#### **ORIGINAL PAPER**



# Effect of calcination temperature on electrochemical properties of spinel-like NiCo<sub>2</sub>O<sub>4</sub> nano-/microstructures

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Received: 20 April 2022 / Revised: 20 May 2022 / Accepted: 28 May 2022 / Published online: 3 June 2022 © The Author(s), under exclusive licence to Springer-Verlag GmbH Germany, part of Springer Nature 2022

#### Abstract

The current study evaluated the impact of the annealing temperature on the electrochemical characteristics of the spinel-like nickel cobaltite (NiCo<sub>2</sub>O<sub>4</sub>) nano-/microstructures. For this purpose, NiCo<sub>2</sub>O<sub>4</sub> samples are prepared through a facile coprecipitation synthesis and calcined at two distinct temperatures of 300 °C and 500 °C. The prepared samples were characterized through X-ray diffraction (XRD), scanning electron microscope (SEM), and energy dispersive X-rays analysis (EDX). The electrochemical characterizations are carried out on prepared samples using the CHI760E electrochemical workstation to determine their suitability as active materials for supercapacitors. The results revealed the battery-type behavior of the prepared electrode materials with the highest value of specific capacity in a 500 °C calcined sample of 224 C/g at 2.00 A/g. The results of EIS are aligned with the outcomes of CV and GCD. The electrochemical analysis indicates that the 500 °C calcined temperature is appropriate for use as electrode material in supercapacitors.

Keywords Supercapacitor · Spinel · Co-precipitation · Cyclic voltammetry · Electrochemical

# Introduction

The time has come when energy has become the priority of humankind. Human beings rely on energy in numerous areas, from the kitchen to the office, from house to market, and from a journey to the destination. As the need for energy expands, so does the requirement for storage devices. Due to certain limitations, traditional batteries and capacitors are unable to meet the ever-increasing demand for energy storage devices [1]. Batteries take a long time to charge in today's fast-paced environment, and capacitors are unable to provide prolonged discharge times [2]. In the last few decades, supercapacitors have achieved a lot of interest from researchers owing to their extraordinary properties of quick charging and discharging, excellent cycle stability, and larger power density [3]. In general, supercapacitors are split into

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<sup>2</sup> Department of Physics, National Institute of Technology, Kurukshetra 136119, India three categories [4]: the first is EDLC (electric double layer capacitor), which uses carbonous materials as the working electrode material [5, 6]. The pseudocapacitor (PC) is the second type of supercapacitor that uses metal oxides or hydroxides as the electrode material and stores energy because of redox processes at the boundary of electrode and electrolyte [7]. The hybrid supercapacitors (HS) fall into the third group, with one electrode made of EDLC and the other made of PC electrode material [8, 9]. But the supercapacitor still lags the batteries in terms of energy density. The research community from all over the globe is focused on enhancing the energy density of supercapacitors. Recently, the battery-type materials have been widely produced and researched in the field of energy storage because of their richer faradaic redox reactions and higher energy density. In electrochemical studies, battery-type electrode materials have clear redox peaks and a nonlinear potential platform, but EDLC and PC materials have different patterns. As a result, the capacity of charge storage for battery-type materials is expressed in C/g rather than F/g for specific capacitance [10]. In general, the type of electrode materials, calcination temperature, synthesis method, and electrolyte deliver a significant impact on the specific capacitance of the electrode material. A high-performance SC is desirable with high specific capacitance/capacity, durable cycling stability, and low specific capacitance. Binary metal oxides have become highly attractive electrode materials in supercapacitors because of their high theoretical specific capacity [11]. Numerous spinels can be used for this purpose like Fe<sub>3</sub>O<sub>4</sub>, Cu<sub>3</sub>O<sub>4</sub>, CuFe<sub>2</sub>O<sub>4</sub>, MnCo<sub>2</sub>O<sub>4</sub>, NiCo<sub>2</sub>O<sub>4</sub>, ZnCo<sub>2</sub>O<sub>4</sub>, MgCo<sub>2</sub>O<sub>4</sub>, and FeCo<sub>2</sub>O<sub>4</sub>, and all these effectively serve outstanding electrochemical properties [4, 12, 13]. For a couple of decades, spinel NiCo<sub>2</sub>O<sub>4</sub> has attracted considerable interest because it is not only cost-effective, plentiful, and ecologically acceptable, but it also has superior electrical conductivity and electrochemical activity to Mn- and V-based materials. There are ample factors that deliver a significant role in the electrochemical presentation of NiCo<sub>2</sub>O<sub>4</sub> as a supercapacitor device such as the method of fabrication [14], precursor taken for synthesis, pH of the synthesis solution [15, 16], type, and concentration of electrolyte, aging period, and the temperature of calcination [17]. H. Qin et al. demonstrated a multi-responsive healable supercapacitor with magnetic  $Fe_3O_4$ @Au/polyacrylamide (MFP) as the electrode material to improve the reliability and lifespan of the device. Herein, results showed the highest areal capacitance of 1264 mF/cm<sup>2</sup> and restore ~ 90% of the initial value of capacitances after ten healing cycles [18]. K. Xu et al. successfully synthesized hollow NiCo<sub>2</sub>O<sub>4</sub> nanostructures with high active surface area using freshly prepared SiO<sub>2</sub> nanospheres. The resulted sample yielded the highest specific capacitance of 1229 F/g at 1.0 A/g, appreciable rate performance of 83% up to 25 A/g current density, and good charging-discharging stability of 86% after 3000 cycles [19]. By producing flower-like NiCo2O4 and balancing asymmetric capacitance, Z. Wang et al. were able to overcome the shortcomings of NiCo<sub>2</sub>O<sub>4</sub> as a supercapacitor electrode, such as a narrow operating voltage and low mass loading. As a result, even with a loading mass of up to 9 mg/cm<sup>2</sup>, the flower-like NiCo2O4 with highly porous ultrathin petals has a high specific capacity of 350 C/g [20]. N. Zhao et al. utilized two-step electrodeposition followed by the calcination method to produce NiCo<sub>2</sub>O<sub>4</sub>@Ni<sub>4.5</sub>Co<sub>4.5</sub>S<sub>8</sub> on the Ni foam to obtain nanosheet morphology. The resulting sample liberated a specific discharge capacity of 369 mAh/g at 1.0 A/g and 258 mAh/g at 20 A/g and excellent cycle stability after 5000 cycles with only 4.8% loss in capacity [21]. H. Chen et al. again fabricated flowerlike NiCo<sub>2</sub>O<sub>4</sub> nanostructures with an enhanced specific capacitance of 658 F/g at 1 A/g current density through a simple hydrothermal method [22]. The prepared NiCo2O4 sample has an extremely extended cycling lifespan, with no signs of degradation even after 10,000 cycles. Recently, A. Manalu et al. fabricated a nanocomposite of NiCo<sub>2</sub>O<sub>4</sub> with rGO via the facile hydrothermal method and reported a specific capacitance of 289.93 F/g as pseudocapacitor [23]. No one has examined the influence of temperature of calcination on the electrochemical presentation of NiCo<sub>2</sub>O<sub>4</sub> nanostructures to our knowledge.

We synthesized NiCo<sub>2</sub>O<sub>4</sub> at two distinct calcination temperatures, 300 °C and 500 °C, for this objective. The synthesized material was characterized by XRD for the analysis of sample purity and crystallinity. SEM analyses were performed for the morphology. To investigate the electrochemical performance, CV, GCD, and EIS were performed. From the GCD, the evaluated value of specific capacity is 224 C/g and 100C/g at current density of 2.00 A/g from NC500 and NC300, respectively. Hence, the calcination temperature influences the electrochemical performance of the NiCo<sub>2</sub>O<sub>4</sub> nanostructures.

# **Materials and methods**

#### **Materials utilized**

All the substances were used exactly as they were obtained, with no further modifications. Potassium hydroxide (KOH) and N-Methyl-2-pyrrolidone (NMP) were purchased with analytical grade from LOBA. Cobalt nitrate hexahydrate ( $Co(NO_3)_2.6H_2O$ , AVARICE, 98% purity) and nickel nitrate hexahydrate ( $Ni(NO_3)_2.6H_2O$ , RANKEM, 98% purity) were purchased. The whole experiment was performed with double distilled water.

## Synthesis

A simple room temperature co-precipitation synthesis followed by a calcination route was followed to fabricate  $NiCo_2O_4$ . The illustration for the synthesis is represented in Fig. 1. Initially, 10 mmol of Ni(NO<sub>3</sub>)<sub>2</sub>.6H<sub>2</sub>O and 20 mmol of Co(NO<sub>3</sub>)<sub>2</sub>.6H<sub>2</sub>O were dissolved in 40 ml of double distilled water to prepare solution (I) and stirred magnetically for half-hour. Furthermore, 6 M KOH was prepared in 10 ml for solution (II). After half-hour stirring, solution (II) was dropwise added to the solution (I). The obtained precipitates were left stirred for another 1 h. The precipitates were then filtered and washed multiple times with distilled water and ethanol. The filtrate was dried in an electric oven at 80 °C overnight. Finally, to obtain NiCo<sub>2</sub>O<sub>4</sub>, the dried precipitates were calcined at a temperature of 300 °C for 3 h in a furnace and named NC300. The sample was prepared again to calcined at 500 °C and named NC500.

## **Electrode preparation**

To prepare the electrodes, 80 mg prepared sample, 10 mg activated carbon, and 10 mg polyvinylidene fluoride (PVDF) were taken for each sample and carefully pulverized in a mortar-pestle for 1 h. The mixture was poured into a 10 ml culture tube, which was then sprinkled with 5–6 drops of N-methyl-2-pyrrolidone (NMP). To generate a homogeneous

Fig. 1 Schematic diagram for the synthesis of  $NiCo_2O_4$  through the Co-precipitation route



slurry, the mixture was magnetically swirled for roughly 12 h. The prepared slurry was dropped onto clean and dry Ni foil in a  $1\text{-cm}^2$  area using the drop-caste method. The pasted electrodes were dried for 12 h in an electric oven at 80 °C. The mass of an empty clean electrode and a slurry pasted, and dried electrodes were used to calculate the weight of active material. The mass of the active material was nearly 1.0 mg.

# **Characterization techniques**

The crystallinity, purity, and structural features of both the manufactured  $NiCo_2O_4$  samples were investigated using an X-ray diffractometer (Rigaku Miniflex Japan) with a Cu-K<sub> $\alpha$ </sub> radiation source and a scanning rate of 2°/min within the range of 10–90° characterization. For surface, morphological, and elemental composition analyses, the JEOL JSM-6390LV SEM was utilized to achieve scanning electron microscope (SEM) micrographs and energy dispersive X-ray spectroscopy (EDX) spectrum. For the three-electrode (3E)

arrangement, the electrochemical investigation was performed using a CHI 760E electrochemical workstation. The synthesized material served as the working electrode, with Ag/AgCl serving as the reference electrode and a platinum wire acting as the counter electrode. Under electrochemical inquiry, cyclic voltammetry (CV), galvanostatic charge–discharge (GCD), and electrostatic impedance spectroscopy (EIS) characterizations are studied in 2 M KOH electrolyte.

# **Results and discussions**

### X-ray diffraction (XRD)

XRD characterization was applied to investigate the crystalline nature and purity of the manufactured samples. The obtained XRD patterns for NC300 and NC500 are presented in Fig. 2(a) and the crystal structure of NiCo<sub>2</sub>O<sub>4</sub> is shown in Fig. 2(b). The diffraction crests in both materials are at 18.7, 31.0, 36.75, 44.50, 55.20, 58.92, 64.7, and 77.8, which

**Fig. 2** a Room-temperature X-ray diffraction of  $NiCo_2O_4$  prepared at 300 °C and 500 °C temperatures. **b** Crystal structure of spinel-like  $NiCo_2O_4$ 



correspond to the (111), (220), (311), (400), (422), (511), (440), and (533) planes, respectively. The result was consistent with the conventional cubic phase  $NiCo_2O_4$  pattern (JCPDS number 73–1702) [24]. There were no more peaks seen, indicating the creation of a pure cubic phase. The higher peak intensity in NC500 as compared to NC300 clear indicates the effect of calcination temperature on the crystallinity of the material. A difference in the electrochemical performance can also be predicted from the XRD pattern.

# Scanning electron microscope and energy dispersive X-ray analysis (EDAX)

To explore the surface morphology and elemental composition, the synthesized materials have gone through the room temperature SEM and EDAX analysis. The resulted images are revealed in Fig. 3(a-d). The NC300 sample has a nonuniform sheet-like morphology as indicated in Fig. 3(a); on the other hand, comparatively smaller sheets and a few rods are visible in Fig. 3(c) for NC500. The EDAX spectrum of NC300 and NC500 is shown in Fig. 3(b) and Fig. 3(d), respectively, indicating the presence of elements Ni, Co, and O which confirms the purity of the samples. The effect of temperature is noticeable on the morphology as well as the weight percentage of the synthesized samples of  $NiCo_2O_4$ . Changes in morphology and weight percentage with temperature can produce a significant impact on the electrochemical properties of materials. Sintering at a higher temperature or for longer periods of time can improve crystal development and crystallinity in the sample. As a result, the capacity of calcined materials at higher temperatures is estimated to be greater.

## **Cyclic voltammetry**

CV is a valuable characterization for determining the charge storage mechanism of electrode material. Figure 4(a) and (b) represents the voltammograms of NC300 and NC500, respectively, within a potential window range of 0.0 to 0.45 V (vs. Ag/AgCl electrode) at different scan rates in 1–20 mV/s. Well-defined redox waves belonging to  $Co^{3+}/$ Co<sup>4+</sup> and Ni<sup>2+</sup>/Ni<sup>3+</sup> redox couples can be seen in pairs on the CV curves, suggesting battery-type characteristics of the material [25]. At 10 mV/s, the redox couple is positioned at 0.33 V/0.24 V in NC300 and 0.32 V/0.23 V in NC500. The minor discrepancy in redox peak potential is related to differences in morphology. With the increase in scan rates, the shape of voltammograms remains unchanged, while the anodic and cathodic peaks shift towards positive and negative potential, respectively. This shift in peaks is more profound in the case of NC500. A close look at the pattern of NC300 voltammograms shows appropriate redox peaks in both forward and backward sweeps, demonstrating that the electrode material is more reversible and has better rate capability. In the case of NC500, however, the reverse scan peaks are missing, indicating a less reversible nature. The following equations describe the electrochemical behavior of nickel cobaltite (NiCo<sub>2</sub>O<sub>4</sub>) in an alkaline electrolyte [26]:

$$NiCo_2O_4 + OH^- + H_2O \leftrightarrow NiOOH + 2CoOOH + e^-$$
 (1)

$$NiOOH + OH^- \leftrightarrow NiO_2 + H_2O + e^-$$
 (2)

$$CoOOH + OH^- \leftrightarrow CoO_2 + H_2O + e^-$$
 (3)

 $Co^{3+}/Co^{4+}$  and  $M^{2+}/M^{3+}$  (M = Co and Ni) redox peaks are so close in the potential that they frequently overlap in

Fig. 3 a SEM image of NiCo<sub>2</sub>O<sub>4</sub> prepared at 300 °C (NC300). b EDAX spectrum of NC300. c SEM image of NiCo<sub>2</sub>O<sub>4</sub> prepared at 500 °C (NC500). d EDAX spectrum of NC500



**Fig. 4** a CV of NiCo<sub>2</sub>O<sub>4</sub> calcined at 300 °C temperature. b CV of NiCo<sub>2</sub>O<sub>4</sub> calcined at 500 °C temperature. c Power-law relationship plot for NiCo<sub>2</sub>O<sub>4</sub> as NC300 and NC500



the CV pattern and only one peak is observed for all the redox couples.

Furthermore, the power-law relationship, i.e.,  $i_P = av^b$ (where  $i_p$  and v denote the peak current and scan rate, respectively) is investigated to better understand the charge storage mechanism. The "b-value" = 1, nearly 1 ( $\approx$ 1), and = 0.5 correspond to pure capacitive EDLC behavior, pseudocapacitive behavior, and battery-like behavior, respectively [27]. Figure 4(c) shows the log( $i_P$ ) vs. log(v) plot used to determine the b-value, which is 0.77 for NC300 and 0.87 for NC500, respectively. It can be concluded that the electrochemical behavior of the as-synthesized electrode materials lies mid-way between battery-type and pseudocapacitor. Furthermore, the GCD characterization is used to better assess the reliability of an electrode material in an energy storage application.

#### Galvanostatic charge-discharge

The charge-discharge behavior, which is evaluated using the Chronopotentiometry technique, is critical for analyzing the practical utility of the material in supercapacitor application. The obtained curves of GCD are represented in Fig. 5(a) and (b) for NC300 and NC500, respectively, within the same potential window as in CV at numerous current densities between 2 and 10 A/g. The quasi-triangular behavior of the curves again confirms the battery-type nature of the electrode materials. Symmetry in charge and discharge arcs indicates the reversibility of the redox reaction. These findings resonate well with the CV observations. The value of specific capacity (C/g) is calculated from the GCD curves using the following equation [28]:

$$C_s = \frac{I}{m} \times \Delta T \tag{4}$$

where  $C_s$  is specific capacitance, I/m is current density, and  $\Delta T$  is the discharge time. Due to the difficulties of reaching all energetic redox sites with a high current density, the discharge time reduces as the current density rises. The calculated values of every specific capacity at several current densities are shown in Table 1. The supreme value of specific capacitance, 224 C/g, is obtained at 2 A/g from NC500, but it is only 100 C/g in NC300. However, NC300 has a rateperformance of 77%, which is higher than 67% in the case of NC500. The variation in the specific capacities with current densities is also elaborated in Fig. 5(c). The cycling stability is another important parameter to judge the practical utility of the electrode material in supercapacitor applications. The NC500 sample results in outstanding retention as shown in Fig. 5(d) of 80% in the specific capacity from the initial after 2000 cycles of GCD at 15.0 A/g current density.

**Fig. 5** a GCD of NiCo<sub>2</sub>O<sub>4</sub> calcined at 300 °C temperature. **b** GCD of NiCo<sub>2</sub>O<sub>4</sub> calcined at 500 °C temperature. **c** Variation in specific capacity (C/g) of NiCo<sub>2</sub>O<sub>4</sub> with the current density. **d** Retention (%) of NiCo<sub>2</sub>O<sub>4</sub> as NC500 for 2000 cycles



Table 1 Values of specific capacitance from GCD curves of NC300 and NC500  $\,$ 

Sr. no	Current density (A/g)	Specific capacity (C/g)	
		NC300	NC500
1	2	100.6	224.06
2	4	88.52	192.45
3	6	82.80	173.58
4	8	79.04	153.52
5	10	77.00	150.20

#### Electrostatic impedance spectroscopy

EIS is performed to investigate electrochemical capacitor performance, such as internal resistance, capacity, and so on by applying AC sinusoidal waves. Nyquist plots are used to examine the EIS data which revealed the electrode/ electrolyte system's frequency response. The imaginary component (Z") of the impedance is plotted against the real component (Z') in a Nyquist plot. Figure 6(a) shows the EIS data obtained for NC300 and NC500 and the corresponding circuit diagrams are represented in Fig. 6(b) and (c), respectively. A Nyquist plot can be studied in two parts: One is a high-frequency semicircle that represents the charge transfer resistance ( $R_{CT}$ ) present at the electrode–electrolyte interface, and another is a low-frequency slanted line that arises due to diffusive resistance and capacitive behavior of the material. Hypothetically, a supercapacitor has a straight line at 90° to the real component of the impedance but, practically, it is somewhere between 45 and 90°. For NC300 and NC500, the R<sub>CT</sub> values determined from the circuit diagram are 43.6 and 21.44  $\Omega$ , respectively. The lower capacitance in NC300 is due to a higher R<sub>CT</sub> value. Furthermore, the equivalent series resistance (R<sub>S</sub>) of NC300 is 0.57  $\Omega$  which is slightly higher than that of 0.49  $\Omega$  for NC500.

## Conclusions

In summary, the impact of the temperature of calcination on the electrochemical presentation of spinel-like nickel cobaltite (NiCo<sub>2</sub>O<sub>4</sub>) microstructures was investigated in this study. NiCo<sub>2</sub>O<sub>4</sub> has been synthesized for this purpose using a robust and simple co-precipitation route at two different temperatures of 300 °C and 500 °C. XRD examination validated the purity of the phase and revealed the increased intensity of diffraction peaks in NC500 that is at the higher temperature of calcination. The electrochemical performance was using a CHI760E electrochemical workstation via CV, GCD, and EIS techniques. In the NC500, the specific capacity was 224 C/g at 2.00 A/g in GCD with 80% retention after 2000 **Fig. 6** a Nyquist plot NiCo<sub>2</sub>O<sub>4</sub> calcined at 300 °C (NC300) and 500 °C (NC500) temperature. **b** Equivalent circuit diagram of NC300. **c** Equivalent circuit diagram of NC500



cycles at 15.0 A/g current density. The rate performance 4.

of the NC300 was slightly better at 77%, even though the capacitance values in the NC500 were higher. The EIS findings were consistent with the CV and GCD findings. Overall, the electrochemical properties of both samples show that the calcined temperature of 500 °C is suitable for usage as electrode material in supercapacitors.

Acknowledgements One of the authors, P. Chand, is thankful to the Science and Engineering Research Board (SERB), Govt. of India, for providing funding through research project No: SERB/F/10804/2017-18. The authors are also grateful to the Director, NIT Kurukshetra, for providing the facilities in Physics Department. We acknowledge RAFM-2022, organized by the Department of Physics, ARSD College, for this submission.

Author contribution Manpreet Kaur: methodology, software, data curation, and writing – original draft; Prakash Chand: conceptualization, writing – review and editing, validation, resources, and supervision; Hardeep Anand: conceptualization, writing – review and editing, and supervision.

# Declarations

Conflict of interest The authors declare no competing interests.

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