

# **Optimizing CBTSSe solar cells for indoor applications through numerical simulation**

Saif M. H. Qaid<sup>1</sup> · Ahmed Shaker<sup>2</sup> · Mohamed Okil<sup>3</sup> · Muath Alkadi<sup>4</sup> · **Abdullah Ahmed Ali Ahmed5 · Walid Zein2**

Received: 28 August 2023 / Accepted: 4 October 2023 / Published online: 26 October 2023 © The Author(s), under exclusive licence to Springer Science+Business Media, LLC, part of Springer Nature 2023

# **Abstract**

Solar cell technologies are pivotal in the transition towards sustainable energy sources. This paper delves into the design and optimization of CBTSSe solar cells for efficient indoor energy harvesting. The tunable bandgap nature of CBTSSe aligns well with the emission spectra of indoor LED lighting, making it a promising candidate for such applications. Our investigations in this paper commence with the design validation of a solar cell device structure, in line with experimental work, ensuring the accuracy of our models and simulation software. Through systematic simulations, we explore the dependence of absorber bandgap on the indoor LED color temperature. The study progresses into a multi-step optimization process, targeting crucial aspects of solar cell design. Investigating interface parameters, we analyze the interplay between the conduction band ofset and surface recombination velocity. Subsequently, the infuence of absorber layer properties on efficiency is examined, uncovering substantial improvements by optimizing thickness and bulk defect density. Further, we scrutinize the impact of electron transport layer thickness and doping, presenting avenues for performance enhancement. Quantitative results underscore the potency of our approach. Under warm and cool LED illumination, the proposed optimizations elevate the power conversion efficiency from  $10.05$  to  $25.58\%$  and  $9.08$  to 25%, respectively, validating the efectiveness of our strategies. These fndings not only underscore the viability of earth-abundant CBTSSe-based solar cells for indoor applications but also pave the way for tailored, high-efficiency low-cost energy harvesting in various indoor environments.

Keywords CBTSSe · Indoor LED · SCAPS-1D · CBO · Power conversion efficiency

# **1 Introduction**

The rising global energy necessity and growing concerns about environmental impact impose the exploration of renewable and clean energy sources. Solar energy, among several renewable options, holds substantial prospective, where photovoltaic (PV) technology has emerged as an efficient way of directly converting solar energy into electricity (Green et al. [2023](#page-19-0); Petrović-Ranđelović et al. [2020](#page-20-0)). Silicon (Si) solar cells currently

Extended author information available on the last page of the article

lead the PV market, representing about 93% of the market segment (Okil et al. [2022\)](#page-20-1). Apart from this, thin flm solar cells (TFSCs) have gained signifcant traction because of their notable advantages involving high power conversion efficiencies (PCEs), easy fabrication processes on fexible substrates, and low processing cost (Huang [2020;](#page-20-2) Gamal et al. [2021](#page-19-1); Farooq et al. [2021;](#page-19-2) Ait Abdelkadir et al. [2023](#page-19-3)). These advantages have led to the widespread application of thin flm PVs, particularly in building-integrated photovoltaics and other related felds.

Among TFSCs, chalcogenide compounds have garnered noteworthy scientifc consideration as semiconducting materials for their tunable bandgap and potential applications in PV devices. While thin-flm materials like CIGS and CdTe have proven potential for renewable energy generation, their large-scale production is hampered by the supply constraints of In, Ga, and Te (Green et al. [2021](#page-19-4)). The alloys of kesterite, specifically  $Cu_2ZnSnS_4$  (CZTS) and  $Cu_2ZnSn(S, Se)_4$  (CZTSSe), offer probable alternatives as they are composed of earth-abundant and non-toxic elements (Kumar [2021;](#page-20-3) Gohri et al. [2023;](#page-19-5) Shafi et al. [2022](#page-20-4)). Nevertheless, the performance of solar cell devices using these materials is constrained by cationic disorder and defect properties, besides their efficiency, which lags behind that of CIGSSe cells (Green et al.  $2021$ ; Sun et al. [2019](#page-20-5)). Recently, Cu<sub>2</sub>BaSnS<sub>4−*x*</sub>Se<sub>*x*</sub> (CBTSSe) has emerged as a prospective alternative due to the larger ionic radius of Ba<sup>2+</sup> (r<sub>i</sub> = 1.56 Å) compared to Zn<sup>2+</sup> (r<sub>i</sub> = 0.74 Å), offering a higher coordination number. The abundant availability of Ba, Cu, and Sn further adds to its interest as a light absorber for PV cells, surpassing the scarcity of In (Sun et al. [2019;](#page-20-5) Chakraborty et al. [2019;](#page-19-6) Ge et al. [2017a\)](#page-19-7). Considerable improvements have been achieved in the synthesis of CBTSSe light absorbers and related PV devices, benefting from theoretical insights, device structures, and functional materials. Moreover, the CBTSSe material exhibits a quasi-direct bandgap, which can be adjusted within a broad range of 1.5–2.0 eV by varying the S:Se ratio (Shin et al. [2017a](#page-20-6)). This tunability ofers promising options for applications in both single-junction and multi-junction PV devices.

Numerous research studies, both experimental and simulation, have been devoted to investigating various fabrication technologies and device confgurations for the CBTSSebased solar cells. In (Shin et al.  $2017b$ ), the authors implemented a two-step process involving precursor sputter deposition and subsequent sulfurization/selenization to produce high-quality, pinhole-free CBTSSe layers incorporating selenium for high-efficiency PV devices. The incorporation of selenium enables the achievement of absorber layers within the CBTSSe with a bandgap of 1.55 eV, ideal for single-junction PV applications. Furthermore, through bandgap tunability, a CBTSSe-based PV cell with a remarkable PCE of 5.2% was reported (Shin et al. [2017b](#page-20-7)). One promising approach to boost the performance of CBTSSe-based cells is to incorporate low electron afnity (EA) electron transport materials such as  $Zn_{1-x}Cd_xS$ ,  $Zn_{1-x}Sn_xO$ , and  $ZnO_{1-x}S_x$  (Cui et al. [2018\)](#page-19-8). An experimental study demonstrated improved open-circuit voltage  $(V<sub>OC</sub>)$  and PCE by incorporating sputtered CdS:O buffer layers, suggesting that further optimization of buffer layers could lead to higher performance in CBTSSe solar cells (Ge et al. [2017b](#page-19-9)). More recently,  $Zn_{1-x}Cd_{x}S$ (ZCS) was employed as a bufer layer with low electron afnity to enhance the band alignment between the CBTSSe absorber and the heterojunction (Cui et al. [2018\)](#page-19-8). To further enhance the band alignment within the cell, a lower electron affinity  $(Zn, Mg)O$   $(ZMO)$ and Al-doped ZnO (AZO) window layers have been incorporated. These modifcations were aimed at achieving improved conduction band minimum (CBM) alignment across the entire device structure, resulting in a PCE of above 6% for the device structure AZO/ZMO/ ZCS/CBTSSe/MoS<sub>2</sub>/Mo (Teymur et al. [2022](#page-20-8)).

On the other hand, various simulation investigations have been published in the literature. In (Ghobadi et al. [2020\)](#page-19-10), CBTSSe solar cells were simulated using SCAPS-1D, and the results were justifed against experimental data. Various back surface feld layer materials were recommended to enhance the performance. Optimization of the MoSe, layer, formed at the interface between the CBTSSe flm and the Mo back electrode, was performed to determine the optimal parameters for the BSF layer. According to the various design techniques, an optimized PCE of 7.31% has been achieved, providing insights for optimizing BSF layers in CBTSSe solar cells (Ghobadi et al. [2020\)](#page-19-10). In (Henni et al. [2023](#page-20-9)), the authors examined the impact of establishing an ultrathin  $Al_2O_3$  insulating layer with a wide bandgap into a CBTS/CdS heterojunction cell, where CBTS has an energy gap of 1.9 eV. Through experimentally calibrated numerical simulations, they investigated the electron tunnelling effect within the  $Mo/MoS<sub>2</sub>/CHTS/AI<sub>2</sub>O<sub>3</sub>/CdS/ZnO/AZO/AI$  structure. The findings indicated that the inclusion of an  $A<sub>1</sub>O<sub>3</sub>$  layer improves the collection efficiency, leading to enhanced cell performance. Specifically, the solar cell with the CBTS/Al<sub>2</sub>O<sub>3</sub>/ CdS structure achieved a PCE of 11.89%, with  $V_{\text{OC}}$  of 1.08 V (Henni et al. [2023\)](#page-20-9). Moreover,  $In_2S_3$  thin films were numerically investigated as an ETL. An experimental CBTSSe PV cell was simulated and validated, followed by the introduction of the  $In_2S_3$  layer as a replacement to toxic CdS. The results demonstrated that the  $In_2S_3$  layer can effectively function as an ETL, and different variations of  $In_2S_3$  properties were explored as potential ETLs for further optimization (Hashemi et al. [2021](#page-19-11)). When comparing the device parameters to theoretical limits, the record CBTSSe device exhibits less than 50% of the maximum voltage ratio ( $V_{OC}$ / $V_{OC}^{SQ}$ ), while optimized CZTSSe devices offer about 60% (Wang et al. [2014\)](#page-20-10). Despite some improvements, CBTSSe devices still face a significant  $V_{OC}$  deficit. To address this issue and assess suitability for practical applications, it is crucial to gather information on defects, carrier lifetime, density, mobility, absorber band levels and more. Moreover, a comprehensive simulation of CBTSSe solar cell devices is crucial to accelerate their development.

As indoor lighting predominantly involves LED sources, which emit light within a specifc range of wavelengths and specifc color temperatures, having an absorber material with a bandgap optimized for the diferent conditions is crucial for enhancing energy harvesting efficiency. The unique property of CBTSSe as a tunable bandgap material allows for the adjustment of its bandgap within the specifc range 1.7–1.9 eV, a range that aligns well with the spectral emission of LED illumination commonly used indoors. Therefore, in this paper, we investigate the suitability of CBTSSe-based PV cells having wide bandgaps for indoor applications. Our investigation begins by designing the solar cell device structure, adhering to the ITO/i-ZnO/CdS/absorber/Mo/glass layer sequence, based on an experimental work (Shin et al. [2017b](#page-20-7)). By replicating the terminal characteristics of the PV cell by simulation under AM1.5G illumination, we provide a validation of the models employed. Next, we present the design of the CBTSSe solar cell under warm and cool LED illumination conditions. Multiple factors impacting CBTSSe solar cell performance across varying LED spectra are investigated. The core objective of this study is to achieve the optimization of cell performance through an in-depth analysis of the intricate interplay between key parameters. Following an initial investigation into the role of the absorber bandgap, we embark on an intricate optimization procedure aimed at designing the solar cell to achieve the utmost achievable PCE. The optimization process unfolds through three main steps, each addressing a diferent aspect of the cell design. These stages encompass the meticulous management of interface parameters (CBO and S), absorber layer parameters (thickness and bulk defects), and ETL parameters (thickness and doping). Finally, we provide a comprehensive overview of the optimization process. By encapsulating an array of simulations and investigations, this study is poised to not only advance the understanding of CBTSSe solar cell performance but also ofer critical insights into the intricate relationship between material parameters and device efficiency. This understanding is anticipated to significantly contribute to the advancement of wide-bandgap solar cell technologies optimized for indoor applications.

### **2 Simulation method and device confguration**

### **2.1 SCAPS simulator**

In this study, we utilize the SCAPS-1D software, evolved by the University of Ghent, Belgium, to simulate and calculate the key PV parameters of the CBTSSe-based thinflm solar cell. SCAPS-1D is a one-dimensional PV cell device simulator program primarily designed for TFSCs. The simulator accurately calculates the performance metrics of the PV devices using numerical solutions of fundamental semiconductor equations. These equations are solved self consistently until convergence is met. Poisson's equa-tion (Eq. ([1\)](#page-3-0)) is used to describe the electrostatic potential  $\Psi$  within the semiconductor material. The continuity equations  $(Eq. (2))$  $(Eq. (2))$  $(Eq. (2))$  describe the continuity of charge carriers (electrons and holes) in a certain semiconductor region. They are crucial for modeling carrier transport and recombination. The bulk Shockley–Read–Hall (SRH) recombination equation (Eq.  $(3)$  $(3)$ ) calculates the recombination rate due to bulk defects, while the interface SRH recombination equation (Eq. [\(4](#page-4-0))) accounts for recombination at the interfaces between the distinct layers. Finally, the transport equations (Eq. [\(5](#page-4-1))) govern the transport of holes and electrons within the material under consideration. They are basically based on the drift–difusion approach. Regarding the mobility modeling, the electron and hole mobilities are taken to be constants (constant mobility model).

$$
\frac{d^2\Psi}{dx^2} = \frac{q}{\varepsilon}(n - p + N_A^- - N_D^+ + n_t - p_t)
$$
 (1)

The ionized acceptor and donor doping concentrations are denoted by  $N_A^-$  and  $N_D^+$ . The free electron and hole concentrations are termed *n*, and *p*, while the corresponding trap densities are  $n_t$  and  $p_t$ .

<span id="page-3-2"></span><span id="page-3-1"></span><span id="page-3-0"></span>
$$
\frac{1}{q}\frac{dJ_n}{dx} = U_n - G(x)
$$
\n
$$
\frac{1}{q}\frac{dJ_p}{dx} = G(x) - U_p
$$
\n(2)

The terms  $U_n$  and  $U_p$  represent carrier recombination rates, while G is the generation rate due to a light source (either AM1.5G or due to LED illumination, etc.).

$$
R_{SRH,bulk} = \frac{(np - n_i^2)}{\tau_p(n + n_1) + \tau_n(p + p_1)}
$$
(3)

 $\tau_n$  and  $\tau_p$  are bulk minority carrier lifetimes for electrons and holes.

<span id="page-4-0"></span>
$$
R_{SRH, int} = \frac{(n_{ij}p_{ij} - n_i^2)}{\frac{p_{ij} + p_1}{S_n} + \frac{n_{ij} + n_1}{S_p}}
$$
(4)

 $S_n$  and  $S_n$  are the interface recombination velocities for electrons and holes.

<span id="page-4-1"></span>
$$
J_n = qn\mu_n E + qD_n \frac{dn}{dx}
$$
  
\n
$$
J_p = qp\mu_p E - qD_p \frac{dp}{dx}
$$
\n(5)

 $D_n$  and  $D_n$  represent the electron and hole diffusion coefficient, while  $\mu_n$  and  $\mu_n$  denote the electron and hole mobility. The electric feld is denoted by *E*.

Upon performing the solution, many terminal characteristics and physical parameters can be extracted such as the current density versus voltage (*J–V*) and external quantum efficiency (EQE), recombination rates, electric field, energy band profiles and other useful characteristics. Other crucial parameters such as PCE,  $V_{\text{OC}}$ ,  $J_{\text{SC}}$  (short-circuit current), and FF (fll factor) can also be easily extracted.

#### **2.2 Solar cell arrangement and materials**

The solar cell device construction is designed based on a fabricated PV cell incorporating the following layer sequence: ITO/i-ZnO/CdS/Cu<sub>2</sub>BaSnS<sub>4-*x*</sub>Se<sub>x</sub>/Mo/glass, where  $x = 3$ . Figure [1a](#page-5-0) demonstrates the PV cell structure, while Fig. [1](#page-5-0)b illustrates the conduction band minimum (CBM) and valence band maximum (VBM) for all cell layers. The CdS thickness is 50 nm, while the thicknesses of i-ZnO and ITO are 50 nm and 150 nm, respectively. The absorber thickness is about 1  $\mu$ m, while its measured bandgap is about 1.54 eV (Shin et al. [2017b\)](#page-20-7). Regarding the work functions of the electrodes, they are adjusted at 4.4 eV for the front ITO contact and 5 eV for the back Mo contact. Most of the defect parameters are obtained from Ref (Shin et al. [2017b\)](#page-20-7), while some other parameters are optimized to ft the experimental data. All other material properties and parameters as well as bulk and interface defects used in SCAPS simulations are extracted from relevant literature as listed in Tables [1](#page-6-0) and [2.](#page-7-0) Further, the energy band profle after contact at short-circuit condition is displayed in Fig. [1c](#page-5-0) (given AM1.5G illumination).

#### **2.3 SCAPS calibration**

The following simulations are conducted at an operating temperature of 300 K, with AM1.5G light illumination (1000  $W/m<sup>2</sup>$ ). The reference experimental cell detailed in Shin et al. [\(2017b](#page-20-7)) was replicated in the SCAPS software. The alignment of the simulated results in the SCAPS environment was achieved by utilizing the physical material parameters outlined in Tables [1](#page-6-0) and [2](#page-7-0). Upon using these parameters and running the simulator, decent agreement between the experimental and SCAPS outcomes is observed, as presented in Fig. [2a](#page-7-1), which shows the *J–V* characteristic curves of both simulated and experimental PV cells. Further, in order to quantitatively validate the simulated results from the SCAPS simulator against the experimental work, we compared the functional PV metrics of the simulated solar cell versus the results from the experimental cell as reported in Shin et al. ([2017b\)](#page-20-7). The congruence between the SCAPS results and the experimental outcomes is evident from the data provided



<span id="page-5-0"></span>**Fig. 1** Typical CBTSSe cell confguration: **a** main cell layers, **b** energy band profle before contact illustrating the accompanying energy levels of each thin flm, and **c** energy band diagram after contact at short circuit current condition under AM1.5G illumination

in Fig. [2a](#page-7-1), where PV parameters, including: PCE of 5.27%,  $V_{OC}$  of 0.618 V,  $J_{SC}$  of 17.32 mA/cm<sup>2</sup>, and FF of 49.17%, were adopted, as highlighted in the figure. The values of the main simulated PV metrics closely align with those obtained experimentally, with a percentage error of just 1.35% for PCE. In addition, EQE characteristic curves of experimental and simulated CBTSSe cells (Shin et al. [2017b\)](#page-20-7) are presented in Fig. [2b](#page-7-1), utilizing the given parameters. It is prominent to notice that the simulated EQE-derived photocurrent is  $17.32 \text{ mA/cm}^2$ , which closely matches the measured  $J_{SC}$ under AM1.5G illumination condition (17.40 mA/cm<sup>[2](#page-7-1)</sup>). The findings revealed in Fig. 2 underscore the accuracy of the applied models and the merit of the software utilized in this study. It should be pointed out here that the experimental EQE-derived photocurrent is 17.59 mA/cm<sup>2</sup>, which is higher than the experimental  $J_{SC}$  derived from the  $J-V$ curve. This can be attributed to the diferent conditions of measurements regarding the *J*–*V* and EQE curves.



<span id="page-6-0"></span>

	ZnO	CdS	CBTSSe <sub>3</sub>	MoSe <sub>2</sub>	CdS/CBTSSe <sub>3</sub>
Defect type	Neutral	Neutral	Donor	Neutral	Donor
Carrier capture cross section			$1 \times 10^{-15}$ cm <sup>2</sup>		
Energetic distribution	Gaussian	Gaussian	Gaussian	Gaussian	Gaussian
Energy level with respect to the highest $E_v$	$1.5 \text{ eV}$	1 eV	$0.75$ eV	0.6 eV	$0.75$ eV
Concentration			$1 \times 10^{15}$ cm <sup>-3</sup> $1 \times 10^{15}$ cm <sup>-3</sup> $2.3 \times 10^{15}$ cm <sup>-3</sup> $1 \times 10^{15}$ cm <sup>-3</sup> $1 \times 10^{15}$ cm <sup>-2</sup>		

<span id="page-7-0"></span>**Table 2** Defects parameters of the cell layers and at the interfaces



<span id="page-7-1"></span>**Fig. 2** Calibration of simulation versus experimental data (Shin et al. [2017b\)](#page-20-7) **a** *J–V* curves under AM1.5G illumination, and **b** EQE curves

# **3 Results and discussion**

In this section, we delve into the outcomes of our simulations and investigations, exploring various factors that infuence the performance of CBTSSe solar cells under diferent LED color temperatures. The design of the CBTSSe cell is based on identifying the infuence of the main factors of the cell on its performance so as to reach the best PCE. The optimization procedure is divided into three main steps concerning the interface between the photoactive layer and the ETL, the photoactive layer and the ETL. The following subsections highlight the key aspects examined in this study, shedding light on their efects on the PCE and characteristics of the PV cells. In the frst subsection, the role of absorber bandgap is explored to validate the appropriate bandgap value that best suits the LED spectra. In the second subsection, the interface parameters are investigated by simultaneously studying the infuence of the CBO and S between the absorber and ETL heterointerface. The subsequent subsection is devoted to the impact of absorber layer parameters on the performance through studying the thickness and bulk defect infuence. Further, the infuence of thickness and doping of the ETL is presented in the fourth subsection. Finally, the optimization phases are summarized in the last subsection, giving a comparison between each step, and explaining the physics beyond the behavior of the PV device.

<span id="page-8-0"></span>

<span id="page-8-1"></span>**Table 3** Main absorber and performance parameters of CBTSSe cell under diferent LED temperatures at optimum bandgap values



### **3.1 Role of absorber bandgap**

Here, we thoroughly examine the role of CBTSSe absorber bandgap under diferent LED color temperatures. Specifcally, we investigate the warm and cool LED efects by analyzing two cases: LED color temperatures of 2900 K and 7500 K. For the two cases, the illumination is taken to be fixed at 200 lx. Figure  $3$  illustrates the relationship between PCE and absorber bandgap across diferent LED temperatures. The fndings indicate that, for our particular design, an optimal bandgap of 1.70 eV aligns with the 2900 K LED spectrum, while a bandgap of 1.73 eV is most proper for the 7500 K LED spectrum. The subtle disparity between these two bandgap values suggests that any value within this range of  $E<sub>a</sub>$ (1.70–1.73 eV) could be efectively employed for both temperature scenarios. The main absorber and performance parameters of CBTSSe cell at optimum bandgap values are listed in Table [3](#page-8-1). It should be pointed out here that the color temperature of LED sources indicates the energy of their emitted photons, with higher temperatures yielding higherenergy photons. Solar cell efficiency relies on aligning the cell's bandgap energy with incident photon energy to avoid energy waste or inadequate electron–hole pair creation. To minimize parasitic effects, such as thermalization occurring for  $E_{photon} \gg E_g$ , it is crucial to match absorber bandgap with LED spectra, making high band gaps suitable for hightemperature LEDs and low band gaps for low-temperature LEDs.

Additionally, we proceed to generate the EQE curves for the initial CBTSSe cell, utilizing the optimized bandgap values, as depicted in Fig. [4](#page-9-0). This graph presents the EQE curves for the CBTSSe cell with the optimized bandgaps, under LED Spectra of 200 lx at 2900K and

<span id="page-9-0"></span>**Fig. 4** EQE curves of initial CBTSSe cell with optimized bandgaps under 200 lx LED Spectra at 2900K, and 7500K showing corresponding LED spectra



7500K, accompanied by the respective LED spectra. The fgure notably illustrates the favorable alignment of the optimized-bandgap CBTSSe material with the spectral range of LED illumination. This alignment underscores the efficacy of the chosen bandgaps in maximizing energy harvesting efficiency under both warm and cool LED conditions.

#### **3.2 Efect of interface parameters (CBO and S)**

The interplay between the conduction band offset (CBO) and surface recombination velocity (S) at the heterointerface of a solar cell is crucial for determining the device's performance. The heterointerface between the absorber and ETL plays a substantial role in charge carrier extraction, recombination, and overall device efficiency (Minemoto and Murata [2015](#page-20-16); Okil et al. [2023](#page-20-17)), and it is described as,

$$
CBO = \Delta E_c = \chi_{absorber} - \chi_{ETL}
$$
 (6)

where  $\chi_{\text{absorber}}$  and  $\chi_{ETL}$  are the affinities of the absorber and the ETL, respectively.

Figure [5](#page-10-0) shows two diferent band profles, representing diferent values of CBO. Figure [5](#page-10-0)a shows the clif band case, which does not deter the extraction of photoexcited electrons from the absorber to the ETL. Still, it alters the activation energy  $(E_a)$  that is related to the charge carrier recombination. In this scenario, the activation energy is less than the absorber bandgap  $(E_{\sigma})$  as can depicted in Fig. [5](#page-10-0)a. As a result, interfacial recombination losses become the dominant recombination process within the cell (Minemoto and Murata [2015](#page-20-16); Tanaka et al. [2009](#page-20-18)). Conversely, Fig. [5](#page-10-0)b demonstrates the formation of a spike-like band ofset at the ETL/ Absorber interface. This spike creates an obstacle to the extraction of the photoexcited electrons from the absorber towards the front electrode (Scheer and Schock [2011\)](#page-20-19). Nevertheless, when the spike is low enough, it no longer has any efect, and the electrons can fow smoothly towards the contact. An optimal CBO ensures balanced carrier injection and reduced recombination.

S quantifes the rate at which charge carriers recombine at the interface, and it is described as,

$$
S = \sigma N_t v_{th} \tag{7}
$$



<span id="page-10-0"></span>**Fig. 5** Energy band diagrams of two cases of CBOs **a** a clif-like band ofset, and **b** a spike-like band ofset

where  $\sigma$  is the carrier capture cross section,  $N_t$  is the interface defect density, and  $v_{th}$  is the thermal velocity. Higher S values indicate more rapid recombination, potentially leading to lower charge carrier extraction efficiency, while a lower S value is desirable as it extends the carrier lifetime and reduces recombination losses. Also, a lower S value allows carriers to reach the contacts before recombination occurs, leading to improved charge extraction efficiency and higher  $V_{OC}$ . An optimal combination of CBO and S is necessary to strike a balance between carrier injection and extraction. A positive CBO should be matched with a lower S value to minimize recombination at the heterointerface. The CBO-S interplay infuences the dynamics of charge carriers as they traverse the heterointerface. Proper alignment enhances carrier transport through the interface, reducing the probability of recombination.

This subsection delves into the infuence of CBO and S on the CBTSSe cell performance. We investigate these factors for two diferent LED spectra (LED 200 lx at 2900K and 7500K) and corresponding bandgap values. By mapping out contour plots of key PV parameters, including  $J_{SC}$ ,  $V_{OC}$ , FF, and PCE, we uncover the optimal ranges of CBO and S that yield the highest efficiency. In this theoretical investigation, CBO is changed from -0.3 to 0.3 eV, while S is varied from  $10^4$  to  $10^7$  cm/s. Figure [6](#page-11-0) shows contour plot of PV performance factors for CBTSSe cell under LED 200 lx at 2900K spectrum at diferent CBO and S values. Similarly, contour plots under LED 200 lx at 7500K spectrum are depicted in Fig. [7.](#page-12-0)

The investigation into the CBO and S reveals an optimum CBO range across various LED spectra, spanning from 0.1 to 0.3 eV. Within this specifed CBO range and at a constant S of  $10^5$  cm/s, the PCE of the CBTSSe cell demonstrates a slight enhancement. Specifcally, for the 2900K LED temperature, the PCE experiences an increment from 11.87 to 11.95% within this CBO range. In parallel, for the 7500K LED temperature, a similar trend is observed, albeit with slightly diferent numerical values. Here, the PCE transitions from 11.45 to 11.54% as the CBO varies within the prescribed optimal range. These results underscore the sensitivity of the CBTSSe solar cell performance to the CBO, particularly within the specifed CBO range, which showcases the direct correlation between CBO and PCE under diferent LED temperature conditions. Furthermore, at  $S = 10^5$  cm/s, we draw the interface recombination current density for three distinct CBO values, as indicated in Fig. [8.](#page-12-1) The results reveal that the recombination current density decreases signifcantly with optimum CBO values under diferent LED



<span id="page-11-0"></span>**Fig.** 6 Contour plots of PV metrics for CBTSSe cell: **a**  $J_{\text{SC}}$ , **b**  $V_{\text{OC}}$ , **c** FF and **d** PCE under LED 200 lx at 2900K spectrum at diferent CBO and S values

temperature conditions. This fnding confrms the importance of CBO engineering in reducing interfacial recombination.

As the optimum CBO range under different spectrums is from 0.1 to 0.3 eV,  $Cd_{0.4}Zn_{0.6}S$ can be utilized as an ETL instead of CdS because of its optimum CBO values of 0.14 eV and 0.11 eV under LED 200 lx at 2900K and 7500K, respectively. The simulation parameters of the ternary alloy  $Cd_{0.4}Zn_{0.6}S$  are given in Table [1](#page-6-0) (Salem et al. [2022b\)](#page-20-15). In this case, performance parameters of CBTSSe cell are a PCE of 12.06%,  $V_{OC}$  of 0.605 V,  $J_{SC}$ of 19.34  $\mu$ A/cm<sup>2</sup>, and FF of 59.65% under LED 200 lx at 2900K, while they are a PCE of 12.02%,  $V_{\text{OC}}$  of 0.639 V,  $J_{\text{SC}}$  of 21.94  $\mu$ A/cm<sup>2</sup>, and FF of 60.15% under LED 200 lx at 7500K.

#### **3.3 Efect of absorber parameters (thickness and bulk defects)**

The interplay between the thickness and defect concentration of the absorber layer in a PV cell is a crucial aspect that signifcantly afects the device's overall performance. Notably, the thickness of the absorber flm directly afects the amount of incident light that is absorbed. A thicker absorber can absorb more light leading to higher electron–hole generation and improved  $J_{SC}$ . However, if the absorber becomes too thick, it might lead to increased recombination losses, as charge carriers may not efficiently reach the contacts before recombining. In line with this, defects within the absorber flm act as recombination centers, facilitating charge carrier recombination. Higher defect densities can lead to



<span id="page-12-0"></span>**Fig. 7** Contour plots of PV metrics for CBTSSe cell: **a**  $J_{\text{SC}}$ , **b**  $V_{\text{OC}}$ , **c** FF and **d** PCE under LED 200 lx at 7500K spectrum at diferent CBO and S values



<span id="page-12-1"></span>**Fig. 8** Interface recombination current density for CBTSSe cell under LED 200 lx at **a** 2900K, and **b** 7500K

increased non-radiative recombination, reducing the carrier lifetime and thereby decreasing  $V_{OC}$  and FF. Lower defect densities are preferred to minimize recombination losses and enhance the overall cell performance. The interplay between thickness and defect density is particularly pronounced in their effects on  $V_{OC}$  and FF. Higher defect densities in a thicker absorber layer can cause more pronounced recombination, leading to a decrease in

 $V_{OC}$  due to a higher recombination current. Moreover, the decreased FF is often attributed to enhanced recombination-related voltage losses as carriers traverse the device. Balancing the thickness and defect density involves optimization to achieve the best compromise between enhanced light absorption and reduced recombination losses. It is essential to fnd the sweet spot where the trade-ofs between increased light absorption and the negative impact of defects on carrier dynamics are minimized.

Here, we explore the infuence of absorber thickness and bulk defect density on the CBTSSe cell's efficiency. Extending our analysis to both LED 200 lx spectra at 2900K and 7500K, we construct contour plots for  $J_{SC}$ ,  $V_{OC}$ , FF, and PCE, revealing the intricate interplay between these parameters. By selecting a representative thickness and defect density, we provide a detailed examination of the cell's performance under these conditions. In this theoretical investigation, defect density of the absorber layer is changed from  $10^{12}$  to  $10^{15}$ cm<sup>-3</sup>, while its thickness is varied from 0.5 to 5 µm. Figures [9](#page-13-0) and [10](#page-14-0) show contour plots of PV performance factors for the CBTSSe cell under LED 200 lx at 2900K and 7500K spectrum at diferent defect density and thickness values, respectively. The observed trends are consistent across various temperatures, as depicted in both fgures. The PCE demonstrates an upward rise with increasing thickness, holding a constant defect density level. However, the PCE tends to reach a nearly saturation point as the thickness continues to increase beyond a certain threshold. The optimum choice is selected to be a thickness of 2 μm and defect density of  $1 \times 10^{14}$  cm<sup>-3</sup>. In this case, PV parameters of CBTSSe cell are a PCE of 20.75%,  $V_{OC}$  of 0.796 V,  $J_{SC}$  of 21.59 μA/cm<sup>2</sup>, and FF of 69.95% under LED 200 lx



<span id="page-13-0"></span>**Fig. 9** Contour plots of PV metrics for CBTSSe cell: **a**  $J_{\text{SC}}$ , **b**  $V_{\text{OC}}$ , **c** FF and **d** PCE under LED 200 lx at 2900K spectrum at diferent thickness and bulk defect concentration of the absorber layer



<span id="page-14-0"></span>**Fig. 10** Contour plots of PV metrics for CBTSSe cell:  $\mathbf{a}$   $J_{SC}$ ,  $\mathbf{b}$   $V_{OC}$ ,  $\mathbf{c}$  FF and  $\mathbf{d}$  PCE under LED 200 lx at 7500K spectrum at diferent thickness and bulk defect concentration of the absorber layer

at 2900K, whereas they are a PCE of 18.36%,  $V_{OC}$  of 0.802 V,  $J_{SC}$  of 22.75  $\mu$ A/cm<sup>2</sup>, and FF of 70.51% under LED 200 lx at 7500K.

### **3.4 Efect of ETL parameters (thickness and doping)**

Notably, the thickness of the ETL directly afects the ease of charge transport within the layer. A thicker ETL might increase the distance that charge carriers need to travel, leading to a longer transit time and potentially higher recombination losses. Conversely, a thinner ETL can facilitate faster carrier extraction and transport to the external circuit, resulting in improved device performance. In addition, the doping density of the ETL determines the concentration of charge carriers (usually electrons) available for transport. Higher doping densities increase the density of free electrons, enhancing the electrical conductivity of the ETL. This can lead to improved carrier extraction efficiency and reduced resistive losses. Striking a balance between thickness and doping density is crucial to prevent excessive recombination. Very high doping densities can lead to increased trap-assisted recombination, especially if the ETL is too thick, as carriers might have a higher probability of interacting with defects and traps.

By creating contour plots of PCE dependence on thickness and doping of the ETL under LED 200 lx at both 2900K and 7500K, we uncover the optimized regions for these parameters. We present performance parameters for the CBTSSe cell at specifc thickness and doping levels, providing insights into the role of ETL characteristics in determining the



<span id="page-15-0"></span>**Fig. 11** Contour plot of PCE for CBTSSe cell at diferent thickness and doping of the ETL layer under LED 200 lx at **a** 2900K and **b** 7500K

cell's efficiency. Figure [10](#page-14-0) shows contour plots of PCE for CBTSSe cell under LED 200 lx at 2900K (Fig. [11a](#page-15-0)) and 7500K (Fig. [11b](#page-15-0)) spectrum at diferent thickness and doping values. As anticipated, the PCE exhibits higher values within regions characterized by thinner and more heavily doped ETL. At a thickness of 20 nm and a doping level of  $1 \times 10^{19}$ cm<sup>-3</sup>, performance parameters of CBTSSe cell are: PCE of 11.23%,  $V_{OC}$  of 0.605 V,  $J_{SC}$ of 18.54  $\mu$ A/cm<sup>2</sup>, and FF of 57.95% under LED 200 lx at 2900K, whereas they are a PCE of 10.70%,  $V_{OC}$  of 0.639 V,  $J_{SC}$  of 20.41  $\mu A/cm^2$ , and FF of 57.54% under LED 200 lx at 7500K.

### **3.5 Final optimization**

In the fnal subsection, we bring together the cumulative insights from the previous analyses and present the ultimate optimization stages for the CBTSSe solar cell. We compile performance parameters under optimal conditions in a comprehensive table, showcasing the improvements achieved. Calculated efficiency enhancements are presented as relative percentages under LED 200 lx spectra at 2900K and 7500K. We compare the illuminated *J–V* characteristics of the initial and final optimized CBTSSe cells, offering a visual representation of the enhanced performance. To further understand the underlying physics of the observed trends, we analyze the electric feld distribution under illuminated short-circuit conditions for both initial and optimized cells. This subsection concludes the section by summarizing the achieved optimizations and their implications for enhancing CBTSSe solar cell efficiency.

Table [4](#page-16-0) summarizes the optoelectronic parameters of the CBTSSe cell during various optimization stages under LED 200 lx spectra at the two studied color temperatures. Further, Fig. [12a](#page-16-1) and b depicts PCE of the diferent optimization stages and the enhancement percentage in PCE over the initial stage. Based on the results, the most substantial factors, that infuence cell performance, among the various factors are the defect concentration and thickness of the photoactive layer, which provide 106.47% and 102.20% enhancement relative percentages under LED 200 lx spectra at 2900K and 7500K, respectively.

In addition, Fig. [13](#page-16-2)a and b display the illuminated *J–V* characteristics curves of initial and fnal optimized CBTSSe cells under LED 200 lx spectra at 2900K and 7500K. The signifcant enhancement in cell performance can be accredited to the comprehensive



<span id="page-16-0"></span>



<span id="page-16-1"></span>Fig. 12 Calculated efficiency improvements for CBTSSe cell showing the enhancement relative percentage under LED 200 lx at **a** 2900K, and **b** 7500K



<span id="page-16-2"></span>**Fig. 13** A comparison between illuminated *J–V* curves of initial and fnal optimized CBTSSe cells under LED 200 lx at **a** 2900K, and **b** 7500K

optimization procedures that led to achieving remarkable PCEs of 25.58% and 25%, accompanied by relative percentage improvements of 154.53% and 175.33%, compared to the initial cell under LED 200 lx spectra at 2900K and 7500K, respectively. To further give a physical explanation about these improvements in cell performance, the electric feld distribution is plotted under illuminated and short-circuit conditions for initial and fnal optimized CBTSSe cells, as depicted in Fig. [14a](#page-17-0) (for the case of 2900K) and Fig. [14b](#page-17-0) (for the case of 7500K). As illustrated in the fgure, according to the optimization steps, the electric feld is substantially increased, leading to accelerated separation of photoexcited carriers, reduced recombination rates, and enhanced cell performance. Moreover, Fig. [15a](#page-18-0) and b display the energy band profles of fnal optimized CBTSSe cells at short circuit current condition under LED 200 lx at 2900K, and 7500K, respectively.

# **4 Conclusion**

In conclusion, this study has systematically examined the feasibility of employing CBTSSe-based solar cells with wide bandgaps for indoor applications. The unique tunable bandgap property of CBTSSe was harnessed to tailor the absorber material's



<span id="page-17-0"></span>**Fig. 14** Electric feld distribution plotted under illuminated short-circuit condition for an initial CBTSSe cell under LED 200 lx at **a** 2900K, and **b** 7500K and the fnal optimized CBTSSe cell under LED 200 lx at **c** 2900K, and **d** 7500K



<span id="page-18-0"></span>**Fig. 15** Energy band profle at illumination and short circuit condition of fnal optimized CBTSSe cells under LED 200 lx at **a** 2900K, and **b** 7500K

characteristics within the  $1.7-1.9$  eV range, aligning aptly with the spectral emission of commonly used indoor LED lighting. Our investigations encompassed a meticulous design of the solar cell device structure, which was validated through simulation against experimental results, thus ensuring the fdelity of our models and the profciency of the software employed. Throughout the study, a comprehensive exploration of factors infuencing CBTSSe solar cell performance under varying LED spectra was undertaken. We meticulously delved into the role of absorber bandgap under warm and cool LED illumination conditions to determine the suitable bandgap range for various LED environment. For our particular design, an optimal bandgap of 1.70 eV with a PCE of 10.05% aligns with the 2900 K LED spectrum, while a bandgap of 1.73 eV with a PCE of 9.08% is most suitable for the 7500 K LED spectrum. Subsequent to this analysis, our focus extended to a three-step optimization procedure, each dedicated to refning a specifc facet of the solar cell design. These phases entailed the intricate management of interface parameters, the meticulous adjustment of absorber layer properties, and the astute manipulation of ETL attributes.

Our fndings demonstrated the criticality of interface parameters; with an optimum CBO range of 0.1 to 0.3 eV, the PCE was enhanced up to 12.06% under the warm LED spectrum and 12.02% under the cool LED spectrum. The thickness and bulk defect density of the photoactive flm were also found to play pivotal roles, showcasing the potential for efficiency improvements, resulting in a PCE of 20.75% under the warm LED spectrum and 18.36% under the cool LED spectrum. Moreover, adjustments to the ETL layer parameters exhibited tangible gains, with the potential for PCE improvements, resulting a PCE of 11.23% under the warm LED spectrum and 10.70% under the cool LED spectrum. Owing to various optimization steps, the cell performance attains a maximum PCEs of 25.58% and 25% with a relative percentage enhancement of 154.53% and 175.33% compared with the initial cell under LED 200 lx spectra at 2900K and 7500K, respectively. In essence, this study has provided a comprehensive understanding of the interplay between material parameters and device performance in CBTSSe-based solar cells for indoor applications.

Furthermore, CBTSSe offers the advantage of being compatible with cost-effective fabrication techniques, making it a promising candidate for scalable indoor PV applications. As the demand for sustainable energy solutions in indoor environments, such as

smart buildings, wearables, and sensor networks, continues to grow, adaptable bandgap properties of CBTSSe make it an attractive choice for meeting the unique requirements of these applications.

**Acknowledgements** This research was funded by the Researchers supporting project number (RSPD2023R762), King Saud University, Riyadh, Saudi Arabia.

**Authors' contribution** SQ, AS, and MO wrote the main manuscript text. WZ, MO, MA, and AAAA performed the simulation and prepared the fgures. All authors shared in conceptualization, methodology, and validation. All authors reviewed the manuscript.

**Funding** This research was funded by the Researchers supporting project number (RSPD2023R762), King Saud University, Riyadh, Saudi Arabia.

**Data availability** Data will be made available on reasonable request.

## **Declarations**

**Confict of interest** The authors declare that there is no confict of interest reported in this paper.

**Ethical approval** Not Applicable.

# **References**

- <span id="page-19-3"></span>Ait Abdelkadir, A., Sahal, M., Oublal, E., Kumar, N., Benami, A.: New Sb<sub>2</sub>Se<sub>3</sub>-based solar cell for achieving high efficiency theoretical modeling. Opt. Quantum Electron. **55**, 514 (2023). [https://doi.org/10.1007/](https://doi.org/10.1007/s11082-023-04797-7) [s11082-023-04797-7](https://doi.org/10.1007/s11082-023-04797-7)
- <span id="page-19-6"></span>Chakraborty, R., Sim, K.M., Shrivastava, M., Adarsh, K.V., Chung, D.S., Nag, A.: Colloidal synthesis, optical properties, and hole transport layer applications of Cu<sub>2</sub>BaSnS<sub>4</sub> (CBTS) nanocrystals. ACS Appl. Energy Mater. **2**, 3049–3055 (2019). <https://doi.org/10.1021/acsaem.9b00473>
- <span id="page-19-8"></span>Cui, X., Sun, K., Huang, J., Lee, C.-Y., Yan, C., Sun, H., Zhang, Y., Liu, F., Hossain, M.A., Zakaria, Y., Wong, L.H., Green, M., Hoex, B., Hao, X.: Enhanced heterojunction interface quality to achieve 93% efficient Cd-free Cu2ZnSnS4 solar cells using atomic layer deposition ZnSnO bufer layer. Chem. Mater. **30**, 7860– 7871 (2018). <https://doi.org/10.1021/acs.chemmater.8b03398>
- <span id="page-19-2"></span>Farooq, W., Alshahrani, T., Kazmi, S.A.A., Iqbal, J., Khan, H.A., Khan, M., Raja, A.A., ur Rehman, A.: Materials optimization for thin-flm copper indium gallium selenide (CIGS) solar cell based on distributed braggs refector. Optik **227**, 165987 (2021).<https://doi.org/10.1016/j.ijleo.2020.165987>
- <span id="page-19-1"></span>Gamal, N., Sedky, S.H., Shaker, A., Fedawy, M.: Design of lead-free perovskite solar cell using  $Zn_{1-x}Mg_xO$  as ETL: SCAPS device simulation. Optik **242**, 167306 (2021). <https://doi.org/10.1016/j.ijleo.2021.167306>
- <span id="page-19-7"></span>Ge, J., Roland, P.J., Koirala, P., Meng, W., Young, J.L., Petersen, R., Deutsch, T.G., Teeter, G., Ellingson, R.J., Collins, R.W., Yan, Y.: Employing overlayers to improve the performance of  $Cu<sub>2</sub>BaSnS<sub>4</sub>$  Thin film based photoelectrochemical water reduction devices. Chem. Mater. **29**, 916–920 (2017a). [https://doi.org/10.](https://doi.org/10.1021/acs.chemmater.6b03347) [1021/acs.chemmater.6b03347](https://doi.org/10.1021/acs.chemmater.6b03347)
- <span id="page-19-9"></span>Ge, J., Koirala, P., Grice, C.R., Roland, P.J., Yu, Y., Tan, X., Ellingson, R.J., Collins, R.W., Yan, Y.: Oxygenated CdS buffer layers enabling high open-circuit voltages in earth-abundant  $Cu$ ,  $BaSnS<sub>4</sub>$  thin-film solar cells. Adv. Energy Mater. **7**, 1601803 (2017b).<https://doi.org/10.1002/aenm.201601803>
- <span id="page-19-10"></span>Ghobadi, A., Yousef, M., Minbashi, M., Kordbacheh, A.H.A., Abdolvahab, A.H., Gorji, N.E.: Simulating the effect of adding BSF layers on Cu<sub>2</sub>BaSnSSe<sub>3</sub> thin film solar cells. Opt. Mater. **107**, 109927 (2020). [https://](https://doi.org/10.1016/j.optmat.2020.109927) [doi.org/10.1016/j.optmat.2020.109927](https://doi.org/10.1016/j.optmat.2020.109927)
- <span id="page-19-5"></span>Gohri, S., Madan, J., Pandey, R., Sharma, R.: Design and analysis of lead-free perovskite-CZTSSe based tandem solar cell. Opt. Quantum Electron. **55**, 171 (2023). <https://doi.org/10.1007/s11082-022-04381-5>
- <span id="page-19-4"></span>Green, M., Dunlop, E., Hohl-Ebinger, J., Yoshita, M., Kopidakis, N., Hao, X.: Solar cell efficiency tables (version 57). Prog. Photovolt. Res. Appl. **29**, 3–15 (2021). <https://doi.org/10.1002/pip.3371>
- <span id="page-19-0"></span>Green, M.A., Dunlop, E.D., Siefer, G., Yoshita, M., Kopidakis, N., Bothe, K., Hao, X.: Solar cell efficiency tables (Version 61). Prog. Photovolt. Res. Appl. **31**, 3–16 (2023).<https://doi.org/10.1002/pip.3506>
- <span id="page-19-11"></span>Hashemi, M., Minbashi, M., Ghorashi, S.M.B., Ghobadi, A.: A modeling study on utilizing low temperature sprayed In<sub>2</sub>S<sub>3</sub> as the buffer layer of CuBaSn(S, Se) solar cells. Sci. Rep. 11, 20038 (2021). [https://doi.org/](https://doi.org/10.1038/s41598-021-99012-6) [10.1038/s41598-021-99012-6](https://doi.org/10.1038/s41598-021-99012-6)
- <span id="page-20-9"></span>Henni, W., Rahal, W.L., Rached, D., Boukortt, A.: Effect of introducing  $A<sub>1</sub>O<sub>3</sub>$  as a tunnelling layer into p-CBTS/n-CdS heterojunction solar cells. J. Comput. Electron. **22**, 897–905 (2023). [https://doi.org/10.](https://doi.org/10.1007/s10825-023-02031-x) [1007/s10825-023-02031-x](https://doi.org/10.1007/s10825-023-02031-x)
- <span id="page-20-2"></span>Huang, X.: Design and development of high performance III-nitrides photovoltaics. PhD dissertation, Arizona State University (2020)
- <span id="page-20-3"></span>Kumar, A.: Theoretical analysis of CZTS/CZTSSe tandem solar cell. Opt. Quantum Electron. **53**, 528 (2021). <https://doi.org/10.1007/s11082-021-03183-5>
- <span id="page-20-16"></span>Minemoto, T., Murata, M.: Theoretical analysis on effect of band offsets in perovskite solar cells. Sol. Energy Mater. Sol. Cells **133**, 8–14 (2015).<https://doi.org/10.1016/j.solmat.2014.10.036>
- <span id="page-20-14"></span>Moustafa, M., Al Zoubi, T., Yasin, S.: Numerical analysis of the role of p-MoSe, interfacial layer in CZTSe thin-flm solar cells using SCAPS simulation. Optik **247**, 167885 (2021). [https://doi.org/10.1016/j.ijleo.](https://doi.org/10.1016/j.ijleo.2021.167885) [2021.167885](https://doi.org/10.1016/j.ijleo.2021.167885)
- <span id="page-20-1"></span>Okil, M., Salem, M.S., Abdolkader, T.M., Shaker, A.: From crystalline to low-cost silicon-based solar cells : a review. Silicon **14**, 1895–1911 (2022).<https://doi.org/10.1007/s12633-021-01032-4>
- <span id="page-20-17"></span>Okil, M., Shaker, A., Ahmed, I.S., Abdolkader, T.M., Salem, M.S.: Design and analysis of Sb<sub>2</sub>S<sub>3</sub>/Si thin film tandem solar cell. Sol. Energy Mater. Sol. Cells **253**, 112210 (2023). [https://doi.org/10.1016/j.solmat.](https://doi.org/10.1016/j.solmat.2023.112210) [2023.112210](https://doi.org/10.1016/j.solmat.2023.112210)
- <span id="page-20-0"></span>Petrović-Ranđelović, M., Kocić, N., Stojanović-Ranđelović, B.: The importance of renewable energy sources for sustainable development. Econ. Sustain. Dev. **4**, 15–24 (2020).<https://doi.org/10.5937/esd2002016p>
- <span id="page-20-11"></span>Salah, M.M., Hassan, K.M., Abouelatta, M., Shaker, A.: A comparative study of diferent ETMs in perovskite solar cell with inorganic copper iodide as HTM. Optik (stuttg). **178**, 958–963 (2019). [https://doi.org/10.](https://doi.org/10.1016/j.ijleo.2018.10.052) [1016/j.ijleo.2018.10.052](https://doi.org/10.1016/j.ijleo.2018.10.052)
- <span id="page-20-12"></span>Salem, M.S., Shaker, A., Abouelatta, M., Alanazi, A., Al-Dhlan, K.A., Almurayziq, T.S.: Numerical analysis of hole transport layer-free antimony selenide solar cells: possible routes for efficiency promotion. Opt. Mater. **129**, 112473 (2022a).<https://doi.org/10.1016/j.optmat.2022.112473>
- <span id="page-20-15"></span>Salem, M.S., Shaker, A., Almurayziq, T.S., Alshammari, M.T.: Prospective efficiency boosting of full-inorganic single-junction Sb<sub>2</sub>(S, Se)<sub>3</sub> solar cell. Sol. Energy Mater. Sol. Cells 248, 112001 (2022b). [https://doi.org/](https://doi.org/10.1016/j.solmat.2022.112001) [10.1016/j.solmat.2022.112001](https://doi.org/10.1016/j.solmat.2022.112001)
- <span id="page-20-19"></span>Scheer, R., Schock, H.W.: Chalcogenide Photovoltaics: Physics, Technologies, and Thin Film Devices. Wiley (2011).<https://doi.org/10.1002/9783527633708>
- <span id="page-20-4"></span>Shaf, M.A., Bouich, A., Khan, L., Ullah, H., Guaita, J.M., Ullah, S., Mari, B.: Optimization of electrodeposition time on the properties of  $Cu<sub>2</sub>ZnSnS<sub>4</sub>$  thin films for thin film solar cell applications. Opt. Quantum Electron. **54**, 533 (2022).<https://doi.org/10.1007/s11082-022-03913-3>
- <span id="page-20-6"></span>Shin, D., Saparov, B., Mitzi, D.B.: Defect engineering in multinary earth-abundant chalcogenide photovoltaic materials. Adv. Energy Mater. **7**, 1602366 (2017a).<https://doi.org/10.1002/aenm.201602366>
- <span id="page-20-7"></span>Shin, D., Zhu, T., Huang, X., Gunawan, O., Blum, V., Mitzi, D.B.: Earth-abundant chalcogenide photovoltaic devices with over 5% efficiency based on a Cu<sub>2</sub>BaSn(S,Se)<sub>4</sub> absorber. Adv. Mater. **29**, 1606945 (2017b). <https://doi.org/10.1002/adma.201606945>
- <span id="page-20-5"></span>Sun, J.-P., Márquez, J.A., Stange, H., Mainz, R., Mitzi, D.B.: Phase and flm formation pathway for vacuumdeposited Cu2BaSn(S,Se)4 absorber layers. Phys. Rev. Mater. **3**, 55402 (2019)
- <span id="page-20-18"></span>Tanaka, K., Minemoto, T., Takakura, H.: Analysis of heterointerface recombination by  $\rm Zn_{1-x}Mg_xO$  for window layer of Cu(In, Ga)Se<sub>2</sub> solar cells. Sol. Energy 83, 477–479 (2009). [https://doi.org/10.1016/j.solener.2008.](https://doi.org/10.1016/j.solener.2008.09.003) [09.003](https://doi.org/10.1016/j.solener.2008.09.003)
- <span id="page-20-13"></span>Teymur, B., Levcenco, S., Hempel, H., Bergmann, E., Márquez, J.A., Choubrac, L., Hill, I.G., Unold, T., Mitzi, D.B.: Optoelectronic and material properties of solution-processed earth-abundant  $Cu_2BaSn(S, Se)_4$  films for solar cell applications. Nano Energy **80**, 105556 (2021).<https://doi.org/10.1016/j.nanoen.2020.105556>
- <span id="page-20-8"></span>Teymur, B., Kim, Y., Huang, J., Sun, K., Hao, X., Mitzi, D.B.: Top stack optimization for Cu<sub>2</sub>BaSn(S,Se)<sub>4</sub> photovoltaic cell leads to improved device power conversion efficiency beyond 6%. Adv. Energy Mater. 12, 2201602 (2022).<https://doi.org/10.1002/aenm.202201602>
- <span id="page-20-10"></span>Wang, W., Winkler, M.T., Gunawan, O., Gokmen, T., Todorov, T.K., Zhu, Y., Mitzi, D.B.: Device characteristics of CZTSSe thin-film solar cells with 12.6% efficiency. Adv. Energy Mater. 4, 1301465 (2014). [https://](https://doi.org/10.1002/aenm.201301465) [doi.org/10.1002/aenm.201301465](https://doi.org/10.1002/aenm.201301465)

**Publisher's Note** Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.

Springer Nature or its licensor (e.g. a society or other partner) holds exclusive rights to this article under a publishing agreement with the author(s) or other rightsholder(s); author self-archiving of the accepted manuscript version of this article is solely governed by the terms of such publishing agreement and applicable law.

# **Authors and Afliations**

Saif M. H. Qaid<sup>1</sup> · Ahmed Shaker<sup>2</sup> · Mohamed Okil<sup>3</sup> · Muath Alkadi<sup>4</sup> · **Abdullah Ahmed Ali Ahmed5 · Walid Zein2**

 $\boxtimes$  Ahmed Shaker ahmed.shaker@eng.asu.edu.eg

> Saif M. H. Qaid sqaid@ksu.edu.sa

Mohamed Okil mohamed.okil@bhit.bu.edu.eg

Muath Alkadi malkadi@kacst.edu.sa

Abdullah Ahmed Ali Ahmed abdullah2803@gmail.com

Walid Zein waleedzein@eng.asu.edu.eg

- <sup>1</sup> Department of Physics and Astronomy, College of Sciences, King Saud University, P.O. Box 2455, 11451 Riyadh, Saudi Arabia
- <sup>2</sup> Department of Engineering Physics and Mathematics, Faculty of Engineering, Ain Shams University, Cairo, Egypt
- <sup>3</sup> Department of Basic Engineering Sciences, Benha Faculty of Engineering, Benha University, Benha, Egypt
- <sup>4</sup> King Abdulaziz City for Science and Technology, Riyadh, Saudi Arabia
- <sup>5</sup> Center for Hybrid Nanostructures (CHyN) and Fachbereich Physik, Universität Hamburg, 20146 Hamburg, Germany