



# String-like core-shell $\text{ZnCo}_2\text{O}_4@ \text{NiWO}_4$ nanowire/nanosheet arrays on Ni foam for binder-free supercapacitor electrodes

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## Abstract

We report the synthesis of 3D, string-like, core-shell, heterostructured  $\text{ZnCo}_2\text{O}_4@ \text{NiWO}_4$  nanowire/nanosheet arrays on nickel foam using a simple two-step hydrothermal method followed by an annealing process, without the needs of conventional electrode preparation. The direct growth of  $\text{ZnCo}_2\text{O}_4$  nanowires serves both as core and main active materials for charge storage with high mechanical stability, while the higher conductive  $\text{NiWO}_4$  nanosheets act as the shell, which could provide enough voids for electrolyte accommodation and facilitate charge transfer and storage. When working as a binder-free electrode, the as-synthesized  $\text{ZnCo}_2\text{O}_4@ \text{NiWO}_4$  nanowire/nanosheet arrays electrode delivers a specific capacitance as high as  $1782 \text{ F g}^{-1}$  ( $2.14 \text{ F cm}^{-2}$ ) at a current density of  $1 \text{ mA cm}^{-2}$  and retains 95.4% of original capacitance after 5000 cycles at  $5 \text{ A cm}^{-2}$ , compared with single  $\text{ZnCo}_2\text{O}_4$  nanowires electrode with  $1300 \text{ F g}^{-1}$  ( $1.04 \text{ F cm}^{-2}$ ) and 90.9%. The excellent capacitance performance demonstrates that constructing 3D, heterostructured, core-shell, nanowire/nanosheet arrays with two different binary transition metal oxides holds great potential for high-performance supercapacitors due to the synergic effects and large specific surface area.

**Keywords** Hydrothermal synthesis ·  $\text{ZnCo}_2\text{O}_4@ \text{NiWO}_4$  · Core-shell structure · Supercapacitor

## Introduction

Supercapacitors, as one of the most promising energy storage devices, have attracted more and more significant research attention in the past few years attributed to extremely higher energy density compared with traditional electrostatic capacitors as well as higher power density, more robust cycling life, and much more excellent reversible charge/discharge characteristics than the commonly used lithium-ion batteries [1–3]. However, their low energy densities are still the main issue, preventing them from many

practical applications. To date, the investigation of supercapacitor electrodes is mainly concentrated on carbon materials and transition metal oxides and hydroxides. Generally, transition metal oxides/hydroxides, which could exhibit exceptionally high specific capacitance and energy density that surpass carbon-based active materials because they can provide efficient faradaic redox reactions, are considered promising candidate of supercapacitive electrodes and thus have been extensively investigated [4]. Recently, many binary transition metal oxides, e.g.,  $\text{NiCo}_2\text{O}_4$  [5],  $\text{ZnCo}_2\text{O}_4$  [6],  $\text{NiMoO}_4$  [7],  $\text{CoMoO}_4$  [8],  $\text{MnMoO}_4$  [9] and  $\text{Zn}_2\text{GeO}_4$  [10], and  $\text{MnCo}_2\text{O}_4$  [11], show higher electrochemical performance than single-transition metal oxides because they could provide more variable oxidation states and enhance electrical conductivities. Among them,  $\text{ZnCo}_2\text{O}_4$ , obtained by replacing one Co atom of the cobalt oxide with Zn atom, could produce high specific capacitance comparable with other binary transition metal oxides. Multifarious  $\text{ZnCo}_2\text{O}_4$ -based materials with novel nanostructure arrays (i.e., nanowires [12], nanosheets [13], nanorods [14], and nanoparticles [15]) have been successfully synthesized by different methods, such as solvothermal method [16] and simple polyol refluxing

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process [6]. For instance, cabbage-like  $\text{ZnCo}_2\text{O}_4$  achieves areal capacitance  $789.11 \text{ mF cm}^{-2}$  as positive electrode [17] and hexagonal-like  $\text{ZnCo}_2\text{O}_4$  reaches specific capacitance  $845.7 \text{ F g}^{-1}$  [18]. The obtained capacitance is still below theoretical level due to inefficient usage of active material since it still suffers from limited surface area and poor electrical conductivity [19]. Therefore,  $\text{ZnCo}_2\text{O}_4$  combined with high conductive carbon-based materials, such as reduced graphene (rGO) [20, 21] and carbon nanotubes [22], has been developed to further improve the electrochemical performance. For instance, the specific capacitance of the nanocomposite that  $\text{ZnCo}_2\text{O}_4$  combined with rGO could reach as high as  $704.2 \text{ F g}^{-1}$  [20]. Despite the great achievements being made, the specific capacitance of the developed electrode is far from the expected level for practical application. Recently, metal oxides featuring core/shell nanostructure, e.g.,  $\text{ZnCo}_2\text{O}_4@ \text{NiO}$  [23],  $\text{ZnCo}_2\text{O}_4@ \text{Ni}_x\text{Co}_{2-x}(\text{OH})_{y6x}$  [24],  $\text{NiCo}_2\text{S}_4@ \text{NiMoO}_4$  [25],  $\text{NiMoO}_4@ \text{CoMoO}_4$  [26], and  $\text{MnMoO}_4 \cdot \text{H}_2\text{O}@ \text{MnO}_2$  [27], have received extensive attention, because the integration of two kinds of pseudocapacitive materials not only increase the loading mass of active materials with high surface area but also significantly enhance the supercapacitive properties due to synergic effects. Recently, binary transition metal oxide  $\text{NiWO}_4$  with a wolframite-type structure, showing higher electrical conductivity than that of  $\text{NiO}$  with the incorporation of the W atoms [21, 28, 29], has been developed for the potential application of sensors [30], catalysis [31], and electrocatalysts [32]. Therefore, constructing  $\text{ZnCo}_2\text{O}_4@ \text{NiWO}_4$  core-shell nanostructured electrode is promising because of the large specific surface area of 3D nanostructure and synergic effects of two different binary transition metal oxides. In this paper, a hierarchical core-shell-nanostructured electrode is fabricated by integrating two different binary transition metal oxide  $\text{ZnCo}_2\text{O}_4$  nanowires with  $\text{NiWO}_4$  nanosheets on Ni foams, which have never been constructed for high-performance supercapacitors. The  $\text{ZnCo}_2\text{O}_4@ \text{NiWO}_4$  core-shell nanowire/nanosheet arrays grown on Ni foam are fabricated via mild hydrothermal and calcination methods with predictable excellent performance.  $\text{ZnCo}_2\text{O}_4$  nanowires could be directly grown on the surface of Ni foam, and then, well-dispersed  $\text{NiWO}_4$  nanosheets are vertically anchored on the surface of  $\text{ZnCo}_2\text{O}_4$  nanowires, avoiding the aggregation of  $\text{NiWO}_4$  particles. Generally, Ni foam with 3D structure could provide large surface area and load more active material per unit area. The direct growth of  $\text{ZnCo}_2\text{O}_4$  nanowires serves both as core and main active materials for charge storage, and the strong adhesion between  $\text{ZnCo}_2\text{O}_4$  nanowires and Ni foam could provide higher mechanical stability and shorten the electron transfer paths. The  $\text{NiWO}_4$  nanosheets act both as the shell and

high surface area active materials, and the well-distributed nanostructure with abundant free space and high electrical conductivity could provide enough voids for electrolyte accommodation and facilitate charge transfer and storage.

## Experimental details

### Material preparation

The raw materials of  $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ ,  $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ ,  $\text{NiCl}_2 \cdot 6\text{H}_2\text{O}$ ,  $\text{Na}_2\text{WO}_4 \cdot 4\text{H}_2\text{O}$ , urea ( $\text{CO}(\text{NH}_2)_2$ ),  $\text{NH}_4\text{F}$ , and  $\text{KOH}$  with analytical pure reagent grade, which were purchased from Sino pharm Chemical Reagents, were used in the experiments without any further purification.

### Pretreatment of Ni Foam

In the pretreatment procedure, Ni foams with a thickness of 1 mm were sheared to a size of  $3 \times 1 \text{ cm}^2$ , carefully treated with acetone under ultrasonic cleaning for 15 min to wipe away the oil contamination on the surface, and then were immersed in 1 M  $\text{HCl}$  for ultrasonic cleaning to dislodge the surface oxide layers. Ni foams were rinsed in anhydrous ethanol and deionized water for 15 min by ultrasonication, respectively. The wet Ni foams were taken out and then dried for 12 h at  $60^\circ\text{C}$  in a vacuum.

### Preparation of $\text{Ni}@ \text{ZnCo}_2\text{O}_4$ nanowire arrays

To synthesis  $\text{ZnCo}_2\text{O}_4$  nanowires on Ni foam as a core in the core/shell nanostructured electrode,  $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$  (0.297 g),  $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$  (0.582 g),  $\text{NH}_4\text{F}$  (0.074 g), and  $\text{CO}(\text{NH}_2)_2$  (0.3 g) were dissolved in 35 mL deionized water to form a homogeneous and mixed solution under robust stirring. The pretreated nickel foam and the obtained solution were mixed into a 50-mL Teflon-lined stainless autoclave and kept at  $120^\circ\text{C}$  for 5 h. After that, the Zn-Co precursor coated on Ni foam was washed with absolute ethanol and deionized water to remove the remaining reactants. After being dried at  $60^\circ\text{C}$  for 12 h in a vacuum oven, the Zn-Co precursor active material was calcined at  $400^\circ\text{C}$  for 2 h in air. The loading mass of the  $\text{ZnCo}_2\text{O}_4$  active material can be obtained by measuring the weight change of the examples before and after hydrothermal and calcination process. The mass loading of  $\text{ZnCo}_2\text{O}_4$  on Ni foam was  $0.8 \text{ mg cm}^{-2}$ .

### Preparation of $\text{ZnCo}_2\text{O}_4@ \text{NiWO}_4$ core-shell nanowire/nanosheet arrays

In a typical synthesis,  $\text{NiCl}_2 \cdot 6\text{H}_2\text{O}$  (0.475 g) was dissolved into 30 mL distilled water by stirring for 10 min. Then, 30 mL distilled water containing  $\text{Na}_2\text{WO}_4 \cdot 4\text{H}_2\text{O}$  (0.659 g) was

added dropwise and stirred for another 10 min. Then, the  $\text{ZnCo}_2\text{O}_4$  on Ni foam was soaked into above solution in a Teflon-lined stainless steel autoclave, which was maintained at 140 °C for 4 h to fabricate  $\text{ZnCo}_2\text{O}_4@NiWO_4$  core-shell nanostructure. Subsequently, the obtained  $\text{ZnCo}_2\text{O}_4@NiWO_4$  precursor was respectively washed in distilled water and anhydrous ethanol and then dried at 60 °C overnight for next fabrication step. Finally, the precursor was annealed at 350 °C in air for 2 h to obtain  $\text{ZnCo}_2\text{O}_4@NiWO_4$  core-shell nanowire/nanosheet arrays. The mass of core-shell hierarchical nanostructure loaded on nickel foam was measured to be  $1.2 \text{ mg cm}^{-2}$ .

## Materials characterization

The crystal structure and chemical composition were evaluated by X-ray diffraction (XRD, Rigaku D/MAX D/MAX-3B) using Cu-K $\alpha$  radiation (1.5418 Å) at 40 kV, 100 mA. The morphological and nanostructured information of the samples were examined using field emission scanning electron microscopy (FESEM, SU 8010) and transmission electron microscope (TEM, JEOL/JEM 2100). The chemical elements and states were investigated using an X-ray photoelectron spectroscopy (XPS, PHI5600 XPS system).

## Electrochemical measurements

Electrochemical performance of all samples was investigated in 2 M KOH aqueous solution using an electrochemical workstation (Zahner). The information of cyclic voltammetry (CV), galvanic charging/discharging (GCD), and electrochemical impedance spectroscopy (EIS) were obtained in a three-electrode system, where the as-prepared active materials ( $\text{ZnCo}_2\text{O}_4$  and  $\text{ZnCo}_2\text{O}_4@NiWO_4$  nanostructures) coated on nickel foam, Pt foil, and Hg/HgO were used as the working, counter, and reference electrodes, respectively. The CV property was respectively recorded from 0 to 0.8 V at different scan rates ranging from 5 to 50  $\text{mV s}^{-1}$ . The GCD property was respectively studied in a potential between 0 and 0.5 V at the current densities ranging from 1 to 10  $\text{mA cm}^{-2}$ . Electrochemical impedance spectra (EIS) were studied with a 5-mV AC voltage in a specific frequency range beginning from 0.01 Hz and ending to 100 kHz. The total area of the testing electrodes was approximately fixed to be  $\sim 1 \times 1 \text{ cm}^2$ .

## Hybrid supercapacitor assembly and measurements

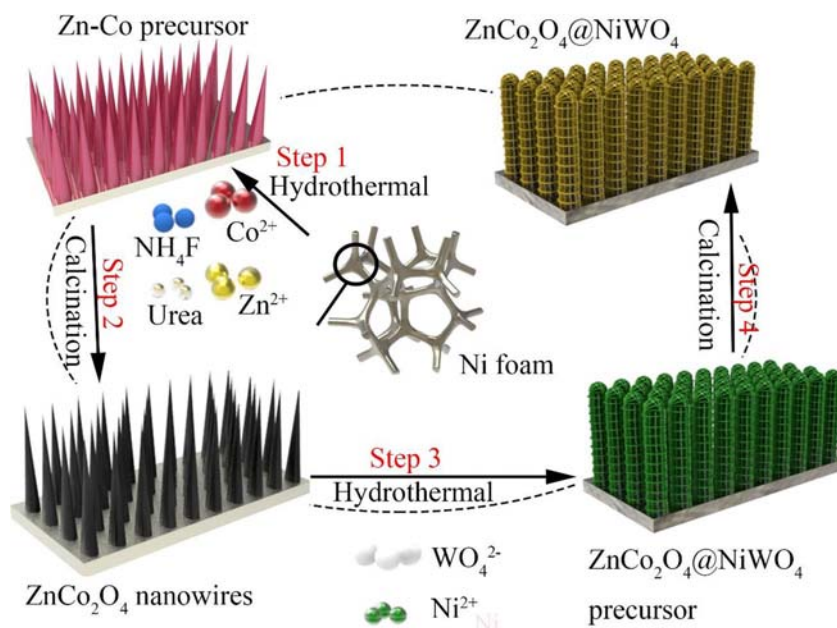
The asymmetric supercapacitor (ASC) was obtained by assembling  $\text{ZnCo}_2\text{O}_4@NiWO_4$  nanowire/nanosheet arrays to be positive electrode and activated carbon (AC) as the negative electrode. Firstly, 80% of AC was mixed with 10% of acetylene black and 10% of polytetrafluoroethylene (PTFE) binder with the aid of NMP solvent. Then, the obtained

mixture was loaded on the nickel foam at a pressure of 6 MPa and subsequently being dried in a vacuum oven at 60 °C overnight to form negative AC electrode. The loading mass of AC on nickel foam was controlled to be 4 mg as the negative electrode material. To prepare the polyvinyl alcohol-potassium hydroxide (PVA/KOH) electrolyte, 6 g PVA was dissolved into a 2 M KOH solution (60 mL) and heated at 85 °C under vigorous stirring until a homogeneous viscous solution appeared. The  $\text{ZnCo}_2\text{O}_4@NiWO_4$  and AC electrodes were firstly immersed in 2 M KOH solution and maintained for 12 h. After, the assembled electrodes were taken out and immersed into the as-prepared PVA/KOH electrolyte with a cellulose separator for 10 min. Finally, an all-solid-state ASC was obtained by assembling the electrodes and cellulose separator into a sandwich structure, which was measured in a two-electrode system.

## Results and discussion

Figure 1 shows the fabrication procedure of  $\text{ZnCo}_2\text{O}_4@NiWO_4$  core-shell nanostructure. Initially, the  $\text{ZnCo}_2\text{O}_4$  nanowire arrays, obtained from the first hydrothermal treatment followed by thermal treatment, were highly ordered aligned on Ni foam. Then, the  $NiWO_4$  nanosheets were loaded on the surface of  $\text{ZnCo}_2\text{O}_4$  nanowire arrays via the second hydrothermal method followed by annealing treatment. Figure 2 a presents top-view photograph of Ni foam before and after the loading of active materials. The color of Ni foam is changed from silver gray to pinkish purple due to the formation of Zn-Co precursor on the nickel foam. The surface of Ni foam turns from pinkish purple to black after thermal treatment. Figure 2 b–g show the morphologies of the  $\text{ZnCo}_2\text{O}_4$  nanowires and the  $\text{ZnCo}_2\text{O}_4@NiWO_4$  core-shell nanostructure. Clearly,  $\text{ZnCo}_2\text{O}_4$  nanowire arrays are uniformly and densely coated on nickel foam to guarantee the enough mass of active materials shown in Fig. 2b, c. Close observation of the  $\text{ZnCo}_2\text{O}_4$  nanowire finds that the  $\text{ZnCo}_2\text{O}_4$  nanowires are uniform in size, with average diameters of 100–200 nm and the length up to several microns (Fig. 2d). The well-separated growth of  $\text{ZnCo}_2\text{O}_4$  nanowires on Ni foam can not only highly enhance mechanical adhesion between active materials and Ni foam but can also be accessible to the electrolyte, which reduces the contact resistance and leads to high capacitive performance. Correspondingly, yellow color can be seen on the current collector indicating the  $NiWO_4$  covering on Ni foam as shown in Fig. 2a. Furthermore, the well-separated  $\text{ZnCo}_2\text{O}_4$  nanowires with much open spaces can act as the backbone to guide the growth of  $NiWO_4$  nanosheet shell. As shown in Fig. 2e–g, the surfaces of the  $\text{ZnCo}_2\text{O}_4$  nanowire are covered by ultrathin and interconnected  $NiWO_4$  nanosheets after the second hydrothermal treatment. The morphological and nanostructured features

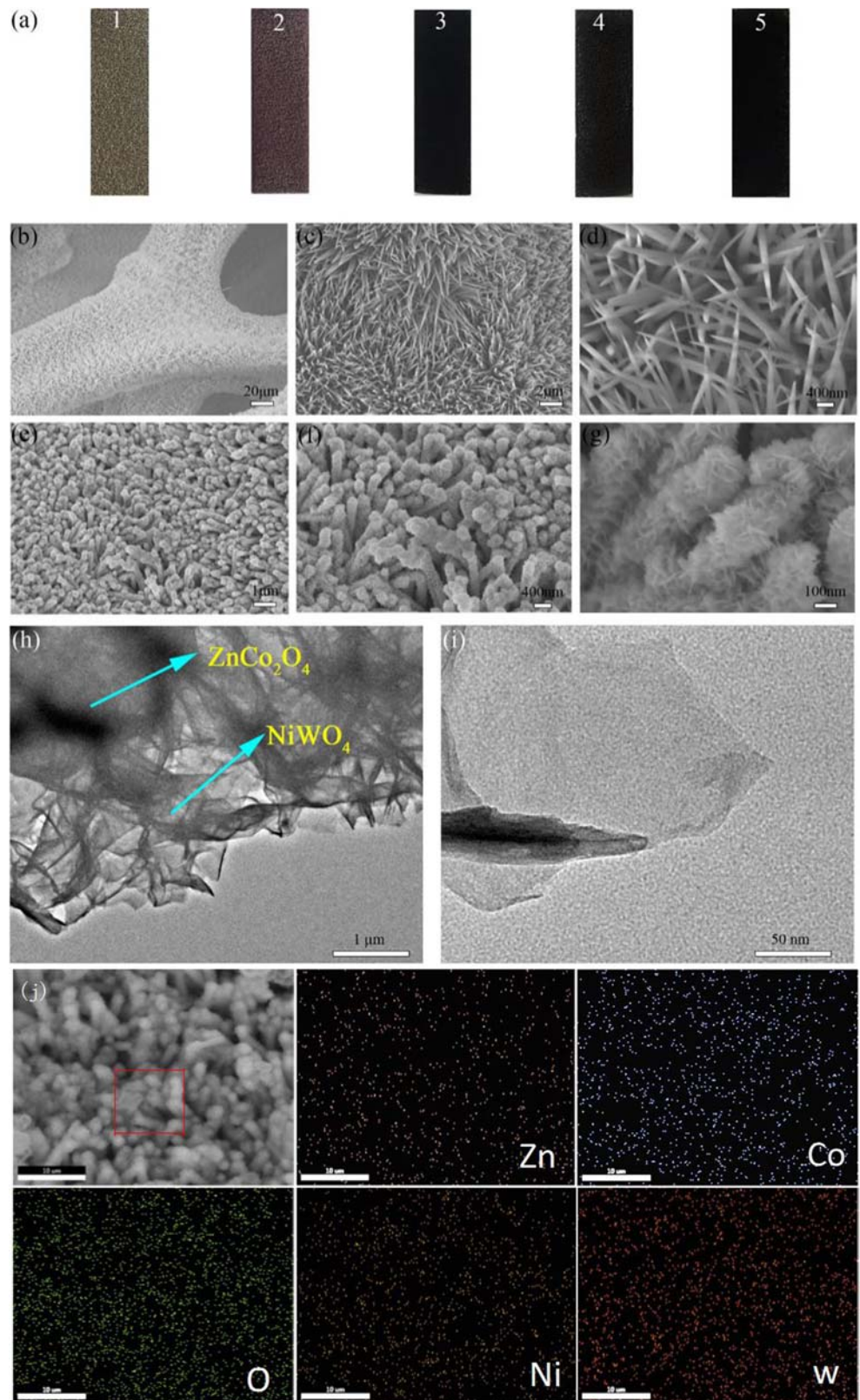
**Fig. 1** Fabrication procedure for the  $\text{ZnCo}_2\text{O}_4@\text{NiWO}_4$  integrated electrode



of  $\text{ZnCo}_2\text{O}_4@\text{NiWO}_4$  core/shell were further characterized by TEM images as shown in Fig. 2h, i. The  $\text{ZnCo}_2\text{O}_4$  nanowire substrates are typically coated with nanosized  $\text{NiWO}_4$ . The Zn, Co, O, Ni, and W elements can be clearly observed by EDS mapping as shown in Fig. 2j. Thus, the formation of the  $\text{ZnCo}_2\text{O}_4@\text{NiWO}_4$  nanostructure was further confirmed. Moreover, the ultrathin  $\text{NiWO}_4$  nanosheets are highly interconnected on  $\text{ZnCo}_2\text{O}_4$  nanowires. The 3D core/shell hierarchical structure can guarantee abundant free space, which can offer more electroactive surface sites and further enhance the electrochemical capacitance. The crystal phases of as-prepared  $\text{ZnCo}_2\text{O}_4$  nanowires and  $\text{ZnCo}_2\text{O}_4@\text{NiWO}_4$  core-shell nanowire/nanosheet arrays were investigated by XRD shown in Fig. 3. The diffraction peaks of  $\text{ZnCo}_2\text{O}_4$  nanowires at  $18.9^\circ$ ,  $31.2^\circ$ ,  $36.8^\circ$ ,  $38.5^\circ$ ,  $44.7^\circ$ ,  $55.6^\circ$ ,  $59.3^\circ$ ,  $65.1^\circ$ ,  $74^\circ$ , and  $77.2^\circ$  correspond well to the planes of (111), (220), (311), (222), (400), (331), (422), (511), (440), and (531), respectively, for the standard cubic spinel of  $\text{ZnCo}_2\text{O}_4$  (JCPDS card no. 23-1390). The hybrid  $\text{ZnCo}_2\text{O}_4@\text{NiWO}_4$  presents the diffraction peaks at  $15.6^\circ$ ,  $19.3^\circ$ ,  $23.9^\circ$ ,  $24.9^\circ$ ,  $30.9^\circ$ ,  $36.6^\circ$ ,  $39.1^\circ$ ,  $41.6^\circ$ ,  $44.7^\circ$ ,  $46.4^\circ$ ,  $49^\circ$ ,  $52.3^\circ$ ,  $54.6^\circ$ ,  $62.3^\circ$ ,  $62.7^\circ$ ,  $65.5^\circ$ ,  $65.8^\circ$ ,  $68.9^\circ$ ,  $72.3^\circ$ , and  $72.6^\circ$  and can be recognized as the planes of (010), (100), (011), (110), ( $-111$ ), (020), (002), (200), ( $-102$ ), (210), ( $-211$ ), ( $-112$ ), ( $-202$ ), (212), (300), (032), ( $-113$ ), ( $-141$ ), ( $-302$ ), and ( $-312$ ) for  $\text{NiWO}_4$  (JCPDS card no. 15-0755) [28]. Notably, the weak diffraction peaks indicate that as-prepared  $\text{ZnCo}_2\text{O}_4@\text{NiWO}_4$  electrode consists of very small crystallines, which are usually linked with high electrochemical performance [33, 34]. The surface composition and chemical states of as-prepared  $\text{ZnCo}_2\text{O}_4@\text{NiWO}_4$  electrode was investigated by XPS shown in Fig. 4. In Fig. 4a, the C 1s peak is related to underlying carbon tape for mounting samples, while the characteristic

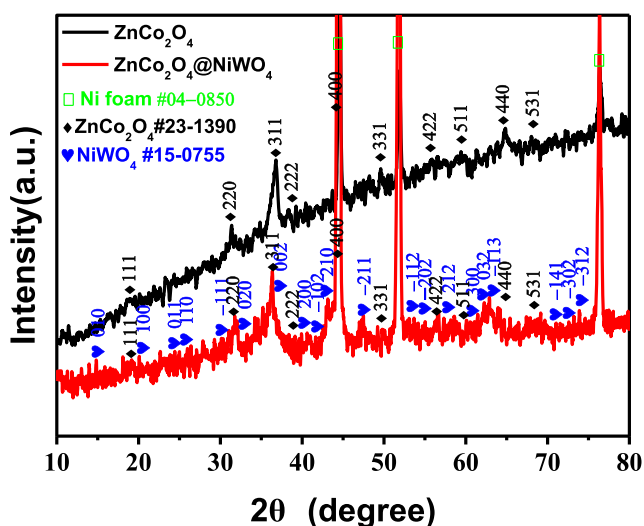
peaks of Zn, Co, O, Ni, and W elements are consistent with the core/shell structure of  $\text{ZnCo}_2\text{O}_4@\text{NiWO}_4$ . Figure 4 b exhibits a major peak at the binding energy of 1024.5 eV, which can be associated with the Zn  $2p_{3/2}$  [19]. Figure 4 c exhibits two major peaks located at the binding energy of 780.4 and 795.2 eV, respectively. The two peaks are related to the Co  $2p_{3/2}$  and Co  $2p_{1/2}$  energy level, respectively, indicating the characteristics oxidation states  $\text{Co}^{2+}$  and  $\text{Co}^{3+}$  [13]. As shown in Fig. 4d, the two main peaks at 529.2 and 530.8 eV are respectively attributed to the characteristic bands of the oxygen atoms in the oxides of  $\text{ZnCo}_2\text{O}_4$  and  $\text{NiWO}_4$ , as well as the oxygen from hydroxide ions. The minor peak at 532.2 eV is corresponding to surface-bound water or adsorbed oxygen [35]. Figure 4 e reveals two major peaks at 856.7 and 872.6 eV, which are respectively associated with Ni  $2p_{3/2}$  and Ni  $2p_{1/2}$  levels, as well as two shake-up satellites [5]. Figure 4 f shows two peaks at 34.1 eV and 36.3 eV, which correspond to W  $4f_{7/2}$  and W  $4f_{5/2}$ , respectively, revealing an oxidation state of  $\text{W}^{6+}$  [7]. The  $\text{ZnCo}_2\text{O}_4$  nanowire arrays and  $\text{ZnCo}_2\text{O}_4@\text{NiWO}_4$  nanowire/nanosheet arrays in situ grown on nickel foams could be directly evaluated by their electrochemical property without extra binder. Figure 5 a shows the CV properties of  $\text{ZnCo}_2\text{O}_4$  and  $\text{ZnCo}_2\text{O}_4@\text{NiWO}_4$  electrodes at a sweep rate of  $20 \text{ mV s}^{-1}$ . As expected, the area integrated within CV curve of the core-shell electrode is larger than that of the single electrode, illustrating that the core-shell electrode can reach a larger storage capacity. It can be understood that the additional  $\text{NiWO}_4$  shells can absorb more cations on the electrode surface for charge storage [36]. In addition, GCD curves were measured at  $1 \text{ mA cm}^{-2}$  to estimate the specific capacitance of the two electrodes, as shown in Fig. 5b. The  $\text{ZnCo}_2\text{O}_4@\text{NiWO}_4$  core/shell nanosheet array electrode spends 1069.0 s completing one discharge process, which is

**Fig. 2** Corresponding top-view photograph of **a** the  $\text{ZnCo}_2\text{O}_4@NiWO_4$  core-shell nanostructure at each step (samples 1, 2, 3, 4, and 5 refer to the pure Ni foam, Zn-Co precursor,  $\text{ZnCo}_2\text{O}_4$  nanowires,  $\text{ZnCo}_2\text{O}_4@NiWO_4$  precursor,  $\text{ZnCo}_2\text{O}_4@NiWO_4$ , respectively). SEM images of **b–d**  $\text{ZnCo}_2\text{O}_4$  and **e–g**  $\text{ZnCo}_2\text{O}_4@NiWO_4$  samples. **h, i** TEM images of  $\text{ZnCo}_2\text{O}_4@NiWO_4$  nanosheet arrays with different magnifications. **j** EDS mapping of the  $\text{ZnCo}_2\text{O}_4@NiWO_4$  electrode



about 2 times that of the  $\text{ZnCo}_2\text{O}_4$  electrode (521.0 s). Since the  $\text{ZnCo}_2\text{O}_4@NiWO_4$  electrode shows much higher specific capacitance, the electrochemical properties of

$\text{ZnCo}_2\text{O}_4@NiWO_4$  electrode were systematically measured as a function of scan rate (current density). Figure 5 c shows the CV curves of the  $\text{ZnCo}_2\text{O}_4@NiWO_4$  electrode obtained at



**Fig. 3** Typical XRD patterns of the  $\text{ZnCo}_2\text{O}_4$  and  $\text{ZnCo}_2\text{O}_4@NiWO_4$  samples, which corresponds well to the standard diffraction patterns of Zinc cobalt oxide (JCPDS card no. 23-1390) and nickel tungsten oxide (JCPDS card No. 15-0755)

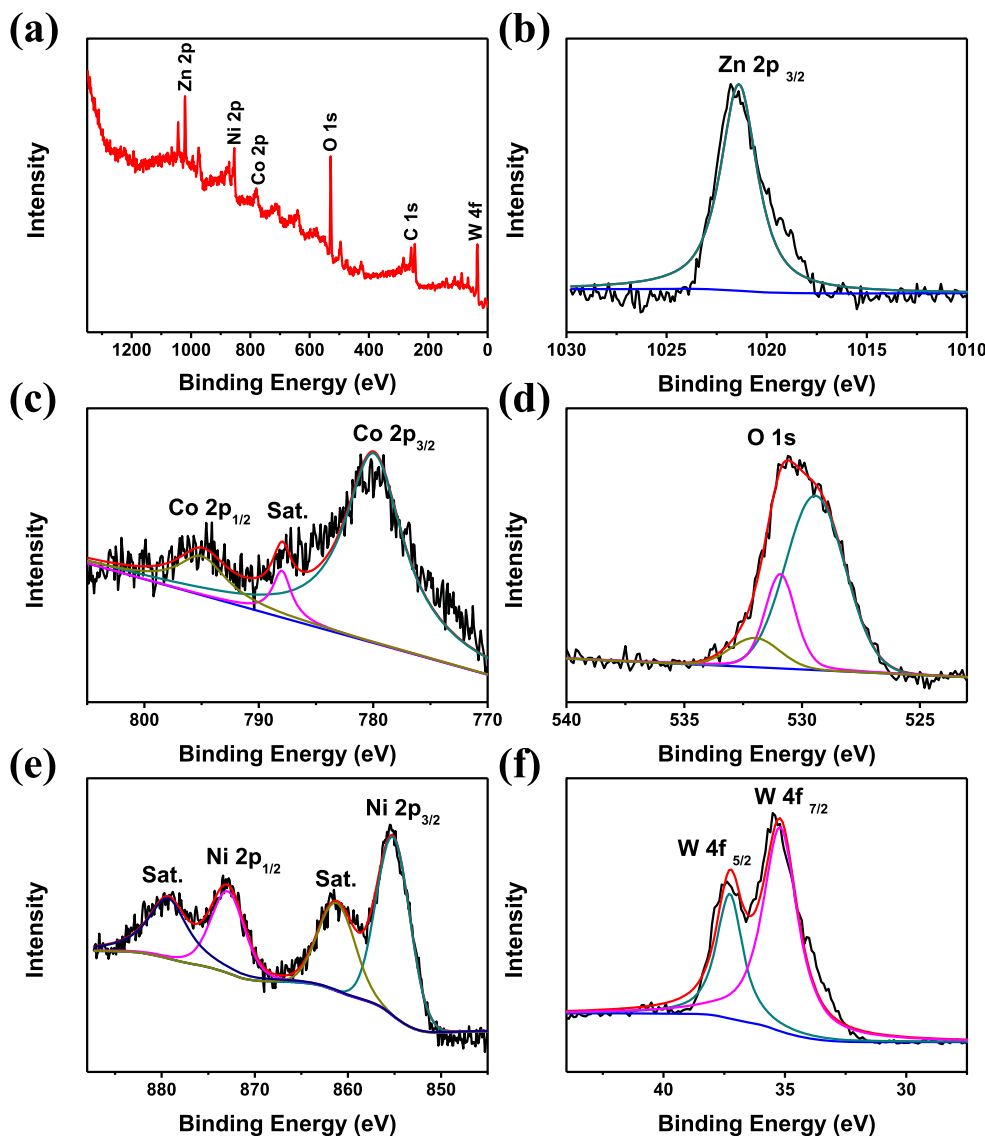
different scan rates in a potential window from 0 to 0.7 V. A pair of redox peaks clearly appeared in the CV curves of  $\text{ZnCo}_2\text{O}_4@NiWO_4$  core-shell electrode, in which one peak appeared at 0.3 V (negative scan) while the other peak appeared at 0.55 V (positive scan), featured by faradaic redox mechanism. The slight shift of redox peaks at various sweep rates is resulted from the polarization phenomenon [37]. Figure 5 d presents GCD curves of the  $\text{ZnCo}_2\text{O}_4@NiWO_4$  in a potential window ranging from 0 to 0.5 V, scanned as a function of current densities. The corresponding discharge time is found to be 1069, 506, 164, 76, and 49 s, respectively, measured at 1, 2, 5, 7.5, and 10  $\text{mA cm}^{-2}$ . The area-specific capacitance  $C_a$  ( $\text{F cm}^{-2}$ ) and mass-specific capacitance  $C_m$  ( $\text{F g}^{-1}$ ) can be respectively calculated by the following equation [1, 38]:

$$C_a = \frac{i\Delta t}{S\Delta V} \text{ and } C_m = \frac{i\Delta t}{m\Delta V} \quad (1)$$

where  $i$  signifies the discharge current,  $S$  is the area of nickel foam immersed in electrolyte, and  $m$  is the loading mass of active material on nickel foam,  $\Delta V$  is the potential window, and  $\Delta t$  designate the discharging time, respectively. For comparison purpose, the areal capacitances of both  $\text{ZnCo}_2\text{O}_4$  and  $\text{ZnCo}_2\text{O}_4@NiWO_4$  electrodes are calculated and plotted as the current density increases, as showed in Fig. 5e. The areal capacitance of  $\text{ZnCo}_2\text{O}_4$  electrode are calculated to be 1.040, 0.864, 0.607, 0.427, and 0.340  $\text{F cm}^{-2}$  at the current densities of 1, 2, 5, 7.5, and 10  $\text{mA cm}^{-2}$ , respectively. Correspondingly, the specific capacitance of  $\text{ZnCo}_2\text{O}_4$  electrode are found to be 1300  $\text{F g}^{-1}$  at a current density 1  $\text{A g}^{-1}$ , which is higher than that of many reported  $\text{ZnCo}_2\text{O}_4$  electrode ranged from 689 to 1664  $\text{F g}^{-1}$  [6, 39–41]. The areal capacitances of the  $\text{ZnCo}_2\text{O}_4@NiWO_4$

electrode are found to be 2.140, 2.024, 1.600, 1.080, and 0.760  $\text{F cm}^{-2}$  at the current densities of 1, 2, 5, 7.5, and 10  $\text{mA cm}^{-2}$ , respectively. Clearly, the  $\text{ZnCo}_2\text{O}_4@NiWO_4$  electrode exhibit much higher areal capacitances than that of single  $\text{ZnCo}_2\text{O}_4$  electrode at all current densities. The corresponding volumetric capacitances are 21.40, 20.24, 16.00, 10.80, and 7.60  $\text{F cm}^{-3}$ , and specific capacitances are found to be 1782, 1687, 1373, 950, and 817  $\text{F g}^{-1}$ , respectively. Therefore, the as-prepared core-shell electrode show higher specific capacitance than that of similar core-shell nanostructure such as  $\text{ZnCo}_2\text{O}_4@NiCo_2\text{O}_4$  (1476  $\text{F g}^{-1}$  at 1  $\text{A g}^{-1}$ ) [42],  $\text{ZnCo}_2\text{O}_4@Ni_3V_2O_8$  (1734  $\text{F g}^{-1}$  at 1  $\text{A g}^{-1}$ ) [43],  $NiCo_2O_4@NiWO_4$  (1384  $\text{F g}^{-1}$  at 1  $\text{A g}^{-1}$ ) [28], and  $NiMoO_4@NiWO_4$  (1290  $\text{F g}^{-1}$  at 1  $\text{A g}^{-1}$ ) [7]. The enhanced performance can be attributed to the 3D hierarchical core/shell structure and the synergic effects of two different binary transition metal oxides. A higher current density leads to a lower area capacitance. The reason is that the diffusion of the ions is quite slow and only the surface of the electrode could be utilized to provide active sites for charge storage. Cycling performance is also a significant parameter to study the quality of electrode for practical applications. Figure 5 f shows the capacitance retention during 5000 GCD cycles at 5  $\text{mA cm}^{-2}$ . At the initial measurement, there is a gradual activation process of the active material, which could contribute to the slight increase of capacitance. The similar phenomenon can be found in many previous literatures [44–46]. Furthermore, it is found that the  $\text{ZnCo}_2\text{O}_4@NiWO_4$  electrode presents an outstanding robust cycling stability with as high as 95.3% capacitance retained after 5000 successive GCD cycles, which is much higher than 90.9% capacitance retention of the single  $\text{ZnCo}_2\text{O}_4$  electrode. The enhanced cycling stability of  $\text{ZnCo}_2\text{O}_4@NiWO_4$  electrode is possibly attributed to the increased electrical conductivity of core-shell-nanostructured electrode, which could be confirmed by the EIS analysis. Figure 5 g shows the comparison of Nyquist plots between the single  $\text{ZnCo}_2\text{O}_4$  and  $\text{ZnCo}_2\text{O}_4@NiWO_4$  electrodes, which was recorded in a frequency range from 0.01 to 100 kHz. The two electrodes exhibit similar impedance spectra with a quasi-semicircle in the high frequency region and a slope in the low frequency region. The EIS curves can be modeled by the equivalent circuit shown in Fig. 5h. As expected, the internal resistance ( $R_s$ ) of the  $\text{ZnCo}_2\text{O}_4@NiWO_4$  electrode (1.053  $\Omega$ ) is slightly lower than that of  $\text{ZnCo}_2\text{O}_4$  electrode (1.063  $\Omega$ ), indicating the enhanced electrical conductivity of  $\text{ZnCo}_2\text{O}_4@NiWO_4$  electrode [47]. Furthermore, the  $\text{ZnCo}_2\text{O}_4@NiWO_4$  electrode exhibits a rather smaller semicircle than that of  $\text{ZnCo}_2\text{O}_4$  electrode, demonstrating that the  $\text{ZnCo}_2\text{O}_4@NiWO_4$  electrode has a much lower charge-transfer resistance ( $R_{ct}$ ) 1.526  $\Omega$  than 2.633  $\Omega$  of the  $\text{ZnCo}_2\text{O}_4$  electrode at electrode–electrolyte interface. In the low frequency region,  $\text{ZnCo}_2\text{O}_4@NiWO_4$  electrode exhibits a more vertical line, revealing the lower diffusion resistance within the electrode [48]. The above results demonstrate the enhanced properties of the  $\text{ZnCo}_2\text{O}_4@NiWO_4$  electrode comes from the

**Fig. 4** XPS spectra of the ZnCo<sub>2</sub>O<sub>4</sub>@NiWO<sub>4</sub> electrode. **a** Survey spectrum. **b** Zn 2p. **c** Co 2p. **d** O 1s. **e** Ni 2p. **f** W 4f



synergistic effects from the two components, the fast charge diffusion and quite low electro-transfer resistance [8]. To demonstrate the more reliable energy storage capability, all-solid-state ASC device of as-obtained ZnCo<sub>2</sub>O<sub>4</sub>@NiWO<sub>4</sub>//AC has been assembled and investigated by integrating the positive electrode of ZnCo<sub>2</sub>O<sub>4</sub>@NiWO<sub>4</sub> core/shell arrays, the negative electrode of AC with PVA-KOH gel electrolyte, as illustrated in Fig. 6a. Figure 6b presents the CV characteristics of the ASC device measured at the scan rates from 10 to 100 mV s<sup>-1</sup>. Clearly, the shape of CV curves and redox peaks are maintained at high scan rate such as 100 mV s<sup>-1</sup>, indicating the fast ion transport and excellent reversible faradaic reaction. Figure 6c presents the GCD properties of the ASC device obtained at various current densities of 5, 10, 15, and 20 mA cm<sup>-2</sup>. The nearly symmetric GCD curves also demonstrate the ZnCo<sub>2</sub>O<sub>4</sub>@NiWO<sub>4</sub>//AC ASC owns a strongly reversible reaction process. Correspondingly, the discharge time is estimated to be 212, 99, 58, and 43 s,

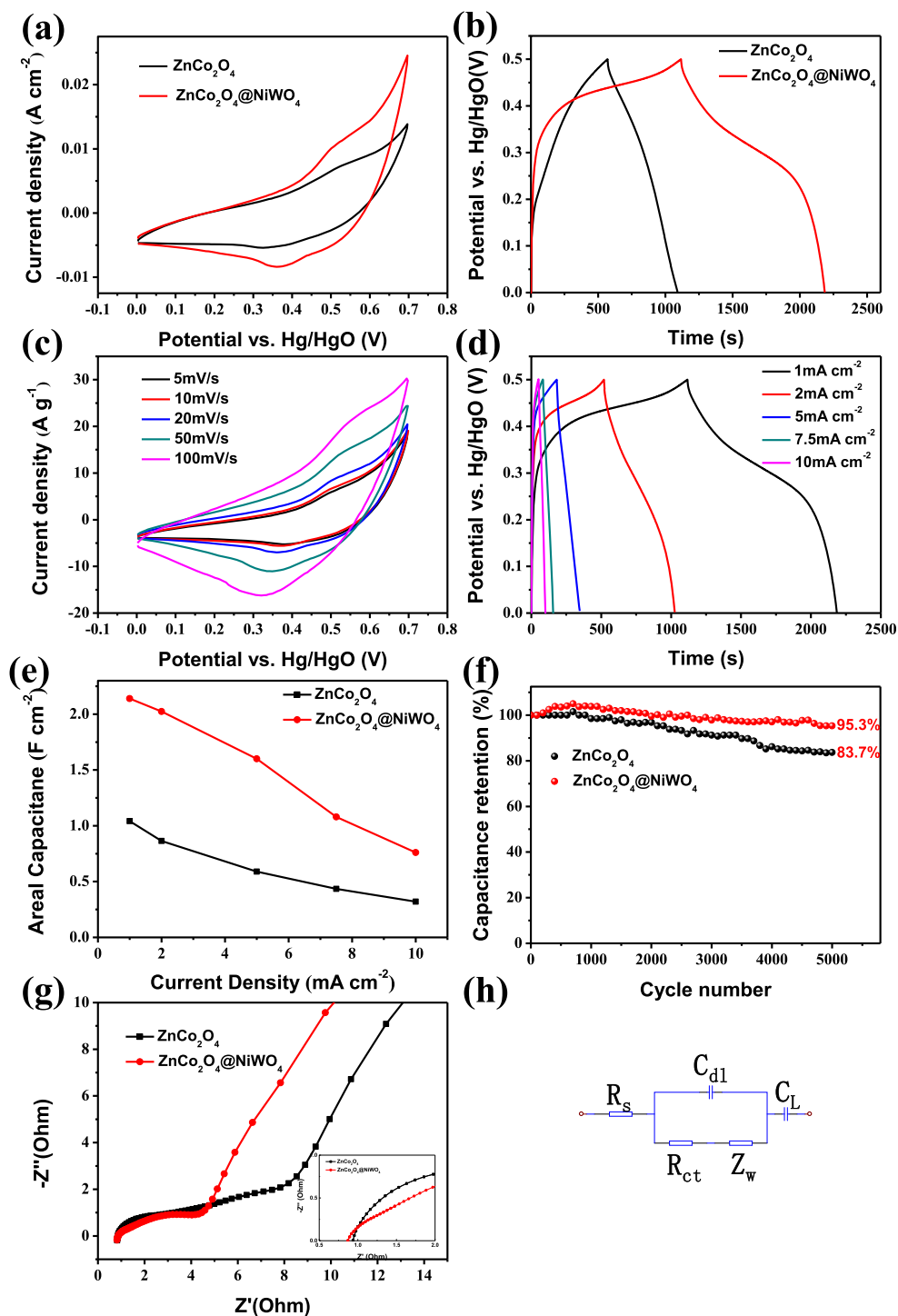
respectively. The specific capacitance of ASC device at different current densities could be calculated by Eq. (2), where *m* is the total mass of all active materials excluding the current collector. The mass ratio of ZnCo<sub>2</sub>O<sub>4</sub>@NiWO<sub>4</sub> arrays and AC can be obtained based on the balance rule of positive and negative charges (*q*<sup>+</sup> = *q*<sup>-</sup>) presented in the following equation [49]:

$$q = C\Delta Vm \tag{2}$$

$$\frac{m_+}{m_-} = \frac{C_- \Delta V_-}{C_+ \Delta V_+} \tag{3}$$

where *C* is the capacitance of the each electrode (F g<sup>-1</sup>), Δ*V* is the potential window (*V*), and *m* represent the mass of each electrode (*g*). The specific capacitance is calculated to be 135, 126, 112, and 110 F g<sup>-1</sup>, respectively, corresponds to different

**Fig. 5** **a** Comparison of CV and **b** GCD curves between the  $\text{ZnCo}_2\text{O}_4$  and  $\text{ZnCo}_2\text{O}_4@\text{NiWO}_4$  electrodes. **c** CV and **d** GCD curves of the  $\text{ZnCo}_2\text{O}_4@\text{NiWO}_4$  electrode. **e** Areal capacitance vs. current density for the  $\text{ZnCo}_2\text{O}_4$  and  $\text{ZnCo}_2\text{O}_4@\text{NiWO}_4$  electrodes. **f** Comparison of cycling behavior of the  $\text{ZnCo}_2\text{O}_4$  and  $\text{ZnCo}_2\text{O}_4@\text{NiWO}_4$  electrodes. **g** EIS and the related enlarged Nyquist plots in the high frequency region. **h** Corresponding electrical equivalent circuit of the  $\text{ZnCo}_2\text{O}_4$  and  $\text{ZnCo}_2\text{O}_4@\text{NiWO}_4$  electrodes



applied currents of 5, 10, 15, and 20  $\text{mA cm}^{-2}$ . Additionally, the energy density ( $E$ ) and power density ( $P$ ) are calculated by the following equation [1]:

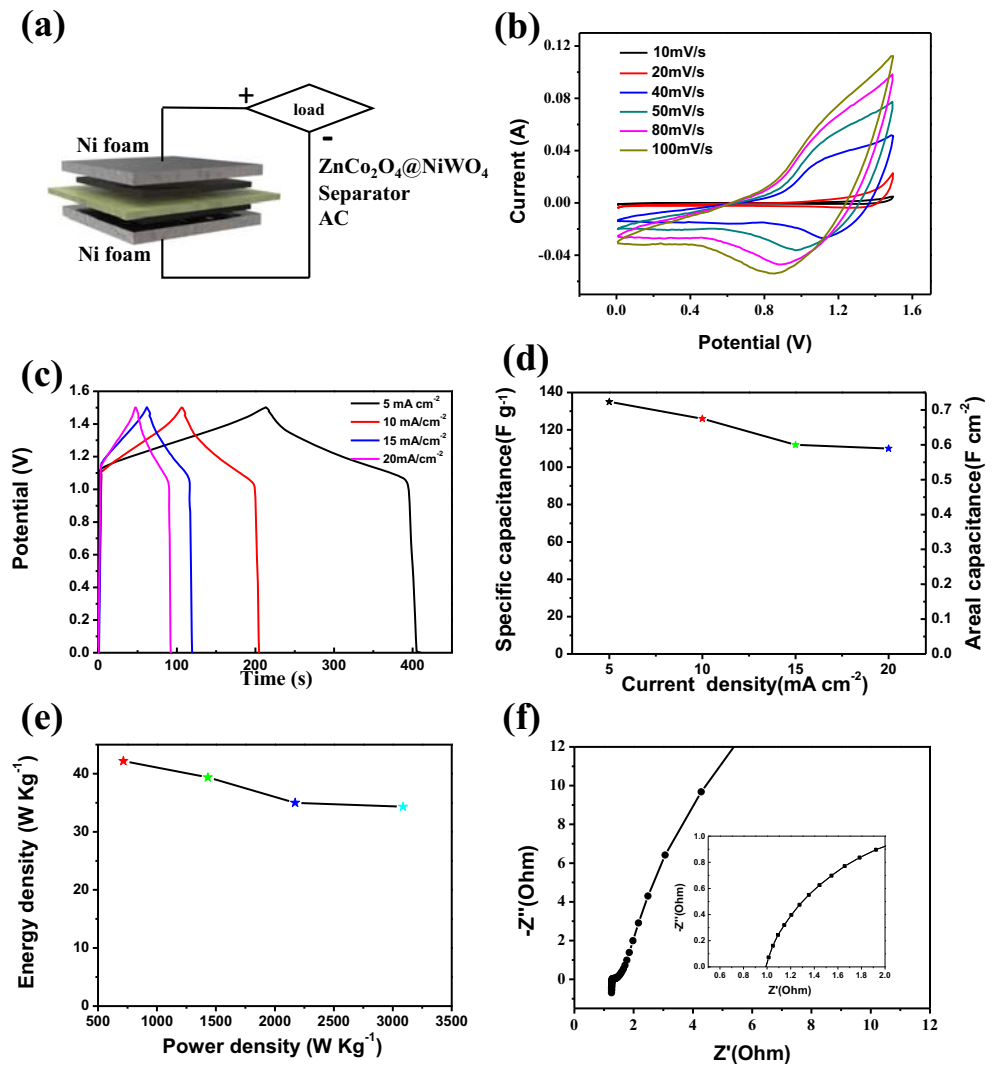
$$E = \frac{1}{7.2} C(\Delta V)^2 \quad (4)$$

$$P = \frac{3600 \times E}{t} \quad (5)$$

where  $C$  is the specific capacitance ( $\text{F g}^{-1}$ ),  $\Delta V$  is the potential window (V),  $t$  is the discharging time (s). From the ragone plot shown in Fig. 6e, the energy densities of the ASC device are found to be 42.2, 39.4, 35.0, and 34.3  $\text{Wh kg}^{-1}$  at power densities of 716, 1432, 2172, and 3087  $\text{W kg}^{-1}$ , respectively. The result is much higher than that of single  $\text{ZnCo}_2\text{O}_4/\text{AC}$  (34  $\text{Wh/kg}$  at 1800  $\text{W/kg}$ ) [50] and  $\text{NiWO}_4/\text{AC}$  (25.3  $\text{Wh/kg}$  at 684  $\text{W/kg}$ ) [29], respectively. Furthermore, the as-prepared



**Fig. 6** **a** Schematic illustration of the ACS device. **b** CV and **c** GCD curves of ACS device. **d** The mass- and area-specific capacitances changed as a function of current density. **e** Ragone plot. **f** EIS and the related enlarged Nyquist plots in the high frequency region of ACS device



electrode also shows much higher energy density and power density than that of similar core-shell nanostructure such as  $\text{ZnCo}_2\text{O}_4@ \text{Ni}_x\text{Co}_{2-x}(\text{OH})_{6x}/\text{AC}$  (26.2 Wh/kg at 511.8 W/kg) [24] and  $\text{ZnCo}_2\text{O}_4@ \text{MO}_2/\alpha\text{-Fe}_2\text{O}_3$  (37.8 Wh/kg at 684 W/kg) [51]. From the EIS spectrum and the inset of enlarged Nyquist plots in the high frequency region shown in Fig. 6f, the ACS device also presents the series resistance of 1.053  $\Omega$  and the charge transfer resistance of 1.526  $\Omega$ , further confirming the higher conductivity and more efficient charge transport within the electrode.

### Conclusions

In conclusion, 3D, string-like, core-shell, heterostructured  $\text{ZnCo}_2\text{O}_4@ \text{NiWO}_4$  nanowire/nanosheet arrays are grown on 3D nickel foam substrate by reliable, two-step hydrothermal processes and subsequent annealing treatment. The  $\text{ZnCo}_2\text{O}_4@ \text{NiWO}_4$  hybrid electrode delivers a specific capacitance as high as 1782  $\text{F g}^{-1}$  at 1  $\text{mA cm}^{-2}$  (2.14  $\text{F cm}^{-2}$ ) and

especially high cycling stability (95.3% capacitance retention after 5000 cycles), which are much higher than individual  $\text{ZnCo}_2\text{O}_4$  electrode without the  $\text{NiWO}_4$  shell. Such fascinating electrochemical performance benefits from the 3D string-like core-shell nanostructure and synergistic effects of the contributions from both  $\text{ZnCo}_2\text{O}_4$  nanowire core and  $\text{NiWO}_4$  shell nanosheets, which could provide more active sites for charge storage, facilitate quick charge penetration of electrolyte, and increase diffusion of ions. Therefore, constructing string-like core-shell hybrid nanowire/nanosheet arrays with two different binary transition metal oxides may serve to be a potential way to develop high-performance supercapacitors

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