\text{CoFe}_2\text{O}_4$ nano particles of size 40 nm with higher coercivity values using hydrothermal method. Gingasu et al. [\(2016](#page-8-10)) revealed the antibacterial effect of Ag-doped  $\text{CoFe}_2\text{O}_4$  nanoparticles prepared through self-combustion and wet ferritization using aqueous extracts of the leaves and fowers of *Hibiscus rosa sinensis*. The photocatalytic activity of these nanoparticles on rhodamine B dye was studied by Nguyen et al. (To Loan et al. [2019\)](#page-9-2). The dependence of the superparamagnetic efect on the temperature and applied magnetic feld was investigated by Ojha and Kant  $(2019)$  $(2019)$ . The photo degradation property of  $\text{CoFe}_2\text{O}_4/\text{SiO}_2$  nanocomposites using methylene blue dye was investigated by Yakob et al. [\(2019](#page-9-3)).

Doping of rare Earth such as Neodymium into cobalt ferrite compounds varies the magnetic anisotropy and results in a large magnetostriction effect even at room temperature (Forester et al. [1978](#page-8-12)). The magneto optic Kerr efect also gets increased on doping ferrites with rare Earth elements (Avazpour et al. [2016](#page-8-13); Cedeño-Mattei et al. [2010\)](#page-8-14) along with an increase in the values of coercivity (Cheng et al. [1999](#page-8-15); Karimi et al. [2014\)](#page-8-16), saturation magnetization (Yan et al. [1998](#page-9-4)) as well as conductivity (Rahman et al. [2014\)](#page-8-17).

Several synthesis techniques were used to prepare Nddoped films of  $\text{CoFe}_2\text{O}_4$  (Abbas et al. [2016](#page-8-18); Mounkachi et al. [2017](#page-8-19); Xavier et al. [2013\)](#page-9-5). There is no report available on the effect of neodymium doping on the structural, morphological, and magnetic properties of cobalt ferrite thin flms deposited through the NSP method. Nebulizer spray pyrolysis is an efective and economic method of depositing various thin flms under controlled conditions. It is a common chemical deposition method of depositing oxides. It relies on the production and transport of fne aerosols produced by the nebulizer head toward the hot substrate surface. The characteristics of the deposited flms were tailored by varying the deposition conditions such as substrate temperature, spray head and substrate distance, spray rate, solvent, etc. The method is much sought due to its favorable advantages such as ease of its assembly, the capability of doping, control over composition, moderate temperature environment, uniformity in thickness and overall quality of the flms. Hence we are focusing on the infuence of neodymium doping on the magnetic properties of  $\text{CoFe}_2\text{O}_4$  thin films fabricated using the economically feasible nebulizer assisted spray pyrolysis technique.

### **Experimental details**

Analytical grade cobalt chloride hexahydrate  $[CoCl<sub>2</sub>.6H<sub>2</sub>O]$ , ferric (III) nitrate (FeN<sub>3</sub>O<sub>9</sub>) neodymium (III) acetate  $(Nd(O_2C_2H_3)_3.H_2O)$  were used as the precursors for cobalt, ferrite and neodymium atoms respectively to deposit the  $CoFe<sub>2</sub>O<sub>4</sub>$ :Nd thin films on bare soda lime glass substrate using nebulizer spray pyrolysis technique. 0.05 M and 0.1 M concentration of cobalt chloride and ferric (III) nitrate was dissolved in de-ionized water (10 ml). The substrate temperature was maintained to be 375 °C ( $\pm$  5 °C) during the fabrication of all the samples. The compressed air is used as a carrier gas in the nebulizer spray unit to produce fne aerosols of the precursor solution and its fow rate was optimized to be 1.5 megapascal (Mpa). A 40 mm distance was fxed between the head of the spray nozzle and substrate throughout the experiment. However, the percentage of neodymium ion doping concentration was varied as 0%, 3% and 5%.

# **Characterization**

The prepared flms were analyzed using standard characterizing techniques to analyze their properties. A PANalytical PW 340/60 x-ray difractometer was used in the range of 10°to 80° with a source wavelength of 1.5416 Å. An S-3400 N model scanning electron microscope supplied by Hitachi annexed with an EDX spectrometer was utilized to analyze the surface morphology and the elemental composition in the flms. The surface morphology was studied further using Digital instruments Nanoscope IV. Perkin Elmer Lambda 35 UV–vis NIR spectrometer was employed to study the optical properties of the samples in the range of 500 nm to 2500 nm. LAKE SHORE 7404 vibrating sample magnetometer was used to study the change in the magnetic properties when Nd ions were doped into the  $CoFe<sub>2</sub>O<sub>4</sub>$ lattice.



<span id="page-2-0"></span>**Fig. 1** XRD patterns of  $\text{CoFe}_{2}\text{O}_{4}$  thin film with different Nd doping concentration

# **Result and discussion**

# **Structural characterization**

Figure [1](#page-2-0) shows the obtained x-ray difraction patterns of undoped  $\text{CoFe}_2\text{O}_4$  and Nd-doped  $\text{CoFe}_2\text{O}_4$  thin films prepared with simple nebulizer spray pyrolysis method. The polycrystalline characteristic of the flms was confrmed using JCPDS card No. 22-1086 corresponding to cubic structured  $\text{CoFe}_2\text{O}_4$  (Yadav et al. [2016](#page-9-6)). In spite of the variation in the Nd doping concentration, all the samples exhibited a dominant peak along the (311) plane representing the major orientation and phase purity of  $\text{CoFe}_2\text{O}_4$ . This XRD pattern also confirms the growth of  $\text{CoFe}_2\text{O}_4$ polycrystalline thin flms with crystallites also oriented along (220) and (511) directions. Similar single phase presence of these polycrystalline flms was reported earlier using the spray pyrolysis technique (Zongyan and Xiang [2015\)](#page-9-7) and also in spin coating methodology (Bagade and Rajpure [2015](#page-8-20)). As observed in Fig. [1,](#page-2-0) the intensity of the  $CoFe<sub>2</sub>O<sub>4</sub>$  films decreases with the increase in the Nd doping concentration from 0 to 5% which might be due to the increase in the values of stress in the deposited flms because of the radius of the Nd<sup>3+</sup> ion (0.998 Å). No other peak is detected in the XRD studies for any other impurities other than neodymium (Nd). A considerable decrease in the peak intensity is seen in the XRD spectra with the increase in the Nd doping concentration particularly along (311) and (220) planes. This decrease may be related to the structural disorder in the flms induced by the introduction of dopant ions (Fitriyanti and Utari [2017](#page-8-21)).

The crystallite size (D) of thin flms was estimated with reference to the plane (311) using Scherrer's formula (Srivastava et al. [1982\)](#page-9-8)

$$
D = \frac{k\lambda}{\beta \cos \theta} \tag{1}
$$

The obtained crystallite size through this equation is associated to the mean minimum dimension of a coherent difraction domain. It is found that the crystallite size estimated from (311) peak of  $\text{CoFe}_2\text{O}_4$  is decreased from 19 to 14 nm when Nd doping concentration was raised from 0 to 5%. When the dopant concentration increases, the number of nucleation site increases and it leads to a decrease in the crystallite size (Garcés Pineda et al. [2015\)](#page-8-22).

The measured crystallite size and additional parameters of the  $CoFe<sub>2</sub>O<sub>4</sub>$  films are presented in Table [1](#page-2-1).

Dislocation density  $(\delta)$  and strain  $(\varepsilon)$  was calculated from crystallite size using equations (Tiwari et al. [2020\)](#page-9-1).

$$
\delta = \frac{1}{D^2} \tag{2}
$$

$$
\varepsilon = \frac{\beta \cot \theta}{4} \tag{3}
$$

The variation of *δ* and *ε* of undoped and Nd-doped  $CoFe<sub>2</sub>O<sub>4</sub>$  thin films are given in Table [1.](#page-2-1) It is observed from Table [1](#page-2-1) that the dislocation density and micro-strain values show an increase when the Nd doping level is increased. The increment in the values of strain causes a decrement in the size of the crystallites.

#### **Morphological analysis using SEM**

In order to understand the structure of grains in the prepared flms, SEM micrographs are taken and Fig. [2a](#page-3-0)–c shows the SEM images of nano grain structured Nd:  $\text{CoFe}_2\text{O}_4$ thin flms with doping concentrations 0%, 3%, and 5%, respectively.

From the fgure, it is observed that all the samples show small nano sized spherical shaped grains uniformly covering the substrate. The grain size gradually decreases as the Nd doping concentration was raised from 0 to 5%. When

<span id="page-2-1"></span>**Table 1** Structural parameters of deposited  $\text{CoFe}_2\text{O}_4$  thin films in different Nd doping concentration

Nd doping concentration $(\%)$	Crystallite $size$ (nm)	Dislocation density $(\delta \times 10^{15} \text{ lines. m}^{-2})$	Strain $(\varepsilon \times 10^{-3})$
$0\%$	19	2.77	2.43
3%	16	3.90	2.75
5%	14	4.43	3.14

flms was calculated to be 12.6 nm, 13.8 nm and 17.1 nm, respectively.

### **Compositional analysis**

The elemental analysis of pristine and 5% Nd-doped  $CoFe<sub>2</sub>O<sub>4</sub>$  thin films was carried out by energy-dispersive x-ray spectroscopic technique.

Figure [4](#page-5-0) depicts the EDX spectra of undoped and 5% Nd-doped spinel cobalt ferrite thin flms. It is seen from Fig. [4a](#page-5-0) that the constituent elements Co, Fe and O are present. The presence of rare earth Nd in the deposited flm was ascertained by its presence in the EDX spectra as shown in Fig. [4b](#page-5-0). The spectra also confrm the fact that the prepared thin flms were free from other impurities except for the peaks for Si from the substrate. The corresponding elemental

<span id="page-3-0"></span>**Fig. 2** SEM images of  $\text{CoFe}_2\text{O}_4$  thin films **a** 0% Nd, **b** 3% Nd, **c** 5% Nd

neodymium ions replace ferric ions in the  $CoFe<sub>2</sub>O<sub>4</sub>$  lattice, strains were produced due to the change in lattice parameters and hence a stress is induced in the lattice which hinders the growth of grains. Thus when the doping concentration increases, grain growth gets decreased (Zubair et al. [2017](#page-9-9)).

# **Morphological analysis using AFM**

An Atomic Force Microscope (AFM) was used for further morphological analysis. The AFM micrographs are shown in Fig. [3.](#page-4-0)

All the flms were found to have good adherence to the substrate. 5% Nd-doped flm shows spherical grains with uniform distribution. The grain size for the 5% doped flm was measured to be 8 nm. The pure and 3% Nd-doped flms exhibit a particle size of 10 and 15 nm, respectively (Nečas and Klapetek [2012\)](#page-8-23). The corresponding roughness of the





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 $\bf{0}$ 1 2 3  $\boldsymbol{4}$ 5 6  $\overline{I}$ 8 9 10

 $\overline{\mathbf{8}}$  $9$  10

<span id="page-4-0"></span>**Fig. 3** AFM images of  $\text{CoFe}_2\text{O}_4$  thin films **a** 0% Nd, **b** 3% Nd, **c** 5% Nd

mapping of undoped and 5% Nd-doped  $\text{CoFe}_2\text{O}_4$  films are shown in Fig. [5](#page-6-0).

#### **Optical studies**

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The graph of incident wavelength (*λ*) vs. optical transmittance (*T*) and absorption spectrum of  $\text{CoFe}_2\text{O}_4$  thin films prepared with changing Nd doping concentration is shown in Fig. [6a](#page-7-0), b.

It is seen that the transmittance of  $CoFe<sub>2</sub>O<sub>4</sub>$  thin film shows an increase with incident wavelength and also with the increases in the Nd doping. It presents a decrementing value in the absorbance of the samples with an increase in the doping concentration of Nd ion. It is also observed that the refectance of the ferrite flm increases when doped with neodymium. Incorporation of Nd increases the refectance of the deposited flm around the wavelength region of 1000 to 2000 nm. Further, the absorption edge shows a blue shift when Nd doping level was increased, which afects the improvement in the band structure.

Optical band gap  $(E_{\varrho})$  of the prepared films was determined from Tauc formula (Tiwari et al. [2020](#page-9-1))

$$
\alpha h \gamma = B(h\gamma - E_g)^n \tag{4}
$$

The band gap estimated from Tauc's plot is shown in Fig. [7.](#page-7-1)

The obtained band gap increases from 1.82 eV to 1.95 eV as the Nd doping concentration was changed from 0 to 5%. The enhancement in the values of bandgap may be attributed to the quantum confinement effect (Arulanantham et al. [2018\)](#page-8-24) and also due to the least values of grain size in the deposited polycrystalline flm. It might also be attributed to the Burstein—Moss efect which shifts the Fermi level into the conduction band and hence necessitates larger energy

<span id="page-5-0"></span>



for the transition from the valence band to conduction band resulting in an increased band gap.

# **Magnetic studies**

The magnetic property of undoped and 5% Nd-doped  $CoFe<sub>2</sub>O<sub>4</sub>$  thin films was studied by employing a vibrating sample magnetometer (VSM) at room temperature (300 K) under an applied magnetic field up to  $\pm$  1.5 T parallel as well as perpendicular to the flm surface. Magnetic hysteresis loops of undoped and 5% Nd-doped  $\text{CoFe}_2\text{O}_4$  thin flms recorded under the application of a parallel magnetic feld are shown in Fig. [8](#page-7-2).

<span id="page-6-0"></span>



<span id="page-7-0"></span>**Fig. 6 a** and **b** depicts the transmittance and absorption spectra of doped and undoped  $\text{CoFe}_2\text{O}_4$ 



<span id="page-7-1"></span>**Fig. 7** Band gap of doped and undoped  $\text{CoFe}_2\text{O}_4$  thin films



<span id="page-7-2"></span>**Fig. 8** The magnetic hysteresis loop of undoped and 5% Nd-doped  $CoFe<sub>2</sub>O<sub>4</sub>$  thin films

A typical hysteresis loop is observed for both pure and Nd-doped  $\text{CoFe}_2\text{O}_4$  thin films supporting the existence of room temperature ferromagnetism. The value of saturation magnetization for pure  $\text{CoFe}_2\text{O}_4$  thin films was observed to be 34.2 emu.gm<sup>-1</sup> and when doped with 5% Nd<sup>3+</sup>, the 40.8 emu.gm−1. The value of saturation magnetization was found to increase with the increase of  $Nd^{3+}$  substitution in cobalt ferrite. These values agree well with the earlier literature (Lee et al. [1998\)](#page-8-25). The Nd ions get substituted in the octahedral site of the  $\text{CoFe}_2\text{O}_4$  spinel structure. Neodymium substitution increases the structural stability of the cobalt ferrite flm due to which inhomogeneous spin structure gets suppressed and hence an increased value in the saturation magnetization (Yan et al. [2007\)](#page-9-10).

# **Conclusion**

In this work, cobalt ferrite thin flms were deposited on the glass substrates with diferent Nd doping concentrations. The deposited undoped and Nd-doped  $\text{CoFe}_2\text{O}_4$  films have excellent adherence and were found to have a good uniformity on the substrate surface. The XRD spectra indicate that the flms have a cubic crystal structure. The Raman analysis established peaks corresponding to octahedral as well as tetrahedral vibrations of the lattice. The presence of the rare earth ions as well as the main elements of the ferrite films were affirmed by EDX analysis. The SEM images show a decrease in the size of the nano spherical  $CoFe<sub>2</sub>O<sub>4</sub>$ grains with the increase in the doping concentration. The magnetic studies show an enhancement in the magnetic properties of  $CoFe<sub>2</sub>O<sub>4</sub>$  thin films with Nd doping. Hence this study reveals that 5% of Nd doping in  $\text{CoFe}_2\text{O}_4$  film enhances its structural, morphological, optical as well as magnetic properties and nebulizer assisted spray pyrolysis can be adopted as cheap technique for mass production of Nd-doped  $\text{CoFe}_2\text{O}_4$  thin films.

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### **Declarations**

**Conflict of interest** We declare no confict of interest among the authors.

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