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# Flexible and stretchable metal oxide gas sensors for healthcare

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Capable of measuring volatile biomarker produced by the metabolism from several secretion pathways, flexible and stretchable metal oxide gas sensors have received increasing attention and their development for healthcare starts to gain momentum. Integration of semiconducting metal oxide on a soft, thin, flexible substrate is the key to enable the flexible property to the gas sensor and such integration typically involves either a direct growth or post transfer of the metal oxide on or to the flexible substrate. In addition to the planar plastic substrate, textile represents another important class of flexible substrates due to its ease of integration with clothing. Moreover, the integration of metal oxide on a single fiber provides a great versatility for different applications. Though flexible sensors can easily conform to the developable surface (e.g., cylinder or cone) from a bending deformation, the conformal contact between the sensor and the tissue surface that is often non-developable requires the sensor to be capable of stretching. Due to the intrinsically brittle nature of the semiconducting metal oxide, several stretchable structures have been explored. Despite the great strides made to the burgeoning area of flexible and stretchable metal oxide gas sensors, grand challenges still need to be overcome before the technology can be applied for the practical application. The selected challenges discussed in this mini-review also represent a fraction of possibilities and opportunities for the research community in the future.

metal oxide gas sensor, flexible and stretchable properties, structural design strategies, volatile biomarker

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## 1 Introduction

Increasing attention has been paid to personal healthcare and human-activity monitoring [1,2]. As an important class in the bio-integrated electronic devices, wearable gas and vapor sensors are being explored to detect gases from the human body and environment for early warning and personal protection [1,3–5]. The volatile biomarker produced by the metabolism has long been known in the medical sciences. The detection of the volatile organic compounds is possible due to several secretion pathways, including breath [6], skin perspiration [7], saliva [8], urine [9,10], and others [11].

Despite the high potential, the continuous monitoring of volatile biomarker has been limited by the challenge of the ultra-low concentration of gas molecules in the complex mixture. Therefore, the design and development of wearable gas and vapor sensors are driven by high sensitivity and selectivity, as well as by low modulus, lightweight, and flexible and stretchable properties of the integrated system.

The gas sensor needs to first recognize the target gas and then convert the recognition event to a measurable signal. Various types of gas sensors have been explored with diverse working principles and methods of integration, resulting in differences between their performances, applicable gas, and occasions. Among the gas sensors that are based on quartz crystal microbalance [12,13], surface and bulk acoustic wave

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devices [14,15], calorimetric sensors [16], electrochemical cells (e.g., lambda probe [17,18]), and gas-sensitive field-effect transistors [19], chemiresistors have attracted considerable attention in both scientific and practical applications [20,21]. Besides providing real-time changes in their electrical resistance in response to the nearby chemical (gases and vapors) environment, chemiresistors are in-expensive and require just simple DC electronics and small devices suitable for miniaturization [22].

Metal oxides, one representative class of chemiresistors, change their electrical resistance upon interaction with reducing or oxidizing gases. The metal oxide gas sensors are typically associated with short response time, accurate conversion of signals, high sensitivity, reversibility, low-cost manufacturing [22-24], and substantially improved selectivity [25], resulting in extensive studies and widespread applications. The common target gases with their physical properties and threshold values for applications in healthcare are listed in Table 1. Though the list is not exhaustive, this mini-review will focus on the discussion of a few representative examples from this list. Due to the intrinsically brittle nature of metal oxide, it is challenging to implement metal oxide gas sensors in a wearable form. Recent advances in materials and mechanics have provided effective means for the design and fabrication of the flexible and stretchable metal oxide gas sensors. The applications of these novel metal oxide gas sensors also go beyond biomedicine to detecting explosion [26], alarming leakage of gaseous hazards [27,28], monitoring gas emission from vehicles [29,30], and etc. Having addressed several key issues that are directly relevant to the wearable gas sensors [1,2,11,31-34], this new class of wearable devices starts to gain momentum. A brief

 Table 1
 List of common target gases measured for applications in healthcare<sup>a)</sup>

highlight of the recent studies is provided here to help facilitate further advances in this burgeoning area. In this minireview, we begin with a brief introduction of the sensing mechanism of the semiconducting metal oxide gas sensors in Section 2. Next, the various strategies and corresponding design considerations of flexible and stretchable metal oxide gas sensors are provided in Section 3 and Section 4, respectively. The review is concluded with a brief discussion on the future perspectives with challenges and potential opportunities for the development of flexible and stretchable metal oxide gas sensors for healthcare.

# 2 Sensing mechanism of semiconducting metal oxide gas sensors

The working principle of a typical resistive semiconducting metal oxide has been discussed in great detail in several references [40,48,49]; thus, only brief highlights will be provided here. The change in resistance of the sensor material is based on a shift in the equilibrium states of the surface oxygen reaction due to the presence of the target gas analyte. The reducing gas results in an increase of the conductivity for n-type semiconductors and a decrease for ptype semiconductors, whereas the effect of the oxidizing gas is vice versa. For instance, oxygen is ionosorbed on the metal oxide surface in presence of oxygen and absence of any humidity for SnO<sub>2</sub>, a wide-bandgap n-type semiconducting material widely used for gas sensing. Acting as electron acceptors, the ionosorbed oxygen species extract electrons to create an electron-depleted surface region (depth of air or the Debye length that is typically on the order of 2–100 nm),

Target gases	Physical properties	Threshold values	Applications in healthcare	Ref.
H <sub>2</sub>	Colorless, highly flammable, readily forms explosive mix- tures with air	20 ppm above basal	Diagnosis of metabolic disor- ders or diseases	[35,36]
NO <sub>2</sub> /NO	Brown gas/colorless, smell like chlorine, poison, oxidizer	206 ppt	Lung injury	[37,38]
CO	Colorless, odorless, toxicity	0.1 ppm	Diagnosis of hemolytic disease	[39,40]
CO <sub>2</sub>	Colorless, odorless at low concentration	60 ppb	Respiration monitoring	[41]
$H_2S$	Colorless, rotten eggs smell, flammable and highly toxic	534 ppt	Asthma, diagnosis of halitosis and lung cancer	[37]
NH <sub>3</sub>	Colorless, strong pungent odor, flammable	10-50 ppb	Liver failure	[42,43]
C <sub>2</sub> H <sub>5</sub> OH	Colorless, alcoholic smell and taste, flammable, low toxicity	0.1 ng	Alcohol consumption	[44,45]
C <sub>3</sub> H <sub>6</sub> O	Colorless, pungent odor, sweetish taste, flammable	10-30 ppb	Hyperglycemia, diabetes	[7,46,47]
Methane	Colorless, odorless, nontoxic yet extremely flammable	20 ppm above basal	Indicating lactose intolerance	[35]

a) H<sub>2</sub>: Hydrogen; NO<sub>2</sub>/NO: Nitrogen dioxide; CO: Carbon monoxide; CO<sub>2</sub>: Carbon dioxide; H<sub>2</sub>S: Hydrogen sulfide; NH<sub>3</sub>: Ammonia; C<sub>2</sub>H<sub>5</sub>OH: Ethanol; C<sub>3</sub>H<sub>6</sub>O: Acetone

with a maximum surface coverage of about  $10^{-3}$  cm<sup>-1</sup> to  $10^{-2}$  cm<sup>-1</sup> ions as dictated by the Weisz limitation [50]. Band bending owing to the negative surface charge generates a surface potential barrier of 0.5 eV to 1.0 eV. Both of the depth and height of the barrier depend on the amount and type of the adsorbed oxygen. In polycrystalline material, the electrical conductivity occurs along the percolation path via the grain-to-grain contact, which depends on the height of the surface potential barrier (or the Schottky barrier)  $eV_{\text{surface}}$ between the adjacent grains. The conductance G can be related to the Schottky barrier as  $G \approx \exp[-eV_{\text{surface}}/(k_B T)]$  [51]. Reacting with the ionosorbed oxygen species, the reducing gases (e.g., H<sub>2</sub>, H<sub>2</sub>S, CO, NH<sub>3</sub>, CH<sub>4</sub>, ethanol, acetone) decrease the adsorbed oxygen to release the surface-trapped electrons back to the bulk. As a result, the reduced Schottky barrier increases the conductance of the sensing material. In contrast, the oxidizing gases (e.g., O<sub>2</sub>, O<sub>3</sub>, NO<sub>x</sub>, CO<sub>2</sub>, or SO<sub>2</sub>) further extract electrons from the semiconductor, resulting in an increased Schottky barrier and thus a decrease in the conductance. For metal oxide in the shape of a nanosized particle, both experimental observation and theoretical evidence indicate a smaller a particle, a higher sensitivity of the sensing material. This is attributed to the fact that the nanostructure dimensions allow the entirety of the sensing materials to be within the Debye length and depleted by the gas analyte for the maximum response. However, the surface accessibility for the exposed gas to the metal oxide depends largely on the tailored properties of the sensing material, which may still include empirical approaches.

Although metal oxide gas sensors are associated with high sensitivity, low cost, and compatibility with modern electronic devices as mentioned above, their relatively poor selectivity among gases makes them impossible to use a single sensor to identify a specific gas from a mixture. One straightforward method is to exploit a suite of coordinated sensors to simultaneously analyze the signals from the array of sensors, often termed as electronic noses [52]. In order to simplify the design in the electronic nose, a separate effort is to use composite metal oxides for enhanced selectivity. Combining two dissimilar materials with a heterojunction at the physical interface can result in a heterostructure. An extensive review has discussed the heterostructure from binary compounds [53] and highlights are briefly reviewed here. Various classes of heterostructures have been achieved through (1) a simple mixture of two constitutes, (2) adding a second oxide on the base metal oxide (e.g., sputtering), and (3) a well-defined partition or interface between two oxides. When analyzing the behavior of the heterostructure for gas sensing, a heterojunction that modulates the sensing property through the p-n junction is a very common interface. In direct contrast to the most commonly used pure n-type gas sensors, the electrons at the higher energy states at the interface in the heterostructure will flow to unoccupied lower energy states (i.e. 'Fermi level-mediated charge transfer') to create a depletion region at the interface. This results in a potential energy barrier due to band bending that the electrons must overcome.

As the adsorption of the oxygen and reaction rates of the analyte gas are temperature dependent, an optimum operating temperature often exists (termed as working temperature) and designs that ensure highly uniform temperature distribution are also desirable [32]. For an n-type sensing material, the response is typically defined as the ratio of the electrical resistance in the presence of a gas to that in the air ( $R_{gas}/R_{air}$ ) for an oxidizing gas and  $R_{air}/R_{gas}$  for a reducing gas. Plotting the response of the sensor as a function of the gas concentration yields a calibration curve of the sensor, from which the calculated slope defines the sensitivity. The response (or recovery) time is typically defined as the time the sensor takes to reach 90% of its steady-state value upon the introduction (or removal) of the analyte gas.

### 3 Flexible metal oxide gas sensors

Due to the geometry mismatch between a flat gas sensor and the curvilinear surface of the human body, a wearable gas sensor is required to be capable of bending. As a structure's bending stiffness typically scales with the cubic of its thickness, reduction in thickness of the structure can significantly decrease its resistance against bending formation. Because the metal oxide gas sensor itself is already thin in its thickness, it is highly desirable to integrate the gas sensor on a thin substrate. Taken together with the fact that the reduced Young's modulus of the substrate can approximately decrease its bending stiffness in a linear manner, a class of soft polymers has been explored as the substrate for the wearable gas sensors, including polyethylene terephthalate (PET), polyimide (PI), polydimethylsiloxane (PDMS) and polyethylene naphthalate (PEN) [54]. Direct growth and posttransfer are the two commonly used strategies for the integration of metal oxide gas sensor on a flexible substrate.

#### 3.1 Direct growth of metal oxide on flexible substrates

To directly grow metal oxide materials on a flexible substrate, the processing temperatures are required to be below the glass transition or thermal degradation temperatures of the flexible substrate [55]. As an example, the single-crystalline ZnO nanowires (NWs) can grow at 80°C via electrodeposition on a transparent PET substrate coated with a patterned indium-tin-oxide (ITO) layer [56]. In a similar approach, thermolysis assisted chemical solution method is used to continuously grow ZnO nanorods on a PI film covered by an 80 nm thick ZnO seed layer from radio-frequency (RF) sputtering [57]. The flexible ZnO nanorod sensors demonstrate the sensing performance of ethanol in a range of 10–100 ppm comparable with those fabricated on a rigid silicon dioxide substrate. Moreover, the rapid hydrothermal synthesis through the use of the variable power microwave sintering system can be used to synthesize undoped ZnO NW arrays and Al-doped ZnO nanostructures (e.g., nanowires and nanosheets) on PET [58].

In addition to the plastic substrate with a thin geometry, the paper is also associated with a cost-effective manufacturing process and a high surface-to-volume ratio due to its hierarchical structure [55], suggesting it as a promising flexible substrate. When compared with the plastic substrate, however, a much lower temperature should be used for the synthesis of the metal oxide. Exploiting low-temperature (i.e., 85 and 90°C) hydrothermal technique allows the direct growth of vertically aligned single crystalline ZnO nanorods and nanoneedles on a paper substrate [59]. As a result, fabrication techniques of metal oxide gas sensors on paper substrates and the influence of paper type on sensing performance have been explored in several studies [44,60,61]. On the top surface of a flexible cellulose glossy paper substrate, a custom-built tool is used to deposit conductive carbon nanotubes as interdigitated electrodes, followed by a sensing material with ITO nanopowders dispersed in a polydiallyldimethylammonium chloride (PDDAC) polymer matrix [44] (Figure 1(a)). Because none of the two components in the sensing material shows a response to ethanol in the experimental condition, the capability of the sensor to detect low concentration of ethanol at room temperature is attributed to the electrostatic interaction between negatively charged ITO particles and positively charged polymer (Figure 1(b)).

Successful growth of one metal oxide such as ZnO nanorods provides a platform for adding a metal catalyst or a second metal oxide layer. As an example, Ag nanoparticles are sputter coated on the as-grown ZnO nanorods that are synthesized on a PI layer pasted onto a polytetrafluoroethylene (PTFE) sheet [54] (Figure 1(c)). With catalytic activities of Ag, the fabricated sensor demonstrates a high response of 27.2, short response (recovery) time of 62 s (39 s) to acetylene at 1000 ppm, and a broad detection range from 3 to 1000 ppm at a low operating temperature of 200°C (Figure 1(d)). Tests of mechanical flexibility and reliability further show a negligible drift of 2.1% for a maximum bending angle of 90° and a decrease of 8% in response after  $5 \times 10^4$  bending/relaxing cycles. In another similar effort to design flexible H<sub>2</sub> sensor [26] (Figure 1(e)), ZnO nanorods are still synthesized on a PI layer attached to a Si substrate. After sputter coating Pd nanoparticles on the as-grown ZnO, the sample is detached from the Si substrate and transferred to a PET substrate for gas sensing and mechanical testing. Upon bending, the sensor even shows an improved response possibly due to a higher surface reactivity from bending for more gas adsorption. Stable sensor response is also observed after 10<sup>5</sup> bending/relaxing cycles (Figure 1(f)). When a freestanding, thin graphene/metal sheet is used as a top electrode on vertically grown ZnO nanorods, the ethanol gas sensor itself can be made into a transparent form with good transmittance over the range of visible light [62].

Growing a second metal oxide on the first creates a metal oxide heterostructure with a heterojunction. Coating In<sub>2</sub>O<sub>3</sub> nanoparticles on vertically aligned ZnO NW by a simple wet-chemical method vields core-shell In<sub>2</sub>O<sub>2</sub>/ZnO heterostructure on a Ti foil [63] (Figure 1(g) and (h)). In comparison to the bare ZnO NW array, the heterostructure conversion of In<sub>2</sub>O<sub>3</sub>/ZnO to In<sub>2</sub>S<sub>3</sub>/ZnO upon exposure to H<sub>2</sub>S changes the electron depletion layer on the ZnO surface to an accumulation layer, resulting in a stronger free-carrier piezoscreening effect than that of gas adsorption of H<sub>2</sub>S. In addition, the In<sub>2</sub>O<sub>3</sub>/ZnO nanoarray nanogenerator can piezoelectric output power as an energy source, providing an important step toward the self-powered active gas sensor. The heterostructure can also form between a metal oxide and other gas sensing materials such as a conducting polymer. In addition, the other techniques can be explored for the synthesis of metal oxide. In an effort to form a polyaniline@SnO<sub>2</sub> heterojunction on a flexible PET [64], a SnO<sub>2</sub> precursor solution is electrospun and calcinated to form SnO<sub>2</sub> nanofibers first. Immersing a piece of pretreated PET film with a mixed solution of aniline monomer in HCl and ammonium persulfate (APS) as an oxidant in ice water in a mass ratio of 1:1 yields polyaniline via an in situ oxidative polymerization method. Simple coating with different amounts of SnO<sub>2</sub> followed by another immersion for polymerization completes the process for PANI-SnO<sub>2</sub> hybrid composites. The hybrid NH<sub>3</sub> sensors show significantly enhanced sensitivity over their constituent components at room temperature (i.e., 5 times and 29 times higher than PANI and SnO<sub>2</sub>). This enhancement is partially due to the improved interaction between NH<sub>3</sub> and N<sup>+</sup>-H sites of PANI, but is mainly attributed to the formation of the p-n heterojunction between the organic p-type PANI and the inorganic n-type SnO<sub>2</sub>.

In addition to ZnO that is mostly reported owing to its good gas sensing performance and biocompatibility [65], several other metal oxides are also studied, including CuO [66,67], SnO<sub>2</sub> [45,68], MoO<sub>3</sub> [69], WO<sub>3</sub> [61,70], TiO<sub>2</sub> [71,72], and etc. A thin layer of Cu sputtered on Kapton substrates with a possibility to pattern through shadow masks can be thermally oxidized to form p-type CuO NWs to detect ethanol and methane of 50 ppm at a working temperature of 400°C (or appreciable response towards CO, H<sub>2</sub>, acetone, H<sub>2</sub>S, and O<sub>3</sub>) [66]. In-plane tensile stresses may also be used during the oxidation process of Cu to effectively promote the CuO NW formation with significantly increased density [67]. Following the synthesis of flower-like WO<sub>3</sub> nanostructure material, an *in situ* chemical oxidation polymeriza-



**Figure 1** (Color online) (a) Optical image and schematic illustration of a fabricated flexible ethanol sensor with the interdigitated electrodes deposited on a glossy paper; (b) schematic illustration of the ethanol sensing enhancement due to the electrostatic interaction between the poly-diallyldimethylammonium chloride (PDDAC) and indium-tin-oxide (ITO) nanopowders. (a)–(b): adapted with permission from ref. [44]. Copyright 2010 by Elsevier; (c) photographs of the acetylene gas sensor and its testing apparatus; (d) responses under different bending angles toward 3 ppm to 30 ppm acetylene at 200°C. (c)–(d): reproduced with permission from ref. [54]. Copyright 2016 by Elsevier; (e) optical image and (f) stability with different bending/relaxing cycles of the hydrogen gas sensor. No significant loss of performance before  $10^6$  cycles. Reproduced with permission from ref. [26]. Copyright 2013 by Elsevier; (g) the schematic illustration that shows the fabrication process of the self-powered gas sensor based on  $\ln_2O_3/ZnO$  core-shell heterostructure (inset shows an optical image of the bent device); (h) Selectivity and sensitivity toward different gases (700 ppm for single gases and 350 ppm+350 ppm for mixed gases). (g)–(h): reproduced with permission from ref. [62]. Copyright 2014 by American Chemical Society.

tion method is used to prepare Polyaniline (PANI)@flowerlike WO<sub>3</sub> nanocomposites on a flexible PET substrate with interdigitated electrodes [73]. Owing to the formation of the p-n heterojunction at the interface between PANI and flowerlike WO<sub>3</sub>, the NH<sub>3</sub> sensor exhibits low detection limit of 500 ppb, rapid response and recovery rates, and outstanding selectivity at room temperature, as well as good moisture resistance. In order to form highly aligned  $TiO_2$  and Nb-doped  $TiO_2$  nanotubes on flexible substrates for a CO gas sensor, electrochemical anodization of Ti and Ti–Nb films sputtered on a Kapton film is used as a low cost and convenient method [74,75]. Other than a chemoresistive gas sensor, the  $TiO_2$  and Nb–TiO<sub>2</sub> nanotubular structures formed on the flexible substrate can also be used for flexible dye-sensitized solar

cells and other applications.

#### 3.2 Post-transfer of metal oxide onto flexible substrates

As the typical processing temperatures of metal oxides are relatively high, the direct growth on a flexible substrate is limited to a certain class of metal oxides. In order to explore a wide range of metal oxides, a transfer process has been exploited to transfer the metal oxide that is synthesized on rigid substrates to the flexible substrate. Various approaches have been developed for the transfer of inorganic semiconducting materials to mechanically flexible substrates [76–79]. A typical transfer process involves the use of a sacrificial layer in the initial growth step. For example, the vertical carbon nanotubes (CNTs)/reduced graphene hybrid film that is first grown on a SiO<sub>2</sub> layer supported on a Si wafer can be detached from the Si substrate by etching the SiO<sub>2</sub> sacrificial layer in a dilute HF solution, followed by a transfer to an Au/polyimide flexible substrate [80]. In comparison to the reduce graphene, the hybrid film exhibits remarkably enhanced sensitivity to NO<sub>2</sub> gas with weak n-p transitions. The slow recovery is also addressed by embedding a Ni/Cu microheater in the polyimide substrate to promote desorption of the  $NO_2$  gas molecules. Due to the use of the flexible polyimide substrate and the reduced graphene thin film, the gas sensor can bend over a radius of 15 mm to detect the presence of 5 ppm  $NO_2$ . Careful choice of the growth substrate may also eliminate the use of a sacrificial layer. Due to the weak adhesion between an ultrathin (1.9 µm) poly(methyl methacrylate) (PMMA) layer and a glass slide, palladium (Pd)-decorated indium-gallium-zinc oxide (IGZO) H<sub>2</sub> sensor fabricated on the PMMA substrate can be readily peeled off the underlying unmodified glass slide [81]. The PMMA substrate with the H<sub>2</sub> gas sensor can then attach to nonplanar substrates such as poly (vinyl chloride) (PVC) gas tubes (Figure 2(a)) or wrinkled hand gloves (Figure 2(b)), demonstrating an excellent sensitivity at room temperature without performance degradation even at a bending radius down to <1 mm (Figure 2(c) and (d)).

Drop-casting is a simple yet effective means to transfer dispersion of sensing materials that are synthesized from a process with a relatively high temperature onto the flexible substrates. For instance, an ethanol sensor results from dropcasting of commercial ZnO nanoparticles in deionized (DI) water on a PET-ITO layer in a parallel electrodes geometry, followed by evaporation of DI water at 70°C for 2 h [82] (Figure 2(e) and (f)). As the UV irradiation generates highdensity free carriers and highly reactive photo-induced oxygen ions, the sensor shows a relatively high response to ethanol even at room temperature. The use of the PET-ITO substrate also shows a transmittance of more than 62% over the visible spectrum range and allows sensing upon bending over a radius of 3.5 mm. In a similar approach, depositing a mixture of SnO<sub>2</sub> nanopowder and PDDAC on a PI substrate with interdigitated electrodes results in an ethanol sensor that can detect 10 ppm at room temperature [45]. In addition to the use of commercial nanoparticles, inks with heterostructures can be prepared for the use of drop-casting. After infiltration and reduction of Pd nanoparticles into the cavity of the Co-based metal-organic framework (MOF, e.g., ZIF-67), a calcination process at 400°C for 1 h results in the formation of PdO-Co<sub>3</sub>O<sub>4</sub> hollow nanocubes (PdO-Co<sub>3</sub>O<sub>4</sub> HNCs) [83]. Mixing PdO-Co<sub>3</sub>O<sub>4</sub> HNCs with SWCNTs increases the conductivity of the sensing material. Drop-casting the SWCNTs-loaded PdO-Co<sub>3</sub>O<sub>4</sub> HNCs inks on a hightemperature-resistant colorless PI with an interdigitated electrode on top and a flexible heater that embeds Au-covered Ni mesh on back generates a flexible NO<sub>2</sub> gas sensor (Figure 2(g)). At elevated operation temperature, the sensor shows improved reaction kinetics, enhanced sensitivity, and a low detection limit for the  $NO_2$  sensing (Figure 2(h)). Moreover, the integrated sensor with a flexible heater exhibits a high mechanical stability with negligible baseline resistance drift over 4000 cyclic bending cycles. Similar to drop casting, gel casting can also be used to disperse the sensing material onto the flexible substrate. With the assistance of a-terpineol, hydrothermally synthesized WO<sub>3</sub> nanoparticles can mix with commercial MWCNTs to form a viscous gel of MWCNTs-WO<sub>3</sub> hybrid [70]. Gel casting the MWCNTs-WO<sub>3</sub> hybrid between two Au electrodes separated by 1 cm on a PET substrate yields a flexible sensor with excellent selectivity to NO2 at room temperature and mechanical robustness over 10<sup>8</sup> bending/relaxing cycles.

The technique to transfer printing gas sensing materials from the growth substrate to the target flexible substrate can also be applied to the other materials. Through the use of a PDMS stamp, ultrahigh-density, highly aligned arrays of silicon NWs are picked up from the silicon-on-insulator substrate and then printed onto a plastic substrate to yield field-effect transistors with parts-per-billion sensitivity to NO<sub>2</sub> [84]. By constructing an array of sensors, the system has demonstrated the capability to distinguish low concentration (1000 ppm) of acetone from hexane solvent vapors. In the specific case of inorganic NWs, aligned assembly techniques also include flow-assisted [85–87], Langmuir–Blodgett [88– 90], bubble-blown [91], electric-field-directed assembly [92,93], contact printing [94–96], and nanoscale combing [97]. A change of the stamp from PDMS to a thermal release tape that modulates the adhesion can also be used for transfer printing of aligned nanotubes onto flexible substrates [98]. These techniques have provided a versatile platform for integrating gas sensors on flexible substrates [34,55,99,100].

#### 3.3 Integration of metal oxide with textile

Besides flat thin films discussed in the above sections, fab-



**Figure 2** (Color online) (a) Optical image and schematic that show the structure and attachment of the Pd-decorated IGZO-based H<sub>2</sub> sensor to a PVC gas tube; (b) the ultra-thin geometry of the PMMA allows attachment of the gas sensor on a wrinkled hand glove, which includes the steps of coating photocurable polymer and UV exposure; (c) sensitivities with different bending diameters; (d) Stability test at a bending radius of 1 mm. (a)–(d): reproduced with permission from ref. [81]. Copyright 2018 by John Wiley and Sons; (e) the measured transmittance over the visible wavelength and corresponding optical images of PET, PET-ITO substrates, and PET-ITO-ZnO gas sensor, respectively; (f) optical image and schematic illustration of the flexible and transparent ethanol gas sensor. The curvature angles of  $61.65^{\circ}$  and  $90^{\circ}$  correspond to the bending radii of 5.12 mm and 3.5 mm, respectively. (e)–(f): reproduced with permission from ref. [82], which is used under CC BY 4.0; (g) schematic illustration of the SWCNTs-PdO-Co<sub>3</sub>O<sub>4</sub> hollow nanocubes on a colorless polyimide layer embedding a Ni/Au core-shell mesh electrode at the back surface; (h) sensitivity upon exposure toward NO<sub>2</sub> under an applied voltage of 0 V and 2.1 V, corresponding to  $22^{\circ}$ C and 100 °C, respectively. (g)–(h): reproduced with permission from ref. [83]. Copyright 2017 by American Chemical Society.

rics and smart textiles have started to gain momentum in the development of flexible gas sensor due to their intrinsically flexible properties, ease of integration with clothing, and possible demonstration for fashion [103]. The metal oxide gas sensor could be integrated on the fabric or a single fiber. The process for the former is similar to the use of the other flat thin films. The low-temperature hydrothermal method also allows the direct growth of ZnO nanorods on a typical fabric lab-coat and the room temperature response of the sensor to H<sub>2</sub> gas demonstrates an electrical resistance change of 27% after 10 min exposure of 1000 ppm  $H_2$  (or 83% with a sputtered ~5 nm thick catalytic Pt layer) [101] (Figure 3(a) and (b)). This ready-to-wear ZnO nanorods device is robust against mechanical deformation and washing cycles, providing a cost-effective manufacturing route for the wearable gas sensor. In another example, the surface of the cotton fabric is modified with ZnO nanostructures by sputtering of the seed layer and sol-gel growth of ZnO [103]. The functionalized cotton fabric shows room temperature gas-sensing of volatile organic compounds (e.g., acetaldehyde, ammonia, and ethanol), as well as an excellent UV filtering capability with enhanced ultraviolet protection factor against UV radiation. The metal oxide sensing material can also be prepared in a fiber mat geometry for a flexible property. Without the need to worry about substrate damage, SnO<sub>2</sub> polycrystalline nanofibers doped with MWCNTs synthesized through electrospinning can be followed by a calcination process at 500°C in the air [104]. Due to the increased binding energy between multiwall carbon nanotubes (MWCNTs) and carbon monoxide (CO) and thereby large electron charge transfer from MWCNTs to CO, the resulting MWCNTs-SnO<sub>2</sub> sensors are capable of detecting CO of

50 ppm at the room temperature. The demonstrated MWCNT doped  $SnO_2$  nanofiber mats have the potential for wearable room temperature gas sensors for their flexible geometry and low-cost manufacturing method.

In comparison to the fabric substrate, the integration of metal oxide on a single fiber offers a great versatility, but it also involves additional challenges bending, twisting, or stretching in the weaving process. Electrodeposition of zinc on a carbon microfiber textile followed by a thermal oxidation yields a hierarchic nano-on-micro ZnO NW structure with a high surface-to-volume ratio [105]. The idea of integrating metal oxide on a single fiber/yarn can also be extended to create the sensor in a fiber geometry. Aligned MWCNTs sheets drop cast with prefabricated metal-organic frameworks (MOFs) (e.g., ZIF-67-Co and MIL-88-Fe) can be twisted with a motor into a desirably aligned MOF/ MWCNT hybrid fiber [31]. Following heat treatment such as

annealing at 300°C for 2 h in air, ZIF-67-Co in the hybrid fiber transforms into Co-based oxide (i.e., Co<sub>3</sub>O<sub>4</sub>). Similarly, annealing MIL-88-Fe/MWCNT in air yields Fe<sub>2</sub>O<sub>3</sub>/MWCNT hybrid fiber (Figure 3(c) and (d)). The Co<sub>3</sub>O<sub>4</sub>/MWCNT hybrid fiber demonstrates remarkable sensitivity to measure  $NO_2$  down to 0.1 ppm at room temperature and (Figure 3(e)) and (f)). A better response in the Co<sub>3</sub>O<sub>4</sub>/MWCNT hybrid fiber sensor is observed upon bending (Figure 3(g)) and it is attributed to the improved electron transfer along the radial direction from the partially straightened fiber due to the tensile strain. Sewing the hybrid fiber into the commercially available textile fabric also demonstrates the easy integration and mechanical robustness of the hybrid fiber. In addition, the hybrid fiber can also serve as a supercapacitor to store energy for wearable devices. It should also be noted that the smart textile is also compatible with the electronic nose design [102] (Figure 3(h)).



**Figure 3** (Color online) (a) Optical images of bare and ZnO nanorod-coated cotton fabric; (b) SEM and optical images of ready-to-wear ZnO nanorods hydrogen gas sensor. (a)–(b): reproduced with permission from ref. [101]. Copyright 2010 by Elsevier; (c) the schematic illustration that shows the fabrication process of metal oxides/MWCNT hybrid fiber gas sensor and the inset shows the SEM image of ZIF-67-Co/MWCNT hybrid fiber; (d) optical image of a hybrid fiber woven into a lab coat and schematic illustration of the hybrid fiber sensor used for NO<sub>2</sub> detection; normalized responses of  $Co_3O_4/MWCNT$  hybrid fiber with different  $Co_3O_4$  loadings to NO<sub>2</sub> in the concentration ranges of (e) [0.1, 20] ppm and (f) [20, 1000] ppm (AC30 indicates 30 wt% of ZIF-67-Co loading, followed by an annealing process); (g) responses of  $Co_3O_4/MWCNT$  hybrid fiber (AC90) with and without bending under different concentrations of NO<sub>2</sub>. (c)–(g): reproduced with permission from ref. [31]. Copyright 2018 by American Chemical Society; (h) schematic diagram and photograph of a novel electronic nose design in a woven textile band and one enlarged gas sensor (upper left). Reproduced with permission from ref. [102]. Copyright 2012 by Elsevier.

# 4 Metal oxide gas sensors integration with stretchable substrates

It is highly desirable for the wearable metal oxide gas sensor to have a conformal contact with the skin of the human body. The capability to bend allows a flat geometry to form conformal contact with a developable surface (Gaussian curvature of zero such as cylinder or cone). For the nondevelopable surface, the stretchable property would be essential to form a conformal contact. To enable the stretchable property to the intrinsically rigid metal oxide, structural design strategies have been explored [106-109]. A wavy geometry of the metal oxide created by the use of pre-strain strategy can allow it to be stretched flat and the small strain in the wavy metal oxide is directly related to the applied tensile strain [110-112]. Instead of creating wavy metal oxide, a flat metal oxide can also be connected by wavy metallic interconnections by adopting the previous established "island-bridge" layout [113–115]. Selectively bonding the metal oxide to a stretchable substrate with a surface relief structure isolates the strain to the region of wavy interconnection [116-119]. Other structural design strategies include interconnected cracks [120,121], 2D/3D helices [122,123], origami/kirigami [124-127], and interlocks [128,129].

As a simple yet effective strategy, pre-strain has been widely used to create stretchable structures. In a recent study [130], graphene electrodes and colloidal PbS quantum dots sensing layer are first transferred to a pre-strained elastomeric substrate via dry transfer and spin-coating, respectively (Figure 4(a)). The release of the pre-strain results in an stretchable gas sensor in a wavy geometry. Owing to the antihumidity interference ability of the colloidal PbS quantum dots, the sensor demonstrates an improved humidity resistance, as well as a high sensitivity in response to NO<sub>2</sub> at room temperature even under 1000 stretch/relax cycles. In addition, the wavy gas sensor exhibits a much smaller variation in the NO<sub>2</sub> detection from a relative humidity of 0% to 86.7% when compared with the flat gas sensor. Due to the great solution processability of colloidal quantum dots, those based on a metal oxide such as SnO<sub>2</sub> can be expected to create stretchable gas sensors with enhanced catalytic activity and fast response to  $H_2S$  [131] (Figure 4(b)).

Apart from the pre-strain strategy, the island-bridge design can also be used to provide a stretchable system, where the rigid sensors are placed on the island region interconnected by deformable bridges. In a multisensor system, an MWNT/ SnO<sub>2</sub> NW NO<sub>2</sub> gas and UV sensor is integrated with a fragmentized graphene foam strain sensor, an array of nine micro-supercapacitors, and a rechargeable wireless radio frequency power receiver all on a stretchable Ecoflex substrate with an embedded liquid metal interconnection [132] (Figure 4(c)). As validated by the experimental demonstration, the prediction from finite element analysis shows that the strain in the relatively rigid island region is much smaller than that of the deformable bridge region upon an applied tensile strain. The stretchable multisensor system attached to the skin could detect various signals with direct relevance to healthcare, including neck pulse, saliva swallowing, voice, body motion, and sensing of NO<sub>2</sub> gas and UV light signal even under a uniaxial strain up to 50% (Figure 4(d)).

Interconnected cracks can be created from multiple approaches and the resulting metal oxide sensing element itself is stretchable. The integration of metal oxide on an elastomeric polymer can be achieved by a platinum transfer technique [135]. Relying on a weak adhesion between the platinum and Si wafer, the Pt with a metal oxide such as ZnO deposited on top can be removed from the Si wafer by casting a PDMS layer on the ZnO layer. Following a reactive ion etching to remove Pt, ZnO in plate-like micro-cracking structures with overlapping edges are left on the PDMS layer. Due to the strong adhesion between ZnO and PDMS, the structure of tectonic plates in ZnO provides an electrical conduction path upon stretching. The transparent and stretchable ZnO film has been shown to result in enhanced gas sensing performance when compared with their rigid counterparts under room temperature conditions [133] (Figure 4(e)–(g)). In addition to a gas sensor for detection of  $H_2$  and NO<sub>2</sub> gases, the metal oxide on PDMS can also be exploited to create ultraviolet light sensors and a tunable diffraction grating for displacement sensing with nanometer accuracy. This approach also allows the creation of electronic devices with patterns in nanometer resolution over a large area.

In addition to the interconnected cracks, the metal oxide itself can be made stretchable by a simple and versatile flame transport synthesis method [136]. As opposed to the purely diffusion-based approaches such as ALD, the flame-transport-synthesis that is based on convection-driven control of the reaction environment allows a faster reaction rate and thereby a significantly enhanced throughput. The resulting interconnected 3D SnO<sub>2</sub> NW network is used for ethanol sensing and UV detection at room and low temperatures (down to 25 K) [137]. Enabling flexible and stretchable properties to brittle materials, the 3D network from this approach can go beyond the sensing to many other applications, including battery electrode, biomedical, and 3D template for growth of new materials [136,138,139].

It should also be noted that several other strategies can also be exploited for stretchable gas sensors though it is not included in the current demonstration. For example, polyurethane (PU) fibers soak-coated with ZnO seed layer, followed by a hydrothermal reaction, yields a multifunctional ZnO NWs@PU [134] (Figure 4(h) and (j)). Owing to the intrinsically stretchable nature of PU fibers, applied tensile strain creates cracks in the ZnO NW array to



**Figure 4** (Color online) (a) A wavy, stretchable quantum dot gas sensor created from the pre-strain strategy (left) and optical images of the gas sensor attached to the finger joint in "paper" and "rock" states (right). Reproduced with permission from ref. [130]. Copyright 2018 by American Chemical Society; (b) responses of the SnO<sub>2</sub> quantum dot gas sensor toward different concentration of H<sub>2</sub>S. Reproduced with permission from ref. [131]. Copyright 2018 by Elsevier; (c) photograph and strain analysis of the multisensory system; (d) responses of the NO<sub>2</sub> gas sensor under different strains. (c)–(d): reproduced with permission from ref. [132]. Copyright 2015 by John Wiley and Sons; (e) an optical image that shows the transparent ZnO-based gas sensor is capable of flexing and twisting; (f) SEM image that shows the surface morphology of ZnO in false color; (g) responses toward H<sub>2</sub> and NO<sub>2</sub> with and without stretching at room temperature. (e)–(g): reproduced with permission from ref. [133]. Copyright 2015 by John Wiley and Sons; (i) schematic (up) and photographic (down) illustrations of the sensor under an applied tensile strain; (j) SEM images of the sensor under different applied tensile strains. (h)–(j): Reproduced with permission from ref. [134]. Copyright 2016 by John Wiley and Sons.

result in a percolating pathway. The stretchable network is capable of detecting strain, temperature, and UV. Its potential use for gas sensing can also be envisioned.

### 5 Conclusion and future perspectives

Aiming to provide only a glance at the recent advances in the

burgeoning area of flexible and stretchable metal oxide gas sensors for healthcare, the review here is by no means an exhaustive list. As briefly discussed in this mini-review, great stride has been made. In order to achieve the desired flexible property, the metal oxide sensing material is either directly grown or post transferred to a soft, thin, and flexible substrate. As for the goal to achieve a conformal contact between the bio-integrated metal oxide gas sensor and the soft tissue, several stretchable structures have been explored. Despite the recent progress, a few challenges still need to be addressed for the practical application.

In order to avoid the adverse thermal effect on the human body and to lower the power budget for long-term operation, it is highly desirable for the wearable metal oxide gas sensor to operate at room temperature. Thus, novel materials and/or heterostructures that provide room-temperature sensing capabilities represent a strong focus for future research. Moreover, the room-temperature processing of the gas sensor is also of interest. For instance, combining a PbS colloidal quantum dots (CQDs) with flexible substrates yields a highly sensitive and fully recoverable NO<sub>2</sub> gas sensor at room temperature [140]. Owing to the large surface area and room-temperature solution processing capability onto various substrates. CODs that provide tunable binding energy to target gases are promising in wearable gas sensing toward healthcare applications. Anchoring SnO<sub>2</sub> quantum wires on reduced graphene oxide nanosheets by a one-step colloidal synthesis without further sintering results in a sensitive and selective room-temperature H<sub>2</sub>S gas sensor [141]. The favorable electron transfer across the interface of the composites could also enhance the room-temperature sensing performance. On a separate route to eliminate the need for an external heater, self-heating is exploited to harness the dissipated power at individual NWs by Joule effect upon application of a biased current [142,143]. With reduced thermal inertia, a single SnO2 NW only requires few mW (from energy harvesting technologies such as thermoelectric microgenerators) to bias, heat, and measure CO and NO<sub>2</sub>, with nearly identical responses to those obtained with integrated microheaters. Similar to the role as an activation energy from the heat, high-density power light energy in the range of the energy gap of the semiconductor has also been used to enable room-temperature sensing. Examples include the detection of NO<sub>2</sub> with SnO<sub>2</sub> [144–146], CO and NO<sub>2</sub> with In<sub>2</sub>O<sub>3</sub> and SnO<sub>2</sub> (with or without catalyst/doping) [147,148], or acetone/acetaldehyde/volatiles with nanoparticulate ZnO [149].

A strain in the metal oxide, even small and limited due to the brittle nature of the semiconducting metal oxide, may still change the signal outputs in the sensing material upon mechanical strains [2,106,107]. Depending on the architecture design of the device and especially the position of the sensing material, the sensing material layer may experience in-plane tensile, compressive, or zero strain upon a bending deformation [67]. Such bending-induced strain would alter the microstructure and transport characteristics of carriers in the semiconducting metal oxide sensing layers, therefore affecting the sensing performance. Though the sensitivity of the sensing material is observed to remain unaffected or even promoted in certain cases [54,81,82,133], the lack of qualitative analysis still leaves the strain effect on the gas sensing performance yet to be explored. The change in the signal outputs may also be a combined effect in the presences of target gas and mechanical deformation. Thus, such coupling should be closely examined and a strategy to decouple the different effects will be of an important interest. In an attempt to address this issue, channel transconductance and its equivalent voltage of a field-effect transistor are measured as temperature and pressure are applied, from which a characteristic matrix can be solved [150]. By using the inverse of the characteristic matrix, both temperature and pressure can be determined for a given channel transconductance and its equivalent voltage, resulting in a bimodal sensing capability within a single device. This approach is effective because channel transconductance and its equivalent voltage are found to have a linear dependence on the temperature and pressure; thus, it is expected to serve as a general approach for the linear problem. Although it is not directly quantified, the improved gas sensing performance from the mechanical deformation seems to show a nonlinear relationship. A general strategy to overcome this grand challenge is still yet to come.

As volatile organic biomarkers often appear at ppb-level for practical applications toward healthcare, the detection of this ultra-low concentration is of significant interest. A wellestablished case is to diagnose the lung cancer with the biomarkers in the exhaled breath [151–153]. Toluene, one of the most common volatile organic biomarkers for lung cancer, is a suitable choice because of their significantly elevated levels in lung cancer patients [151]. When WO<sub>3</sub> nanofibers are decorated by Pd catalysts on both inner and outer layers, they show a low limit of toluene detection of 20 ppb in a simulated human exhaled breath at operating temperature of 350°C, suitable for diagnoses of lung cancer [152]. A low limit of detection as small as 2 ppb, as well as high response and good selectivity toward toluene, is also demonstrated by using the ordered mesoporous NiFe<sub>2</sub>O<sub>4</sub> with the ultrathin framework and a large specific surface area [153]. Another representative example is to diagnose diabetes via ppb-level acetone detection in the human breath. A chemoresistive detector based on Si-doped WO3 demonstrates an ultra-low acetone detection of 20 ppb, offering a potential for the noninvasive diagnosis of diabetes [47]. Despite great strides have been made in the ppb-level gas detection, challenges to detect this ultralow-level concentration of target gases in a complex environment at a fast, reliable, and cost-effective manner still need to be overcome to further promote the development of flexible and stretchable gas sensors for healthcare.

In addition, combining the wearable gas sensor with the other sensing modalities provides a comprehensive system for the healthcare, as demonstrated in several initial attempts. In addition to the example discussed in Section 4.3, a multisensor platform on a flexible plastic foil has been demonstrated to simultaneously measure oxidizing and reducing gases, volatile organic compounds (VOCs), as well as humidity and temperature [154]. In such system integration, many practical issues will also need to be considered, including the high density of the active components or small footprint of the entire system, cost-effective manufacturing methods for large-scale production, high reliability and durability for long-term use, and others. Nonetheless, these grand challenges also represent a fraction of possibilities and opportunities for the research community. Due to the multidisciplinary nature of the burgeoning area of flexible and stretchable metal oxide gas sensors for healthcare, researchers from diverse fields are encouraged to work together to provide collective wisdom for additional discoveries in the future.

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