Experimental Study on Electrical Conductivity of MnO-CaO-SiO₂ Slags at 1723 K to 1823 K (1450 °C to 1550 °C) and Various Oxygen Potentials



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The electrical conductivity of molten slag has many important and practical effects in modeling and operating the electric smelting furnace. In the present study, the electrical conductivities (total and electronic/ionic properties) of MnO-CaO-SiO₂ slags were measured by a four-electrode method at different oxygen potentials and temperatures. Experimental results show that the effects of temperature on the total, electronic, and ionic conductivities obey the Arrhenius law, and all conductivities increase when increasing the temperature. The stepped potential chronoamperometry method was used to measure the electronic transference number, which is affected strongly by oxygen potential but is unaffected by temperature. The total electrical, electronic, and ionic conductivities present similar increasing trends when increasing the CO/CO₂ ratio, which resulted from increasing Mn²⁺.

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I. INTRODUCTION

AS one of the most basic physical-chemical properties, the electrical conductivity of molten slag has many important and practical effects. On one hand, the electrical conductivity of molten slag plays a prominent role in modeling and operating the electric smelting furnace and optimizing the metallurgical process^[1-6]; on the other hand, it is important for understanding the structure of molten slags.^[7-12] For instance, a large number of metals, including titanium, aluminum, and magnesium, have been produced by the electrolysis of molten slags and salts, and the earliest evidence of the ionic structure of oxide melts resulted from electrical conductance measurements. For the preceding aspects, the study of electrical properties of metallurgical melts is particularly necessary and important. Molten oxide electrolysis^[13–16] is a carbon-neutral electrochemical technique, which can decompose metal oxide directly into liquid metal and oxygen gas. Compared to other technologies, its greatest advantage is no greenhouse gas emissions. As an efficient electrolysis, the melt must be

predominantly an ionic conductor. Knowledge of the electrical conductivity of slag can help in the design and selection of the proper slag for the electrolysis process. So far, the electrical conductivities of abundant different slags have been reported.^[10,17–22] High melting point and the charge transfer between the slag and gas make it hard for the conductivity measurement of MnO-bearing slags, which leads to a lack of data of electrical conductivity of the present work was to experimentally study the electrical conductivity of MnO-CaO-SiO₂ slags at different oxygen potentials controlled by CO-CO₂ gas mixtures in the temperature range of 1723 K to 1823 K (1450 °C to 1550 °C).

II. EXPERIMENTS

The initial slag compositions are provided in Table I. In each group, the molar ratio of CaO and SiO₂ remains constant (CaO:SiO₂ = 1:1), but the content of MnO gradually increases. Slag samples were prepared using reagent grade SiO₂ and CaCO₃ powders (analytically pure, Sinopharm Chemical Reagent Co., Ltd., Beijing, China), both of which were calcined at 1273 K (1000 °C) for 10 hours in a muffle furnace to decompose any carbonate and hydroxide before being used. Then, about 12 g mixtures were precisely weighted according to the compositions shown in Table I and mixed in an agate mortar thoroughly to ensure the components were uniform. In the present study, a four-terminal method was employed to accomplish the electrical conductivity measurement, and this method had already been

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successfully used to measure the electrical conductivity of molten slags.^[17,22] In this way, the effects of polarization at the current-carrying electrodes can be largely eliminated and the effect of alternative paths is reduced. Due to the necessity of calibration, determinations of the cell constant *C* and resistance R_x are obligatory to the measurement accuracy. The relation among the total electrical conductivity σ_t , constant *C*, and resistance R_x can be expressed by Eq. [1]:

$$\sigma_{\rm t} = C/R_{\rm x}.$$
 [1]

MnO-bearing slags can be regarded as mixed conductors, since they exhibit both ionic and electronic conductivity (σ_i and σ_e). The electronic transference numbers (the ratio of electronic conductivity to the total conductivity) were measured by the stepped potential chronoamperometry method.^[17] The electronic transference number is defined as

 $t_{\rm e} = i_{\rm e}/(i_{\rm e}+i_{\rm i})$

or

$$t_{\rm e} = \sigma_{\rm e}/(\sigma_{\rm e} + \sigma_{\rm i}),$$
 [3]

[2]

where i_e and i_i are the currents carried by the electronic and ionic charge carriers, respectively. Therefore, the electronic conductivity and ionic conductivity can be calculated by total electrical conductivity and electronic transference number.

The descriptions of the device and experimental principle have been described already in detail in our previous study.^[23–25] In the experimental process, the oxygen partial pressure was controlled by the ratio of CO to CO₂, whose gas rate was controlled by a flowmeter. The total flow rate of CO₂ and CO was fixed at 200 mL min⁻¹. The slag was kept for 2 to 3 hours in each atmosphere for the purpose of equilibrium and uniformity of slag. Once the slag and gas reached equilibrium, the electrical measurements were carried out at every 25 K interval on cooling from 1823 K (1550 °C). All the measurements were completed using a CHI 660a electrochemical workstation (Shanghai Chenhua Instrument Co., Ltd., Shanghai, China).

III. RESULTS AND DISCUSSION

A. Influence of Temperature

The Arrhenius law can be used to express the effect of temperature on electrical conductivity:

$$\sigma = A \exp(-E/RT)$$
 [4]

or

$$\ln \sigma = \ln A - E/RT, \qquad [5]$$

where σ is the electrical conductivity, Ω^{-1} cm⁻¹; A is the pre-exponent term; E is the activation energy, J (mol K)⁻¹; R is the gas constant, 8.314 J (mol K)⁻¹; and T is the absolute temperature, in Kelvin. The changes of total electrical conductivity as functions of temperature for different slags at $CO/CO_2 = 1$ are shown in Figure 1. As seen in this figure, the temperature dependence of total electrical conductivity obeys the Arrhenius law well; furthermore, electrical conductivity increases when increasing the temperature, since the charge transport rate is faster at the higher temperature.

The electronic transference number was measured under conditions identical to those of the electrical conductivity measurements. The results are provided in

 Table I.
 Composition of Slag Samples (Mole Percent)

| MnO | CaO | SiO ₂ |
|-----|------|------------------|
| 5 | 47.5 | 47.5 |
| 10 | 45 | 45 |
| 20 | 40 | 40 |
| 30 | 35 | 35 |
| | | |



Fig. 1—Changes of total electrical conductivity as a function of temperature for different slags at $CO/CO_2 = 1$.



Fig. 2-Effect of temperature on the electronic transference number.

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Fig. 3—Effect of temperature on the ionic conductivity at CO/ CO_2 = 1.

Figure 2, from which it can be seen that under the present experimental conditions, the electronic transference number is essentially independent of the temperature. The negligible effect of temperature on the transference number has also been reported by other authors.^[18,26] Furthermore, it is obvious from Figure 2 that when increasing the MnO content, there is an increase of the electronic transference number.

The effect of temperature on the ionic conductivity and electronic conductivity is shown in Figures 3 and 4, respectively. Similar to the case of total electrical conductivity, the ionic conductivity and electronic conductivity increase when increasing the temperature, and the relationships between temperature and both ionic and electronic conductivities also follow the Arrhenius law. The positive effect of temperature on ionic conductivity can be related to the increased mobility of cations at higher temperature, due both to a greater diffusion coefficient and to more depolymerization of the silicate structure.

B. Influence of Oxygen Potential

The partial pressure of oxygen P_{O_2} can be controlled by CO-CO₂ gas mixtures according to Eq. [6]:

$$CO(g) + 1/2O_2(g) = CO_2(g),$$
 [6]

$$\Delta G^{\theta}(\mathbf{J}) = -279,710 + 84.08T = -RT \ln\left(\frac{P_{\rm CO_2}}{P_{\rm CO}} \left(\frac{P^{\theta}}{P_{\rm O_2}}\right)^{0.5}\right)$$

Figure 5 shows the total electrical conductivity for different compositions at 1823 K (1550 °C) as a function of CO/CO₂ ratio. Figure 5 shows that for all experimental slags, the total electrical conductivities increase when increasing the CO/CO₂ ratio in fixed MnO content, and in the case of fixed CO/CO₂ ratio, the higher the MnO content is, the higher the total electrical conductivities will be.



Fig. 4—Effect of temperature on the electronic conductivity at CO/ CO_2 = 1.



Fig. 5—Total electrical conductivity for different compositions at 1823 K (1550 $^{\circ}\rm{C})$ as a function of CO/CO₂ ratio.

The change of the electronic transference number as different CO/CO_2 ratios is shown in Figure 6. From Figure 6, under the current experimental conditions, the electronic transference numbers vary from about 21 to 50 pct and always increase when increasing the MnO content and CO/CO_2 ratio.

The change of the ionic conductivity at 1823 K (1550 °C) as different CO/CO₂ ratios is shown in Figure 7. From Figure 7, for all experimental slags, the ionic conductivity increases when increasing the CO/CO₂ ratio. There are several ions in the present MnO-CaO-SiO₂ slags, but Mn^{2+} and Ca^{2+} ions are the most important charge carriers. The reaction between manganese ions and gas is shown as follows:

$$Mn^{2+} + 3O^{2-} + 1/2O_2 = (MnO_4)^{4-}.$$
 [7]

The existence form of ${Mn}^{4+}$ is a much weaker mobility compared with ${Mn}^{2+}$ ion. On the basis of



Fig. 6—Electronic transference number for different slags at 1823 K (1550 °C) as a function of CO/CO_2 ratio.



Fig. 7—Ionic conductivity at 1823 K (1550 °C) as a function of CO/ CO_2 ratio.

Eq. [7], more and more Mn^{2+} will replace Mn^{4+} when increasing the CO/CO₂ ratio, which will lead to an increase of the ionic conductivity.

Figure 8 shows the effect of the CO/CO_2 ratio on electronic conductivity at 1823 K (1550 °C). From Figure 8, for slags with various MnO contents, the electronic conductivity increases when increasing the CO/CO_2 ratio, and in the case of the fixed CO/CO_2 ratio, the electronic conductivity increases when increasing the MnO content. Barati and Coley proposed a diffusion-assisted charge transfer model that has already successfully explained the experimental phenomenon of electronic conductivity for CaO-SiO₂-FeO_x slags.^[27]

The variation phenomenon of the present electronic conductivity can be explained by the diffusion-assisted charge transfer model. This model requires that the electron hopping between Mn^{2+} and Mn^{4+} ions takes place when they are at appropriate distance. That is to say, the ions that are far apart from each other first



Fig. 8—Electronic conductivity at 1823 K (1550 $^{\circ}\text{C})$ as a function of CO/CO_2 ratio.



ions travel to reach separation distances sufficiently short for electron hopping



travel to reach separation distances sufficiently short, and then the activated charge hopping between Mn^{2+} and Mn⁴⁺ ions can take place. The schematic diagram of the diffusion-assisted charge transfer model is shown in Figure 9. According to the diffusion-assisted charge transfer model, when the slag composition, temperature, and oxygen partial pressure are constant, the electronic conductivity is related to the concentration of Mn^{2} and Mn⁴⁺; therefore, the expression of electronic conductivity can be simplified and shown as $\sigma_{\rm e} = k c_{({\rm Mn}^{2+})} c_{({\rm Mn}^{4+})}$; in other words, the electronic conductivity is proportional to the product of the concentration of Mn^{2+} and Mn^{4+} . When the content of manganese oxide is fixed, the amounts of Mn^{2+} and Mn^{4+} are constant, that is, $\frac{c_{(Mn^{2+})}}{c_{(Mn0)}} + \frac{c_{(Mn^{4+})}}{c_{(Mn0)}} = 1$, and the product of the concentration of Mn^{2+} and Mn^{4+} will be maximal when $c_{(Mn^{2+})} = c_{(Mn^{4+})}$. According to Eqs. [6] and [7], when increasing the CO/CO_2 ratio, the concentration of Mn²⁺ will increase and the concentration of $(MnO_4)^{4-}$ will decrease, which leads to the value of the electronic conductivity increasing first and then decreasing; there should be a maximum value when $c_{(Mn^{2+})} = c_{(Mn^{4+})}$. However, the present experimental results show that the electronic conductivity increases monotonously when increasing the CO/CO₂ ratio, which may result from the concentration of Mn^2 always being less than that of Mn⁴⁺ in current experimental conditions, even in the largest CO/CO₂ ratio, just like Figure 8. In addition, it also can be seen from Figure 8 that the higher the MnO content, the higher the electronic conductivity will be. When increasing the MnO content, the concentration of Mn^{2+} and Mn^{4+} will increase, which results in larger contact probability between Mn^{2+} and Mn^{4+} , eventually leading to the increase of the electronic conductivity.

IV. CONCLUSIONS

- 1. The effects of temperature on the electrical conductivity of MnO-SiO₂-CaO slags obey the Arrhenius law; when increasing the temperature, all the electrical conductivities increase.
- 2. The electronic transference number is affected strongly by oxygen potential but is unaffected by temperature.
- 3. When increasing the CO/CO_2 ratio, the ionic conductivity exhibits the increasing trend because of a higher concentration of Mn^{2+} .
- In the current experimental conditions, the electronic conductivity increases when increasing the ratio of CO/CO₂. The effect of CO/CO₂ on electronic con-

ductivity is suggestive of a small polaron-hopping mechanism between Mn^{2+} and Mn^{4+} ions.

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