

Research on a High‑Threshold‑Voltage AlGaN/GaN HEMT with P‑GaN Cap and Recessed Gate in Combination with Graded AlGaN Barrier Layer

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Received: 13 August 2023 / Accepted: 24 January 2024 / Published online: 5 March 2024 © The Minerals, Metals & Materials Society 2024

Abstract

Due to the enhanced-mode (E-mode) operation, AlGaN/GaN high-electron-mobility transistors (HEMTs) are considered to be safer for circuit operation. In order to improve the threshold voltage (V_{th}) of the device, this work provides a hybrid gate structure HEMT by embedding a P-GaN cap on the etched graded AlGaN barrier layer. Through simulation calculations, the P-GaN cap (thickness of P-GaN = 50 nm, concentration of *P*-type= 2×10^{18} cm⁻³) and the aluminum (Al) composition (Al:0.3 \rightarrow 0.24), in the graded AlGaN barrier layer were optimized. Although simulation calculations show that the optimized P-GaN layer can significantly increase the device's V_{th} to 8.6 V and transconductance (g_m) to 94.7 mS/mm, the device exhibits a lower saturation current (I_{sat}) . Therefore, to improve the output characteristics of the devices, the addition of an N-well in the GaN channel layer of such structures was proposed. It can increase the device's source–drain current while maintaining a steady V_{th}. Compared with the HEMT structure/combined P-GaN cap with recessed gate and a graded AlGaN barrier layer, the device with the added N-well exhibits a signifcant improvement of 11.2% in the saturation current $(I_{sat} = 718$ mA/mm). The results demonstrate that HEMT structures combining recessed gates and P-GaN with N-well have promising applications in next-generation high-power devices.

Keywords E-mode · AlGaN/GaN HEMT · P-GaN cap · graded AlGaN barrier layer · N-well

Introduction

Gallium nitride (GaN) has emerged as an excellent candidate for high-power and high-temperature applications due to its wide bandgap. It also possesses high breakdown feld strength and high electron saturation velocity.^{[1](#page-9-0)[–3](#page-9-1)} These properties make GaN-based high-electron-mobility transistors (HEMTs) potential candidates for next-generation power electronic devices in applications such as power supplies, photovoltaic inverters, and motor drives.^{[4](#page-9-2)} HEMT devices formed by AlGaN/GaN heterostructures at the two-dimensional electron gas (2DEG) are naturally in the conducting state, which increases power losses and poses challenges to circuit safety.^{[5](#page-9-3)} However, enhancement-mode HEMTs^{[6](#page-9-4)} have been developed to address these issues. They not only reduce the complexity and cost of circuit design in applications such as radio frequency (RF) and low-noise power amplifers, but also signifcantly improve circuit safety and system reliability in high-power switching applications. Moreover, enhancement-mode HEMTs can alleviate the P-channel defciency in nitride semiconductors. As a result, a growing number of researchers are exploring enhancement-mode HEMTs. Currently, there are several methods to achieve device enhancement mode: recessed gate structures,^{[7](#page-9-5)} tech-nology of P-GaN cap,^{[8](#page-9-6)} F⁻ implantation technology,^{[9](#page-9-7)} and polarization engineering techniques.¹⁰ However, the temperature reliability of devices using fuorine implantation is not suitable for practical applications. $11,12$ Polarization engineering techniques have extreme process complexity and cost, making them impractical. The recessed gate technique, while efective, results in a thinner barrier layer due to etching of the AlGaN barrier, causing signifcant gate leakage current and afecting gate swing and device reliability. On the other hand, P-GaN gate technology employs a P-GaN layer with an activated hole concentration that efectively depletes the

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2DEG at the AlGaN/GaN interface, rendering the gate nonconductive under zero bias voltage.¹³ Su et al.¹⁴ successfully achieved enhancement-mode HEMT devices by employing heavily doped P-GaN cap layers, resulting in a threshold voltage as high as 4.3 V and a maximum drain–source current of 208 mA/mm. However, the elevated doping concentration of Mg in the P-GaN layer introduces defects, $15-21$ $15-21$ $15-21$ leading to a decrease in device performance. Moreover, many related studies have been reported focusing on individual device parameters, and achieving high threshold voltages and drain–source currents simultaneously still poses signifcant challenges. Therefore, in this paper, we propose a hybrid gate structure combining P-GaN with a recessed gate, and determine the optimal P-GaN cap layer thickness and concentration, leading to a signifcant improvement in the V_{th} of the HEMT device. In addition, the Al component in the graded AlGaN barrier layer is optimized to reduce the remote scattering effect^{[22](#page-9-15)} and improve the electron mobility in the channel. Finally, an N-well is added to the GaN channel layer, efectively increasing the source–drain current of the device. The structure with a P-GaN gate grown on an etched AlGaN barrier layer and an N-well inserted in the GaN channel was realized using Silvaco TCAD (technology computer-aided design) simulation software. This not only significantly increases the V_{th} and g_m of the device but also enhances the maximum drain–source (saturation) current.

Device Structure

Figure [1](#page-1-0)a, b, and c illustrate the architecture of structure A, consisting of a conventional MIS-recessed gate HEMT; structure B, the recessed gate HEMT covered with a P-GaN layer $(Al_{0.3}Ga_{0.7}N/GaN)$; and structure C, the P-GaN cap and recessed gate HEMT with a graded AlGaN layer $(Al_{0.30\rightarrow0.24}Ga_{0.70\rightarrow0.76}N/GaN)$. The mole fraction of Al in the AlGaN barrier layer of structure A is 0.24. The mole fraction of Al in the AlGaN barrier layer of structures B and C is 0.3, and the P-GaN cap layer

Fig. 1 The structure cross sections of the three devices: (a) structure A: a conventional MIS-recessed gate HEMT, (b) structure B: the recessed gate HEMT covered with a P-GaN layer HEMT $(Al_{0.3}Ga_{0.7}N/GaN)$, and (c) structure C: the P-GaN cap and recessed gate HEMT with the graded AlGaN layer HEMT structure $(Al_{0.30\rightarrow0.24}Ga_{0.70\rightarrow0.76}N/GaN).$

is doped with a *P*-type concentration of 1×10^{18} cm⁻³. Among them, the GaN buffer layer is unintentionally doped. Table [I](#page-2-0) summarizes the detailed information of the three main structural parameters (a) , (b) , and (c) .

Simulation Model

In this study, we employed the Silvaco ATLAS TCAD tool for 2D simulations. 23 23 23 The numerical solution was performed for the Poisson equation, electron continuity equation based on the drift–diffusion (DD) model. 24 The transport equations were solved using the drift–diffusion model, 25 and all calculations in the simulations were based on Fermi–Dirac statistics. Additionally, the Shockley–Read–Hall (SRH) model was utilized to simulate the impact of traps in the bandgap on carrier recombination, 25 25 25 and a polarization model was employed to simulate the polarization charges along the heterojunction interface. Considering that the electron mobility in the GaN channel varies with the electric field strength, we considered the effect of the electric field on the electron mobility of the nitride in our simulations and chose a specific fielddependent mobility model. We used the Silvaco ATLAS simulation software to simulate the transfer characteristics of the conventional AlGaN/GaN HEMT and compared the results with previously reported experimental $data.²⁶$ As shown in Fig. [2,](#page-2-1) the results demonstrate excellent agreement between the simulation and experimental data, indicating that the physical parameters chosen in our simulations are consistent with the actual conditions.

Results and Discussion

The polarization model is crucial for the simulation of GaNbased HEMT devices. In Fig. [3,](#page-3-0) the polarization charge distribution at the AlGaN/GaN heterojunction interface is illustrated for structure B and structure C. The spontaneous polarization arises from the asymmetry along the interface of the wurtzite crystal, while the piezoelectric polarization results from the stress and strain caused by the lattice mismatch between the GaN and AlGaN layers. In the absence of an external electric feld, the total polarization intensity

Fig. 2 $I_d - V_g$ characteristic comparison of conventional AlGaN/ GaN HEMT^{[26](#page-9-19)} with TCAD simulation for validation at $V_{ds} = 2$ V, $V_g = -4 \sim 1$ V.

Fig. 3 Polarization-induced charge profle along the vertical axis.

 (P_{To}) is obtained by adding the spontaneous polarization strength (P_{SP}) and the piezoelectric polarization strength (P_{PF}) . The 2DEG at the interface of the AlGaN/GaN HEMT heterojunction can be described by the polarization charge formula, Eq. [1](#page-3-1):

$$
\sigma = P_{\text{spont (AIGaN)}}(x) - P_{\text{spont (GaN)}}(0) + P_{\text{pie (AIGaN)}}(x) \tag{1}
$$

Here, the spontaneous polarization strength of the GaN layer is $P_{\text{spont(GaN)}}$ =−0.034, and the spontaneous polarization of the AlGaN layer is $P_{\text{spont}(A|GaN)}$ =−0.052*x*−0.034.

The piezoelectric polarization intensity acting on the AlGaN barrier layer due to the lattice mismatch in the AlGaN/GaN heterojunction can be expressed by Eq. [2](#page-3-2):

$$
P_{\text{pie}}(\text{AlGaN}) = 2\frac{a_s - a_0}{a_0} \left(E_{31} - \frac{C_{13}}{C_{33}} E_{33} \right) \tag{2}
$$

Here, *x* represents the mole fraction of Al in the AlGaN layer, E_{31} and E_{33} are piezoelectric constants, C_{13} and C_{33} are elastic constants, a_0 and a_s are the lattice constant in the strained layer and the substrate lattice constant, respectively.

$$
V_{\text{th}} = \varphi_B - \Delta E_c - \frac{qN_{D_{\text{AlGaN}}}d_{\text{AlGaN}}^2}{2\varepsilon_0 \varepsilon_{\text{AlGaN}}} - \frac{\sigma}{\varepsilon_{\text{AlGaN}}}d_{\text{AlGaN}} \tag{3}
$$

In Eq. [3,](#page-3-3) φ_B is the Schottky barrier height of the gate metal, ΔE_c is the conduction band offset at the AlGaN/GaN interface, q is the electron charge, σ is the net polarization charge density at the interface, *d* is the thickness of the AlGaN barrier, and ϵ_0 and ϵ_{AlGaN} are the vacuum permittivity and AlGaN permittivity, respectively.

Fig. 4 Schematic of structure B.

Fig. 5 Transfer characteristic curves of structure B with diferent thicknesses of P-GaN cap.

DC characteristics of the recessed gate HEMT covered with a P‑GaN layer HEMT

We investigated the effect of the thickness of P-GaN on the transfer characteristics of the device when the P-GaN cap was doped with 2×10^{18} cm⁻³ *P*-type concentration. Figure [4](#page-3-4) shows a schematic of structure B.

As shown in Figs. 5 and 6 , we consider that when the etching thickness of the AlGaN barrier layer is fxed to 10 nm, the thickness of P-GaN is increased to 20 nm, 40 nm, 60 nm, 80 nm, and 100 nm, respectively. With an increase in the thickness of the P-GaN cap, the distance between the gate and the GaN channel increases, and so does the V_{th} . However, the transconductance decreases signifcantly. This is because the increased P-GaN thickness weakens the gate control over the channel. Figure [7](#page-4-1) presents the relationship between the threshold voltage and transconductance for

Fig. 6 Transconductance curves of structure B with diferent P-GaN caps.

Fig. 7 Relationship between threshold voltage and transconductance for diferent thicknesses of GaN cap.

diferent thicknesses of the GaN cap layer. From Fig. [7,](#page-4-1) we can conclude that a P-GaN thickness of 50 nm is optimal.

After determining a P-GaN cap thickness of 50 nm, we further discuss the efect of *P*-type concentrations on the device. Figure [8](#page-4-2) illustrates the transfer characteristics for diferent *P*-type concentrations in the P-GaN cap layer. For *P*-type concentrations below 2×10^{18} cm⁻³, although the threshold voltage increases with the doping concentration, the increase in V_{th} is not significant. However, when the *P*-type concentration is 2×10^{18} cm⁻³ and 3×10^{18} cm⁻³, the threshold voltage increases signifcantly to 8 V and 12.8 V, respectively. This is because the hole concentration can deplete the 2DEG under the gate, leading to an increase in the conduction band and the threshold voltage. 27 When

Fig. 8 Transfer characteristics of P-GaN cap with diferent P-doping concentrations.

Fig. 9 Output characteristics of P-GaN cap with diferent P-doping concentrations.

the *P*-type concentration is 1×10^{18} cm⁻³, the 2DEG in the channel is entirely depleted. Thus, at hole concentrations of 2×10^{18} cm⁻³ and 3×10^{18} cm⁻³, P-GaN increases the energy gap between the conduction band and the Fermi level of the AlGaN/GaN heterojunction. As a result, the gate bias voltage needs to provide additional voltage to reduce the energy gap between the conduction band and the Fermi level, resulting in a signifcant increase in the threshold voltage. Figure [9](#page-4-3) shows the output characteristics for diferent *P*-type concentrations in the P-GaN cap. Although the device exhibits the highest threshold voltage at a hole concentration of 3×10^{18} cm⁻³, the source–drain current is practically zero. This is because the high concentration of holes difuses into

Fig. 10 Schematic of structure C.

Fig. 11 Transfer characteristics and transconductance curves of structure A.

the GaN channel layer, leading to the recombination of the 2DEG with the holes, resulting in only a few electrons participating in conduction in the channel under the on state, and consequently leading to a very low source-drain current. Additionally, the difusion of holes into the GaN channel, combined with the lower hole mobility compared with electron mobility, also contributes to the low source–drain current. To strike a balance between the threshold voltage and the source–drain current, we chose a hole concentration of 2×10^{18} cm⁻³ in this study.

To examine the performance of structure C, we propose three diferent structures: structure A, structure B, and structure C. Figure [10](#page-5-0) shows a schematic of structure C. Figures [11](#page-5-1) and [12](#page-6-0)a and b depict the transfer characteristics and transconductance (g_m) curves of the three structures. For structure A, the V_{th} and g_m can reach 0.5 V and 339 mS/mm, respectively. In the case of structure A, when the thickness of the AlGaN barrier layer beneath the gate is etched to 3 nm, the device has a V_{th} of only 0.5 V, indicating the limitation in etching the AlGaN barrier layer to improve the V_{th} . For structure B, the holes in the P-GaN cap can efectively deplete the electrons in the GaN channel layer, signifcantly increasing the V_{th} to 8.0 V. However, the activation energy of Mg in the P-GaN layer is mostly high (0.15–0.21 eV), and the compensation efect of H in P-GaN prevents the activation concentration of P-GaN from reaching high levels. This is also a critical challenge faced by GaN technology. Therefore, according to Eq. [3](#page-3-3), we can reduce the 2DEG beneath the recessed gate caused by polarization charges by reducing the lattice mismatch between the AlGaN barrier layer and the GaN channel layer. The V_{th} of structure C is 8.6 V, which is 0.6 V higher than that of structure B. The transconductance (95.7 mS/mm) of structure C is higher than that of structure B (90.4 mS/mm). This is attributed to the gradual change in the Al composition in the AlGaN barrier layer from 0.3 at the P-GaN/AlGaN interface to 0.24 at the AlGaN/GaN interface. The lower Al composition at the AlGaN/GaN interface reduces the scattering probability for electrons in the GaN channel and thus increases the electron mobility in the GaN channel. Deng et al. reported that increasing the threshold voltage of the device would decrease the density or mobility of 2DEG in the GaN channel, resulting in a decrease in the maximum saturation current output.[28](#page-10-1) However, according to Fig. [12](#page-6-0)a and b, for the transfer and output characteristics of structure C and structure B, not only does the threshold voltage of structure C increase by 0.5 V, but the saturation current of the device is slightly larger than that of structure B. Moreover, structure C also has higher transconductance, which characterizes the controllability of the gate to the channel, 29 and shows that the controllability of structure C to the channel is higher than that of structure B.

To investigate the impact of the gradual change in the Al composition in the AlGaN barrier layer from the P-GaN/ AlGaN interface to the AlGaN/GaN interface on the GaN channel, we set up six groups of barrier layers with different Al composition grades: Al: $0.3 \rightarrow 0.20$; $0.3 \rightarrow 0.22$; $0.3 \rightarrow 0.24$; $0.3 \rightarrow 0.26$; $0.3 \rightarrow 0.28$; $0.3 \rightarrow 0.30$. Figure [13a](#page-6-1) illustrates the transfer characteristics of the six sets of barrier layers with varying Al fractions. The threshold voltage of the device exhibits an increasing trend as the Al tapers from 0.3 to 0.2, attributed to the reduction in the 2DEG concentration resulting from the weakening of the polarization electric feld. Figure [13](#page-6-1)b shows the gate leakage currents of the AlGaN barrier layers with six diferent Al compositions. The gate leakage current exhibits an increasing trend as the Al content decreases. Notably, the gate leakage current reaches its maximum severity at Al values of 0.20 and 0.22. Conversely, when Al is set to 0.3 and 0.28, the gate leakage current is almost zero. When Al is 0.24 and 0.26, the gate leakage is also considered small for P-GaN gate HEMT devices with V_{gs} greater than 15 V.^{[30](#page-10-3)[–34](#page-10-4)} As the AlGaN barrier increases, the gate leakage current is signifcantly reduced. However, an excessive Al fraction leads to a decrease in the device's threshold voltage. Based on the

Fig. 12 (a) Transfer characteristic, (b) transconductance and (c) output characteristic curves of structure B and structure C.

Fig. 13 (a) Transfer characteristic curves of diferent Al compositions in the graded AlGaN barrier layer, (b) gate leakage current of diferent Al compositions in the graded AlGaN barrier layer at $V_{ds}=10$ V and $V_{gs}=0-30$ V.

Fig. 14 Structure D: P-GaN cap and recessed gate HEMT with graded AlGaN layer and added N-well in GaN channel.

Fig. 15 Schematic of structure D in the simulation software, showing cutline at two locations perpendicular to the *x*-axis.

aforementioned analysis, we determined that an Al composition gradient of 0.24 represents the optimal value for the AlGaN barrier layer.

Figure [14](#page-7-0) shows the addition of two 20-nm-thick N-wells with N-doping in the GaN channel of structure D (Fig. [1c](#page-1-0)). The remaining structural parameters remain unchanged. Figure [15](#page-7-1) presents a schematic of structure D with two regions perpendicular to the *x*-axis, designated as region 1 and region 2, in the simulation software. In Fig. [16a](#page-8-0) and b, the conduction band profles along the cutline in region 1 and region 2 are shown when N-doping in the N-well is set to 4×10^{18} cm⁻³. In region 1, the triangular potential well widens below the Fermi level due to the infuence of the N-well potential. In region 2, without the presence of an N-well, the rise in the conduction band of the P-GaN cap causes the triangular potential well at the AlGaN/GaN interface to be higher than the Fermi level, leading to the depletion of electrons in the channel. If the N-well is confned within region 1, the doping distribution in the N-well will not afect the electron concentration in the channel below the gate, and thus it will not afect the threshold voltage.

Figure [17](#page-8-1) shows the transfer characteristics of I_{ds} (V_{gs}) under different N-doping distributions in the N-well. Although the doping concentration in the N-well does not afect the threshold voltage of the device, the current density of the device increases with the doping concentration in the N-well. Figure [18](#page-8-2) illustrates the output characteristics of structure D devices under diferent *N*-type doping concentrations in the N-well, in which the inset shows the efect of diferent doping profles on the N-well drain current at $V_{\text{gs}}=15$ V. The output characteristics of the structure D devices increase with increasing N-doping concentration. This indicates that the addition of the N-well expands the triangular potential well below the Fermi level, thereby increasing the density of carriers participating in the conduction current in the channel. When the N-doping concentration exceeds 8×10^{18} cm⁻³, the saturation current of the device tends to level off. To achieve a better doping effect, we chose an N-well doping concentration of 8×10^{18} cm⁻³. Compared with the saturation current of the enhanced structure C device (646 mA/mm) (represented by the light purple dashed line in the inset), the saturation current of the structure D device is signifcantly higher. When the N-doping concentration is 8×10^{18} cm⁻³, the saturation current of the structure D device is 718 mA/mm. Although the ionized impurity scattering produced by doping *N*-type impurities in the GaN channel layer affects the carrier mobility in the GaN channel layer, the charge compensation effect of the N-well plays a major role. 35 35 35 As a result, the electrons after the ionization of *N*-type impurities participate in the conductivity, which leads to a signifcant increase in the saturation current of the device. Additionally, Fig. [19](#page-9-20) summarizes the threshold voltage and saturation current performance of the proposed composite trench-gate HEMT compared with other reported GaN-based HEMTs.[25,](#page-9-18)[34,](#page-10-4)[36](#page-10-6)[–40](#page-10-7) It can be observed that the HEMT of structure D exhibits a signifcant improvement in output performance at high threshold voltages compared with other GaN-based HEMTs. This points to a wide range of applications of this approach in high-power electronic devices.

Conclusion

In this paper, we propose a HEMT structure combining a P-GaN cap with a recessed gate and an AlGaN barrier layer with a graded Al component. Optimization of the P-GaN cap thickness and P-doping was performed using the Silvaco TCAD simulation tool to achieve an enhanced HEMT device. Additionally, the influence of

Fig. 16 Conduction band profle of structure D along (**a**) region 1 and (**b**) region 2 below the gate, showing triangular potential wells below the Fermi level at zero gate voltage in Fig. [15.](#page-7-1)

Fig. 17 Transfer characteristic curves for diferent N-doping concentrations in the N-well.

700 600 500 I_{ds} (m A /mm) 76 400 740 $\frac{1}{2}$ $sat(mA/mm)$ 700 300 680 v_{gs} =15 v 660 4×10^{18} cm-3 200 640 6×10^{18} cm-3 Current level of normally-off HEM $62($ 8×10^{18} cm-3 without N-wells 100 600 $4.0E+18$ $6.0E+18$ $8.0E+18$ $1.0E+19$ 1×10^{19} cm-3 Uniform donor concentration(cm) W/T Well $\overline{0}$ $\overline{0}$ 5 10 30 15 20 25 $V_{ds}(V)$

Fig. 18 Output characteristic curves for diferent N-doping concentrations in the N-well. The inset shows the efect of diferent doping profiles on the N-well drain current at V_{gs} = 15 V.

piezoelectric polarization in the AlGaN barrier layer under the gate electrode was investigated using polarization engineering techniques. Through comparative analysis, it was found that when the Al composition is graded from 0.3 to 0.24, the device exhibits a threshold voltage of $V_{\text{th}} = 8.6$ V and a transconductance of $g_m = 97.5$ mS/mm. Most design schemes for enhanced-mode HEMTs significantly impact the source–drain current of the device.

Therefore, in the optimized structure mentioned above, an N-well is added to the GaN channel layer, and the effect of the N-doping concentration on the device is discussed. Simulation results demonstrate that the participation of electrons from the N-well in conduction compensates for the channel degradation, thereby improving the device's current density. In particular, the saturation current of the device shows the most significant increase when the N-doping concentration is 8×10^{18} cm⁻³ ($I_{sat} = 718$ mA/ mm). Therefore, this structure holds great potential for applications in the field of high-power devices.

Fig. 19 Comparison of output and transfer characteristics of GaNbased HEMTs. The red star represents this work (HEMT devices of structure D).

Acknowledgments This work was supported by the Natural Science Foundation of Fujian Province (2023J011458, 2021I0025, 2022J011274).

Data Availability The data that support the fndings of this study are available from the corresponding authors upon reasonable request.

Conflict of interest The authors declare that they have no confict of interest.

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