



# PH<sub>3</sub> gas adsorption on S and Mo vacancy MoS<sub>2</sub> monolayer: a first principle study

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**Abstract** The sensing nature and change in the electron transport behavior of S vacancy armchair MoS<sub>2</sub> (AmS-MoS<sub>2</sub>), Mo vacancy armchair MoS<sub>2</sub> monolayer (AmMo-MoS<sub>2</sub>), S vacancy zigzag MoS<sub>2</sub> (ZigS-MoS<sub>2</sub>), and zigzag Mo vacancy MoS<sub>2</sub> (ZigMo-MoS<sub>2</sub>) monolayer before and after PH<sub>3</sub> adsorption were theoretically investigated using Density Functional Theory (DFT) in combination with Non-Equilibrium Greens Function (NEGF) based on first principle calculations. To study the feasibility of armchair and zigzag MoS<sub>2</sub> device as PH<sub>3</sub> gas sensor, we conducted an analysis of the changes in the geometrical structures, density of states (DOS), transmission spectrum, and I–V characteristic. Our results predicted that PH<sub>3</sub> adsorption on all four devices is through van der Waals interactions. Among the four devices, AmMo-MoS<sub>2</sub> shows enhanced adsorption behavior with the adsorption energy – 1.8048 eV and charge transfer of – 0.2120e. The I–V characteristic of AmMo-MoS<sub>2</sub> shows a significant change in the conductivity compared with the other devices. Thus, our work concluded that AmMo-MoS<sub>2</sub> is considered to be a better device for PH<sub>3</sub> adsorption compared with the other devices.

**Keywords** Density Functional Theory (DFT) · Non-Equilibrium Greens Function (NEGF) · Phosphine (PH<sub>3</sub>) · Van der Waals interactions · Electron transport study · Molybdenum disulfide (MoS<sub>2</sub>) · Adsorption · Sensor

## Introduction

Phosphine (PH<sub>3</sub>) is a highly poisonous gas that is commonly used in semiconductor industries and for fumigating grains. PH<sub>3</sub> gas molecules are exhausted during the production of acetylene and flame-retardant industries. The emission of PH<sub>3</sub> gas is very harmful to human beings, which causes cancer, headache, vomiting, fatigue, and even cause damage to the heart [1–4]. Therefore, rapid and precise sensing and monitoring of PH<sub>3</sub> gas molecules play an essential role in prevention.

Two-dimensional materials (2D) have a tremendous attractive interest due to their unique and extraordinary mechanical, physical, and chemical properties [5]. It has been considered as a suitable material for various potential applications. It is also considered as flexible material for next-generation optoelectronic devices, electronic devices, and gas sensors [6–8]. For the past few years, the researchers found that graphene has earned much more attention for its unique properties, and it has been used in many applications. However, the absence of bandgap in graphene has limited their progress [6, 9]. Transition

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metal dichalcogenides (TMDs) are thin semiconductors of type  $MX_2$  [10], where  $M$  is the transition metal element from groups IV, V, or VI and  $X$  represents the chalcogen elements like S, Se, and Te [11]. 2D materials have some attractive and interesting features, including high carrier mobility, high surface to volume ratio, high chemical stability, high thermal stability, low electronic temperature noise and fast response time [12, 13], and low-cost effect [14]. Materials with these characteristics are considered to be ideal for sensing applications [11, 15]. Among TMDs,  $MoS_2$  (molybdenum disulfide) is considered one of the most suitable materials for electronic device and sensor applications due to its excellent electrical and mechanical properties [14] and tunable bandgap when compared with graphene [12, 16]. The crystal structure of  $MoS_2$  consists of a weakly coupled S-Mo-S sandwich layer. The structure of  $MoS_2$  monolayer can be fabricated by the micromechanical cleavage or exfoliation method [9, 17, 18].

Shokri and Salami analyzed the sensing capabilities of  $MoS_2$  monolayer transducer with CO,  $CO_2$  and NO gas molecules and concluded that the NO gas molecule shows more changes in the electronic properties and charge transfer while compared with CO,  $CO_2$  gas molecules [19]. Jasmine et al. theoretically analyzed the sensing behavior of  $Cl_2$ ,  $PH_3$ ,  $AsH_3$ ,  $BBr_3$ , and  $SF_4$  gas molecules on Mo/S vacancy  $MoS_2$  monolayer and concluded that the  $PH_3$  gas molecule shows more adsorption towards S/Mo vacancy  $MoS_2$  monolayer [10]. Ren et al. theoretically investigated the adsorption behavior of  $CH_3$  gas molecule on S and Mo vacancy  $MoS_2$  monolayer, and they suggested that the different vacancies have a different effect on adsorption behavior [20]. Feng et al. concluded that the material's conductivity is increased due to the vacancy creation [21]. Zhao et al. theoretically investigated the adsorption of various gas molecules, including CO,  $CO_2$ ,  $NH_3$ , NO,  $NO_2$ ,  $CH_4$ ,  $H_2O$ ,  $N_2$ ,  $O_2$ , and  $SO_2$  on  $MoS_2$  monolayer using DFT. The results indicate that NO and  $NO_2$  show better adsorption than other gas molecules [22]. Wei et al. theoretically investigated the sensing behavior of Ni-doped  $MoS_2$  monolayer towards  $SO_2$ ,  $H_2S$ , and  $SF_6$  gas molecules. The results indicate that  $H_2S$  and  $SO_2$  tend to adsorb on the surface of Ni- $MoS_2$  monolayer by chemisorption, and the adsorption energy of the  $H_2S$  and  $SO_2$  are  $-1.319$  eV and  $-1.382$  eV, respectively [23]. Kumar et al. experimentally analyzed the recent

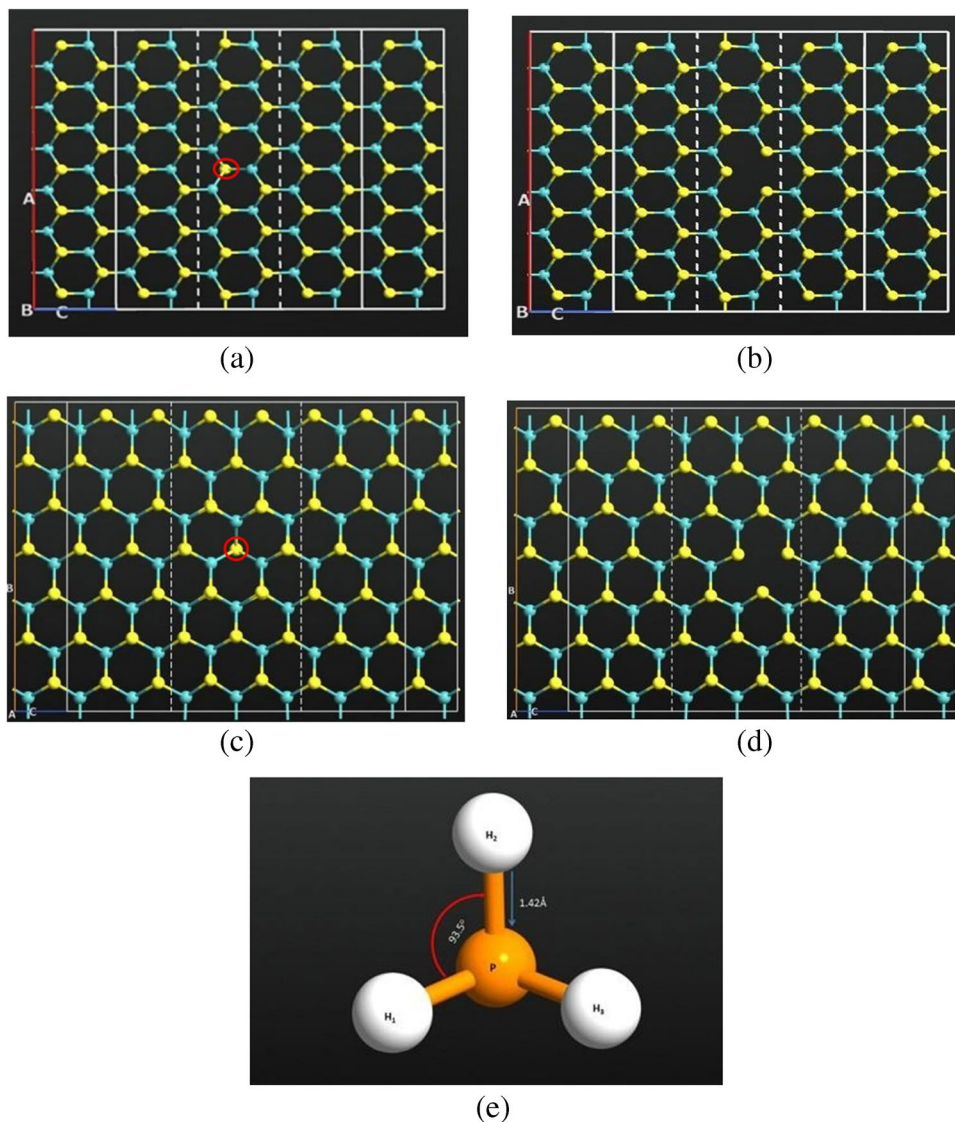
progress and remarkable development in gas sensing field by using the 2D  $MoS_2$ . They have developed various fabrication techniques for synthesizing a wide range of different nanostructures and morphologies of the  $MoS_2$  on rigid as well as flexible substrates. They concluded that all the exciting gas sensing results of the 2D  $MoS_2$  could be the best candidate for developing a high-performance room temperature gas sensor [24]. Chacko et al. analyzed the experimental study of Ni and Pd functionalized  $MoS_2$  devices towards  $H_2S$  and NO. The  $MoS_2$ -based sensors showed excellent sensing performances with high sensitivity at room temperature, which can serve as an excellent alternative to the standard semiconductor metal oxide gas sensors that require high optimal working temperatures for good response [25].

The main novelty of this work is to analyze the sensing nature, adsorption behavior, and the changes in the electron transport properties of S and Mo vacancy created  $MoS_2$  towards  $PH_3$  gas molecule. In recent works, researchers have been using  $MoS_2$  for gas sensing applications. But we have analyzed the sensing nature of S and Mo vacancy created  $MoS_2$  towards  $PH_3$  gas molecule. Moreover, we have also constructed the armchair and zigzag device with two electrodes model and analyzed the changes in the electron transport properties using DFT combined with Non-Equilibrium Green's Function (NEGF) for  $PH_3$  gas adsorption.

## Computational details

A modeled structure of the  $MoS_2$  device with S and Mo vacancy is represented in Fig. 1. We have constructed the  $MoS_2$  device with 50 Mo atoms and 100 S atoms. The device consists of three parts, i.e., the left and right electrodes and central scattering region. The size of the central region is  $19.55 \text{ \AA}$  (the central region is long enough to study the adsorption behavior of the gas molecule), and the size of the electrode is  $3.16 \text{ \AA}$  which is chosen in such a way to study the effect of adsorption between the  $MoS_2$  monolayer and the  $PH_3$  gas molecule.

The electron transport properties for adsorption effects of  $PH_3$  gas molecule on S and Mo vacancy  $MoS_2$  monolayer are performed using DFT combined with NEGF [26]. The empirical correction DFT+D2 has been used to correct the effect of van der Waals



**Fig. 1** Optimized structure of **a** AmS-MoS<sub>2</sub> device, **b** AmMo-MoS<sub>2</sub> device, **c** ZigS-MoS<sub>2</sub> device, and **d** ZigMo-MoS<sub>2</sub> device and **e** PH<sub>3</sub> gas molecule

interaction [27]. We utilized the virtual NanoLab simulation tool for constructing the MoS<sub>2</sub> device and the QuantumWise Atomistix Toolkit (ATK) package for performing the DFT calculations [27, 28].

The optimized geometry of Am-MoS<sub>2</sub> and Zig-MoS<sub>2</sub> with S and Mo vacancy is represented in Fig. 1. For optimization, we have used Linear Combination of Atomic Orbitals (LCAO) as basic set and Double Zeta plus polarization to solve the Kohn–Sham equations. For the geometrical optimization, we have used generalized gradient approximation (GGA)

as an exchange–correlation function with Perdew–Burke–Ernzerhof (PBE) functional [19]. For energy tolerance, we have set the convergence criteria of  $1.0 \times 10^{-5}$  and the maximum force of 0.002 Ha/Å and 0.005 Å for the displacement of geometrical optimization. For accuracy calculation, the cut-off ratio has been set as 5.0 Å in the real space grid [27]. Zone integration is sampled with  $2 \times 1 \times 100$  grid mesh along *x*-, *y*-, and *z*-directions, where the electron transport is along the *z*-direction. For relaxation calculation, the lattice parameters were set as

$a = b = 12.66 \text{ \AA}$  and  $c = 20.00 \text{ \AA}$  [10, 27]. The atomic positions of all the geometries were fully optimized until the force on each atom becomes less than  $0.05 \text{ eV/\AA}$  [10, 27, 29]. The temperature of the electron is taken as  $300 \text{ K}$  throughout the calculations [29–31]. The transmission spectrum was calculated using the transmission function at a particular bias. The transmission function  $T(E, V)$  is given:

$$T(E, V) = Tr[\Gamma_L(E, V)G^R(E)\Gamma_R(E, V)G^A(E)] \quad (1)$$

Here,  $\Gamma_L$  and  $\Gamma_R$  represent the contact broadening functions of the left and right electrodes respectively.  $G^R$  and  $G^A$  represent the retarded advance Green's function. The current  $I(V)$  can be calculated using  $T(E, V)$ , and it can be written as follows:

$$I(V) = \frac{2e^2}{h} \int_{\mu_L}^{\mu_R} [f(E - \mu_L) - f(E - \mu_R)] T(E, V) dE \quad (2)$$

where  $e$ ,  $h$ ,  $f$ , and  $E$  represented the electron charge, Plank's constant, Fermi function, energy respectively.  $\mu_L$  and  $\mu_R$  are the chemical potentials of the left and right electrodes.

The adsorption energy between the  $\text{PH}_3$  gas molecule and armchair and zigzag  $\text{MoS}_2$  monolayer with S and Mo vacancy is defined as follows:

$$E_{\text{ads}} = E_{\text{PH}_3\text{V}_{\text{S/Mo}}\text{MoS}_2} - (E_{\text{V}_{\text{S/Mo}}\text{MoS}_2} + E_{\text{PH}_3}) \quad (3)$$

where  $E_{\text{PH}_3\text{V}_{\text{S/Mo}}\text{MoS}_2}$  represents the total energy of S/Mo vacancy  $\text{MoS}_2$  device after  $\text{PH}_3$  adsorption on.  $E_{\text{V}_{\text{S/Mo}}\text{MoS}_2}$  represents the total energy of S/Mo vacancy created  $\text{MoS}_2$  device.  $E_{\text{PH}_3}$  represents the total energy of the gas molecule. We have used Mulliken population analysis for charge transfer ( $Q$ ) calculation. The difference between the actual valence of S/Mo atom and the valency charge obtained from Mulliken population analysis gives the charge of each and every individual atom in  $\text{V}_\text{S}$  and  $\text{V}_{\text{Mo}}$   $\text{MoS}_2$  monolayer. The net charge transfer of the system is calculated by the summation of all the differences obtained from Mulliken population analysis. From the net total, the negative sign indicated the charge transfer from  $\text{V}_\text{S}/\text{V}_{\text{Mo}}$   $\text{MoS}_2$  monolayer to  $\text{PH}_3$  gas molecule and the positive value indicated the charge transfer from the  $\text{PH}_3$  gas molecule to the  $\text{V}_\text{S}/\text{V}_{\text{Mo}}$   $\text{MoS}_2$  monolayer [10, 29] which are represented in Table 1. To get a deeper understanding of the electronic and

**Table 1** The values of adsorption distance ( $d$ ), adsorption energy ( $E_{\text{ads}}$ ), charge transfer ( $Q$ ), and changes in the bond length and bond angle

MoS <sub>2</sub> model	S/Mo vacancy	Adsorption distance $d(\text{\AA})$	Bond length ( $\text{\AA}$ )			Bond angle (deg)		Adsorption energy $E_{\text{ads}}$ (eV)	Charge transfer $Q(e)$
			(H <sub>1</sub> P)	(H <sub>2</sub> P)	(H <sub>3</sub> P)	(H <sub>1</sub> PH <sub>2</sub> )	(H <sub>2</sub> PH <sub>3</sub> )		
PH <sub>3</sub> before adsorption	-	-	1.4200	1.4200	1.4200	93.50°	93.50°	93.5°	-
	S	2.6837	1.4262	1.3973	1.4404	96.72°	97.78°	96.17°	-0.9006
Armchair	Mo	2.3420	1.4428	1.4439	1.4433	98.17°	98.19°	98.32°	-1.8048
	S	2.5204	1.4461	1.4429	1.4431	96.38°	96.04°	96.12°	-0.1234
Zigzag	Mo	2.3627	1.4503	1.5142	1.4532	93.23°	114.30°	85.59°	-0.3522
	S	2.3627	1.4503	1.5142	1.4532	93.23°	114.30°	85.59°	-0.3522



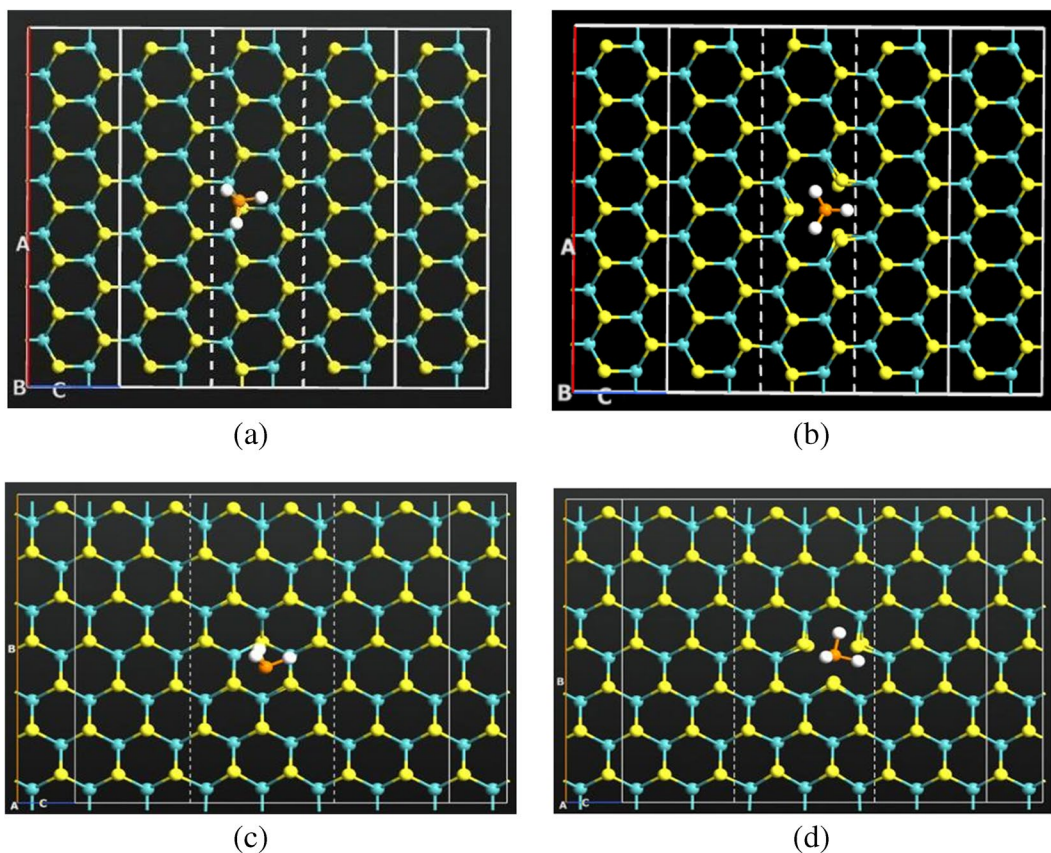
sensing property, the changes caused in the charges after the adsorption of gas molecule are analyzed. In Fig. 1, the bond length between Mo atom and S atom is 2.415 Å, and between two S is 3.131 Å.

## Results and discussion

To understand the adsorption effects of  $\text{PH}_3$  gas molecule towards  $\text{MoS}_2$  device, different adsorption configurations are calculated. The results include DOS, transmission spectrum, I-V curve, adsorption energy, and charge transfer.

Figure 2a–d shows the optimized structure of  $\text{PH}_3$  adsorption on armchair S vacancy  $\text{MoS}_2$  monolayer ( $\text{AmS-MoS}_2\text{PH}_3$ ),  $\text{PH}_3$  adsorption on armchair Mo vacancy  $\text{MoS}_2$  monolayer ( $\text{AmMo-MoS}_2\text{PH}_3$ ),  $\text{PH}_3$  adsorption on zigzag S vacancy  $\text{MoS}_2$  monolayer ( $\text{ZigS-MoS}_2\text{PH}_3$ ), and  $\text{PH}_3$

adsorption on zigzag Mo vacancy  $\text{MoS}_2$  monolayer ( $\text{ZigMo-MoS}_2\text{PH}_3$ ), respectively. In order to get a more stable configuration, the optimized  $\text{PH}_3$  gas molecule is placed vertically above the vacancy created on the  $\text{MoS}_2$  monolayer. From Fig. 2, we observed that there is a non-bonding interaction between the device and the gas molecule. This shows that the adsorption is through physisorption [32, 33]. After the adsorption, the structure of the  $\text{PH}_3$  gas molecule dislocates, where the bond length P–H and bond angle HPH have been increased and decreased, and the changes are listed in Table 1. The changes in the bond length and bond angles are caused due to the van der Waals interaction [33]. From Table 1, we observed that the  $\text{PH}_3$  adsorption on  $\text{AmMo-MoS}_2$  and  $\text{ZigMo-MoS}_2$  shows more change in the bond length and bond angle. For  $\text{AmMo-MoS}_2\text{PH}_3$ , the changes in the bond length are 1.4428 Å, 1.4439 Å, and 1.4433 Å



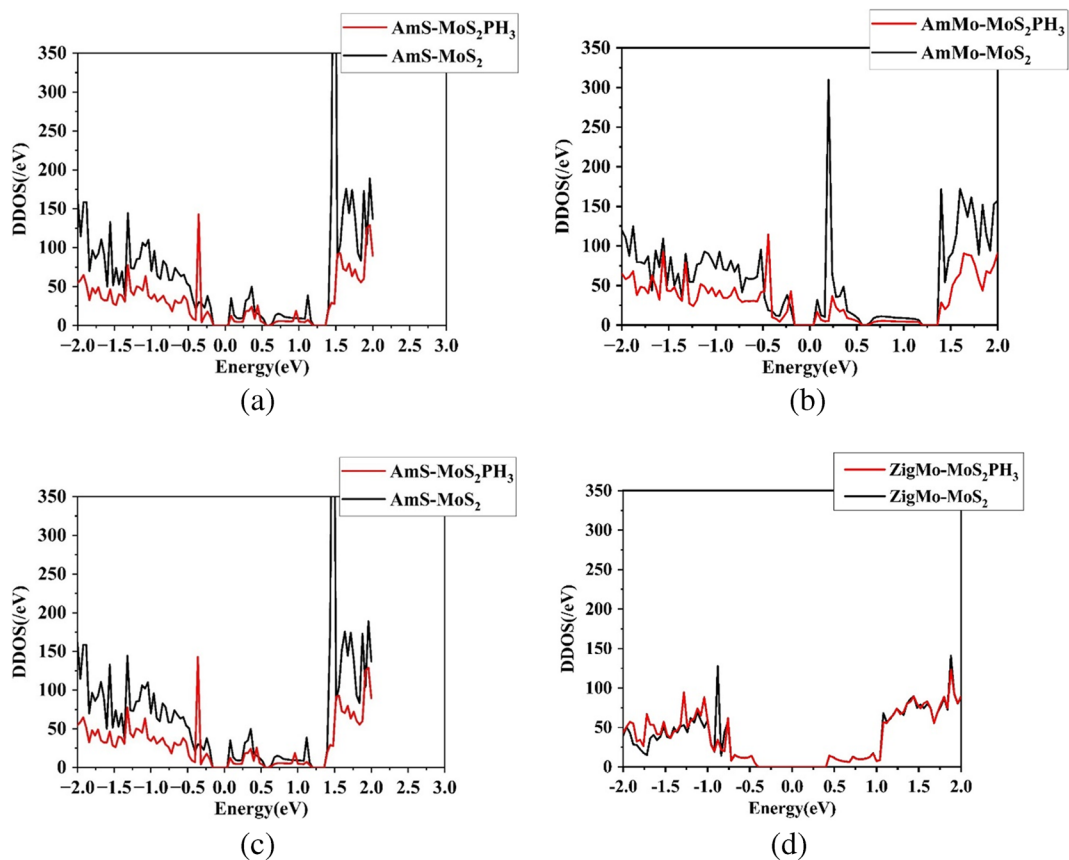
**Fig. 2** Optimized structure of **a**  $\text{AmS-MoS}_2\text{PH}_3$  device, **b**  $\text{AmMo-MoS}_2\text{PH}_3$  device, **c**  $\text{ZigS-MoS}_2\text{PH}_3$  device, and **d**  $\text{ZigMo-MoS}_2\text{PH}_3$  device

for P-H<sub>1</sub>, P-H<sub>2</sub>, and P-H<sub>3</sub> (Fig. 1e), and the changes in the bond angles are 98.17°, 98.19°, and 98.32° for H<sub>1</sub>PH<sub>2</sub>, H<sub>2</sub>PH<sub>3</sub>, and H<sub>3</sub>PH<sub>1</sub>, respectively. Similarly, for ZigMo-MoS<sub>2</sub>-PH<sub>3</sub> the changes in the bond length are 1.4503 Å, 1.5142 Å, and 1.4532 Å, and the changes in the bond angle are 93.23°, 114.30°, and 85.59°, respectively. The adsorption distance, adsorption energy, and charge transfer of the four devices are also listed in Table 1. From the Table 1, AmMo-MoS<sub>2</sub> and ZigMo-MoS<sub>2</sub> show less adsorption distance of 2.3420 Å and 2.3626 Å, more adsorption energy -1.1048 eV and -0.3522 eV, and more charge transfer of 0.112e and -0.18e, respectively. From this, we observed that the PH<sub>3</sub> adsorption on AmMo-MoS<sub>2</sub> and ZigMo-MoS<sub>2</sub> is comparatively more.

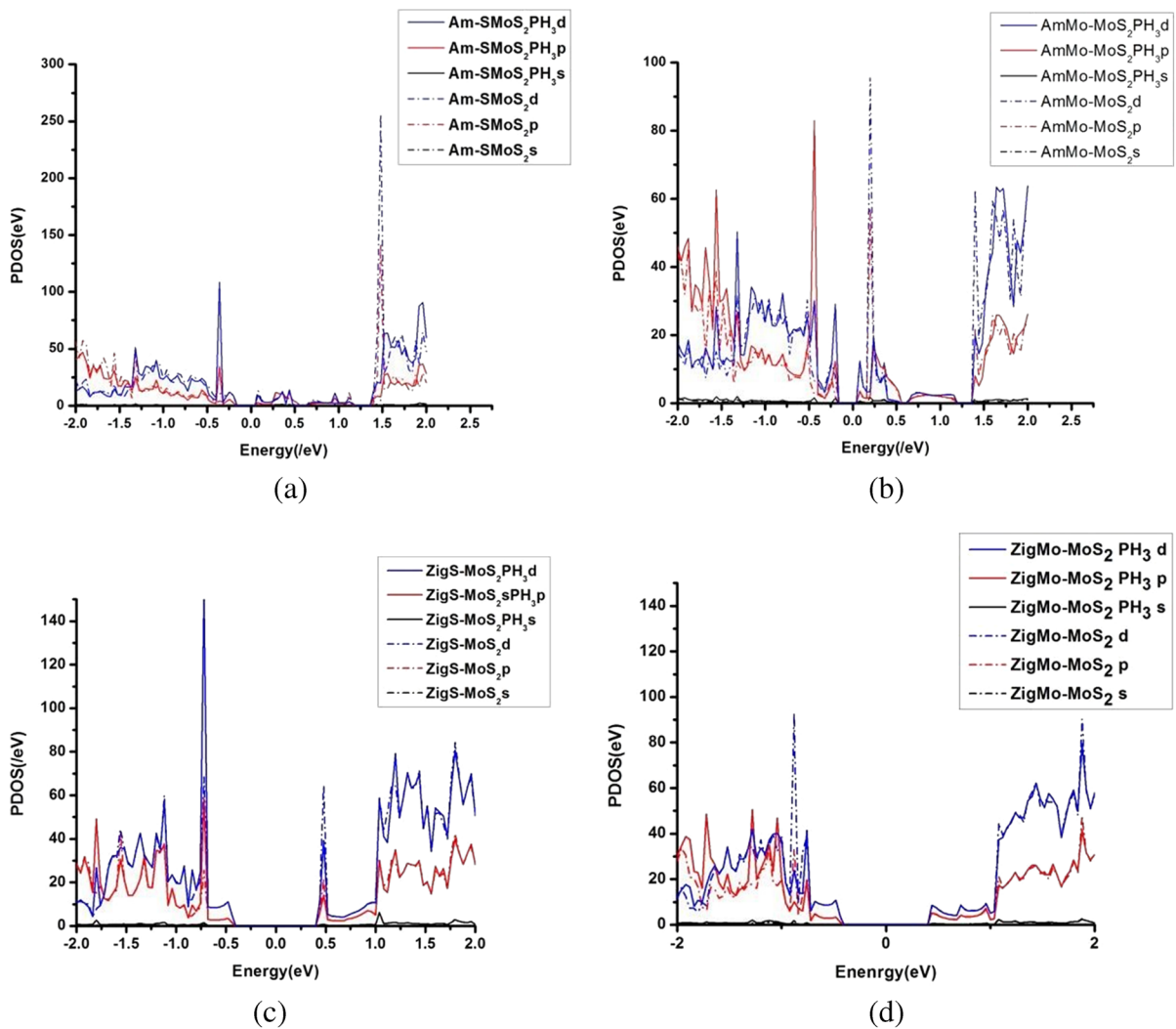
Figure 3 shows the TDOS curve of PH<sub>3</sub> adsorption on AmS-MoS<sub>2</sub>, AmMo-MoS<sub>2</sub>, ZigS-MoS<sub>2</sub>, and ZigMo-MoS<sub>2</sub>. The Fermi Energy ( $E_F$ ) is set

to 0 eV, and this represents the zero carrier density at the Fermi region. All the four systems have not produced any changes in the bandgap after the adsorption of PH<sub>3</sub> gas molecule; this shows that adsorption does not introduce any mid-gap states, but near the Fermi region, some states of the AmS/Mo-MoS<sub>2</sub> has been changed after the PH<sub>3</sub> adsorption. This leads to a change in the electrical conductivity. For PH<sub>3</sub> adsorption on ZigS and ZigMo-MoS<sub>2</sub> (Fig. 3c, d), no changes are observed near the Fermi region but ZigS/Mo-MoS<sub>2</sub> shows a slight change (-2.0 to -0.5 eV) after the adsorption of PH<sub>3</sub> gas molecule. This indicates that the adsorption of PH<sub>3</sub> gas molecule considerably affects the electrical conductivity of the AmS/Mo-MoS<sub>2</sub> when compared with ZigS/Mo-MoS<sub>2</sub>.

Figure 4a-d shows the projected density of states (PDOS) of AmMo-MoS<sub>2</sub>, AmS-MoS<sub>2</sub>, ZigS-MoS<sub>2</sub>, and ZigMo-MoS<sub>2</sub> before and after the



**Fig. 3** TDOS of **a** AmS-MoS<sub>2</sub> and **b** AmMo-MoS<sub>2</sub> and **c** ZigS-MoS<sub>2</sub> and **d** ZigMo-MoS<sub>2</sub> before and after adsorption of PH<sub>3</sub> gas molecule

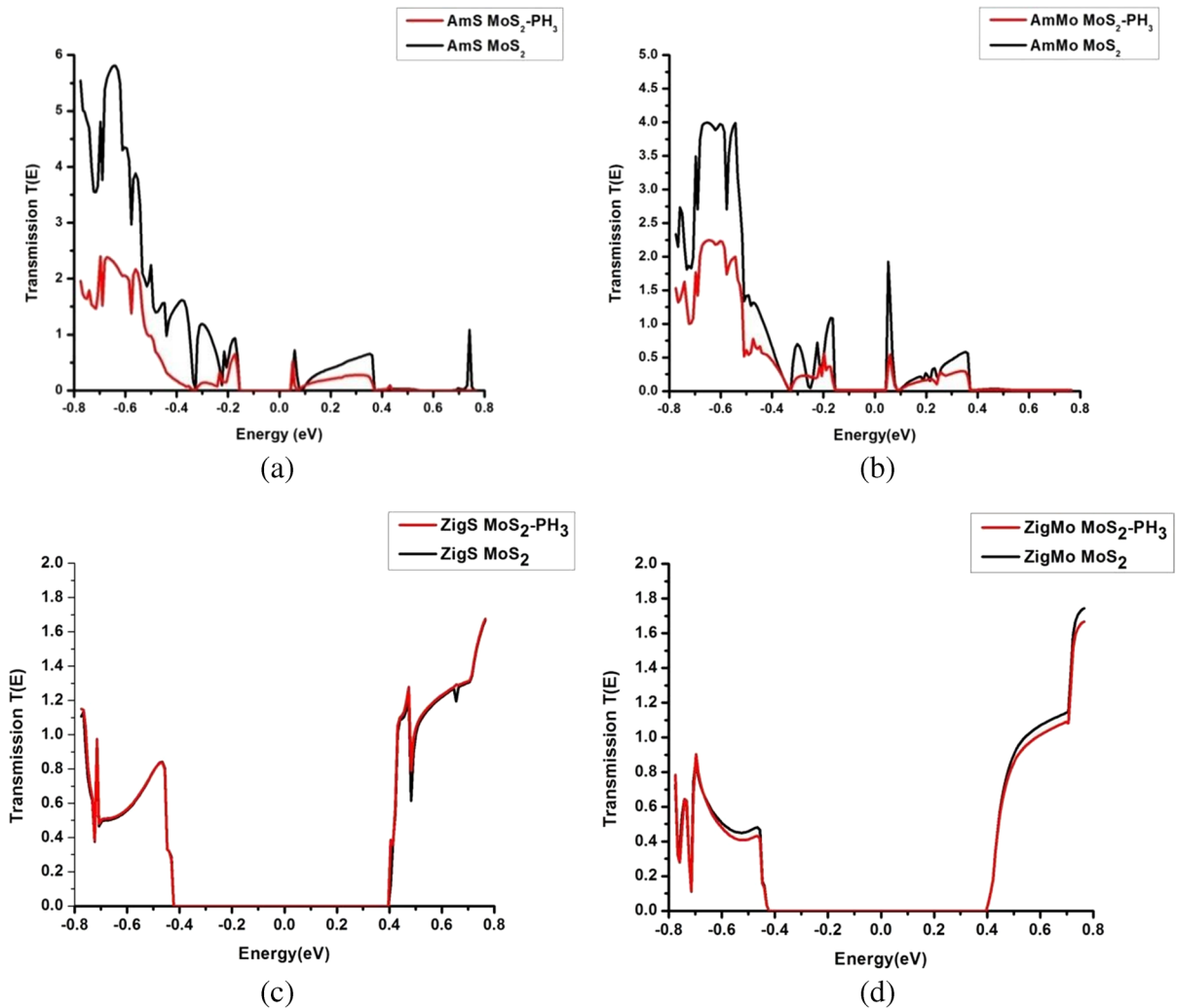


**Fig. 4** PDOS of **a** AmS-MoS<sub>2</sub> and **b** AmMo-MoS<sub>2</sub> and **c** ZigS-MoS<sub>2</sub> and **d** ZigMo-MoS<sub>2</sub> before and after adsorption of PH<sub>3</sub> gas molecule

adsorption of PH<sub>3</sub> gas molecule. From Fig. 4, we observed that the PH<sub>3</sub> gas molecule affects the p and d orbitals of all the four systems, thus causes the changes in the TDOS after the adsorption on PH<sub>3</sub> gas molecule. Near the Fermi region, most of the peaks of s and d orbitals have been changed after the adsorption of PH<sub>3</sub> gas molecule. This indicates that there is a significant charge transfer occurred between PH<sub>3</sub> gas molecule and MoS<sub>2</sub> device and no bond formation between the gas molecule and MoS<sub>2</sub> device. This indicates that the changes in the peaks are caused due to van der Waals’s force between the gas molecule and the

system [10]. Due to these changes, the electrical conductivity of the system was changed.

The transmission spectrum of V<sub>S</sub> and V<sub>MO</sub> on the armchair and zigzag MoS<sub>2</sub> monolayer with and without PH<sub>3</sub> gas molecule are illustrated in Fig. 5a–d. From the figure, we observed that at zero bias, there is zero transmission coefficient and the width of the transmission gap are about 0.22 eV and 0.17 eV for AmS-MoS<sub>2</sub> and AmMo-MoS<sub>2</sub> and 0.82 eV and 0.81 eV for ZigS-MoS<sub>2</sub> and ZigMo-MoS<sub>2</sub>, respectively, and the transmission gap acts as a barrier for electron transmission. This shows that the material has semiconducting



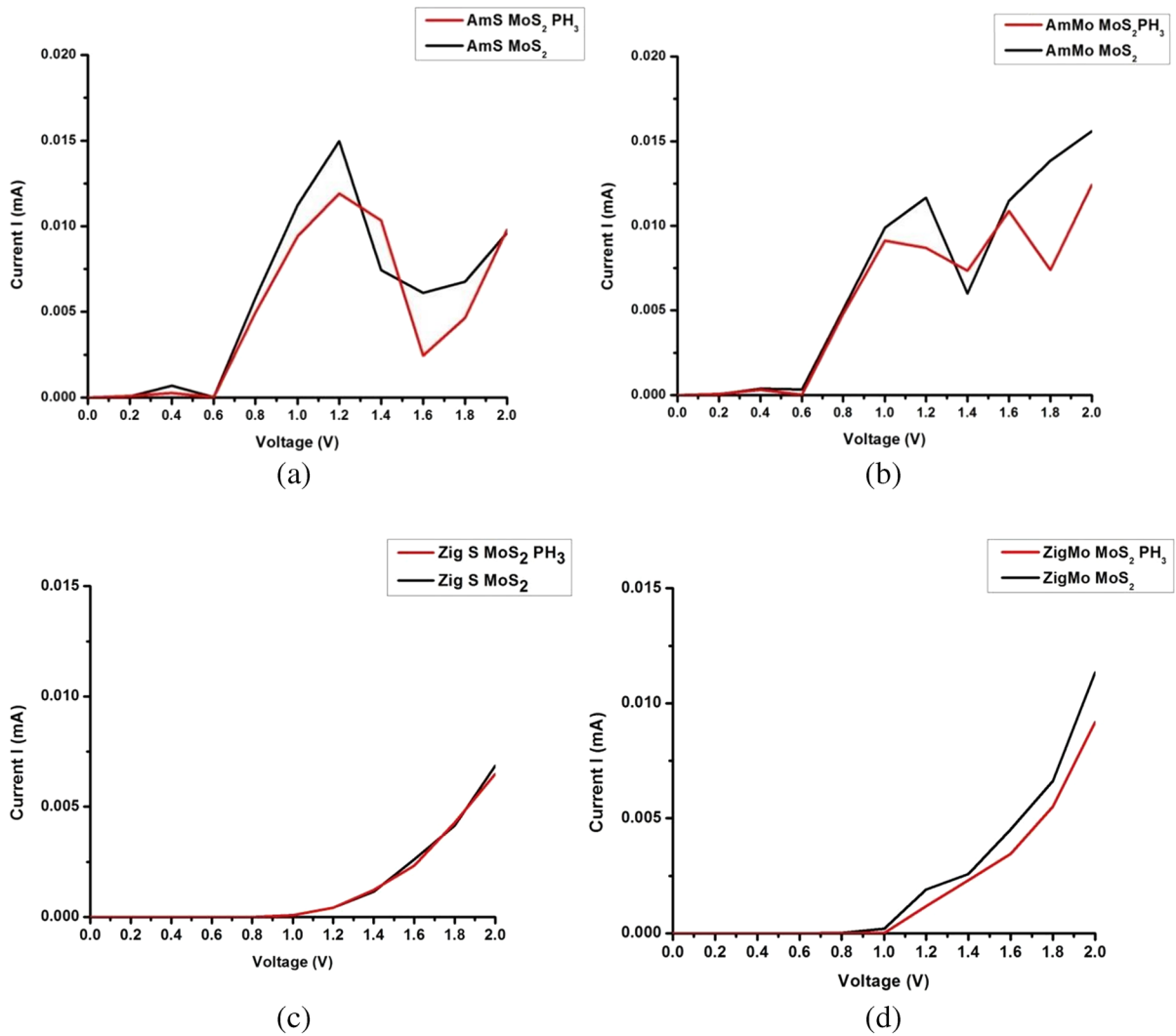
**Fig. 5** Transmission spectrum of **a** AmS-MoS<sub>2</sub> and **b** AmMo-MoS<sub>2</sub> and **c** ZigS-MoS<sub>2</sub> and **d** ZigMo-MoS<sub>2</sub> before and after adsorption of PH<sub>3</sub> gas molecule

nature [34]. For AmS/Mo-MoS<sub>2</sub> (Fig. 5a, b), we observed that there is a decrease in the transmission after the adsorption of PH<sub>3</sub> gas molecule. The peaks in the transmission spectrum indicated the conduction channels. The reduction in the transmission peaks leads to the reduction in the current. This indicates that AmS/Mo-MoS<sub>2</sub> is considerably affected by the PH<sub>3</sub> gas molecule. For ZigS/Mo-MoS<sub>2</sub>, the changes in the transmission are comparatively lower than AmS/Mo-MoS<sub>2</sub>.

To clearly observe the modification in the conductivity and to qualitatively evaluate the performance of the S and Mo vacancy armchair and

zigzag MoS<sub>2</sub> monolayer as a PH<sub>3</sub> sensor, the I–V characteristics of the system were analyzed, and their respective I–V graph is shown in Fig. 6. Figure 6a shows the I–V characteristics of AmS-MoS<sub>2</sub> with and without PH<sub>3</sub> adsorption. It exhibits a non-linear behavior before and after the adsorption of PH<sub>3</sub> gas molecule. Figure 6a shows a linear increase in the current for the bias voltage of about 0.6 to 1.2 V; the increase in the current ( $I_{\text{peak}}$ ) reaches a maximum value of 0.1497 mA before the adsorption of PH<sub>3</sub> and 0.1192 mA after the adsorption of PH<sub>3</sub>. This shows that the  $I_{\text{peak}}$  decreases after the adsorption of the PH<sub>3</sub> gas molecule.





**Fig. 6** I–V characteristics of **a** AmS-MoS<sub>2</sub> and **b** AmMo-MoS<sub>2</sub> and **c** ZigS-MoS<sub>2</sub> and **d** ZigMo-MoS<sub>2</sub> before and after adsorption of PH<sub>3</sub> gas molecule

Beyond the bias voltage of 1.2 to 1.6 V, there is a rapid decrease in the current, and the non-differential resistance (NDR) phenomena were observed [35, 36]. The decrease in the current ( $I_{\text{valley}}$ ) reaches the minimum current value of 0.7345 mA before and 0.2450 mA after the adsorption of PH<sub>3</sub> gas molecule. This shows that there is a change in the conductivity of the material after the adsorption of PH<sub>3</sub> gas molecule. For PH<sub>3</sub> adsorption on AmMo-MoS<sub>2</sub> shows an increase in the current for the bias voltage of about 0.6 to 1.2 V before adsorption and 0.6 to 1.0 V after the adsorption. Here, the  $I_{\text{valley}}$  starts from 1.2 V for AmMo-MoS<sub>2</sub>

and 1.0 V for AmMo-MoS<sub>2</sub>PH<sub>3</sub>; after 1.0 V, there is a reduction in current for AmMo-MoS<sub>2</sub> PH<sub>3</sub> compared with AmMo-MoS<sub>2</sub>. The current reduction indicated the increase in the resistance of the AmS/Mo-MoS<sub>2</sub> PH<sub>3</sub> material after the adsorption of PH<sub>3</sub> gas molecule.

Figure 6c, d represents the I–V characteristic of PH<sub>3</sub> adsorption on ZigS/Mo-MoS<sub>2</sub> monolayer. The magnitude of the current along the ZigS/Mo-MoS<sub>2</sub> is smaller than AmS/Mo-MoS<sub>2</sub>. For PH<sub>3</sub> adsorption on ZigS-MoS<sub>2</sub>, there is no significant changes in the current between before and after the adsorption of PH<sub>3</sub> gas molecule. This shows that PH<sub>3</sub> gas

**Table 2** The values of percentage of sensitivity of PH<sub>3</sub> gas molecule on AmS-MoS<sub>2</sub>, AmMo-MoS<sub>2</sub>, ZigS-MoS<sub>2</sub>, and ZigMo-MoS<sub>2</sub> device under voltage from 0 to 2.0 V

Bias voltage/model	Percentage of sensitivity			
	AmS-MoS <sub>2</sub>	AmMo-MoS <sub>2</sub>	ZigS-MoS <sub>2</sub>	ZigMo-MoS <sub>2</sub>
0.2 V	10.45	16.78	10.78	11.32
0.4 V	32.76	35.70	12.67	12.43
0.6 V	12.67	15.87	13.98	12.99
0.8 V	25.78	14.78	15.89	17.68
1.0 V	50.65	30.67	17.84	20.67
1.2 V	50.85	88.79	19.78	70.69
1.4 V	77.98	34.98	20.65	35.89
1.6 V	70.57	40.87	23.98	60.45
1.8 V	62.87	96.87	24.67	50.65
2.0 V	13.85	86.65	24.99	72.75

molecule does not causes any change in the conductivity of the material. For PH<sub>3</sub> adsorption on ZigMo-MoS<sub>2</sub>, the current flow is zero until the bias voltage is 0.9 V before the absorption of PH<sub>3</sub> and 1.0 V after the adsorption of PH<sub>3</sub> gas molecule. After this, the current starts increasing dramatically. Under the bias voltage of 2.0 V, the current flow through the material is 0.017 mA before and 0.014 mA after the adsorption of PH<sub>3</sub> gas molecule. This shows that the adsorption of PH<sub>3</sub> gas molecule is slightly more in ZigMo-MoS<sub>2</sub> when compared with the ZigS-MoS<sub>2</sub>. The conductance for the armchair S/Mo vacancy MoS<sub>2</sub> monolayer is non-linear. The changes in the conductance before and absence of PH<sub>3</sub> gas molecule attribute to the response of the sensor. Therefore, for estimating the sensor response, the selectivity of the sensor is calculated using the following:

$$S = |G - G_0|/G_0 \quad (4)$$

where  $G$  and  $G_0$  represent the conductance of Am/ZigS/Mo-MoS<sub>2</sub> after and before the gas adsorption of PH<sub>3</sub> gas molecule, respectively [27].

The estimated percentage of sensitivity values of the system after the adsorption of PH<sub>3</sub> gas molecule is listed in Table 2. Here, all the system shows different sensitivity at different bias voltage. From Table 2, we observed that the PH<sub>3</sub> adsorption on AmMo-MoS<sub>2</sub> shows more adsorption energy,

charge transfer, and percentage of sensitivity of  $-1.8048$  eV,  $-0.2120$ e, and 96.87% under the bias voltage of 1.8 V when compared with AmS-MoS<sub>2</sub>, ZigS-MoS<sub>2</sub>, and ZigMo-MoS<sub>2</sub> [13].

## Conclusion

In this study, we have investigated the sensing behavior and electron transport property of S/Mo vacancy armchair and zigzag MoS<sub>2</sub> monolayer using NEGF-DFT techniques. The result shows that the PH<sub>3</sub> gas molecule is allowed to be adsorbed in all the four systems through van der Waals interaction. The structural optimization results indicate that the changes in the bond length, bond angle, and the value of the adsorption energy and charge transfer of PH<sub>3</sub> adsorption on AmMo-MoS<sub>2</sub> are more when compared with AmS-MoS<sub>2</sub>, ZigS-MoS<sub>2</sub>, and ZigMo-MoS<sub>2</sub>. PH<sub>3</sub> adsorption on AmMo-MoS<sub>2</sub> shows better adsorption. Moreover, the changes in the PDOS and transmission spectrum are comparatively more in AmS/Mo-MoS<sub>2</sub> which leads to more changes in the I-V curve of AmS/Mo MoS<sub>2</sub>, and ZigMo-MoS<sub>2</sub> indicated that there will be a change in the conductivity of the material after the adsorption of PH<sub>3</sub> gas molecule. Moreover, the NDR behavior was observed in both AmMo/S-MoS<sub>2</sub>. AmMo-MoS<sub>2</sub> shows more changes in the current at  $I_{\text{vally}}$  after the adsorption of PH<sub>3</sub> gas molecule. Thus, we concluded that AmMo-MoS<sub>2</sub> shows better adsorption towards PH<sub>3</sub> gas molecule when compared with the other devices.

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**Data availability** The raw/processed data required to reproduce these findings cannot be shared at this time due to technical or time limitations.

**Compliance with ethical standards**

**Conflict of interest** The authors declare no competing interests.

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