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# Visible Light-induced Cr-doped SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub> Composite for Improved Photocatalytic Performance

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> **Abstract:** Novel visible light-induced Cr-doped  $SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub>$  composite photocatalysts were synthesized by introducing polymeric  $g$ -C<sub>3</sub>N<sub>4</sub>. The composite photocatalyst was characterized by X-ray diffraction (XRD), high-resolution transmission electron microscopy (HRTEM), Fourier transform infrared (FT-IR) spectroscopy, UV-vis diffuse reflection spectroscopy, photoluminescence (PL) spectroscopy and BET surface area measurements. The photocatalytic oxidation ability of the novel composite photocatalyst was evaluated using methyl orange (MO) as a target pollutant. The photocatalysts exhibited a significantly enhanced photocatalytic performance in degrading MO. The optimal  $g - C_3 N_4$  content for the photodegradation activity of the composite photocatalysts was determined. The as-prepared composite photocatalyst exhibits an improved photocatalytic activity due to enhancement of photo-generated electron-hole separation at the interface.

**Key words:** photocatalysis;  $g - C_3 N_4$ ; Cr-doped SrTiO<sub>3</sub>; composite

## 1 Introduction

 Photocatalysis has received considerable interest for application in removal of pollutant and in water splitting for hydrogen or oxygen generation by using solar energy. However, to realize the photocatalysis technique for large-scale industrial applications, highly efficient photocatalysts are still most important<sup>[1-2]</sup>. As one of the promising photocatalytic candidates, SrTiO<sub>3</sub> has attracted more and more attention due to its excellent photocatalytic performance<sup>[3]</sup>. However, due to the relatively large band gap  $(3.2 \text{ eV})$ , SrTiO<sub>3</sub> only responds to the UV light. To decrease its band gap, doping of foreign elements into  $SrTiO<sub>3</sub>$  has been reported<sup>[4]</sup>. In particular, chromium (Cr) doping has attracted great interest because the occupied  $Cr<sup>3+</sup>$ level is usually 1.0 eV higher than the valence band top formed by O 2p orbital or 2.2 eV lower than the

conduction band bottom formed by Ti 3d orbital, which means that the band gap of  $SrTiO<sub>3</sub>$  can be narrowed and the visible light response can be achieved through the Cr doping<sup>[5,6]</sup>.

To increase the separation efficiency of photogenerated electron-hole pairs is usually a good method to improve the photocatalytic activities, such as the use of co-catalysts<sup>[7]</sup>. Another feasible route is to form a composite photocatalyst between two kinds of semiconductors, and the suitably matching band level of conduction and valance bands can induce improved photocatalytic activity<sup>[8]</sup>. Polymer photocatalyst, graphitic carbon nitride (g- $C_3N_4$ ) which has been used for hydrogen or oxygen generation from water splitting under visible light irradiation has been reported by Wang *et al* for the first time<sup>[9]</sup>. Very recently, some composite photocatalysts, such as  $TiO<sub>2</sub>$  $g - C_3 N_4^{[10]}$ ,  $C_3 N_4$ -TaON<sup>[11]</sup>,  $g - C_3 N_4/Bi_2 W O_6^{[12]}$ , graphene/  $C_3N_4^{[13,14]}$ ,  $C_3N_4/BiPO_4^{[15]}$ , MWNTs/g- $C_3N_4^{[16]}$ , g- $C_3N_4$ - $SrTiO<sub>3</sub>:Rh<sup>[17]</sup>, g-C<sub>3</sub>N<sub>4</sub>/ZnO<sup>[18]</sup>, C<sub>3</sub>N<sub>4</sub>/ZnWO<sub>4</sub><sup>[19]</sup> and$ BiOBr-C<sub>3</sub>N<sub>4</sub><sup>[20]</sup>, have been prepared. To the best of our knowledge, there has been no report regarding the introduction of  $g - C_3N_4$  to Cr-doped SrTiO<sub>3</sub> to enhance the photocatalytic activity.

In the present study, considering the synergic

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effect between Cr-doped SrTiO<sub>3</sub> and polymeric g-C<sub>3</sub>N<sub>4</sub>, the novel heterostructured Cr-doped SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub> was prepared for the first time as a photocatalyst via mixing and heating methods. The characteristics of the organic-inorganic heterogeneous composite powders with various amounts of  $g - C_3N_4$  were analyzed. The amount of  $g - C_3N_4$  was optimized in terms of maximizing the photocatalytic activity for enhancing the MO photodegradation activity under visible light irradiation. The results demonstrated that compared with pure Cr-doped SrTiO<sub>3</sub> and  $g - C_3N_4$ , the  $Cr$ -doped  $SrTiO<sub>3</sub>$ -g- $C<sub>3</sub>N<sub>4</sub>$  heterojunction photocatalyst had an improved MO photodegradation activity, and the possible photodegradation mechanism was also discussed.

# 2 Experimental

#### 2.1 Materials

The starting materials utilized were  $Sr(NO<sub>3</sub>)<sub>2</sub>$ ,  $Cr(NO<sub>3</sub>)<sub>3</sub>•9H<sub>2</sub>O$ ,  $Ti(C<sub>4</sub>H<sub>9</sub>O)<sub>4</sub>$ , NaOH and ethylene glycol (EG) and melamine  $(C_3H_6N_6)$  (analysis purity grade, Sinopharm Chemical Reagent Co. Ltd.).

### 2.2 Synthesis of Cr-doped SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub> composite

The Cr-doped  $SrTiO<sub>3</sub>$  was prepared by a sol-gel hydrothermal route<sup>[6]</sup>. In a typical process,  $Sr(NO<sub>3</sub>)<sub>2</sub>$ ,  $Cr(NO<sub>3</sub>)<sub>3</sub>•9H<sub>2</sub>O$  and  $Ti(C<sub>4</sub>H<sub>9</sub>O)<sub>4</sub>$  were weighed out according to the stoichiometry of  $Sr<sub>0.95</sub>Cr<sub>0.05</sub>TiO<sub>3</sub>$ .  $Sr(NO<sub>3</sub>)<sub>2</sub>$  and  $Cr(NO<sub>3</sub>)<sub>3</sub>·9H<sub>2</sub>O$  were dissolved in EG and the solution was stirred at 80 °C. Then Ti $(C_4H_9O)_4$ was dropped into the solution and a gel was prepared with the evaporation of EG. Then 5 mol $\cdot L^{-1}$  NaOH solution (30 mL) was added to the gel. After stirred for a certain time, it was transferred into a Teflon-lined stainless steel autoclave and heated at 200 ℃ for 24 h. The product was washed with distilled water until the pH value was 7-8 and dried at 80 ℃ for 12 h. The  $g - C_3 N_4$  photocatalyst was prepared by directly heating melamine at 500 °C (heating rate: 20 °C·min<sup>-1</sup>) for 2 h, and further deammonation treatment was set at 520 °C for 2  $h^{[21]}$  in the semiclosed system to prevent sublimation of melamine. Then, the mixture of Crdoped SrTiO<sub>3</sub> and  $g - C_3N_4$  powder was ground together with 20wt%, 40wt%, 60wt%, and 80wt% g-C<sub>3</sub>N<sub>4</sub> (denoted as 20wt%, 40wt%, 60wt%, and 80wt% Crdoped  $\text{SrTiO}_3\text{-g-C}_3\text{N}_4$ , respectively). Then, the mixture was calcined at 300 ℃ for 1 h in a tubular furnace under N<sub>2</sub> atmosphere. A mechanical mixed  $g - C_3N_4$ and Cr-doped SrTiO<sub>3</sub> with 60wt% g-C<sub>3</sub>N<sub>4</sub> and no heat

treatment was also prepared as a reference (denoted as 60% Cr-doped SrTiO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>).

### 2.3 Characterization

The products were characterized by X-ray diffraction (XRD) for phase identification on a Rigaku Ultima III diffractometer with Cu Kα radiation (λ  $= 0.154$  nm, 40 kV, 40 mA) and a scan rate of 10  $\degree$ •min<sup>-1</sup>. The specific surface area of the as-prepared powders was obtained on a Micromeritics TriStar 3000 instrument (USA) at 77 K and Brunauer-Emmett-Teller (BET) equation was used to calculate the specific surface area. The infrared optical properties were measured on NEXUS870 IR spectrometer using KBr pellet technique. Ultraviolet visible (UVvis) diffuse reflection spectra were recorded using a UV-vis spectrophotometer (Shimadzu UV-2550, Japan) and converted from reflection to absorbance by the Kubelka-Munk method. Their band gaps are calculated by the equation of  $\alpha h v = A(hv - E_g)^{n/2}$ , in which  $\alpha$ ,  $\nu$ ,  $A$ , and  $E_e$  signify the absorption coefficient, light frequency, proportionality constant, and band gap, respectively, and *n* equals 1 or 4, depending on whether the transition is direct or indirect, respectively. Here  $n = 1$  because these materials are direct-gap semiconductors. The microstructure of the sample was observed by a high resolution transmission electron microscope (HR-TEM; JEM-2100, 200 kV, JEOL Ltd.). The acquired photoluminescence (PL) properties of the as-prepared composite catalysts were detected with a spectrofluorometer (VARIAN, Cary Eclipse, USA) at room temperature.

#### 2.4 Photocatalytic activity tests

The methyl orange (MO) dye was used to test the photocatalytic activities of samples. The photocatalytic reaction was preformed in a Pyrex reactor. The catalyst (0.1 g) was dispersed in MO aqueous solution (100 mL,  $4.2 \text{ mg} \cdot \text{L}^{-1}$ ). The light irradiation system contains a 300 W Xe lamp with cut-off filter L42 for visible light and a water filter to remove heating effects. The reaction solutions for all experiments of the MO photodegradation were first stirred in the dark for 1 h to reach the adsorption-desorption equilibrium of MO on catalysts. The degradation efficiency of MO was evaluated using the UV-vis absorption spectra to measure the peak value of a maximum absorption of MO solution at the wavelength of 463 nm.

### 3 Results and discussion

The powder XRD patterns of  $g - C_3N_4$ , Cr-doped

Fig.2 shows the Fourier transform infrared (FT-

inter-layer staking peak of aromatic systems) and (100) diffraction planes (the in-plane structural packing motif) of the graphite-like carbon nitride<sup>[21]</sup> respectively. The  $XRD$  pattern (Fig.1) indicates that the Cr-doped SrTiO<sub>3</sub> sample is very similar to the cubic  $SrTiO<sub>3</sub>$ , and all Bragg diffraction peaks which can be indexed as  $(110)$ ,  $(111)$ , (200), (211), (220) and (310) are in good agreement with JPCDS No. 89-4934. The inset of Fig.1 shows a magnified image of the diffraction peaks for the  $(110)$ plane. The peak is shifted toward a higher angle by the doping of Cr ions, which confirmed the incorporation of the Cr ions into the  $SrTiO<sub>3</sub>$  crystal lattice due to the smaller ionic radius of the  $Cr^{3+}$  (0.0615 nm) than that of  $Sr^{2+}$  (0.118 nm)<sup>[6]</sup>. The Cr-doped SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub> composite samples present two components:  $g - C_3N_4$  and Cr-doped SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub>, and the intensity of the (002) peak of  $g - C_3N_4$  becomes stronger with the increase of the content of  $g - C_3 N_4$ .

 $SrTiO_3$  and Cr-doped  $SrTiO_3-g-C_3N_4$  composite are shown in Fig.1. It can be seen that there are two peaks found in sample with 100% g-C<sub>3</sub>N<sub>4</sub> at 27.40°and 13.08°, which can be indexed to (002) peak (the characteristic



Fig.1 XRD patterns of the Cr-doped SrTiO, and  $g - C_3 N_4$ composite with different proportions. Inset shows diffraction peak positions of the (110) plane in the range of  $2\theta = 32-33^\circ$ 



Fig.2 FTIR spectra of the Cr-doped SrTiO<sub>3</sub> and g-C<sub>3</sub>N<sub>4</sub> composite with different proportions

IR) spectra of  $g - C_3N_4$ , Cr-doped SrTiO<sub>3</sub> and Cr-doped  $SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub>$  composite respectively. The FT-IR spectra of the as-prepared  $g - C_3N_4$  show the features very similar to those of the published results<sup>[10]</sup>. The absorption band near  $1640 \text{ cm}^{-1}$  is attributed to C-N stretching, while the other three absorption bands near 1 240, 1 320 and 1 408 cm<sup>-1</sup> are attributed to aromatic C-N stretching. A broad band near 3140 cm<sup>−</sup><sup>1</sup> corresponds to the stretching modes of terminal NH<sub>2</sub> or NH groups at the defect sites of the aromatic ring<sup>[10]</sup>. The band near 810 cm<sup>-1</sup> is attributed to out-of plane bending modes of C-N heterocycles. The FT-IR spectra of Cr-doped SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub> composites show that the characteristic bands for  $g - C_3N_4$  still remain. For the pure Cr-doped  $SrTiO<sub>3</sub>$ , the broad absorption in the range of 500-800  $cm^{-1}$  may be attributed to  $TiO<sub>6</sub>$  octahedron stretching vibration<sup>[22]</sup>, while for the composite, with the increase of the content of  $g - C_3N_4$ , the intensity of the broad band decreases maybe due to the existence of the organic component.



 $SrTiO<sub>3</sub>$  and g-C<sub>3</sub>N<sub>4</sub> composite with different proportions

UV-vis diffuse reflectance spectroscopy was used to investigate the optical properties of the samples. Fig.3 shows the UV-vis absorption spectra of g-C<sub>3</sub>N<sub>4</sub>, Cr-doped SrTiO<sub>3</sub> and Cr-doped SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub> composite. The absorption edge of the Cr-doped  $SrTiO<sub>3</sub>$ sample occurs at ca. 540 nm, and the band gap energy is calculated as 2.30 eV.

After coupled with  $g - C_3N_4$ , the absorption edge shifts to the higher energy region. It can be seen that the absorption edges of the composite samples shift remarkably to shorter wavelengths with increasing amount of  $g - C_3N_4$ . The increase in band gaps of the samples is from 2.30 eV of Cr-doped SrTiO<sub>3</sub> to 2.70 eV of g-C<sub>3</sub>N<sub>4</sub> when the content of g-C<sub>3</sub>N<sub>4</sub> is increased from 0% to 100%. The morphology of 60wt% Cr-doped  $SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub>$  composite material is investigated using TEM and high-resolution TEM (HRTEM), as shown in Figs.4(a) and 4(b) respectively. The particle size of

the Cr-doped SrTiO<sub>3</sub> is estimated to be in the range of approximately 20-30 nm. The HRTEM image (Fig.4(b)) shows that the lattice fringes have a spacing of 0.27 nm corresponding to interplanar spacing of (110) plane of  $Cr$ -doped  $SrTiO<sub>3</sub>$ . This observation is consistent with the XRD result.



Fig.4 TEM (a) and HRTEM (b) images of 60wt% Cr-doped  $SrTiO<sub>3</sub> - g-C<sub>3</sub>N<sub>4</sub>$  composite



Fig.5 MO photodegradation over the Cr-doped SrTiO, and  $g - C_3 N_4$  composite with different  $g - C_3 N_4$  proportions: (a) 0%; (b) 20%; (c) 40%; (d) 100%; (e) 80% and (f) 60%

 The activities for MO photodegradation over the investigated samples are shown in Fig.5. In our experiments, MO photolysis under visible light irradiation in the absence of catalyst is not observable. This indicates that MO is stable under visible light irradiation. From Fig.5, it can be seen that, after 360 min of visible light illumination, MO removal over Crdoped SrTiO<sub>3</sub> is as low as  $44\%$  and that g-C<sub>3</sub>N<sub>4</sub> shows a photocatalytic performance with a degradation ratio of 85%. The as-prepared composite photocatalysts with 60wt% or 80wt% g-C<sub>3</sub>N<sub>4</sub> show better activity than single-phase  $g - C_3N_4$  or Cr-doped SrTiO<sub>3</sub>. Significantly, the 60wt% Cr-doped SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub> presents a sharp increase in the catalytic activity for MO decomposition, which induced 100% degradation within 240 min light irradiation. 8.9%, 12.9%, 12.4%, 13.8%, and 12.6% MO before the photoreaction was adsorbed on the surface of g- $C_3N_4$ , Cr-doped SrTiO<sub>3</sub> and 20%, 40%, 60%, 80% Cr-doped SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub>, respectively. After 240 min light irradiation, 59.1%, 21.2%, 39.0%, 47.3%, 100%, 79.5% MO was photocatalytically degraded over  $g - C_3 N_4$ , Cr-doped SrTiO<sub>3</sub> and 20%, 40%, 60%, 80% Crdoped  $SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub>$ , respectively (see Fig.5). For the Cr-doped SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub> composite, the g-C<sub>3</sub>N<sub>4</sub> content was important to achieve the high photocatalytic activity. The suitable content of  $g - C_3N_4$  caused its well dispersion on the Cr-doped  $SrTiO<sub>3</sub>$  surface, which promoted the transfer and separation of photogenerated electrons and holes. However, at contents higher than 60wt%, the heterojunction structures and interfaces between the  $g - C_3N_4$  and Cr-doped SrTiO<sub>3</sub> particles decreased. The interfacial charge transfer was suppressed and thus the photocatalytic activity was reduced. Therefore, the  $60wt\%$  Cr-doped SrTiO<sub>3</sub>-g- $C_3N_4$  was the best performing sample and was selected for the cycling tests.



Fig.6 Stability test for the  $60wt\%$  Cr-doped SrTiO<sub>3</sub>-g- $C_3N_4$  photocatalysis system: (a) cycling runs for photodegradation of MO under visible light irradiation; (b) XRD patterns for 60wt% Cr-doped SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub> composite before and after the cycling photocatalytic reaction

The stability of a photocatalyst is a crucial factor for its application and assessment. The cycling reactions for three times for the photodegradation of MO over the 60wt% Cr-doped SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub> photocatalyst were performed to evaluate its photocatalytic stability and the results are shown in Fig.6. The separated photocatalyst was washed with deionized water and dried after every 6 h of photodegradation. Fig.6(a) shows that a high MO photodegradation could be maintained after 3 cycling runs and there was no obvious catalyst deactivation. The XRD patterns (Fig.6(b)) also show that the crystal structure of the Cr-doped  $SrTiO_3-g-C_3N_4$  photocatalysts did not change after the photocatalytic reaction. These evidences indicate that the Cr-doped SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub> photocatalyst can be concluded as stable.

For comparison, the commercial nitrogendoped-TiO<sub>2</sub> with a specific surface area about

 $65 \text{ m}^2 \cdot \text{g}^{-1}$  (TPS201, Sumitomo Corp. Japan) and the mixed material of  $g - C_3N_4$  and Cr-doped SrTiO<sub>3</sub> without heat treatment (60wt% Cr-doped SrTiO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>) were also used as photocatalysts in degradation of MO. Fig.7 shows the degradation activities of MO over different photocatalysts under visible light irradiation. As shown in Fig.7, the 60% Cr-doped SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub> composite show much higher activity than that of nitrogen-doped  $TiO<sub>2</sub>$  and 60wt% Cr-doped SrTiO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> photocatalysts. This clearly indicates that the advantage of heating treatment is the formation of chemically bonded interfaces between the two materials, and the Cr-doped SrTiO<sub>2</sub>-g-C<sub>2</sub>N<sub>4</sub> was determined as an efficient visible light photocatalyst for the degradation of MO.



Fig.7 Comparison of MO degradation over different photocatalysts under visible light irradiation

In general, hydroxyl radicals ('OH) and superoxide  $(O_2$  or HOO') reactive oxidation species, are formed during the photocatalytic reaction<sup>[23]</sup>. Hydroxyl radical may be the active species for composite system $^{[24]}$ . OH is generated via the photogenerated hole oxidation<sup>[25]</sup> or multistep reduction of  $O_2$  induced by photogenerated electron  $(O_2 + e$  $\rightarrow$  O<sub>2</sub><sup>--</sup>, O<sub>2</sub><sup>--</sup> + e + 2H<sup>+</sup>  $\rightarrow$  H<sub>2</sub>O<sub>2</sub>, H<sub>2</sub>O<sub>2</sub> + e  $\rightarrow$  'OH + OH<sup>-</sup>)  $[26]$ . The photodegradation activity of photocatalysts is generally associated with the oxidation ability of photogenerated holes in valence band and reduction ability of photogenerated electrons in conduction band<sup>[23]</sup>. The oxidation potential of g-C<sub>3</sub>N<sub>4</sub> is 1.57  $V^{[12]}$ , which means that the photogenerated holes are incapable of directly oxidizing adsorbed hydroxyl groups to generate hydroxyl radicals  $(2.7 \text{ V} \text{ vs } \text{NHE})^{[23]}.$ Therefore, a reliable speculation may be that the 'OH in the composite photocatalysis system is formed from the multistep reduction of  $O<sub>2</sub>$  under light irradiation.

The BET specific surface area of the  $g - C_3N_4$ , Crdoped SrTiO<sub>3</sub>, 20wt%, 40wt%, 60wt%, and 80wt% Cr-doped SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub> sample is 8.20, 23.10, 20.10, 18.20, 16.01, and 14.40  $m^2 \cdot g^{-1}$ , respectively. Thus, the improved photocatalytic activity is not attributed to the effect of the specific surface area. The enhancement of photocatalytic performance of the composite materials







Fig.9 Photoluminescence spectra of the  $g - C_3N_4$ -Cr-doped SrTiO<sub>3</sub> composite photocatalysts

is supposed to be attributed to the more effective separation of the photogenerated electron-hole pairs. Based on the band gap positions, the CB and VB edge potentials of polymeric  $g - C_3N_4$  were determined as  $-1.13$  and  $+1.57$  eV at pH 7 (versus the normal hydrogen electrode)<sup>[12]</sup>, respectively. The conduction band level of SrTiO<sub>3</sub> is  $-0.2$  eV<sup>[27]</sup>. Therefore, the occupied  $Cr^{3+}$  level in Cr-doped SrTiO<sub>3</sub> is estimated to be 2.1 eV according to the band gap of 2.3 eV which was determined from the UV-Vis absorption spectra (Fig. 3). Since the CB edge potential of g-C<sub>3</sub>N<sub>4</sub> (−1.13) eV) is more negative than that of Cr-doped SrTiO<sub>3</sub> ( $-0.2$ ) eV), the photo-induced electrons on  $g - C_3N_4$  particle surfaces transfer more easily to  $Cr$ -doped  $SrTiO<sub>3</sub>$  via the well developed interface. Similarly, the photo-induced holes on the Cr-doped SrTiO<sub>3</sub> surface move to  $g - C_3N_4$ due to the large difference in VB edge potentials. The scheme for electron-hole separation and transport at the visible-light-driven organic-inorganic composite photo catalyst interface is shown in Fig.8. This reduces the probability of electron-hole recombination and leads to a larger amount of electrons on the Cr-doped  $SrTiO<sub>3</sub>$ surface and holes on the  $g - C_3 N_4$  surface, respectively, which promotes the photocatalytic reactions to decompose MO.

PL analysis has been used to investigate the

migration and recombination processes of photogenerated electrons and holes in semiconductors $[12,13]$ . To further confirm the above-proposed mechanism, the PL spectra of the Cr-doped  $SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub>$  composite photocatalysts at an excitation wavelength of 330 nm are investigated and shown in Fig.9. The emission band for pure  $g - C_3 N_4$  is centered at 478 nm, which is attributed to the radiative recombination process of self-trapped excitations<sup>[12]</sup>. The positions of the Cr-doped SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub> emission peaks are similar to that of pure  $g-C_3N_4$ . Obviously, compared with pure  $g - C_3N_4$ , the emission intensity of the Cr-doped  $SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub>$  composite samples is much lower. The PL results indicate that the recombination of photogenerated electrons and holes was inhibited in the composite semi-conductors, which conforms to the discussion about the separation of charge carriers and photocatalytic experiments.

### 4 Conclusions

In summary, we have fabricated a Cr-doped  $SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub>$  composite photocatalyst by introducing polymeric  $g - C_3 N_4$  via a mixing and heating method. The stable dye MO was selected as a substrate to evaluate the photocatalytic activity of Cr-doped  $SrTiO<sub>3</sub>$  $g - C_3N_4$ . Our results clearly indicate that the visiblelight-driven Cr-doped SrTiO<sub>3</sub>-g-C<sub>3</sub>N<sub>4</sub> heterojunction has a good performance in photo-oxidation of this organic pollutant, which resulted from the suitably matching conduction and valance band levels that improved the separation efficiency of photogenerated electron-hole pairs.

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