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Synthesis and electrochemical properties of $rGO-MoS₂$ heterostructures for highly sensitive nitrite detection

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Abstract In this paper, the reduced graphene oxide (rGO) and molybdenum disulfide $(MoS₂)$ $(rGO-MoS₂)$ heterostructures have been successfully synthesized by a facile hydrothermal method. The crystal phase, surface morphology, and chemical composition of the obtained heterostructures were characterized by X-ray diffraction (XRD), scanning electron microscopy (SEM), transmission electron microscopy (TEM), and X-ray photoelectron spectroscopy (XPS) techniques. The electrochemical properties of the nitrite sensor attached with $rGO-MoS₂$ heterostructures were investigated using cyclic voltammetry (CV) and single-potential amperometry methods. The measured results show that the as-prepared sensor based on $rGO-MoS₂/GCE$ exhibits a wide linear measurement range $(0.2-4800 \mu M)$, low detection limit (0.17 μ M), high sensitivity (0.46 μ A μ M⁻¹ cm⁻²), and

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good selectivity and reproducibility towards nitrite detection. The anti-interference property and real sample analysis were also investigated, which shows that the as-prepared $rGO-MoS₂$ heterostructures present great potential for practical applications.

Keywords Reduced graphene oxide · Molybdenum disulfide . Nitrite . Electrochemical sensor

Introduction

In the past decades, great attentions have been attracted on determination and detection of nitrite $(NO₂⁻)$ because the high concentrations of nitrite can become poisonous to human beings and animals. Nitrite can form carcinogenic nitrosamines [[1\]](#page-9-0) and cause the loss of oxygencarrying ability for hemoglobin in the human body [\[2](#page-9-0)]. Despite these facts, nitrite is widely used in the environment, beverages, and food products as a preservative [\[3\]](#page-9-0), which has extensive existence in soil, natural water, and physiological systems [[4](#page-9-0)]. In order to protect human health, the European Community has fixed the maximum limit of 0.1 mg L^{-1} (~2.2 mM) for nitrite in drinking water [\[5](#page-9-0)]. Therefore, it is necessary to accurately monitor nitrite in living environment and food technologies. Up to now, several analytical techniques such as spectrophotometry, flow injection analysis, chemiluminescence, and chromatography, have been developed for nitrite detection $[6-10]$ $[6-10]$ $[6-10]$ $[6-10]$ $[6-10]$. Among these methods, the electrochemical method is a simple, rapid, and low-cost methodology based on the oxidation or reduction of nitrite [\[11](#page-9-0)]. Moreover, the direct oxidation of nitrite on conventional electrode requires large overpotential. Hence, different kinds of electrochemical

nitrite sensors have been developed based on the modified electrodes [[12](#page-9-0)–[14](#page-9-0)].

Recently, numerous two-dimensional (2D) nanostructured electrode materials, such as graphene, tungsten disulfide (WS₂), molybdenum disulfide (MoS₂), and conductive polymer materials have received considerable interests because of their intriguing quantum confined properties and excellent electrochemical performance [\[15](#page-9-0)–[17\]](#page-9-0). Among all studied 2D nanostructures, graphene has been demonstrated to be one of the most promising electrode materials in electrochemical sensors owing to its good electrical conductivity, large contact surface area, strong mechanical strength, and a wide electrochemical window compared to the other 2D materials [\[18](#page-9-0)]. Moreover, graphene can be easily integrated with other 2D nanocrystals such as $MoS₂$, boron nitride (BN), or WS_2 to form heterostructures [\[19](#page-9-0)–[21](#page-9-0)]. Among them, $MoS₂$, as a 2D layered structure, has attracted much attention in electrochemical terms due to its excellent electrocatalytic properties. For example, Huang et al. [\[22\]](#page-9-0) reported the $MoS₂$ and chitosan-gold nanoparticle composites modified electrode as a novel electrochemical sensor constructed for the determination of bisphenol A (BPA). Chang et al. [[23\]](#page-9-0) synthesized the graphenelike $MoS₂/amorphous carbon composites, exhibiting a$ high reversible capacity and excellent cyclic stability as anode materials for lithium ion batteries. Moreover, $rGO-MoS₂$ composites have been widely used in electrochemistry, such as reversible lithium storage, DNA detection, and supercapacitor [[19](#page-9-0), [24,](#page-9-0) [25\]](#page-9-0). Notably, Zhang et al. [[26\]](#page-9-0) demonstrated that a layered Mb/ $MoS₂$ –graphene/Nafion biosensor exhibited catalytic performance for the reduction of sodium nitrite $(NaNO₂)$. However, its detection limit was poor and the direct reduction of NO_2^- on the layered MoS_2 -graphene/ Nafion was not investigated. Although considerable efforts have been devoted to the electrochemical applications of two-dimensional materials, to the best of our knowledge, seldom works have been conducted on rGO-MoS2 heterostructures for nitrite detection.

In this work, the electrochemical sensor based on rGO- $MoS₂$ heterostructures has been developed for the nitrite detection. The microstructural characterization, elemental composition, and chemical state analysis of the $rGO-MoS₂$ heterostructures were analyzed through the different material characterization techniques. The electro-catalytic performances of as-fabricated sensor towards the nitrite detection were evaluated by cyclic voltammetry (CV) and single-potential amperometry. The results obtained have demonstrated that the as-prepared $rGO-MoS_2/GCE$ sensor exhibits high sensitivity, low detection limit, wide linear concentration range, and high stability and availability for accurate NO_2^- detection.

Experimental

Chemicals and reagents

Graphite (325 meshes) was purchased from Qingdao Huatai Lubrication Sealing Technology Co. Ltd. Sodium molybdate $(Na_2MoO_4·2H_2O, >99.99%)$, thiourea (NH_2CSNH_2) , $≥99.0\%$), and sodium nitrite (NaNO₂, $≥99.0\%$) were used from Sigma-Aldrich (Shanghai, China). Potassium permanganate (KMnO₄, $>99.5\%$) was obtained from Sinopharm Chemical Reagent Co, Ltd. (Shanghai, China). All other reagents used in our experiments were of analytical reagent grade and were used without further purification. The deionized water (Milli-Q Millipore 18.2 M Ω cm resistivity) was used in all dilutions. The 0.1 M phosphate buffer solution (PBS) was prepared by mixing solutions of $Na₂HPO₄$ and $NaH₂PO₄$ and used as the supporting electrolyte.

Synthesis of the $rGO-MoS₂$ heterostructures

Graphene oxide (GO) was prepared by a modified Hummers method [[27\]](#page-9-0). Firstly, 0.25 g of graphite and 0.25 g of $KMnO₄$ were mixed with 7.5 mL of H_2SO_4 (98%). Secondly, the mixture was stirred for 10 min at room temperature. Then, the reactants were kept at 50 °C and stirred for 3.5 h. After that, the mixture was diluted and the excess $KMnO₄$ removed by adding H_2O_2 . Next, the mixed aqueous solution was washed with 10% HCl (36–38%). Subsequently, the precipitate was suspended in 150 mL of deionized water and underwent ultrasonic peeling for 1.5 h. Finally, the aqueous phase suspension of GO was obtained after centrifugation at 3500 rpm, and the concentration of GO solution was approximately 0.5 mg/mL.

The $rGO-MoS₂$ heterostructures were synthesized using a modified one-step hydrothermal method [\[28](#page-9-0)]. In a typical synthesis, 0.6 g Na₂MoO₄ \cdot 2H₂O and 0.75 g NH₂CSNH₂ were dissolved into 30 mL GO suspension, and the solution was stirred for 20 min at room temperature. After that, the mixture was transferred into Teflon-lined autoclave and heated at 220 °C for 24 h. Then, the GO was reduced to the reduced graphene oxide (rGO). After cooling to room temperature, the black product was washed with DI water and anhydrous ethanol. Finally, the $rGO-MoS₂$ heterostructures were obtained by drying at 60 °C for 12 h.

Prior to modification, the surface of the GC electrode was polished to a mirror using 1.0 and 0.3 μm alumina particles, sonicated in ethanol and deionized water, and dried in air. Then, 2 mg as-prepared samples were dispersed into 1 mL deionized water to obtain homogeneous slurry (2 mg/mL). After that, a certain amount of $rGO-MoS₂$ slurry was dropped onto the surface of the GCE by pipette and dried at 60 °C for 1 h, which was defined as $rGO-MoS₂/GCE$. For comparison, rGO/GCE and $MoS₂/GCE$ were prepared using the similar procedure. All of the modified electrodes were stored at 4 °C before use.

Characterization

XRD analysis was made on a MiniFlex II (Rigaku Corporation Cu-K α 1 radiation $\lambda = 1.5406$ Å) with a scanning speed of $3^{\circ}/$ min. The morphology of as-synthesized hybrid structures was observed by scanning electron microscopy (SEM, JSM-7001F) with an accelerating voltage of 10 kV. The crystalline structures of the sample were further measured by transmission electron microscopy (TEM, JEM-2100F). The X-ray photoelectron spectroscopy (XPS, VG ESCALAB 250) measurements were performed on an Escalab 250Xi instrument with Mg-K α radiation. All the electrochemical measurements were carried out on a Zahner IM6 electrochemical workstation with a conventional threeelectrode system. A bare or modified GCE was used as the working electrode, and a Pt wire mesh electrode was used as the counter electrode. A silver chloride electrode (saturated KCl) was used as the reference electrode.

Results and discussion

Morphological and structural characterization

Figure 1 shows the XRD patterns of the as-synthesized rGO, $MoS₂$, and rGO- $MoS₂$ heterostructures, respectively. For rGO (Fig. 1a), the two peaks at about $2\theta = 24.40^{\circ}$ and $2\theta = 42.85^{\circ}$ can be assigned to the (002) and (100) planes of graphene sheets [[29](#page-9-0)]. Compared with pure rGO, both $MoS₂$ and rGO- $MoS₂$ heterostructures exhibit the similar diffraction peaks corresponding to the (002), (100), (103), (110), which match well with the phase of $MoS₂$ (JCPDS 37-1492). Furthermore, it is worth noticing that the intensity of all the diffraction peaks of $rGO-MoS₂$ hybrid structures are much weaker than that of the pure $MoS₂$ structure, especially the (002) plane peaks, which suggests that the incorporation of the graphene

considerably inhibited the (002) plane growth of the MoS₂ crystals in the heterostructured composites [\[30\]](#page-10-0).

The surface morphology and structure of the asprepared rGO-MoS₂ samples were further characterized by SEM. Figure [2](#page-3-0)a depicts the SEM micrograph of the rGO and shows that the surface of the rGO has a wrinkled texture associated with the presence of the flexible and ultrathin graphene sheets. On the other hand, the morphology characterizations of the pure $MoS₂$ (Fig. [2b](#page-3-0)) revealed that the pristine $MoS₂$ nanoflakes tend to stack together tightly and exhibit a flower-like nanostructure. The thickness of petal-like nanoflakes is only about several nanometers. As a contrast, the $rGO-MoS₂$ heterostructures (Fig. [2c](#page-3-0), d) exhibit the irregular three-dimensional nanoarchitecture. The drastic difference indicates the novel role of GO as an efficient substrate material for the growth of $MoS₂$, suppressing the stacking of $MoS₂$ nanoflakes [[31](#page-10-0)]. The unique nano-architecture of rGO- $MoS₂$ can not only increase the surface contact area with the analytes but can also facilitate the rapid electronic transport of electrochemical reactions. Moreover, the developed rGO-MoS₂ heterostructures also possess an excellent stability due to the super strength of graphene.

To further study the morphology and the interplanar spacing of the synthesized $rGO-MoS₂$ heterostructures, TEM, HRTEM, and the corresponding selected-area electron diffraction (SAED) were performed on the assynthesized $rGO-MoS₂$ heterostructures, as shown in Fig. [3.](#page-3-0) From the TEM images of $rGO-MoS₂$ (Fig. [3](#page-3-0)a, b), we can clearly observe that a large number of curved and wrinkled $MoS₂$ nanoflakes were dispersed on the surface of rGO, which was consistent well with the SEM image of $rGO-MoS₂$. Figure [3](#page-3-0)c exhibits the typical HRTEM image of $rGO-MoS₂$ heterostructures, and the measured interlayer distance is about 0.62 nm, which corresponds to the (002) lattice plane of MoS₂ [[32\]](#page-10-0). Figure [3](#page-3-0)d illustrates the SAED pattern of the rG O-MoS₂ heterostructures, which is similar to the crystalline structure of pure $MoS₂$. The measured results unambiguously indicate that the stacking between the adjacent rGO sheets is suppressed due to the selective growth of $MoS₂$, which is consistent with the XRD results [[31\]](#page-10-0).

To determine the elemental composition and chemical state, XPS analysis was conducted on the $rGO-MoS₂$ heterostructures and the main results are illustrated in Fig. [4.](#page-4-0) The binding energies were calibrated to the C1s peak at 284.4 ± 0.1 eV of the surface adventitious carbon. It can be observed that only signals of Mo, S, C, and O elements present in the survey spectrum (Fig. [4](#page-4-0)a), which indicates the high purity of as-synthesized $rGO-MoS₂$ sample. Figure [4b](#page-4-0) displays the XPS profile of Mo 3d spectrum, which can be deconvoluted into three peaks. The measured two strong peaks Fig. 1 XRD of the as-synthesized rGO (a), $MoS₂(b)$, and rGO-MoS₂ (c) with binding energy values of 229.4 and 232.5 eV can be

Fig. 2 Typical SEM images of the as-synthesized rGO (a) , MoS₂ (b), and $rGO-MoS₂$ heterostructures (c, d)

assigned to the $Mo3d_{5/2}$ and $Mo3d_{3/2}$ peaks, respectively, while the weak peak is located at 226.4 eV is corresponding to S2s of $MoS₂$. Meanwhile, the S 2p spectrum shows two peaks at 162.2 eV and 163.4 eV (Fig. [4](#page-4-0)c), which could be attributed to the spin–orbit couple $S2p_{3/2}$ and $S2p_{1/2}$, respectively. These binding energies are all consistent with the

Fig. 3 a TEM image of rGO-MoS₂. **b**, **c** HRTEM images of rGO-MoS₂ (zoomed from the selected area in a, b respectively). d The corresponding SAED pattern

previous reported values of $MoS₂$ crystal [[33\]](#page-10-0). Figure [4](#page-4-0)d displays the spectrum of C1s, and the two peaks located at 284.6 and 286.8 eV can be ascribed to the C–C and C–O, respectively. Moreover, the low content of O suggests that GO has been adequately reduced to rGO by the solvothermal treatment.

Fig. 4 XPS survey spectrum of rGO-MoS₂ heterostructures, a full survey scan spectrum, b Mo 3d, and c S2p and d C1s regions

Electrocatalytic oxidation of nitrite on electrodes

It is well known that the pH value can significantly influence the electrochemical performance of the electrochemical sensor. Here, the electrochemical response of $rGO-MoS₂/GCE$ towards 500 μM nitrite was initially investigated by testing the CVs in the pH range of 6.0 to 8.0 in 0.1 M PBS. Figure 5a shows CV curves of rGO-MoS₂/GCE at different solution pH values in 500 μM nitrite. Clearly, the current response climbed up at first and reached to the maximum at pH 7.0, and then declined significantly with further increase of the pH value, as illustrated in Fig. 5b. Consequently, the pH 7.0 has been selected for the following experiments to obtain the high sensitivity.

The relationship between the peak current and modification amounts of $rGO-MoS₂$ were further investigated. Figure [6a](#page-5-0) shows the CV curves of $rGO-MoS_2/GCE$ sensors towards 500μ M nitrite. The measured results reveal that the maximum peak current can reach to 45.3 μA, when the modification rGO-MoS₂ slurry is about 25 μ L, as shown in Fig. [6](#page-5-0)b. The highest peak current can be ascribed to the reasons as follows: when a small amount of $rGO-MoS₂$ slurry was coated on the

Fig. 5 a CV curves of rGO-MoS2/GCE at different solution pH values in 500 μM nitrite. b Effect of different solution pH values on peak current in 500 μM nitrite

Fig. 6 a CV curves of GCE modified with different amounts of rGO-MoS₂ slurry in 500 μM nitrite. b Effect of different amounts of rGO-MoS₂ slurry on peak current in 500 μM nitrite

surface of GCE, the modified electrode surface could not be thoroughly covered by $rGO-MoS₂$ heterostructures, which will cause insufficient active sites for nitrite catalysis. On the other hand, when excessive of $rGO-MoS₂$ heterostructures was modified on the electrode, the large amount of rGO- $MoS₂$ slurry may cause the aggregation, blocking the electron transport. As a result, the optimum content of rG O-MoS₂ heterostructures is about 25 μL, and the subsequent measurements were carried out under this condition. Furthermore, the thickness of the $rGO-MoS₂$ layer was characterized using SEM. As shown in Fig. S1(a, b), the bare GCE is uniformly covered by the $rGO-MoS₂$ heterostructures. Figure $S1(c)$ reveals the cross-section image of $rGO-MoS₂$ heterostructures, and the thickness of rGO-MoS₂ slurry is approximately 60 μ m on the surface of GCE.

To compare the electrocatalytic activity of the asprepared electrodes (GCE, rGO/GCE , $MoS₂/GCE$, and $rGO-MoS₂/GCE$) towards nitrite detection, the electrochemical properties of the as-prepared electrodes were studied by cyclic voltammograms (CVs) in the potential range of 0.0–1.2 V at a scan rate of 50 mV s⁻¹ (vs. Ag/ AgCl). Figure 7a illustrates the CVs of as-prepared electrodes in 0.1 M PBS ($pH = 7.0$) towards 500 μM nitrite. For the bare GCE, a broad oxidation peak appears at the potential of ~ 0.90 V with a peak current of ~ 17.3 μ A. For the rGO/GCE electrode, a significant oxidation peak response can be observed at ~ 0.81 V with current about 33.2 μA, which indicates a certain catalytic property on rGO in this current potential window. A higher oxidation current peak (\sim 38.1 μA) can be achieved for MoS₂/GCE comparing with GCE and rGO/GCE but with a higher catalytic potential of ~0.95 V. Moreover, it can be clearly observed that the measured oxidation peak current value can reach to 45.0 μ A for rGO-MoS₂/GCE with the potential of \sim 0.85 V, and the highest catalytic current and more negative potential (than $MoS₂/GCE$) of rGO-MoS₂ hybrid structures suggest a better catalytic ability towards nitrite than rGO and $MoS₂$. The results obtained demonstrate that the introduction of rGO can improve the electronic transmission rate as well as increase the surface area to

Fig. 7 a CV curves of the as-prepared electrodes in 0.1 M PBS (pH = 7.0) with 500 μM nitrite. b CV curves of the rGO-MoS2/GCE electrode in 0.1 M PBS (pH = 7.0) under different concentrations of nitrite: 100, 300, 500, 700, and 1000 μ M (scan rate: 50 mV s⁻¹)

rate (inset image: the peak potential and logarithm of the scan rate (log ν))

capture a large amount of analytes, which consequently can efficiently promote the electro-oxidation of NO_2^- .

Meanwhile, the sensing properties of $rGO-MoS₂$ heterostructures were further studied by detecting different concentrations of NO_2^- , as shown in Fig. [7](#page-5-0)b. The measured results demonstrated that the oxidation peak current value proportionally increases with the increasing concentration of $NO₂⁻$ from 100 to 1000 μ M, which confirmed that the rGO- $MoS₂/GCE$ could effectively determine the different concentrations of NO_2^- in current conditions. As a result, the rGO-MoS2 heterostructures have shown excellent electrocatalytic properties, which confirmed that the $rGO-MoS₂$ is indeed a very promising sensing electrode material for the electrochemical NO_2 ⁻ sensor.

On the present $rGO-MoS₂$ electrode, the nitrite could be oxidized at a suitable potential in the PBS solution. As reported by Guidelli et al., the nitrite oxidation is a second-order homogeneous disproportionation process [\[34\]](#page-10-0). Thus, the mechanism and overall reaction can be explained with the following process (Eqs. $(1-2)$):

$$
NO_2^- - e^- \rightarrow NO_2 \tag{1}
$$

$$
2NO2 + H2O \to NO2- + NO3- + 2H+
$$
 (2)

From Eqs. (1–2), NO_3 ⁻ is the only plausible final product. The influence of the scan rate on the cyclic voltammetric performance for the rGO-MoS₂/GCE was also carried out in 0.1 M PBS ($pH = 7.0$) with 500 μ M nitrite. Figure 8a depicts the CV responses of $rGO-MoS_2/GCE$ at the presence of 500 μM nitrite at the different scan rates of 10, 25, 50, 75, 100, 125, and 150 mV/s, respectively. The results show that oxidation peak current (I_p) increases linearly with the increasing of the square root of the scan rate $(v^{1/2})$, indicating the diffusion-controlled electrocatalytic process (Fig. 8b) [[35\]](#page-10-0). Moreover, it can be observed that the peak potential E_p

Fig. 9 a The amperometric current responses of rGO-MoS₂/GCE for successive addition of nitrite range from 0.2 to 4800 μ M in 0.1 M PBS (pH = 7.0) (inset image (i): amperometric current response of 0.2 to 50 μM). b The linear plot of oxidation current plateau value vs. nitrite concentration

increases with the rising of scan rate, which also exhibited a linear dependence with the logarithm of scan rate (log v) (Fig. [8b](#page-6-0) inset). The measured results have clearly demonstrated that the electrocatalytic oxidation process of nitrite is irreversible [[36\]](#page-10-0).

Amperometric $(i-t)$ response of rGO-MoS₂/GCE sensor

The amperometric response experiments were also performed on the $rGO-MoS₂$ heterostructures with stirring rate 200 rpm. Figure [9a](#page-6-0) shows the stepped amperometric current (i-t) curve of sensor based on $rGO-MoS₂/GCE$ in 0.2 μM–4800 μM nitrite at 0.80 V. And, the inset image illustrates the amplified amperometric current response for the lower nitrite concentration of 0.2–50 μM. The obtained experimental results show that the developed sensor could achieve 95% of the response steady-state value within 3 s after incorporation of a certain nitrite concentration into solution. The outstanding electrochemical properties may be due to the synergistic effect of rGO and $MoS₂$, which provides fast and efficient electrocatalytic abilities towards nitrite. Meanwhile, the electrochemical performances of rGO-MoS₂/GCE sensor were also studied under the different stirring speeds of the solution [[37](#page-10-0)]. Table S1 (see Supplementary section) illustrates the measured linear range, detection limit, and sensitivity of rGO- $MoS₂/GCE$ sensor, which confirmed that the rGO-MoS₂/ GCE sensor exhibits excellent electrochemical properties with stirring rate of 200 rpm. Therefore, the optimum stirring rate is set to be about 200 rpm for the amperometric experiments.

Figure [9](#page-6-0)b displays the corresponding calibration curve (current (μA) vs. concentration (μM)) of the amperometric response with stirring rate of 200 rpm. The $rGO-MoS₂/GCE$ shows an excellent linear response in the range of 0.2 to 4800 μM. The linear equation of $rGO-MoS₂/GCE$ can be defined as $y = 0.09 \times + 3.51$ with a correlation coefficient R^2 = 0.999, and the minimum detection limit is estimated to be 0.17 μ M under a signal-to-noise ratio (S/N) of 3. Furthermore, to compare the performances of the fabricated rGO-MoS2/GCE nitrite electrochemical sensor, various nitrite electrochemical sensors based on rGO or $MoS₂$ sensing materials and non-electrochemical means were summarized in Table 1. Compared with previous works [\[6](#page-9-0), [8](#page-9-0)–[10,](#page-9-0) [12](#page-9-0)–[14,](#page-9-0) [16,](#page-9-0) [38](#page-10-0)–[43\]](#page-10-0), the electrochemical sensor modified with $rGO-MoS₂$ heterostructures exhibits a very large linear range (0.2 μM-4800 μM) with a detection limit of 0.17 μM, and higher sensitivity (0.46 μA μM⁻¹ cm⁻²) at the potential of 0.8 V. Consequently, it can be concluded that the sensor based on $rGO-MoS₂$ heterostructures demostrates the best overall response characteristics due to the unique composite nanostructure and synergistic effect, which is promising for nitrite detection.

Selectivity and reproducibility

To evaluate the selectivity of the as-fabricated sensor towards nitrite, the anti-interference ability of the $rGO-MoS₂$ heterostructures has also been examined by incorporation of such additives as $Na₂SO₄$, KCl, NaNO₃, Na₂CO₃, glucose, and alcohol into the solution. As shown in Fig. 10a, when nitrite was added, the current increased significantly and reached a steady-state value immediately. However, no apparent amperometric responses were observed when additives were subsequently injected at the regular intervals. Stability of the output sensor's signal suggests that none of the incorporated chemicals have effect on the performance of the sensor in the current conditions. Meanwhile, the reproducibility of the sensor was also determined by detecting the current response with three rGO-MoS₂ modified electrodes, which was prepared using the same method (Fig. 10b). The measured result indicates that the amperometric responses towards the same concentration of nitrite for the different sensors are almost the same. The standard deviation of the sensitivities is calculated to be 0.12, which proves an excellent reproducibility for the developed electrochemical sensor attached with rGO-MoS₂/GCE heterostructures.

Table 2 Determination results of nitrite in the real samples

Sample	Added (µM)	Found (μM)	Recovery $(\%)$
1	50	52.1 ± 3.3	104.2
2	100	105.0 ± 8.1	105.0
3	300	303.1 ± 4.2	101.0
4	500	503.2 ± 3.4	100.6
5	1000	983.8 ± 15.2	98.4
6	1500	1494.3 ± 21.4	99.6
7	2000	1991.8 ± 155.3	99.6
8	2500	2493.3 ± 32.5	99.7
9	3000	2993.2 ± 132.5	99.8
10	3500	3494.4 ± 154.2	99.8
11	4000	3981.2 ± 178.3	99.5

Application to real samples

In order to demonstrate the practicability of the fabricated $rGO-MoS₂/GCE$ sensor, the amperometric response measurements were conducted in known concentrations of nitrite solution. Eleven different concentrations of nitrite samples (50, 100, 300, 500, 1000, 1500, 2000, 2500, 3000, 3500, 4000 μM) were prepared by tap water. Figure S2 (see Supplementary section) exhibits the amperometric responses of the $rGO-MoS₂/GCE$ sensor for the real samples, which indicates that the current can reach a stable value in a relatively short time. Table 2 illustrates the determination results of rGO- $MoS₂/GCE$ sensor under the different nitrite concentrations. The measured recovery values for $rGO-MoS₂/GCE$ sensor vary from 98 to 104%, which confirmed that the fabricated sensor has great potential application nitrite detection in the real samples.

Conclusions

In this work, a simple and facial hydrothermal method for the synthesis of $rGO-MoS₂$ heterostructures is demonstrated. The electrochemical sensor based on developed $rGO-MoS₂$ heterostructures was fabricated and used for nitrite measurements. The electrochemical results obtained show that the sensor attached with synthesized $rGO-MoS₂$ heterostructures can greatly improve the electrocatalytic performance towards the nitrite determination in a solution compared with other sensors based on the conventional solid electrodes. Moreover, the experimental results clearly demonstrated that the fabricated sensor exhibited several advantages including wide linear range (0.2 to 4800 μ M); high sensitivity (0.46 μA μM⁻¹ cm⁻²); low detection limit (0.17 μM); and good selectivity, reproducibility, and availability. The enhanced electrochemical properties of the rGO- $MoS₂/GCE$ sensor can be attributed to the unique hybrid structure sensing electrode and synergistic effect. The as-synthesized $rGO-MoS₂$ heterostructures can

therefore be utilized as promising and unique sensing platform for the development of various nitrite electrochemical devices with enhanced sensing capabilities.

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