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Synthesis of transition metal oxide incorporated MOF-5 and NH₂-MOF-5 as efficient photoanode for oxygen evolution reaction

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Abstract

Development of a highly active, stable, and facile-synthesized photoelectrocatalyst for water oxidation (OER) is very challenging and has attracted great research attention. In this article, highly efficient MOF-based photoelectrocatalysts (MOF-5 and aminefunctionalized MOF (NH₂-MOF-5)) have been synthesized at room temperature and have been successfully characterized. For the photoelectrochemical studies, working electrodes are prepared by coating the synthesized photoelectrocatalysts on Ni-foam. All the synthesized materials have been successfully characterized via powder X-ray diffraction (PXRD), Fourier-transform infrared (FTIR) spectroscopy, scanning electron microscopy (SEM), energy-dispersive X-ray (EDX) spectroscopy, elemental mapping, and ultraviolet-visible (UV-Vis) spectroscopy. Photoelectrochemical measurements for oxygen evolution reaction are performed via cyclic voltammetry and linear sweep voltammetry. It has been observed that among all the synthesized catalysts, $Co₃O₄@NH₂-MOF-5/NF$ has emerged as an efficient, stable, and highly active photoelectrocatalyst towards oxygen evolution reaction (OER) as compared to all other synthesized catalysts. It requires just 223 mV overpotential to deliver the 10 mA cm−² current density and exhibits the lowest Tafel slope 52 mV dec⁻¹ as compared to all other synthesized samples and some of the previously reported catalysts. Furthermore, long-term catalytic stability is studied via continuous linear sweep voltammetry and chronoamperometric measurements. This study encourages the development of a more efficient MOF-based catalyst for different photoelectrochemical studies.

Keywords Water oxidation \cdot Oxygen evolution reaction \cdot Amine-functionalized MOFs \cdot Co₃O₄ \cdot Linear sweep voltammetry

Introduction

In the twenty-first century, society is facing a great challenge to deal with environmental pollution and energy crisis due to dependence on carbon-based fuels. So, it motivated a huge number of scientists and researchers to search for an alternate source of energy [[1\]](#page-10-0). Different strategies have been proposed to solve this and one of them is hydrogen production from water splitting. Water splitting is divided into two half-reactions: cathodic half-reaction (where hydrogen evolution reaction (HER) takes place) and anodic half-reaction (where

 \boxtimes Muhammad Athar athar.qr@bzu.edu.pk oxygen evolution reaction (OER) takes place) [\[2](#page-10-0)]. Both HER and OER are kinetically sluggish reactions and huge overpotential is required to occur at a suitable rate. OER is four electron-proton coupled reaction (4OH^{$-$} → 2H₂O + 4e^{$-$} + O_2) while HER is just a two-electron transfer reaction (2H⁺ + $2e^- \rightarrow H_2$). So, OER is kinetically more sluggish and requires higher overpotential as compared to HER to overcome the energy barrier and splits the water [\[3](#page-10-0)].

Therefore, extensive research has been focused on the development of highly efficient OER catalysts. Among various active OER catalysts, $RuO₂$ and IrO₂ are considered as highly active OER catalysts both in acidic and basic electrolytes because they require very less overpotential for oxygen evolution from water splitting [\[4](#page-10-0), [5\]](#page-10-0). But, they cannot be used in large-scale practical applications due to their scarcity and expensive nature. So, researchers and scientists have devoted their research attention to the development of efficient, stable, and cost-effective non-noble metal OER catalysts, i.e., 3dtransition metal oxides, perovskites, transition metal sulfides, hydroxides, and non-metal compounds [[6](#page-10-0)–[10](#page-10-0)].

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Metal-organic frameworks (MOFs) are porous crystallinematerials, composed of inorganic metal and an organic ligand. Due to their unique properties such as high porosity and large surface area, they have been widely used in a large number of potential applications such as gas adsorption/separation [[11\]](#page-10-0), magnetism [\[12](#page-10-0)], optoelectronics [\[13](#page-10-0)], and catalysis [\[14\]](#page-10-0). In recent years, due to their tunable pore structure and easy functionalization, MOFs are emerging as promising catalysts in a basic medium for water splitting [\[15](#page-10-0)]. Recently, highly efficient MOF-based OER catalysts have been reported such as NiFe-MOF-74 supported on Ni-foam delivered the bench-mark of 10 mA cm⁻² at 223 mV overpotential [\[16](#page-10-0)]. Similarly, Fe-doped Ni-MOF $Fe^{II}_{0.1}Ni_{0.4}$ -MOF, Co-BPDC/Co-BDC heterojunction, UTSA-16, and Co-MOF have been reported as efficient OER catalysts and they delivered the 10 mA cm^{-2} current density at 294, 335, 408, and 360 mV overpotential, respectively [\[17](#page-10-0)–[20](#page-10-0)]. Therefore, metal-organic frameworks (MOFs) have been used as potential catalysts for oxygen evolution reaction (OER), oxygen reduction reaction (ORR), and hydrogen evolution reaction (HER) such as NiO-MOF/rGo, rGO-NiO/CuO MOF, Ni-BTC-MOF/rGO, Co-BTC MOF/ rGO, MnBDC@rGO, Cu-ZIF-67, NiCo/graphene, NiCo-ZIF, Cu-MOF/GO, and NiCo/NCNTs [\[21](#page-10-0)–[30\]](#page-11-0). It has been also observed that catalytic efficiency of MOFs can be further increased by using 2-aminoterephthalic acid as an organic linker instead of terephthalic acid because amine functionalization shifts the band gap towards the visible region, increases the absorption of visible light, improves the charge separation, and reduces the e^-/h^+ (electron-hole pair) recombination [[31](#page-11-0)].

First row 3d-transition metal oxide nanoparticles have been widely used in catalytic applications due to their low band gap, non-irritating, easy availability, and cost-effectiveness. Transition metal oxide (Mn, Fe, Co, Ni, and Cu) is also used as efficient HER and OER catalysts [\[32](#page-11-0)].

Therefore, in this research, we report the facile and lowcost synthesis of highly efficient OER $M_xO_y@MOF-5$ and $M_xO_y@NH_2-MOF-5$ ($M_xO_y = Co_3O_4$, NiO, CuO, and ZnO) catalysts at room temperature via in situ incorporation of four metal oxide nanoparticles into host MOFs. It is observed that amine functionalization has increased the photoelectrocatalytic activity towards OER. $M_xO_y@NH_2$ -MOF-5 photoelectrocatalysts show better OER activity as compared to $M_xO_y@MOF-5$ and some of previously reported MOF-based OER catalysts.

Zinc acetate dihydrate $(Zn(CH_3COO)_2.2H_2O$, Merck, 99.99%), 1,4-benzenedicarboxylic acid $(H₂BDC, Merck,$

Experimental

Chemicals

99%), 2-aminoterephthalic acid (Merck, 99%), N,Ndimethylformamide (DMF, Merck, 99.8%), sodium hydroxide (NaOH, Merck, 97%), ethanol (CH₃CH₂OH, Merck, 95%), oleic acid $(C_{18}H_{34}O_2,$ Merck, 99%), iron sulfate heptahydrate $(Fe_2(SO_4)_3.7H_2O$, Merck, 21-23% Fe basis), cobalt acetate tetrahydrate $(Co(C_2H_3O_2)_2.4H_2O$, Merck, 97%), ammonia solution (NH4OH, Merck, 28%), nickel chloride hexahydrate (NiCl₂·6H₂O, Merck, 97%), copper nitrate trihydrate $(Cu(NO₃)₂·3H₂O, Merck, 99%),$ and deionized water.

Synthesis of transition metal oxide (M_xO_y) nanoparticles

Four transition metal oxide ($M_xO_y = Co_3O_4$, NiO, CuO and ZnO) nanoparticles were synthesized [\[33](#page-11-0)–[36\]](#page-11-0). Except for $Co₃O₄$ nanoparticles, the remaining metal oxide (NiO, CuO, and ZnO) nanoparticles were synthesized by using the precipitation method. In brief, a particular amount of metal salt was dissolved in 100 mL distilled water by continuous stirring for 30 min. After that, 0.2 M NaOH was added dropwise and stirred for a specific period of time at a specific temperature as mentioned in Table [1.](#page-2-0) Then, precipitates were obtained, separated by centrifugation, washed with distilled water, and dried at 50 °C. After that, dried precipitates were calcinated at 500 °C for 2 h, while Co_3O_4 nanoparticles were prepared by using the one-step hydrothermal method. Cobalt acetate tetrahydrate was dissolved in distilled water by continuous stirring for 30 min. After that, 10 mL of ammonia solution was added dropwise under continuous stirring and transferred the whole mixture into Teflon-lined autoclave and heated at 180 ° C for 12 h. Finally, the precipitates of $Co₃O₄$ nanoparticles were obtained.

Synthesis of amine-functionalized MOF-5

MOF-5 was synthesized at room temperature. In a roundbottom flask, 2.7 g of $Zn(CH_3COO)_2$: $2H_2O$ was dissolved into 100 mL DMF by continuous stirring. Then, 50 mL DMF solution containing 0.8 g H_2 BDC was added dropwise to the mixture under constant magnetic stirring. The whole mixture was stirred at room temperature for 24 h. Precipitates were obtained, collected by centrifugation, and washed with distilled water and DMF several times. The resulting product was dried at 50 \degree C in a vacuum oven for 3 h.

Amine-functionalized MOF-5, NH₂-MOF-5, was synthesized under the same conditions as mentioned above for MOF-5 by using 2-aminoterephtahlic acid instead of 1,4 benzenedicarboxylic acid as an organic linker.

Table 1 Conditions for the synthesis of NiO, CuO, and ZnO nanoparticles via the precipitation method

Synthesis of $M_xO_y@MOF-5$ and $M_xO_y@NH_2$ -MOF-5 composites

 $M_xO_y@MOF-5$ and $M_xO_y@NH_2-MOF-5$ composites were synthesized at room temperature via in situ incorporation of pre-synthesized metal oxide nanoparticles into host MOFs. In a typical procedure, 0.8 g of 1,4-benzenedicarboxylic acid was dissolved in 20 mL DMF by continuous stirring for 30 min. After that, dropwise added 10 mL DMF suspension of 0.01 g metal oxide nanoparticles and 2.7 g/20 mL DMF solution of $Zn(CH_3COO)_2.2H_2O$, respectively. Then, the resultant solution was continuously stirred for 24 h and precipitates were separated by centrifugation, washed with distilled water and DMF, and dried overnight at 50 °C in an oven under vacuum. In the same way, $M_xO_y@NH_2-MOF-5$ composites were synthesized by using 0.8 g/20 mL DMF of 2-aminoterephthalic acid instead of 1,4-benzenedicarboxylic acid.

Instruments

Powder X-ray diffraction spectra (PXRD) were obtained in 2θ range 5 to 80° via Shimadzu X-ray diffractometer with Cu-Kα radiation ($\lambda = 0.15406$ nm) at 40 kV and 40 mA. A Nicolet Nexus 870 spectrometer was used to collect FTIR spectra in the wavelength range from 400 to 4000 cm^{-1} . Morphology and elemental composition of synthesized samples were evaluated by using Philips XL30 Environmental SEM attached with an Oxford Instrument Inca 500 energy-dispersive X-ray spectrometer (EDX). Optical properties were studied by using Shimadzu UV-2600 spectrophotometer in the range of 200– 900 nm.

Formation of working electrodes

The slurry of the catalyst was uniformly pasted on a piece (1 cm²) of Ni-foam (NF) to form working electrodes. Before pasting slurry, Ni-foam was cleaned with acetone and water via sonication and dried at 50 °C in an oven for 2 h. To prepare the slurry, 10 mg of synthesized sample was dispersed in 1 mL deionized water via sonication for 30 min. After that, it was uniformly pasted on Ni-foam to form a photoanode for OER.

Photoelectrochemical studies

Photoelectrochemical studies towards OER were studied in a common three-electrode setup with Ag/AgCl as a reference electrode, Pt as a counter electrode, and catalyst coated on Nifoam as photoanode. The photoelectrochemical studies were studied both in dark and in presence of visible light. The electrodes were dipped in 1M KOH (aq) electrolyte solution. A Uniscan instrument 3100 Poteniostat/Galvanostat attached with NOVA 1.10 software for automated data taking and recording was used for OER analysis. OER activity was studied via cyclic voltammetry (CV) and linear sweep voltammetry (LSV) measurements at 30 and 1 mV s^{-1} , respectively. All the potential data was converted into reversible hydrogen electrode (RHE) potential by using the following equation.

$$
E_{RHE} = E_{Ag/AgCl/Sat.KCl} + 0.059\ pH + 0.197\tag{1}
$$

For OER, overpotential (n) was calculated by using the following equations

$$
\eta = E_{RHE} - 1.23\tag{2}
$$

Tafel slope was calculated from the Tafel plot by using the following equation

$$
\eta = a + b \log j \tag{3}
$$

where η is the overpotential, $\dot{\eta}$ is the current density, and $\dot{\theta}$ is the Tafel slope

Furthermore, the stability of working electrodes was found via continuous 1000 LSV sweeps and chronoamperometric studies at an applied potential of 1.5 V in the presence of visible light.

Results and discussion

Crystallographic analysis

The powder X-ray diffraction patterns (PXRD) of bare MOF-5 and $M_xO_y@MOF-5$ in comparison with the simulated pattern of MOF-5 generated from original CIF file are represented in Fig. [1a](#page-3-0). It can be seen that all the characteristic diffraction peaks of MOF-5 are consistent with the simulated pattern and it reveals the successful synthesis of

Fig. 1 a PXRD patterns of pure MOF-5, Co₃O@MOF-5, NiO@MOF-5, CuO@MOF-5, and ZnO@MOF-5 compared with the simulated pattern of MOF-5; b PXRD patterns of pure NH₂-MOF-5, Co₃O₄@NH₂-MOF-5, NiO@NH₂-MOF-5, CuO@NH₂-MOF-5, and ZnO@NH₂-MOF-5

Zn-based metal-organic framework, MOF-5. After the incorporation of metal oxide nanoparticles into MOF-5, $M_xO_y@MOF-5$ spectra exhibit some new diffraction peaks besides the diffraction peaks of MOF-5, which are in good agreement with the diffraction pattern of metal oxide $(Co₃O₄$, NiO, CuO, and ZnO) nanoparticles. Figure 1b represents the crystallographic patterns of $NH₂-MOF-5$ and $M_xO_y@NH_2-MOF-5$ composites. It is observed that NH₂-MOF-5 has a different crystal structure and PXRD pattern as compared to MOF-5 due to the use of 2-aminoterephtalic acid instead of simple terephthalic acid. However, like $M_xO_y@MOF-5$, PXRD spectra of $M_xO_y@NH_2-MOF-5$ indicate that amine-functionalized MOF, $NH₂$ -MOF-5, maintains its crystal structure and remains intact after incorporation of metal oxide nanoparticles. No extra diffraction peaks for impurity are observed, which indicates the single-phase formation of all these samples. From powder XRD analysis of $M_xO_y@MOF-5$ and $M_xO_y@NH_2-MOF-5$, it can be concluded that incorporation of metal oxide nanoparticles does not significantly affect the crystalline structure of host MOFs, and there is no chemical interaction between incorporated nanoparticles and a central metallic cluster of MOFs; they are intact with each other via physical interaction.

Fourier-transform infrared spectroscopic analysis

The surface functional groups and formation of pure Zn-based MOFs MOF-5 and $NH₂$ -MOF-5 as well as their composites $M_xO_y@MOF-5$ and $M_xO_y@NH_2-MOF-5$ are confirmed by FTIR spectroscopy. Figure [2a](#page-4-0) represents the FTIR spectra of MOF-5 and $M_xO_y@MOF-5$ composites. In the FTIR spectrum of MOF-5, two strong transmittance peaks related to symmetric and asymmetric stretching vibrations of the carboxylic group are observed at 1575 and 1375 cm⁻¹. A weak transmittance peak can also be seen at 642 cm^{-1} due to symmetric stretching vibration of the central Zn_4O^{6+} cluster of MOF-5 $[37]$ $[37]$. However, in the FTIR spectrum of NH₂-MOF-5, a weak peak is observed at 1242 cm^{-1} for C–N stretching vibration of the aromatic amine group and two strong transmittance peaks are also observed at 1540 and 1378 cm^{-1} indexed to the vibrational motion of carboxylate group of 2 aminoterephthalic acid, shown by Fig. [2b](#page-4-0) [\[38\]](#page-11-0), while FTIR spectra of $M_xO_y@MOF-5$ and $M_xO_y@NH_2-MOF-5$ composites exhibit some additional peaks at 576, 517, 535, and 555 cm−¹ for vibrations of Co–O, Ni–O, Cu–O, and Zn–O [\[39](#page-11-0)–[41\]](#page-11-0), respectively in addition to vibrational bands of both Zn-based MOFs MOF-5 and NH₂-MOF-5, which indicates that metal oxide nanoparticles have been successfully incorporated into MOFs.

Morphological and compositional analysis

The morphology and composition of all the synthesized samples have been studied by scanning electron microscopy images, SEM-based energy-dispersive X-ray (EDX) analysis, and elemental mapping. Figure [3A](#page-4-0) (a–e) represents the SEM images of synthesized bare MOF-5 and $M_xO_y@MOF-5$ composites. It appears that synthesized MOF-5 has grown in well-defined rectangular morphology and is uniformly distributed, shown by Fig. [3A](#page-4-0) (a). Similarly, the incorporated samples, $M_xO_y@MOF-5$, have also grown in almost rectangular morphology but due to the incorporation of metal oxide nanoparticle coarse surfaces, variations in size and slight

Fig. 2 a FTIR spectrum of pure MOF-5, Co₃O@MOF-5, NiO@MOF-5, CuO@MOF-5, and ZnO@MOF-5; b FTIR spectrum of NH₂-MOF-5, Co3O4@NH2-MOF-5, NiO@NH2-MOF-5, CuO@NH2-MOF-5, and ZnO@NH2-MOF-5

agglomerations are also observed, represented by Fig. 3A (b-e). Figure $3A$ (a'–e') represents the SEM-based mix elemental maps of synthesized MOF-5 and its composites. It indicates that metal oxide nanoparticles have been successfully incorporated into cavities of MOF-5 and these particles are uniformly distributed.

The SEM images reveal that bare amine-functionalized MOF NH₂-MOF-5 and its composites $M_xO_y@NH_2-MOF-5$ have grown in spherical shapes with spongy smooth surfaces, as shown in Fig. $3B$ (a–e). However, as a result of the incorporation of metal oxide nanoparticles, some variations in size and agglomeration are observed in incorporated samples. Mix elemental maps of $M_xO_y@MOF-5$ and $M_xO_y@NH_2-MOF-5$ indicate that metal oxide nanoparticles have been successfully incorporated into cavities of both Zn-based MOFs and these particles are uniformly distributed, as shown in Fig. $3B$ (a'–e'). Individual elemental maps of basic components of bare MOFs (MOF-5 and

Fig. 3 A) SEM (a–e) and SEM-based mix elemental maps (a′–e′) of MOF-5, Co₃O@MOF-5, NiO@MOF-5, CuO@MOF-5, and ZnO@MOF-5, respectively; B) SEM (a–e) and SEM-based mix

elemental maps (a'-e') of NH₂-MOF-5, $Co₃O₄@NH₂$ -MOF-5, NiO@NH₂-MOF-5, CuO@NH₂-MOF-5, and ZnO@NH₂-MOF-5, respectively

 $NH_2-MOF-5$) and their composites $(M_xO_y@MOF-5$ and $M_xO_y@NH_2-MOF-5$) are shown in Figure S1 and S2.

The composition of all the synthesized samples has been studied through SEM-based EDX images. Figure S3 and S4 represent the EDX images of MOF-5/ $M_xO_y@MOF-5$ and $NH_2\text{-MOF-5}/M_xO_y@NH_2\text{-MOF-5}$, respectively. It is evaluated from EDX images of bare MOF-5 and NH₂-MOF-5 that they only contain diffraction peaks of basic elements Zn, C, and O of MOF-5 and Zn, C, O, and N of $NH₂-MOF-5$, respectively. Similarly, incorporated samples contain all basic diffraction peaks of their respective MOF and they also contain additional diffraction peaks of Co, Cu, Ni, and Zn. It indicates the successful incorporation of metal oxide nanoparticles into both Zn-based MOFs. No additional peak of impurity is observed, which indicates a high degree of purity of these synthesized samples.

Optical analysis

The UV-Vis absorption spectra of as-synthesized bare Zn-based MOF MOF-5 and amine-functionalized MOF NH₂-MOF-5 as well as their composites are represented by Fig. 4a, b. Figure 4a shows that MOF-5 exhibits maximum absorption at 207 nm with an absorption edge at 320 nm due to the organic linker terephthalate and inorganic central Zn_4O_{13} cluster. It indicates that in MOF-5 ligand-to-metal charge transfer (LMCT) takes place. Upon loading with metal oxide nanoparticles into MOF-5, a redshift is observed in the UV region with maximum absorption at 235, 213, 245, and 256 nm in $Co₃O₄@MOF-5$, NiO@MOF-5, CuO@MOF-5, and ZnO@MOF-5, respectively. It has been also observed that in $M_xO_y@MOF-5$, absorption edges have extended up to 400 nm and new absorption peaks are also

Fig. 4 a UV-Vis absorption spectra of MOF-5, $Co₃O@MOF-5$, NiO@MOF-5, CuO@MOF-5, and ZnO@MOF-5; b UV-Vis absorption spectra of NH₂-MOF-5, $Co_3O_4@NH_2-MOF-5$, NiO@NH₂-MOF-5, CuO@NH2-MOF-5, and ZnO@NH2-MOF-5; c Tauc plot of MOF-5,

Co₃O@MOF-5, NiO@MOF-5, CuO@MOF-5, and ZnO@MOF-5; d Tauc plot of NH₂-MOF-5, $Co₃O₄@NH₂$ -MOF-5, NiO@NH₂-MOF-5, CuO@NH₂-MOF-5, and ZnO@NH₂-MOF-5

observed in the visible region with λ_{max} at 695 and 754 nm in $Co₃O₄@MOF-5$ and CuO@MOF-5, respectively. The band gap of all these synthesized samples has been calculated from the Tauc plot, Fig. [4c](#page-5-0), and is 4.03,3.01, 3.94, 3.48, and 3.70 eV for MOF-5, Co₃O₄@MOF-5, NiO@MOF-5, CuO@MOF-5, and ZnO@MOF-5, respectively. All these MOF-5-based samples exhibit low efficiency towards water splitting due to a large effective band gap. Therefore, their efficiency has been further improved by the amine-functionalization of MOF-5. The water-splitting efficiency of $NH₂$ -MOF-5 is significantly improved due to amine functionality. $NH₂$ -MOF-5 exhibits absorption edge at around 438 nm and upon loading with metal oxide nanoparticles, $M_xO_y@NH_2-MOF-5$ composites, absorption edges extend up to 600 nm, falling in the visible region, as shown in Fig. [4b](#page-5-0). So, a significant redshift is observed. Their calculated band gaps are 3.45, 2.78, 3.39, 3.23, and 2.89 eV for $NH_2-MOF-5$, $Co_3O_4@NH_2-MOF-5$, $NiO@NH_2-MOF-5$, $CuO@NH_2-MOF-5$, and $ZnO@NH_2-MOF-5$, respectively. It indicates that amine functionalization reduces the band gap, increases the charge separation, and improves the catalytic activity.

Photoelectrochemical studies towards oxygen evolution reaction

Photoelectrochemical studies towards oxygen evolution reaction are studied by cyclic voltammetry (CV) and linear sweep voltammetry (LSV) at 30 and 1 mV s^{-1} , respectively. Photoelectrochemical studies towards OER are studied both in dark and in the presence of visible light. Figure [5a](#page-7-0) represents the CV curves of MOF-5/NF and $M_xO_y@MOF-5/NF$ electrodes. It is observed that almost zero current density is produced in dark due to negligible OER activity. However, all the synthesized samples exhibit anodic current density under illumination which further increases gradually with the incorporation of pre-synthesized nanoparticles. The bare MOF-5/NF generates only 0.012 mA cm−² current density and after incorporation of metal oxide nanoparticles, a tremendous increase in current density has been observed in $M_xO_y@MOF-5/NF$ electrodes and among all these catalysts, $Co₃O₄@MOF-5/NF$ shows the lowest onset potential and generates maximum current density for OER. The catalytic OER activity of these synthesized samples has been further increased by amine functionalization of terephthalic acid. Figure [5b](#page-7-0) represents the CV curves of synthesized NH₂-MOF-5/NF and $M_xO_y@NH_2-MOF-5/$ NF. It can be observed from CV curves that after amine functionalization, lower onset potential and higher current density are achieved towards OER as compared to bare MOF-5/NF and $M_xO_y@$ MOF-5/NF, respectively. Like $M_xO_y@MOF-5/NF$, among $M_xO_y@NH_2-MOF-5/NF$ electrodes, $Co_3O_4@NH_2-MOF-5/NF$ shows the lowest onset potential and generates the highest 34.48 mA cm−² current

density as compared to bare NH2-MOF-5/NF (6.33 mA cm^{-2}), NiO@NH₂-MOF-5/NF (8.06 mA cm⁻²), $CuO@NH₂-MOF-5/NF (16.77 mA cm⁻²)$, and ZnO@NH₂-MOF-5/NF (20.67 mA cm⁻²), respectively.

Figure [5c and d](#page-7-0) represent the LSV curves of bare MOF-5 and amine-functionalized MOF, NH₂-MOF-5/NF-based electrodes, and it can be seen that amine-functionalized-based electrodes exhibit better OER activity as compared to nonamine-functionalized working electrodes. It is also observed that the $Co₃O₄@NH₂-MOF-5/NF$ electrode exhibits the largest current density (32.93 mA cm⁻² at 1.49 V vs. RHE) for OER as compared to all other electrodes. For comparison, overpotential (η) requires to achieve 5 mA cm⁻² current density is considered, represented by Fig. [5e, f.](#page-7-0) It is found that $Co₃O₄@NH₂-MOF-5/NF$ needs only 203 mV overpotential to deliver 5 mA cm^{-2} as compared to all other aminefunctionalized-based electrodes NH₂-MOF-5/NF ($\eta_5 = 353$) mV), NiO@NH₂-MOF-5/NF ($\eta_5 = 243$ mV), CuO@NH₂-MOF-5/NF (η ₅ = 223 mV), and ZnO@NH₂-MOF-5/NF (η ₅ $= 333$ mV) as well as non-amine-functionalized-based electrodes $Co₃O₄@MOF-5/NF$ (η ₅ = 323 mV), NiO@MOF-5/NF $(\eta_5 = 353 \text{ mV})$, CuO@MOF-5/NF ($\eta_5 = 343 \text{ mV}$), and ZnO@MOF-5/NF ($\eta_5 = 333$ mV). Furthermore, the overpotential required to reach the benchmark of 10 mA cm⁻² current density is compared and it has been found that $Co₃O₄@NH₂$ -MOF-5/NF requires the lowest 223 mV overpotential to deliver 10 mA cm^{-2} as compared to both amine-functionalized and non-aminefunctionalized-based electrodes CuO@NH2-MOF-5/NF $(\eta_{10} = 253 \text{ mV})$, ZnO@NH₂-MOF-5/NF ($\eta_{10} = 353 \text{ mV}$), $Co₃O₄@MOF-5/NF$ ($\eta_{10} = 343$ mV), NiO@MOF-5/NF $(\eta_{10} = 363 \text{ mV})$, and CuO@MOF-5/NF $(\eta_{10} = 353 \text{ mV})$, represented by Fig. 6a. This indicates that the $Co₃O₄@NH₂-MOF-5/NF$ electrode is a highly efficient photoelectrocatalyst for photoelectrochemical OER as compared to all other synthesized photoelectrocatalysts as well as previously reported Co-based and MOF-based OER catalysts, as represented in Table [2](#page-8-0).

Tafel slope is an important parameter to understand the reaction kinetics. So, the Tafel plot is derived from LSV curves of MOF-5/NF and NH₂-MOF-5/NF-based electrodes, as shown by Fig. [6b, c](#page-8-0). It is revealed from Tafel plots that amine-functionalized electrodes show a lower Tafel slope as compared to non-amine-functionalized-based electrodes, which indicates that amine-functionalized electrodes have more favorable electron transfer and better OER activity. It is observed that $Co₃O₄@NH₂$ -MOF-5/NF exhibits the lowest Tafel slope 52 mV dec^{-1} as compared to all other synthesized catalysts as well as previously reported MOF-based OER catalysts as shown in Table [2.](#page-8-0) This result agrees with the highest current density of $Co₃O₄@NH₂$ -MOF-5/NF towards OER activity. So, it can be concluded that $Co₃O₄@NH₂$ -MOF-5/NF is a highly efficient catalyst for OER as compared to all

Fig. 5 a and b CV curves of bare MOF-5/NF, NH₂-MOF-5/NF, and their composites, respectively; c and d LSV curves of bare MOF-5/NF, NH₂-MOF-5/NF, and their composites, respectively; e and f overpotential required to 5 mA cm−² current density for MOF-5- and NH2-MOF-5-based composites

synthesized photoelectrocatalysts. It is expected that in basic medium, OER is occurred by the following equations

$$
OH \rightarrow OH_{ads} + e^{-}
$$
 (4)

$$
\rm OH_{ads} + OH^{-} {\rightarrow} O_{ads} + H_{2}O + e^{-} \qquad \qquad (5)
$$

$$
O_{ads} + OH^- \rightarrow OOH_{ads} + e^-
$$
 (6)

$$
\mathrm{OOH}_{ads} + \mathrm{OH}^- {\rightarrow} \mathrm{O}_{2(g)} + e^- \qquad \qquad (7)
$$

So, this overall catalytic OER study reveals that amine functionalization of MOF-5 by using 2 aminoterephthalic acid has shifted the band gap towards the visible region, increased the charge separation, enhanced the absorption of visible light, and improved the photoelectrocatalytic OER activity. Thus, amine functionalization of MOF-5 has significantly enhanced the OER activity via water splitting.

Table 2 Comparison of the OER activity of $Co_3O_4@NH_2-MOF-5/NF$ with previously reported Co- and MOF-based materials in literature

The stability of the working electrode is another important parameter for photoelectrochemical studies. Therefore, the stability of $Co_3O_4@MOF-5/NF$ and $Co_3O_4@NH_2-MOF-5/$ NF is studied by 1000 continuous LSV sweeps. Figure [7a](#page-9-0) [and b](#page-9-0) represent the 1st and 1000th LSV cycle of $Co₃O₄@MOF-5/NF$ and $Co₃O₄@NH₂-MOF-5/NF$, respectively. It reveals that there is no degradation in current density before and after the 1000th LSV cycle. In addition to 1000 LSV cycles, long-term photoelectrocatalytic stability is also important for commercial applications. The long-term stability of all the synthesized samples is carried out for 6000 s by using the chronoamperometry technique. Figure [7c and d](#page-9-0) show the current vs. time graph of all the synthesized working electrodes at the static potential of 1.5 V vs. RHE for MOF-5-based and NH_2 -MOF-5based working electrodes, respectively in the presence of visible light. The stability study reveals the satisfactory and superior stability of $Co₃O₄@NH₂-MOF-5/NF$ photoelectrocatalyst towards OER in alkaline solution.

Another important parameter to study reaction kinetics is impedance spectroscopy (EIS). It gives an idea about solution induced resistance as well as resistance at electrode/electrolyte

 $b)_{0.36}$

 0.3

 0.32

 0.30

 0.25

 0.26

 0.22

Co₃O₄@MOF-5/NF

Co.O.@NH.-MOF-5/N

CuO@NH₂-MOF-5/NF

CuO@MOF-5/NF

ZnO@MOF-5/NF ZnO@NH₃-MOF-5/NF Co.O.@MOF-5/N

NiO@MOF-5/NF

CuO@MOF-5/NF

ZnO@MOF-5/NF

a)

 351

30

 \sum_{E}^{250}

음 150

50

interface (charge transfer resistance). The small diameter of semicircle indicates low induced resistance, low charge transfer resistance, and higher catalytic activity. Figure S5 represents the Nyquist plot of $Co₃O₄@NH₂$ -MOF-5/NF and bare MOF NH₂-MOF-5. It shows that $Co₃O₄@NH₂$ -MOF-5/NF has a small diameter of the semicircle as compared to bare MOF NH₂-MOF-5/NF, which reveals that $Co₃O₄@NH₂$ -MOF-5/NF has low charge transfer resistance, maximum conductance, and better OER activity as compared to bare MOF NH₂-MOF-5/NF.

Proposed mechanism of photoelectrochemical oxygen evolution reaction

In MOFs, organic linkers act as an antenna, capture the visible light, and generate the photogenerated electrons. Therefore, it is hypothesized that photogenerated electrons are transferred from the valence band of NH2-MOF-5 to its conduction band and subsequently transferred to the energy level of incorporated $Co₃O₄$ nanoparticles. Furthermore, these electrons are transferred to the counter electrode via an external circuit and increased the charge separation, reduced the electron-

NH_-MOF-5/

 0.3

 $\frac{8}{5}$ 0.25

 0.15

 0.10

 \overline{a} 0.30

č 0.20

r
62mVdec

- Co₃O₄@NH₂-MOF-5/NF
- NiO@NH₂-MOF-5/NF

ZnO@NH_-MOF-5/NF

MOE-S/NE

73mVdec

116mVdec

67mVdec

m
Vder

$\overline{0.8}$ -0.4 0.0 0.0 0.2 0.4 log j 0.6 1.0 1.2 0.4
Iog j mAcm 1.2 -0.2 0.8 $\ddot{ }$ 10mAcm⁻² Current density **Fig. 6** a Overpotential required to achieve the 10 mA cm⁻² current density; **b** Tafel plot of MOF-5-based composites; **c** Tafel plot of NH₂-MOF-5-based samples

115mVder

104mVd

Fig. 7 a and b 1st and 1000th LSV cycle of Co₃O₄@MOF-5/NF and Co₃O₄@NH₂-MOF-5/NF, respectively; c and d long-term stability test of all MOF-5-based and NH2-MOF-5-based photoelectrocatalysts

hole pair recombination, and enhanced the catalytic activity. The resulting holes at the surface of the photoanode cause the photooxidation of water into O_2 and H⁺. The H⁺ is transported

towards the counter electrode via electrolyte where they undergo photoreduction by photogenerated electrons and generate hydrogen gas, as shown in Fig. 8.

Fig. 8 Proposed mechanism for electron transfer and oxygen evolution reaction by Co₃O₄@NH₂-MOF-5 under sunlight exposure

Conclusion

In summary, the photoelectrocatalytic activity towards water oxidation (OER) of MOF-5-based photoelectrocatalysts has been successfully increased by amine functionalization. So, $NH_2-MOF-5/NF$ and $M_xO_y@NH_2-MOF-5/NF$ photoelectrocatalysts exhibit better and enhanced catalytic activity towards OER as compared to MOF-5/NF and $M_xO_y@MOF-5/NF$ catalysts. Among all the synthesized photoelectrocatalysts, $Co₃O₄@NH₂$ -MOF-5/NF exhibits excellent catalysis towards OER in alkaline medium and it requires the lowest overpotential 203 mV and 223 mV to delivers 5 and 10 mA cm−² , respectively. In addition, it exhibits the lowest Tafel slope of 52 mV dec⁻¹ and excellent long-term stability as compared to all other synthesized catalysts. A large number of active sites, fast electron transfers, and reduce electron-hole pair recombination lead to enhance the OER activity of $Co₃O₄@NH₂$ -MOF-5/NF as compared to all other synthesized catalysts and even some of the previously reported catalysts. This study encourages the development of more nanoparticles incorporated amine-functionalized MOF-based photoelectrocatalysts and used them in electrocatalytic applications.

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Compliance with ethical standards

Conflict of interest The authors declare that they have no conflict of interest.Supplementary Information The online version contains supplementary material available at [https://doi.org/10.1007/s11581-020-](https://doi.org/10.1007/s11581-020-03866-1) [03866-1](https://doi.org/10.1007/s11581-020-03866-1).

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