Study of the Viscosity of Mold Flux Based on the Vogel–Fulcher–Tammann (VFT) Model



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Viscosity is one of the most important properties of mold flux and affects the process of continuous casting significantly. In order to describe the variation of viscosity of mold flux accurately in a wide range of temperature occurring in the casting mold, a non-Arrhenius Vogel–Fulcher–Tammann (VFT) model was adopted in this study. The results showed that the adjusted coefficient of determination (Adj. R^2) of non-Arrhenius VFT Model ranges from 0.92 to 0.96, which suggests this model could be well adapted to predict the relationship between viscosity and temperature of mold flux. The temperature at which viscosity becomes infinite, $T_{\rm VFT}$, increased with the addition of Cr_2O_3 and improvement of basicity, while it decreased with the addition of B_2O_3 , as it was determined by both the degree of polymerization of the melt structure and crystallization behavior of the melt. Also, the pseudo-activation energy, $E_{\rm VFT}$, of Samples 1 to 5 was 60.1 ± 3.6 , 94.7 ± 14.9 , 101.7 ± 19.0 , 38.0 ± 4.8 , and 32.4 ± 4.0 kJ/mol, respectively; it increased with the addition of Cr_2O_3 and B_2O_3 , but deceased with the increase of basicity.

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I. INTRODUCTION

THE mold flux plays important roles in the process of continuous casting, such as the lubrication of shell during the mold oscillation, in-mold heat transfer control, and adsorption of inclusions on top of molten steel, which are greatly influenced by the viscosity of mold flux. Mold flux with improper viscosity would cause lots of problems, for example, the presence of sever oscillation marks, surface cracks, breakouts,^[1,2] the entrapment of slag on top of molten steel,^[3] the erosion of the nozzle in the slag layer,^[4] and the formation of large slag rim in the vicinity of meniscus.^[5] Therefore, it is significantly important to describe the viscosity of mold flux accurately during the process of continuous casting.

A variety of approaches and models have been used to describe the viscosity of liquid mold flux as a function of temperature and chemical composition. Among them, the temperature-dependent viscosity models include empirical,^[6] Arrhenius,^[7,8] and Weymann–Frenkel mod-el,^[9]while the composition-dependent viscosity models include quasi-structural,^[10] basicity index,^[11,12] and optical basicity model.^[13,14] However, most of those models are limited to a relatively small range of temperature and viscosity. The mold flux in continuous

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casting mold experiences a wide temperature gradient ranging from more than 1773 K (1500 °C) to room temperature; as the temperature of liquid mold flux is close to the molten steel when it contacts with molten steel; while the temperature decreases to the break temperature after it infiltrates into the gap between mold wall and shell; and then it further decreases to room temperature when the mold flux exists out with the slab from the bottom of mold. Thus, the viscosity of mold flux shows a strong function of temperature.

The Vogel–Fulcher–Tammann (VFT) model was proposed independently by Vogel (in 1921), Fulcher (in 1925), and Tammann (in 1926),^[15–17] which is one of the most widely used non-Arrhenius temperature-dependent viscosity model. The VFT relation has been found to be a good prediction model for many classes of materials, and has been cited in hundreds of articles. For example, Giordano, *et al.*^[18,19] used the VFT model to study the rheology of magma. Research works from Angel, *et al.*^[20] suggested that the viscosity of glassy alloys shows perfect VFT equation behavior. Lu, *et al.*^[21] found that the VFT model was able to provide an accurate prediction for the temperature-dependent relaxation behavior of shape-memory polymers. Mokhtarani, *et al.*^[22] also successfully fitted the experimental viscosity of pure and binary ionic liquids using the VFT viscosity model.

Although the VFT model has been widely used to characterize the viscosity of many materials, it has never been adopted to study the rheological property of mold flux. Therefore, in this paper, the viscosity of mold fluxes at the temperature ranging from 1200 K (927 °C) to 1573 K (1300 °C) was measured firstly; then, the relationship between viscosity and temperature was analyzed by both Arrhenius model and non-Arrhenius

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VFT model; finally, the effects of mold flux components on the VFT temperature and pseudo-activation energy were also discussed.

II. EXPERIMENT METHOD

A. The Raw Materials

The designed mold fluxes in this study are listed in Table I. Among them, Sample 1 is a decarburized commercial mold flux for the casting of low-carbon steel by placing it into a programmable furnace at 1073 K (800 °C) for 24 hours. Samples 2 to 5 were prepared by adding different amount of reagent grade chemicals of CaCO₃, SiO₂, Al₂O₃, MnCO₃, Na₂CO₃, Li₂CO₃, CaF₂, Cr₂O₃, and B₂O₃ (Supplier: Fine chemical engineering and technology research and development center, Guangdong, China) into Sample 1 to adjust their compositions. All samples were stirred in a blender for 120 minutes to homogenize their compositions before the viscosity test.

B. Experiment Apparatus and Process

The viscosity measurements were carried out using a Brookfield DV-II + viscometer (Brookfield Inc.) through the rotating cylinder method, which is schematically shown in Figure 1. A calibration measurement was carried out at room temperature using stand oil with known viscosity.^[23]

When measuring the viscosity of the designed mold flux, 250 g of the sample powders was firstly placed in a graphite crucible with a diameter and an internal height of 50 and 80 mm, respectively (Table II). Second, the crucible was heated to 1773 K (1500 °C) and held for 10 minutes to obtain a homogeneous melt in an electric resistance furnace with $MoSi_2$ as the heating element. Then, the melt was cooled to the target temperature. After that, a bob, which is made of molybdenum with the height of 18 mm and the diameter of 15 mm (Table II), was immersed into liquid slag bath and rotated to obtain the value of viscosity at the target temperature.

The composition of mold fluxes after viscosity measurement tests was also analyzed by the X-ray fluoroscopy (XRF, S4Pioneer; Bruker AXS; GmbH Karlsruhe, Germany) and inductively coupled plasma mass spectrometry (ICP, SPECTRO, Germany). The results are shown in Table III. It could be found that the evaporative loss of mold flux components is relatively small, and the influence caused by the evaporation can be ignored, which is consistent with our previous study.^[24]

In order to investigate the precipitated phase of mold fluxes after the viscosity tests, parts of mold fluxes after the viscosity measurements at the target temperature of 1573 K (1300 °C) were obtained by a molybdenum spoon and quenched in water. Then, those quenched mold fluxes were sampled and observed through scanning electron microscope (Japanese Electronics Company JSM-6360LV) with an acceleration voltage of 20 kW and 200 times magnification.

C. The Vogel–Fulcher–Tammann (VFT) Non-Arrhenius Model

The generic relation for the Vogel–Fulcher–Tammann (VFT) model is as following:

$$\log \eta = A_{\rm VFT} + \frac{E_{\rm VFT}}{T - T_{\rm VFT}},$$
[1]

where, η (Pa·s) is the viscosity of mold flux, and the adjustable parameters $A_{\rm VFT}$, $E_{\rm VFT}$, and $T_{\rm VFT}$ are dependent on the composition of mold flux. The parameter $A_{\rm VFT}$ is the value of log η at infinite temperature, and $T_{\rm VFT}$ is the temperature (K) at which viscosity becomes infinite. The parameter $E_{\rm VFT}$ corresponds to the pseudo-activation energy associated with viscous flow, and is thought to represent a potential energy barrier obstructing the structural rearrangement of the melt.

The quality of regressions can be measured by the parameter Adj. R^2 , which is a value between 0 and 1. Generally speaking, if it is close to 1, it means that the predicted values are very close to the measured ones. The Adj. R^2 can be written as

$$\overline{R}^2 = 1 - \frac{(1 - R^2)(n - 1)}{n - p - 1},$$
[2]

where p is the total number of explanatory variables in the model, and n is the sample size. R^2 is the coefficient of determination that can be calculated as

$$R^2 = 1 - \frac{SS_{\rm res}}{SS_{\rm tot}},\tag{3}$$

where SS_{res} is the regression sum of the square measured deviation, and SS_{tot} is the total sum of the square of the predicted deviations. They can be written as

Table I. The Chemical Composition of Mold Fluxes with Different Content (Weight Percent)

Samples	SiO ₂	CaO	Al_2O_3	$(Na_2O + Li_2O)$	F	MnO	Cr_2O_3	B_2O_3	R^*
1	33.78	32.5	3.51	14.56	12.94	0	0	0	0.96
2	32.21	30.92	3.51	14.56	12.94	1.1	2.1	0	0.96
3	30.68	29.45	3.51	14.56	12.94	1.1	2.1	3	0.96
4	29.27	30.86	3.51	14.56	12.94	1.1	2.1	3	1.05
5	27.94	32.14	3.51	14.56	12.94	1.1	2.1	3	1.15
* R is the	basicity = C	aO/SiO.							



Fig. 1-Schematic figure of viscometer.

Fig. 2—The variation of viscosity with temperature of Sample 1.

Table II. Dimension and Material of Crucible and Bob

	Crucible		Bob			
Graphite	Inner diameter Height in total Wall thickness	50 mm 80 mm 10	molybdenum 	diameter height angle for the tapers	15 mm 18 mm deg	

Table III. The Chemical Composition of Mold Fluxes After the Viscosity Tests (Weight Percent)

Samples	SiO ₂	CaO	Al ₂ O ₃	$(Na_2O + Li_2O)$	F	MnO	Cr ₂ O ₃	B_2O_3	R
1	33.88	32.71	3.72	14.25	12.74	0	0	0	0.96
2	32.35	31.13	3.68	14.24	12.72	1.11	2.12	0	0.96
3	30.79	29.66	3.71	14.22	12.71	1.12	2.12	3.00	0.96
4	29.37	30.92	3.62	14.29	12.81	1.11	2.11	3.01	1.05
5	27.99	32.19	3.61	14.35	12.84	1.10	2.10	3.01	1.15

$$SS_{\text{tot}} = \sum_{i} (y_i - \overline{y})^2, \qquad [4]$$

$$SS_{\rm reg} = \sum_{i} (f_i - \overline{y})^2, \qquad [5]$$

where y_i is the measured values of $\log \eta_i$, f_i is the predicted value of $\log \eta_i$, and $\overline{y} = \frac{1}{n} \sum_{i=1}^{n} y_i$.

III. RESULTS AND DISCUSSIONS

A. Comparison of the Arrhenius and the Non-Arrhenius VFT Model

Figure 2 shows the variation of viscosity of Sample 1 vs temperature, which is a typical temperature–viscosity curve of $\log \eta \ vs \ 1/T$ for the mold flux in a wide temperature range from 1200 K (927 °C) to 1573 K (1300 °C). According to the suggestion by Tweer,^[25] the viscosity in Figure 2 can be divided into three zones. The first one is the high-temperature Arrhenius zone, where the viscosity of Sample 1 increased with temperature slowly and exhibited an Arrhenius behavior. The second one is the intermediate-temperature zone, in which the viscosity departures from the inverse temperature dependence near the break temperature and starts to show non-Arrhenius characteristics. The viscosity in the third zone returns to the Arrhenius behavior again, although the viscosity increased rapidly, which is called low-temperature Arrhenius zone. Due to the different behavior of viscosity in different temperature zone, the overall variation of viscosity with temperature for Sample 1 shows non-Arrhenius behavior in the whole temperature range. Besides, the beak temperature of mold flux, at which the viscosity increases rapidly, can also be obtained from the curve of $\log \eta vs 1/T$, and it was about 1302 K (1029 °C) for Sample 1 as shown in Figure 2.

The measured values for the viscosity and temperature of mold flux Samples 1 to 5 during the tests were fitted by Arrhenius ($\log \eta = A_{VFT} + \frac{E_{VFT}}{T - T_{VFT}}$) and non-Arrhenius VFT models, and are shown in Figures 3(a) and (b),^[26] respectively. It can be seen intuitively in Arrhenius model, as shown in Figure 3(a), that the



Fig. 3—The fitting of viscosity vs temperature of mold fluxes by using Arrhenius and VFT models. (a) Results from Arrhenius model, (b) Results from VFT model (This figure is being adapted from Ref. [26]).

Table IV. The Adj. R^2 of the Arrhenius and Non-Arrhenius VFT Model

Samples	1	2	3	4	5
Arrhenius Model	0.57	0.62	0.80	0.67	0.65
Non-Arrhenius VFT Model	0.96	0.94	0.92	0.93	0.93



Fig. 4—The $T_{\rm VFT}$ and $T_{\rm br}$ temperature of mold fluxes Sample 1 to 5.

deviation between the measured data and the fitting ones is obvious, especially in the low-temperature region, which confirmed that the viscosity of mold flux is not Arrhenius dependent, especially in the low-temperature range. The reason for that is mainly due to the fact that the Arrhenius model is linear related; however, the viscosity of mold flux increases slowly in the high-temperature Arrhenius zone, while it increases sharply in the low-temperature Arrhenius zone, which indicates the relation of log η vs T is obviously non-linear related in the whole temperature range. However, the measured data distribute along the fitting line evenly in the VFT model as shown in Figure 3(b), which suggests that the VFT model can be used to well describe the relationship between viscosity and temperature of mold flux. In order to quantify the quality of linear regression by Arrhenius and VFT models, the adjusted coefficient of determination (Adj. R^2) was calculated, and is shown in Table IV. The values for Adj. R^2 of non-Arrhenius VFT Model were from 0.92 to 0.96 that is greatly larger than those of Arrhenius model ranging from 0.57 to 0.80. Therefore, the non-Arrhenius VFT is better for the description of the relationship between viscosity and temperature of mold flux in a wide temperature range.

B. Effect of Components on the VFT Temperature of Mold Flux

Through the fitting process of viscosity and temperature using the VFT model, the parameter $T_{\rm VFT}$ (VFT temperature) can be obtained. According to the Eq. [1], the T_{VFT} presents the temperature (K) at which viscosity becomes infinite. So, the $T_{\rm VFT}$ can be considered as another key benchmark to characterize the lubrication ability of mold flux besides the break temperature $(T_{\rm br})$ and glass transition temperature (T_g) , as all of them indicate that the mold flux loses flowability when the temperature becomes low. The $T_{\rm VFT}$ and $T_{\rm br}$ for mold flux Samples 1 to 5 are shown in Figure 4, where the $T_{\rm VFT}$ of Samples 1 to 5 are 1242 K (969 °C), 1250 K (977 °C), 1165 K (892 °C), 1212 K (939 °C), and 1240 K (967 °C), while the $T_{\rm br}$ are 1302 K (1029 °C), (1029 °C), 1330 K (1057 °C), 1224 K (951 °C), 1246 K (973 °C), and 1268 K (995 °C), respectively. It can be found that the $T_{\rm br}$ of mold flux are apparently higher than $T_{\rm VFT}$. The main reason is that the break temperature is obtained at the temperature where the viscosity of mold flux starts to increase rapidly, but the VFT temperature is corresponding to the temperature at which the



Fig. 5—SEM of mold flux samples after viscosity test. (a) Sample 1, (b) Sample 2, (c) Sample 3, (d) Sample 4, and (e) Sample 5.

viscosity becomes infinite. Besides, the viscosity of mold flux always increases with the reduction in temperature. So, the $T_{\rm VFT}$ should be lower than $T_{\rm br}$. Although the variation trend of $T_{\rm VFT}$ and $T_{\rm br}$ is consistent with each other as shown in Figure 4, it is convenient to use the $T_{\rm VFT}$ to describe mold flux lubrication ability, as it can be calculated directly and accurately from the VFT models based on the viscosity and temperature data from the measurements; however, it may introduce errors when obtaining $T_{\rm br}$ due to the arbitrary estimation on the slope of viscosity–temperature curve.

Besides, the VFT temperature of Samples 1 to 5 is different from each other, which suggests that the $T_{\rm VFT}$ of mold flux is dependent on its individual chemical composition. Comparing Sample 1 with Sample 2, the

 $T_{\rm VFT}$ increased from 1242 K (969 °C) to 1250 K (977 °C), which attributes to the addition of 1.1 pct MnO and 2.1 pct Cr₂O₃ in the mold flux Sample 2. Although MnO is considered to be a network modifier in molten slag system to lower the degree of polymerization of the melt, the Cr₂O₃ would expect to behave as a network former and increase the viscosity of the mold flux greatly according to our previous paper.^[27] Thus, the temperature for the viscosity to become infinity would be improved and lead to the increase of $T_{\rm VFT}$. The $T_{\rm VFT}$ of Sample 3 deceased a lot, from 1250 K (977 °C) of Sample 2 to 1165 K (892 °C), with the further addition of 3 pct B₂O₃. It is resulted from that the B₂O₃ is a typical effective fluxing agent with a low melting point,^[28,29] which can lower the melting



Fig. 6—The pseudo-activation energy E_{VFT} of mold fluxes.

temperature of mold flux system greatly through the formation of low melting point substances with other components. For Samples 4 and 5, the $T_{\rm VFT}$ increases continuously to 1212 K (939 °C) and 1240 K (967 °C), respectively, as the basicity (R) is improved from 0.96 to 1.05 and 1.15, which leads to the enhancement of crystallization of mold flux.^[23,24] Thus, the precipitation temperature of crystal would be high, then the $T_{\rm VFT}$ of mold flux increased. The SEM photos of the above 5 mold flux samples after the viscosity testes are shown in Figure 5, where the crystals are formed and become very obvious especially in the Samples 4 and 5, which suggests the $T_{\rm VFT}$ is also affected by the crystallization of mold flux.

C. Effect of Components on the Pseudo-activation Energy of Mold Flux

The parameter $E_{\rm VFT}$ in Eq. [1] is the pseudo-activation energy, representing the energy barrier that the ion clusters should overcome during the transport process in the melt. The $E_{\rm VFT}$ of the above 5 mold flux samples are calculated and shown in Figure 6. They were 60.1 ± 3.6 , 94.7 ± 14.9 , 101.7 ± 19.0 , 38.0 ± 4.8 , and 32.4 ± 4.0 kJ/mol for Sample 1 to 5, respectively. It can be seen from Figure 6 that the $E_{\rm VFT}$ is also varied with the mold flux composition. It increases from $60.1 \pm 3.6 \text{ kJ/mol}$ (Sample 1) to $94.7 \pm 14.9 \text{ kJ/mol}$ (Sample 2), and then to $101.7 \pm 19.0 \text{ kJ/mol}$ (Sample 3) with the addition of MnO and Cr_2O_3 in Sample 2, as well as the addition of B_2O_3 in Sample 3. The main reason is Cr_2O_3 and B_2O_3 are network formers, which make the ion clusters in the molten mold flux larger and more complex; thus, the energy barrier for the ion clusters to transport increases. On the contrary, the increase of basicity for Samples 4 and 5 would provide more O^{2-} to break the bond of Si-Si, and simplify silicate structure leading the movement of ion clusters in molten mold flux easier; thus, the $E_{\rm VFT}$ for above samples was reduced.

IV. CONCLUSIONS

The rheological behavior of mold flux was investigated in this paper using the non-Arrhenius temperature-dependent Vogel–Fulcher–Tammann (VFT) model, and the specific important conclusions were summarized as follows:

- (1) The Adj. R^2 of non-Arrhenius VFT Model ranges from 0.92 to 0.96, while those values for Arrhenius model range from 0.57 to 0.80, which suggests that the non-Arrhenius VFT model is better for the description of the relationship between viscosity and temperature of mold flux in a wide range of temperature.
- (2) The $T_{\rm VFT}$ of Samples 1 to 5 are 1242 K (969 °C), 1250 K (977 °C), 1165 K (892 °C), 1212 K (939 °C), and 1240 K (967 °C), respectively, which are apparently lower than the $T_{\rm br}$. It is convenient to use the $T_{\rm VFT}$ to describe mold flux lubrication ability, as it can be obtained directly and accurately from the VFT model based on the measured viscosity and temperature data.
- (3) The T_{VFT} of mold flux increased with the increase of Cr_2O_3 and basicity, while it decreased with the addition of B_2O_3 , as it was determined by both the degree of polymerization of the melt structure and crystallization behavior of the melt.
- (4) The $E_{\rm VFT}$ of the five mold flux samples were 60.1 ± 3.6, 94.7 ± 14.9, 101.7 ± 19.0, 38.0 ± 4.8, and 32.4 ± 4.0 kJ/mol for Sample 1 to 5, respectively. It increased with the addition of Cr₂O₃ and B₂O₃, but deceased with the increase of basicity.

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