

Coprecipitation nanoarchitectonics for the synthesis of magnetite: a review of mechanism and characterization

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Abstract

Magnetite nanoparticles have attracted interest of researchers from diferent science felds such as medicine, biology, physics, or chemistry due to their multifunctional properties including biocompatibility, superparamagnetism and low toxicity. Several methods have been developed to synthesize magnetic nanoparticles with controlled size, shape, and magnetic properties. Among them, coprecipitation is the most widely used because of several advantages such as high-yield production of nanoparticles, its simplicity, low-cost and its eco-friendly reaction conditions, but it also has low reproducibility. To achieve the magnetic nanoparticles with the desired properties, the investigation and control of reaction parameters are essential. In this article, the mechanism of coprecipitation reaction for magnetite nanoparticle synthesis and recent studies in reaction parameters controlling the particle properties will be reviewed. In addition, the most used methods for structural and magnetic characterization of magnetite and magnetite functionalized nanoparticles are presented and exemplifed.

Keywords Magnetite · Nanoparticle · Coprecipitation · Structural characterization · Superparamagnetism

Introduction

During the recent decades, magnetite (Fe_3O_4) nanoparticles have been attracted interest in many domains due to their specifc properties such as biocompatibility, low toxicity, strong magnetization, high durability, and low price [[1\]](#page-30-0). These characteristics of magnetite nanoparticles have been widely used in environmental remediation [[2–](#page-30-1)[7\]](#page-30-2), drug delivery [[8,](#page-30-3) [9\]](#page-30-4), magnetic resonance imaging (MRI) [[10\]](#page-30-5), as anti-wear and friction reducing additives in lubricating oil [\[11](#page-30-6)] catalysis [[12–](#page-30-7)[14\]](#page-30-8), electronics [[15\]](#page-30-9), biosensors [\[16](#page-30-10), [17\]](#page-30-11) and many other applications. All mentioned

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potential applications require specifc properties of magnetite nanoparticles, so in order to obtain the necessary chemical composition, shape, size, surface properties and magnetic properties, diverse methodologies to synthesize and functionalize them have been developed. These methods include chemical coprecipitation [\[18](#page-30-12)], thermal decomposition [[19\]](#page-30-13), micro emulsion [\[20](#page-31-0)], sol–gel [[21\]](#page-31-1), hydrothermal method [[22\]](#page-31-2), electrochemical method [\[23](#page-31-3)], bacterial method [\[24](#page-31-4)], ultrasonic assisted reverse coprecipitation [[25\]](#page-31-5), other combined procedures [[26\]](#page-31-6), etc.

The coprecipitation method is the most widely used process for magnetite nanoparticles synthesis because of the high yield of nanoparticles that can be synthesized, its simplicity, low-cost and its eco-friendly reaction conditions $[27-29]$ $[27-29]$, very important characteristics for industrial applications. Compared to some other techniques, the coprecipitation method also has some disadvantages, such as low repeatability, inappropriate size distribution of the particles and low crystallinity degree [\[30](#page-31-9)[–32](#page-31-10)]. The concentration, molar ratio of the reactants, pH and temperature are some important factors for controlling the phase and size of nanoparticles and saturation magnetization [\[33](#page-31-11)]. To achieve the magnetic nanoparticles with the desired properties, the investigation and control of the mentioned parameters are essential.

Although the coprecipitation method is one of the most used techniques for the synthesis of magnetite nanoparticles, the number of integrative papers analyzing the process parameters that infuence the yield and the properties of the particles is scarce. In this work, a comprehensive review focused on coprecipitation technique for synthesis of magnetite nanoparticles is presented. It highlights the synthesis parameters infuencing the properties of the magnetite nanoparticles and provides a short guide to the young and senior researchers working in this domain to work with convenient process parameters and to properly characterize them. The pH of the solution and the precipitating agent, temperature, ion precursor and ionic strength, and stirring rate have been discussed, exemplifed, and summarized. The mechanism of magnetite synthesis by coprecipitation method and the impurities formed during the process are also revised. Furthermore, a summary of magnetite and functionalized magnetite nanoparticles characterization methods is presented, with examples of each method and best practices, emphasizing their advantages and limitations (Scheme [1](#page-2-0)).

Crystal structure and properties of magnetite nanoparticles

Iron oxides exist in many forms in nature, among them magnetite (Fe_3O_4) , maghemite (γ -Fe₂O₃) and hematite (α -Fe₂O₃) being most popular. Magnetite, known also as magnetic iron ore or ferrous ferrite [[34\]](#page-31-12) is the most useful iron oxide and has been employed in various felds of applications, exhibiting the strongest magnetism among the transition metal oxides [[35\]](#page-31-13). According to Teja and Koh [\[34](#page-31-12)], magnetite has an inverse spinel structure with Fe(III) ions distributed randomly between octahedral and tetrahedral sites, and Fe(II) ions in octahedral sites. The inverse spinel cubic structure, as reported by Kozlenko et al. [[36\]](#page-31-14) is given in Fig. [1.](#page-2-1) Magnetite crystallizes with an inverse spinel cubic structure, with $Fe³⁺$ ions in the sites with tetrahedral oxygen coordination (A) and a mixture of Fe^{3+} and Fe^{2+} (equal proportion) in

Scheme 1 Chemical representation of magnetite and maghemite structure [[39,](#page-31-16) [40](#page-32-0)] Here t is tetrahedral site, o is octahedral site, and \square is vacancy

Fig. 1 The inverse spinel cubic (left) and post-spinel orthorhombic (right) crystal structure of magnetite. The **A** iron sites with tetrahedral (in the spinel structure) or prismatic (in the post-spinel structure) oxygen coordination and **B** iron sites with the octahedral oxygen coordination are shown—reproduced from [[36\]](#page-31-14)

the sites with octahedral oxygen coordination (B). The iron spins at the A and B sub-lattices arranges ferromagnetically below the Néel temperature [[36\]](#page-31-14). As mentioned by Narang and Pubby [\[37](#page-31-15)] the spinel ferrites have 8 divalent metal cations (Fe²⁺ for magnetite) at 8 of the available 16B-sites and 16 trivalent ferric cations ($Fe³⁺$) at 8 A-sites and the remaining 8B-sites as per formula $[Fe^{3+} \downarrow]_A [Fe^{2+} \uparrow Fe^{3+} \uparrow]_B O_4^{2-}$, where \downarrow and \uparrow are the opposite spin directions of A and B sites. In magnetite case, $Fe³⁺$ ions acts both as network formers and network modifiers.

In magnetite, the ferrimagnetic moment appears due to the unpaired spins of $Fe²⁺$ in octahedral coordination. Concerning the $Fe³⁺$ ions, they are coupled antiferromagnetically via the oxygen atom, resulting in a net zero magnetization in the Fe³⁺ sub lattice. Octahedrally coordinated Fe³⁺ and Fe²⁺ ions are coupled

ferromagnetically through a double exchange mechanism, with an electron whose spin is directed in the opposite direction of the others that can be exchanged between two octahedral coordination sites [[38\]](#page-31-17). The physical properties of magnetite are presented in Table [1](#page-3-0).

Maghemite is formed by oxidation of magnetite. Maghemite has a similar cubic structure to magnetite, a spinel-inverted type structure, and presents unit cells with a vacancy of 2.67 cationic iron atoms located in octahedral sites. The chemical repre-sentation of magnetite and maghemite may be indicated as follows [[39,](#page-31-16) [40\]](#page-32-0):

Coprecipitation vs. other methods

Coprecipitation is the traditional method for producing $Fe₃O₄$ [\[41](#page-32-1), [42](#page-32-2)] involving the combination of ferric and ferrous ions in extremely basic solutions in a 1:2 molar ratio. Surfactants like dextran or polyvinyl alcohol can be added to the reaction media, or the particles can be coated in a subsequent stage, to produce welldispersed magnetic nanoparticles [[3\]](#page-30-14). Surfactants protect colloidal dispersions by regulating particle size and stabilizing it. However, because of the large particle size distribution, secondary size selection techniques are needed. Even while aqueous synthesis approaches for the preparation of magnetite nanoparticles have improved over the past ten years, they is still one significant drawback, the difficulty to properly control particle size [[43\]](#page-32-3).

Properties		
Molecular formula	Fe ₃ O ₄	
Color	Black	
Crystal structure	Cubic	
Molecular structure Inverse spinel		
Cell dimension/Lattice parameters (nm)	$a = 0.8396$	
Formula units, per unit cell, Z	8	
Density (g/cm^3)	5.18	
Mohs hardness	5.5	
Type of magnetism	Ferrimagnetic (bulk) super- paramagnetic (nanopar- ticles)	
Saturation magnetization, Ms (emu/g) at 300 K	$92 - 100$	
Band gap energy (eV)	2.6	
Curie temperature (K)	858	
Standard free energy of formation, ΔG_{f}^{0} (kJ/mol)	-1012.6	
Melting point $(^{\circ}C)$	1583-1597	
Boiling point $(^{\circ}C)$	2623	

Table 1 The properties of magnetite [\[29](#page-31-8), [145](#page-37-0)]

Thermal decomposition syntheses are currently viewed as the most popular way to obtain magnetic nanoparticles, and alternatives to other approaches. The production of the magnetic magnetite/maghemite phases, on the other hand, happens later owing to oxidation upon exposure to air oxygen, according to recent research [[44\]](#page-32-4). This suggests that the thermal decomposition routes may begin with the formation of a non-magnetic Wüstite phase. Additionally, thorough research using high-resolution transmission electron microscopy points to the presence of faws and polycrystalline particles, which could ultimately infuence the magnetic and functional properties of the nanoparticles. Due to this aspect, post-synthesis thermal oxidative treatments have been used to enhance the magnetic characteristics of the nanoparticles. These treatments work well for small particles (15 nm), but when particle size increases (above 20 nm), these approaches lose their efectiveness or require extremely long $(>30 \text{ h})$ processing times [[19\]](#page-30-13).

Precipitation in microemulsion has been proved as a potential technique for producing magnetic nanoparticles with improved control over size and morphology. A microemulsion is a stable isotropic dispersion of two immiscible liquids that is stabilized by an interfacial coating of surface-active molecules. It consists of nanosized domains of one liquid in the other. The confnement efect created by the surfactant molecules restricts the growth, nucleation, and agglomeration of particles. In this process, tiny water droplets (referred to as "water pools") that are uniformly dispersed in an oil phase co-precipitate with surfactant molecules. The water-to-surfactant molar ratio thermodynamically determines the size of these "water pools," also known as reverse micelles, which serve as micro reactors for the synthesis of nanoparticles [\[20](#page-31-0)].

Hydrothermal synthesis is one of the most popular methods for creating magnetic nanoparticles. As opposed to conventional crystal-making synthesis, which involves high temperatures and pressures, this technique produces single-crystal nanoparticles by depending on the solubility of an aqueous solution (water). It is essentially a synthesis approach based on solution reactions. From room temperature to extremely high temperatures (130–250 °C), hydrothermal synthesis of magnetic nanoparticles can take place at a variety of temperatures. It is possible to use the various vapor pressures in the reaction, either in low-pressure or high-pressure conditions (0 and 3–4 MPa), to control the morphology of the materials to be synthesized, resulting in the material crystallizing in a sealed container and greatly improving the preferential growth of diferent planes of the MNPs. However, compared to other, more straightforward, less expensive synthesis processes, the high pressure and temperatures sacrifce the magnetic properties of the nanoparticles [[45\]](#page-32-5).

The wet-chemical approach's classic and widely utilized sol–gel synthesis process is mostly employed to fabricate nanometric size materials (usually metal oxides) [\[46](#page-32-6), [47\]](#page-32-7). A colloidal solution typically acts as the foundation for an interconnected network of separated particles. A gel, a continuous three-dimensional network that encloses the colloidal liquid phase, and a sol phase, a stable dispersion of colloidal particles in a solvent, shape the system. In a colloidal gel, the network is created by the collection of colloidal particles. A sub-colloidal aggregate's deposition on the particles in a polymer gel creates a polymeric sub-structure for those particles.

Due to hydrophilic ligands on the surface of magnetic nanoparticles, which are used in the surface coating and for high control over size and shape, the sol–gel reaction has defnite advantages over the prior wet-chemical or aqueous synthesis method, such as the simple dispersion in solvents. However, the low yield compared to other approaches, like coprecipitation, the expensive costs of the reagent precursors (i.e., alkoxides), and the difficulty obtaining magnetite structure limit the convenience of this technique [\[45](#page-32-5)].

Among the traditional techniques for producing magnetic nanoparticles, electrochemical synthesis is starting to be promising. One beneft of this approach is the ability to regulate particle size by altering the electrooxidation system's imposed current density (i) or potential (E) [[48,](#page-32-8) [49\]](#page-32-9). Additionally, it is possible to prevent particle aggregation if the synthesis is carried out in the presence of a surfactant. Using this technique, Cabrera et al. [[23\]](#page-31-3) produced $Fe₃O₄$ nanoparticles between 20 and 30 nm in size. Using standard techniques, it is impossible to obtain this size. Co-precipitation, for instance, produces smaller particles in the 5–12 nm range, whereas hydrolysis or sol–gel techniques can produce larger particles, but they can never be larger than 20 nm, and most of the time, they are difficult to suspend in an aqueous colloidal solution.

This promotes the quest for a method of water-based synthesis where the particle size might be around 20 nm or greater, which has been suggested as being excellent for heating and separation processes [\[50](#page-32-10)]. The produced particles, on the other hand, would be hydrophilic because the electrochemical synthesis of $Fe₃O₄$ is carried out in aqueous solution; as a result, their surface might be easily altered by exchange reactions with biomolecules of interest.

Magnetite nanoparticles synthesis by coprecipitation method

Mechanism

The synthesis of magnetite by coprecipitation is the most widely used method but the mechanism of particle formation is still not sufficiently explained, due to a lack of information on how intermediates are formed which may limit the reproducibility of the coprecipitation reaction. A comprehensive understanding of the mechanism is important not only for consistent synthesis, but also for adjusting particle properties for specifc applications.

The overall chemical reaction of $Fe₃O₄$ precipitation is given in Eq. [1](#page-6-0) [[51\]](#page-32-11). The reaction is performed by using aqueous solutions of ferric (Fe^{3+}) and ferrous ions $(Fe²⁺)$ in an alkaline medium. The main steps of the mechanism consist of nucleation of iron oxide particles and post growth of the nuclei. The number of nucleus formed in the reaction and the size of the particles are strongly dependent on the experimental conditions, wherein the pH of the solution is considered to best control the nucleation and growth of the magnetite nanoparticles and can infuence the magnetite properties, e.g., particle size and size distribution or saturation magnetization [\[52](#page-32-12)].

$$
Fe^{2+} + 2Fe^{3+} + 8OH^{-} \rightarrow Fe_3O_4 + 4H_2O
$$
 (1)

The possible reaction mechanism of iron oxide nanoparticles formation is shown in the following chemical reactions (Eqs. $2-5$) [\[52](#page-32-12)]:

$$
\text{Fe}^{3+} + 3\text{OH}^- \rightarrow \text{Fe(OH)}_3\text{(s)}\tag{2}
$$

$$
Fe(OH)_3(s) \rightarrow FeOOH(s) + H_2O
$$
 (3)

$$
\text{Fe}^{2+} + 2\text{OH}^- \rightarrow \text{Fe}(\text{OH})_2\text{(s)}\tag{4}
$$

$$
2FeOOH(s) + Fe(OH)2(s) \rightarrow Fe3O4(s) + 2H2O
$$
 (5)

Initially, the Fe^{3+} and Fe^{2+} hydroxides are precipitated very fast, followed by the ferric hydroxide decomposition to FeOOH. Then a solid state reaction between FeOOH (s) and Fe(OH)₂ (s) takes place with the formation of magnetite $[52]$ $[52]$. The most commonly noticed impurity compounds in magnetite synthesized by the coprecipitation method are maghemite, goethite and some hematite but absence or low oxygen can limit the transformation of magnetite to other phases [\[30](#page-31-9)]. A side reaction that takes place in the presence of oxygen is oxidation of Fe^{2+} and Fe^{3+} . To overcome this problem, two approaches have been used. In the frst approach, the nitrogen (N_2) gas is bubbled into the reaction mixture or, other works presents the synthesis of magnetite nanoparticles under oxidizing environment (air atmosphere) by lowering the initial molar ratio (Fe³⁺: Fe²⁺ <2:1), so that after the oxidation of $Fe²⁺$ to $Fe³⁺$ the ratio approaches to 2:1.

The growth mechanism of iron oxide nanoparticles using $Na₂CO₃$ as a precipitating agent was studied by LaGrow et al. $[53]$ $[53]$. Na₂CO₃ was chosen as a precipitating base because of the slower growth kinetics than the kinetics when using stronger bases [\[54](#page-32-14)]. The experiments were carried out at 60 \degree C in a flow reactor and therefore the particle analysis was carried out in solution, during the synthesis, with the help of synchrotron X-Ray difraction. At pH 12, two initial phases were formed: ferrihydrite phase, formed mainly of $Fe³⁺$ (poorly crystalline) and iron hydroxide carbonate, formed primarily of Fe^{2+} (crystalline) [[53\]](#page-32-13).

Using X-ray difraction, electron microscopy, and Fourier-transform infrared spectroscopy (FTIR) Ahn et al. [[55\]](#page-32-15) showed that when the pH of the iron salt solution increased slowly and continuously, frst nucleated was akageneite, which transformed to goethite, that had a topotactic transition to magnetite (the upper route in Fig. [2\)](#page-7-0). An abrupt pH change results in diferent particle formation routes than slow base additions, showing the precipitation of ferrous hydroxide $[Fe(OH)₂]$ and its transition to several intermediate phases including lepidocrocite (the under route in Fig. [2](#page-7-0)). Magnetite nanoparticles were formed within the coprecipitation process by the phase transformation of iron oxyhydroxides, instead of the direct reaction of Fe^{2+} and Fe^{3+} in the aqueous phase [\[55](#page-32-15)].

The "LaMer model" is often used to interpret the size distributions of the synthesized iron oxide nanoparticles. According to the LaMer model, a short instantaneous

Fig. 2 Formation pathways of magnetite nanoparticles by coprecipitation method, infuenced by the base addition speed, Reprinted with permission from Ref. [\[55](#page-32-15)], J. Phys. Chem. C. (2012), 116, Copyright © American Chemical Society

nucleation followed by slow controlled growth, without further nucleation, is the key to producing colloids with a narrow size distribution [\[56](#page-32-16)]. For iron oxide nanoparticles, the model is often cited as a diagrammatic drawing and description of the model is only associated with words. A further perspective on mechanism elucidation is the experimental evidence that the LaMer model fts the experimental nucleation kinetics data of iron oxide nanoparticles.

Operating parameters

The selectivity, morphology, and particle size of magnetite in the solution is quite sensitive and depends on operating parameters such as pH, reaction temperature, iron precursors and their concentration, optional presence of stabilizers/ligands, sequence of reagents etc. Table [2](#page-8-0) summarizes the influence of several coprecipitation reaction parameters on diferent properties of magnetic particles. Hereinafter, a detailed description of each parameter is presented.

Efect of pH and precipitating agent

Magnetic iron nanoparticles can be synthesized by coprecipitation method, in alkaline conditions using diferent bases (NaOH, KOH, NH4OH etc.). According to the thermodynamics of the reaction, complete precipitation of $Fe₃O₄$ should be expected at a pH between 8 and 14, with a stoichiometric ratio of 2:1 (Fe³⁺ /Fe²⁺) in a non-oxidizing environment [[57\]](#page-32-17). A strong alkaline medium (such as NaOH, KOH, or LiOH) can lead to formation of nonmagnetic iron components, so in order to avoid this, weak bases such as $NH₄OH$ or $Na₂CO₃$ are recommended in many studies [\[58,](#page-32-18) [59](#page-32-19)]. Gnanaprakash et al. [\[58\]](#page-32-18) studied the efect of initial pH of iron salt solutions (before adding the precipitating agent) on the formation of magnetite nanoparticles. The saturation magnetization values decreased from about 58 emu/g to about 17 emu/g as the initial pH increased from 4.7 to 6.7 at room temperature of 25 °C (reaction conditions: $Fe^{2+/}Fe^{3+}$ molar ratio = 1:2; precipitating agent 25% of aqueous ammonia; final $pH = 10$; oleic acid coating), due to the formation of goethite which contributes to a drop in saturation magnetization of the magnetic particles. This occurs because of $Fe²⁺$ ions oxidation to the

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ferric state driving to an excess of Fe^{3+} ions in the system. Furthermore, if the iron precursors are frst dissolved in an acidic aqueous medium, the individual precipitation of iron hydroxides is avoided.

Sodium carbonate may be a convenient precipitating base to study the mechanism of magnetite formation because the precipitation is progressive, and it is possible to record the reaction changes. Also, by reducing the abrupt pH variations, a narrow particle size distribution is obtained [\[54](#page-32-14)]. This was also observed by Peternele et al. [\[39](#page-31-16)], in the case of the size of magnetic nanoparticles obtained using weak bases as precipitating agent (e.g. $NH₄OH$) is more uniform than those obtained with strong bases (e.g., NaOH) $[39]$ $[39]$. The effect of pH on the properties of the magnetic nanostructured particles was studied by Blanco-Andujar et al. [\[54](#page-32-14)], at pH values between 6 and 10, room temperature, stoichiometric mixture of ferrous and ferric chloride salts and using sodium carbonate as a precipitating agent. The morphology, particle size and formed phases were investigated by powder XRD and TEM and are presented in Table [3](#page-9-0). The obtained nanoparticles showed an increase in average particle size of about 10 nm per pH unit for the magnetite phase for pH values of 8, 9 and 10.

The nature of alkali, the slow or fast addition of alkaline solution and the fnal pH of the solution were studied to evaluate their infuence on the particle size of the magnetite, by Mascolo et al. [\[52](#page-32-12)]. To investigate how the nature of precipitating agent afects the particle sizes, experiments were carried out using diferent amounts of NaOH, KOH or $(C_2H_5)_4NOH$ (TEAOH) alkaline solutions. Smaller magnetite particles were obtained using KOH and TEAOH compared to NaOH. The fast addition of the base enabled the formation of smaller magnetite nanoparticles in comparison to those of the corresponding magnetite obtained by slow addition of the base. This is related to the continuous nucleation against growth [\[52\]](#page-32-12). The authors also concluded that a stoichiometric ratio of Fe^{3+}/Fe^{2+} equal to 2:1 in solid phase is best achieved at room temperature, in a range of pH values, from 10.0 to 13.0. At higher temperatures (70–80 $^{\circ}$ C), the ratio between Fe³⁺/ $Fe²⁺$ is 2/1 at pH values between 9 and 12.

	pH Particle size	Morphology	Phase
6	30 ± 4 nm long and 3.2 ± 0.4 nm Homogeneous acicular shape Goethite (a-FeOOH) wide		
7	$440 + 65$ nm 31 ± 1 nm and 4 ± 1 nm width	Nearly square nanoparticles Acicular nanoparticles	Siderite (Fe($CO3$))-primary phase Goethite-secondary phase
8	$6.9 + 0.4$ nm Forming aggregates of $50 - 70$ nm	Nearly square nanoparticles Acicular particles	Magnetite-primary phase Goethite-secondary phase Siderite-tertiary phase
9	$18 + 3$ nm Forming aggregates of 34 nm	Spheroidal nanoparticles	Magnetite
10	28 ± 5 nm without aggregates	Square nanoparticles Acicular particles	Magnetite-primary phase Goethite-secondary phase (low percentage)

Table 3 Iron oxides nanoparticles and their characteristics, obtained by coprecipitation method at different pH values, using $Na₂CO₃$ as precipitating agent. Copyright with the permission of © The Royal Society of Chemistry 2012 [[54\]](#page-32-14)

In most cases the syntheses of magnetic nanoparticles were performed at 70–80 °C or higher temperature and the size of the magnetic nanoparticles increases with reaction temperature, but there also are some studies that indicate an inconsiderable temperature dependence of magnetic nanoparticles properties.

Niu and Zheng [[60](#page-33-1)] analyzed the properties of $Fe₃O₄$ nanoparticles (size and magnetic properties) synthesized by coprecipitation, at temperatures from 60 °C to 90 °C and observed an increase of particle size with the reaction temperature. $Fe₃O₄$ nanoparticles (determined by SEM), with a narrow size distribution were obtained by coprecipitation method, at 1.5 h reaction time, a stirring speed of 200 rpm and using ammonia solution as a precipitating agent. The size of nanoparticles varied from 30 to 60 nm, in proportion to the increasing reaction temperatures. It can be seen that the particle size has almost doubled (e.g., presented in the article 31.26 nm for 60 $^{\circ}$ C and 58.06 for 90 $^{\circ}$ C). Although all the samples have typical superparamagnetic behavior, the saturation magnetization (M_s) also changes with temperature. The highest saturation magnetization was for the particles synthesized at 70 °C (58.9 emu/g), probably due to the better degree of crystallinity than the others $[60]$ $[60]$ $[60]$ and the lowest M_s was 45.6 emu/g for a synthesis temperature of 60 °C. Petcharoen and Sirivat [[61\]](#page-33-6) successfully controlled the nanoparticle size to be in the range of 10–40 nm, by varying the reaction temperature and surface modifcation. Therein, the particle size of bare magnetite also increases with temperature. The same efect was observed for hexanoic acid coated magnetite, but not for the oleic acid coated magnetite, where the particle size decrease with increasing temperature, probably due to steric hindrance of acid molecule [[61](#page-33-6)].

The infuence of temperature on magnetite nanoparticle band gap, size and morphology has been studied by Saragi et al. [[62](#page-33-2)] from room temperature to 80 $^{\circ}$ C, at pH = 10. They obtained spherical nanoparticles and using TEM and Particle Size Analyzer observed that the average size of $Fe₃O₄$ nanoparticles has an irrelevant increase with temperature. They assigned it to the acceleration of chemical reaction between iron ions. The average particle size calculated from TEM measurement was 10.14 nm, 10.32 nm, 10.95 nm and 11.66 nm for synthesis temperature of 25 °C, 40 °C, 60 °C, and 80 °C. Even if an increase in particle size with temperature has been observed, it is not very significant. Comparable results have been obtained by Meng et al. [[63](#page-33-7)], who investigated the temperature effect, among other parameters (pH, iron salts ratio and concentration, crystallization temperature) on particle size and saturation magnetization. No more than 2 nm, but, usually, a sub nanometer variation of particle size was observed in 40–70 °C temperature range. Also, the saturation magnetization was insignifcantly afected by temperature, with a variation of 3 emu/g, in the same pH medium.

At high temperatures larger particles can be produced because at elevated temperatures the growth of nucleus is easier to happen, and the nucleation rate is slow down $[64]$ $[64]$.

Efect of iron precursors and Fe3+/Fe2+ratio

Recent reports reveal that the kind of precursors infuence the mean size, shape and magnetic properties of the magnetic nanoparticles synthesized by the coprecipitation method. The most used anions of iron salts consist of chloride, sulfate, and nitrate. The growth of oxide nanoparticles in aqueous solution can be controlled by controlling the ionic strength of the reaction solution [[65\]](#page-33-9). So, by changing the iron salt, the ion strength of the solution changes.

Yazdani and Seddigh [\[66](#page-33-10)] explained the infuence of precursor's anions on nanoparticle size based on ionic strength (I) and the double layer theory and it was attributed to the size of precursor anions. They used various iron salts, Fe²⁺–FeCl₂· 4H₂O, FeSO₄·7H₂O and Fe³⁺–FeCl₃·6H₂O, Fe₂(SO₄)₃ and Fe(NO₃)₃·9H₂O precursors in varied combination, to obtain magnetite, using a NaOH solution, at 25 °C and under a nitrogen atmosphere. The increase in ionic strength results in reducing the size of the magnetic nanoparticles. For the solutions with the same ionic strength but diferent mean size of the particles the behavior was explained by the double layer theory. As the anion size of the precursor salt increases, the synthesized nanoparticles mean size decreases (Fig. [3](#page-11-0)).

Iwasaki et al. [\[67](#page-33-11)] studied the infuence of iron precursors and coexisting anions in the formation process of $Fe₃O₄$ nanoparticles (at room temperature, sodium hydroxide- as precipitating agent, pH > 12, and argon atmosphere). They used in their experiments FeCl₃ 6H₂O as Fe³⁺ ion source and FeSO₄ 7H₂O, FeCl₂ 4H₂O or Fe(C₃H₅O₃)₂ 3H₂O as Fe²⁺ ion source. The role of coexisting anions was investigated by adding sodium sulphate $(SO₄^{2−} source)$, sodium chloride $(CI[−] source)$ or sodium lactate $(C_3H_5O_3^-$ source), in water before the iron salts were dissolved. It was found that the presence of SO_4^2 coexisting ion has the effect of increasing the particle size. Contrarily, chloride and lactate contributed to size reduction of the Fe₃O_{[4](#page-12-0)}. Table 4 presents the average crystallite size of magnetite nanoparti-cles obtained from XRD analysis [\[67](#page-33-11)]. Furthermore, when FeSO₄ was used as Fe²⁺ sources, goethite particles were also observed in the fnal product.

Fig. 3 The variation of magnetite nanoparticles size with ionic size and strength of the iron precursors, Reprinted with permission from Ref. [[66\]](#page-33-10), Mater. Chem. Phys. (2016), 184 © Copyright Elsevier

Not only the type of iron precursor is important for acute control of the properties of magnetic nanoparticles, but also the ratio of Fe^{3+}/Fe^{2+} and the concentration of iron salts. Among other variables, Roth et al. [\[68](#page-33-3)] have explored the impact of Fe^{3+}/Fe^{2+} ratio and iron salts concentration in magnetization and in size of magnetic nanoparticles. They found that by modifying Fe^{3+}/Fe^{2+} the saturation magnetization (M_s) may be improved. The M_S increased from 62 to 82 A m² /kg by reducing the Fe^{3+}/Fe^{2+} molar ratio from 3 to 1.5 [\[68](#page-33-3)]. A similar influence of Fe^{3+}/Fe^{2+} ratio on M_S was reported earlier in literature [\[69](#page-33-12), [70](#page-33-13)].

Efect of stirring rate

The precipitation reactions are very fast, and the efect of mixing is signifcant down to slow mixing speed. This phenomenon is more prominent at fast mixing conditions and will depend on the fow regime in the reactor. Only a few studies designed the infuence of stirring rate on the particle size and distributions of the magnetite nanoparticles synthesized by the coprecipitation method. Thus, Valenzuela et al. [\[71](#page-33-4)] found that a stirring rate of 10,000 rpm is suitable to obtain magnetite nanoparticles with a mean diameter of about 10 nm. As expected, decreasing the stirring velocity led to larger particle size and a wider size distribution. An interesting observation is related to a higher stirring rate (25,000 rpm) leading to an increase in the solutions' core temperature during synthesis, that promotes the partial thermal transformation of magnetite to goethite [\[71](#page-33-4)]. Fadli et al. [\[72](#page-33-14)] synthesized magnetite crystallites of 10–12 nm, by coprecipitation method (Fe $^{3+}/$ Fe $^{2+}$ 2:1, 40–80 °C, pH 10 using ammonia solution as precipitating agent) at a stirring speed between 300 and 500 rpm. The higher the stirring speed was (500 rpm), the smallest magnetite crystallites were obtained (10 nm), the agglomeration of the particles became more uniform and the crystallinity of the magnetite phase increased. Other authors observed a similar efect too [[73–](#page-33-0)[75\]](#page-33-15). A higher stirring rate produces a faster seeding and thus, more seeds with smaller particles are formed compared to slower stirring rates [\[73](#page-33-0)]. The magnetite content in the iron oxide nanoparticles was only slightly affected by the stirring rate $[73]$ $[73]$. In another research study, magnetite nanoparticles with crystallite size ranging from 22 to 25 nm were reported in the case of using stirring rates between 500 and 900 rpm. Contrary to the previous studies, from

600 to 800 rpm the crystallite size was about 22 nm and for a higher stirring rate (900 rpm) the particle size increased to 25 nm [\[76](#page-33-5)].

An explanation for producing small-sized particles at high stirring speed to is that the sample obtains transport energy in the form of kinetic energy that can reduce the molecular weight of $Fe₃O₄$ and break the agglomeration so that the particle becomes small fragments [\[76](#page-33-5)]. As mentioned by Rahmawati et al. [[76\]](#page-33-5), due to the strong magnetic force, the $Fe₃O₄$ molecules can recombine to form a large particle.

Magnetite nanoparticles stabilization

Magnetic nanoparticles are very sensitive to oxidation and agglomeration, so important issues of magnetic nanoparticle are their protection against accumulation, chemical reactivity and degradation, because a change of the properties may afect and limit the use of magnetic nanoparticles [\[77](#page-33-16)]. When they are in dispersions medium, there are several possibilities to stabilize the magnetic nanoparticles: electrostatic, steric, or combined methods. Electrostatic stabilization consists in the addition of charged functionalized molecules to the particle surface or the choice of the solvent used in particle suspension. Electrostatic repulsion can be controlled through knowledge of the diffusion potential, instead steric forces are more difficult to anticipate and quantify [[78\]](#page-33-17).

Zeta potential (ζ) determination is very useful for determining the dispersion stability of nanoparticles in diferent media. In general, colloids with values for the ζ-potential in the range of 20–30 mV or higher are considered stable. When the potential is small, attractive forces may exceed this repulsion and the dispersion may break and focculate. Highly positively or negatively charged particles tend to repel each other, thus forming stable colloidal solutions which show only minor trends to agglomerate. So, dispersions with high zeta potential (negative or positive) are electrically stabilized, while those with low zeta potentials tend to aggregate [[79\]](#page-33-18).

Xiong et al. [[80\]](#page-34-0) prepared stable (trimethoxysilyl)propylmethacrylate (MPS) chemically modifed magnetite particles for over three months at room temperature. The authors measured the zeta potential the chemically modifed magnetite particles in aqueous media, at diferent pH values, using a Malvern Zetasizer Nano ZS. The zeta potentials were higher in neutral and alkaline conditions than that in acidic medium, indicating a high stability of the particles in neutral and alkaline condition. The explanation to this behavior is related to the extended polymer chains and high electrostatic repulsions on the surface of the nanoparticles.

The stability of the coated magnetite nanoparticles is recommended to be tested in a dispersing media like the application environment of the nanoparticles. For example, if magnetic nanoparticles are used in medicine, the dispersing medium for zeta potential measurements should be close to that of biological environment and the biological fuid. Due to the interaction with the biological media, magnetite nanoparticles tend to destabilize and undergo to aggregation processes, leading to the signifcant increase in their hydrodynamic size, as well as changes in nanoparticles surface properties.Yu et al. [\[81\]](#page-34-1)tested the stability of tetramethylammonium hydroxide coated iron oxide nanoparticles (T-SPIONs) and citrate coated iron oxide nanoparticles (C-SPIONs) in water and three different biological media of PBS (phosphate buffered saline), RPMI (Roswell Park Memorial Institute Medium) and DMEM (Dulbecco's Modifed Eagle's Medium). The stability was tested in terms of hydrodynamic diameter as a function of time, and zeta potential. Both SPIONs show good stability in water since the hydrodynamic diameter did not change up to 3 months. C-SPIONs showed robust stability in biological media of PBS and RPMI and destabilized in DMEM, while T-SPIONs were found to aggregate rapidly and in all tested biological media. The authors succeed to stabilize the T-SPIONs and C-SPIONs by adding BSA (bovine serum albumin) on their surface, which signifcantly enhanced their stability, with no aggregation phenomenon detected in 24 h.

To avoid oxidation, protect and stabilize the magnetite core, a wide range of coating agents can be employed to coat the magnetic particles: natural and synthetic polymers, silica, carboxyl group, noble metals, metal oxides and sulfdes and so on. Furthermore, they can also increase wetting properties and binding fexibility for various applications. Coating $Fe₃O₄$ nanoparticles with silica (SiO₂) may prevent or reduce the agglomeration of the particles, and protect them from dissolution in acidic solution, because $SiO₂$ is chemically stable in acidic solution. Furthermore, silica has tunable properties and further functionalization can be easily achieved and it has a good biocompatibility and hydrophilicity, which are indispensable for biomedical applications [\[82\]](#page-34-2). There are two methods for coating $Fe₃O₄$ with silica, by acidic hydrolysis of silicate in aqueous solutions and the modifed Stöber process. The silica shell thickness can be controlled by varying the experimental parameters: the reaction time, the ratio of TEOS/Fe₃O₄ and the concentration of Fe₃O₄ nanoparticles [[83](#page-34-3)]. Karimi Pasandi-deh et al. [\[84\]](#page-34-4)have coated $Fe₃O₄$ particles with SiO₂ to form core–shell particles that have been successfully applied in humic acid removal from water resources, in acidic conditions. Silica coated $Fe₃O₄$ particles have also been regenerated and reused several times. The carbon coating also provides an efficient oxidation protection and prevents corrosion of iron oxide [[85](#page-34-5)].Besides stability over the entire pH range, the carbon coating also enhance the biocompatibility and dispersity of the particles. Noble metals such as gold and silver facilitate further functionalization and are also known to improve stability and compatibility [\[86\]](#page-34-6). The surface of the iron oxide can be coated with various natural polymers (e.g. chitosan, gelatine, dextran) or synthetic polymers (such as polylactic acid (PLA), polyethylene glycol (PEG), polyvinyl alcohol (PVA)) avoiding aggregation and improving chemical and thermal stability, hydrophilicity and biocompatibility [\[87](#page-34-7)[–89\]](#page-34-8).

Castelló et al. [[90\]](#page-34-9) have synthesized alginate and chitosan coated magnetic nanoparticles and tested their stability by monitoring their hydrodynamic diameters as a function of time. After a period of six months the hydrodynamic diameter did not change and no focculation or settling of the particles was observed. In biological medium, alginate magnetic nanoparticles were stable for more than 9 days, while chitosan magnetic nanoparticles were stable for only 2 days.

Magnetite nanoparticles characterization

The characterization of any synthesized material is necessary to understand the scientifc reason of its properties. A great review dealing with practices for characterization of magnetic nanoparticles for biomedical applications was written by Sandleret al. [\[91\]](#page-34-10) who offered a simple explanation and background for preferred practices in the characterization methods and technologies associated with the magnetic nanoparticle feld. The most common analytical techniques for the characterization of magnetic nanoparticles, but not limited to, with the advantages and restrictions collected from the literature, are summarized in Table [5.](#page-16-0)

Fourier transform infrared spectroscopy (FTIR) is a technique based on the measurement of the absorption of electromagnetic radiation with wavelengths within the mid-infrared region $(4000-400 \text{ cm}^{-1})$.

FTIR analysis is an excellent technique to identify the functionalization of magnetite, but it can also be a useful technique to diferentiate between diferent iron oxides. The absorption band at a high wavenumber region is due to OH stretching, and at lower wavenumber as a result of Fe–O lattice vibration. Namduri and Nasrazadani [\[92\]](#page-34-11) established a quantitative analysis of iron oxides using FTIR. Binary mixtures of iron oxides were prepared with known compositions containing pure commercial magnetite, maghemite and hematite for calibration purposes. The authors selected the bands at 570 cm⁻¹ (for magnetite), 540 cm⁻¹ (for hematite) and 630 cm^{-1} (for maghemite) to set the calibration curve for known amounts of iron oxide. Table [6](#page-17-0) summarizes FTIR absorption bands for different iron oxides, selected from the literature.

The presence of silica layer on the surface of magnetite was demonstrated using FTIR characterization by the vibration bands of Si–O bonds and silanol groups at wavenumbers around 800 and 1000–1100 cm^{-1} , bands that increased with silica loading. These bands were shifted from original, to smaller wavenumbers, indicating the interaction of siloxane with hydroxyl functionality of the magnetite nanoparticles, forming bonds of the type of Si–OH–Fe. Examples of the FTIR spectra of functionalized magnetite nanoparticles with diferent amounts of silica are shown in Fig. [4](#page-17-1) [[93](#page-34-12)]

Another category of widely used materials for magnetite functionalization are the polymers and the monomers because they can increase the stability of the nanoparticles and protect them in vitro and in vivo. The most used polymers or monomers for magnetite nanoparticles are polyethylene glycol (PEG) [[94](#page-34-13)], glucose, dextran [[95](#page-34-14)], chitosan [[96\]](#page-34-15), alginate [[97](#page-34-16)], polyvinyl alcohol [\[98\]](#page-34-17). Sathyanarayanan and Raina [[99](#page-35-0)] investigated the coating of magnetite surface with polymers like PEG and chitosan. Using FTIR, they demonstrated the absorption of PEG and chitosan on magnetite nanoparticles through absorption bands of C–H methyl stretch (2956 cm⁻¹), amide stretch (1641 cm⁻¹ and 1371 cm⁻¹), C–C and C–O asymmetric and symmetric stretch (1087 cm−1, 1041 cm−1). The electrostatic interaction between the negatively charged obtained magnetite and positively charged chitosan (due to the amine groups) gave a minor shift of the peaks compared to the pure material.

Iron oxide/hydroxide	Wavenumber $(cm-1)$	References			
Magnetite (Fe ₃ O ₄)	Broad bands at 570–590 and 360–400	$[158 - 160]$			
Maghemite (γ -Fe ₂ O ₃)	700, 630–660, 620, 430 range (Fe–O range)	[161, 162]			
Hematite (α -Fe ₂ O ₃)	540, 470 and 352	[163, 164]			
Goethite (α -FeOOH)	1124, 890 and 795–810 cm ⁻¹ for OH stretch, 635	$[164]$ [159, 163, 165]			
Lepidocrocite $(\gamma$ -FeOOH)	1020 cm^{-1} (in plane vibration) and 750 cm ⁻¹ (out of plane vibration), 620	[159, 166]			

Table 6 Infrared bands of diferent iron oxides and hydroxides

Fig. 4 FTIR spectra of: (a) Fe_3O_4 ; (b-h) Fe_3O_4 @SiO₂ with different amounts of SiO₂ coatings; and (i) SiO₂, Reprinted with permission from Ref. [\[93](#page-34-12)], Surf. Interfaces (2019), 14 © Copyright Elsevier

Ma et al. [[100](#page-35-1)] have prepared alginate-coated magnetic particles and the binding of magnetite to alginate was emphasized by FTIR analysis by characteristic carboxyl group adsorption bands (from β-(1–4)-p-mannuronic and α -L-gluronic acid units) around 1610 cm^{-1} and 1410 cm^{-1} . Compared to alginate sample the bands were shifted to lower frequencies indicating the interaction between iron oxide and alginate. Magnetic solid acid catalyst, designed as 12-tungstophosphoric acid loaded on the surface of mesoporous silica supported on magnetite core (Fe₃O4@SBA-15@HPW) was synthesized by Zhang et al. [[101](#page-35-2)], as a catalyst for biodiesel production. The successful coating of magnetite was investigated for the Keggin structure of HPW in the FTIR spectra, through the presence of the stretching frequency of P–O in the central PO₄ tetrahedron (1120 cm⁻¹), (terminal bands for W=O in the exterior WO6 octahedron (1000 cm^{-1}) and bands for the W–Ob–W and W–Oc–W bridge (879 and 800 cm⁻¹) [[101](#page-35-2)].

Table [7](#page-18-0) summarizes wavenumbers, characteristics and functional group of selected groups present on the surface of functionalized magnetite.

Wavenumber (cm^{-1})	Functional group vibrations	Characteristics	Refs.
570-580	$Me-O$ (Fe-O)	Crystalline lattice of Fe_3O_4	$[167 - 170]$
1095 and 1193	$S=O$	$SO3H$ groups	[171]
800, 1000-1100	Si-O and Si-O-Si	Silica	[101, 172]
1120	$P-O$	PO ₄	[101]
1030-1330	$C-O$	Alcohol group	[173]
1080	$C-O-C$	Glycols groups	$[173]$
1400-1500; 1600	$C = C$	Aromatic ring	[172, 174]
1550-1650	$C=N$		[174]
1650-1750	$C=O$	Carbonyl group, carboxylic acid, esters	[112, 175]
2850 and 2950	$C-H$	Hydrocarbon chains	[173, 174]
1630 broad 3200-3600 broad	OH bending OH stretching	Adsorbed water molecules	[112]
3300-3500	$N-H$	Amines	[176]

Table 7 Selected vibrational assignments and characteristics for magnetite and some of the most common groups present on the surface of functionalized magnetite

X-ray difraction (XRD) is one of the most extensively used nondestructive techniques for the characterization of materials that provides information regarding the crystalline structure, lattice parameters, chemical composition, physical properties of materials and crystallite sizes [[102](#page-35-3)]. The information about crystal structure and quantitative phase analyses can be obtained based on the peak intensity and the peak shape gives information about micro strains and crystallite size. If the width of the peaks in XRD patterns is narrow the particles are nano-sized [\[103\]](#page-35-4). Diffraction data, usually presented as intensity distribution as a function of the 2θ angle, can give several useful information presented in Fig. [5.](#page-19-0)

The crystallite sizes parameter is estimated using the Scherrer equation (Eq. [6](#page-18-1)) from the broadening of the most intense peak of an XRD measurement for a specifc sample [[104](#page-35-5)].

$$
FWHM(2\theta) = \frac{b \cdot \lambda}{D \cdot \cos \theta} \tag{6}
$$

The value of the dimension of the crystallite considered as cubes (D) in Eq. [6](#page-18-1) can be calculated by Eq. [7](#page-18-2):

$$
D = \frac{2 \cdot b \cdot \pi}{FWHM(Q)}\tag{7}
$$

Here FWHM is the full width of the peak, *2θ* is the scattering angle in radians, λ is the wavelength, Q is the magnitude of the scattering vector, b is a constant, which normally takes a value between 0.89 and 0.94 depending on the function

used to fit the peak and D is the dimension of the crystallites as if they were cubes, monodisperse in size. Therefore, for diferent refections and diferent crystallite shapes, the interpretation of D must be treated with some care. For polydisperse systems, D is a measure of the volume-averaged crystallite size, which can be determined from Eq. [8](#page-19-1):

$$
d = \frac{K \cdot \lambda}{B \cdot \cos \theta} \tag{8}
$$

Here D is the crystallite average size; K is the shape factor; λ is the x-ray wavelength; B is the instrument broadening (FWHM—full width at half maximum); θ is the theta, half of the Bragg angle (radians).

For magnetite, the crystallite size of the samples is determined by Scherrer equation from the measured width of the 311 peak, which is the most intense one $[105]$ $[105]$. The composition of the particles can be determined by comparing the position and intensity of the peaks with the reference patterns available from database: ICDD (International Centre for Difraction Data), previously JCPDS (Joint Committee on Powder Difraction Standards).

Alfredo Reyes Villegas et al. [\[25](#page-31-5)] determined the average crystallite size of magnetite nanoparticles by application of the Scherrer equation to the refection 311 peak with $2\theta \sim 35.5$ and it was found to be in the range of 16 nm [[25\]](#page-31-5). The reference pattern for synthesized nanoparticles was the JCDPS Card 19-0629. The crystallinity of the magnetic nanoparticles is assigned to strong and sharp peaks, while the broadening indicates the nanosized of the particles. Similarly, Mascolo et al. [\[52](#page-32-12)] noticed that in the XRD patterns, the peak positions shift slightly forward to larger 2θ values with the increasing pH of the synthesis solution. The (311) XRD peak of

Fig. 5 Graphical representation of difraction data and information that can be extracted, Reprinted with permission from Ref. [[150\]](#page-37-11), *Materials Characterization Using Nondestructive Evaluation (NDE) Methods* (2016) pp. 81–124 © Copyright Elsevier

the products showed the peak shift, indicating that the lattice parameter contracts as the particle size decreases.

For magnetite, the refection planes with the respective difraction angles, relative peak intensities and interplanar spacing are shown in Table [8](#page-20-0) (according to Shi et al. [\[106](#page-35-8)]).

An example of XRD pattern of $Fe₃O₄$ nanoparticles obtained by Predescu et al. [\[107](#page-35-9)] is shown in Fig. [6](#page-21-0), where the several well-defined reflections can be observed.

The diferentiation between magnetite and maghemite on the nanoscale using XRD is difcult because both iron oxides have cubic structure, and their lattice parameters are almost identical. Minor diferences in the difractogram of maghemite exist in the form of a few extra peaks at 210 and 213 (weak intensities) which can possibly be used to distinguish magnetite from maghemite [[108\]](#page-35-10). A method proposed by Kim et al. [[109\]](#page-35-11) can be used to identify and quantify mixtures of magnetite-maghemite by scanning the 511 peak (Fig. [7](#page-22-0)) and resolved through the deconvolution routine. Another possible way to diferentiate between magnetite and maghemite is Raman spectroscopy [[73\]](#page-33-0) or UV–Vis spectroscopy [\[110](#page-35-12)].

Thermal gravimetric analysis (TGA) can provide information concerning the mass and composition of the surface modifers of the magnetic nanoparticles. With this technique, the sample is heated and the changes in weight are correlated with the sample composition and stability. TGA analysis is extensively utilized to measure the thermal stability, composition, moisture, oxidative stability, and volatile content, kinetics of decomposition, sample dehydration and lifetime.

Soh et al. [[111\]](#page-35-13) recorded a TG curve of $Fe₃O₄$ from 25 to 950 °C, under a nitrogen atmosphere and attributed the three mass losses to (1) the absorbed water molecules (25 °C–200 °C), (2) the physisorption and chemisorption processes occurring at the surface of the magnetite (250 °C–400 °C) and (3) phase transition from magnetite to hematite (500 $^{\circ}$ C–750 $^{\circ}$ C). If the thermogravimetric analysis is carried out in air atmosphere, a weight increase assigned to the oxidation of $Fe₃O₄$ to $Fe₂O₃$ can be observed from TG curve [\[112](#page-35-6), [113\]](#page-35-14). The weight increase was observed by Liang and Lu $[112]$ $[112]$ between 140 and 210 °C.

Fig. 6 Example of a XRD pattern for Fe₃O₄ nanoparticles, Reprinted with permission from Ref. [[107\]](#page-35-9), Powder Technol (2019), 432 © Copyright Elsevier

Thermogravimetric analysis is essential for the determination of the weight of surfactant covering the particle surface. For instance, magnetite nanoparticles with oleic acid and amphiphilic block copolymers adsorbed on their surface were investigated with a thermogravimetric analyzer. The grafting density of molecules on the surface of the nanoparticles was studied by TGA under the nitrogen atmosphere, with a heating rate of 20 $^{\circ}$ C/min from 30 to 800 $^{\circ}$ C [[114](#page-35-15)]. The thermal stability of magnetic materials coated by blends of modifed chitosan and poly(quaternary ammonium) salt has been investigated by Ziegler–Ziegler–Borowska et al. [[115](#page-35-16)] using thermogravimetric analysis in air and nitrogen atmosphere, from 20 to 800 °C. In nitrogen atmosphere, TG curve of chitosan showed two degradation moments at 20 $^{\circ}$ C–160 $^{\circ}$ C and 160 $^{\circ}$ C–400 $^{\circ}$ C ranges attributed to the evolution of water adsorbed by polymer in the frst stage, and by the rupture of polysaccharide chains (including dehydration, deamination, deacetylation, breaking of glycoside bonds, and pyranose ring opening), vaporization, and elimination of degradation products, in the second stage. The chemical modifcation of chitosan was also confrmed by diferences in the TG curves of chitosan and modifed chitosan [\[115\]](#page-35-16). Similarly, Lesiak et al. [[116\]](#page-35-17) studied the stability of various magnetite nanoparticles functionalized with carboxyl groups (for example succinic acid, oxalic acid, citric acid) and amino acid (glutamic acid, L-arginine). TG curves were recorded in the temperature range of 30 $^{\circ}$ C–600 $^{\circ}$ C.

The carboxyl, hydroxyl, carbonyl, and nitrogen groups of chemically adsorbed molecules undergo decomposition, as it was observed in the second temperature region of 150 \degree C–600 \degree C.

Dynamic light scattering (DLS) is a widely employed technique to fnd the size of nanoparticles in colloidal suspensions typically in the sub micrometer range. The size that is measured in DLS is the hydrodynamic diameter and it refers to how a particle difuses within a fuid. It is calculated using the Stokes–Einstein equation in Eq. [\(4](#page-6-3)) [[117\]](#page-35-18):

$$
d(H) = \frac{k \cdot T}{3 \cdot \pi \cdot \eta \cdot D} \tag{9}
$$

Here: $d(H)$ is the hydrodynamic diameter, D is the translational diffusion coefficient, k is the Boltzmann's constant, T is the absolute temperature and η is the viscosity.

The main factors that afect the difusion speed of particles are ionic strength of medium, surface structure and the shape of the particles. A low conductivity medium will produce an extended double layer of ions around the particle, reducing

the difusion speed and resulting in a larger, apparent hydrodynamic diameter. When measuring particle size using DLS, concentration is also an important parameter to consider because it infuences scattering of the light and may give inaccurate results. A good standardized method which highlights the sample preparation and analysis is presented in *ISO 22412:2017(E) Particle size analysis — Dynamic light scattering (DLS)* [\[118](#page-36-0)]. For an analytical review on the characterization of magnetic particles by DLS, researchers are encouraged to read the paper by Lim et al. [[119\]](#page-36-1), who have reviewed the infuence of the parameters (suspension concentration, particle shape, colloidal stability and surface coating) on the size value of magnetic nanoparticles obtained by DLS measurements.

Measurement of hydrodynamic diameter is essential in describing particle aggregation, colloidal stability of the particles in various environments and the degree of surface modifcation of the magnetic nanoparticles. For instance, Arévalo-Cid et al. $[120]$ $[120]$ examined the stability of chitosan-coated $Fe₃O₄$ nanoparticles over time. Coating was modifed by varying the amount of the crosslinker glutaraldehyde and the reaction time between magnetite and chitosan. For DLS measurements they used 0.01 g/l water suspensions of Fe₃O₄ and Fe₃O₄@chitosan crosslinked with different amounts of glutaraldehyde and they observed that the hydrodynamic size of bare magnetite nanoparticles increased over time due to the aggregation of the particles. The aggregation process was also observed for the $Fe₃O₄$ @chitosan samples prepared with small amounts of glutaraldehyde (0.5 ml). Increasing the amount of glutaraldehyde, smaller size of the aggregates was initially observed explained by a better dispersion in water of the samples, as a result of a thicker formed chitosan coating. Even if the chitosan presence improves the initial dispersion of the particles in water, it does not prevent the aggregation of constituent particles over time. [[120\]](#page-36-2). The aggregation of commercially available magnetite nanoparticles (Skyspring Nanomaterials) using DLS was determined as a function of pH, in a pH range of 2–12. Magnetite nanoparticles were dispersed in MilliQ fltered water and the pH was adjusted to the wanted value, using nitric acid or potassium hydroxide. Results indicated that as the pH value increases, the efective diameter increases and the stability decreases. Furthermore, particle aggregation occurred at pH 7.5, the isoelectric point of magnetite [\[121](#page-36-3)].

Magnetic nanoparticles coated with a layer of surfactants (e.g.) prevent their agglomeration and keep them stable, but also prevent the particles from not dispersing in aqueous solutions, an important property because for biomedical applications nanoparticles must be water soluble [[122\]](#page-36-4). Gonzales and Krishnan [\[123](#page-36-5)] succeeded to make iron oxide nanoparticles soluble in aqueous solutions by coating them with copolymer Pluronic F127. DLS measurements were used to analyze the hydrodynamic diameter of the oleic acid-magnetite nanoparticles before and after the coating and thus, the success of the magnetic nanoparticles transferred in water. The hydrodynamic diameter of the nanoparticles before coating with Pluronic F 127 was analyzed in toluene, because the oleic acid-magnetite nanoparticles were soluble in toluene. After Pluronic F 127 coating, the magnetite nanoparticles were transferred to the water phase and the hydrodynamic diameter was measured in aqueous medium.

X-ray photoelectron spectroscopy (XPS) is a powerful quantitative technique, useful to elucidate the electronic structure, elemental composition, and oxidation states of elements in a material. The main drawbacks of XPS technique are the sample preparation and data collection and interpretation. Baer [\[124](#page-36-6)] elaborated a short comprehensive guide with good practices on XPS measurement objectives, sample preparation and analysis approaches when characterizing nanoparticles.

The XPS analysis of mixed iron oxide systems are complex because of peak asymmetries, multiple splitting, shake-up satellite structure and overlapping binding energies [\[125](#page-36-7)] and should include corroborating evidence from O 1 s analysis and other analytical techniques. Grosvenor et al. [[126\]](#page-36-8) paper is an excellent starting point for qualitative ferrous (Fe²⁺) and ferric (Fe³⁺) compounds determination. An XPS analysis of the chemical states of Fe oxides and hydroxides was made by Biesinger et al. [\[127](#page-36-9)] and the ftting Fe 2p3/2 parameters in magnetite are presented in Table [9.](#page-25-0)

A sample with more than three species cannot be accurately ftted. For example, the resulting fit for a mixture of iron oxides components, $Fe₃O₄$ and $Fe₂O₃$, is shown in Fig. [8.](#page-26-0) Because the higher binding energy satellite peaks for $Fe₂O₃$ are omitted from the fitting, as are the assumed satellites for the $Fe₃O₄$ components, it causes a slight overestimation of the metal compared to the two oxide components [\[127](#page-36-9)].

The XPS technique can also analyze the ligand exchange interactions and surface functionalization of magnetic nanoparticles. The mechanism of arsenate adsorption on a magnetite-doped activated carbon fber was investigated by Zhang et al. [\[128](#page-36-10)] using the XPS technique, by tracking the change in core level peak position of the adsorbents, thus deducing the direction of electron transfer during adsorption. In another report, magnetite nanoparticles coated with L-ascorbic acid (LAA) were investigated by FTIR and XPS to confrm that LAA is capped on the surface of magnetite nanoparticles $[129]$ $[129]$. The presence of C=O of LAA capped on the surface of magnetite nanoparticles was proved from XPS spectra of the O1 peak (located at 531.8 eV) and from XPS spectrum of Fe2p which had higher binding energy peaks than the reference $Fe₃O₄$ without LAA. Furthermore, the XPS technique can also distinguish between core–shell and homogeneous alloy structures. Many studies have demonstrated that if particles possessed a core–shell system, the core element would be screened by the shell element, and thus, the compositions in the shell layer become gradually more dominant. The chemical information from the particles surface analyzed by XPS can be used to assess the thickness of the coatings [\[130](#page-36-12), [131](#page-36-13)].

Scanning electron microscopy (SEM) is excellent for nanoparticles shape and size characterization, and it is one of the most used techniques for this purpose. Sample preparation in SEM analysis is very important because it can result in artefacts that can infuence the quality of the results. The ISO 19749:2021—*Nanotechnologies — Measurements of particle size and shape distributions by scanning electron microscopy* [\[132](#page-36-14)] provide useful information with examples about nanoparticles morphology and size characterization by SEM technique.

In early years SEM analysis could not distinguish between iron oxide phases (e.g. magnetite, maghemite, hematite or goethite) due to insufficient precision of conven-tional EDS [\[133](#page-36-15)]. Innovative SEMs can analyze different iron oxides if a sufficient number of counts is acquired. John Konopka could diferentiate between magnetite,

Fe 2+ 708.4 1.2 16.6 709.2 1.2 14.8 – – – Fe 3+ 710.2 1.4 23.7 711.2 1.4 17.8 712.3 1.4 12.2

 709.2
 711.2

 16.6
23.7

 $\frac{1}{2}$ $\frac{4}{4}$

 708.4
 710.2

Fe $\overset{2+}{1}$ Fe $\overset{3+}{1}$

 1.4

 12.2 $\overline{1}$

 $\overline{14}$ \mathbf{I}

712.3 $\overline{1}$

 14.8
17.8

Fig. 8 Curve ftted Fe 2p spectrum of a mixed metal/oxide system of Fe based nanoparticles, Reprinted with permission from Ref. [\[127](#page-36-9)], Appl. Surf. Sci (2011), 257 © Copyright Elsevier

hematite and goethite with 350 k counts in the Fe peak.[[134\]](#page-36-16), but this can apply for larger particles in μm domain [[135\]](#page-36-17).

Transmission electron microscopy (TEM) is another helping tool for magnetic nanoparticles morphological characterization and size and size distribution determination in order to defne the unique set of physical properties such as magnetic, optical, catalytic and so on (measured/analyzed with other techniques) [[136,](#page-36-18) [137](#page-36-19)]. Like SEM analysis, the ISO 21363:2020—*Nanotechnologies—Measurements of particle size and shape distributions by transmission electron microscopy* [[138\]](#page-36-20) provide useful information on capturing, measuring and analyzing transmission electron microscopy images to obtain particle size and shape distributions in the nanoscale range. Compared to SEM analysis, with the help of TEM we can get information about the core structure too, and not only about the external structure, which is very important because the magnetic nanoparticles may have diferent behavior for varied core and coating dimensions [[91\]](#page-34-10).

Deng et al. [[139](#page-37-12)] used TEM analysis to evaluate how the solvent used in preparation of silica coated magnetic nanoparticles (alcohol type: methanol, ethanol i-propanol and n–propanol) infuences the coating, size and morphology of silica coated magnetite particles. Based on TEM results, they concluded that ethanol is the best candidate alcohol to obtain uniform silica coated magnetite particles and they associated thus to the polarity of the alcohol. The decrease of polarity of alcohol, (from methanol to ethanol to i-propanol to n-propanol) conducted to larger clusters of more irregular particles. TEM is also capable of distinguishing between core and shell(s). The formation

of core–shell nanoparticle may be observed from the diferences in contrast to typical TEM micrograph. ZnO layer of about 0.25 nm, of spherical core–shell Fe₃O₄@ZnO nanoparticle was observed and measured using TEM by Gupta et al. [[140](#page-37-13)]. Similar results was also observed by Faaliyan et al. $[141]$ $[141]$ in Fe₃O₄@SiO₂ core–shell nanoparticles with the mean size of the cores of about 20 nm and the shell thickness, 2.5 nm. Organic surface functionalization compounds, such as polystyrene (PS) or polyacrylic acid (PAA) could also be observed in TEM micrograph [\[106\]](#page-35-8). Fig. [9](#page-27-0) shows a typical TEM micrograph of a core–shell $Fe₃O₄@SiO₂$ (nonporous) $@SiO₂$ (mesoporous) nanoparticle, where the black core is $Fe₃O₄$, the dark grey shell is nonporous SiO₂ layer $(nSiO₂)$ and the lighter shell is the mesoporous $SiO₂$ layer $(mSiO₂)$ [\[142\]](#page-37-15).

Magnetometry techniques with the help of vibrating sample magnetometer *(VSM)* () or superconducting quantum interference device *(SQUID)* () are the most used to measure the magnetic behavior and they give quantifable results on the magnetic properties of a material. The measured properties are the magnetization (M) as a function of applied feld (H), temperature and time.

The saturation magnetization (Ms) of bulk magnetite at room temperature is 92–100 emu/g. The saturation magnetization of magnetite nanoparticles is smaller than the bulk magnetite value and it decreases with the reduction of the particle size. From the analyzed literature it was found that iron oxide nanoparticles often display superparamagnetism.

Superparamagnetism is one type of magnetism that occurs in ferromagnetic and ferrimagnetic nanostructures with size range of 20 nm or less. Due to the presence of single domain state the nanoparticles behave magnetically only under the infuence of an external magnetic feld and are rendered inactive once removed- they retain no remnant magnetization upon the removal of the external magnetic feld [\[143,](#page-37-16) [144\]](#page-37-17), The reduction in particle size of magnetite and maghemite, from bulk to nanoscale, reduces the

quantity of exchange-coupled spins that rebels magnetic reorientation spontaneously, resulting in superparamagnetic behaviour [[145](#page-37-0)].

Maity et al. [\[146\]](#page-37-18) used VSM and SQUID magnetometry to carry out the magnetic measurements of a stable aqueous magnetite suspension based on $Fe₃O₄$ functionalized with tri(ethylene glycol)-TREG. Their magnetization results showed a saturation magnetization of 65 emu/g and a superparamagnetic behavior of the magnetite nanoparticles indicated by zero coercivity and zero remanence on the magnetization curves. The blocking temperature of the particles was determined to be 100 K, below this temperature the particles are ferromagnetic. Yau and colleagues prepared dispersions of magnetite covered with l-ascorbic acid and the magnetic behavior was demonstrated by VSM. The saturation magnetization (Ms) of magnetite particles with L -ascorbic acid was smaller than that of bare magnetite, but the particles maintained their superparamagnetism property as evidenced by the lack of hysteresis loop [\[129](#page-36-11)]. Generally, the saturation magnetization of the functionalized magnetite is smaller than that of the bare one, due to the decrease of the efective atoms contributing to the magnetism of the samples [[147](#page-37-19), [148](#page-37-20)]. Cîrcu et al. [\[149](#page-37-21)] succeeded in enhancing the saturation magnetization of the starting magnetite nanoparticles using α -hydroxy acids (glycolic acid, malic acid, and mandelic acid) stabilizers able to form chelate complexes with the magnetite.

Sandler et al. [\[91](#page-34-10)] published a paper on magnetic nanoparticle characterization, which is suggested to researchers new to the feld but not limited to, that highlight 'best practices' for structural, chemical, and mainly magnetic characterization.

Summary and perspectives

There is a great potential for the application of magnetite nanoparticles and a lot of research studies have been conducted to synthesize the best material for targeted application. This paper highlighted the features of the coprecipitation method, which is the most widely used method for magnetite nanoparticles production. The parameters infuencing the process such as pH of the solution and the precipitating agent, temperature, ion precursor, ionic strength, and stirring rate have been discussed, exemplifed, and summarized to ensure that young and senior researchers understand the process and may select the most suitable reaction conditions for their samples considering the feld of application.

From the selected literature, it results that the size, shape, saturation magnetization and composition (iron oxide phase) of the resulting nanoparticles are very much dependent upon the presented parameters:

- Usually, magnetite nanoparticles are prepared in a pH value between 9 and 13.
- Smaller magnetite nanoparticles are obtained at higher pH values.
- A strong alkaline medium (such as NaOH, KOH, or LiOH) can lead to formation of nonmagnetic iron components, so to avoid this, weak bases such as NH4OH or $Na₂CO₃$ are recommended.
- Magnetite nanoparticles with high purity are obtained at room temperature and at higher temperature too (up to 85 °C), but usually at high temperatures larger particles can be produced with higher saturation magnetization values.
- The most used anions of iron salts consist of chloride, sulfate, and nitrate.
- The growth of oxide nanoparticles in aqueous solution can be controlled by controlling the ionic strength of the reaction solution- the increase in ionic strength results in reducing the size of the magnetic nanoparticles.
- At fast stirring speed the growth of the particles is reduced, resulting in smaller size particles.

To ensure they best describe these materials a comprehensive summary of magnetic nanoparticle characterization methods was presented, emphasizing their advantages and limitations.

In order to prevent magnetite oxidation, most of the studies are conducted in an inert atmosphere or in the presence of protective agents. Future work in this area must be focused on the research of the stability of bare, or surface functionalized magnetic nanoparticles, in time and in the media, they are intended to be used. Many studies present the stability of synthesized magnetic nanoparticles by zeta potential measurements. While zeta potential is a good indication of particles stability in suspensions, tests regarding chemical stability or the stability of the particles over time should be more often addressed.

Although many papers are dedicated to the synthesis of magnetic nanoparticles by coprecipitation, challenges remain in diferent aspects of working with this. The infuence of mixing order of the reactants and the contact time of the ions, before adding the precipitating agent, on structural and magnetic properties of the particles had not been clearly demonstrated.

Some outlooks on further developments of magnetic nanoparticles using coprecipitation method, include study of the infuence of shell thickness of magnetic core–shell particles and large-scale (industrially) production of ideal functionalized iron oxide materials. The "LaMer model" is often named to interpret the size distributions of the synthesized iron oxide nanoparticles. For iron oxide nanoparticles, the model is often cited as a diagrammatic drawing and the description of the model is only associated by words. A further perspective on mechanism elucidation is the experimental evidence that LaMer model is f the experimental nucleation kinetics data of iron oxide nanoparticles.

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Declarations

Conficts of interest There are no conficts to declare.

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