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# **Preparation of F-Doped TiO<sub>2</sub> Photocatalysts by Gas–Liquid Plasma at Atmospheric Pressure**

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Abstract F-Doped TiO<sub>2</sub> was successfully fabricated using gas-liquid cold plasma with different fluorine sources, including NaF, NaF combined with [C<sub>4</sub>MIM]  $HSO_4$ , and  $[C_4MIM]BF_4$ . The samples were characterized by XRD (X-ray diffraction), XPS (X-ray photoelectron spectroscopy), and BET (N<sub>2</sub> adsorption-desorption isotherms). Photocatalytic activity of the samples was tested by degradation of methylene blue (MB) dye under simulated sunlight. The results show that doping with different fluorine sources was beneficial to form interstitial F-doped TiO<sub>2</sub>, and ionic liquids used in our work were most favorable to the formation of interstitial F-doped TiO<sub>2</sub>. Calcination of the 0.6-F-[C4MIM]BF4-TiO2 sample at 500°C shows that interstitial F-doped TiO<sub>2</sub> has an excellent thermal stability, a suitable mesoporous structure, higher specific surface area (60.5  $m^2 g^{-1}$ ) and the best photocatalytic activity. MB degradation was about 98% under the stimulated sunlight for 30 min, and its apparent rate constant was 42 times higher than photolysis and was about 2.9 times as that prepared without addition of ionic liquid. Furthermore, the stability of ionic liquids in the gas-liquid plasma was also investigated and discussed with the assistance of Fourier transform infrared (FTIR) spectroscopy.

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<sup>1</sup> College of Physical Science and Technology, Dalian University, Dalian 116622, China **Keywords** Gas–liquid plasma  $\cdot$  Dielectric barrier discharge  $\cdot$  F-Doped  $\cdot$  TiO<sub>2</sub>  $\cdot$  Ionic liquid

### **1** Introduction

Titanium dioxide (TiO<sub>2</sub>) is the most promising photocatalyst and has drawn much research attentions due to its high efficiency, low cost, non-toxicity and high photo-stability [1-5]. However, the application of TiO<sub>2</sub> has been limited owing to its large band gap energy (3.20 eV for anatase TiO<sub>2</sub>), which is not helpful to fully use of the sunlight. Doping with non-metal ions such as N, C, B and F has been proved to be an efficient method to narrow the band gap of TiO<sub>2</sub> and to broader light response range towards visible wavelength [6-9]. Among these non-metal elements, doping TiO<sub>2</sub> with F has been demonstrated an effective approach to enhance its visible light photocatalytic activity. Yu et al. [10] synthesized F-doped TiO<sub>2</sub> with NH<sub>4</sub>F and tetrabutyl titanate using a modified sol-gel method including a process of drying in air at 100 °C for 24 h and then heating at 500 °C. The F-doped TiO<sub>2</sub> exhibited an enhanced photocatalytic performance for MB degradation with a degradation rate of 91% compared to pure TiO<sub>2</sub> (32%). Pan et al. [11] prepared mesoporous F-TiO<sub>2</sub> spheres consisting of anatase nanocrystallites using titanium alkoxides and NH<sub>4</sub>F as Ti and F precursors, respectively. After calcination at 450 °C for 2 h, the obtained mesoporous  $F-TiO_2$ spheres showed superior photocatalytic activity in dye degradation. These traditional sol-gel or solvothermal methods indicated that synthesizing F-doped TiO<sub>2</sub> is a good strategy to improve the photocatalytic performance of TiO<sub>2</sub>.

Recently, gas-liquid plasma technology has been developed and has been used to treat waste water, promote chemical reactions and prepare nanomaterials [12–15]. Ionic liquids (ILs), which consists of only positive and negative ions, have extremely low vapor pressure, high heat capacity and chemical stability. They have been successfully used in different kinds of reactions as green reaction media. The interaction of plasma and ionic liquid may change the properties of plasma and thus vary the intensity and direction of the gaseous phase reaction in the plasma. Yu et al. [13] reported the catalytic action of nine imidazolium-based ionic liquids for methane conversion in a direct current (DC) discharge plasma. Ionic liquid C<sub>6</sub>MIMHSO<sub>4</sub> (1-hexyl-3-methylimidazolium hydrogen sulphate) enhanced the conversion of methane and C<sub>6</sub>MIMBF<sub>4</sub> (1-hexyl-3-methylimidazolium tetrafluoroborate) increased the selectivity of  $C_2$  hydrocarbons. Zhou et al. [16] investigated SiO<sub>2</sub> supported ionic liquid with Pd systems for the selective hydrogenation of acetylene with the assistance of RF non-thermal plasma. The results suggested that supported Pd catalysts loaded with ionic liquids were effective for selective hydrogenation of acetylene in ethylene. The catalysts with ionic liquids of BF<sub>4</sub><sup>-</sup> anions usually achieve higher conversion but lower selectivity, and those catalysts with  $PF_6^-$  anions give lower conversion but higher selectivity. Zhang et al. [17] studied the influences of  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> supported C<sub>6</sub>MIMBF<sub>4</sub> and Pd on methane conversion to C<sub>2</sub> hydrocarbons under cold plasma. The results indicated that the selectivity and yield of  $C_2$ hydrocarbons were increased with the increase in immobilization amount of C<sub>6</sub>MIMBF<sub>4</sub>. Furthermore, plasma also enhances the synthetic reaction in the ionic liquid or water and ionic liquid mixed solution. Wei et al. [18] gained gold nanoparticles using the subatmospheric dielectric barrier discharge plasma reduction in the C<sub>4</sub>MIMBF<sub>4</sub> (1-butyl-3-methylimidazolium tetrafluoroborate). Hatakeyama et al. [14] presented an effective preparation method of Au nanoparticles (NPs) in an arc plasma deposition technique combined with ionic liquids as capture media. Tran et al. [15] synthesized Platinum nanoparticles (PtNPs) using a liquid plasma system with C<sub>4</sub>MIMBF<sub>4</sub> under atmospheric pressure. Zhang et al. [19] prepared anatase-rutile mixed phase  $TiO_2$  with high photocatalytic activity by atmospheric-pressure dielectric barrier discharge (DBD) gas-liquid cold plasma using C<sub>4</sub>MIMBF<sub>4</sub> as assiantance.

In this work, F-doped anatase  $\text{TiO}_2$  photocatalysts were synthesized by atmospheric-pressure gas-liquid cold plasma, using titanium oxysulfate as Ti precursor and NaF solution, mixed solution of NaF and [C<sub>4</sub>MIM] HSO<sub>4</sub>, and [C<sub>4</sub>MIM]BF<sub>4</sub> as the F precursor. The phase structure, doping type, thermal stability and photocatalytic performance of the F-doped TiO<sub>2</sub> samples have been investigated.

### 2 Experimental

## 2.1 Materials

Titanium oxysulfate [TiOSO<sub>4</sub>, analytical reagent (AR)grade] was obtained from Tianjin Guangfu Fine Chemical Research Institute, China. Sodium fluoride [NaF, analytical reagent (AR)-grade] was obtained from Aladdin. 1-Butyl-3-methylimidazolium tetrafluoroborate ([C<sub>4</sub>MIM]BF<sub>4</sub> ionic liquid, purity >99%) and 1-butyl-3-methyl-imidazolium hydrogen sulphate ([C<sub>4</sub>MIM]HSO<sub>4</sub> ionic liquid, purity >99%) were purchased from Lanzhou Institute of Chemical Physics, Chinese Academy of Sciences, China. All of the materials were used as received.

### 2.2 Catalyst Preparation

The preparation process was similar with our previous work [19]. Briefly, certain amount of fluorine sources (0.8 g NaF, the mixture of 0.8 g NaF and 0.6 mL [ $C_4$ MIM]  $HSO_4$ , or 0.6 mL [C<sub>4</sub>MIM]BF<sub>4</sub>) were added to the solution of 10 g TiOSO<sub>4</sub> and 50 mL deionized water under vigorous stirring to obtain the mixture of Ti precursors solution containing the fluorine source. Then the process of plasma treatment was conducted to prepare the fluoride-doped TiO<sub>2</sub> (F-TiO<sub>2</sub>) photocatalysts, which was performed at atmospheric pressure in open air for twice, with an interval of 5 min between the two operations. Each treatment just takes 5 min. Then the obtained solution was filtered, washed and dried under vacuum for 2 h at 100 °C to remove the impurities remaining in the TiO<sub>2</sub> networks. Corresponding to the different fluorine sources (NaF, NaF combined with  $[C_4MIM]HSO_4$ , and  $[C_4MIM]BF_4$ ), the obtained samples are denoted as F-TiO<sub>2</sub>, F-[C<sub>4</sub>MIM]HSO<sub>4</sub>-TiO<sub>2</sub> and  $F-[C_4MIM]BF_4-TiO_2$ , respectively.

The schematic diagram of the atmospheric-pressure DBD gas–liquid cold plasma device for preparing F-doped  $\text{TiO}_2$  photocatalysts can be seen in previous work [20]. A quartz reaction cell (60 mm in diameter) was placed between the high-voltage electrode and ground electrode, both of which were made of stainless steel (50 mm in diameter). The two quartz dielectrics were both 1 mm thick, and the discharge gap was 4 mm. A sinusoidal high voltage at a frequency of 14.2 kHz was applied to conduct the plasma treatment with a CTP-2000K power source provided by Nanjing Suman Electronic Co. Ltd, China.

#### 2.3 Catalyst Characterization

The F-doped TiO<sub>2</sub> samples were characterized by powder X-ray diffraction (XRD) on an X-ray diffractometer (DX-2700, Dandong Haoyuan, China) using Cu K $\alpha$ 1 radiation ( $\lambda$ =1.54178 Å) with a step-scan of 0.03 degree per step,

corresponding to 3.6 degree per minute. The chemical compositions of the samples were investigated by X-ray photoelectron spectroscopy (XPS, ESCALAN250 Thermo VG) using a monochromatized AlK $\alpha$  (1486.6 eV) X-ray source. All binding energies were referenced to the XPS peak of carbon 1 s at 284.6 eV. The pore size ( $D_p$ ) distributions, Brunauer–Emmett–Teller (BET) surface areas ( $S_{BET}$ ), and pore volumes ( $V_p$ ) were measured by nitrogen adsorption and desorption using a gas sorption analyzer (NOVA 2200e, Quantachrome Corp., USA). Prior to the adsorption analysis, the samples were outgassed at 200 °C for 5 h in the degassing port of the analyzer. The C<sub>4</sub>MIMBF<sub>4</sub> used before and after gas–liquid plasma treatment was detected on a Spectrum one-B FTIR (PerkinElmer).

### 2.4 Photocatalytic Activity Test

Photocatalytic activity of the F-TiO<sub>2</sub> samples was tested based on the degradation of a methylene blue (MB) dye solution using a 300 W Xe lamp. Detailed information about the photocatalytic activity test can be found in our previous work [21]. In brief, the photocatalytic reaction was carried out with 0.1 g catalyst suspended in 100 mL of a MB solution (10 mg  $L^{-1}$ ) in a Pyrex glass cell. The solution, which was kept at pH=6, was stirred in the dark for 30 min to obtain a good dispersion and established the adsorption-desorption equilibrium between the MB molecules and the catalyst surface. At given time intervals, the slurry samples including the photocatalysts and MB were centrifuged and the solution was analyzed with a UV-visible spectrometer (721 Spectrophotometer, Shanghai Jinghua Group Co., Ltd, China). Since good linear relationship between the concentration and the absorption of MB was obtained at the wavelength of 665 nm, the MB degradation  $\eta$  is defined as

$$\eta = \frac{c_0 - c}{c_0} \times 100\%$$
(1)

where  $c_0$  is the initial concentration of MB (mg L<sup>-1</sup>) and c is the concentration of MB (mg L<sup>-1</sup>) at a certain time.

### **3** Results and Discussion

# 3.1 Effect of Fluorine Sources on the Phase Structure of TiO<sub>2</sub>

Figure 1 shows the XRD patterns of  $TiO_2$  and F-doped  $TiO_2$  samples prepared with different fluorine sources (NaF, NaF combined with  $[C_4MIM]HSO_4$ , and  $[C_4MIM]BF_4$ ), namely F-TiO\_2, F- $[C_4MIM]HSO_4$ -TiO\_2 and F- $[C_4MIM]BF_4$ -TiO\_2 samples. As shown in Fig. 1, the diffraction peaks at 25.3° and 48.0° appeared, which were indexed



Fig. 1 XRD patterns of F-doped  $\text{TiO}_2$  prepared by gas-liquid plasma with different fluorine sources

to (101) and (200) reflections of anatase TiO<sub>2</sub> (PDF No. 21-1272). By carefully comparing the diffraction peaks for (101) and (200) in the three F-doped TiO<sub>2</sub>, we can find that the (101) diffraction peaks at round 25° shift to lower  $2\theta$  values, while the (200) diffraction peaks at around  $48.0^{\circ}$  shift toward higher  $2\theta$  values. Similar phenomenon was also observed in the work when doping with fluorine in preparing ZnWO<sub>4</sub> catalyst [22, 23]. The shift of the diffraction peak for F-doped TiO<sub>2</sub> samples indicates that lattice imperfection was formed due to F doping. The fluorine atoms enter into the lattice of TiO<sub>2</sub>, therefore breaks the equilibrium of original TiO<sub>2</sub> atoms and varies the interplanar crystal spacing of anatase TiO<sub>2</sub>. The Bragg equation illustrated below clearly reveals the functional relationship between the interplanar crystal spacing and the diffraction angle:

$$d_{(hkl)} = \frac{n\lambda}{2\sin\theta} \tag{2}$$

where  $d_{(hkl)}$  is the interplanar crystal spacing,  $\lambda$  is the wavelength of X ray and  $\theta$  is the angle between X ray and the (hkl) crystal face. When doping with fluorine, the  $d_{(hkl)}$  value is varied by the lattice deformation and the value of  $\theta$  changes correspondingly, and the change of  $\theta$  value becomes greater when the dopants of fluorine increases.

Table 1 demonstrates the relative shift values of diffraction angle at around 25.3° and 48.0° of F-doped TiO<sub>2</sub> prepared with different fluorine sources. The relative shift value of diffraction angle for the diffraction peak at around 25.3° for F–TiO<sub>2</sub> samples is 0.8°, while the relative shift value for the diffraction peak at around 48.0° is 0.2°. As for the F–[C<sub>4</sub>MIM]HSO<sub>4</sub>–TiO<sub>2</sub> and F–[C<sub>4</sub>MIM]BF<sub>4</sub>–TiO<sub>2</sub>

Samples	20	Relative shift (toward lower $2\theta$ value)	20	Relative shift (toward higher $2\theta$ value)
Anatase TiO <sub>2</sub>	25.3°	_	48.0°	-
F-TiO <sub>2</sub>	24.5°	0.8°	48.2°	0.2°
F-[C <sub>4</sub> MIM]HSO <sub>4</sub> -TiO <sub>2</sub>	24.3°	1.0°	48.4°	0.4°
F-[C <sub>4</sub> MIM]BF <sub>4</sub> -TiO <sub>2</sub>	24.3°	1.0°	48.4°	0.4°



Fig. 2 XPS spectra of F1s in F–TiO<sub>2</sub>, F–[C<sub>4</sub>MIM]HSO<sub>4</sub>–TiO<sub>2</sub> and F–[C<sub>4</sub>MIM]BF<sub>4</sub>–TiO<sub>2</sub>

samples, the shift values are both  $1.0^{\circ}$  for the diffraction peak at around  $25.3^{\circ}$  and  $0.4^{\circ}$  for the diffraction peak at around  $48.4^{\circ}$ , which are greater than the F–TiO<sub>2</sub> samples. This reveals that [C<sub>4</sub>MIM]HSO<sub>4</sub> and [C<sub>4</sub>MIM]BF<sub>4</sub> used here are helpful for the formation of F-doped TiO<sub>2</sub>.

# 3.2 Effect of Fluorine Sources on the F Doping Types of TiO<sub>2</sub>

The XPS spectra of F1s in F–TiO<sub>2</sub>, F–[C<sub>4</sub>MIM]HSO<sub>4</sub>–TiO<sub>2</sub> and F–[C<sub>4</sub>MIM]BF<sub>4</sub>–TiO<sub>2</sub> samples are shown in Fig. 2. The F1s peaks were observed in all of the F-doped TiO<sub>2</sub> samples, and may be divided into two peaks at 684.2 and 684.6 eV with Gaussian distributions. Specifically, the peak at 684.2 eV is assigned to the F<sup>-</sup> ions that are physically adsorbed on the surface of TiO<sub>2</sub> [23–25]. According to the literature [24, 25], the peak at around 688.7 eV is the F in solid solution TiO<sub>2-x</sub>F<sub>x</sub> formed by the substitution of O<sup>2-</sup> with F<sup>-</sup>. Therefore the peak at 684.6 eV observed here should be ascribed to the interstitial doping F<sup>-</sup> ions in TiO<sub>2</sub> lattice. As we know, the structure of anatase TiO<sub>2</sub> belongs to the tetragonal system, which is composed of TiO<sub>6</sub> octahedron. The O<sup>2-</sup> ions in TiO<sub>2</sub> are closely packed to form octahedral interstices and tetrahedral interstices. Half of the octahedral interstices may be occupied by the Ti<sup>4+</sup> ions to form TiO<sub>6</sub> octahedron, while the left octahedral interstices and tetrahedral interstices are vacant. There are two methods for preparing F-doped TiO<sub>2</sub>. One is substitution of  $F^{-}$  for  $O^{2-}$  in the TiO<sub>2</sub> lattice crystal. Since the atomic radii of  $F^-$  (0.133 nm) and  $O^{2-}$  (0.140 nm) are about the same, the substitutional F-doping will not change the lattice spacing of TiO<sub>2</sub>. Therefore, as confirmed by the XRD test and the Bragg equation discussed above, the observed shift of diffraction peak toward lower angle may not be due to the smaller lattice parameter resulted from the substitution of  $O^{2-}$  by F<sup>-</sup> [24–28]. The other kind of F-doping is interstitial doping. Since the atomic radius of F<sup>-</sup> is large, it can only enter into the octahedral void with large space, and a small change in lattice parameter may occur. This may result in the change of lattice space and the shift of diffraction peak [22, 23].

Furthermore, the F1s peak areas of the three F-doped  $TiO_2$  samples are significantly different. The peak area of interstitial doping F<sup>-</sup> ions for F-TiO<sub>2</sub> is much smaller than the F-[C<sub>4</sub>MIM]HSO<sub>4</sub>-TiO<sub>2</sub> and F-[C<sub>4</sub>MIM]BF<sub>4</sub>-TiO<sub>2</sub> samples. In conclusion, all of the F-doped TiO<sub>2</sub> samples prepared with gas-liquid plasma present the interstitial F doping, and the ionic liquids of [C<sub>4</sub>MIM]HSO<sub>4</sub> and [C<sub>4</sub>MIM]BF<sub>4</sub> are beneficial to the entrance of F<sup>-</sup> ions into the interstitial space of TiO<sub>2</sub> lattice.

#### 3.3 Thermal Stability of the F-Doped TiO<sub>2</sub>

In order to investigate the thermal stability of the F-doped TiO<sub>2</sub> samples, F–TiO<sub>2</sub>, F–[C<sub>4</sub>MIM]HSO<sub>4</sub>–TiO<sub>2</sub> and F–[C<sub>4</sub>MIM]BF<sub>4</sub>–TiO<sub>2</sub> were calcined at 500 °C for 2 h, and denoted as F–TiO<sub>2</sub>-500 °C, F–[C<sub>4</sub>MIM]HSO<sub>4</sub>–TiO<sub>2</sub>-500 °C and F–[C<sub>4</sub>MIM]BF<sub>4</sub>–TiO<sub>2</sub>-500 °C, respectively. XRD patterns of the F-doped TiO<sub>2</sub> samples are shown in Fig. 3. It demonstrates that after calcination the diffraction peak at around 25.3° has no shift for the F–TiO<sub>2</sub>-500 °C sample, however, it still shifts to a lower diffraction angle for F–[C<sub>4</sub>MIM]HSO<sub>4</sub>–TiO<sub>2</sub>-500 °C and F–[C<sub>4</sub>MIM]BF<sub>4</sub>–TiO<sub>2</sub>-500 °C. It reveals that the interstitially doped fluorine impurities within F–TiO<sub>2</sub>-500 °C are not as thermally stable as the other two samples. Besides, the shift of diffraction peaks of F–[C<sub>4</sub>MIM]HSO<sub>4</sub>–TiO<sub>2</sub>-500 °C and F–[C<sub>4</sub>MIM]



Fig. 3 XRD patterns of F-doped TiO\_2 with different fluorine sources calcined at 500  $^{\circ}\mathrm{C}$ 

 $BF_4$ -TiO<sub>2</sub>-500 °C sample is smaller (from 1.0° decreasing to 0.4°) after calcination, indicating the decrease of interstitial doping F contents in these two samples. Furthermore, all of the diffraction peaks of samples after calcination show a higher intensity, which shows that the calcination process may enhance the crystallization of F-doped TiO<sub>2</sub>. Interestingly, the diffraction peaks of the TiO<sub>2</sub> samples are greater enhanced when using ionic liquids as the fluorine sources. Thus it may be concluded that the fluorine sources have significant effect both on the thermal stability and crystallinity of the F-doped TiO<sub>2</sub> samples.

XPS spectra of F1s in F-doped TiO<sub>2</sub> with different fluorine sources calcined at 500 °C were also obtained, as shown in Fig. 4. As for F–TiO<sub>2</sub>-500 °C, the physically adsorbed and interstitial doping F<sup>-</sup> ions at around 684 eV disappeared. For F–[C<sub>4</sub>MIM]HSO<sub>4</sub>–TiO<sub>2</sub>-500 °C and F–C<sub>4</sub>MIM]BF<sub>4</sub>–TiO<sub>2</sub>-500 °C, only the interstitial doping F<sup>-</sup> ions at the binding energy of 684.6 eV were observed, and the intensities were decreased compared with the samples without calcination. Vanish of the physically adsorbed F<sup>-</sup> ions for F-doped TiO<sub>2</sub> samples indicated that calcination played an important role on the stability of F-doped TiO<sub>2</sub>, and ionic liquid is beneficial to enhance the stability of interstitial F-doping TiO<sub>2</sub>.

The N<sub>2</sub> absorption-desorption isotherms and pore size distributions of F-doped TiO<sub>2</sub> samples with different fluorine sources calcined at 500 °C are also observed and illustrated in Fig. 5 and Table 2. All of the calcined F-doped TiO<sub>2</sub> samples exhibit type-IV isotherm plots and H3 hysteresis loop [20], as shown in Fig. 5a. In addition, there is no significant difference between their specific surface area, which indicates that the pore structure of F-doped



Fig. 4 XPS spectra of F1s in F-doped TiO2 with different fluorine sources calcined at 500  $^{\circ}\mathrm{C}$ 

TiO<sub>2</sub> samples is thermally stable against the calcination at 500 °C.

## 3.4 Photocatalytic Activity of the Post-Calcined F-Doped TiO<sub>2</sub>

Figure 6a represents the MB photodegradation over F–TiO<sub>2</sub>-500 °C, F–[C<sub>4</sub>MIM]HSO<sub>4</sub>–TiO<sub>2</sub>-500 °C and F–[C<sub>4</sub>MIM]BF<sub>4</sub>\_TiO<sub>2</sub>-500 °C under simulated sunlight. It has been shown that the F-doped TiO<sub>2</sub> photocatalysts prepared by gas–liquid plasma using different fluorine sources and calcined at 500 °C all showed high photocatalytic activity for MB degradation. The F–[C<sub>4</sub>MIM]BF<sub>4</sub>\_TiO<sub>2</sub> shows the best activity for MB photodegradation. According to the values taken from Fig. 6a,  $\ln(c_0/c)$  for the samples was plotted against the irradiation time, as shown in Fig. 6b. Good linear relationships between the two parameters were obtained, which confirms the pseudo-first-order reaction for the degradation of MB as discussed in previous work [19].

The apparent rate constants (*k*) of MB photodegradation, obtained from the slopes of the best-fit lines, are 0.0252, 0.0390 and 0.0719 min<sup>-1</sup>, for F–TiO<sub>2</sub>-500 °C, F–[C<sub>4</sub>MIM] HSO<sub>4</sub>–TiO<sub>2</sub>-500 °C and F–[C<sub>4</sub>MIM]BF<sub>4</sub>–TiO<sub>2</sub>-500 °C, respectively (Fig. 6c). The apparent rate constant for MB photodegradation over F–[C<sub>4</sub>MIM]BF<sub>4</sub>–TiO<sub>2</sub>-500 °C is the highest and is 42 times higher than photolysis (0.0017 min<sup>-1</sup>), and was about 2.9 times as that prepared without addition of ionic liquid.

### 3.5 The Stability of ILs in Preparing F-Doped TiO<sub>2</sub> by Gas–Liquid Plasma

The process of preparing F-doped  $TiO_2$  by gas-liquid plasma is a fast reaction. During the preparation process,

Fig. 5 a  $N_2$  absorption–desorption isotherms and **b** pore size distributions of F-doped TiO<sub>2</sub> samples with different fluorine calcined at 500 °C



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**Table 2** Pore volume ( $V_p$ ), pore diameter ( $D_p$ ) and specific surface area ( $S_{BET}$ ) of F-doped TiO<sub>2</sub> prepared with different fluorine sources calcined at 500 °C

Samples	$V_{\rm p}/({\rm cm}^3 {\rm g}^{-1})$	$D_{\rm p}/({\rm nm})$	$S_{\rm BET}/({\rm m}^2~{\rm g}^{-1})$
F–TiO <sub>2</sub> -500 °C	0.02	7.21	64.1
F–[C <sub>4</sub> MIM]HSO <sub>4</sub> –TiO <sub>2</sub> - 500 °C	0.02	9.47	68.1
F–[C <sub>4</sub> MIM]BF <sub>4</sub> –TiO <sub>2</sub> -500 °C	0.03	3.79	65.0

part of the F<sup>-</sup> ions are physically absorbed on the surface of TiO<sub>6</sub> precursors, while others enter into the tetrahedral interstices formed by O<sup>2-</sup> ions, which results in interstitial doping F<sup>-</sup> ions. As discussed above, the doped F<sup>-</sup> ions are provided by NaF for F–TiO<sub>2</sub> and F–[C<sub>4</sub>MIM]HSO<sub>4</sub>–TiO<sub>2</sub> samples. However, as for F–[C<sub>4</sub>MIM]BF<sub>4</sub>–TiO<sub>2</sub> sample, they may be derived from the decomposition of [C<sub>4</sub>MIM] BF<sub>4</sub> ionic liquid. Therefore it would be necessary to investigate the stability of [C<sub>4</sub>MIM]BF<sub>4</sub> ionic liquid during the gas–liquid plasma process.

The FTIR spectra of the C<sub>4</sub>MIMBF<sub>4</sub> before (Fig. 7a) and after (Fig. 7b) gas–liquid plasma treatment are showed in Fig. 7. The spectra contains imidazole C–H stretching peaks at 3160 and 2121 cm<sup>-1</sup>; an aliphatic C–H stretching at 2958, 2933, and 2862 cm<sup>-1</sup>; an imidazole C–N stretching at 1574 cm<sup>-1</sup>; a methylene C–H scissoring at 1467 cm<sup>-1</sup>. The peaks at 1170 and 1062 cm<sup>-1</sup> contribute to the aromatic C–H in-plane bending vibration and B–F stretching, respectively [13]. There is no much difference between the spectra of C<sub>4</sub>MIMBF<sub>4</sub> before and after gas–liquid plasma treatment. This indicated that [C<sub>4</sub>MIM]BF<sub>4</sub> ionic liquid in plasma is stable. This has also been observed for ionic liquid [C<sub>4</sub>MIM]HSO<sub>4</sub> (Fig. S1).

Furthermore, the amount of F dopants is different for F–TiO<sub>2</sub>, F–[C<sub>4</sub>MIM]HSO<sub>4</sub>–TiO<sub>2</sub> and F–[C<sub>4</sub>MIM]BF<sub>4</sub>–TiO<sub>2</sub> samples, as revealed by Figs. 1 and



**Fig. 6** Photocatalytic activity of F-doped TiO<sub>2</sub> with different fluorine sources calcined at 500 °C **a** MB photodegradation, **b**  $\ln(c_0/c)$  as a function of time and **c** the apparent rate constants (*k*)



Fig. 7 FTIR spectra of the  $C_4$ MIMBF<sub>4</sub> ionic liquid *a* before and *b* after gas–liquid plasma treatment

2. The interstitial F dopants in F-[C<sub>4</sub>MIM]HSO<sub>4</sub>-TiO<sub>2</sub> and  $F-[C_4MIM]BF_4-TiO_2$  are relatively higher, which indicates that ionic liquids are helpful for forming interstitial F dopants. This may be explained by the formation of the  $\pi$ - $\pi$ bond between imidazole cationic, and the strong hydrogen bond between TiO<sub>6</sub> octahedral and HSO<sub>4</sub> and BF<sub>4</sub> anionic. The  $\pi$ - $\pi$  bond, hydrogen bond and Coulomb force could be structure-oriented in the formation of TiO<sub>2</sub>, therefore F<sup>-</sup> ions are easier in entering the tetrahedron intervals formed by  $O^{2-}$  ions and forming interstitial dopants. Besides, F-doped TiO<sub>2</sub> would nucleate faster in ionic liquids since its interfacial tension is lower. In conclusion, ionic liquids combined with plasma promote the formation of F-doped TiO<sub>2</sub>, proper amount of ionic liquids is conductive for preparing highly photocatalytic activity F-doped TiO2. Further work will be conducted to investigate the influence of F doping amounts and discharge parameters.

### 4 Conclusion

Interstitial fluorine doping TiO<sub>2</sub> photocatalysts were successfully prepared by gas–liquid plasma technology using NaF, NaF combined with [C<sub>4</sub>MIM]HSO<sub>4</sub>, and [C<sub>4</sub>MIM] BF<sub>4</sub>, as the fluorine precursors, respectively. Ionic liquids used in our work were beneficial to the formation of interstitial F-doped TiO<sub>2</sub>. The interstitial doping fluorine ions were thermally stable against calcination under the temperature of 500 °C. And they showed mesoporous structure, better crystallinity and higher photocatalytic activity for MB degradation. Notably, the F–[C<sub>4</sub>MIM]BF<sub>4</sub>–TiO<sub>2</sub>

sample showed the highest photocatalytic activity, which was 42 times higher than its photolysis and was about 2.9 times as that prepared without addition of ionic liquid. FTIR results indicate that  $[C_4MIM]BF_4$  ionic liquid is stable under plasma treatment.

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