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# **Electrochemical performance optimization of the polyaniline electrodeposited on ITO substrate**

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#### **Abstract**

We have elaborated polyaniline flms on ITO substrate (indium tin oxide), by electrochemical process in diferent electrolytes (HCl,  $H_2SO_4$ , HNO<sub>3</sub>, and  $H_3BO_3$ ), which allowed us to study the effect of the counter ion on electrochemical energy storage performances of polyaniline as electrode material in supercapacitors. The study of the diferent obtained flms performances was carried out by cyclic voltammetry and galvanostatic charge–discharge method and is interpreted by the SEM technique. We found that there is a clear dependence on the specifc capacitance of the counter ion. Justifed by its porous structure, the PANI/ITO electrode doped with  $SO_4^{2-}$  has the highest specific capacitance, 57.3 mF/cm<sup>2</sup> at a current density of 0.2 mA/cm<sup>2</sup> and 64.8 mF/cm<sup>2</sup> at 5 mV/s. The deep analysis by Dunn's method allowed us to conclude that the faradic process dominates the energy storage in the case of PANI/ITO electrode elaborated in boric acid (99%). On the contrary, the capacitive character is the most contributory in the case of electrodes elaborated in  $H_2SO_4$ , HCl, and HNO<sub>3</sub>. The study at diferent potentials (0.80, 0.85, 0.90, 0.95, and 1.0 V/SCE) from 0.2 M monomer aniline showed that the deposition at 0.95 V/SCE leads to higher specific capacitance (24.3 mF/cm<sup>2</sup> at scan rate 5 mV/s and 23.6 mF/cm<sup>2</sup> at 0.2 mA/cm<sup>2</sup>) with a coulombic efficiency of 94%. By varying the concentration of the monomer while keeping a potential fixed at 0.95 V/SCE, we also found that the specifc capacitance increases with monomeric concentration.

**Keywords** Polyaniline · Electrodeposition · Cyclic voltammetry · Specifc capacitance · Energy storage

## **Introduction**

The global crisis caused by fossil fuel–based energy and its climate impact has played an important role in shaping research and development activities in the feld of renewable energy conversion and storage. This challenge requires the improvement of conventional energy storage systems such as lead-acid batteries and the discovery of other technologies. Competing with carbon used in Li-ion batteries, polymers are excellent materials with high performance, better cyclic stability and durability and a good contribution to green energy. Recently, a variety of polymers have

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 $\boxtimes$  Aziz Aynaou aynaou18@gmail.com been developed for use in batteries such as polyvinylidene fuoride used as collector (Shi et al. [2020](#page-12-0)), polyethylene as separator (Gu et al. [2021\)](#page-11-0), Nafion as membrane (Okonkwo et al. [2021](#page-12-1)), poly(3-hexylthiophene) in photovoltaic cells (Dang et al. [2011](#page-11-1)), and polyaniline (P. Liu et al. [2019](#page-12-2); Ryu et al. [2002;](#page-12-3) Zhou et al. [2005](#page-13-0)) and polypyrrole (Muthulakshmi et al. [2006](#page-12-4); Sharma et al. [2008](#page-12-5)) as dielectric capacitors (Wang et al. [2010\)](#page-12-6). Supercapacitors occupy an intermediate place between accumulators and capacitors with an energy density that can reach hundreds of kilojoules per kilogram and a power density ranging from 10 W/kg to 100 kW/kg (Lokhande et al. [2020\)](#page-12-7), which allows them to store a large amount of energy and to release it quickly and therefore to fill the energy deficit of capacitors and the power deficit of accumulators. A supercapacitor is also characterized by its specifc capacitance, which depends essentially on the materials making up its electrodes, of which we distinguish carbon materials (graphene oxide 425 F/g in a symmetric cell) (Zhao et al. [2017](#page-13-1)), metal oxides (RuO<sub>2</sub> 1400 ~ 2000 F/g) (Chen et al. [2013\)](#page-11-2),  $MnO_2$  1370 F/g (Jabeen et al. [2016](#page-11-3)),

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and conducting polymers (Snook et al. [2011\)](#page-12-8). Energy storage in these so-called pseudocapacitive materials is achieved by two processes, in the electrochemical double layer and by rapid and reversible redox reactions at the electrode/electrolyte interface. In polymers, or more precisely in the polyaniline which is the subject of this work, the charging/discharging mechanism is mainly due to the existence of three oxidation states, leucoemeraldine (the most reduced state), pernigraniline (the most oxidized state), and emeraldine (intermediate state) (Angelopoulos et al. [1988;](#page-11-4) Nakajima et al. [1989\)](#page-12-9). This doping/undoping process is also produced by the insertion and de-insertion of counter ions into the polymer chain (Snook et al. [2011](#page-12-8)). In addition to its use in supercapacitors particularly and energy storage generally, polyaniline fnds its applications in many other felds such as nanoelectronic devices (feld efect transistors) (Salikhov et al. [2015](#page-12-10)), chemical (Fratoddi et al. [2015\)](#page-11-5) or biological sensors (Dhand et al. [2011](#page-11-6)), catalysis or electrocatalysis (Eskandari et al. [2020](#page-11-7)), microwave absorption and electromagnetic shielding (Saini et al. [2009](#page-12-11)), electrorheological fuids (Yin et al. [2008\)](#page-12-12), and biomedicine (Zare et al. [2019](#page-12-13)).

It is well known that polyaniline is a material whose physical and chemical characteristics (morphology, conductivity, degree of crystallinity) depend strongly on the method of preparation, experimental parameters, and synthesis procedure. In the present work, we have studied the efect of counter-ion, electrodeposition potential, and monomer concentration on the electrochemical performance of polyaniline. For this purpose, we have elaborated polyaniline flms by electrodeposition using cyclic voltammetry in four acids  $H_2SO_4$ , HCl, HNO<sub>3</sub>, and  $H_3BO_3$  and, then, by the potentiostatic method at diferent potentials and various monomer concentrations. The specific capacitance measurements are performed by cyclic voltammetry and galvanostatic charge–discharge.

# **Materials and experimental**

## **Materials and reagents**

All chemicals used were of analytical grade. Aniline  $C_6H_5NH_2$  99.5% (ACS reagent), sulfuric acid  $H_2SO_4$  98% (ACS reagent), nitric acid  $HNO<sub>3</sub> 65%$  (ACS reagent), hydrochloric acid HCl 37% (ACS reagent), boric acid  $H_3BO_3$ 99.5% (ACS reagent), sodium sulfate  $Na<sub>2</sub>SO<sub>4</sub>$  (Merck), and indium tin oxide (ITO)–coated glass (20  $\Omega$ /cm) were purchased from SOLEMS. The ITO was used as the working electrode, the emerged extremity of the ITO in the electrolyte has an area of 1 cm<sup>2</sup> (1 cm  $\times$  1 cm), and the other extremity was used to establish electrical contact with an alligator clip.

#### **Preparation of electrodes**

Electrochemical measurements were performed using a threeelectrode cell connected to a Versa STAT 3 potentiostat–galvanostat combined with Versa Studio software. The platinum (Pt) electrode was used as the counter electrode, the saturated calomel electrode (SCE) as the reference electrode (all potentials are given relative to this reference), and indium tin oxide (ITO) as the working electrode. Prior to electropolymerization, the ITO substrate was cleaned in an ultrasonic bath using acetone for 10 min ethanol for 10 min and fnally distilled water for 5 min. The surface area of the working electrode was set at  $1 \text{ cm}^2$ .

**Counter‑ion efect** The electropolymerization of polyaniline (PANI) was carried out by cyclic voltammetry (CV) using 0.2 M aniline dissolved in diferent acidic electrolytes  $(H_2SO_4, HNO_3, HCl, and H_3BO_3)$  of the same concentrations (1 M) using cyclic voltammetry for two cycles.

**Efect of electrodeposition potential** The elaboration of PANI/ ITO electrodes was carried out by galvanostatic method (at 0.80, 0.85, 0.90, 0.95, and 1.0 V/SCE) using 0.2 M aniline dissolved in acidic electrolyte  $(H_2SO_4 1 M, Na_2SO_4 0.5 M)$ .

**Effect of monomer concentration** To study the effect of monomer concentration on the electrochemical performance of PANI/ ITO electrodes, we developed them from diferent concentrations of aniline (0.05, 0.1, 0.2, and 0.4 M) dissolved in acidic electrolyte  $(H_2SO_4 \, 1 \, M, Na_2SO_4 \, 0.5 \, M)$  by the potentiostatic method at the same electrodeposition potential 0.95 V/SCE.

#### **Electrochemical measurement and characterization**

The electrochemical performances of PANI electrodeposited (noted PANI/ITO in the following) were characterized using the same three electrodes system. However, the electrodeposited flm changed the working electrode and a solution of  $H_2SO_4$  1 M/Na<sub>2</sub>SO<sub>4</sub> 0.5 M was used as electrolyte. Cyclic voltammetry measurements were performed at diferent scan rate 5, 10, 20, 30, 40, and 50 mV/s. The morphology of the PANI samples was studied by scanning electron microscopy (SEM) using a Philips XL 30FEG.

## **Results and discussions**

#### **Cyclic voltammetry**

In order to study the efect of the counter ion on the polyaniline electrodeposited on the ITO, we registered in Fig. [1](#page-2-0)a the frst cyclic voltammograms from−0.2 to 1 V/SCE with scan rate 15 mV/s in each acid (HNO<sub>3</sub>, H<sub>2</sub>SO<sub>4</sub>, HCl, and H<sub>3</sub>BO<sub>3</sub>). All four cycles exhibit a single oxidation peak corresponding to the oxidation of aniline to PANI (pernigraniline) during the scan to the anodic potentials. During the return to the cathodic potentials, we found two peaks corresponding successively to the reduction of pernigraniline to emeraldine and the reduction of emeraldine to leucoemeraldine (Córdova et al. [1994;](#page-11-8) Sayah et al. [2021;](#page-12-14) Aynaou et al. [2022](#page-11-9)). A nucleation loop appears in the case of  $HNO<sub>3</sub>$  and  $H<sub>3</sub>BO<sub>3</sub>$  and not in the case of  $H_2SO_4$  and HCl. On the other hand, the position and the current density of oxidation peak of aniline to PANI in its pernigraniline form vary from one acid to another (the peak starts at 0.7, 0.8, 0.85, and 0.9 V/SCE for  $H_3BO_3$ ,  $H_2SO_4$ , HCl, and HNO<sub>3</sub> respectively).

On the second cycles, shown in Fig. [1](#page-2-0)b, to the aniline oxidation peak in PANI are added two anodic waves attributed to the oxidation of leucoemeraldine to emeraldine and emeraldine to pernigraniline. The current density of the anodic peak (aniline to PANI) on second cycle increases strongly compared to the first cycle in the case of  $H_2SO_4$ ,  $HNO_3$ , and HCl, while it decreases in the case of  $H_3BO_3$  because of the low electrical conductivity of polyaniline elaborated in the latter (Yakuphanoglu and Şenkai [2008\)](#page-12-15). Borate ions are likely to form bonds with polyaniline (Scheme [1\)](#page-2-1), which limits the movement of radicals along the polymer chain and thus the decrease in electrical conductivity (Suematsu et al. [2000](#page-12-16); Yakuphanoglu and Şenkai [2008\)](#page-12-15).

This suggests that the nature of the acid infuences the electrochemical behavior of the polymer. The growth rate of PANI flms therefore depends on the acid anion (in the following order:  $H_2SO_4$  >  $\rightarrow$  HCl > HNO<sub>3</sub> >  $H_3BO_3$  and the pH (oxidation of protonated aniline is easier than non-protonated) which is in agreement with previous studies (Arsov et al. [1998](#page-11-10); Lippe & Holze [1992\)](#page-12-17). It should be noted that the infuence of the counter-ion and the ionic strength of the medium on the initial stages of nucleation and growth are manifested mainly in the intensity and shape of the redox peaks, the amplitude of the nucleation loop, the redox



<span id="page-2-1"></span>**Scheme 1** Illustration of the cross-linking of polyaniline by complexation with borate ions



<span id="page-2-0"></span>**Fig. 1** Cyclic voltammograms at a scan rate of 15 mV/s of ITO substrate in 0.2 M aniline and the different acids H<sub>2</sub>SO<sub>4</sub>, HNO<sub>3</sub>, HCl, and  $H_3BO_3$ : **a** first cycles, **b** second cycles

potential of the monomers, and the evolution of current beyond the frst cycle.

In order to compare the efect of the acid on the electrochemical performances of polyaniline, the PANI flms are electrodeposited by CV for two cycles in the four acids  $(H_2SO_4, HNO_3, HCl, and H_3BO_3).$ 

#### **Counter‑ion efect**

The electrochemical performance of the obtained PANI/ ITO electrodes was studied by CV in an electrolytic bath containing  $H_2SO_4$  1 M and  $Na_2SO_4$  0.5 M. Figure [2](#page-3-0) shows the evolution of the cyclic voltammograms as a function of the scan rate for each electrode. Cyclic voltammetry shows comparable electrochemical responses for the electrodes elaborated in  $H_2SO_4$ , HCl, and HNO<sub>3</sub>; the quasi-rectangular shapes of the voltammograms with symmetrical redox peaks refect the pseudocapacitive nature of these three electrodes (Mathis et al. [2019a,](#page-12-18) [b\)](#page-12-19).

During the charge, the polyaniline undergoes oxidation, which creates positively charged sites in the polymer skeleton (Le et al. [2017](#page-11-11)). During the discharge, the polyaniline

undergoes a reduction and the charged sites return to the neutral state; it thus behaves like a battery (Gannett et al. [2021](#page-11-12)). Simultaneously with these processes of oxidation/reduction (charge/discharge), we witness the insertion/disinsertion of anions of the electrolyte in the porous matrix of the polymer, which behaves like a capacitor (Schoetz et al. [2018](#page-12-20); Le et al. [2017](#page-11-11)). This pattern is modifed in the case of the electrode prepared in  $H_3BO_3$ ; it can be seen that the redox peaks are no longer symmetrical, and hence, this electrode tends to have a battery-type behavior; more precisely, the faradic process prevails over the capacitive character (Pu et al. [2021\)](#page-12-21).

We can use cyclic voltammetry to determine the predominant process of charge storage. A representative power law relationship between peak current  $i_{\text{peak}}$  and scan rate  $v$  gives an overview of the charge storage mechanism in PANI/ITO electrodes using Eq. [1](#page-3-1) (J. Liu et al. [2018](#page-12-22)).

$$
i_{\text{peak}} = av^b \tag{1}
$$

<span id="page-3-2"></span><span id="page-3-1"></span>We introduced the logarithm to linearize this equation:

$$
\log(i_{\text{peak}}) = \log(a) + b\log(v) \tag{2}
$$



<span id="page-3-0"></span>**Fig. 2** Cyclic voltammograms at scan rates 5, 10, 20, 30, 40, and 50 mV/s for PANI/ITO electrodes elaborated in diferent acids  $H_2SO_4$ , HNO<sub>3</sub>, HCl, and  $H_3BO_3$ 

*a* and *b* are adjustable parameters which can be calculated from a plot of  $log(i)$  vs.  $log(v)$ ;  $log a$  is determined by the ordinate at the origin and b from the slope. A *b* value of 0.5 corresponds to a process totally controlled by difusion, and a value of 1.0 indicates the dominance of the capacitive process (J. Liu et al. [2018](#page-12-22)).

On Figs. [3](#page-4-0) and [4](#page-5-0), we have plotted  $log(i_{\text{pic}})$  versus  $log(v)$  of the anodic and cathodic peaks for each electrode. It is clear that these curves are straight lines with correlation coefficients close to one (1) which is in agreement with Eq. [2](#page-3-2). We can therefore easily determine the values of *b* from the slopes.

We note that *b* converges towards 1 in the case of the electrodes elaborated in  $H_2SO_4$ , HCl, and HNO<sub>3</sub>, which confrms the predominance of the capacitive character for these electrodes. While in the case of the electrode elaborated in  $H_3BO_3$ , the charge storage under diffusional control is justifed by the value of *b*, close to 0.5.

The specifc capacitance was estimated from the cyclic voltammograms using Eq. [3:](#page-4-1)

$$
C + \left(\int j dE\right) / 2\nu \Delta E \tag{3}
$$

<span id="page-4-0"></span>**Fig. 3** Functional relationship of the peak current (i/mA) vs the scan rate  $(v/mV s^{-1})$  for dif-

ferent electrodes

where *C* is the specific capacitance (mF/cm<sup>2</sup>),  $\int \hat{j}dE$  is the voltammetric charge obtained by integration of curve in (CV),  $\Delta E = E_{initial} - E_{final} = 0.7$  (V/SCE) – 0.0 (V/SCE) = 0.7 V is the potential window, and  $v$  is the scan rate (V/s).

In Fig. [5a](#page-6-0), we have plotted the variation of the specifc capacitance of each electrode as a function of the scan rate. The specific capacitance of the electrodes electrodeposited in the diferent acids decreases with the increase of the scan rate from 5 to 60 mV/s. This evolution can be explained by the decrease of the faradic process contribution when the scan rate increases. In other words, the redox reactions do not have enough time to take place. Recent in situ atomic force microscopy studies have demonstrated that the pseudocapacitive behavior of conductive polymers is favored for high charge conditions and battery behavior at low charge (Schoetz et al. [2018](#page-12-20)). On the other hand, we note that the specifc capacitance of the PANI/ ITO electrode electrodeposited in  $H_2SO_4$  is higher than those electrodeposited in the other three acids  $(65 \text{ mF/cm}^2)$ at 5 mV/s). Previous studies have found that the electrochemical performance of PANI depends on its structure and morphology, which in turn depends on the nature of

<span id="page-4-1"></span>

<span id="page-5-0"></span>**Fig. 4 a** Specifc capacitance values of PANI/ITO electrode elaborated in diferent acids, **b** separation of capacitive contribution (gray area) for PANI/ITO electrode elaborated in  $H_2SO_4$ at a scan rate of 5 mV/s, **c**–**f** percentages of pseudocapacitive contribution at diferent scan rates for PANI/ITO electrodes elaborated in  $H_2SO_4$ , HCl,  $HNO<sub>3</sub>$ , and  $H<sub>3</sub>BO<sub>3</sub>$  respectively



the doping anion, which justifes the diference in specifc capacitance depending on the acid used (Li et al. [2014](#page-12-23)).

Using Dunn's method, we can quantify the ratios of the capacitive contribution. This method consists to separate the contributions of the surface reaction and the diffusion control to the total current. At a fxed scan rate, in cyclic voltammetry measurements, the total current can be interpreted as the sum of the capacitive current related to the charge of the electrochemical double layer and/or the fast redox reactions at the interface  $(i<sub>cap</sub>)$  and the current related to the slow diffusion-controlled processes  $(i_{\text{diff}})$ (Guo et al. [2018\)](#page-11-13).

<span id="page-6-0"></span>**Fig. 5** SEM image of polyaniline flms obtained by cyclic voltammetry in diferent acid:  $\mathbf{a}$  H<sub>2</sub>SO<sub>4</sub>,  $\mathbf{b}$  HCl,  $\mathbf{c}$  HNO<sub>3</sub>, and  $dH_3BO_3$ 



 $b)$ 

 $\rm d$ 

$$
i = i_{\text{cap}} + i_{\text{diff}} \tag{4}
$$

For a strictly difusion-limited redox reaction, the current is proportional to the square root of the scan rate:

$$
i = i_{\text{diff}} = k_d v^{0.5} \tag{5}
$$

whereas, the capacitive current from the double layer and pseudocapacitance varies linearly with the scan rate according to equation:

$$
i = i_{\rm cap} = k_c v \tag{6}
$$

From which the total current can be described by the following empirical equation:

$$
i = k_c v + k_d v^{0.5}
$$
 (7)

$$
(i/\nu^{0.5}) = k_d + k_c \nu^{0.5}
$$
 (8)

where  $k_c$  and  $k_d$  are constants,  $k_c v$  indicates the capacitive contribution to the overall current, and  $k_d v^{0.5}$  indicates the difusion-controlled contribution.

The plot of  $i(E)/v^{0.5}$ , as a function of  $v^{0.5}$ , allows to determine  $k_c$  and  $k_d$  and thus the separation of the capacitive charges and the difusion-controlled charge. The contribution ratios of the two processes at diferent scan rates for each electrode were also determined and presented in Fig. [5](#page-6-0)c–d. It is noted that the pseudocapacitive contribution gradually increases with increasing scan rate. This can be explained by the fact that at high scan rates, i.e., high driving force, the difusion of anions into the polymer matrix cannot keep up with the rate of creation of positive sites (Schoetz et al. [2018\)](#page-12-20). Therefore, the redox process is governed by the difusion of counter ions within the flm. The capacitive contribution ratios at scan rates of 5, 10, 20, 30, 40, and 50 mV/s are higher for the electrodes developed in  $H_2SO_4$ , HCl, and  $HNO<sub>3</sub>$  (75.8%, 82.2%, and 89.2% respectively at

50 mV/s), suggesting that the capacitive process is more contributory in charge storage for these electrodes. For the electrode developed in  $H_3BO_3$ , the percentage of capacitive charge is less than 1% and therefore the difusion-controlled capacitance plays a decisive role in the charge storage performance in this electrode.

To confrm these fndings, we analyzed the morphology of the four flms using the SEM technique (Fig. [5](#page-6-0)). A particulate (granular), less porous and very dense structure, was revealed in the case of the electrode developed in  $H_3BO_3$ , which justifies its low specific capacitance. This structure is slightly modified in the case of the  $HNO<sub>3</sub>$  electrode, in which the presence of large clusters and micropores is marked. The hierarchical structure of polyaniline and the existence of large pores in the case of the electrodes elaborated in HCl and  $H_2SO_4$  facilitate the incorporation of ions and electrons within the electrode and thus a relatively high doping/undoping degree. This results in an improved specifc capacitance (Cui et al. [2014\)](#page-11-14).

These fndings are in agreement with the results of Yang et al. ([2017\)](#page-12-24) studying the relationship between pore size and charge transfer resistance of carbon aerogels for organic double-layer capacitor electrodes.

The galvanostatic charge–discharge curves of each electrode (Fig. [6\)](#page-7-0) were carried out over the 0–0.7-V window for the current density 0.2 mA/cm<sup>2</sup>. Quasi-linear and symmetrical variations of the voltage with time observed during the charging and discharging phase confrm the pseudocapacitive nature of PANI/ITO electrodes developed in  $H_2SO_4$ ,  $HNO<sub>3</sub>$ , and HCl. However, the galvanostatic charge–discharge curve of the electrode prepared in  $H_3BO_3$  shows a horizontal part characteristic of battery-type electrodes (Storage [2018\)](#page-12-25).

Figure [6](#page-7-0) b shows the specifc capacitance (evaluated by Eq. [9\)](#page-7-1) of the PANI/ITO electrodes electrodeposited in the diferent mediums.

<span id="page-7-1"></span>
$$
C = j\Delta t / \Delta E \tag{9}
$$

where *j* is the applied current density  $(A/cm<sup>2</sup>)$ ,  $\Delta t$  is the discharge time (s), and  $\Delta E$ =0.7 (V) is the potential window.

The galvanostatic charge–discharge confrms the results of the cyclic voltammetry. The highest specifc capacitance is that of the PANI/ITO electrode elaborated in  $H_2SO_4$ medium. This justifes the choice of this acid as an electrodeposition medium in the following work to study the efect of the electrodeposition potential on the electrochemical performance of the PANI/ITO electrode.

### **Efect of electrodeposition potential and monomeric concentration**

In a frst step, the elaboration of PANI/ITO electrodes was performed by the potentiostatic method at 0.80 V, 0.85 V, 0.90, 0.95, and 1.0 V/SCE potentials from a fxed concentration of monomeric aniline (0.2 M). The recorded current density-time transients (*j*-*t*), which are presented in Fig. [7](#page-8-0)a, exhibit the same pattern, a rapid decay part corresponding to the charging of the electrochemical double layer and adsorption of ions onto the substrate, followed by an increase in current density whose speed depends on the electrodeposition potential and, fnally, the current density converges towards a plateau. On the exponentially ascending part, we note the presence of a more or less discernible infection point, which indicates that the growth of polyaniline flms goes through two stages. The frst one corresponds to the processes of nucleation and growth of a polyaniline layer on the ITO until its complete coverage and during the second one the electropolymerization continues by growth of the polymer chains, accompanied by the ramifcations (Aynaou et al. [2022;](#page-11-9) Bade et al. [1992\)](#page-11-15). This interpretation can be completed by the intervention of an autocatalytic radical growth mechanism

<span id="page-7-0"></span>**Fig. 6 a** Galvanostatic charge– discharge of PANI/ITO electrodes elaborated in diferent acids between 0 and 0.7 V in  $H_2SO_4$  (0.5 M)/Na<sub>2</sub>SO<sub>4</sub> (0.1 M) at 0.2 mA/cm<sup>2</sup>. **b** Specific capacitance of four PANI/ITO electrodes at 0.2 mA/cm.<sup>2</sup>



<span id="page-8-0"></span>**Fig. 7** Chronoamperometric curves obtained on ITO electrode, **a** in 0.2 M aniline at 0.8, 0.85, 0.9, 0.95, and 1.0 V/SCE and **b** in 0.05 M, 0.1 M, 0.2 M, and 0.4 M of aniline at 0.95 V/ **SCE** 



(Stilwell & Park [1988\)](#page-12-26). Often, electropolymerization starts with the formation of soluble oligomers in the solution followed by a nuclei formation step on the substrate surface. This step can be done in two ways, successive additions of monomers or oligomers on the oxidized monomers attached to the surface or by precipitation of dissolved oligomers on the electrode. The frst way is similar to the electrocrystallization of metals and metal oxides; the nucleation rate depends directly on the potential. While in the second way, the potential acts only on the generation of oligomers in the solution (Komsiyska et al. [2007\)](#page-11-16). By analyzing Fig. [1](#page-2-0)a, we can see that the current density and thus the electropolymerization rate depend signifcantly on the potential, which validates the hypothesis of successive additions. Figure [7](#page-8-0) b highlights the dependence of the growth rate of polyaniline flms on the potential in conformity with previous studies that have found that the electropolymerization is frst order in aniline concentration (aniline) (Eq. [10](#page-8-1)) (Wei et al. [1989](#page-12-27)):

$$
r = k_{\rm app}[\text{aniline}] \tag{10}
$$

where  $r$  is the electropolymerization reaction rate, and  $k_{\text{app}}$  is the apparent reaction rate constant. The rate of electropolymerization depends on the concentration of oligomers [oligomers] according to an autocatalysis mechanism (Eq. [11\)](#page-8-2) (Mondal et al. [2007\)](#page-12-28).

<span id="page-8-2"></span>
$$
r = k[\text{aniline}][\text{oligomers}]^{1/2} \tag{11}
$$

We retain that the concentration and the potential (via the creation of oligomers) afect the rate of formation of polyaniline flms, and thus their morphologies and textures.

In Fig. [8](#page-8-3)a, we have plotted the cyclic voltammograms recorded at the scan rate 20 mV/s for the five electrodes over the electrochemical window 0–0.7 V. We notice that these voltammograms keep the same shape, an almost rectangular shape with the presence of oxidation and reduction peaks which is a characteristic of pseudocapacitive materials (Wang et al. [2016](#page-12-29); Gogotsi and Penner [2018;](#page-11-17) Nguyen et al. [2021](#page-12-30)).

<span id="page-8-1"></span>We have plotted in Fig. [8b](#page-8-3) the variation of the specific capacitance of each electrode as a function of the scan rate.



<span id="page-8-3"></span>**Fig. 8 a** Cyclic voltammograms at scan rate 20 mV/s for PANI/ ITO electrodes elaborated at diferent potentials; **b** evolution of the specifc capacitance as a function of the scan rate

The specifc capacitance of the electrodes electrodeposited at the diferent potentials decreases with the increase of the scan rate from 5 to 60 mV/s. This evolution can be explained by the decrease of the faradic process contribution when the scan rate increases (Lindstrom et al. [1997;](#page-12-31) Wang et al. [2007](#page-12-32)). In other words, the redox reactions do not have enough time to take place. On the other hand, the specifc capacitance increases with the electrodeposition potential; it passes by a maximum at 0.95 V/SCE, and then, it decreases. These fndings can be justifed by the density of polarons and the form of polyaniline electrodeposited on each electrodeposition potential range [28, 29]. In the zone of increase of the specifc capacitance (potential lower than 0.95 V/SCE), one obtains mainly the emeraldine salt, the most conductive form, within which the density of the polarons increases with the potential. In the range of potential higher than 0.95 V/SCE, we obtain mainly the less conductive pernigraniline. For potentials below 0.95 V/SCE, the increase of specifc capacitance with potential can be correlated to the morphology of the flms. At low potentials, the radical activation and therefore the electropolymerization are slow; we obtain a denser and less porous structure and hence a less important stored charge (Andrade et al. [1998\)](#page-11-18).

The galvanostatic charge–discharge curves of each electrode (Fig. [9a](#page-9-0)) were carried out over the 0–0.7-V window at diferent current densities. Quasi-linear and symmetrical variations of the voltage with time observed during the charging and discharging phase confrms the pseudocapacitive nature of PANI/ITO electrodes.

Figure [9](#page-9-0) b shows the specifc capacitance (evaluated by Eq. [9\)](#page-7-1) (Mathis et al. [2019a,](#page-12-18) [b](#page-12-19); Shieh et al. [2016](#page-12-33)) of the PANI/ITO electrodes electrodeposited at diferent potentials.

The galvanostatic charge–discharge confrms the results of the cyclic voltammetry. The highest specifc capacitance is that of the PANI/ITO electrode developed at 0.95 V/SCE.



<span id="page-9-2"></span>Fig. 10 Coulombic efficiency as a function of current density of different electrodes

It varies from 23.6 mF/cm<sup>2</sup> for 0.2 mA/cm<sup>2</sup> to 18.8 mF/cm<sup>2</sup> for  $1.0 \text{ mA/cm}^2$ .

Using the galvanostatic charge–discharge curves, we can also determine the coulombic efficiency defined as the ratio between the discharge time  $t_D$  and the charge time  $t_C$  when the charge–discharge current densities are equal (Eq. [12](#page-9-1)):

<span id="page-9-1"></span>
$$
coulombic efficiency = \frac{t_D}{t_C} \times 100
$$
 (12)

By analyzing Fig. [10](#page-9-2) representing the coulombic efficiency as a function of current density for each electrode, it

<span id="page-9-0"></span>**Fig. 9 a** Galvanostatic charge–discharge of PANI/ ITO electrodes elaborated at diferent potentials between 0 and 0.7 V in  $H_2SO_4$  (0.5 M)/  $Na<sub>2</sub>SO<sub>4</sub> (0.1 M)$  at 0.2 mA/  $cm<sup>2</sup>$ . **b** Evolution of the specific capacitance as a function of current density



can be seen that it increases with the current density, that is to say that the galvanostatic charge–discharge curves become more and more symmetrical. This evolution is explained by the predominance of the capacitive character at high current densities. On the other hand, the highest specifc capacitance is obtained for electrodes elaborated at low electrodeposition potentials.

To study the infuence of the monomer concentration on the electrochemical properties of the PANI/ITO electrode, we have elaborated by the potentiostatic method for electrodes at 0.95 V/SCE, from diferent concentrations of monomeric aniline (0.05, 0.1, 0.2, and 0.4 M) dissolved in 1 M  $H<sub>2</sub>SO<sub>4</sub>$ . In Fig. [11a](#page-10-0), b, the evolution of the specific capacitance serves as a function of the scan rate and discharge current density respectively. We can see that the specifc capacitance increases with increasing aniline concentration over this range. The amount of monomer near the substrate afects the morphology of the flm and the way the deposit is formed. For low concentrations, the flm that forms slowly is relatively ordered and has a less porous structure. At high concentrations of aniline, the flm forms rapidly, which makes it less ordered and more porous. The SEM images (Fig. [12](#page-10-1)a, b) confrm these fndings.

## **Conclusion**

The analysis of the specifc capacitance values obtained by cyclic voltammetry and galvanostatic charge–discharge shows that the counter ion acts directly on the electrochemical performance of the PANI/ITO electrode. For the same rate, the specifc capacitance increases in the following order of the counter ion  $BO_3^{3-} < NO_3^{-} < Cl^{-} < SO_4^{2-}$ . It increases from 1.6 mF/cm<sup>2</sup> in case  $BO_3^{3-}$  to 64.8 mF/ cm<sup>2</sup> in case  $SO_4^2$ <sup>-</sup> for a scan rate of 5 mV/s. This result was also confrmed by the galvanostatic charge–discharge. SEM images show a granular structure for the PANI/ITO electrode developed in  $H_3BO_3$  and more or less porous

<span id="page-10-0"></span>**Fig. 11** Evolution of the specifc capacitance values of PANI/ITO electrode: **a** with the scan rate, **b** discharge current density



<span id="page-10-1"></span>**Fig. 12** SEM image of polyaniline flms obtained at 0.95 V, **a** [aniline]=0.2 M, **b** [ani $line] = 0.4 M$ 



structures for the other electrodes. The calculation of capacitive and faradic contributions confrmed the pseudocapacitive nature of the electrodes elaborated in  $H_2SO_4$ , HCl, and HNO<sub>3</sub>, while the electrode elaborated in  $H_3BO_3$ can be used as a cathode in batteries. The polyaniline electrodeposition potential considerably afects the electrochemical performance of PANI/ITO electrodes. The specifc capacitance determined by cyclic voltammetry and galvanostatic charge–discharge method is maximum in the case of PANI/ITO electrode elaborated at 0.95 V/SCE. For this electrode, the specifc capacitance which decreases with current density and scan rate, increases from 23.6 mF/cm<sup>2</sup> at 0.2 mA/cm<sup>2</sup> to 18.8 mF/cm<sup>2</sup> at 1 mA/cm<sup>2</sup>, and from 24.3 to 19.6m $F/cm<sup>2</sup>$  when the scan rate varies from 5 to 60 mV/s.

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**Data availability** The authors confrm that the data supporting the fndings of this study are available within the article.

## **Declarations**

**Ethical approval** This manuscript is the author's original work, which has not been previously published elsewhere, and the paper is not currently being considered for publication elsewhere. We agree with the above statements and declare that this submission follows the policies of the Journal of Materials Science as outlined in the Guide for Authors and in the Ethical Statement. All authors have been personally and actively involved in substantial work leading to the paper and will take public responsibility for its content.

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**Competing interests** The authors declare no competing interests.

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