

# **High‑Performance Field‑Efect Transistor Fabricated on CVD‑Grown MoS<sub>2</sub> Monolayers with Indium Contacts**

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#### **Abstract**

Molybdenum disulfide  $(MoS<sub>2</sub>)$ , an emerging two-dimensional semiconductor material, has been keenly studied for field-effect transistors (FETs). In this work, we explored the optical and electrical properties of FETs fabricated by MoS<sub>2</sub> flakes grown by chemical vapor deposition (CVD) and transferred to the electrodes through propylene carbonate flm. Large-area, highquality and highly crystalline MoS<sub>2</sub> monolayers up to 58  $\mu$ m are obtained through CVD. Flakes are characterized by optical microscopy, atomic force microscopy, Raman spectroscopy, and photoluminescence analysis. The back-gated measurements are performed in ambient conditions without any encapsulation of the device. The fabricated device reveals *n*-type behavior with high field-effect mobility of 32 cm<sup>2</sup>/V s and high current ON/OFF ratio of  $10^6$ . Good ohmic contact is achieved while using indium as source/drain electrodes. The large sized, highly crystalline flakes of  $MoS<sub>2</sub>$  and the fabricated device showing high feld-efect mobility and ON/OFF ratio make them potential candidates for high-performance nanoelectronics and optoelectronics devices.

**Keywords** Two-dimensional materials  $\cdot$  MoS<sub>2</sub>  $\cdot$  chemical vapor deposition  $\cdot$  propylene carbonate  $\cdot$  field-effect transistor  $\cdot$ indium

# **Introduction**

Because of its unique and remarkable electrical properties, graphene has potential for substitution in typical Si-based semiconductor devices.<sup>[1](#page-5-0)</sup> Although graphene has extraordinary high carrier mobility, $2$  its zero band gap limits its use in logic applications. Inspired by the invention of graphene monolayers, transition metal dichalcogenides (TMDs) such as  $MoS<sub>2</sub>$ ,  $MoSe<sub>2</sub>$ ,  $WS<sub>2</sub>$  and  $WSe<sub>2</sub>$  have gained significant consideration for their excellent potentials in research felds of nanotechnology, microelectronics, photonics, and opto-electronics.<sup>[3,](#page-5-2)[4](#page-5-3)</sup> TMDs possess a layered hexagonal structure having a dangling bond-free surface, excellent thermal con-ductivity, and stability.<sup>[5](#page-5-4)</sup> Among these TMDs,  $MoS_2$  has been

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studied widely as it exhibits excellent fexibility, electrical and optical properties.<sup>[6](#page-5-5)</sup> In  $MoS<sub>2</sub>$ , each Mo atom is covalently bonded to two sulfur atoms. This covalent bonding provides great mechanical strength to  $MoS<sub>2</sub>$ . MoS<sub>2</sub> has an outstanding ON/OFF ratio of about  $10^8$  and carrier mobility almost equal to 200 cm<sup>2</sup>/V s at standard temperature.<sup>[7](#page-5-6)</sup> MoS<sub>2</sub> has a tunable bandgap, from 1 to 2 eV. $8.9$  $8.9$  The bandgap of  $MoS_2$ changed from an indirect to a direct one from bulk  $MoS<sub>2</sub>$ to monolayer  $MoS<sup>9</sup>$  $MoS<sup>9</sup>$  $MoS<sup>9</sup>$  Monolayer  $MoS<sub>2</sub>$  has a direct bandgap of 1.9 eV which makes it suitable for use in electronic and optoelectronic devices.[10](#page-5-9)

Thin layers of  $MoS<sub>2</sub>$  can be obtained by various meth-ods such as mechanical exfoliation,<sup>[7](#page-5-6)</sup> liquid exfoliation,<sup>[11](#page-5-10)</sup> chemical vapor deposition  $(CVD)$ ,<sup>[12](#page-5-11)</sup> sulfurization,<sup>13</sup> physical vapor deposition, $^{14}$  and hydrothermal<sup>15</sup> and electrochemical synthesis. $16$  Flakes of high quality can be achieved by exfoliation, but these fakes are randomly dispersed, limiting control of the number of layers of  $MoS<sub>2</sub>$ . Hence, these methods are not designed for large-scale device fabrication. By synthesis,  $MoS<sub>2</sub>$  forms nanotubes (one-dimensional) or closed (zero-dimensional) structures. It is very rare to obtain large-area, high-quality, layer-controlled and highyield  $MoS<sub>2</sub>$  for bulk fabrication of devices with excellent electrical properties. CVD is used extensively to produce large-area layers of  $MoS<sub>2</sub>$  with control over the number of layers, but this method does not ensure the high quality of the fakes. Enhanced scalability has been reported by using various methods, for example sulfurization of molybdenum and its oxides, vapor-liquid growth by using powder  $MoO<sub>3</sub>$ and sulfur, thermal decomposition of ammonium thiomolybdates, and vapor-solid growth of  $MoS<sub>2</sub>$  monolayers by using powder of  $MoS_2$ .<sup>[17](#page-5-16)</sup> In emerging CVD methods, sulfurization is done after spin coating of a solution of metal on a sub-strate.<sup>[18](#page-5-17)</sup> Sulfurization with sulfur vapor has been demonstrated to be useful for fabricating sputtered-multilayer  $MoS<sub>2</sub>$ flms with an approximately 4-nm thickness with enhanced mobility.<sup>19</sup>Monolayers of MoS<sub>2</sub> have been achieved by CVD on a mica substrate. A 7.5-nm-thick  $MoS<sub>2</sub>$  film was produced. Except for monolayers, no other layers of  $MoS<sub>2</sub>$  have been fabricated. With the help of a nucleation promoter, perylene-3,4,9,10-tetracarboxylic acid tetrapotassium salt (PTAS), a highly crystalline and uniform monolayer of  $MoS<sub>2</sub>$  has been fabricated.<sup>[20](#page-5-19)</sup> However, it is very hard to use this monolayer uniformly on a large substrate. Wafer-scale monolayers of  $MoS<sub>2</sub>$  have been synthesized which exhibit on/off ratios ranging from  $10^5$  to  $10^8$  and carrier mobility up to  $1.2 \text{ cm}^2/\text{V s}$ . A recent work investigating field-effect transistors (FETs) fabricated on monolayer  $MoS<sub>2</sub>$  produced by CVD showed carrier mobility of  $1.45 \text{ cm}^2/\text{V}$  s with an ON/OFF ratio of  $10^{4.22}$  $10^{4.22}$  $10^{4.22}$  MoS<sub>2</sub> FETs fabricated by mechanical exfoliation have shown good carrier mobility, but FETs based on CVD-grown  $MoS<sub>2</sub>$  exhibited low electrical performance. Growing a layer-controlled, contamination-free, high-quality, large-scale  $MoS<sub>2</sub>$  films for nano-devices with excellent performance is still challenging. Moreover, no optimized parameters and methods have been reported for the growth of monolayer  $MoS<sub>2</sub>$ .

In this work, we fabricated uniform, highly crystalline, large-area, high-quality monolayers of  $MoS<sub>2</sub>$  by CVD on  $SiO<sub>2</sub>$  substrates. Atomic force microscopy (AFM), Raman spectroscopy, and photoluminescence (PL) analysis confirmed that our monolayers of  $MoS<sub>2</sub>$  are highly scalable, uniform, and crystalline in nature. We also fabricated FETs on CVD-synthesized monolayer  $MoS<sub>2</sub>$  films which exhibited an on/off ratio of  $10^6$  with comparative carrier mobility of 32  $\text{cm}^2$ /V s. The contact between the metal electrodes and the  $MoS<sub>2</sub>$  becomes one of the most critical aspects restricting device performance. Good ohmic contact is attained when we use indium (In) as electrodes, which improves the performance of the  $MoS<sub>2</sub>$ -based FETs. Our research work demonstrates that excellent electrical performance of devices based on  $MoS<sub>2</sub>$  can be achieved by controlled growth of large-area and high-quality  $MoS<sub>2</sub>$  films.

# **Experimental Setup**

#### **MoS<sub>2</sub>** Synthesis

Figure [1](#page-1-0) demonstrates the setup of the experiment. Thin films of  $MoS<sub>2</sub>$  are grown on  $O<sub>2</sub>$  plasma treated 100 nm Si $O<sub>2</sub>$ / Si substrates in 2-zone CVD with 5 cm quartz. The substrate was pre-treated with  $O_2$  plasma for 120 seconds at 110 W to attain a hydrophilic surface.  $MoO<sub>3</sub>$  and sulfur powders purchased from Sigma-Aldrich are used as precursors. Two milligrams of  $MoO<sub>3</sub>$  powder was taken in a quartz boat with the substrate in a face-down position arranged on the top of boat. Next, 300 mg of sulfur powder was placed in a separate boat. A distance of 15 cm was maintained between boats containing the  $MoO<sub>3</sub>$  and sulfur powder. Sulfur was placed in an upstream position in the furnace. Thin layers of  $MoS<sub>2</sub>$ were grown at 750°C under 500 sccm Ar flow.

The shape, density, and growth of the  $MoS<sub>2</sub>$ , flakes were observed by optical microscopy (OM). The uniformity,



<span id="page-1-0"></span>**Fig. 1** (a) Schematic diagram of the CVD setup for growth of MoS<sub>2</sub>. (b) Temperature profile of S and MoO<sub>3</sub> during growth process.



<span id="page-2-0"></span>**Fig. 2** (a) OM image of MoS<sub>2</sub> flakes. (b) PL image of MoS<sub>2</sub> flakes. (c, d) Density of MoS<sub>2</sub> monolayer flakes. (e) Area of MoS<sub>2</sub> monolayer flake. (f) AFM image of  $MoS<sub>2</sub>$  flake.

smoothness and surface structure were examined by a Nano-Magnetics ezAFM system. Raman and PL analysis of  $MoS<sub>2</sub>$ fakes were performed using a Renishaw inVia confocal Raman microscope system with an objective lens of 100× with numerical aperture of 0.75 A. continuous wave laser of 532 nm with power of 1 mW was used. Integration time of 0.3 s was used for Raman spectra and 0.02 for PL analysis.

#### **Device Fabrication and Characterizations**

The back-gated FET was fabricated by transferring the CVD-grown  $MoS_2$  monolayer on the  $SiO_2/Si$  substrate, and Au/In metal electrodes were deposited on it by optical lithography. A total of 80 nm of Au was deposited on 20 nm indium (In) to make electrodes. The  $MoS<sub>2</sub>$  flakes synthesized via CVD were relocated onto a  $SiO<sub>2</sub>/Si$  substrate using a pick-up technique involving the application of propylene carbonate (PC). The PC solution was made by combining





<span id="page-3-0"></span>Fig. 3 (a) Raman spectra of MoS<sub>2</sub>. (b) PL spectra of MoS<sub>2</sub>.

methoxybenzene with PC particles at a mass fraction of around 10%. PC solution was drop-cast on the  $SiO<sub>2</sub>/Si$  substrate containing CVD-grown  $MoS<sub>2</sub>$  flakes. The PC solution was hardened into the PC flm by placing it on a hot plate for 10 min at 110°C. The PC flm worked as a stamp and carried the  $MoS<sub>2</sub>$  layers out of the substrate. Then this PC film was moved to the targeted substrate. At 80°C, the PC film loosened its adhesion and  $MoS<sub>2</sub>$  layers were transferred on the desired substrate by removing the PC flm. The transfer characteristics of the  $MoS<sub>2</sub>$ -based FET were measured with a Keithley 2400 source meter.

# **Results and Discussion**

#### **Optical Characterization**

Figure [2](#page-2-0)a presents the OM images of CVD-grown  $MoS<sub>2</sub>$ flakes with triangular shape. PL images of  $MoS<sub>2</sub>$  flakes can be seen in Fig. [2b](#page-2-0). In Fig. [2](#page-2-0)c and d the density of the CVDgrown  $MoS<sub>2</sub>$  flakes can be seen. We observed maximum crystal size up to 58 µm for this growth. The OM image of a  $MoS<sub>2</sub>$  triangle with edges longer than 52.58 µm can be seen in Fig. [2](#page-2-0)e. In Fig. [2f](#page-2-0), a smooth, uniform, highly crystalline and large-area layer structure of  $MoS<sub>2</sub>$  flake can be observed with AFM.

Figure [3a](#page-3-0) presents the Raman spectra of  $MoS<sub>2</sub>$ . Most significantly, the Raman spectra of  $MoS<sub>2</sub>$  show two prominent peaks of  $MoS<sub>2</sub>$ . Two non-resonant characteristic peaks of  $MoS<sub>2</sub>$  are found. These two peaks are dominated by two vibrational modes.  $E_{2g}^{1}$  (in-plane vibration) is due to the in-plane vibration of two sulfur atoms with Mo atoms. The  $E_{2g}^1$  mode appeared at 382 c/m. The  $A_{1g}$  mode (out of plane) refers to the out-of-plane vibrations of sulfur atoms in opposite directions. The  $A_{1g}$  mode is observed at 404 c/m. These two modes are very sensitive to the thickness of the  $MoS<sub>2</sub>$  structure. The thickness and number of layers of  $MoS<sub>2</sub>$  are determined by the peak positions of these two modes. The diference between these two peaks tells us about the number of layers of  $MoS<sub>2</sub>$ . The difference between these two peaks, observed from our Raman spectra, is  $A_{1g} - E_{2g}^1 = 405 - 385 = 20$  c/m and ratio of  $A_{1g}/E_{2g}^1 = 1.057$  $\approx$  1. The difference of 20 c/m shows that our MoS<sub>2</sub> is mon-olayer.<sup>[23,](#page-6-0)[24](#page-6-1)</sup> Figure [3b](#page-3-0) presents the PL spectra of MoS<sub>2</sub>. Excitons are the electron-hole pairs which are formed by optical excitation of electrons from valence band to conduction band when  $MoS<sub>2</sub>$  is exposed to laser of energy below than 670 nm. As these excited electrons are unstable, they come back from the conduction band to the valence band and relax. In the relaxation state, these electrons recombine with the holes produced in the valence band after their excitation. Radiation is induced because of this recombination. As a result, we observed various peaks of  $MoS<sub>2</sub>$  in its PL spectra. Two



<span id="page-4-0"></span>**Fig. 4** (a) Cross-sectional diagram of CVD-grown MoS<sub>2</sub> -based FET device. (b) Optical image of fabricated MoS<sub>2</sub> -based FET. (c) Output curves of device by sweeping the  $V_G$  from −60 to 60 V. (d, e) Transfer characteristics  $(I_D-V_G)$  of monolayer MoS<sub>2</sub>-based FET.

well-defined peaks are observed in bare  $MoS<sub>2</sub>$  spectra at 1.84 eV and 2.08 eV. These peaks are referred to as A exciton and B exciton, respectively. These peaks demonstrate the exciton formation, and this diference in energy between these two peaks is attributable to the spin-orbit splitting of the valence band. $25$ 

## **Electrical Characterization**

Figure [4](#page-4-0)a shows a cross-sectional illustration of the CVDgrown  $MoS<sub>2</sub>$ -based FET device. The device is fabricated on a  $SiO<sub>2</sub>/Si$  substrate.  $SiO<sub>2</sub>$  with a thickness of 300 nm acts as the gate oxide layer deposited on 500-nm-thick Si. It acts as a back gate to control the fabricated FET. The channel length  $L_c$  of the active region is 5  $\mu$ m and channel width  $W_c$ is 6 µm. Figure [4b](#page-4-0) shows the optical image of the fabricated  $MoS<sub>2</sub>$ -based FET. Figure [4](#page-4-0)c, d demonstrates the output and transfer characteristics of the monolayer  $MoS<sub>2</sub>$ -based FET in contact with In. Output curves of the FET were obtained by taking the Vg from −60 V to 60 V. Good ohmic contact with In was achieved. All electrical measurements were taken in ambient conditions. It is obvious from the transfer curve in Fig. [3](#page-3-0)d that  $MoS_2$ -based FET exhibits *n*-type behavior. From the saturation region of the on-state conducting channel, threshold voltage  $V_{TH}$  is defined.  $V_{TH}$  is obtained from

the intercept made from linear extrapolation on the  $I_D-V_G$ graph. FET is usually in the on state when  $V_G = 0$  V, which represents the collective conduction of electrons. Figure [3e](#page-3-0) represents the  $I_D-V_G$  curves demonstrating the distinct *n*-type characteristics with ON/OFF =  $10<sup>6</sup>$ 

The field-effect mobility of the  $MoS<sub>2</sub>$ -based FET is obtained by using the following equation.

$$
\mu_{\rm eff} = (\frac{L}{C_{0x}WV_{ds}})(\frac{dI_{ds}}{dV_{gs}})
$$

where  $\mu_{FE}$  is field-effective mobility,  $g_m$  is transconductance  $=\frac{\partial Id}{\partial Vgs}$ ,  $L_c$  is the length of the channel,  $C_{go}$  is the capacitance of the gate oxide,  $V_D$  is the drain voltage and  $W_c$  is the width of the channel.  $C_{\text{go}} = \varepsilon_{\text{g0}} / t_{\text{og}}$  where  $\varepsilon_{\text{go}}$  is the dielectric constant of the gate oxide and  $t_{\text{op}}$  is the thickness of the gate oxide. This device shows good electronic performance. High mobility of 32 cm<sup>2</sup>/V s and an ON/OFF ratio of  $10^6$  is attained, which is comparatively higher than the previous reported devices fabricated on CVD-grown  $MoS<sub>2</sub>$  monolayers.<sup>22,[26](#page-6-3)[,27](#page-6-4)</sup>

# **Conclusion**

We have demonstrated the optical and electronic properties of a FET device based on  $MoS<sub>2</sub>$  flakes grown by CVD. A uniform, highly crystalline, large-area, high-quality monolayer of  $MoS<sub>2</sub>$  is achieved by CVD on  $SiO<sub>2</sub>$  substrates. Electrodes are fabricated by optical lithography and  $MoS<sub>2</sub>$ , grown by CVD, is transferred through PC film. AFM, Raman, PL, and back-gated measurements are performed in ambient conditions without any encapsulation of the device. The fabricated device reveals *n*-type behavior with high field-effect mobility of 32  $\text{cm}^2$ /V s and high current ON/OFF ratio of  $10<sup>6</sup>$ . Good ohmic contact is achieved when using In as source/drain electrodes. The large-sized, highly crystalline flakes of  $MoS<sub>2</sub>$  and the fabricated device showing high feld-efect mobility and ON/OFF ratio make them potential candidates for high-performance nanoelectronic and optoelectronic devices.

**Conflict of interest** The authors declare that they have no confict of interest.

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