



Junction Parameters and Electrical Characterization of the Al/n-Si/Cu₂CoSnS₄/Au Heterojunction

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Quaternary kesterite thin films of Cu₂CoSnS₄ were deposited on the n-Si substrate to fabricate Cu₂CoSnS₄/n-Si heterojunctions. The x-ray diffraction and field emission scanning electron microscopy were employed to study the structural properties of the Cu₂CoSnS₄ deposited on to n-Si single crystal substrate. The capacitance–voltage measurements of the Cu₂CoSnS₄/n-Si heterojunction were investigated to study the junction nature which displays an abrupt junction. The dark *I–V* characteristics of the Cu₂CoSnS₄/n-Si heterojunction displays a rectification behavior. We characterized the influence of the annealing temperature on the magnitudes of the diode parameters of the Cu₂CoSnS₄/n-Si heterojunction. The barrier height ϕ_b of the Cu₂CoSnS₄/n-Si heterojunction was increased with raising the annealing temperature while the ideality factor *n* has a reverse performance. The illuminated *J–V* plot of the Cu₂CoSnS₄/n-Si heterojunction displays an efficiency of 6.17% for the prepared junction.

Key words: Copper cobalt tin sulfide, spray pyrolysis technique, XRD, FE-SEM, photovoltaic, *I–V* and *C–V* characteristics, heterojunction

INTRODUCTION

Recently, Earth-abundant kesterite thin films like Cu₂ZnSnS₄, Cu₂CdSnS₄, Cu₂FeSnS₄, Cu₂MnSnS₄, and Cu₂CoSnS₄ attracted high interest in the last years owing to they are inexpensive, nontoxic, more abundant and exhibit good optical and electrical properties.^{1–3} These materials exhibit a *p*-type conductivity and excellent optoelectronic properties, so they are ideal absorber materials for solar cells.⁴ The high stability of the kesterite thin films makes it suitable for various applications like solar cells and photonic devices.⁵ Cu₂ZnSnS₄ films

display a high absorption coefficient and a suitable band gap so they are considered a good competitor for CdTe and CuIn_xGa_{1–x}S₂ in the fabrication of solar cells. The Cu₂ZnSnS₄ solar cell reached an efficiency of 12%.⁶ So, the study of the other kesterite thin films, such as Cu₂CdSnS₄, Cu₂FeSnS₄, Cu₂MnSnS₄, and Cu₂CoSnS₄ is very important to fabricate low-cost solar cells. Cu₂CoSnS₄ is an alternative solar absorber with excellent optical, thermoelectric and magnetic properties which is well suited for utilization in solar energy conversion.⁷ Different preparation techniques were utilized to fabricate the Cu₂CoSnS₄ powder and thin films like spray pyrolysis,⁸ hydrothermal,⁹ solvothermal,¹⁰ sol–gel method,¹¹ and hot injection¹² techniques. The previous articles are available for studying the structural and optical

properties of the CCTS thin films while there is no interest in the fabrication of CCSS/n-Si heterojunction. So, the present work is aimed to fabricate and study the structural and electrical properties of the Cu₂CoSnS₄/n-Si heterojunction synthesized by spray pyrolysis method. The junction parameters of the Cu₂CoSnS₄/n-Si heterojunction have been estimated from the current–voltage characteristics curve. The junction nature and the donor carrier concentrations of the Cu₂CoSnS₄/n-Si heterojunction were evaluated from the capacitance–voltage characteristics in the dark.

EXPERIMENTAL DETAILS

Synthesis of the Cu₂CoSnS₄ Thin Film and Heterojunction

High quality Cu₂CoSnS₄ thin film was synthesized by the interaction between 0.1 M copper chloride hydrate, 0.05 M cobalt chloride hydrate, 0.05 M stannic chloride hydrate and 0.2 M thiourea. A mixture of propanol to water ratio of 1:3 ml has been used for solution preparation. The pH of the solution set at 10. The final Cu₂CoSnS₄ solution was sprayed into a heated glass substrate held at 300°C. For heterojunction fabrication, an *n*-type Si substrate was chemically etched by a mixture of different acids (HF:HNO₃:CH₃COOH) with a ratio of 1:6:1¹³ Then the n-Si substrate was cleaned well with deionized water then dried. After the cleaning process, an Al film with a thickness of 200 nm was deposited on the bottom side of the n-Si substrate. A Cu₂CoSnS₄ thin film of thickness 479 nm was sprayed on the top surface of the n-Si substrate by spray pyrolysis system. Then Au electrode with a thickness of 200 nm was deposited on the surface of the Cu₂CoSnS₄ thin film. The Al and Au electrodes were performed by a thermal evaporation technique type (Edward 306 A). Figure 1 displays the Al/n-Si/Cu₂CoSnS₄/Au heterojunction.

Characterization of the Cu₂CoSnS₄ Thin Film and Heterojunction

XRD scan was carried out to examine the structure of the Cu₂CoSnS₄ thin film. Quanta FEG 250 field emission scanning electron microscopy was used to scan the surface morphology of the

Cu₂CoSnS₄ thin film. A computerized *C–V* meter was used to study the dark capacitance–voltage (*C–V*) characteristics of the Al/n-Si/Cu₂CoSnS₄/Au heterojunction. A programmable electrometer (Type Keithley-2635 A) was employed to measure the current–voltage *I–V* characteristics of the Al/n-Si/Cu₂CoSnS₄/Au heterojunction. A halogen lamp was employed for the illumination process. A solar power meter was used to measure the intensity of the incident light.

RESULTS AND DISCUSSION

Structural Characterizations

Morphology and EDAX of the Cu₂CoSnS₄ Thin Film

Representative surface morphology image of the Cu₂CoSnS₄ film is presented in Fig. 2. The morphology of the Cu₂CoSnS₄ layer displayed that the film has a crack-free smooth surface and formed from spherical grains. The EDAX pattern demonstrated that the existence of Cu, Co, Sn and S in the Cu₂CoSnS₄ film and its atomic ratio are Cu = 21.62, Co = 11.43, Sn = 10.16, S = 38.84 and Si = 17.95 at.%, respectively.

XRD Study

The XRD pattern of the Cu₂CoSnS₄ thin film grown on the n-Si substrate is presented in Fig. 3. The pattern shows that Cu₂CoSnS₄ thin films are polycrystalline in nature. The previous work illustrates that the Cu₂CoSnS₄ thin films exhibit tetragonal structure. Maldar et al.¹⁴ demonstrated that Cu₂CoSnS₄ thin films prepared by spray pyrolysis reveal a tetragonal phase. Mokurala et al.¹⁵ confirmed that Cu₂CoSnS₄ thin films fabricated by spin coating display a tetragonal phase. In this work, the study of the diffraction pattern of the Cu₂CoSnS₄ thin films displays that the appeared planes were matched with the tetragonal crystal structure Cu₂CoSnS₄ phase according to JCPDS card No. 26-0513.

The structural constants of the Cu₂CoSnS₄ film represented in the crystallite size (*D*), the lattice strain (ϵ), and dislocation density (δ) have been given via the following Scherer's formulas^{16–18}:

$$D = \frac{0.9\lambda}{\beta \cos(\theta)}, \quad (1)$$

$$\epsilon = \frac{\beta \cos(\theta)}{4}, \quad (2)$$

$$\delta = \frac{1}{D^2}. \quad (3)$$

Here β denotes and the experimental full width at the half maximum (FWHM) in radians and θ represents the Bragg diffraction angle.

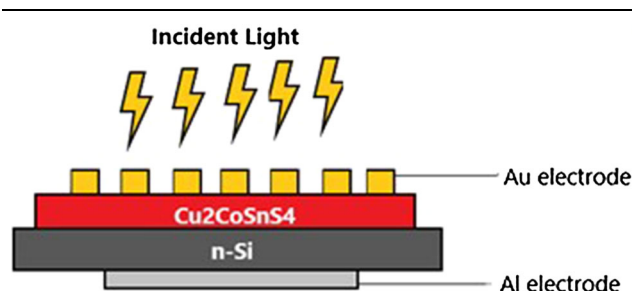


Fig. 1. Schematic diagram of Al/n-Si/Cu₂CoSnS₄/Au heterojunction.

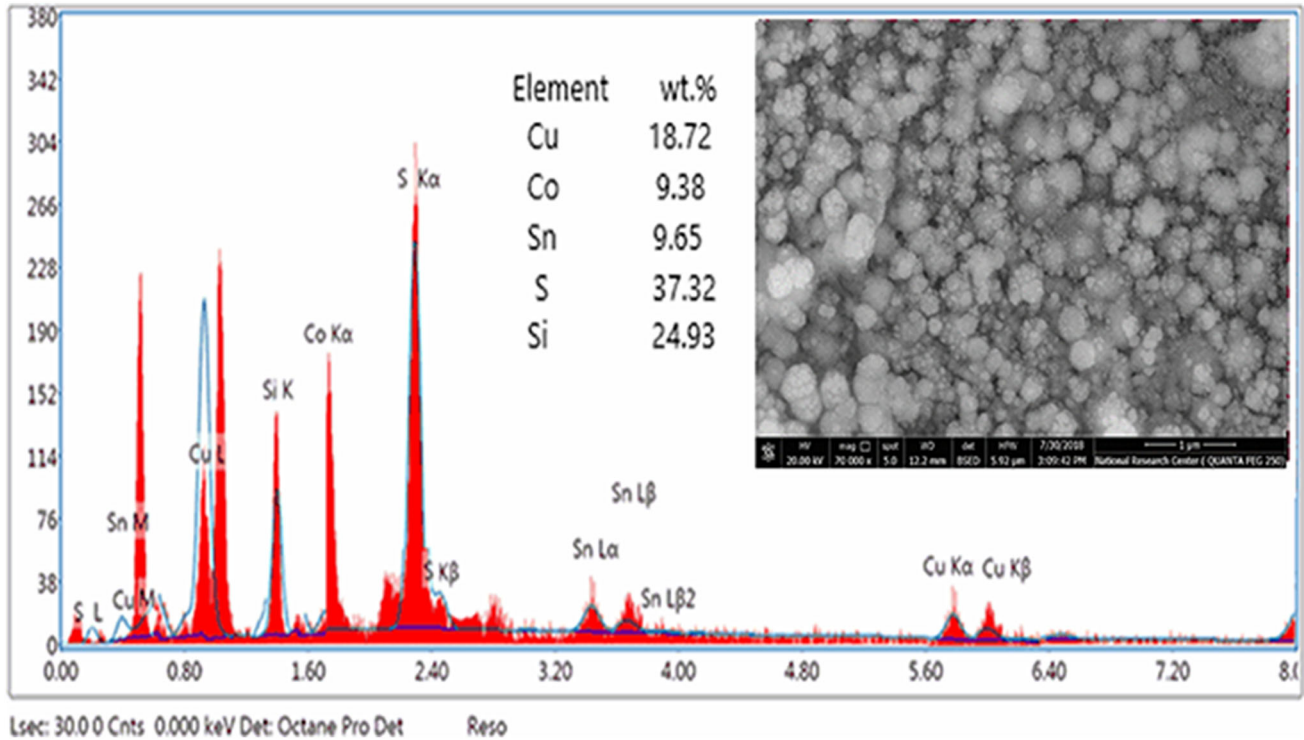


Fig. 2. The FE-SEM and EDAX spectrum of the $\text{Cu}_2\text{CoSnS}_4$ thin film deposited on the n-Si substrate.

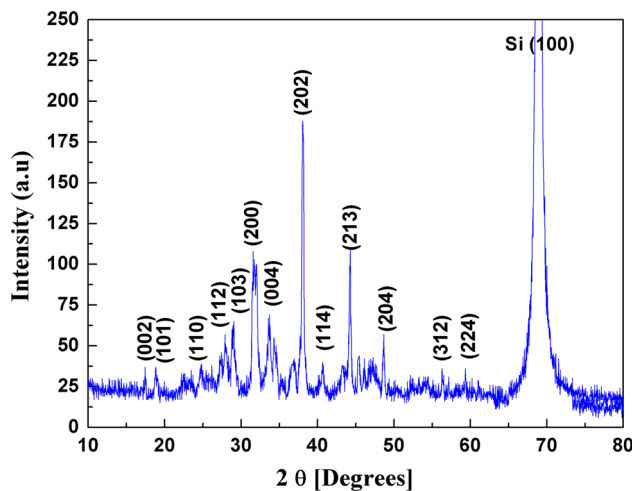


Fig. 3. XRD of the sprayed $\text{Cu}_2\text{CoSnS}_4$ thin film fabricated on the n-Si substrate.

The magnitudes of the structural constants D , ε and δ of the $\text{Cu}_2\text{CoSnS}_4$ thin film are 47.68 nm, 3.047×10^{-3} and $4.39 \times 10^{-4} \text{ nm}^{-2}$ respectively.

Capacitance–Voltage Characterizations

The Capacitance–Voltage (C – V) measurements of the $\text{Cu}_2\text{CoSnS}_4$ /n-Si heterojunction occurred at 1 MHz in the dark condition. The C^{-2} – V curve of the $\text{Cu}_2\text{CoSnS}_4$ /n-Si heterojunction is presented in Fig. 4. The curve displays a linear relation, which

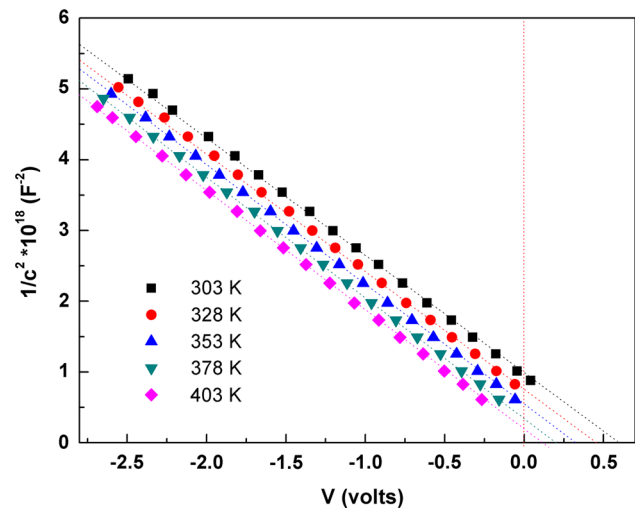


Fig. 4. The dark C – V characteristics of the Al/n-Si/ $\text{Cu}_2\text{CoSnS}_4$ /Au heterojunction.

revealed that the junction exhibits an abrupt nature. The dependence of the junction capacitance on the reverse bias potential for the $\text{Cu}_2\text{CoSnS}_4$ /n-Si heterojunction was given via^{19,20}:

$$C^2 = \frac{qN_A N_D \varepsilon_1 \varepsilon_2}{2(\varepsilon_1 N_A + \varepsilon_2 N_D)} \frac{1}{(V_{bi} - V)}. \quad (4)$$

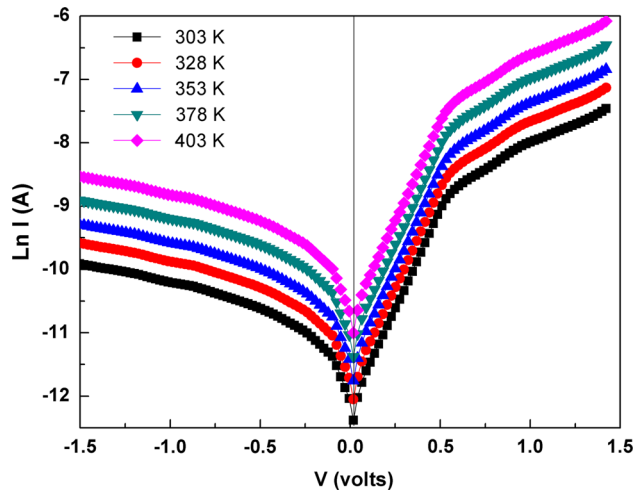
Here ε_1 denotes the dielectric constant of the Si substrate and equal $1.044 \times 10^{-11} \text{ F cm}^{-1}$, q

Table I. The magnitudes of the V_{bi} and N evaluated from the dark $C-V$ characteristics for the (Al/n-Si/Cu₂CoSnS₄/Au) heterojunction

T (K)	V_{bi} (V)	$N * 10^{19}$ (cm ⁻³)
308	0.54	3.91
323	0.432	4.28
343	0.35	5.31
353	0.217	5.87
373	0.13	6.26

Table II. The heterojunction parameters evaluated from the dark $I-V$ characteristics of the (Al/n-Si/Cu₂CoSnS₄/Au) heterojunction

T (K)	n	ϕ_b (eV)	R_S (Ω)	R_{sh} (k Ω)
308	2.68	0.785	205	52.43
323	2.47	0.812	191	43.61
343	2.31	0.843	165	35.15
353	2.19	0.875	147	31.82
373	2.06	0.893	133	28.76


 Fig. 5. Forward and reverse $I-V$ characteristics of the Al/n-Si/Cu₂CoSnS₄/Au heterojunction in the dark conditions.

denotes the charge of the electron and equals 1.6×10^{-19} C, N_D denotes the donor concentration of n-Si and equal 10^{15} cm⁻³, N_A denotes the acceptor concentration of the Cu₂CoSnS₄ and ϵ_2 denotes the dielectric constant of the Cu₂CoSnS₄ thin film.

By extrapolating the line in the Fig. 4 to $C^{-2} = 0$, the built-in voltages V_{bi} of the Cu₂CoSnS₄/n-Si heterojunction can be estimated. The magnitudes of V_{bi} and N for the Cu₂CoSnS₄/n-Si device are listed in Table I. Table I illustrated that the variance of built-in voltages (V_{bi}) and carrier concentration (N) with temperature for the Cu₂CoSnS₄/n-Si heterojunction. It is observable that the magnitudes of V_{bi} were reduced with raising the temperature while the magnitudes of N were increased with increasing the temperature. Actually, the interface density of charges improves by raising the temperature. It has a significant impact on the apparent built-in voltage.

Evidently, raising the interface density of charges reduces the built-in voltage. The large interface density of states likewise acts as practical tunneling centers.²¹

Dark Current-Voltage Characterization

The dark $I-V$ characteristic is very useful to determine the diode parameters like the ideality factor n , shunt resistance R_s , and the series resistance R_{sh} and barrier height ϕ_b . Figure 5 depicts the $I-V$ measurements of the Al/n-Si/Cu₂CoSnS₄/Au heterojunction in the dark at various temperatures. It is observed from this curve that the $I-V$ plot of the Cu₂CoSnS₄/n-Si device displays a rectification performance.

The $I-V$ curve of the Al/n-Si/Cu₂CoSnS₄/Au heterojunction in the forward bias region was studied according to the thermionic emission theory by the presented relation^{19,20}:

$$I = I_0 \left(\exp \left(\frac{qV}{nKT} \right) - 1 \right), \quad (5)$$

where n represents ideality factor, T represents the temperature in Kelvin, q represents the electronic charge, V represent applied voltage and K is the Boltzmann constant.

The reverse saturation current of the Cu₂CoSnS₄/n-Si device was estimated by the below formula^{21,22}:

$$I_0 = AA^* T^2 \exp \left[\frac{-q\phi_b}{KT} \right]. \quad (6)$$

Here ϕ_b represents the barrier height, A^* denotes the Richardson constant for the n-Si (which equals $A^* = 122 \text{ A/cm}^2 \text{ K}^2$)²³ and A denotes the Schottky contact area.

The magnitudes of ideality factor n , of the Al/n-Si/Cu₂CoSnS₄/Au heterojunction, was estimated from the slope of the linear plot of Fig. 5 according to the below expression^{24,25}:

$$\ln(I) = \frac{qV}{nKT} + \ln(I_0). \quad (7)$$

The ideality factor values n of the Al/n-Si/Cu₂CoSnS₄/Au heterojunction are given in Table II. It is noted from Table II that the n values of the Al/n-Si/Cu₂CoSnS₄/Au heterojunction were decreased via increasing the annealed temperature. As observed the magnitudes of ideality factor n for n values of the Al/n-Si/Cu₂CoSnS₄/Au heterojunction were higher than unity. This performance occurred

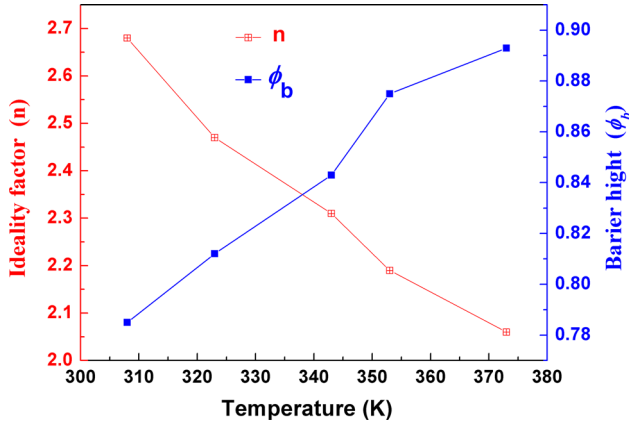


Fig. 6. Dependence of ideality factor and barrier height on temperature for the Al/n-Si/Cu₂CoSnS₄/Au heterojunction.

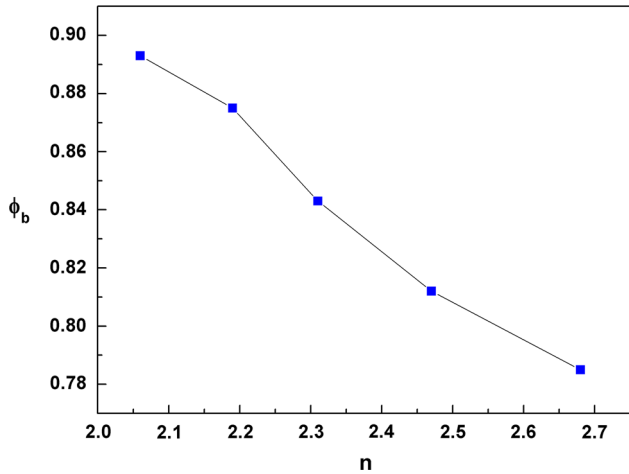


Fig. 7. Plot of the ideality factor versus barrier height for the Al/n-Si/Cu₂CoSnS₄/Au heterojunction.

because of increasing of the leakage current due to the recombination of the electron and holes in the depletion region.²⁶

The barrier height ϕ_b at zero bias for the Al/n-Si/Cu₂CoSnS₄/Au heterojunction is given by the next relation^{27,28}:

$$\phi_b = \frac{K_B T}{q} \ln \left(\frac{A A^* T^2}{I_0} \right). \quad (8)$$

The magnitudes of barrier height ϕ_b of the Al/n-Si/Cu₂CoSnS₄/Au heterojunction are listed in Table II.

Figure 6 presents the temperature effect on the ϕ_b and n of the Al/n-Si/Cu₂CoSnS₄/Au heterojunction. By raising the annealing temperature on the Al/n-Si/Cu₂CoSnS₄/Au heterojunction, the ϕ_b values were increased while the n values were decreased. This performance is in agreement with various articles like Farag²⁹ and agree with Uluşan.³⁰

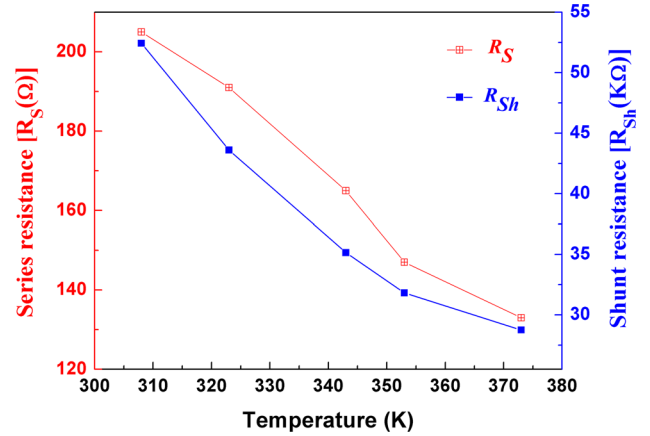


Fig. 8. Dependence of series and shunt resistances on temperature for the Al/n-Si/Cu₂CoSnS₄/Au heterojunction.

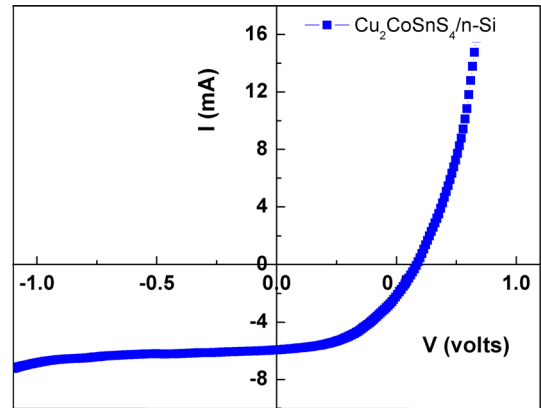


Fig. 9. The I - V characteristics of the Al/n-Si/Cu₂CoSnS₄/Au heterojunction under illumination.

The relation between the ideality factor n and the barrier height ϕ_b for the Al/n-Si/Cu₂CoSnS₄/Au heterojunction is presented in Fig. 7. It is observed from this plot that the magnitudes of the ideality factor n increase as the barrier height ϕ_b decreases.

The series resistance R_S of the Al/n-Si/Cu₂CoSnS₄/Au heterojunction was estimated via the slope of the linear plot of the forward I - V plot according to the following relation³¹:

$$R_S = \frac{\Delta V_{\text{Forward Bias}}}{\Delta I_{\text{Forward Bias}}}. \quad (9)$$

While the magnitude of shunt resistance R_{Sh} for the Al/n-Si/Cu₂CoSnS₄/Au heterojunction was evaluated via the slope of the linear plot of the reverse I - V plot according to the following expression³⁰:

$$R_{Sh} = \frac{\Delta V_{\text{Reverse Bias}}}{\Delta I_{\text{Reverse Bias}}}. \quad (10)$$

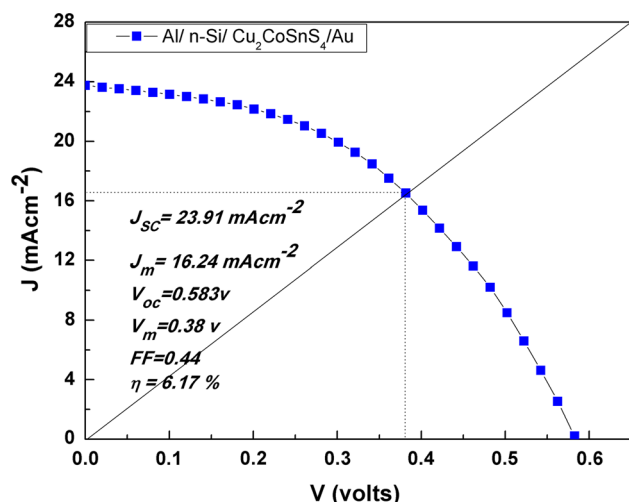


Fig. 10. The J - V characteristics for the Al/n-Si/Cu₂CoSnS₄/Au heterojunction under illumination of 100 mW cm⁻².

Figure 8 displays the temperature effect on the magnitudes of the shunt resistance R_{Sh} and the series resistance R_S for the Al/n-Si/Cu₂CoSnS₄/Au heterojunction. There is an enhancement in the conductivity of the fabricated device resulting from the decreasing of R_S and R_{Sh} with increasing the temperature.³² The magnitudes of R_S and R_{Sh} are given in Table II.

Photovoltaic Analysis of the Al/n-Si/Cu₂CoSnS₄/Au

The photovoltaic analysis of the Al/n-Si/Cu₂CoSnS₄/Au heterojunction has been investigated via determining the I - V characteristics below the illumination conditions. The illuminated I - V curve of the Al/n-Si/Cu₂CoSnS₄/Au heterojunction is presented in Fig. 9. As observed the current values of the Al/n-Si/Cu₂CoSnS₄/Au heterojunction under illumination conditions was higher than the current values in the dark. This performance is owing to the production of electron-hole pairs.³³⁻³⁵

Figure 10 depicts the J - V curve of the Al/n-Si/Cu₂CoSnS₄/Au heterojunction. Moreover, the photovoltaic constants of the prepared heterojunction as $J_{SC} = 23.91$ mA cm⁻², $J_m = 16.24$ mA cm⁻², $V_{OC} = 0.583$ V, $V_m = 0.38$ V and the efficiency were estimated via this plot.

The magnitudes of the device efficiency (η) and fill factor of the Al/n-Si/Cu₂CoSnS₄/Au heterojunction have been evaluated according to the below relations^{36,37}:

$$\eta = \frac{P_{\max}}{P_{\text{in}}} = \frac{\text{FF} \times V_{OC} \times J_{SC}}{P_{\text{in}}} \times 100\%, \quad (11)$$

$$\text{FF} = \frac{V_m \times J_m}{V_{oc} \times J_{sc}}, \quad (12)$$

where P_{in} denotes the input energy from the sun, P_{max} denotes the output energy from the device, FF is the fill factor, J_{sc} is the short-circuit current density and V_{oc} represents the open-circuit voltage. The magnitudes of the FF and the η of the Al/n-Si/Cu₂CoSnS₄/Au heterojunction were 0.44% and 6.17% respectively.

CONCLUSION

In the present study, Cu₂CoSnS₄ thin film was sprayed on the n-Si via spray pyrolysis technique. The formation of the Cu₂CoSnS₄ film was confirmed by XRD and FE-SEM. The Cu₂CoSnS₄ thin films are polycrystalline. The EDX pattern displays that the Cu₂CoSnS₄ thin film is near stoichiometric in composition. The Capacitance- Voltage study of the Cu₂CoSnS₄/n-Si heterojunction displays that the junction nature is an abrupt junction. The diode parameters of the Cu₂CoSnS₄/n-Si heterojunction were determined by investigating temperature dependences of the I - V characteristics. By increasing the temperature, the magnitudes of the barrier height ϕ_b of the Cu₂CoSnS₄/n-Si heterojunction increases while the magnitudes of the ideality factor n , shunt resistance R_{Sh} and series resistance R_S were decreased. The photovoltaic constants of the Cu₂CoSnS₄/n-Si heterojunction were evaluated for display efficiency of 6.17%.

REFERENCES

1. S. Siebentritt and S. Schorr, *J. Prog. Photovolt.* 20, 512 (2012). <https://doi.org/10.1002/pip.2156>.
2. M. Nakashima, T. Yamaguchi, S. Yukawa, J. Sasano, and M. Izaki, *J. Thin Solid Films* 621, 47 (2017). <https://doi.org/10.1016/j.tsf.2016.11.035>.
3. A. Walsh, S. Chen, S.-H. Wei, and X.-G. Gong, *Adv. Eng. Mater.* 2, 400 (2012). <https://doi.org/10.1002/aenm.201100630>.
4. S.S. Fouad, I.M. El Radaf, P. Sharma, and M.S. El-Bana, *J. Alloys Compd.* 757, 124 (2018). <https://doi.org/10.1016/j.jallcom.2018.05.033>.
5. A. Ennaoui, M. Lux-Steiner, A. Weber, D. Abou-Ras, I. Kötschau, H.-W. Schock, R. Schurr, A. Hölzling, S. Jost, and R. Hock, *J. Thin Solid Films* 517, 2511 (2009). <https://doi.org/10.1016/j.tsf.2008.11.061>.
6. T.K. Todorov, J. Tang, S. Bag, O. Gunawan, T. Gokmen, Y. Zhu, and D.B. Mitzi, *Adv. Eng. Mater.* 3, 34 (2013). <https://doi.org/10.1002/aenm.201200348>.
7. Y. Cui, R. Deng, G. Wang, and D. Pan, *J. Mater. Chem.* 22, 23136 (2012). <https://doi.org/10.1039/C2JM33574C>.
8. P.S. Maldar, A.A. Mane, S.S. Nikam, S.D. Giri, A. Sarkar, and A.V. Moholkar, *J. Mater. Sci. Mater. Electron.* 28, 18891 (2017). <https://doi.org/10.1007/s10854-017-7842-1>.
9. C. An, K. Tang, G. Shen, C. Wang, L. Huang, and Y. Qian, *Mater. Res. Bull.* 38, 823 (2003). [https://doi.org/10.1016/S0255-5408\(03\)00046-1](https://doi.org/10.1016/S0255-5408(03)00046-1).
10. J.Y. Chane-Ching, A. Gillorin, O. Zaberca, A. Balocchi, and X. Marie, *Chem. Commun.* 47, 5229 (2011). <https://doi.org/10.1039/C1CC10749F>.
11. B. Murali, M. Madhuri, and S.B. Krupanidhi, *Cryst. Growth* 14, 3685 (2014). <https://doi.org/10.1021/cg500622f>.
12. A. Gupta, K. Mokurla, A. Kamble, S. Shankar, S. Mallick, and P. Bhargava, *AIP Conf. Proc.* 1665, 140022 (2015).
13. A.M. Mansour, I.S. Yahia, and I.M. El Radaf, *Mater. Res. Express* 5, 076406 (2018). <https://doi.org/10.1088/2053-1591/aad15b>.

14. P.S. Maldara, M.A. Gaikwada, A.A. Manea, S.S. Nikama, S.P. Desai, S.D. Girib, A. Sarkar, and A.V. Moholkara, *J. Solar Energy* 158, 89 (2017). <https://doi.org/10.1016/j.solene.2017.09.036>.
15. M. Krishnaiah, R.K. Mishra, S.G. Seo, S.H. Jin, and J.T. Park, *J. Alloys Compd.* 781, 1091 (2019). <https://doi.org/10.1016/j.jallcom.2018.12.036>.
16. M.S. El-Bana, I.M. El Radaf, S.S. Fouad, and G.B. Sakr, *J. Alloys Compd.* 705, 333 (2017). <https://doi.org/10.1016/j.jallcom.2017.02.106>.
17. I.M. El Radaf and R.M. Abdelhameed, *J. Alloys Compd.* 765, 1174 (2018). <https://doi.org/10.1016/j.jallcom.2018.06.277>.
18. I.M. El Radaf, S.S. Fouad, A.M. Ismail, and G.B. Sakr, *Mater. Res. Express* 5, 046406 (2018). <https://doi.org/10.1088/2053-1591/aaba0a>.
19. A. Ashery, A.A.M. Farag, and M. Zeama, *Superlattices Microstruct.* 66, 136 (2014). <https://doi.org/10.1016/j.spmi.2013.12.002>.
20. T.A. Hameed, I.M. El Radaf, and H.E. Elsayed-Ali, *J. Mater. Sci. Mater. Electron.* 29, 12584 (2018). <https://doi.org/10.1007/s10854-018-9375-7>.
21. M. Nasr, A.M. Mansour, and I.M. El Radaf, *Mater. Res. Express* 6, 036405 (2019). <https://doi.org/10.1088/2053-1591/aaf3f3>.
22. I.S. Yahia, A.A.M. Farag, F. Yakuphanoglu, and W.A. Farooq, *Synth. Met.* 161, 881 (2011). <https://doi.org/10.1016/j.synthmet.2011.02.016>.
23. R.K. Gupta, M.E. Aydın, and F. Yakuphanoglu, *Synth. Met.* 161, 2355 (2001). <https://doi.org/10.1016/j.synthmet.2011.09.002>.
24. A. Ashery, I.M. El-Radaf, and M.M.M. Elnasharty, *J. Silicon* 1876, 9918 (2018). <https://doi.org/10.1007/s12633-018-0047-2>.
25. I.M. El Radaf, M.S. Al-Kotb, M. Nasr, and I.S. Yahia, *J. Alloys Compd.* 788, 206 (2019). <https://doi.org/10.1016/j.jallcom.2019.02.189>.
26. A.A.M. Farag and I.S. Yahia, *Synth. Met.* 161, 32 (2011). <https://doi.org/10.1016/j.synthmet.2010.10.030>.
27. I.M. El Radaf, A.M. Mansour, and G.B. Sakr, *J. Semicond.* 39, 124010 (2018). <https://doi.org/10.1088/1674-4926/39/12/124010>.
28. İ. Taşcıoğlu, S.O. Tan, F. Yakuphanoğlu, and Ş. Altındal, *J. Electron. Mater.* 47, 6059 (2018). <https://doi.org/10.1007/s11664-018-6495-z>.
29. V. Ganesh, M.A. Manthrammel, M. Shkir, I.S. Yahia, H.Y. Zahran, F. Yakuphanoglu, and S. AlFaify, *Appl. Phys.* 124, 424 (2018). <https://doi.org/10.1007/s00339-018-1832-x>.
30. B. Saha, K. Sarkar, A. Bera, K. Deb, and R. Thapa, *Appl. Surf. Sci.* 418, 328 (2017). <https://doi.org/10.1016/j.apsusc.2017.01.142>.
31. G.K. Rao, *Appl. Phys. A* 224, 1 (2017). <https://doi.org/10.1007/s00339-017-0850-4>.
32. A.A.M. Farag, I.S. Yahia, and M. Fadel, *Int. J. Hydrog. Eng.* 34, 4906 (2009). <https://doi.org/10.1016/j.ijhydene.2009.03.034>.
33. A.B. Uluşan, A. Tataroglu, Y. Azizian-Kalandaragh, and S. Altind, *Mater. Sci. Mater. Electron.* 29, 159 (2018). <https://doi.org/10.1007/s10854-017-7900-8>.
34. I.M. El Radaf, M. Nasr, and A.M. Mansour, *Mater. Res. Express* 5, 015904 (2018). <https://doi.org/10.1088/2053-1591/aaa25e>.
35. A.A.M. Farag, F.S. Terra, A. Ashery, and A.M. Mansour, *J. Alloys Compd.* 615, 604 (2014). <https://doi.org/10.1016/j.jallcom.2014.06.058>.
36. I.M. El Radaf, T.A. Hamid, and I.S. Yahia, *J. Mater. Res. Express* 5, 066416 (2018). <https://doi.org/10.1088/2053-1591/aaca7b>.
37. M. Nasr, I.M. El Radaf, and A.M. Mansour, *J. Phys. Chem. Solids* 115, 283 (2018). <https://doi.org/10.1016/j.jpcs.2017.12.029>.

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