

Enhanced room temperature ammonia gas sensing properties of Al‑doped ZnO nanostructured thin flms

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Abstract

In this present research work, we report the preparation of successfully synthesized ZnO films as pristine and doped with Al(Al-ZnO) on glass via facile, eco-friendly, and controllable SILAR method. A systematic evolution of structural, surface morphology, composition, photoluminescence and ammonia gas sensing behaviour of the system was investigated with a variation of Al dopant. XRD examination disclosed polycrystalline nature with the hexagonal system of all flms and crystallite size was noticed between 37 and 51 nm. EDX study approves the presence of Al doping in ZnO. Surface morphological tests through SEM presented the formation of nanoparticles and nanorods with a variation of Al content. The photoluminescence study revealed that due to Al doping the PL intensity was quenched which signifes the reduction of defects in the flms. It was shown that the estimated values of the energy gap are enlarged to 3.12 from 3.01 on rising the Al content till 3wt.% and fnally decreased for 5wt.% Al content. The gas sensing analysis showed that Al doping content was made to drastically increase the gas sensing response. Compared with other dopant levels, the 3wt.% Al-ZnO nanorods unveiled the uppermost retort when tested to 100 ppm ammonia $(NH₃)$ gas concentration at room temperature.

Keywords Controllable SILAR method · Al-ZnO thin flms · X-ray difraction · Optical properties · Gas sensor

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1 Introduction

Environmental pollution and various hazards gases released from various chemical factories initiate the scientists to advance chemical sensors able to precisely monitor and alert for industrial and technological related systems in real-time. These developed sensors also can be properly utilized to prevent fre or explosion. Currently, nanostructured metal oxides have drawn signifcant attention in the feld of nanotechnology for making active sensors to solve environmental issues. Semiconducting metal oxide nanostructures (SMONs) are currently focused extensively and plays a key role in enhanced gas sensing. These nanostructures are conventional for resistance type gas sensors (Li et al. [2019\)](#page-18-0). When exposed to hazardous gases Nanostructured materials exhibit higher sensitivity and rabid response under relatively low operating temperatures. This fast response might be owed to a large surface/size ratio of nanomaterials particularly in nanoscale, that enhance response kinetics of surface through growing the no. of adsorptions centers on the surface of the sensor coating (Dilova et al. [2020](#page-14-0)). Reducing the grain size to nanoscale may lead to induce new efects such as bandgap widening, room temperature (RT) photoluminescence, etc. due to the quantum confnement (Viter and Iatsunskyi [2019](#page-17-0)). It is worthy to note that compared to bulk counterparts these nanostructured materials have extraordinary unique and enhanced chemical, physical, and mechanical properties (Nunes et al. [2019](#page-16-0)). Diferent morphological structures and shapes also have attained signifcant interest in employing novel type gas sensor (Aslani [2011\)](#page-14-1). Various patterns like nanoparticles, nanoarrays, nanowires, nanosheets, nanorods, nanotubes, tower-like structures, nanobelts, nanosprings, nanocombs, nanorings and nanoflms, are the latest progress for developing or designing advanced chemical gas sensor (Zhang and Gao [2019](#page-18-1); Schmidt-Mende and MacManus-Driscoll [2007](#page-17-1)). However, for real usages, the key task is focused on discovering low cost, high performance, the sensitive and scalable fabrication process for the manufacture of metal oxide gas sensors.

ZnO is a talented, supreme, and trustworthy material for the above-mentioned purpose. In addition ZnO has prospective application in many felds such as piezoelectric devices (Qin et al. [2016](#page-16-1)), spintronics (Shaaban et al. [2019](#page-17-2)), acoustic waveguides (Fu et al. [2010](#page-14-2)), light emitting diode (Kwon et al. [2018\)](#page-15-0), Photodetector (Liu et al. [2009](#page-15-1)) photocatalyst (Binjing et al. [2019\)](#page-14-3), solar cells (Linhua et al. [2018\)](#page-15-2), nonlinear characteristics (Sofani et al. [2007;](#page-18-2) Abed et al. [2015](#page-14-4)), etc. Recently ZnO got immense consideration in the gas sensitivity feld because of its explicit chattels, like fnest conductivity/low toxicity/cost/chemthermosolidity (Dilova et al. [2020\)](#page-14-0). It also possesses higher mobility, lifetime, and intrinsic impurities. To enhance gas sensitivity V_0 , Zn_i , and O_i are possible defects for ZnO material (Schmidt-Mende and MacManus-Driscoll [2007](#page-17-1)). Nanostructures point of view also it can be easily grown with a variety of nanostructures. ZnO nanostructures have been extensively utilized for proving their excellency in detecting plenty of gases, like $H₂/CO/NO₂/$ $NH₃/H₂Stethanol/acetone etc.$ (Dilova et al. [2020\)](#page-14-0). The aforementioned gases ammonia is mostly considered as one of the highly toxic gas because of long duration $NH₃$ inhalation primes to demise (Liu et al. [2019\)](#page-15-3).

Further, based on ZnO-gas sensing devices possess some drawbacks as well like operation at high temperature/limited detection content/selectivity, etc. (Eisele et al. [2001](#page-14-5); Stegmeier et al. [2011](#page-17-3)). Hence, such disadvantages of such sensors limit their use in commercially. To overcome this issue various techniques like doping has been implemented to operate at room temperature in order to surge the efficacy/choosiness and sensitivity of ZnO based sensors. It is worth mentioning that thin flm gas sensors that are operated at room temperature (RT)

have many advantages like less power consumption, a long lifetime, and catching fre is not possible (Imad and Kadhim [2016\)](#page-15-4). Generally, group III dopants (transition metals) as Al, Ga, In, Mn, Co, Cr, Ni, Fe, and Cu have been suitably added to improve gas sensing properties. These dopants can boost the oxygen type for sensing and expand gas molecules adsorption (Sankar Ganesh et al. [2017e\)](#page-17-4). Recently, Al dopant is an important candidate, which plays a key role in making ZnO-based gas sensors. This is because Al dopant ofers improvement in the gas detection properties of ZnO by changing the energy-band structure, surface morphology, bandgap, transparency, resistivity or conductivity. This alternation is related to small ionic radii of Al^{3+} ion related to Zn^{2+} ion (Aydın and Yakuphanoglu [2019](#page-14-6)), hence it is called as fast difusion dopant.

In the past few eforts have been taken by numerous research workers to sense the leakage of various gases by appropriate doping in ZnO. Zhang et al. prepared Al-doped narcissus-like ZnO for NO₂ gas by hydrothermal method. The narcissus-like structure showed a fast response and recovery for the lower operating temperature of 240 $^{\circ}$ C (Zhang et al. [2019\)](#page-18-3). Ghosh et al. tuned the visible photoluminescence in Al-doped ZnO thin flm for detecting the glucose on the basis of quenching in fuorescence. Advanced sensitivity was attained by structure tuning [\(2018](#page-15-5)). Sheeba et al. [\(2017\)](#page-17-5) studied sensing of LPG by a clear Al:ZnO flms obtained by the thermo-evaporation process, an appreciable response time of 40 s was achieved for 3 wt.% Al-doped ZnO. Benelmadjat et al. ([2013](#page-14-7)) fabricated highly oriented Al and Al/Sb:ZnO films for H_2 gas on Si wafer by sol–gel method. Rabid responds to the low H_2 concentrations were noted. Gas detecting action of Al-ZnO film for H₂S recognition was systematically studied by Kolhe et al. [\(2018\)](#page-15-6) the Al dopant increased the oxygen vacancy related defects, which enhances the gas response at a minimum operating temperature of 200 °C. Madhumita Sinha et al. synthesized quick response/sensing at a low temperature of acetone/ethanol by Al:ZnO, they achieved acetone detection at RT and ethanol detection at a low temperature of 150 °C (Sinha et al. [2020](#page-15-7)). Aydınk et al. [\(2019\)](#page-14-6) studied the ammonia sensing through Al doping into ZnO for diferent ppm level in which 5 wt.% Al-doped flms operated at room temperature shows the fast response for 660 ppm. Ravichandran et al. [\(2019\)](#page-16-2) prepared W:ZnO flms for sensing of ammonia gas by spray technique, in which 1% tungsten shows a fast response towards 100 ppm ammonia level operated under room temperature.

Up to our knowledge, only a few scientifc reports are available for Al-ZnO flms which could be operated at RT to detect NH_3 gas. Moreover, fast response and recovery time is necessary to detect a low-level concentration of such harmful gas. So, to achieve these two objectives, we synthesized Al-ZnO films for the NH_3 sensor by the SILAR method. Compared with several other techniques SILAR technique is preferred most because this promising route has the following advantages such as simplicity, non-hazardous, inexpensive, less time consuming, does not require sophisticated instruments and high-quality substrates, easy route for the introduction of diferent dopants, suitable for large areas deposition and also to build functional structures (Deshmukh et al. [2018;](#page-14-8) Raut et al. [2017](#page-16-3); Omran et al. [2019\)](#page-14-9). Si, current work is focuses to fabricate Al-ZnO flms via SILAR process and investigate their characteristics towards ammonia gas sensing.

2 Investigational procedures

2.1 Preparation on Al‑ZnO thin flms

Here SILAR technique was adopted for the deposition of Al-ZnO thin flms on glass slides. Zinc sulfate $[ZnSO_4]$, NaOH, and Hexahydrate aluminium chloride (AlCl₃ 0.6H₂O) were collected from locally available commercial markets in India and they were used without further purifed. In order to remove the visible dirt, all the glass slides were initially cleaned with soapy water. Then they were successively washed with chromic acid, well rinsed with deionized water and then dried.

To deposit the flms the cleaned substrates were vertically dipped into the sodium zincate solution, which can be prepared by mixing 0.1 M ZnSO₄and 0.2 M NaOH. Before dipping the temperature of this solution was maintained at 90 °C. The pH of sodium zincate solution bath was maintained at 9. After immersing the substrate for 10 s in a sodium zincate bath, it was immersed in hot water (90 $^{\circ}$ C) for 10 s. This course was repetitive for 80 times to get pristine ZnO flms. Aluminium doping was done by adding hexahydrate aluminium chloride at diferent proportions of Al, such as 0, 1, 3, and 5wt.%. After the completion of the dipping cycles, the substrates drying was done in the air and then annealed at 400 °C for about 2 h.

The equations for the formation of ZnO thin flms are as follows:

$$
ZnSO4 + 4NaOH \rightarrow Na2ZnO2 + Na2SO4 + 2H2O
$$

$$
Na2ZnO2 + H2O \rightarrow ZnO + 2NaOH
$$

2.2 Characterization

The prepared high-quality ZnO thin flms were analyzed and recorded to recognize their phase, morphology, composition, and most importantly ammonia gas sensing behaviors affected by Al content. The structure information of undoped and Al-ZnO films was obtained by XRD (Bruker D8 Advance Eco XRD). A scanning electron microscope (EVO18 CARL ZEISS) was employed to inveterate the evolution of the morphology of samples. The room temperature Photoluminescence (PL) data were collected using a Varian Carry Eclipse PL Spectrometer. The optical properties were accessed through Ocean Optics HR 2000 optical fber-based spectrophotometer. Stylus profle meter (Mitutoyo SJ-301) was employed to determine the thickness of prepared films and are noted to $~> 920$, 980, 1020, and 880 nm for 0, 1, 3, and 5 wt.% of Al dopings, respectively.

3 Result and conversation

3.1 XRD studies

Figure [1](#page-4-0) depicted detailed XRD profiles of pristine and Al-ZnO films. From Fig. 1 we noted that the nature of all flms is polycrystalline and exhibited multiple peaks with hexagonal structure. Moreover, it is strongly oriented along the (002) direction.

Fig. 1 a X-ray difraction patterns of Al-ZnO thin flms with diferent Al doping concentrations. **b** Lattice paramerers of Al-ZnO thin flms with diferent Al doping concentrations

This strongest peak is usually observed in ZnO thin flms suggesting that free energy of the surface of (002) plane is the lowermost in ZnO flms (Fidal and Chandra [2018;](#page-14-10) Taabouche et al. [2016](#page-17-6)). The other difraction peaks correspond to (100), (101), (102), (112), (103) (112) and (201) orientations also appeared. All the difraction peaks of ZnO which matched with the JCPDS card no. 005–0664 (Linhua et al. [2014\)](#page-15-8). All appeared difraction peaks are sharp and well distinct which suggests the formation of high-quality crystalline phase of ZnO.No additional secondary phases related to Al or $A1₂O₃$ species were occurred not even for the highest proportion. Hence, we assumed that Al^{3+} substituted into the Zn^{2+} sites/incorporated interstitially in host ZnO lattice with unchanged hexagonal structure crystal structure (Humayan Kabir et al. [2018\)](#page-15-9). This may indicate that the precursors have been fully converted into ZnO phase (Sinha and Mahapatra [2020\)](#page-17-7). The intensity of (002) and (101) planes increases with increasing the Al concentration up to 3% doping concentration confrms the incorporation of Al atoms into ZnO lattice and its crystallite size was increased. Furthermore, for 5wt.% Al concentration the peak intensity is considerably decreased due to Al atoms placed in interstitial position or surface of the grains causes structural disorder in the crystal structure. A decrease in the crystallite size was an indication of deterioration of the crystallinity of the flms. A similar type of observation was reported by Kamal Rudra et al. for Mn doping of nanostructure ZnO (Kamal Rudra and Prajapati [2020\)](#page-16-4). Goktas et al. also reported the same type of observation for various transition metal dopants like Mn, Co, Fe into ZnO flm (Goktasa et al. [2016\)](#page-15-10).

The structural constraints like size of crystallites (D) , micro strain (ε) , number of dislocations (δ) and quantity of crystallites (N) were evaluated by (Deva Arun Kumar et al. [2018\)](#page-14-11)

$$
D = \frac{0.9\lambda}{\beta \cos \theta} \tag{1}
$$

$$
\varepsilon = \frac{\beta \cot \theta}{4} \tag{2}
$$

$$
\delta = \frac{1}{D^2} \tag{3}
$$

$$
N = \frac{t}{D^3} \tag{4}
$$

Here all codes are well-recognised.

It has been noticed that the crystallite size became bigger when the concentration of Al reached 3wt.% then leads to a decrease for 5wt.% Al level. The undoped ZnO possesses the smallest crystalline size. The FWHM of XRD peaks decreases frstly then rises on rising of Al impurity content, which supports the variation of crystallite size. The calculated crystallite size is comparable with previously reported values by Zaiour et al. [\(2019](#page-17-8)). The calculated microstrain and dislocation density values are low for 3wt.% Al content which indicates the formation of fewer defects. The N values initially declined and fnally enlarged with Al impurity. Table [1](#page-5-0) shows the calculated structural parameter values.

The cell volume '*V'* and lattice constants '*a'* and 'c' are evaluated for (002) peak by Vijayprasath et al. ([2014\)](#page-17-9).

$$
v = a^2 c(\sin 60^\circ) \tag{5}
$$

$$
\frac{1}{d^2} = \frac{4}{3} \left\{ \frac{h^2 + hk + k^2}{a^2} \right\} + \left\{ \frac{l^2}{c^2} \right\}
$$
(6)

here (hkl) and d have their usual meanings. The calculated lattice constants '*a'*, '*c'* and 'V' for Al- ZnO flms were depicted in Fig. [2](#page-6-0)b revealed that the cell volume has been decreased with Al doping and the highest value was noticed for undoped ZnO. It is noticed that a rise in Al content roots the decrement in both *a* and *c* parameters. This is due to the small ionic radius of Al^{3+} (0.68 Å) to the ionic radius of Zn^{2+} (0.74 Å). The change in the lattice parameter suggesting the formation of uniform strain (Hong and Choi [2018\)](#page-15-11). The estimated lattice parameter values comparable with earlier data reported by Osali et. al. (*a*=3.25 Å and *c*=5.207 Å) (Osali and Karami [2018\)](#page-16-5).

Table 1 Film thickness, structural parameters and bandgap of SILAR deposited Al-ZnO thin flms

Al con- centration $(wt,\%)$	Film thickness (nm)		Crystallite Dislocation size (nm) density $(\times 10^{15})$ $lines.m^{-2}$	Strain $(\times 10^{-3})$	Number of crystallites $(x10^{16}m^{-2})$	Bandgap (eV)
Ω	920	37	0.750	3.18	1.89	3.01
$\mathbf{1}$	980	41	0.583	2.80	1.38	3.04
3	1020	51	0.378	2.25	0.75	3.12
5	880	45	0.504	2.60	0.99	3.09

Fig. 2 SEM images of **a** pure, **b** 1, **c** 3 and **d** 5 wt.% Al-ZnO thin flms

3.2 Morphological studies

Figure [2](#page-6-0)a–d provided the surface morphological images of as-deposited Al-ZnO flms. Some notable changes between the samples were observed as the doping percentage of Al content increases. It can be observed that the surface of the un-doped flm has been covered with nano. From Fig. [2b](#page-6-0), it is observed that flm contains small size nanowire like morphology. However, with 3wt% Al doping, the flm morphology is completely nanowire shape [see Fig. [2](#page-6-0)c] is very clear, which may be due to higher concentration doping of Al. The 5% doped flm exhibits an entire wire-like nanostructure and its size is higher than previous samples. It is worthy to note the increment in the length of the nanowire has been observed. Moreover, a small diference was observed in all the upper parts of the fabricated flms. The highly packed particles are observed on the surface of pure ZnO flm along with few particles on top leads to agglomerate. These suggest that Al impurity is deposited preferably on the upper parts of ZnO which leads to the formation of a nano wire-like structure. Yuanyuan Lv et al. [\(2018](#page-17-10)) fabricated ZnO Nanowire by infuencing Al dopant into the host lattice of ZnO. Antaryami Mohanta et al. ([2019\)](#page-14-12) also prepared ZnO nanowires through Al addition. F:ZnO NWs achieved by plasma action was reported by Wang et al. [\(2018](#page-17-11)). Kim et al. [\(2019](#page-15-12)) formed Pd on ZnO NWs for improved sensitivity/selectivity towards H_2 gas. Room temperature gas nanosensors based Au-modifed ZnO nanowires synthesized by Lupan et al. [\(2019](#page-15-13)). Mohd Arif et al. ([2019\)](#page-16-6) applied Sol–gel spin coating to fabricate the pure and diferent concentrations of aluminum-doped ZnO nanostructure. Sankar Ganesh et al. $(2017e)$ $(2017e)$ tune the response of NH₃ sensing through Cu-in ZnO. Hence it is worth to our nano wire-like morphology results also resembles these results.

The elemental analysis of Al-doped thin flms was achieved by EDX is projected in Fig. [3](#page-7-0). The EDX spectrum obtained for the SILAR prepared ZnO thin flm deposited with

Fig. 3 EDX spectrum of Al-ZnO(3wt.%) thin flm

3wt.% Al which confrmed that Zn, O, and Al elements have emerged within the structure. The appearance of additional peaks may be owing to the glass substrate.

3.3 Photoluminescence studies

PL spectroscopy is an inevitable tool to analyze Photoluminescence behavior because it furnishes more information about sensing behaviors, point defects, and band edge emissions. Figure [4](#page-7-1) displays RT PL spectra of Al doped ZnO flms, which signify that luminescence character dependency on Al content. On increasing Al content emission intensity reduced to 3 wt.%, however, increased for 5 wt.%. Two intense PLE bands exist in all flms

from which one is for NBE occurs owing to recombining the excitons among CB and VB. Figure [4](#page-7-1) contains 4 sharp peaks at 407, 420, 486, and 517 nm and two small peaks at 444, and 459 nm total of six defects were presented. Generally, six kinds of intrinsic imperfections are likely names as V_0 , Zn_i , O_i , Zn_v , and Zn/O antisite (Nkosi et al. [2020](#page-16-7)). Moreover, it is worthy to note that the presence of defects and vacancies are benefcial for gas sens-ing applications (Patil et al. [2019\)](#page-16-8). The PLE band at \sim 406 nm is for NBE, the other PLE bands at 458 and 484 nm are ascribed to Zn_i and at 540 nm is for Vo. The low-intensity blue PLE at 445 nm is for Zn_i , an intense PLE band at 518 nm (green) is for V_o (Anita et al. [2016;](#page-14-13) Renitta and Vijayalakshmi [2017\)](#page-16-9). Normally higher the ZnO crystal quality possesses stronger UV emission (Ahmed et al. [2019](#page-14-14)). It can be known from XRD patterns that 3 wt. %Al doped ZnO shows the strongest diffraction peak; e.g. some ZnO with better crystallinity like NRs, NWs possess an exceptional UV emission act (Linhua et al. [2019](#page-15-14)).

3.4 Optical studies

A detailed study on optical transmittance can be used to analyze the efect (optical energy bandgap, optical transitions) of Aluminium dopant on the optical nature of ZnO flm. Figure [5](#page-9-0)a demonstrates the optical transparency for Al-ZnO flms prepared with various Al concentrations. All samples exhibit average transmittance values between 30 and 82%, depending on the dopant concentrations in the visible wavelength region. While increasing Al in ZnO the transparency is enhanced signifcantly in VIS-region. The 3 wt.% Al-ZnO flm exhibits the highest transmission, whereas the lowest one around 30% is shown by the undoped flm. This maximum optical transmittance spectrum specifed that the Al-ZnO flms can have good structural homogeneity (Hou and Jayatissa [2017\)](#page-15-15). Compared to 5% Al-ZnO flms, the flm doped with 3 wt.% shows slightly increasing transmittance values shows a maximum optical transmission which indicates the degradation in the crystal quality for higher doping. It is matched with XRD results. These transmittance fndings are perfectly matched with the workers Musaab Khudhr et al. (xxxx), Zhai et al. [\(2016](#page-17-12)), and Mkawi et al. (2015) (2015) . Further, a drop in T is observed in all the prepared films at around 400 nm. This drop may be due to the fundamental absorption by excitonic transitions.

The E_{σ} values were estimated via Tauc's rule (Firdous et al. [2013\)](#page-14-15)

$$
\alpha h v = B(hv - E_g)^n \tag{7}
$$

Here, α is the coefficient of absorption, E_{ϱ} is the direct energy gap, and B is a constant. The inset in Fig. [5](#page-9-0)b displayed the attained E_g of all Al-ZnO films. As can be seen from Fig. [5](#page-9-0)b the E_ρ value of undoped ZnO film was found to be equal to 3.01 eV. When the Al concentration is increased up to 3wt.% the E_g value is enlarged from 3.01 to 3.12 eV then fnally decreased for a higher doping level. This result is in accordance with the report of Rajendra Prasad et al. (Rajendra Prasad and Haris [2018](#page-16-11)).

3.5 Gas sensing studies

An indigenously designed set-up (Keithley) was employed to investigate the sensing act of undoped and Al-ZnO flms. A High-quality Ag paste was used to make contacts on the surface. The sensing element was kept in the sensor testing chamber. Baseline resistance was noted at RT for the Al-ZnO sensor by keeping an open chamber to allow atmosphere air. Then ammonia is injected by syringe to the tube which is connected by

Fig. 5 a Transmittance and **b** Bandgap spectra of Al-ZnO thin flms with diferent Al doping concentrations

a heater. Ammonia vapour is produced by applying the required heat to the tube that relates to flm and so lessening resistance. All sensor-related tests are done at RT by keeping the sensing area constant. For sensing of gas, the applied voltage was fxed at 5 V. Sensitivity is defined by Sivalingam et al. ([2012\)](#page-17-13): $S = R_a/R_g$, where R_a and R_g are strengths of air and gas (Table [2](#page-10-0)).

Figure [6](#page-11-0) reveals the response/recovery of Al–ZnO $(0, 1, 3,$ and 5wt. $\%)$ films for 100 ppm of NH_3 gas. The observed timings of responses are 52, 47, 34, and 39 $\,$ s, and recoveries are 11, 10, 6 and 9 s for Al–ZnO (0, 1, 3, and 5wt. %) samples, respectively. The observed response time initially increased on rising Al doping content in ZnO and maximum response time was observed for 3 wt% Al-doped flm due to high crystalline and nanowire structure. The sensitivity of the samples is 547, 6470, 10,700, and 12,300 for Al $(0, 1, 3, \text{ and 5wt. } \%)$ doped ZnO film. The excellent sensitivity observed for Al 3wt.% film rather than other doped flms due to its high crystalline quality and nanowire structure is suitable for difusion and interaction of more ammonia vapours. Figure [7](#page-11-1) elaborates sensing features of 3wt.% Al-ZnO film for diverse contents of $NH₃$ at RT operating condition. Sensitivity increases as increasing ammonia vapour ppm from 100 to 200. It is clear from the observations that the amount of vapour interact on the surface of Al-ZnO flms is a

123456789

substrate with exposure and

ent ppm

Fig. 6 Response time (t_{res}) and recovery time (t_{rec}) of Al-ZnO thin films deposited on glass substrate with exposure and removal of NH_3 gas of 100 ppm concentration condition

vital constraint governing the response. It is also confrmed from the fgure that Al-ZnO display an enhancement in the sensitivity and high selectivity for ammonia.

The observed value of response and recovery time of present work is compared with other works. Sankar Ganesh et al. $(2017d)$ synthesized enriched of NH₃ gas sensor as Ag/ZnO by hydrothermal method. They showed that the addition of Ag exhibited quick response/recovery timings of 13/20 s for operating temperature 150 °C. On the other hand,

Fairose et al. [\(2019](#page-14-16)) lessened the response/recovery timing to 8/37 s through a nanostructured ZnO sensor by the SILAR method but the operating temperature of the sample was 200 °C which is higher than our work. Ravichandran et al. [\(2019](#page-16-2)) studied room temperature ammonia vapour sensing ability for tungsten doped ZnO. The loaded tungsten showed a response and recovery time of 293 s and 13 s for the 100 ppm level. This achievement was due to a greater number of V_0 and Z_i these are the key reasons for the enrichment of sensing capability. But Kulandaisamy et al. ([2016\)](#page-15-17) reduced the response and recovery time up to 34 s and 28 s by introducing Mg into ZnO host lattice. This room-temperature sensing nature might be because of lesser grain size. Ganesh Kumar Mani et al. ([2014\)](#page-14-17) attained much better performance than those already discussed so far with a quicker response/ recovery of 30/14 s for room temperature ammonia sensor for 50 ppm. Smaller grain size refected faster response/recovery time. Thus, compared to the above-mentioned reports the Al-ZnO flm is showing a best response/recovery time of 34/9 s for 100 ppm. This fast response is might be because of the generation of nanowire like morphology formed by Al content. Mariappan et al. ([2014\)](#page-16-16) reported Na-ZnO NRs films by NSP and their HN_3 sensing activity. The uniformly distributed nanorods show the sensing response of 1.7 for 500 ppm operated at 400 $^{\circ}$ C. Sankar Ganesh et al. ([2017a\)](#page-17-17) increased the sensitivity of 15.1 for the operating temperature of 400 $^{\circ}$ C when exposed to 100 ppm. It was due to enhanced catalytic activity. On the other hand Pt nanoparticles (NPs) are deposited on a sputtered aluminum-doped zinc oxide exhibited a sensing response of 2183 once exposed to 1000 ppm but the working temperature was 300 $^{\circ}$ C by Chen et al. [\(2018](#page-14-18)). Mani et al. (2015) (2015) presents a very selective co-doped ZnO RT NH₃ sensor. Its sensing response was 225. Sankar Ganesh et al. [\(2017b](#page-17-18)) investigated the infuence of Al doping on RT ammonia sensing by ZnO nanorods. It displayed a fast sensing response of 350 owed to the existence of frequent V_0 , which indorses oxygen adsorption. In current, we accomplished a high response of 12,300 towards RT $NH₃$ sensing due to the nanostructure of nanowire-like structure. Furthermore, we have listed some work that was reported and published already.

3.5.1 Sensing mechanism

Figure [8](#page-13-0) presents the mechanism for NH_3 sensing towards pure and 3wt. % Al-ZnO films. The sensing of $NH₃$ behavior is primarily built on chemical reactions amongst test gas and targeted specimen. This process might be discussed in 2 main steps. Initially, when the undoped ZnO thin flm was bare to ambient air, oxygen molecules were captured by electrons apprehending from CB of ZnO. As a result, the width of the potential barrier changed which leads to an increase of resistance. It is expressed through Eq. ([8\)](#page-12-0), once plummeting gas like NH_3 is exposed to ZnO films surface, NH_3 vapours interact its surface adsorbed oxygen, and electrons gets freed to surface, thus reduces the width of the potential barrier. Consequently, the resistance decreases and gas sensing response increases. This express able by Eq. [\(9\)](#page-12-1) (Rana and Rajasree [2016;](#page-16-17) Kanaparthi and Singh [2020](#page-15-18)).

$$
O_{2(\text{atmosphere})} + e_{(ZnO\text{ surface})}^- \to O_2^-(ZnO\text{ surface})
$$
 (8)

$$
4NH_3 + 3O_{2(ad)}^- \to 6H_2O + 6e^-
$$
 (9)

Similarly, 3wt.%Al-doped ZnO flm contains more nanowires causes, more oxygen species can interact than undoped ZnO flm thus the depletion layer width is reduced.

Fig. 8 NH_3 sensing mechanism towards **a** ZnO and **b** Al-ZnO (3wt.%) thin film

Hence, owing to its thin depletion layer, additional electrons are injected when $NH₃$ reacts with oxygen species.

4 Conclusions

In this present work, we report the sensor-based Al-doped ZnO flms using the SILAR route. Aluminium ions were successfully incorporated into the ZnO structure which was confrmed from the XRD and further confrmation was carried out by EDX. The morphological analysis approves the development of nanowire like structure formed by dopant impurity which is favorable for sensing. The evaluated Eg varied from 3.01 to 3.12 eV on rising the Al doping content. RT ammonia sensing analysis showed an excellent response towards ammonia. In addition, the flm doped with 3wt.% displays improvement in fast response and recovery characteristics. The outcomes signify that the grown Al-doped ZnO flms are of great importance for gas sensing.

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