

# Photocatalytic degradation of orange II on the novel hetero-system WS<sub>2</sub>/TiO<sub>2</sub> under UV light

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Received: 12 November 2014/Accepted: 10 February 2015/Published online: 19 February 2015 © Akadémiai Kiadó, Budapest, Hungary 2015

Abstract The photodegradation of orange II (OII) is successfully achieved on the new hetero-system WS<sub>2</sub>/TiO<sub>2</sub> under UV illumination. The nano sized bi-functional catalyst is prepared by sol gel and calcined at ~450 °C. The X-ray diffraction shows noticeably broad peaks and TiO<sub>2</sub> is therefore less crystallized with the anatase phase. The sensitizer WS<sub>2</sub> presents a direct optical transition at 1.72 eV. The Mott–Schottky plot (C<sup>-2</sup>–V) is characteristic of *n* type conduction from which a flat band potential of 0.17 V<sub>SCE</sub> and a donor density of 5.30 × 10<sup>17</sup> cm<sup>-3</sup> are determined. An energy band diagram is established, predicting an electron transfer from WS<sub>2</sub> to TiO<sub>2</sub>; the injection process is confirmed by the photocurrent measurements at different concentrations. Upon increasing the mass of WS<sub>2</sub>, the activity increases and the best performance occurs on 30 % WS<sub>2</sub>/TiO<sub>2</sub>. A conversion of 98 % is reported in aerated solution for OII concentration of 10 ppm in less than 90 min. The oxidation follows a first order kinetic with a rate constant of  $4.8 \times 10^{-2} \text{ min}^{-1}$ .

Keywords Hetero-system  $WS_2/TiO_2 \cdot Orange II \cdot Photocatalytic degradation \cdot Kinetic$ 

## Introduction

Conventionally, wastewaters are treated by physical and/or biological methods that reduce the pollution level, but not enough to comply with the directives of the

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World Health Organization [1, 2]. In this respect, the advanced oxidation process (AOP) has emerged as a promising technique for the remediation of the aquatic environment, particularly for the effluents produced by the textile industry [3–6]. The SC-liquid junction may be viewed as infinity of micro-photo-electrochemical (PEC) cells. TiO<sub>2</sub> is a catalyst of choice for many applications and remains very popular owing of its superior activity, chemical stability and non-toxicity. It can practically degrade many hazardous by-products including dyes, drugs and pesticides [7, 8] or produce hydrogen from water [9]. However, TiO<sub>2</sub> based solar devices are efficient at short wavelengths (<400 nm) where only 5 % of the solar spectrum accounts for the UV radiation. Many efforts have been focused on heterosystems which could not only extend the spectral response of wide band gap semiconductors (SCs) toward lower energies but also inhibit the charge recombination and in this way increase the photocatalytic performance [10]. TiO<sub>2</sub> can mediate the electron transfer and plays a crucial role when the potential of its conduction band (CB) is less cathodic than that of the photosensitizer [11].

On the other hand, the layered chalcogenides MX<sub>2</sub>, where M is commonly a transition metal and X an element of group  $VI_A$  (X = S, Se and Te), have been actively used in PEC applications [12]. In this category, WS<sub>2</sub> is an attractive material for the solar energy conversion because it combines a narrow band gap, a chemical inertness over a fair pH range and occurs in *n*- as well in *p*-types [13]. In contrast to oxides that have a valence band (VB) deriving from  $O^{2-}$ : 2p parentage, WS<sub>2</sub>-VB is made up of less electronegative S: 3p orbital. This leads to a lower gap and a CB of cationic character with a high reducing ability. The optical transition is of d-d characteristic and involves lower and upper bands of  $W^{4+}$ : 5d: orbital. Consequently, the bond W-S is unaffected and the photocorrosion is less pronounced. In addition, the position of the bands does not change with pH and can be positioned adequately with respect to those of TiO<sub>2</sub>. The PEC properties of WS<sub>2</sub> and the position of the band edges, imposed by the crystal structure itself, are essential to predict the mechanistic and transport of the charges carriers. With a gap of  $\sim 1.8$  eV, WS<sub>2</sub> can absorb  $\sim 50$  % of the sunlight and possesses high absorption coefficients ( $\sim 10^5 \text{ cm}^{-1}$ ) [14].

The main goal of this work is to report the preparation of the hetero-system  $TiO_2/WS_2$  by sol gel and its PEC characterization. As an application, the photocatalytic performance is successfully tested toward the oxidation of orange II (OII) upon UV light on the bi-functional hetero-system under short circuited conditions. OII is a model molecule with weak biodegradability, it is widely used in the textile industry because of its coloring properties and its degradation is of potential significance. The photocatalytic activity is dependent on many parameters among which the OII concentration and WS<sub>2</sub> loading.

#### Experimental

The hetero-system  $TiO_2/WS_2$  is synthesized by sol gel; titanium iso-propoxyde { $Ti(OC_3H_7)_4$ , Aldrich, 97 %}, ethanol (Fluka, 98 %) and methanol (Fluka, >99.5 %) are used as starting reagents. An equimolar mixture of ethanol/methanol

is introduced in a Pyrex beaker (250 mL capacity) and Ti(OC<sub>3</sub>H<sub>7</sub>)<sub>4</sub> is added dropwise. Then, the desired amount of WS<sub>2</sub> is added. The proportions of the sensitizer (WS<sub>2</sub>) with respect to the mass of TiO<sub>2</sub> (prepared alone in a first part) were 10 % (0.167 g), 20 % (0.334 g) and 30 % (0.501 g). The resulting mixture is heated during 3 h at 75 °C, after which hot water is added dropwise. After gelling, the samples were dried overnight at 110 °C and then heat-treated in air for 2 h at 450 °C with a heating rate of 3 °C/min. To confirm the formation of the phases, X-ray diffraction (XRD) is performed with a Siemens diffractometer (Model D-5000) using Cu K<sub>α</sub> radiation. BET surface areas are determined from nitrogen adsorption at 77 K using a Micromeritics apparatus (Coulter SA 3100 and Belsorpmini porosimeter); the samples are outgassed at 140 °C (4 h) to vacuum (10<sup>-4</sup> Pa) in order to have a clean surface.

The optical absorption spectra are recorded with a double beam spectrophotometer (Specord 200 Plus) equipped with an integrating sphere, PTFE is used as reference. The point of zero zeta potential (pzzp) of WS<sub>2</sub> is determined from the equilibrium pH of an aqueous solution containing a powder suspension; a value of  $\sim 1$  is obtained.

WS<sub>2</sub> is mechanically soft with poor sintering. It is cold pressed into circular pellets ( $\emptyset = 13$  mm, thickness ~1 mm) under a pressure of 4 tons cm<sup>-2</sup> and heated at 300 °C in evacuated Pyrex ampoule, the compactness approximates 70 %. Ohmic contact onto the back pellets is made by soldering copper wires with silver paint. The working electrodes (WE) are prepared by encapsulating the pellets in glass holders. Electrochemical (EC) and PEC characterizations are carried out in a closed cell with a three electrode arrangement using Pt sheet (1 cm<sup>2</sup>) as auxiliary electrode. All potentials are given with respect to a saturated calomel electrode (SCE) and controlled with a PGZ301 potentiostat (Radiometer analytical). WS<sub>2</sub> is illuminated with a tungsten lamp (200 W) through an optical window. The differential capacitance is measured at a frequency of 10 kHz.

The photocatalytic tests are done in batch mode. 50 mg of  $TiO_2/WS_2$  are dispersed by magnetic stirring in 100 mL of OII solution (10 mg/L). The temperature is regulated at 25 °C with a thermostated bath. The powder suspension is achieved by magnetic stirring; ~2 h is required to reach the dark adsorption equilibrium. The light source consists of a high pressure mercury lamp (HPK 125 W Phillips), mounted on the central axis of a quartz reactor. The aliquots are withdrawn at regular times, subjected to a vigorous centrifugation (3000 rpm, 15 mn) to remove the solid particles and filtered through a 0.45-mm Millipore filter. The disappearance of OII is followed by UV–Visible spectrophotometry (Shimadzu 1800,  $\lambda_{max} = 484$  nm) using 1 cm quartz cell. The photocatalytic yield is calculated from the relation:

$$\eta = 100 \times (C_0 - C_t)/C_0 \tag{1}$$

Here,  $C_0$  is the initial concentration and  $C_t$  the concentration after irradiation for time (*t*). No OII is degraded in the dark and all solutions are made up with CO<sub>2</sub> free distilled water.

### **Results and discussion**

Structure and optical properties of WS<sub>2</sub>

TiO<sub>2</sub> presents three crystallographic varieties, the anatase is the most photoactive and is thermally stable up to ~500 °C, above which it converts irreversibly to the rutile variety. After calcination at 450 °C, the system is mixed phases and the XRD patterns show the peaks of WS<sub>2</sub> and TiO<sub>2</sub> anatase (Fig. 1). WS<sub>2</sub> is an excellent solid lubricant, which adopts the MoS<sub>2</sub> molybdenite structure, intermediate between the ionic rutile SnO<sub>2</sub> and covalent CdI<sub>2</sub> [15]. It has two-dimensional structure with S– W–S sandwich sheets connected to each other by Van der Waals type interactions. It crystallizes in platelet morphology, which forms layered crystal with a grey color, the structural refinement is performed with the space group P6<sub>3</sub>/mmc (No 194) in agreement with the JCPDS card No 21-1272. The structure is highly anisotropic; it is described as a cubic unit cell containing sulfur atoms at the corners and W<sub>4</sub>S tetrahedron unit in the center.

The optical properties of WS<sub>2</sub> are typical of insulators and therefore not different from those of other sulfides. The fundamental absorption, which corresponds to electron excitation, is used to determine both the nature of the optical transition and its value. The relation between the gap (E<sub>g</sub>), the absorption coefficients ( $\alpha$ ) and the incident energy (hv) is given by:

$$(\alpha hv)^{n} = C(hv - E_{g}) \tag{2}$$

In Eq. 2, *C* is a constant. For crystalline SCs, the exponent *n* depends on the type of transition, for direct allowed, n = 1/2 and for indirect allowed transition, n = 2. The extrapolation of the linear part  $(\alpha hv)^n$  to the energy axis shows that the inter band transitions are directly and indirectly allowed (Fig. 2). Due to the grey color,



Fig. 1 X-ray diffraction pattern of the hetero-system TiO<sub>2</sub>/WS<sub>2</sub> (10 %)



Fig. 2 Direct and indirect optical transitions of  $WS_2$  obtained from the diffuse reflectance spectrum



**Fig. 3** The photocurrent-potential characteristics of WS<sub>2</sub> in aerated OII solution at different concentrations using Na<sub>2</sub>SO<sub>4</sub> as support electrolyte at 25 °C, scanning rate 10 mV/s, *a* 0 ppm; *b* 5 ppm; *c* 10 ppm; *d* 15 ppm; *e* 20 ppm. *Inset* the J(V) plot of WS<sub>2</sub> in the dark in Na<sub>2</sub>SO<sub>4</sub> electrolyte

 $E_g$  can be reasonably taken at 1.72 eV, which is attributed to the transition 5d: W<sup>4+</sup>:  $e_g \rightarrow t_{2g}$ . A further transition at 1.91 eV, directly allowed is presumably due to the charge transfer S<sup>2-</sup>:  $3p \rightarrow W^{4+}$ :  $e_g$ . Such energy is smaller than that observed in oxides because of the higher energetic position of S<sup>2-</sup>: 3p orbital by ~1 eV above O<sup>2-</sup>: 2p orbital and the weak ionicity of the W–S chemical bond.

## EC and PEC characterizations

The current density-potential J(V) characteristics of WS<sub>2</sub> are done in OII solution (10 ppm, pH ~7) using Na<sub>2</sub>SO<sub>4</sub> (10<sup>-3</sup> M) as support electrolyte (Fig. 3, Inset). The plot exhibits a plateau region with a dark current less than 5  $\mu$ A cm<sup>-2</sup> that becomes cathodically large at potentials negative of ~ -0.9 V owing to hydrogen evolution. The reduction of S<sup>2-</sup> to elemental sulfur appears at ~ -0.5 V in close agreement with the redox potential of S/S<sup>2-</sup> couple (-0.75 V). The photocurrent (J<sub>ph</sub>) starts to flow at ~0 V (photocurrent onset potential: V<sub>on</sub>, figure not shown) and increases toward anodic potentials, indicating *n*-type conductivity. One of the most important parameters to determine in the PEC conversion is the flat band potential (V<sub>fb</sub>) under the operating conditions, it is given by the relation:

$$\frac{1}{C^2} = \left(\frac{2}{\varepsilon \varepsilon_0 e N_A}\right) \left(V - V_{fb} - kT/e\right) \tag{3}$$

Here *e* is the electronic charge,  $\varepsilon$  the dielectric constant of WS<sub>2</sub> (= 8) [16],  $\varepsilon_o$  the dielectric constant of vacuum (8.85 × 10<sup>-12</sup> F m<sup>-1</sup>) and *kT* the thermal energy (~26 meV at 25 °C). The intercept of the fitted line at C<sup>-2</sup> = 0 (Fig. 4) and the slope give the potential V<sub>fb</sub> (-0.77 V) and the electron density (N<sub>D</sub> = 5.30 × 10<sup>17</sup> cm<sup>-3</sup>). The difference between the potentials V<sub>on</sub> and V<sub>fb</sub> indicates the presence of surface states within the gap region, while the positive slope confirms the *n* type behavior. WS<sub>2</sub> is lightly doped and the depletion width  $\delta$  (~20 nm), extends over many crystallographic unit cells:

$$W = \left(\frac{2\varepsilon\varepsilon_0 \Delta V}{eN_D}\right)^{0.5} \tag{4}$$

This is advantageous in photocatalysis, where most electron/hole  $(e^{-}/h^{+})$  pairs generated within the depletion length contribute to the photoactivity. The potential





$$E_{CB} = 4.75 + |e|V_{fb} - E_a$$
(5)

The  $E_{CB}$  value is -0.87 V (3.88 eV); hence the VB is located at (0.85 V/ 5.6 eV =  $E_{CB} + E_g$ ). Such a result confirms the cationic character of the electronic bands of WS<sub>2</sub>, made up of W<sup>4+</sup> 5*d* orbital. This study underlines once more the role played by the crystal structure, which determines not only the position of the electronic bands, but also the width of the gap through the ionicity of the chemical bond. The electronic band are determined by *d* levels, the lower filled  $e_g$  levels providing VB though being non-bonding, whereas CB consists mainly of empty  $t_{2g}$ levels. Such electronic structure is compatible with the chemical inertness and gives WS<sub>2</sub> its semi-conductivity. The *d*–*d* transitions are not allowed by the Laporte rules. However, they are observed because the CB has not pure *d* character but is rather a hybridization of W<sup>4+</sup>: 5*d* and S<sup>2-</sup>: 3*p* orbitals. The CB of the anatase has not been measured because of its poorly sintered character, its value (-0.75 V) is taken from Ref. [17].

## Photocatalysis

Dyes affect dramatically the aquatic life by reducing the light flux, thus inhibiting the photosynthesis; they are discharged in water and recovered by conventional methods. The adsorption is a displacement of the pollution otherwise the catalyst should be regenerated or replaced by a new one and this requires a further investment. By contrast, environmental photoelectrochemistry is an emerging technique for the water treatment [18], where the dye is converted to less harmful forms (ideally into CO<sub>2</sub> and H<sub>2</sub>O). A literature survey has led to the conclusion that AOP occurs via radicals  $O_2^{\bullet-}$  and/or OH<sup> $\bullet$ </sup> generated respectively from photoelectrons in WS<sub>2</sub>-CB and/or photoholes in WS<sub>2</sub>-VB and the presence of molecular oxygen is crucial for the photodegradation. It has been observed that OII is not converted by photolysis; therefore any change in the OII concentration is due to the photocatalytic process (see below). According to the PEC characterization, the energetic diagram of the hetero-system WS<sub>2</sub>/TiO<sub>2</sub> (Fig. 5) provides insights on the feasible reactions; the higher occupied molecular orbital (HOMO, 0.17 V) and the lower unoccupied molecular orbital (LUMO, -1.86 V) of OII are taken from Ref. [19]. The absorption spectrum of OII shows two main bands at 430 and 485 nm, which corresponds to the  $n \rightarrow \pi_*$  transition of the azo and hydrazone forms, respectively [20].

The thermodynamic requirement is such that the potential of  $O_2/O_2^{\bullet-}$  couple (-0.52 V) [21] is less negative than WS<sub>2</sub>-CB. However, the difference between the  $O_2/O_2^{\bullet-}$  couple and WS<sub>2</sub>-CB is large for a direct electron exchange and a low activity is expected since the electron transfer should occur iso-energetically. Some strategies have been attempted to overcome such drawback and a growing interest

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<sup>&</sup>lt;sup>1</sup> The activation  $E_a$  (~0.1 V) is determined from the conductivity measurements on sintered pellets.



Fig. 5 The energy band diagram of the hetero-system n-WS<sub>2</sub>/n-TiO<sub>2</sub> in the orange II solution at pH  $\sim 6$ 

has been paid to hetero-systems [22, 23]. As mentioned above, coupled SCs enhance the photocatalytic efficiency by extending the spectral photoresponse toward longer wavelengths and hinder the loss of  $(e^{-}/h^{+})$  pairs [24].

WS<sub>2</sub>-CB is pH insensitive whereas TiO<sub>2</sub>-CB changes usually by -0.06 V pH<sup>-1</sup>, we have exploited this property to have an optimal band bending at neutral pH. Owing to the adequate position of its CB, TiO2 provides a bridge between WS2 and the  $O_2/O_2^{\bullet}$  couple since WS<sub>2</sub>-CB (-0.87 V) is more cathodic than TiO<sub>2</sub>-CB (-(0.75 V) [17]. Moreover, the synthesis of TiO<sub>2</sub> in the presence of WS<sub>2</sub> in suspension byn the sol-gel method leads to intimate contact, which facilitates the electrons transfer. The photoactivity is dependent on the morphology of the bi-functional catalyst. Then, the question arises about the effect of decreasing the dimension of the crystallite on the OII oxidation. The strategy is that the charge carriers should have a lifetime as long as possible to reach the interface particularly for small polaron SCs with a low carrier mobility; this can be achieved by decreasing the path the carriers have to diffuse to reach the interface, and nano sized dimension is desirable for that purpose [25]. In this respect, the sol-gel has gained a growing popularity in the solid state chemistry for preparing particles with high surface-tovolume ratio [26]. In addition, the porous nature of TiO<sub>2</sub> reduces the over-voltages by increasing the active surfaces; the specific surface area determined by BET isotherms works out to be 110 m<sup>2</sup> g<sup>-1</sup>.

The ability of excited OII to inject electrons into WS<sub>2</sub>-CB with visible light is measured through the J(V) curves (Fig. 3). At high potentials (>V<sub>fb</sub>) in the plateau region, the limiting current is diffusion controlled and depends only on the OII concentration i.e. the mass transport of the electro-active species toward the interface and the system is under kinetic control. Another possibility to explain the photocurrent increase can be due to the hole injection from WS<sub>2</sub>-VB to HOMO, but this hypothesis has not been checked in our case. The dark adsorption favors the OII oxidation and is a first step in the photocatalytic process. The adsorption of azo dyes occurs through the sulfonic groups; the primary amines groups are likely responsible for OII binding to favour its access to catalytic sites onto  $TiO_2$  by electrostatic interactions. However, the OII adsorption is weak, less than 5 % as shown by measuring of the concentration before and after keeping the powder overnight in OII solution (10 ppm).

WS<sub>2</sub> is photoexcited and the electrons are injected to TiO<sub>2</sub>-CB and subsequently transferred to dissolved oxygen which is reduced preferentially because of its redox potential (-0.50 V). Therefore, if the solution is purged with nitrogen, OII degradation is strongly inhibited. The mechanistic pathway generally accepted is the following [27]:

$$O_{2 ads} + e^{-} \rightarrow O_{2}^{\bullet} + (H^{+}) \rightarrow HO_{2}^{\bullet} \rightarrow H_{2}O_{2} + (e^{-}) \rightarrow O_{2} + OH^{\bullet} + OH^{-}$$
(6)

The illumination time is fixed at 4 h and the main parameters that influence the photoactivity are the loading of the sensitizer WS<sub>2</sub> and the OII concentration ( $C_0$ ). The photoactivity is investigated by varying the mass of WS<sub>2</sub> keeping that of TiO<sub>2</sub> constant (Fig. 6). As expected, the performance increases with increasing the amount WS<sub>2</sub>, due to its effective dispersion over TiO<sub>2</sub>, which provides a high reception surface for the incident photons and in this way the contribution of a large number of ( $e^-/h^+$ ) pairs. Similarly, there is an optimal concentration above which OII act as optical filter; the light is absorbed before reaching the catalyst and competes with WS<sub>2</sub> absorption, thus lowering the photoactivity. The half life ( $t_{1/2}$ ), the time needed to degrade half of OII present initially, is found to be concentration dependent. The linear relation between  $ln(C_0/C_t)$  and irradiation time (t) of the photoactalytic degradation of OII obeys pseudo-first order kinetics (Fig. 7):

$$\ln C_t = -kt + \ln C_0 \tag{7}$$



Fig. 6 Effect of  $WS_2$  on the photocatalytic decolorization of orange II on the hetero-system  $TiO_2/WS_2$ 



Fig. 7 Linear plots  $ln(C_0/C)$  versus illumination time of orange II on the hetero-system  $TiO_2/WS_2$ 

Table 1 First order rate constant (k) and half-lives  $(t_{1/2})$  for the photocatalytic degradation of orange II on TiO<sub>2</sub>/WS<sub>2</sub> system

Catalysts	$\mathrm{K_{app}}~(\mathrm{min}^{-1}) \times ~10^{2}$	t <sub>1/2</sub> (min)	$R^2$
TiO <sub>2</sub> –SG	2.11	31.5	0.998
TiO2-SG/10 %WS2	2.61	26.7	0.990
TiO <sub>2</sub> -SG/20 %WS <sub>2</sub>	3.22	21.7	0.989
TiO <sub>2</sub> –SG/30 %WS <sub>2</sub>	4.80	14.4	0.989

The rate constants (k) and half-lives (Table 1) show that the best performance is obtained on 30 %WS<sub>2</sub>/TiO<sub>2</sub> with a constant k of  $4.8 \times 10^{-2} \text{ mm}^{-1}$ . The kinetics exhibits an initial period of relatively rapid degradation. Over time, the slopes decrease progressively and become smaller followed by the gradual cessation of the photoactivity (Fig. 6b). This tendency to saturation indicates that the layers already adsorbed are oxidized after which the process becomes governed kinetically by the diffusion of OII toward the active sites at the interface in which the radicals OH<sup>•</sup> are generated for further adsorption/photodegradation. So, the process is self limited due to adsorbed layer and availability of photocatalytic sites. In this respect, the EC impedance would be helpful to bring insights on the interfacial process. This study is presently under way and will be consecutively reported.

## Conclusion

The feasibility of WS<sub>2</sub> loaded TiO<sub>2</sub> is demonstrated for the light induced oxidation of OII in aerated aqueous solution under UV light. WS<sub>2</sub>/TiO<sub>2</sub> achieves the colloidal photochemical hetero-system synthesized by sol gel with intimate contact. The choice of WS<sub>2</sub> is motivated by the chemical stability over a fair pH range and high energetic position of the CB, made up of W: 5d orbital, more cathodic than TiO<sub>2</sub>-CB resulting in the electron transfer. The combination of the flat band potentials and optical gaps permitted to build the energetic diagram. Electron injection from OII to WS<sub>2</sub> upon visible illumination is confirmed by the photocurrent measurement. Oxygen improves the photoactivity considerably and the oxidation of OII obeys to first order kinetics. The photocatalytic process is governed by the diffusion of OII to active sites in which the radicals OH<sup>•</sup> are generated.

**Acknowledgments** The authors would like to express their gratitude to the Faculty of Chemistry for the financial support of this research. The authors are also grateful to H. Zouaoui for his technical assistance.

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