

## Formation and biocorrosion behavior of Zr-Al-Co-Nb bulk metallic glasses

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Ni- and Cu-free Zr-Al-Co-Nb glassy alloys with different Nb and Co contents were synthesized by melt spinning and copper mold casting. The effects of Nb addition to partially replace Co in the  $Zr_{55}Al_{20}Co_{25}$  glassy alloy on the glass-forming ability, thermal properties, *in-vitro* biocorrosion behavior and surface wettability of the metallic glasses were investigated. Although addition of Nb up to 5 at.% slightly decreased the supercooled liquid region and the glass-forming ability (GFA), the alloys could be casted in a bulk glassy rod form with diameters up to 3 mm. The Zr-Al-Co-Nb glassy alloys were spontaneously passivated with low passive current densities in phosphate buffered saline and Hanks' solution. Substitution of "toxic" Co by Nb is effective in improved the corrosion resistance of the Zr-Al-Co glassy alloy. Water contact angle measurements showed that Nb addition increased the hydrophilicity of the glassy alloys, which may enhance cell adhesion of the alloys in biomedical applications.

**Zr-based alloy, Nb, metallic glass, corrosion, biocompatibility**

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Because of the lack of long-range atomic order, Zr-based bulk metallic glasses (BMGs) exhibit superior strength, high elastic strain limits and fatigue endurance limits, relatively low Young's modulus, and excellent wear and corrosion resistance [1–5], which attracted attention in biomedical applications. However, toxic elements such as nickel and copper are often included in Zr-based metallic glasses to achieve high glass-forming ability (GFA), which are thought to impair cellular metabolism [6]. In recent years, many efforts have been made to remove these elements from Zr-based BMGs to further improve their biocompatibility. Major efforts have been made in removing Ni from the alloying elements, because Ni is a known allergen. Ni-free glass forming systems such as Zr-Al-Cu [7], Zr-Al-Co [8–11], Zr-Al-Co-Cu [12], Zr-Al-Fe-Cu [13], Zr-Cu-Pd-Al-Nb [14], Zr-(Cu, Ag)-Al [15], Zr-Al-Cu-Fe-(Ti/Nb) [16], and Zr-Al-Co-Ag [17,18] have been developed. Among the

Ni-free Zr-based BMGs, Zr-Al-Co BMGs are attractive because they are free from copper, which may result in high cytotoxicity if excessive release occurs during corrosion and is also an element of concern in biomaterials [19]. Based on the ternary  $Zr_{55}Al_{20}Co_{25}$  alloy with high GFA, quaternary  $Zr_{55}Al_{20-x}Co_{25}Nb_x$  ( $x = 2.5$  at.% and 5 at.%) metallic glasses have been developed, and addition of Nb were found to enhance their resistance to pitting corrosion in 3 mass% NaCl solution [20]. From the viewpoint of biomedical engineering, Nb is recognized as a highly biocompatible element. In the quaternary Zr-Al-Co-Nb alloys, Co is more cytotoxic than Al, and excessive amounts of Co or prolonged exposure to Co may result in Co poisoning [21–23]. Therefore, rather than substitution of Al with Nb as in the previous study [20], substitution of Co with Nb could further improve the biocompatibility of the alloys. In consideration of potential medical applications, the GFA and *in vitro* biocorrosion behavior of  $Zr_{55}Al_{20}Co_{25-x}Nb_x$  glassy alloys were investigated in the present work. Surface wettability of the

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$Zr_{55}Al_{20}Co_{25-x}Nb_x$  glassy alloys was also studied, which can be related to cell attachment/adhesion with relatively higher hydrophilicity enhancing cell adhesion [24,25]. The effects of substitution of Co with Nb on the formation and corrosion resistance of the Zr-Al-Co glassy alloy are discussed.

## 1 Experimental

Master alloys with nominal compositions of  $Zr_{55}Al_{20}Co_{25-x}Nb_x$  ( $x=0, 2.5$  at.% and 5 at.%) were prepared by arc melting mixtures of the pure metals in a Ti-gettered high-purity argon atmosphere. Ribbon samples (thickness 0.02 mm, width 1 or 4 mm) and cylindrical alloy rods with different diameters were prepared from crushed alloy ingots in an argon atmosphere by melt spinning and injection copper mold casting, respectively. The microstructures of the specimens were examined by X-ray diffraction (XRD) with Cu-K $\alpha$  radiation. The thermal stability of the alloys was investigated by differential scanning calorimetry (DSC) at a heating rate of 0.33 K/s.

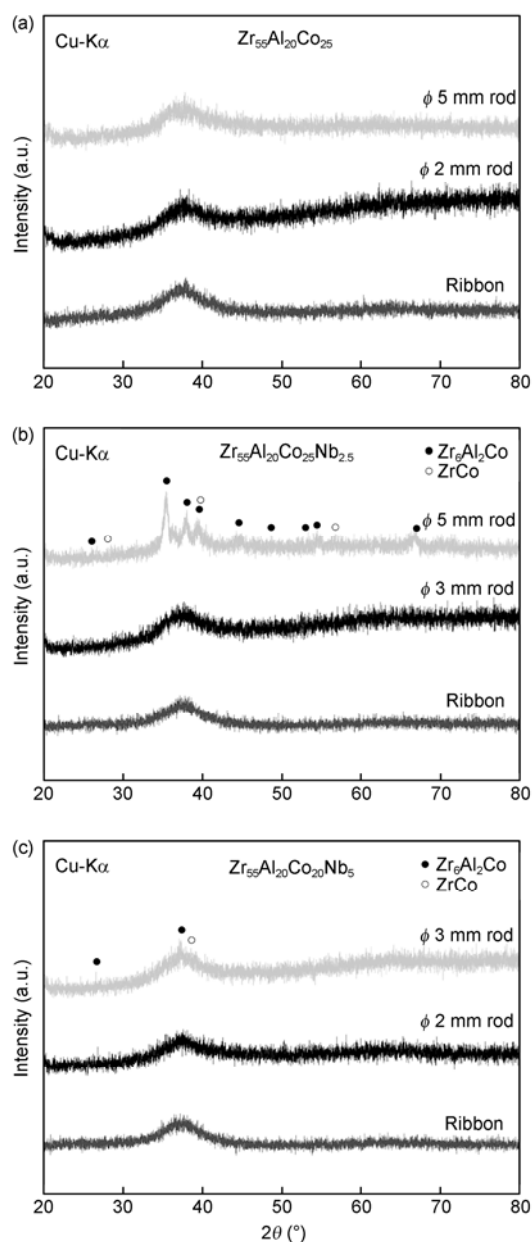
The electrochemical behaviors of the Zr-Al-Co-Nb glassy alloys were characterized using ribbon samples of 1 mm wide in phosphate-buffered saline (PBS) and Hanks' balanced salt solution. Before the tests, the solutions were heated to 310 K in a water bath and aerated with 4 vol.%  $O_2/N_2$  gas mixture at a flow rate of 50 mL/min. Aeration was continued throughout the tests. The surface of each specimen was mechanically polished in cyclohexane with silicon carbide paper up to # 2000 (average particle diameter 10  $\mu$ m), degreased in acetone, cleaned in ethanol and distilled water, dried in air, and then exposed to air for about 24 h for good reproducibility. A three-electrode cell system consisting of a corrosion sample as the working electrode, a saturated calomel reference electrode (SCE,  $E_{SCE}=0.242$  V), and a platinum counter electrode was adopted. Prior to potentiodynamic polarizations, the test samples were immersed in the electrolytes for 4000 s to allow open circuit potentials (OCPs) to reach steady state. Anodic potentiodynamic polarization tests were started at about 0.05 V below the OCPs at a potential sweep rate of 50 mV/min.

For the characterization of the surface wettability of the glassy alloys, water contact angles were measured using the sessile drop technique at 298 K with 4-mm-wide ribbon samples. The surface of each ribbon sample was polished with silicon carbide paper up to # 2000, cleaned in an ultrasonic cleaner with acetone, ethanol and distilled water, and then dried in air. Deionized water droplets (about 0.5  $\mu$ L) were dropped carefully onto the surface. Measurements were performed at five different positions along each sample and the average contact angle was calculated.

## 2 Results and discussion

XRD patterns of the  $Zr_{55}Al_{20}Co_{25-x}Nb_x$  (at.%) alloy samples

with different geometries are shown in Figure 1. The XRD patterns of the  $Zr_{55}Al_{20}Co_{25}$  alloys with diameters up to 5 mm show a broad peak at around  $2\theta=37^\circ$ , which is characteristic of the amorphous structure. Nb addition negatively impact glass formation of the Zr-Al-Co alloy. Substitution of Co by Nb at 2.5 at.% decreases the critical diameter for glass formation to 3 mm. Crystallization peaks for  $Zr_6Al_2Co$  and ZrCo can be observed in the diffraction pattern of the  $\phi$  5 mm sample. When the Nb content is increased to 5 at.%, the critical diameter of the  $Zr_{55}Al_{20}Co_{20}Nb_5$  BMG decreases to 2 mm. The XRD patterns of partially crystallized Zr-Al-Co-Nb alloys indicate that Nb addition influences the GFA of the Zr-Al-Co BMGs by enhancing formation of  $Zr_6Al_2Co$



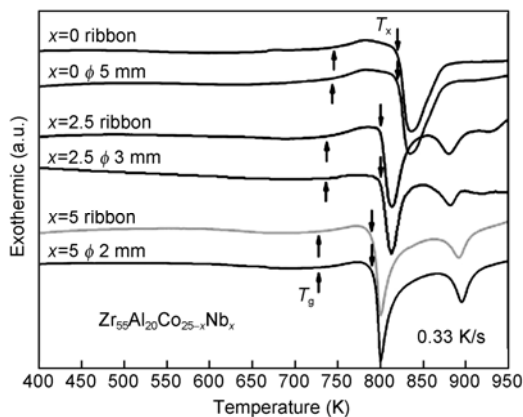
**Figure 1** XRD patterns of  $Zr_{55}Al_{20}Co_{25-x}Nb_x$  ( $x=0, 2.5$  at.% and 5 at.%) ribbon and rod samples.

and ZrCo phases.

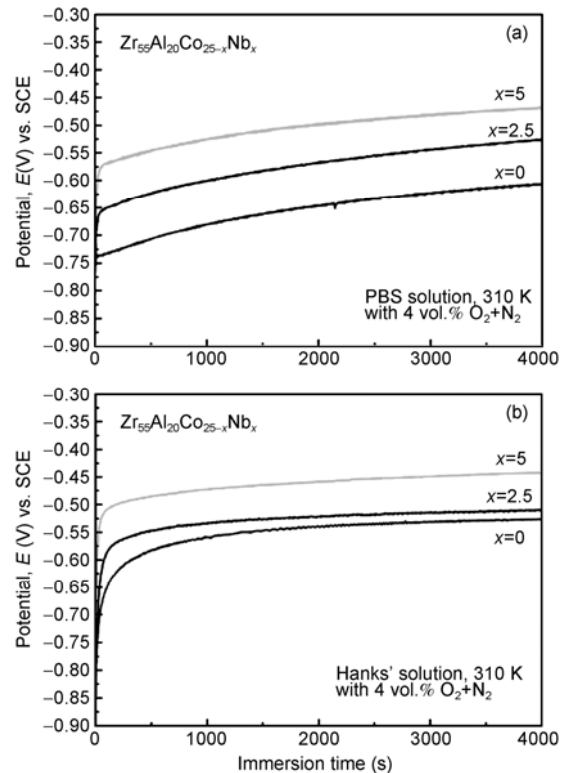
DSC curves of the Zr-Al-Co-Nb alloy ribbons and rods with their critical diameters are shown in Figure 2, where  $T_g$  and  $T_x$  are glass transition temperature and onset temperature of crystallization, respectively. Upon heating, each of the samples exhibits a distinct glass transition followed by a wide supercooled liquid region ( $\Delta T_x = T_x - T_g$ ) before crystallization. The values of  $T_g$ ,  $T_x$ , and heat release of the crystallization for the rod samples are almost identical to those of the ribbon samples with the same compositions, which further confirms the glassy structure of the bulk alloys. With the increase in Nb content,  $T_g$  decreases from 746 to 728 K, and  $T_x$  decreases from 821 to 790 K, which causes a decrease in  $\Delta T_x$  from 75 to 62 K. The decrease in  $\Delta T_x$  can be correlated with the decrease in the GFA of the alloys.

To simulate body fluid, PBS and Hank's solution were used in corrosion tests. The former is a simple solution with a similar Cl<sup>-</sup> concentration and pH value to body fluid, and the later is more complex, which additionally contains Mg<sup>2+</sup>, Ca<sup>2+</sup>, and glucose. Changes in the open circuit potential with immersion time for the Zr<sub>55</sub>Al<sub>20</sub>Co<sub>25-x</sub>Nb<sub>x</sub> ( $x=0, 2.5$  at.% and 5 at.%) glassy alloys in PBS and Hank's solution are illustrated in Figure 3. During immersion, the potential of each alloy initially rises steeply and then slows down until reaches constant values, which are determined as OCPs. The OCP values of the alloys in both solutions increase with the increase in Nb content, indicating that Nb addition improved the stability of the surface films of the alloys.

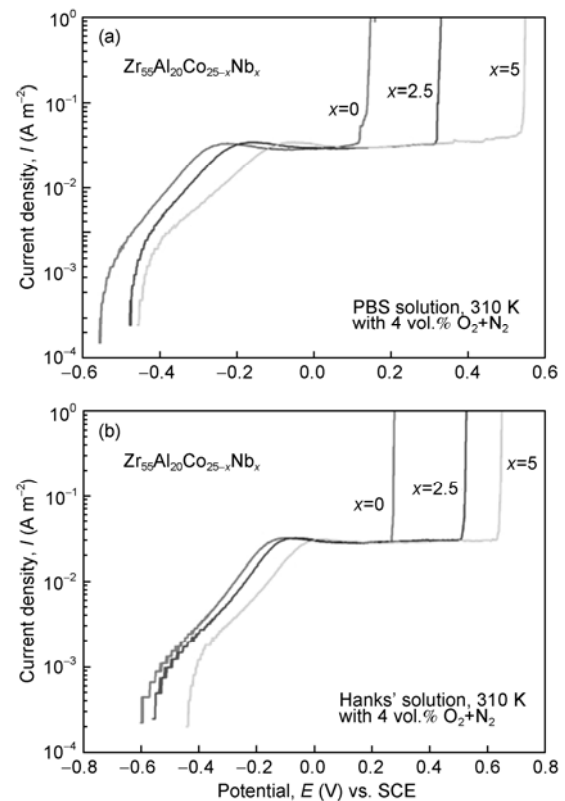
Figure 4 shows the anodic polarization curves of the glassy alloys in PBS and Hank's solution at 310 K with aeration of O<sub>2</sub>/N<sub>2</sub> gas mixture at 50 mL/min. All alloys are spontaneously passivated with wide passive regions before pitting in both electrolytes, and their passive current densities are around 0.08 A/m<sup>2</sup>. Figure 5 shows that the pitting potential of the Zr<sub>55</sub>Al<sub>20</sub>Co<sub>25-x</sub>Nb<sub>x</sub> BMGs increases considerably with increasing Nb content. Therefore, partial substitution of Co by Nb effectively improves the corrosion resistance of the Zr<sub>55</sub>Al<sub>20</sub>Co<sub>25</sub> BMG in the simulated body



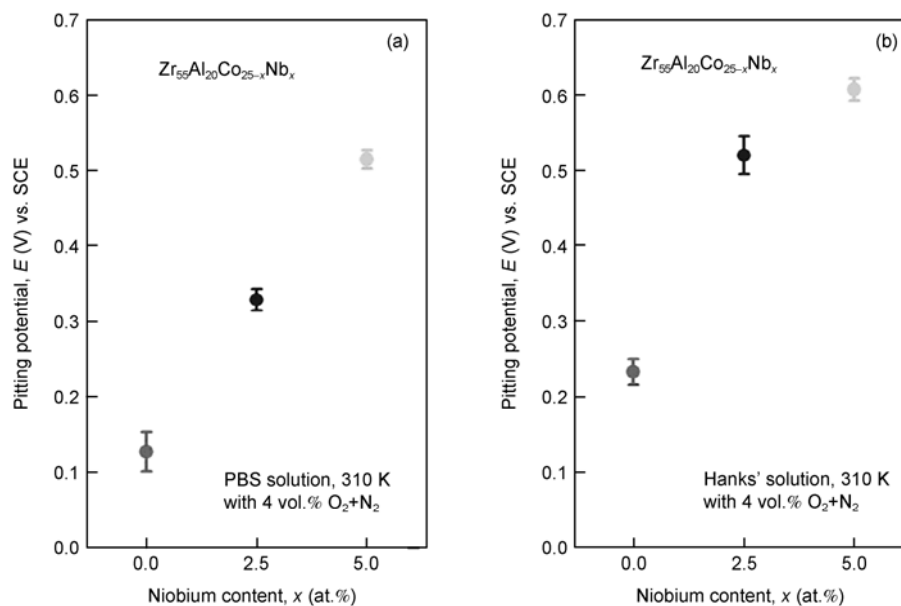
**Figure 2** DSC curves of Zr<sub>55</sub>Al<sub>20</sub>Co<sub>25-x</sub>Nb<sub>x</sub> ( $x=0, 2.5$  at.% and 5 at.%) ribbon and rod samples.



**Figure 3** Changes in the open-circuit potential with immersion time for Zr<sub>55</sub>Al<sub>20</sub>Co<sub>25-x</sub>Nb<sub>x</sub> ( $x=0, 2.5$  at.% and 5 at.%) glassy alloys in PBS and Hank's solution.



**Figure 4** Anodic polarization curves of Zr<sub>55</sub>Al<sub>20</sub>Co<sub>25-x</sub>Nb<sub>x</sub> ( $x=0, 2.5$  at.% and 5 at.%) glassy alloys in PBS and Hank's solution.



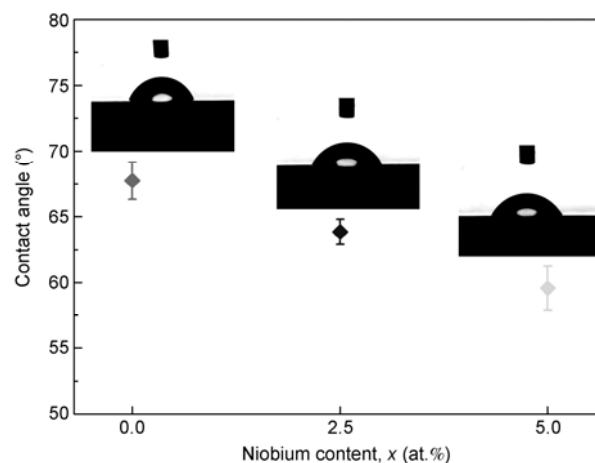
**Figure 5** Changes in the pitting potentials of  $Zr_{55}Al_{20}Co_{25-x}Nb_x$  ( $x=0, 2.5$  at.% and 5 at.%) glassy alloys with Nb content in PBS and Hanks' solution.

fluids. The positive effect on the corrosion resistance of the Zr-Al-Co glassy alloy by Nb addition can be attributed to the change in the composition of the passive film. It has been reported that Nb addition contributes to the formation of a protective surface film with higher chemical stability [20].

Water contact angle results for the Zr-Al-Co-Nb glassy alloys are presented in Figure 6. The same abrasion procedure diminishes the effect of surface roughness on the surface wettability. The water contact angle decreases considerably with the addition of Nb, which indicates higher hydrophilicity of the Nb-containing alloys. When the Nb content reaches 5 at.%, the water contact angles become smaller than the Berg limit ( $\theta=65^\circ$ ), which is beneficial for biomedical applications [26]. The present experimental results indicate that Nb addition simultaneously improves both the hydrophilicity and the corrosion resistance of the Zr-Al-Co-Nb glassy alloys. This is different from an earlier finding that related high corrosion resistance with low hydrophilicity for ZrTiCuNiBe metallic glasses [27]. The improvement in the corrosion resistance of the Zr-Al-Co glassy alloy with Nb addition is attributed to the change in the composition of the passive film, because Nb addition to Zr-based glassy alloys induces the formation of a more protective passive film enriched in Nb [20]. Further research will be carried out to clarify whether the composition or wettability is dominant in determining the corrosion resistance of various glassy alloy systems.

### 3 Conclusions

Ni- and Cu-free  $Zr_{55}Al_{20}Co_{25-x}Nb_x$  ( $x=2.5$  at.% and 5 at.%)



**Figure 6** Contact angles of water droplets measured on  $Zr_{55}Al_{20}Co_{25-x}Nb_x$  ( $x=0, 2.5$  at.% and 5 at.%) glassy alloys.

BMGs with critical diameters up to 3 mm were developed. The glassy alloys exhibited high thermal stability, and good biocorrosion resistance and surface wettability. Nb addition was slightly detrimental to the GFA. The corrosion resistance of the Zr-Al-Co BMG in PBS and Hanks' solution was improved considerably with the increase in Nb content, which was indicated by the widening of the passive region, and the increases in pitting potential and OCP. Water contact angle measurements showed that Nb addition improved the hydrophilicity of the alloy, which is beneficial for biomedical applications.

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