# **Fabrication and Characterization of Plasmonic Au Nanoparticles on ITO‑Coated Glass Sheets**

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Received: 2 May 2022 / Accepted: 27 July 2022 / Published online: 25 August 2022 © The Author(s), under exclusive licence to Springer Science+Business Media, LLC, part of Springer Nature 2022

#### **Abstract**

We report here the fabrication of sputter-deposited Au nanoparticles on ITO (Sn-doped indium oxide)-coated glass substrate. The efect of varying annealing parameters, viz., time and temperature in conjunction with varying thickness parameters has been thoroughly investigated. Synthesized Au nanoparticles were characterized using a feld emission scanning electron microscope (FE-SEM), Rutherford backscattering (RBS) measurement, X-ray difraction (XRD), and optical absorbance spectroscopy. The role of annealing temperature and time duration on the evolution of Au nanoparticles has been discussed in detail based on optical absorbance spectroscopy. The present study shows that 5 nm thickness Au thin flm which was annealed at 500 °C for 90 min exhibited the strongest surface plasmon resonance (SPR) efect.

**Keywords** Au nanoparticles · ITO · Sputtering · Annealing · Plasmons

# **Introduction**

Plasmonic metal nanoparticles of noble metals such as Au, Ag, and Cu are of great interest because of their unique optical, electrical, and subsequent application in chemical sensors, biosensors, photonic devices, resistive random access memory (RRAM) and DNA engineering, etc., due to their unique property of surface plasmon resonance (SPR). However, the SPR properties of these nanoparticles depend upon

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their size and shape as well as the dielectric constant of the medium [[1\]](#page-9-0). Altering the shape and size of nanostructures in the form of nanorings, nanorods, and nanospheres is quite tricky to get desired SPR properties and requires the use of various chemical and physical methods [[2\]](#page-9-1). The efect of the annealing process on the evolution of Au nanorings at the surface of indium tin oxide (ITO) was reported by Ruffino et al. [\[3](#page-9-2)]. The size of nanoparticles and their distribution depends upon the environment in which they are grown. A recent report by Meischein et al. shows the efect on the size distribution of Ag, Au, and Cu in diferent ionic liquids [[4\]](#page-9-3). On a similar note, the role of viscosity of ionic liquids surface in controlling the size of Au nanoparticles deposited by sputtering has been reported in the literature [[5](#page-9-4), [6](#page-9-5)]. In addition to this, the efect of annealing temperature and the atmosphere was investigated by M. Bechelany and they observed the transformation of the honeycomb mask of Au nanoparticles deposited by sputtering into the hexagonal arrays [[7\]](#page-9-6). Mishra et al. investigated the efect of annealing temperature and annealing conditions on the formation of Au nanorings on the quartz substrate. In another report, a systematic investigation was made on tuning localized surface plasmon resonance (LSPR) properties of Au nanoparticles embedded in the ZnO matrix with varying temperatures from 200 to 600 °C. The redshift in LSPR was observed with increasing annealing temperature on account of the agglomeration of Au nanoparticles. The process of difusion of



small Au nanocrystals during the annealing was responsible for the growth of Au nanoparticles. In another examination, the formation of silver nano cups on the surface of quartz using ultra-thin silver flm and their subsequent annealing in an inert environment was also investigated by Mishra et al. The atomic force microscopy (AFM) measurements show the presence of unsymmetric islands in the as-deposited film, while annealing at 900  $\degree$ C in Ar environment shows the evolution of nanocup-type structure. Ostwald ripening and buckling phenomenons were responsible for their evolution because of the existence of a metastable state during the process of annealing [[8–](#page-9-7)[10\]](#page-9-8). Pannu et al. demonstrated the formation of Au nanoparticles on graphene substrate by the way of varying thin flm thickness which was annealed at 400 °C. The formation of Au nanoparticles with optimum thickness was responsible for engineering strain in graphene nanosheets, which in turn alters the semiconducting properties [[11](#page-9-9)]. The formation of Ag nanorings was illustrated by Mohapatra et al. The nanorings are formed by the atom beam co-sputtering method through the self-assembly of Ag nanoparticles embedded in the silica matrix. The study reveals that the change in ring dimensions is due to a change in surface difusivity of Ag nanoparticles in diferent substrates that is carbon and silica substrates [\[12](#page-9-10)]. Khan et al. adopted the glancing angle deposition method to develop aligned nanodots on a rippled silica substrate. Low energy ion beam irradiation was used to fabricate the rippled silica substrates. An atomistic simulation was performed on the annealing and deposition process by using Monte Carlo techniques to develop aligned nanodots. The study reveals that the rippled silica template/transparent substrate could be of great interest to obtain aligned metal nanoparticles with narrow size distribution for application in biosensing [[13\]](#page-9-11). Singhal et al. reported the formation of  $Au-C_{60}$  nanocomposite by thermal co-evaporation followed by thermal annealing treatment from 150 to 300 °C. Evolution of broad SPR peak was observed at 250 °C and with the rise in temperature, a blue shift in SPR peak was realized due to transformation of  $C_{60}$ fullerene matrix into amorphous carbon [\[14](#page-9-12)]. All the reports discussed above confrm the signifcant role of annealing

parameters and the substrate on the plasmonic properties of metal nanoparticles.

ITO (Sn-doped indium oxide)-coated glass substrates are most commonly used to prepare thin flms for their application in resistive random access memory (RRAM), solar cell electrodes, photoelectrochemical water splitting studies, etc. [[15](#page-9-13)[–17](#page-9-14)]. The authors have prepared Au nanoparticles by various routes such as electrodeposition and citrate reduction method and examined their role in enhancing photoelectrochemical response for hydrogen generation for metal oxides viz. Fe<sub>2</sub>O<sub>3</sub> and BiVO<sub>4</sub>/Fe<sub>2</sub>O<sub>3</sub> heterojunction, respectively [[17,](#page-9-14) [18\]](#page-9-15). The motivation behind the present investigation is to get the optimum thickness, and annealing parameters to derive maximum SPR efect by adopting the sputtering method to deposit Au nanoparticles on ITO-coated glass substrates. The parameters so obtained were utilized to prepare thin films of Au/TiO<sub>2</sub> which were further irradiated with 500 keV  $Ar^{2+}$  ion beams for their application in PEC water splitting for hydrogen generation, and the results have been reported by the authors [[19\]](#page-9-16).

The present work reports the highlights of the efect of flm thickness, annealing temperature, and time on the evolution of Au nanoparticles on ITO-coated glass substrate deposited by the sputtering method. Various sets of samples prepared under the present investigation are shown below in Fig. [1.](#page-1-0)

The absorbance spectrum of all 27 samples along with FE-SEM imaging was performed. The thickness of the sample was confrmed by Rutherford backscattering (RBS) measurements.

#### **Experimental**

Au thin flms were deposited under inert conditions using Q150T S, Quoram Technologies, a high vacuum turbo pumped sputter coater system available at Inter-University Accelerator Center, New Delhi. Plasma in the deposition chamber was created using Ar gas under vacuum conditions and by increasing time of deposition time from 10 to 50 s to

<span id="page-1-0"></span>**Fig. 1** Schematic presentation of samples prepared by adopting diferent thicknesses and annealing parameters



deposit three diferent thicknesses of 2 nm, 5 nm, and 8 nm on cleaned ITO-coated glass substrates. The as-deposited thin flms were then annealed for the diferent time durations (30 min, 60 min, and 90 min) and at each annealing temperature, viz. 400 °C, 500 °C, and 600 °C in an inert  $N_2$ atmosphere.

Thin flms so obtained were characterized by FE-SEM instrument (MIRA II LMH from TESCAN) for surface morphology at IUAC, New Delhi. The chemical analysis of the samples was carried out using Oxford Instruments energy dispersive spectrometry (EDS) attached with the AZTEC program at a working distance of 15 mm, voltage 15 kV, and X-ray counts up to 90 s, utilizing a facility at the Department of Chemistry, DEI, Dayalbagh, Agra. The elemental peaks were resolved with the AUTOID program of the system. EDS facility is the ftment of a JEOL JSM-7610F PLUS feld emission scanning electron microscope (FESEM).

The samples were also observed in XRD, which was performed with Bruker D8 AdVANCE with Cu Kα radiation source with 2θ scan range of 20–60° with a scan speed of 2 s/ step and step size of 0.05°. UV–visible optical absorbance spectroscopy was performed by a double-beam UV–visible spectrometer, UV-1800, Shimadzu, Japan. XRD and optical absorbance facilities were availed at the Department of Chemistry, DEI, Dayalbagh, Agra.

The areal concentration of Au in the sample was determined by RBS experimental technique where a 2-MeV He ion beam from a 1.7-MV Pelletron accelerator was used to record the Rutherford backscattering spectrum. A silicon surface barrier detector having a resolution of 20 keV was used for fnding backscattered He ions at an angle of 165° from the beam direction with a solid angle of 1 msr. The SIMNRA software, which is a Microsoft Windows program, is utilized to simulate and analyze the backscattering spectra. RBS measurements were performed using tandem accelerator facilities available at Inter University Accelerator Center, New Delhi.

## **Results and Discussion**

#### <span id="page-2-1"></span>**Microscopic Imaging and EDS Measurements**

The particle size distribution of all the samples was determined from FE-SEM images. The ImageJ software was used to get the average particle size of particles. A typical FE-SEM image of plane/pristine ITO glass substrate is given below in Fig. [2.](#page-2-0)

Figures [3](#page-3-0), [4](#page-4-0), and  $5(a-i)$  show the FESEM image for various annealing time and temperature. It is shown for all three different thicknesses. The inset of FE-SEM micrographs shows the particle size distribution.





**Fig. 2** FE-SEM micrograph of plane ITO substrate

<span id="page-2-0"></span>In the case of a 2-nm-thick Au flm, the average particle size lies in the range of 24–34 nm. From the particle size distribution shown in Fig. [6](#page-6-0), with an increase in time and temperature for annealing, the particle size tends to increase. A similar trend of increase in particle size with a rise in temperature rise on  $SiO<sub>2</sub>$  substrate was reported by Kwon et al., which is showing an increasing trend [[5\]](#page-9-4).

As can be seen in Figs. [4](#page-4-0) and [5](#page-5-0), for 5- and 8-nm thicknesses at the lowest annealing temperature of 400 °C, dewetting was observed. An increase in annealing time for 5-nmthick Au flm at 500 °C led to an increase in particle size from 77 to 83 nm. At  $600 \degree C$ , 78 to 94 nm increase in particle size was observed by increasing annealing time. When the duration was raised to 90 min, particle fragmentation was observed, indicating that high temperature and time for annealing are responsible for the dissolution of particles at the surface.

At the highest thickness, i.e., 8 nm taken in the present study, dewetting was observed at the lowest temperature of 400 °C. On increasing annealing temperature to 500 °C, the particle size of 171 and 170 nm was obtained at 60- and 90-min annealing time, while for 30 min annealing time average particle size was 256 nm, showing agglomeration of Au nanoparticles. On further increasing annealing temperature up to 600 °C, particle dissolution was observed showing a reduction in particle size.

EDS measurements of pure ITO glass and Au thin flm of 5 nm thickness (500  $^{\circ}$ C, 90 min) and 8 nm thickness (400 °C, 30 min) were also performed as shown below in Fig. [7.](#page-6-1) Pure ITO substrate shows the peak of In, O, and Sn while the peak of Au is clearly visible in Au deposited and



<span id="page-3-0"></span>**Fig. 3 (a-i)** FE-SEM micrographs of 2-nm-thick Au flm annealed for diferent time durations at varied annealing temperatures

annealed on ITO glass substrate. The EDS spectrum shows 5.4 wt% of Au in 5 nm thickness flm and 9.6 wt% of Au in 8-nm-thickness flm.

# <span id="page-3-1"></span>**Optical Measurements**

To confrm the efect of annealing on sputter-deposited nanoparticles, the absorbance spectra of plane ITO sheet and as-deposited flms were recorded and are presented in Fig. [8](#page-6-2). It is clearly visible from the fgure that the absorbance spectra of as-deposited Au thin flms match well with the plane ITO, which confrms the absence of nanoparticle formation that was responsible for the SPR peak.

Figure [9a](#page-7-0)–c shows absorbance spectra for 2-nm-thick flm. The broad absorbance is visible on account of the variable size of particles up to 500 °C. At a high temperature of 600 °C, maximum absorbance was attained at 90 min annealing time which shows direct relation with increasing particle size.



<span id="page-4-0"></span>**Fig. 4 (a-i)** FE-SEM micrographs of 5-nm-thick Au flm annealed for diferent time durations at varied annealing temperatures

In the case of 5-nm-thick films, absorbance at low temperature does not show SPR due to dewetting of flms (Fig. [9](#page-7-0)d). At 500 and 600 °C, maximum intensity SPR peaks are visible at 90 and 60 min annealing time, respectively, and the SPR peak intensity shows direct dependency on the particle size of nanoparticles so formed (Fig. [9e](#page-7-0), f).

Likewise for 8-nm-thick film, dewetting was prominent at a low annealing temperature of 400 °C. At high temperatures, particle size shows harmony with the calculated average particle size. The shift in SPR peak toward higher wavelength was observed due to the bigger-sized Au nanoparticles, and in addition to this, reduced absorbance is a result of prevalent light scattering by larger size Au particles.

From the UV–vis spectroscopy, it is clear that for 2- and 8-nm-thick Au thin flms, no clear signature of SPR is as such visible. Thus, for 5-nm-thick Au flm (annealed at 500 and



**Fig. 5 (a-i)** FE-SEM micrographs of 8-nm-thick Au flm annealed for diferent time durations at varied annealing temperatures

<span id="page-5-0"></span>600 °C) was taken to see the dependence of surface plasmon resonance peak position on average particle size (Fig. [10](#page-8-0)). For 5-nm-thick Au flm annealed at 400 °C, flm dewetting was prominent. At 500 °C, with an increase in annealing time, particle size increases, and SPR peak is shifting toward higher wavelength, i.e., redshift is visible. On increasing annealing temperature to 600 °C, the redshift in SPR peak is observed up to 60 min annealing time, but for 90 min annealing, particle size reduces from 94 (60 min annealing) to 86 nm on account of particle dissolution, which results in a shift of SPR peak from 633 to 629 nm. This can also be inferred that high annealing temperature does not infuence SPR peak position at large with increasing annealing time.

#### **RBS Measurements**

The RBS spectra of three thicknesses of Au flms on ITOcoated glass are shown in Fig. [11](#page-8-1), and all RBS peaks related to Au, Sn, In, Si, and O have been assigned. The area under the RBS peak reveals the areal concentration of a particular



<span id="page-6-0"></span>**Fig. 6** Average particle size distribution of **a** 2-nm-, **b** 5-nm-, and **c** 8-nm-thick flms recorded at varying annealing parameters

element. With an increase in the thickness of Au from 2 to 8 nm, the area under the Au RBS peak is increased, and this confrms that the area concentration of Au has been increased,



<span id="page-6-1"></span>**Fig. 7** EDS spectrum of **a** ITO substrate, **b** Au thin flm of 5 nm, and **c** 8 nm thickness

as also suggested by an experiment. The areal concentration of the element is calculated using the formula given below [\[20](#page-9-17)]:

$$
N_c = \frac{Y \sin \alpha}{N_p \frac{\partial \sigma}{\partial \Omega} \Omega}
$$

where  $N_c$  is an areal concentration for Au element in the RBS spectra,  $N_p$  is the number of incidents of He ions,  $\Omega$  $(=1 \text{ ms})$  is the solid angle subtended by the detector,  $\alpha$  $(=90^{\circ})$  is the angle between beam direction and the sample surface, and  $(\partial \sigma / \partial \Omega)$  is the differential Rutherford backscattering cross-section. The areal concentration of Au on the ITO-coated glass substrate is calculated and mentioned in Table [1](#page-8-2). The recorded spectrum is shown in Fig. [11](#page-8-1) confrms the presence of Au at channel no. 1800.

In literature, the RBS studies reported by the Kumar et al. reveal difusion of Ag atoms in glass as a result of post-annealing that leads to a decrease in peak height of Ag [\[21,](#page-9-18) [22\]](#page-9-19), while in the present study, as such, no change in peak intensity can be seen as an efect of post-annealing.



<span id="page-6-2"></span>**Fig. 8** Absorbance spectra of plane ITO along with as-deposited thin flms



<span id="page-7-0"></span>**Fig. 9** Absorbance spectra of Au thin flms. **a**–**c** 2 nm Au thin flms annealed at 400 °C, 500 °C, and 600 °C for diferent time durations; **d**–**f** 5 nm Au thin flms annealed at 400 °C, 500 °C, and 600 °C for

diferent time durations; **g**–**i** 8 nm Au thin flms annealed at 400 °C, 500 °C, and 600 °C for diferent time durations at varied annealing temperatures

#### **X‑ray Diffraction Studies**

To study the efect of thermal treatment on Au thin flms and, correspondingly, on ITO substrate, 5-nm-thick Au flm was selected for XRD measurements (see Fig. [12](#page-8-3)). The selected thin films were annealed for 60-min duration at 400 °C, 500 °C, and 600 °C. The XRD patterns do not show any impurity peaks except for the peaks of Au (JCPDS card no. 04–0784) and ITO (JCPDS card no. 44–1087), which confrms that annealing at high temperatures does not afect the crystallinity of Au as well as ITO substrate. Moreover, ITO is stable up to  $600 \text{ °C}$  [\[23,](#page-9-20) [24](#page-9-21)]. In the case of Au, it can be seen that with a temperature rise, Au peak intensity is increasing on account of an increase in particle size. The fndings match the average particle size obtained from FESEM analysis. Kumar et al. also reported an increase in peak intensity of Ag nanoparticles with an increase in annealing temperature [\[21\]](#page-9-18).

#### **A Mechanism for the Growth of Nanoparticles**

The process of thermal annealing affects the optical properties of noble metal particles. Cu nanoparticles deposited on phosphate glass, borosilicate glass, and natural plagioclase mineral crystal have been investigated regarding the role of thermal annealing on SPR evolution in-depth in past reports by various authors [[25](#page-9-22)–[27](#page-9-23)]. Likewise, the efect of thermal annealing in silver nanocrystals in sodalime glass, and Au nanoparticles in sodium zinc borate glass have been studied in detail by Kumar et al. [[21,](#page-9-18) [22,](#page-9-19) [28,](#page-9-24) [29](#page-9-25)].

From the reports discussed above, this can be said that role of thermal annealing is well established in afecting the SPR properties of Cu, Ag, and Au nanoparticles. In the present study, the optical absorbance spectra of asdeposited Au thin flms of 2 nm, 5 nm, and 8 nm thickness



<span id="page-8-0"></span>**Fig. 10** SPR peak versus average particle size distribution plot for 5-nm-thick Au flm annealed at 500 and 600 °C

was matching with the ITO substrate except there is an increase in peak intensity with increasing thickness of Au flm (see Fig. [8\)](#page-6-2). During the process of thermal annealing, there starts a process of formation of nanoparticles and this is clearly visible from the evolution of SPR peaks in optical absorbance spectra provided in Fig. [9.](#page-7-0) Though for the lowest thickness of 2 nm Au flm, particles of 24–34 nm



<span id="page-8-1"></span>**Fig. 11** RBS spectrum of Au thin flms

<span id="page-8-2"></span>**Table 1** Thickness of Au thin flms by SIMNRA simulation

<b>Film identity</b>	<b>Areal concentration</b>	<b>Film thickness</b>
$2 \text{ nm}$ , 500 °C, 60 min	$17 \times 10^{15}$ atoms/cm <sup>2</sup>	$2.8 \text{ nm}$
5 nm, 500 °C, 90 min	$41 \times 10^{15}$ atoms/cm <sup>2</sup>	$6.9 \text{ nm}$
8 nm, 400 °C, 30 min	$63 \times 10^{15}$ atoms/cm <sup>2</sup>	$10.6$ nm



<span id="page-8-3"></span>**Fig. 12** XRD plots of 5 nm Au thin flm annealed for 60 min at 400 °C, 500 °C, and 600 °C

size are present. In the case of 5-nm- and 8-nm-thick Au flm, initially, a low temperature of 400 °C does not facilitate nanoparticle formation even for a longer duration of annealing, and flm dewetting was noticed. But, as the temperature rises to 500 °C and 600 °C, the formation of nanoparticles is visible and has been discussed in detail in the "[Microscopic Imaging and EDS Measurements"](#page-2-1) and "[Optical Measurements](#page-3-1)" sections. During the process of annealing, particle growth is observed by the agglomeration process up to a limited annealing duration by a process of Ostwald ripening, and then the dissolution of the particle is seen as a consequence of low binding energy of surface atoms [[2,](#page-9-1) [8\]](#page-9-7). The growth of nanoparticles under thermal annealing is well described by Oswald ripening and the equations governing the process. The details of the process and equations have been well described in the literature [[30\]](#page-10-0). In the current investigation, ITO substrate does affect the formation of nanoparticles, yet thermal annealing does not ensue implantation of Au nanoparticles in ITO, which has been confrmed through RBS and XRD studies.

## **Conclusion**

The investigation on annealing of Au thin flms with different thicknesses for diferent time durations revealed that 5 nm thickness annealed at 500 °C temperature for 90 min duration offers the best SPR effect. Besides this, it was also observed that dewetting of flms occurs at low annealing temperature while increased temperature and annealing time leads to particle fragmentation. The study aimed to obtain suitable conditions to use these nanoparticles as visible-light sensitizers.

**Acknowledgements** The authors would like to thank Dr. Sunil Kumar for FE-SEM characterization and Mr. Sunil Ojha for RBS measurements at IUAC, New Delhi. Author AV would like to thank Dr. M. Shyam Prasad for carrying out EDS measurements at Department of Chemistry, Dayalbagh Educational Institute, Agra.

**Author Contribution** Anuradha Verma: conceptualization, methodology, writing—original draft preparation, investigation. Sakshi Saxena: data curation, editing, and formal analysis. Neeraj Kumar Biswas: visualization and data curation. Anupam Srivastav*:* conceptualization and visualization. Uday Bhan Singh: software and validation. Saif A Khan: visualization and data curation. Rohit Shrivastav: data curation and visualization. Devesh Kumar Avasthi: supervision, writing—reviewing and editing, conceptualization, and validation. Sahab Dass: supervision, research activity planning, writing—reviewing and editing.

**Funding** We thank fnancial support by MNRE, Govt. of India under the project (103/241/2015-NT).

**Data Availability** The data used to support the fndings of this study are included within the article and no additional source data are required.

#### **Declarations**

**Competing Interests** The authors declare no competing interests.

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