ORIGINAL RESEARCH ARTICLE

Efect of Annealing Temperature on the Electrochromic Properties of Electrochemically Oxidized Nickel

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Abstract

Electrochromic (EC) flms are widely used in smart windows, automobile mirrors, and low-power displays due to their capability of reversibly changing color under an external potential. Transition metal oxides, due to their ability to switch between multiple oxidation states, are primarily used as EC flms. They require low operating power and provide high coloration efficiency in EC applications. However, their thin-film fabrication through a low-cost and scalable process is still challenging. Here, nickel oxide (NiO) EC flms have been synthesized by room-temperature electrochemical oxidation of Ni thin films. The Ni films were grown by electroless deposition and annealed at three different temperatures of 100° C, 200° C, and 300°C. The morphological, optoelectronic, and electrochemical properties of the flms have been comparatively analyzed to obtain an optimized EC flm. The NiO flm annealed at 100°C exhibited superior properties with coloration and bleaching times of 1.8 s and 2.9 s, respectively, and a coloration efficiency of 20 cm²/C. This annealed film is further used to fabricate a EC device with NiO and tungsten oxide (WO_3) as the electrode layers and a phosphoric acid-based gel as the electrolyte. The low-temperature deposition process used here can also be extended to fexible substrates.

Keywords Electroless deposition · electrochemical oxidation · nickel oxide · cyclic voltammetry · chronoamperometry

Introduction

The optoelectronic properties of transition metal oxides make them useful in a wide range of applications, such as photodetectors, gas sensors, displays, and solar cells. $1-5$ $1-5$ The fast-switching speed and coloration efficiency of these metal oxides facilitate their use in a particular type of display device called an electrochromic (EC) device. Electrochromism is a phenomenon in which a persistent and reversible change in the optical absorption of a material occurs

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due to the insertion and extraction of electrons and ions.^{[6](#page-8-2)} EC devices are of considerable commercial and technological interest since they are non-scattering, stable, and retain their properties when switched back and forth between the colored and bleached states. They are preferred for transparent glazing or see-through applications, smart windows, displays, and rear view mirrors.^{7[,8](#page-8-4)} EC windows can be easily powered because of their low voltage requirement, which is helpful in energy conservation by reducing cooling/heat-ing loads and by improving indoor comfort.^{[9](#page-8-5)} EC oxides are primarily of two types, cathodic coloring oxides (Ti-, Nb-, Mo-, and W-based oxides) which color upon ion insertion, and anodic oxides (Cr-, Mn-, Fe-, Ni-, and Co-based oxides) which color upon extraction of ions. WO_3 , NiO, MoO₃, and V_2O_5 are the most commonly used EC metal oxides, of which NiO and WO_3 are widely studied.^{[10,](#page-8-6)[11](#page-8-7)}

NiO has unique electrochemical and optoelectronic properties which make it popular in the EC industry. It possess high cyclic stability, high electrical conductivity, fast response time, well-defned and stable redox kinetics, and an efficient charge transfer mechanism.¹² Various deposition methods have been investigated in the preparation of

NiO-based EC films, such as sputter coating, $13-15$ $13-15$ $13-15$ electron beam evaporation,¹⁶ thermal evaporation, 17 templateassisted hydrolysis, $18 \text{ sol–gel deposition}$ $18 \text{ sol–gel deposition}$, $19 \text{ aerosol-assisted}$ chemical vapor deposition,^{[9](#page-8-5)} and hydrothermal synthesis.²⁰ Studies have been carried out by varying the deposition method and thickness to obtain the maximum coloration efficiency and fast response time. It has also been shown that a porous morphology in the NiO flm enhances the EC performance by facilitating the fast intercalation/deintercalation of ions. $9,21,22$ $9,21,22$ $9,21,22$ $9,21,22$ $9,21,22$ Most of the current deposition techniques require high annealing or deposition temperatures. An alternate route is electroless nickel (Ni) deposition, which is an auto-catalytic process and can be carried out below 100° C.²³ However, to oxidize the deposited Ni, annealing above 500° C is usually required.²⁴ To avoid hightemperature annealing, room-temperature electrochemical oxidation can be carried out to convert Ni into NiO. This also converts the granular structure of the metallic flm to a porous morphology which enhances the EC properties of the film.^{[25](#page-8-20)}

In this work, the electroless Ni deposition technique has been used to deposit Ni thin flms, followed by electrochemical oxidation to produce NiO. The as-deposited Ni flms are amorphous in nature 26 26 26 and we investigated the effect of annealing temperature of the Ni flm on the EC properties. Uniform Ni deposition was carried out on fuorine-doped tin oxide (FTO) substrates and the flm was annealed at 100°C, 200°C, and 300°C, followed by room-temperature electrochemical oxidation to convert the Ni to NiO. The surface, optical, and EC properties of the formed NiO flms have been extensively characterized and, based on our results, the flm annealed at 100°C was found to possess superior properties, including fast response time and high coloration efficiency. A prototype solid-state EC device was also fabricated using tungsten oxide (WO_3) and NiO as the electrodes and a phosphoric acid-based gel as the electrolyte. Given the low annealing temperature, the process developed here is also highly suitable for use with fexible substrates.

Materials and Methods

Materials

Nickel sulfate $(NiO₄.6H₂O)$ and potassium hydroxide (KOH) were purchased from Merck Life Sciences (India), sodium citrate $(C_6H_5Na_3O_7.2H_2O)$ and ammonium sulfate $((NH_4)_2SO_4)$ were obtained from Avantor Performance Materials India, sodium hypophosphite monohydrate $(H₂NaO₂P.H₂O)$ was purchased from Spectrochem (India), and polyvinyl alcohol (PVA; molecular weight 195,000) and orthophosphoric acid $(H_3PO_4; 85\%$ purity) were procured from Merck and Labchem, respectively. The solvents and chemicals were used without further purifcation. FTOcoated glass slides (sheet resistance $<$ 15 Ω /sq.) were purchased from Technistro India.

Preparation of NiO EC Film

The method followed in this work is adapted from a previous work in the group²⁵ and is described briefly below. The FTO glass slides were cleaned by sonication in acetone, ethanol, isopropyl alcohol, and de-ionized water for 7 min each using a Branson CPX2800H bath sonicator. The slides were then masked at the edges using a Kapton polyimide tape to create a 1 cm \times 1 cm window at the center for Ni deposition. The glass surface was then activated by Au-Pd sputter-coating using a Polaron Series SC7640 high-resolution sputter coater (Quorum Technologies) operating at 1-kV DC output and 25-mA process current for 60 s. The chemical bath for electroless Ni deposition was prepared by adding 6 g each of nickel sulfate, ammonium sulfate, and sodium citrate, and 2.73 g of sodium hypophosphite in 200 ml of de-ionized water. The solution was stirred at room temperature using an IKA C-MAG HS-7 stirrer until the chemicals were completely dissolved. 26 The pH of the bath was 5.6. The bath was then heated to 60°C in an IKA C-MAG HS-7 digital hot plate with an attached thermocouple and electroless deposition was carried out for 30 s. The samples were further annealed in an oven at 100°C for 60 min to evaporate the solvents and moisture, and the obtained flm was labeled EN-100. Other samples were further annealed at 200°C and 300°C for 60 min and denoted as EN-200 and EN-300, respectively. These Ni flms were then electrochemically oxidized at room temperature to convert them into NiO.^{[25](#page-8-20)}

Fabrication of NiO‑Based EC Device

Of the flms annealed at diferent temperatures, the EN-100 flm was used to fabricate an EC device by incorporating a 150-nm-thick WO_3 film (2.5 cm \times 2 cm) as the anode. The $WO₃ film was DC magnetron-sputtered at room temperature$ on an FTO substrate from a pure tungsten target at 100-W power in an argon and oxygen atmosphere, with fow rates of 20 and 25 standard cubic centimeters per minute, respectively. The fabrication and electrochromic properties of this $WO₃$ film were detailed in our previous work.²⁷ A PVA/ H_3PO_4 polymer gel was used as the electrolyte for this solidstate EC device. First, a PVA gel was prepared by mixing 1 g of PVA in 10 mL distilled water using a REMI 2MLH stirrer at 80 \pm 2°C for 2 h. H₃PO₄ was then added to this viscous solution to obtain the electrolyte solution of 1.86 M concentration, which was next coated over both the NiO and $WO₃$ films via spin coating. These electrolyte-coated films were then sandwiched and heated at 60°C for 30 min. The electrical connections were made using copper tape.

Characterization

The surface characterization of the as-deposited and oxidized samples was carried out using a Apreo S fieldemission scanning electron microscope (FESEM) with an energy dispersive spectroscopy (EDS) setup. The thickness of the films were measured using a Nanomap 1000WLI (AEP Technology) optical profilometer. The optical transmittance spectra were obtained using a JASCO V-660 UV–Vis spectrophotometer. Cyclic voltammetry (CV) and chronoamperometry (CA) studies of the samples were carried out using a Biologic SP-150 and a Metrohm Autolab potentiostat, respectively. A threeelectrode system in 1-M KOH solution with the Ni film on FTO as the working electrode, Pt mesh as the counter electrode, and Ag/AgCl in 3-M KCl electrolyte as the reference electrode was run for 420 cycles in the potential window -0.2 V to 1 V at a scan rate of 20 mV/s to convert the Ni film to NiO film. For the CV measurements, a linear potential sweep was applied from − 0.2 V to 1 V for 10 cycles to observe the cyclic stability of the film. CA was performed at a potential of -0.2 V and 1 V to study the switching behavior of the film from the colored to the bleached state and vice versa.

Results and Discussion

Morphological Studies

Figure [1a](#page-2-0), b, and c shows SEM images of the Ni flms annealed at diferent temperatures, while Fig. [1d](#page-2-0), e, and f shows the same flms after electrochemical oxidation. It can be seen that there are no visible changes in the microstructure of the flms with the increasing annealing temperature, except for a slight increase in the average grain size. During electroless deposition, the Ni grains are deposited on the substrate in an island growth mode²³ and, as more material is deposited, these islands grow and meet to form a continuous layer on the substrate. For the deposition conditions used in this work, a flm of thickness approximately 47 nm (as measured using the optical profler) was obtained. The thickness was optimized based on our previous work on EN films. 25 Thinner films did not show a stable and repeatable EC behavior while thicker flms had a lower optical transmittance, both in the colored and bleached states. From Fig. [1](#page-2-0), it can be seen that the granular morphology of the Ni flm is changed to a porous nanofake morphology upon electrochemical oxidation. A similar morphology change was also observed for the electroless Ni flms, without any post-annealing treatment. 25 The morphology change occurs because, during the oxidation and reduction cycles, OH− ions are, respectively, inserted and extracted from the Ni film.²⁸

Fig. 1 Scanning electron microscope images of the as-deposited and oxidized flms: (a, b, and c) the deposited EN-100, EN-200, and EN-300, respectively; (d, e, and f) the electrochemically oxidized EN-100, EN-200, and EN-300, respectively. With electrochemical

oxidation, the granular morphology transforms into a porous nanofake morphology which helps in the insertion/extraction of the ions and improves the EC flm. The flm annealed at 100°C has the smallest nanofake size and hence shows the best EC properties.

The rough morphology of the oxidized samples enhances the EC properties of the flm, since the increased surface area improves the ion insertion kinetics in the film.^{29,30} It can be observed qualitatively from the SEM images that the size of the nanofakes increases with the annealing temperature. The smaller fakes in EN-100 make it more porous and hence enhance the EC behavior due to the enhanced surface area compared to the flms annealed at higher temperatures.

Cyclic Voltammetry Analysis

The Ni flms were subjected to electrochemical oxidation to convert them into NiO which is optically transparent. CV was performed on the NiO flms to study their electrochemical properties. Figure [2](#page-3-0) compares ten CV cycles for the EN100, EN-200, and EN-300 flms.

The EC behavior of NiO is attributed to the transition between the Ni^{2+} and Ni^{3+} oxidation states.³¹ Coloration occurs when the transparent NiO and Ni(OH)− are oxidized to NiOOH and bleaching occurs when the NiOOH is reduced. In the anodic scan, the intercalation of OH− ions into the transparent NiO film results in the oxidation of Ni^{2+} to Ni^{3+} , which changes the color of the flm to dark brown. During the reverse cathodic scan, the de-intercalation of OH− ions from the colored NiOOH film results in the reduction of $Ni³⁺$ back into Ni^{2+} , which bleaches the film back to the transparent state. The whole process can be summarized using: $32-35$:

$$
NiO + H_2O \rightleftharpoons Ni(OH)_2
$$
 (1)

$$
NiO + x(OH^{-}) \rightleftharpoons NiOOH + xe^{-}
$$
 (2)

$$
Ni(OH)2 + x(OH-) \Rightarrow NiOOH + H2O + xe-
$$
 (3)

All three samples (EN-100, EN-200, and EN-300) were cycled ten times to observe the cyclic stability and the data are shown in Fig. 2 . The diffusion coefficient of the OH− ions within the flm has been calculated from the Randles–Sevcik relationship $36-38$ $36-38$:

$$
i_p = 2.75 \times 10^5 \times n^{2/3} \times D^{1/2} \times C_0 \times v^{1/2}
$$
 (4)

where I_p is the anodic peak current density, *n* is the number of electrons, *D* is the diffusion coefficient, C_0 is the number of active ions in the electrolyte, and ν is the scan rate. The calculated diffusion coefficient for the samples are 8.6×10^{-12} cm²/s, 8.7×10^{-12} cm²/s, and 8.6×10^{-12} cm²/s for the samples EN-100, EN-200, and EN-300, respectively. These values are consistent with values reported in the litera-ture.^{[36](#page-8-29)[,39](#page-9-1)} Typically, the higher the value of D , the faster the EC reaction, since movement of ions is easier. The comparable values of *D* for all three flms indicate that the movement of ions within the NiO is not afected by the annealing conditions prior to oxidation. However, when comparing the CV curves from Fig. [2](#page-3-0), it can be seen that the curve for EN-300 is shifted to slightly higher voltages compared to EN-100 and EN-200. This leads to slightly longer switching times. Similarly, for EN-200, the cyclic stability is lower since the curve for the tenth cycle is shifted by approximately 5% to the left compared to the frst cycle. This sort of shift is not observed for EN-100 and EN-300.

Chronoamperometry Studies

For an EC device, switching (response) time is an important factor to determine its efficiency. Switching time is the time taken by the device to reach 90% of the complete response. To obtain the switching time between the colored and bleached states of the NiO flms, CA studies were carried out. A potential of 1 V (for coloration) and − 0.2 V (for bleaching) were applied for 30 s each for a total time of 300 s, and the resulting graphs are shown in Fig. [3](#page-4-0) for the flms annealed at the three diferent temperatures.

Fig. 2 Ten cyclic voltammograms recorded in the 1-M KOH electrolyte at a scan rate of 20 mV/s for (a) EN-100, (b) EN-200, and (c) EN-300; the data clearly establish the cyclic stability of the flms.

The response times, t_c (coloration time) and t_b (bleaching time), for each flm were calculated from the CA curves and shown in Table [I.](#page-4-1) The EN-100 flm has the shortest response time for coloration compared to the other two samples. This is because a higher annealing temperature makes the flm more dense, decreasing the rate of ion intercalation and de-intercalation in the film. 36 The coloration and bleaching times for EN-100 and EN-200 are constant throughout the cycles, giving a sharp switch between the states, whereas it was found to vary in the case of EN-300.

Electrochromic Properties

The as-deposited Ni flms appear gray in color and, upon electrochemical oxidation, turn transparent due to the formation of NiO. During voltage cycling, the transparent flms turn dark brown on oxidation and back to transparent/ bleached state on reduction (Fig. [4](#page-5-0)). UV-Vis transmittance spectra were obtained for the as-deposited, colored, and bleached states for all three samples, i.e., annealed at the diferent temperatures.

Coloration and bleaching were carried out at 1 V and − 0.2 V, respectively, for a duration of 30 s. Figure [5](#page-5-1) shows the transmittance data of the flms and the values measured at the middle of the visible spectrum, i.e., 550 nm. EN-100 shows a maximum transmittance of 96.4% in the bleached state and 21.2% in the colored state. This diference (between the colored and bleached states) is defned as the transmittance modulation (ΔT) and is given by:

$$
\Delta T = \left(T_b - T_c\right)_{\lambda} \tag{5}
$$

where T_b and T_c are the transmittance in the bleached and colored state, respectively, and λ is the wavelength at which the modulation is measured (i.e., 550 nm). the values being set out in Table [I](#page-4-1). The transmittance modulation was found to decrease with the increase in annealing temperature, with EN-100 having the highest transmittance modulation of 75.2% and EN-300 with the lowest at 55.6%. Another related parameter to measure the change from the colored to the bleached state is the change in the optical density $(\Delta OD)^{35,44}$ $(\Delta OD)^{35,44}$ $(\Delta OD)^{35,44}$:

$$
\Delta OD = \ln(T_b/T_c)_{\lambda} \tag{6}
$$

The ∆OD values are set out in Table [II](#page-5-2), and the highest value is for the EN-100 flm. The optical density value is used to calculate the coloration efficiency (η) . This is defined

Fig. 3 Chronoamperometry curves of the oxidized NiO flms: (a) EN-100, (b) EN-200, and (c) EN-300.

Fig. 5 UV-Vis transmittance spectra showing the transmittance in the as-deposited, colored, and bleached states for (a) EN-100, (b) EN-200, and (c) EN-300; the *vertical dotted line* is drawn at 550 nm

as the ratio of the change in optical density at a specifc wavelength to the injected/ejected electronic charge (Q_1) , and given by $33,45$ $33,45$:

$$
\eta = \left(\frac{OD}{Q_{\rm i}}\right)_{\lambda = 550 \,\text{nm}} = \frac{\ln(T_{\rm b}/T_{\rm c})}{Q_{\rm i}}\tag{7}
$$

where Q_I is calculated using:

parameters of the EN films

$$
Q_{\mathbf{i}} = \int_{t_1}^{t_2} \mathbf{j} \, \mathrm{d}t \tag{8}
$$

where *j* is the current density, t_1 and t_2 are the starting and ending times, respectively, of the coloration or bleaching

to obtain the transmittance values for calculation of the EC properties. The as-deposited transmittance curve is the same for all the samples since this represents the starting Ni flm.

process. The current density, *j*, and the coloration and bleaching times, t_1 and t_2 , are calculated from the CA curves along with the cathodic and anodic charge densities $(Q_c$ and *Q*a) which are obtained by calculating the area under the CA curves. The calculated values for the flms annealed at diferent temperatures are compiled in Tables [I](#page-4-1) and [II](#page-5-2).

From the tables, it can be seen that EN-100 gives the maximum coloration efficiency of 20.2 cm^2/C and that the efficiency decreases with the increasing film annealing temperature. This decrease can be attributed to the decrease in the optical density and, at the same time, the increased charge density requirement for flms annealed at higher temperatures.

The data obtained in this work can be compared to the literature values shown in Table [I](#page-4-1), including our previous work on as-deposited EN films. 25 From both the tables, it can be seen that EN-100 has a better performance in terms of response time, transmittance modulation, and efficiency. The porous morphology of the flm due to electrochemical oxidation provides a larger surface area, which helps in faster ion difusion compared with the other deposition techniques.

This helps the flm to have a faster switching between the colored and bleached states compared to flms prepared by other techniques.

Electrochromic Solid‑State Display Device

The EN-100 flm, due to its optimum performance, was further used as an anode to fabricate a solid-state EC display. A display of 2.5 cm \times 2 cm size was fabricated by sandwiching $PVA/H₃PO₄$ polymer gel electrolyte between $WO₃$ and NiO electrodes. The cyclic voltammetry curve for the device is shown in Fig. $6a$ $6a$. WO₃ shows cathodic coloring behavior, which means that it reversibly changes from colorless to blue color. This color change is due to electron insertion, which changes the oxidation state of tungsten from W^{6+} to W^{5+} ²⁷. The involved redox equations for the EC behavior are:

(9) Cathodic : $WO_3(bleached) + x(H^+ + e^-) \rightleftharpoons H_xWO_3(colored)$

(10) Anodic : H_xNiO(colored) + y(H⁺ + e⁻) \Rightarrow H_{x-y}NiO(bleached)

The response time and the efficiency of the device were evaluated using CA curves, which were obtained by applying potentials of -3 V (coloration) and 1 V (bleaching) for 30 s each until 300 s (Fig. [6](#page-6-0)b). The bleaching and coloration times were \sim 4.2 and \sim 3.9 s, respectively. The longer switching time for the device compared to the EN-100 flms can be attributed to the frustrated difusion in the solid gel electrolyte compared to difusion in the liquid electrolyte. The transition between the deep blue-colored and transparent states at − 3 V and 1 V, respectively, of the EC device are shown in Fig. [7a](#page-7-0). These states represent the EC behavior of the solid-state device.

The coloring and bleaching of the device under negative and positive voltages, respectively, can be attributed to the higher complementary charge capacity ratio, which depends on the thickness of the EC films.⁴⁶ The complementary charge capacity ratio is the ratio of the surface charge capacity of WO_3 during the insertion of electrons to the surface charge capacity of NiO during the extraction. Hence, the coloration of WO_3 is more prominent due to its greater thickness, while high transparency is maintained during

Fig. 6 (a) Cyclic voltammograms of the device recorded in 1.18 M PVA/H_3PO_4 polymer gel electrolyte at a scan rate of 20 mV/s, showing anodic and cathodic peaks at 0.3 and -0.16 V, respectively; (b)

CA curves of the device subjected to potentials of -3 and 1 V; (c) bleaching and (d) coloration times of the device.

Fig. 7 (a) Images and (b) transmittance spectra of the EC device in bleached (1 V) and colored (− 3 V) states; the deep blue color represents the oxidation of WO_3 at a negative applied voltage.

Table III Comparison of various NiO/WO₃ based EC devices from the literature with the present work

$t_c(s)$	$t_{\rm h}$ (s)	$\Delta T (\%)$	η (cm ² /C)	Ref.
3.1	4.6	46	90	46
10	20	55	87	47
7.9	2.8		109.6	48
10	13	75.4	131.9	49
10	11.7	84	55.2	50
3.9	4.2	69	47.3	This work

bleaching due to the lower thickness of the NiO flm. Figure [7b](#page-7-0) shows the UV-visible transmittance of the colored and bleached states of the fabricated device. The transmittance of the colored (T_c) and bleached states (T_b) of the device at 550 nm are 10.9% and 79.9%, respectively, from which ΔT was obtained at 69.0% and the optical density was 0.87 at 550 nm. From the CA curve of the device, Q_c is 18.3 mC/ cm². From Eq. [\(7\),](#page-5-3) the efficiency of the device is calculated to be $47.3 \text{ cm}^2/\text{C}$. The properties of the fabricated device were comparable to other devices reported in the literature, as seen in Table [III.](#page-7-1)

This EC device can be successfully applied in various applications, such as smart windows, color displays, and sensors. Its applicability can also be extended to smart wearable and fexible electronic devices by utilizing fexible substrates for the deposition of the WO_3 and NiO films.

Conclusions

Uniform NiO flms were prepared on FTO substrates using electroless nickel deposition, with diferent annealing temperatures, followed by electrochemical oxidation in a 1-M

KOH bath. Surface morphology studies revealed that the granular morphology of the as-deposited flm was changed into a porous structure on electrochemical oxidation. The flm annealed at 100°C was found to give superior EC and optical properties with a high transmittance modulation of 75.2%. It also exhibited the fastest response times with 1.7 s and 2.8 s as the coloration and bleaching times, respectively. The efficiency of the EN-100 film was calculated to be 20.2 cm^2/C . The fabrication of an electrochromic device was demonstrated using an EN-100 NiO flm as the counter electrode. The good performance of the NiO flm annealed at low temperature provides a promising route to use this technique to develop fexible EC devices.

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Author Contributions M.M.S. designed the experiments, deposited and characterized the NiO flms, and drafted the manuscript. N.S. assisted in writing of the manuscript. M.M.S., N.S., and A.M.K. fabricated and characterized the EC device. F.A helped in the deposition and characterisation of the films. A.M.K and S.A. deposited the WO_3 films. P.S. supervised and coordinated this study and revised the manuscript.

Data Availability The authors confrm that the data supporting the fndings of this study are available within the article.

Conflict of interest The authors declare that they have no confict of interest.

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