

# **Synthesis and characterization of tin (IV) oxide thin flms**

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# **Abstract**

Tin  $(IV)$  oxide, SnO<sub>2</sub> films have been successfully synthesized in argon gas using a magnetron sputtering device. The morphology, structure, optical, photoluminescence, and photoresponse features of the samples have been analyzed via feld electron scanning electron microscope, X-ray difractograms, UV–Vis spectrometer, and spectro fuorophotometer. Compact nano grained morphologies with tetragonal structure and high absorbance were obtained. Increasing the annealing temperature led to a slight rise in the bandgap energies of the deposited samples.  $SnO<sub>2</sub>$  films exhibited good photoluminescence features with increasing photoresponse with time as the annealing temperature reduced. The flms can be potentially applied to optical and solar cell devices.

# **Graphic abstract**



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# **1 Introduction**

Transparent conducting oxides (TCOs) have been researched because they exhibit combined optical and electrical features that grant them useful access in batteries, sensing devices, optoelectronics, solar cells, and photocatalyst devices (Batzill and Diebold [2005](#page-9-0)). Tin (IV) oxide is an important TCO that undergoes phase transition during the synthesis process, has a transparent conducting surface, great sensitivity, and a useful catalyst during oxidation processes (Batzill and Diebold [2005](#page-9-0)). The different forms of fabricating  $SnO<sub>2</sub>$ sensors could be as whiskers, pallets, thick or thin flms (Mitra and Mondal [2008\)](#page-10-0). Tin (IV) oxide is an n-type semiconducting material with good stability, oxidation state of  $+4$ , has a wide and direct bandgap (Eifert et al. [2017](#page-10-1)), high conductivity (Kılıç and Zunger [2002](#page-10-2)), great optical features (Kang [2010\)](#page-10-3), and can exist in the tetragonal or orthorhombic phase. It fnds useful application in optical devices, solar cells, gas sensors, perovskite cells, and dye-sensitized solar cells (Kumara et al. [2001\)](#page-10-4).

Several methods of synthesizing tin (IV) oxide flms include chemical bath deposition (Amma et al. [2005](#page-9-1)), spray deposition (Baranauskas et al. [2005;](#page-9-2) Kasar et al. [2008;](#page-10-5) Chacko [2006;](#page-9-3) Thangaraju [2002\)](#page-11-0), successive ionic layer adsorption and reaction (Mitra and Mon-dal [2008](#page-10-0); Deshpande et al. [2008;](#page-9-4) Pusawale et al. [2011](#page-10-6)), evaporation technique (Geurts et al. [1984](#page-10-7); Jaiswal et al. [2013](#page-10-8)), sequential infltration synthesis (Barick et al. [2019\)](#page-9-5), ion beam irradiation (Kang [2010](#page-10-3)), chemical vapor deposition (Nagirnyak et al. [2016](#page-10-9); Naeem et al. [2015](#page-10-10)), magnetron sputtering (Adamchuk et al. [2019;](#page-9-6) Chub et al. [2020](#page-9-7)), atomic layer deposition (Mai [2019](#page-10-11); Maximov [2017](#page-10-12)). The diverse synthesis methods infuence the quality and properties exhibited by the flms. Sputter deposition allows flm materials to be ejected from a target to a substrate with minimal heating efects. Sputtered flms adhere more on substrate surfaces, sputter materials of high melting point, and produce flms with similar compositions as the source material. Annealing flms improve the crystal structure, enhances surface features, reduces strain, and improves optical features (Nkele [2019](#page-10-13)). Tin (IV) oxide flms are usefully applied in optoelectronic, catalytic devices (Barick et al. [2019\)](#page-9-5), and sensors (Chub et al. [2020](#page-9-7)).

Several works have been carried out on the synthesis of  $SnO<sub>2</sub>$  by the sputtering technique. Sangaletti et al. ([1997\)](#page-10-14) thermally treated tin flms in the air via RF sputtering to obtain a mixed orthorhombic and tetragonal phase. Camacho-Lopéz ([2013\)](#page-9-8) characterized reactive DC-sputtered  $SnO<sub>2</sub>$  film and obtained tetragonal-phased and high transmittance flms. The purpose of this research is to synthesize and characterize the obtained morphologies, structure, optical, and photoluminescence characteristics of the sputtered tin (IV) oxide flms.

## **2 Experimental details**

#### **2.1 Materials and methods**

The  $SnO<sub>2</sub>$  material was synthesized on the glass substrate via an RF magnetron sputtering device.  $SnO<sub>2</sub>$  ceramic target (Kurt. J. Lesker, 99.99% pure) and 2 mm diameter with 6 mm thickness was deposited by RF power supply. First, the substrates were cleaned

intensively before coating. The layering stage was performed in the argon (99.99% pure) atmosphere. The chamber was frst vacuumed by a turbo molecular pump to an initial pressure of  $5 \times 10^{-5}$  torr while the working pressure was kept constant at  $4.5 \times 10^{-3}$  torr. Before starting the coating process, the chamber was fashed thrice with argon (Ar) gas to eliminate oxygen and other contaminants from the chamber. A thin film of  $SnO<sub>2</sub>$  with a power of 80 watts was layered to a thickness of 100 nm. After coating, the samples were heated at 300 °C (S3), 400 °C (S2) and 500 °C (S1) for 2 h in the air atmosphere. One sample (S4) was left without heating treatment.

The Au metal interdigital electrodes (IDEs) were deposited on the prepared  $SnO<sub>2</sub>$  samples by the sputtering technique as schematically illustrated in Fig. [1](#page-2-0). The thickness and active area of the patterned IDEs were kept constant at  $100 \text{ nm}$  and  $3.5 \text{ mm}^2$ , respectively.

#### **2.2 Characterizations**

The synthesized tin (IV) oxide flms were respectively analyzed to understand the morphology, structure, optical, photoluminescence, and photoresponse features using feld electric scanning electron microscope (FESEM) (Sigma 300-HV Zeiss), X-ray difraction (XRD) (ADVANCE-D8 Bruker) equipped with Cu<sub>kα</sub> radiation source with  $\lambda = 1.5406$  Å, UV–Vis spectrometer (1800 UV/Vis SHIMADZU), Spectro fuorophotometer (RF-6000 SHIMADZU). I–V features of the flms were obtained at room temperature and atmospheric pressure. The LED was blue with  $0.11 \text{ W/m}^2$  intensity with 2 min OFF/ON switching cycles under 5 V bias voltage.

# **3 Results and discussion**

#### **3.1 Morphological studies**

FESEM images revealed tiny clusters of nanoparticles distributed over the substrate surface, as seen in Fig. [2](#page-3-0). Similar tiny morphology has also been reported by Barick et al. ([2019\)](#page-9-5). Synthesizing the  $SnO<sub>2</sub>$  film produced clustered nano grains evenly distributed with surface cracks. The surface cracks are usually attributed to  $SnO<sub>2</sub>$  films because developing oxide flms leads to stress development in the flms. The cracks could also be due to

<span id="page-2-0"></span>



<span id="page-3-0"></span>**Fig. 2** FESEM images of the flms synthesized at **a** 500 **b** 400 **c** 300 **d** 0 °C

deposition conditions, low surface porosity, and difusion of oxygen into the pores of the oxide layer (Camacho-López et al. [2013\)](#page-9-8).

A cross-section of the FESEM images for the synthesized flms has been shown in Fig. [3a](#page-3-1)–d. Uniform flm distribution can be observed throughout the substrate surface



<span id="page-3-1"></span>**Fig. 3** Cross-sectional FESEM images of the flms synthesized at **a** 500 **b** 400 **c** 300 **d** 0 °C



<span id="page-4-0"></span>**Fig. 4** AFM images of the samples synthesized at **a** 500 **b** 400 **c** 300 **d** 0 °C



<span id="page-4-1"></span>**Table 1** Roughness values for the deposited films

with efficient material build-up. The most annealed film recorded the highest optical transparency due to the applied post-heating efect. This property makes the flms useful in optical devices and light-emitting diodes (Kang [2010\)](#page-10-3).

Atomic force microscope images of the flms obtained at several temperatures in tapping modes are shown in Fig. [4](#page-4-0)a–d. Round homogenous bulges were seen as agglomerations on the substrate surface. Spikes of diferent densities emanating from the substrate surface can be observed at varying temperatures. Varying the annealing temperature signifcantly afected the density of the tin (IV) oxide nanostructures. Thickness values of 240 nm, 219 nm, 210 nm, and 200 nm were obtained for the  $SnO<sub>2</sub>$  films annealed respectively at 0 °C, 300 °C, 400 °C, and 500 °C with the help of Digimizer software program. The flm's thickness decreased with increasing annealing temperature because the thermal energy overcomes the adhesion and bonding energies of loosely bound atoms, thereby reducing the number of atoms that are adsorbed on the surface.

The roughness values for the samples were obtained using Eqs.  $(1)$  $(1)$  and  $(2)$  $(2)$  and are displayed in Table [1.](#page-4-1)

<span id="page-5-0"></span>
$$
R_a = \frac{1}{n} \sum y_i
$$
 (1)

<span id="page-5-1"></span>
$$
R_q = \sqrt{\frac{1}{n} \sum y_i^2}
$$
 (2)

It can be observed that the unannealed flm recorded the maximum roughness value while the film annealed at 500  $^{\circ}$ C had the least roughness value. Increasing the annealing temperature led to more growth of the nanoclusters, and accounts for the thickness variations.

### **3.2 Structural analysis**

Figure  $5$  displays a tetragonal crystalline structure of the SnO<sub>2</sub> films at 2theta degrees ranging from 10° to 80°. The lattice planes of the X-ray difractograms and their corresponding 2theta angles are 26.57° (1 1 0), 33.86° (1 0 1), 37.94° (2 0 0), 51.76° (2 1 1), 54.74° (2 2 0), 57.81° (0 0 2), 61.86° (3 1 0), 64.72° (1 1 2), 71.26° (2 0 2), 78.68° (3 2 1). Tamilalagan et al. [\(2020](#page-10-15)) obtained similar lattice planes. The non-distinct peaks were indicative of sputtered tin (IV) oxide films (Popovich et al. [2016\)](#page-10-16). Applying low and high temperatures to tin oxide produced poorly crystalline flms (Singh [2019](#page-10-17)). Annealing the flms led to a



<span id="page-5-2"></span>**Fig. 5** Structural patterns of the  $SnO<sub>2</sub>$  films synthesized at varying temperatures

transformation to the polycrystalline tin (IV) oxide phase, reduction of water content, and stabilization the TiO<sub>2</sub> structure (Adamchuk et al. [2019\)](#page-9-6). The structural parameters obtained from the most prominent peaks have been outlined in Table [2.](#page-6-0)

Table [2](#page-6-0) showed that increasing the annealing temperature reduced the crystallite size, D and interplanar distance, d. The decreasing crystallite size could be attributed to the high density of the localized state. The lattice constants, a and c have also been outlined in Table [2](#page-6-0), where the lattice constant a equals that of b.

#### **3.3 Optical studies**

Figure [6](#page-7-0) gives the optical transmittance, absorbance, and refectance plots for the deposited flms. The deposited flms recorded high transmittance and absorbance. The flms were transparent to light in the visible electromagnetic spectrum. Reduced and fuctuating refectance values were observed at increasing wavelength regions. Annealing the flms reduced the refectance of the flms in the visible spectrum. Similar transmittance and absorbance features have been reported in the literature (Singh [2019](#page-10-17); Ivanova [2020\)](#page-10-18). These optical features make the  $SnO<sub>2</sub>$  films potential materials for solar cells.

The absorption coefficient versus wavelength plot of the  $SnO<sub>2</sub>$  films are exhibited in Fig. [7a](#page-7-1). A similar absorption coefficient trend was obtained with the most annealed film recording the highest absorption. Figure [7](#page-7-1)b shows the bandgap energy plots of the synthesized flms. Tauc plot was employed in determining the bandgap energies of the flms. The bandgap energy values of the flms increased from 3.18 to 3.21 eV as the annealing temperature increased, due to confnement of the electrons and holes. Subjecting the flms to annealing conditions did not signifcantly alter the band structure of the flms (Bazargan et al. [2012\)](#page-9-9). Similar results have also been obtained (Sangaletti [1997](#page-10-14); Kong et al. [2010](#page-10-19)).

## **3.4 Photoluminescence (PL) studies**

Photoluminescence analysis is a non-destructive technique that gives information on the optical, point defects, and photochemical features of the flms under study. Figure [8](#page-8-0) provides the photoluminescence plot of the unannealed and annealed  $SnO<sub>2</sub>$  films measured at a PL wavelength of 200 nm. The unannealed sample exhibited higher photoluminescence quenching than the annealed samples. Maximum PL emission occurred at 369 nm within the visible emission region. Strong emission bands were obtained at 369 nm and 766 nm, while weak emission bands were visible at 470 nm, 626 nm, and 739 nm. The unannealed sample exhibited lower luminous intensity than the annealed samples; thereby making the heat-treated  $SnO<sub>2</sub>$  films suitable for optical materials. The role of oxygen vacancies in PL analysis is evident from the decrease in PL intensity (Nehru et al. [2012\)](#page-10-20).

<span id="page-6-0"></span>



<span id="page-7-0"></span>**Fig. 6** Optical **a** transmittance, **b** absorbance, and **c** refectance plots of the synthesized samples



<span id="page-7-1"></span>Fig. 7 Plots showing the a absorption coefficient and **b** bandgap energies of the samples

# **3.5 Photoresponse analysis**

The photoresponse (current versus time) plots of the flms are shown in Fig. [9](#page-8-1). Figure [9](#page-8-1) depicts the time-dependent photoresponse behavior when the devices are under darkness and irradiation by blue lights with 2 min OFF/ON switching cycles under 5 V bias voltage.

**500 oC**

<span id="page-8-0"></span>

<span id="page-8-1"></span>

<span id="page-8-2"></span>The more the annealing temperature, the better the photoresponse of the flms (Breddels and Blasse [1984](#page-9-10)). The flm annealed at 500 °C recorded the best photoresponse as outlined in Table [3.](#page-8-2)

The photocurrent  $I_{ph}$  is defined in Eq. ([3\)](#page-8-3) (Tian and Fan [2018\)](#page-11-1) as:

<span id="page-8-3"></span>
$$
I_{ph} = I_{\text{illustrated}} - I_{\text{dark}} \tag{3}
$$

Two parameters of photodetector are calculate (Zhong [2017](#page-11-2)):

$$
Photo responsibility (R) = I_{ph}/PS
$$
 (4)

$$
Detectivity(D*) = [(RS)1/2]/[(2eIdark)1/2]
$$
 (5)

where  $I_{ph}$  is the photocurrent, P is the light power intensity, and S is the effective exposure area of the photodetector, and  $I_{dark}$  is the dark current. The results obtained are displayed in Table [1](#page-4-1).

Table [3](#page-8-2) shows increased photoresponse of the tin (IV) oxide flms at the higher annealing temperatures. The sample annealed at  $400^{\circ}$ C recorded the highest photodetection ability because more photogenerated carriers were created at that temperature. This increased sensitivity makes  $SnO<sub>2</sub>$  films useful as sensors (Bazargan et al.  $2012$ ; Savaniu [1999](#page-10-21)).

# **4 Conclusion**

This work successfully synthesized SnO<sub>2</sub> films unannealed and annealed at 300 °C, 400 °C, and 500 °C via the RF magnetron sputtering technique. The flms were characterized for their morphological, structural, optical, photoluminescence, and photoresponse features. The unannealed flm had compact nanog rained morphology with surface cracks, while AFM images showed spikes emanating from the substrate surface. The flms exhibited a tetragonal crystal structure with the high absorbance feature. The bandgap energy value of the SnO<sub>2</sub> films increased with annealing temperature. The highest photoluminescence intensity was recorded at 369 nm within the visible electromagnetic region. The flms also recorded good photoresponse to current. The synthesized  $SnO<sub>2</sub>$  films could be potentially applied in optical and solar cell devices.

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