

# Efficient SnO<sub>2</sub>/CuO/porous silicon nanocomposites structure **for NH3 gas sensing by incorporating CuO nanoparticles**

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### **Abstract**

In this paper, a porous silicon (Psi) surface modifcation process, including the incorporating of pure and CuO mixed SnO<sub>2</sub> within a Psi layer, was investigated. A Psi layer with a pore-like structure prepared via laser-assisted etching process with a laser illumination intensity of (40) mW/cm<sup>2</sup> was employed as a substrate for materialization  $SnO_2/CuO$ Psi nanocomposites for gas sensing application. Pure  $SnO<sub>2</sub>$  and  $SnO<sub>2</sub>:CuO$  nanoparticles were synthesized via a chemical spray pyrolysis process on the Psi surface at 350 °C. The materialization of  $SnO<sub>2</sub>:CuO/Psi$  nanocomposites were recognized through Field-Emission Scanning Electron Microscopy, X-ray Difraction, Atomic Force Microscopy, and Energydispersive X-ray Analysis. The results displayed that the morphological properties of SnO<sub>2</sub>:CuO nanoparticles were definitely controlled by varying the CuO contents. Different shapes (nearly cubic shape and spherical shape) with various sizes of  $SnO<sub>2</sub>$  and  $SnO<sub>2</sub>$ :CuO within the Psi surface were achieved by increasing the CuO content. SnO<sub>2</sub>:CuO/Psi nanocomposites, plans, grain size, average surface roughness, homogeneity and specifc surface area of SnO<sub>2</sub>:CuO nanoparticles were considerably influenced by CuO content. The lower grain size and higher average surface roughness and specifc surface area were attained when the percentage of CuO was about  $30\%$  within the structure of  $70\%$  SnO<sub>2</sub>:30% CuO, and the performance of the  $NH<sub>3</sub>$  gas sensor revealed a short response time with a high sensitivity of (88%).

**Keywords** Nanocomposites  $\cdot$  SnO<sub>2</sub>:CuO  $\cdot$  Spray pyrolysis  $\cdot$  Porous silicon  $\cdot$  NH<sub>3</sub>

# **1 Introduction**

Applications of Psi layer as a basic substrate for gas sensors process are of abundant signifcance as a global issue. The Psi surface modifcation process with a metal oxide semiconductor (MOS) ensures the extremely concerned attention owing to their stability and economic cost (An and Yang [2012;](#page-11-0) Kemmler et al. [2013;](#page-11-1) Huang and Wan [2009\)](#page-11-2). Among the whole types of MOS, pure  $SnO<sub>2</sub>$  as a wide band gap of (3.5 eV) is broadly used in the

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optoelectronics and solar cell applications (Dattoli et al. [2007](#page-11-3); Chappel et al. [2002](#page-11-4)). On the other hand, the pure  $SnO<sub>2</sub>$  sensing element cannot provide the necessity for extreme detection for poisonous gases. So, a lot of alterations have been tried to enhance the gas sensing properties of  $SnO<sub>2</sub>$ , like element incorporating (Zhang and Yang [2012\)](#page-12-0), and combined with further metal oxide (Katoch et al.  $2015$ ). The more efficient process to enhance the physical and chemical properties and hence the gas sensing process is incorporating an alternative material to create a composite material (Paek et al. [2009](#page-11-6); Huang et al. [2011;](#page-11-7) Kim et al.  $2010$ ). Several types of MOS, like  $In_2O_3$ , WO<sub>3</sub>, ZnO, and CuO are incorporated to boost the gas sensing property. A unique favorable approach to improve the gas detection process is by growing the surface porosity while preserving the suitable sensing material at the surface, like using Psi substrates (Moshnikov et al. [2012\)](#page-11-9). Many applications have been done on n-type PSi (Alwan and Dheyab [2017,](#page-11-10) Alwan et al. [2017,](#page-11-11) [2018;](#page-11-12) Ismail et al. [2016\)](#page-11-13). One of the most important advantages of Psi layer is the simplistic preparation, inexpensive, low-temperature operation. The excellent specifc surface area with well-regulated pore sizes makes the penetration of metal oxides into the silicon pores probable and also makes the Psi with metal oxide nanoparticles as a supreme convenient material for the preparation of MOS/Psi nanocomposites structure for the chemical gas sensors (Moshnikov et al. [2012;](#page-11-9) Bolotov et al. [2012](#page-11-14); Naderi et al. [2012;](#page-11-15) Salonen et al. [2015\)](#page-11-16). In the present work, a Psi surface modification process, including the incorporating of pure and  $CuO$  mixed  $SnO<sub>2</sub>$  nanoparticles within the Psi layer was investigated. An efficient  $SnO<sub>2</sub>:CuO/Psi$  nanocomposites structure with excellent morphological properties is the main aim of this study, and hence to controlling and enhancing the gas sensing properties by improving the exposure area of nanocomposites/porous silicon structures through employing a specifc form of SnO2:CuO nanoparticles at specifc positions within the porous layer.

### **2 Experimental work**

### **2.1 Porous silicon formation**

A Psi layer was prepared using mirror-like (100)-oriented n-type Si substrate with a resistivity of 10  $\Omega$  cm. Prior to the etching process; the SiO<sub>2</sub> was eliminated from the Si by rinsing the substrates in 10% HF aqueous solution for 5 min. The Psi layer was created via a laser-assisted etching process in an etching combination solution of 1:1 of 40% HF and 99.999% ethanol. The Si substrate was etched at a constant current density of around (18 mA/cm<sup>2</sup>) for 6 min with a laser wavelength of 650 nm and an intensity of 40 mW/  $\text{cm}^2$  cm. Next to the etching process, the Psi substrate was washed with deionized pure H2O and permitted to dry in the ambient atmosphere for a few minutes, Fig. [1](#page-2-0) illustrates the experimental setup of the laser-assisted etching process.

### **2.2 Preparation of SnO<sub>2</sub>:CuO/Psi nanocomposites structure**

After the etching process, the Psi layer substrate is ready to deposit on by a spray pyrolysis process. As shown in Fig. [2](#page-2-1), the experimental setup of the chemical spray pyrolysis process includes a solution that is sprayed on top of a Psi layer held at  $350^{\circ}$ C, where the solution reacts to synthesize the preferred nanoparticle layer. The preliminary solution was achieved by aqueous solutions of 0.1 M  $SnCl<sub>4</sub>5H<sub>2</sub>O$  from (BDH, ENGLAND) and 0.1 M CuCl<sub>2</sub> from (BDH, ENGLAND) and used as a doping agent to form multi mixing concentrations



<span id="page-2-0"></span>**Fig. 1** Laser assisted etching setup



<span id="page-2-1"></span>**Fig. 2** Schematic diagram of spray pyrolysis experimental setup

 $(90\%SnO<sub>2</sub>:10\%CuO, 70\%SnO<sub>2</sub>:30\%CuO, 50\%SnO<sub>2</sub>:50\%CuO).$  These compounds were melted in distilled deionized water, to form the required spray solution of 40 ml for every deposition route. The optimum conditions of the spraying procedure were found by using a glass atomizer, which has an output nozzle around 0.8 mm, for a spray time of 6 s, and the spray interval is about 60 s with a constant carrier nitrogen gas at a pressure of 3 bars, the distance between the nozzle and the Psi layer was about 15 cm with a fow rate of about 2.3 ml/min.

### **2.3 Characterization**

The crystal structures of  $SnO<sub>2</sub>$  and  $SnO<sub>2</sub>$ :CuO films were characterized by XRD diffract meter (Model-XRD- 6000 Shimadzu) using CuK $\alpha$  ( $\lambda$  = 1.54056 Å) radiation. Morphological properties of Psi surface before and after the modifcation process with pure and CuO mixed  $SnO<sub>2</sub>$  were investigated via using (FE-SEM) along with (EDXS) spectroscopy (INSPECT-550) and (CSP model AA3000 AFM supplied by Angstrom Company). The  $NH<sub>3</sub>$  sensing performances were made in a stationary gas performances characterization system comprising of a glass test cavity, a regular heating plate, and a programmable digital readout system. Aluminum ohmic contacts of  $1 \text{ mm} \times 0.5 \text{ mm}$  were deposited by using thermal evaporation technique with a grid mask having a thickness of (300 nm) on the top of the SnO<sub>2</sub>/CuO film surface. Sensing performances of Al/Psi/Al, Al/SnO<sub>2</sub>/Psi/Al, and Al/ SnO<sub>2</sub>:CuO/Psi/Al device sensors have been successfully fabricated. The prepared sensors were investigated essentially in a vacuum case and with  $NH<sub>3</sub>$  gas of different concentrations 50 ppm, 100 ppm, and 150 ppm at the room temperature. The  $NH<sub>3</sub>$  was presented into the cavity straight to acquire the wanted concentration, and a programmable readout (UNI-UT70D) was employed for nonstop monitoring the current variation of the sensors by changing the voltage. The specimen interlude was set to 1 s, and the test temperature was the room temperature.

## **3 Results and discussion**

#### **3.1 X‑ray difraction patterns**

The XRD patterns of the Psi layer,  $SnO<sub>2</sub>$ , and  $SnO<sub>2</sub>$ :CuO/Psi nanocomposites are elucidated in Fig. [3.](#page-4-0) Broader and defnite XRD peaks were detected at 2 theta: 26.6, 33.9, 38.08 and 51.8 corresponding to (110), (101), (200) and (211) planes, respectively. This pattern is in a good agreement with the polycrystalline rutile structure of  $SnO<sub>2</sub>$  layer. Also, it is important to mention that the specifc two peaks at around 2 theta: 33 and 38.9 are observed in the XRD pattern of  $SnO<sub>2</sub>:CuO$  nanocomposite corresponding to (110) and (002) planes of CuO (Hwang et al. [2009\)](#page-11-17), respectively. XRD patterns of  $SnO<sub>2</sub>$  and CuO displayed a good agreement with the difraction data from JCPDS 41-1445 and 05-0661, correspondingly. From this pattern, it is clear that as the CuO content increases in  $SnO<sub>2</sub>:CuO$  nanocomposite, the crystal size decreases, therefore, incorporating the CuO nanoparticles within the  $SnO<sub>2</sub>$  layer will improve the additional specific surface area over the Psi layer.

The crystallite size (D) was calculated using the Scherrer formula (Klug and Alexander [1974\)](#page-11-18),

$$
D = \frac{0.94\lambda}{L(2\theta)\cos\theta} \tag{1}
$$



<span id="page-4-0"></span>**Fig. 3** XRD patterns of: (a) Bare Psi, (b)  $SnO<sub>2</sub>$ , (c)  $90\%SnO<sub>2</sub>:10\%CuO$ , (d)  $70\%SnO<sub>2</sub>:30\%CuO$  and (e) 50%SnO<sub>2</sub>:50%CuO

where  $\lambda$  is the wavelength of the X-ray of Cu K $\alpha$  radiation, and L is the full width at the half maximum of diffraction peaks. The values of crystal size were in the range from  $(25.1)$ to (13.3) nm for the unincorporated and the CuO incorporated  $SnO<sub>2</sub>$ , correspondingly. Table [1](#page-4-1) lists the crystal size and the specific surface area as a function of CuO content within SnO<sub>2</sub>:CuO nanocomposite.

The specific surface area  $(S.S.A)$  of  $SnO<sub>2</sub>:CuO$  nanocomposite is given by Hashim et al. ([2018](#page-11-19)):

$$
S.S.A = \frac{6000}{D \times \rho d} \tag{2}
$$

where  $\rho$  is the density of SnO<sub>2</sub>:CuO nanocomposite and it is calculated based on the density of  $SnO<sub>2</sub>$  and CuO by taking into account the percentage of elements in the EDS analysis (Hashim et al. [2018](#page-11-19)).

$$
\rho_{\rm d} = \frac{(a \times \rho)/(b \times \rho)}{a + b} \tag{3}
$$

<span id="page-4-1"></span>**Table 1** Diffraction peaks, planes, crystallite size and the specific surface area within SnO<sub>2</sub>:CuO nanocomposite

Sample	Main plane	$2\theta$	Crystal size (nm)	Specific surface area $(m^2/m)$
Pure $SnO2$	110	26.66	25.1	34
90% SnO <sub>2</sub> :10% CuO	110	26.73	18.3	47
70% SnO <sub>2</sub> :30% CuO	110	26.7	13.5	63
50% SnO <sub>2</sub> :50% CuO	110	26.69	15.4	56
90% SnO <sub>2</sub> :10% CuO	110 (CuO)	33	17	55
70% SnO <sub>2</sub> :30% CuO	110 (CuO)	33.05	13.3	71
50% SnO <sub>2</sub> :50% CuO	110 (CuO)	33.02	14	67

where  $\rho$  is the density of the SnO<sub>2</sub> and CuO NPs,  $\rho_d$  is the density of SnO<sub>2</sub>:CuO nanocomposite in g/cm<sup>3</sup>,  $a = wt\%$  SnO<sub>2</sub> and  $b = wt\%$  CuO in the EDS analysis. The increase of S.S.A leads to enhance the interaction desorption and adsorption rates between the  $SnO<sub>2</sub>-CuO$ layer and the gas molecules.

#### **3.2 Morphological features of SnO2:CuO/Psi Nanocomposites structures**

The morphological nature of  $SnO<sub>2</sub>$  and  $SnO<sub>2</sub>:CuO$  was investigated by analyzing the FESEM image, as shown in Fig. [4](#page-6-0). The surface morphology of the Psi in Fig. [4a](#page-6-0) reveals that the structure is a pore-look like, rectangular and asymmetrical in shape due to the pores overlapping among the individual pores, and the pores are arbitrarily dispersed on the surface. This overlapping occurs due to the excessive etching process at a high laser illumination intensity (Alwan and Abdulrazaq [2008](#page-11-20)). The sizes of the pores are in the range of (0.8–2.7) µm, and the depth of the porous layer and the cross-sectional morphology of the deposited layer are evinced in the inset of each image. For pure  $SnO<sub>2</sub>/Psi$  nanocom-posites structure in Fig. [4b](#page-6-0), the shapes of the pure  $SnO<sub>2</sub>$  nanoparticles deposited over the porous structure are nearly cubic with a size range of 80–95 nm. For the surface morphology of SnO2:CuO/Psi nanocomposites at diferent CuO contents, Figs. [4](#page-6-0)c–e, it is clear that the SnO<sub>2</sub>:CuO nanoparticles have a nearly spherical shape. Most of the nanoparticles are agglomerated at the silicon walls, and some of the nano sized CuO particles entered inside the pores of silicon, this is in a good agreement with (Ismail et al. [2016\)](#page-11-13). With the increasing of CuO concentration from 10 to 50%, the resulting nanoparticles size decreases from 90 to 30 nm with improving the surface homogeneity, this refects the important features for controlling the  $SnO<sub>2</sub>:CuO$  nanoparticles and hence develops the extra specific surface over the native Psi layer. The nature of the sensing layer  $(SnO<sub>2</sub>–CuO/PSi$  nanocomposites) is regulated by controlling the CuO content in the layer. This process is refect a signifcant scientifc touch for improving the sensing parameters.

The chemical structure of the  $SnO<sub>2</sub>:CuO$  Nanocomposite was inspected by the EDS spectrum, as exposed in Fig. [5a](#page-7-0)–d. The elements Cu, Sn, O and Si are obvious, certainly confirming the existence of  $SnO<sub>2</sub>:CuO$  nanocomposite. Also, from these figures, it can be noted that the peak of the specifc elements was increased with increasing their contents in the synthesized  $SnO<sub>2</sub>:CuO$  nanocomposite. The results of the element mapping for Sn, Cu and O are depicted in Fig. [5e](#page-7-0)–h. The spatial distributions of Sn, Cu, Si, and O clearly distinguish the CuO nanoparticles from  $SnO<sub>2</sub>$  ones.

#### **3.3 AFM analysis**

The topographical properties of Psi,  $SnO<sub>2</sub>$ , and  $SnO<sub>2</sub>$ : CuO nanocomposite are manifested in 2-D and 3-D AFM images in Fig. [6.](#page-8-0) The images clarify that the surface of all structures has a high degree of homogeneity without any cracks. The measured average surface roughness and the root mean square (RMS) obtained were increased with the increasing of CuO content, as shown in Table [2.](#page-9-0) In general, from this table, the roughness was increased with increasing CuO concentration, and a higher value of about (42.3 nm) was obtained at 30% CuO concentration. The increasing of the surface roughness will lead to enlarge the adsorption and the desorption rates between the  $NH<sub>3</sub>$  molecules and the active sensing elements. The granularity size spreading of the prepared particle sizes, as displayed in Fig. [6](#page-8-0), specifes that the CuO content was defnitely controlled by varying the surface properties of the deposited nanoparticles.



<span id="page-6-0"></span>**Fig.** 4 FE-SEM images of: **a** PSi, **b** Pure SnO<sub>2</sub>, **c**  $90\%$ SnO<sub>2</sub>:10%CuO, **d**  $70\%$ SnO<sub>2</sub>:30%CuO and **e** 50%SnO<sub>2</sub>:50%CuO Nanocomposite. The inset images are cross section images



<span id="page-7-0"></span>**Fig. 5** EDS spectra: **a** Pure SnO<sub>2</sub>, **b**  $90\%$ SnO<sub>2</sub>:10%CuO, **c**  $70\%$ SnO<sub>2</sub>:30%CuO, **d**  $50\%$ SnO<sub>2</sub>:50%CuO and **e-h** Elemental mapping images of 70%SnO<sub>2</sub>:30%CuO

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#### **3.4 Gas sensing characteristics**

Al/PSi, Al/SnO<sub>2</sub>, and Al/SnO<sub>2</sub>:CuO sensors have been successfully fabricated, and the resistance as a function of time of  $NH<sub>3</sub>$  gas with different concentrations has been illustrated in Fig. [7.](#page-9-1) All sensors have been tested three times to confrm their gas performance. The Psi sensor showed a response and recovery times of values 15 s and 17 s, respectively. When the  $SnO<sub>2</sub>$  nanostructure was sprayed on the Psi, the response time enhanced by the value of 11 s, but with a long recovery time of 40 s. CuO nanoparticles have developed the sensor when  $10\%$  of them were loaded to  $90\%$  SnO<sub>2</sub> with a response and recovery times of 12 s and 14 s, respectively. The excellent sensor was obtained when 30% of CuO nanoparticles was incorporated into  $70\%$  SnO<sub>2</sub>, this sensor elucidated a response and recovery times of 7 s and 20 s, respectively. The development in this sensor could be attributed to the small grain size and high specific surface area. The sensor  $50\%$  SnO<sub>2</sub>:50% CuO evidenced a long response and recovery times. The decreasing of the response time (fast

<span id="page-9-1"></span>

<span id="page-10-0"></span>

sensing) of PSi with incorporating of  $SnO<sub>2</sub>-CuO$  nanocomposites is due to the increasing of the adsorption rate between the sensing elements and the  $NH<sub>3</sub>$  molecules. Also the increasing of the recovery time is related to the desorption rate between the gas molecules and the active elements.

The sensitivity as a function of  $NH<sub>3</sub>$  gas concentrations is displayed in Fig. [8](#page-10-0). Psi sensor showed the lowest sensitivity. This sensitivity improved when the  $SnO<sub>2</sub>$  nanostructure was deposited on the Psi layer with the value of (61%). As the 10% of CuO nanoparticles was modified to  $90\%$  SnO<sub>2</sub>, the sensitivity increased to  $(68\%)$ . The best value of sensitivity was observed when 30% of CuO was added to  $70\%$  SnO<sub>2</sub> and it recorded (88%), this value is better than that reported in Yan et al. [\(2019](#page-12-1)), Ganesh et al. [\(2018](#page-11-21)) and Jahromi and Behzad ([2018\)](#page-11-22). The sensitivity was calculated by Alwan et al. [\(2019](#page-11-23)):

$$
Sensitivity = \frac{R_g}{R_a} \tag{4}
$$

where,  $R<sub>g</sub>$  is the resistance measured in the presence of the target gas, and  $R<sub>a</sub>$  is the resistance measured in the presence of air.

# **4 Conclusions**

In this work, efficient MOS/Psi nanocomposites structure was successfully synthesized by incorporating pure SnO<sub>2</sub> and CuO mixed with SnO<sub>2</sub> nanoparticles into a Psi layer. Controlling the CuO content changes efficiently the morphology of the Psi surface. A high degree of homogeneity with a high average surface roughness, high root means square (RMS) and highly extra specifc surface over the native Psi layer were achieved by 30% CuO content layer. The gas sensor properties evidenced a short response and recovery times with a very high sensitivity. Further works in the application of these new MOS/Psi nanocomposites will expose a unique prospect in the feld of applied Nanosciences.

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