



# A wide solar spectrum light harvesting Ag<sub>2</sub>Se quantum dot-sensitized porous TiO<sub>2</sub> nanofibers as photoanode for high-performance QDSC

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**Abstract** A wide spectrum of light harvesting silver selenide (Ag<sub>2</sub>Se) quantum dots (QDs) with an average size of ~5 nm has been synthesized by a low-temperature one-pot hot injection method. To finely control the size of Ag<sub>2</sub>Se QDs, oleylamine was used as the solvent, and dodecanethiol was used as the capping agent as well as a stabilizer. The prepared Ag<sub>2</sub>Se QDs were ex situ sensitized on the porous TiO<sub>2</sub> nanofiber (NF) substrate via a direct adsorption method to use as efficient photoanode for quantum dot-sensitized solar cell (QDSC). The UV-Vis-NIR absorption and photoluminescence studies revealed that Ag<sub>2</sub>Se QD-sensitized porous TiO<sub>2</sub> NF photoanode (Ag<sub>2</sub>Se/P-TiO<sub>2</sub>) exhibited near-infrared (NIR) absorption and fast electron kinetics. Further, the QDSC fabricated using Ag<sub>2</sub>Se/P-TiO<sub>2</sub> NFs as the photoanode, Cu<sub>2</sub>S as the counter electrode, and liquid polysulfide (S<sup>2-</sup>/S<sub>x</sub><sup>2-</sup> redox couple) as the electrolyte exhibited a photoconversion efficiency of 2.50% with an improved photocurrent density of 11.12 mA/cm<sup>2</sup>.

**Keywords** Electrospinning · Ag<sub>2</sub>Se quantum dots · TiO<sub>2</sub> nanofibers · Quantum dot-sensitized solar cell

## Introduction

Quantum dot-sensitized solar cells (QDSCs) have garnered immense interest on account of the ability of quantum dots that elicited a theoretical efficiency up to ~44%. This higher efficiency (more than the Shockley-Queisser limit (~31%)) is attributed to the unique optical properties of QDs such as bandgap tuning by size control, large dipole interaction, resistivity to photobleaching, high extinction coefficient, and multiple exciton generations by means of impact ionization (Kamat 2013; Kouhnavard et al. 2014). The steady progress in the class of QDs has mainly focused on binary QDs to explore as sensitizers in QDSCs. Cd- and Pb-based QDs such as CdSe (Gao et al. 2009), CdS (Lai et al. 2014), and PbSe (Benekohal et al. 2012) have been studied extensively as sensitizers; however, its practical applicability inflicts the resolution of a green and clean world. Hence, replacement with environment benign QDs has to turn into the trend in QDSCs. In addition to toxicity issue, the broad light absorbing range, high CB edge position than semiconductor metal oxide, and adsorption of maximum QDs over the photoanode substrate are the relevant factors need to focus on. In this aspect, a low toxic Ag<sub>2</sub>Se QDs came into attention due to its narrow bandgap (~0.15 eV) which could capable to absorb more solar flux from the visible to the near-infrared region (Sahu et al. 2012; Langevin et al. 2013). The small size of Ag<sub>2</sub>Se QDs would also help to tune the conduction band edge and cause fast electron transfer to the CB of semiconductor oxide film substrate (Kamat 2008). Suitability of

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Ag<sub>2</sub>Se QDs as the sensitizer has been studied by Tubtimtae et al., where Ag<sub>2</sub>Se QDs is found to achieve low efficiency of 1.57%. This low performance of Ag<sub>2</sub>Se QD-based QDSC might have contributed to in situ sensitization of QDs over the TiO<sub>2</sub> NP film and electron hopping while carrying to the external circuit of the device. The slow electron transport, electron-hole recombination, and poor coverage of QDs on the photoanode can be overcome by employing Ag<sub>2</sub>Se QDs over the porous TiO<sub>2</sub> NFs for proper adsorption of Ag<sub>2</sub>Se QDs on the surface of the photoanode substrate.

The in situ sensitization such as chemical bath deposition (CBD) and successive ionic layer adsorption and reaction (SILAR) for QDs adsorption has found the cause for lower photoconversion efficiency in the QDSCs due to the wide range of size distribution and disparity in desired morphology, whereas an ex situ sensitization method encourages the formation of small-sized QDs and tuning of bandgap as well (Lai et al. 2014). Most of the ex situ adsorption are done by using a bilinker agent like 3-mercaptopropionic acid (MPA), while few preferred to anchor via direct adsorption method, where direct adsorption has the advantage of maximum adsorption on the surface of photoanode and fast electron transport kinetics in comparison with the assistance by bilinker agent (Pemik et al. 2011).

In order to increase the photocurrent density of QDSCs, the maximum uptake of QDs over the photoanode substrate have been made to prevent the electron-hole recombination. The porous TiO<sub>2</sub> NF-based photoanode has offered a better way to improve the power conversion efficiency of QDSC; this would accommodate the high quantity of QDs over the surface by providing high surface area and proper contact between the counter electrode and electrolyte. Electrospun TiO<sub>2</sub> NFs have the higher surface area for superior adsorption of QDs and faster semidirected electron transport and lesser transit time to the FTO plate in addition to internal light scattering, and the highly fibrous structure which facilitates higher electrolyte uptake than the conventional nanoparticle morphology.

In the present investigation, we prepared Ag<sub>2</sub>Se QDs by a simple hot injection method at a low temperature using 1-DDT as a capping agent. Later, these QDs are directly adsorbed on the porous TiO<sub>2</sub> NFs in a solvent medium. Finally, the optical and photovoltaic performances of Ag<sub>2</sub>Se QD-sensitized porous TiO<sub>2</sub> NF-based photoanode for QDSC are studied in detail.

## Experimental

### Materials

Titanium IV isopropoxide (TiP, 99%), polyvinyl pyrrolidone (PVP,  $M_w = 1,300,000$ ), silver chloride (AgCl, 99.9%), selenium powder (99.9%), oleylamine (technical grade, 70%), and 1-dodecanethiol (> 98%) were procured from Sigma-Aldrich. Methanol (> 99.5%), ethanol (> 99%), hexane (> 95%), acetic acid glacial (99–100%), and glycerol (GR 87%) were purchased from Merck India Ltd. and used without further purification.

### Synthesis of Ag<sub>2</sub>Se QDs

Ag<sub>2</sub>Se QDs were prepared by a simple hot injection method using oleylamine as the solvent and dodecanethiol as the effective capping agent as well as a stabilizer for the reaction to control the particle size. The synthesis was carried out in two steps. First, selenium precursor solution was prepared in which 2 mM of Se powder in 10 mL of oleylamine was taken in a two-neck round-bottom flask and then refluxed at 150 °C for 1 h to form Se-OAm complex. Second, 0.4 mM of silver precursor, 5 mL of OAm (solvent), and 1 mL of 1-DDT (capping agent) were taken in a 50-mL round-bottom flask and heated initially up to 130 °C under N<sub>2</sub> gas atmosphere. At this temperature, the preprepared OAm-Se complex solution was injected into the flask under continuous stirring for 15 min. In this condition, it starts the process of nucleation and growth which led to the formation of quantum dots that was evident by the change in color from yellow to the dark gray. Later, the reaction was stopped, and then, the solution was left for cooling followed by precipitation by adding 10 mL of cold methanol. The pure QDs were obtained after centrifugation using by ethanol and methanol as the solvent, and then they were dispersed in a nonpolar solvent (hexane).

### Preparation of Ag<sub>2</sub>Se-sensitized porous TiO<sub>2</sub> NFs

The highly porous TiO<sub>2</sub> NFs were prepared by the solvasonication process by dispersing conventional TiO<sub>2</sub> NFs in glycerol as a pore-forming solvent. The ultrasonication (Leela Sonic, 50 W and 30 kHz) was done for 90 min. Then, the TiO<sub>2</sub> NFs are removed from glycerol and calcined at 300 °C for 10 min in a muffle

furnace to get highly porous TiO<sub>2</sub> NFs. This porous TiO<sub>2</sub> NF has interconnected porous structure with higher porosity and surface area than the conventional TiO<sub>2</sub> NFs (Nisha Singh et al. 2018).

The porous TiO<sub>2</sub> NF paste was then prepared by mixing 30 wt% of porous TiO<sub>2</sub> NFs with terpineol (50 wt%), dibutyl phthalate (5 wt%), and ethyl cellulose (15 wt%) in ethanol. The FTO glass substrates (12 Ω sq<sup>-1</sup>) were thoroughly rinsed with deionized water, acetone, and ethanol in an ultrasonic bath for 5 min and dried in a hot air oven. The prepared porous TiO<sub>2</sub> NF paste was coated on pretreated FTO substrates by a doctor blade technique using adhesive scotch tape so that the thickness was maintained at 11–12 μm and then the coatings were sintered at 450 °C for 30 min and cooled to 80 °C. The coated substrate was then dipped in Ag<sub>2</sub>Se QDs colloidal solution (prepared from 5 mg QDs dispersed in 5 mL of hexane) for 24 h to complete the adsorption of Ag<sub>2</sub>Se QDs onto the porous TiO<sub>2</sub> NF-based photoanode. The resultant photoanode was finally rinsed with hexane and then dried.

### Physical characterization

The phase purity and crystal structure of prepared Ag<sub>2</sub>Se QDs and Ag<sub>2</sub>Se QD-sensitized porous TiO<sub>2</sub> NFs were confirmed by X-ray diffraction analysis (Rigaku, Ultima IV) using Cu-Kα source over the scan range of 20–80° with an increment of 0.02°. The particle size of Ag<sub>2</sub>Se QDs was confirmed by the dynamic light scattering (DLS) technique (Malvern Panalytical), and the attachment of Ag<sub>2</sub>Se QDs on the porous TiO<sub>2</sub> NFs was confirmed by transmission electron microscope (JEOL, FEG-TEM, JSM-7600F) with selected area electron diffraction (SAED) pattern. The morphological analysis with the elemental composition of Ag<sub>2</sub>Se QDs and Ag<sub>2</sub>Se QD-sensitized porous TiO<sub>2</sub> NFs was determined by using scanning electron microscopy (Hitachi, model: S-4200) coupled with energy-dispersive X-ray spectrometer (EDX).

The absorbance spectra were recorded at 300–1200 nm using UV-Vis-NIR spectrophotometer (PerkinElmer, model: L-650) for Ag<sub>2</sub>Se QDs and Ag<sub>2</sub>Se QD-sensitized porous TiO<sub>2</sub> NFs. Photoluminescence studies were performed using a luminescence spectrometer (NanoLog, HORIBA Jobin Yvon) with xenon lamp as the source of excitation for Ag<sub>2</sub>Se QDs and Ag<sub>2</sub>Se QD-sensitized porous TiO<sub>2</sub> NFs.

The performance of fabricated Ag<sub>2</sub>Se QD-sensitized porous TiO<sub>2</sub> NF-based QDSCs was determined by a calibrated AM 1.5 solar illumination (Newport, Oriel Instruments, USA; 150 W, model: 67005) with a light intensity of 100 mWcm<sup>-2</sup> calibrated using standard silicon solar cell (Newport, Oriel Instruments, model: 91150 V) and a computer-controlled digital source meter (Keithley, model: 2420). Electrochemical AC impedance measurements (Biologic VSP, France) were carried out under AM 1.5 sun illumination at room temperature in the frequency range of 1 mHz to 100 kHz with an AC amplitude of 10 mV (Subramania et al. 2015)

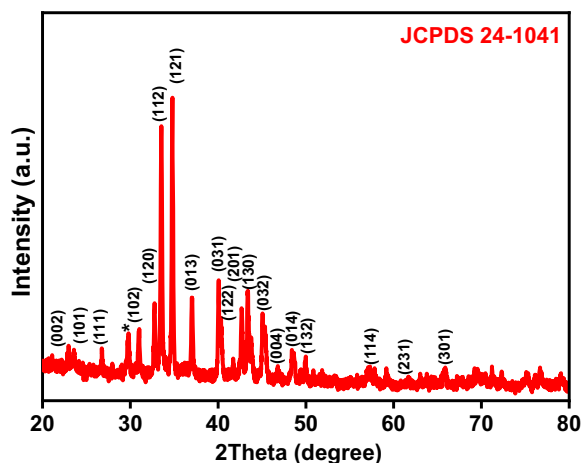
### Fabrication of QDSC devices

The photoanode was prepared as in the “[Preparation of Ag<sub>2</sub>Se-sensitized porous TiO<sub>2</sub> NFs](#)” section, and the counter electrode was prepared by mixing 95 wt% of Cu<sub>2</sub>S nanoparticles with 5 wt% N-methyl-2-pyrrolidone (binder) and coated on pretreated FTO glass plates by the doctor blade technique and then dried at 80 °C for overnight in a vacuum oven to remove excess solvents. The prepared photoanode and the counter electrode were sealed using 60-μm surlyn hot melt spacer, and the space of cell was filled with polysulfide redox electrolyte containing 2 M Na<sub>2</sub>S, 2 M S, and 0.2 M KCl in 3:7 (v/v) mixture of methanol and deionized water through the predrilled holes in the counter electrode. Later, the holes were closed by using a surlyn strip and left as such for few hours so that complete diffusion of the electrolyte into both electrodes takes place. The active area of the cell was 0.30 cm<sup>2</sup>.

## Results and discussion

### Characterization of Ag<sub>2</sub>Se QDs

The X-ray diffraction (XRD) pattern of prepared Ag<sub>2</sub>Se QDs is shown in Fig. 1. Its characteristic peaks resemble the orthorhombic low-temperature β-phase of Ag<sub>2</sub>Se (Hu et al. 2000) This is well matched with the JCPDS Card No. 024-1041. No significant peaks corresponding to the organic ligands are observed. The peak at  $\theta = 29.9^\circ$  (represented by \*) does not match with any silver selenide pattern (Panneerselvam et al. 2009). The broadening of XRD peaks is attributed to the smaller size of Ag<sub>2</sub>Se QDs. It has been reported that organic ligands play a significant role in the stability of the quantum dots

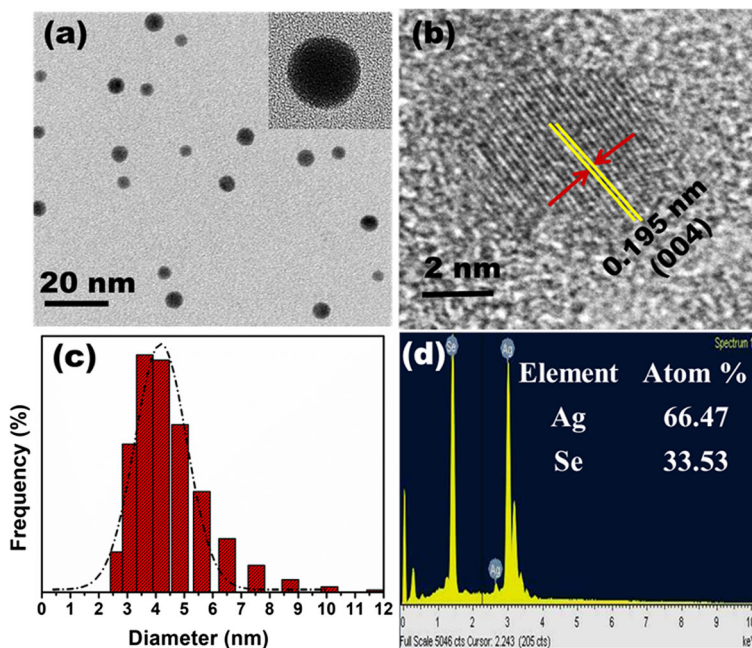


**Fig. 1** XRD pattern of  $\text{Ag}_2\text{Se}$  QDs

(Chen et al. 1997; Wickham et al. 2000). On nucleation of  $\text{Ag}_2\text{Se}$  QDs, the organic ligands bound on its surface and the smaller  $\text{Ag}_2\text{Se}$  QDs would be formed with the broadened XRD peaks (Xing et al. 2008). The transmission electron microscopy (TEM) image (Fig. 2a) confirms that the  $\text{Ag}_2\text{Se}$  quantum dots have an average size of  $\sim 5$  nm. Figure 2 b shows the lattice fringe that represents the (004) plane corresponding to 0.195 nm lattice spacing. These results confirm the formation of  $\text{Ag}_2\text{Se}$  QDs.

The  $\text{Ag}_2\text{Se}$  QDs size was further confirmed from DLS analysis, and its histogram is shown in Fig. 2c.

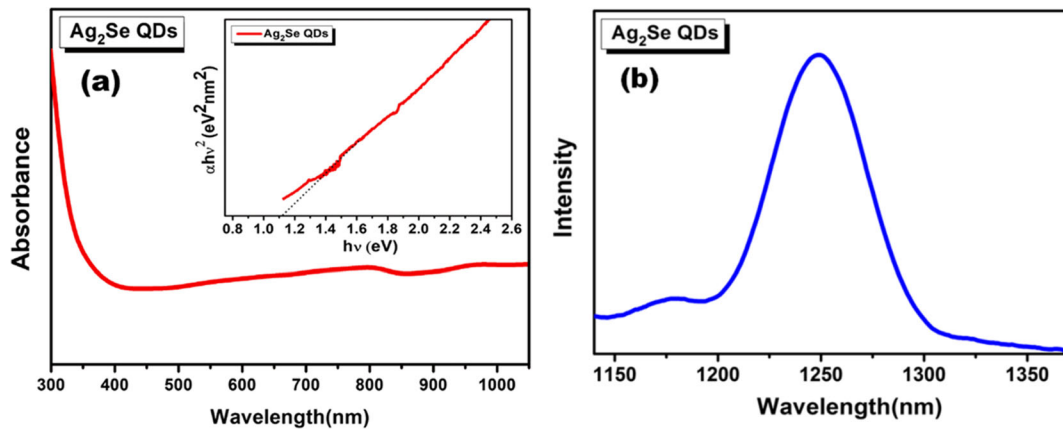
**Fig. 2** a, b TEM images of  $\text{Ag}_2\text{Se}$  QDs. c Histogram of  $\text{Ag}_2\text{Se}$  QDs from DLS analysis. d EDX spectrum of  $\text{Ag}_2\text{Se}$  QDs



The average size of  $\text{Ag}_2\text{Se}$  QDs is 5 nm. The EDX spectrum of  $\text{Ag}_2\text{Se}$  is shown in Fig. 2d. It gives information on the elemental composition of  $\text{Ag}_2\text{Se}$  QDs, and the stoichiometric ratio of silver/selenide is found to be 1.98 which is close to the standard 2:1 ratio of  $\text{Ag}_2\text{Se}$ .

The optical absorption spectrum recorded for  $\text{Ag}_2\text{Se}$  QDs dispersed in hexane is shown in Fig. 3a, and its Tauc plot is calculated to determine the bandgap of obtained absorbance from 400 to 1100 nm. The absorption spectrum exhibits the characteristic excitonic peak at 826 nm which is a low-energy peak of  $\text{Ag}_2\text{Se}$  QDs and attributes blue shift from the bulk  $\text{Ag}_2\text{Se}$ . The excitonic peak shows the existence of fewer surface defects which is prerequisite criteria of an efficient sensitizer for QDSC (Jara et al. 2014). The resolved feature of blue shifting towards lower wavelength represents the quantum confinement effect of  $\text{Ag}_2\text{Se}$  QDs. From the Tauc plot, the optical bandgap of 1.15 eV is calculated by using the energy relation of  $(\alpha h\nu)^2$  vs  $h\nu$ , where  $\alpha$  is the absorption coefficient and  $h\nu$  is photon energy. This extended absorption to NIR region will help to amplify in harvesting more photons from the solar spectrum and generate more electron-hole pairs upon illumination, and hence, such QDs own wide solar absorption capability and can be used as an efficient sensitizer for QDSC.

Photoluminescence study is carried out for  $\text{Ag}_2\text{Se}$  QDs as shown in Fig. 3b. It shows a discernible



**Fig. 3** **a** UV-Vis-NIR absorption spectrum with inset Tauc plot of  $\text{Ag}_2\text{Se}$  QDs. **b** Photoluminescence spectrum of  $\text{Ag}_2\text{Se}$  QDs

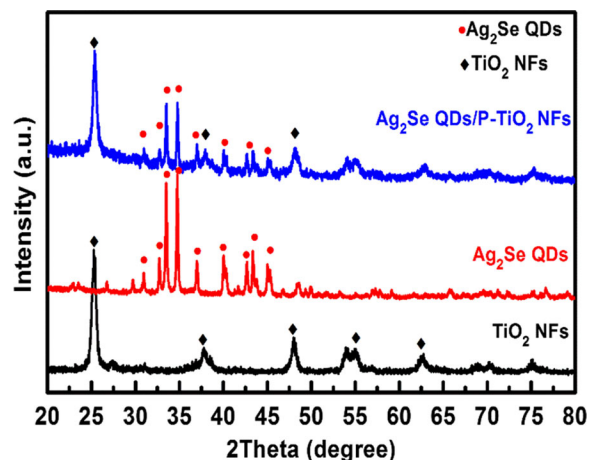
emission peak at 1249 nm in the narrow range from 1200 to 1300 nm. The calculated FWHM for the emission peak was found to be 52 nm. This clearly shows the narrow size distribution of  $\text{Ag}_2\text{Se}$  QDs. In addition, sharp emission intensity attributes to the less density of defect sites (Cao and Che 2014), thereby  $\text{Ag}_2\text{Se}$  QDs can be proved efficient in improving photovoltaic performance as QDs inbuilt with the low defect and high PL emission quality. The appearance of small shoulder peak along with the main emission line may be due to the transition from a selenium vacancy that acts as a donor for  $\text{Ag}_2\text{Se}$ . (U. N. Roy et al. 2005)

#### Characterization of $\text{Ag}_2\text{Se}$ QD-sensitized porous $\text{TiO}_2$ NFs

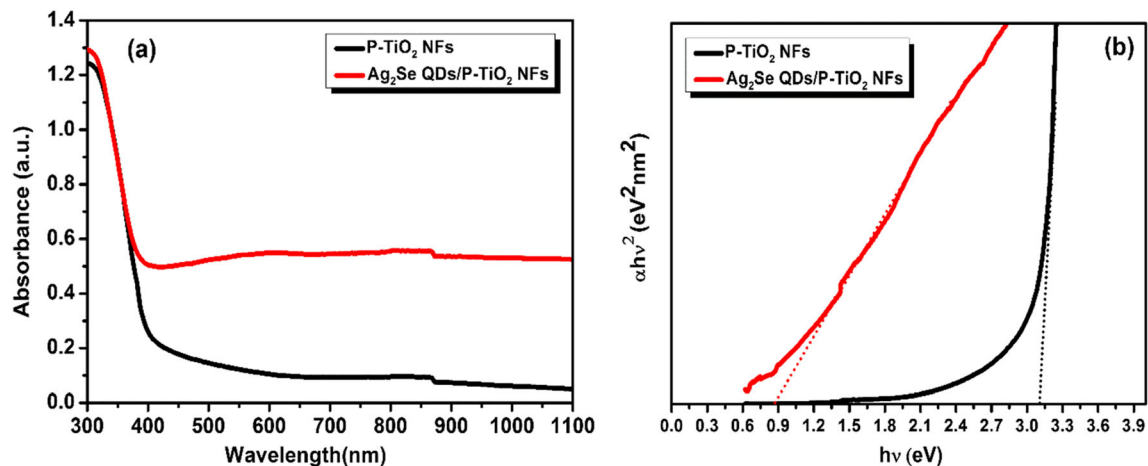
The XRD pattern of  $\text{Ag}_2\text{Se}$  QD-sensitized porous  $\text{TiO}_2$  NFs is studied to investigate the presence of ex situ adsorbed  $\text{Ag}_2\text{Se}$  QDs onto the porous  $\text{TiO}_2$  NFs, and it is previously sensitized using the direct ex situ direct adsorption method (Pernik et al. 2011). Figure 4 shows the XRD patterns of porous  $\text{TiO}_2$  NFs and  $\text{Ag}_2\text{Se}$  QD-sensitized porous  $\text{TiO}_2$  NFs, where diffraction peaks of porous  $\text{TiO}_2$  NFs clearly indexed to the anatase phase structure. The  $\text{TiO}_2$  NFs being porous in nature have shown the enhanced adsorption of  $\text{Ag}_2\text{Se}$  QDs on  $\text{TiO}_2$  photoanode that can be seen from the XRD pattern. Along with the peaks of  $\text{TiO}_2$  NFs, there are eight major peaks of  $\text{Ag}_2\text{Se}$  QDs such as (102), (120), (112), (121), (013), (031), (113), and (032) that were observed which justify the adsorption of  $\text{Ag}_2\text{Se}$  QDs over the  $\text{TiO}_2$  NF-based photoanode. Moreover, broadening in the XRD peaks is

due to the existence of small sizes of  $\text{TiO}_2$  nanoparticles composed of  $\text{TiO}_2$  NFs and QDs as well (Wei et al. 2015).

Figure 5 a illustrates the UV-Vis-NIR absorbance spectra of porous  $\text{TiO}_2$  NFs with and without sensitized  $\text{Ag}_2\text{Se}$  QDs. The plain porous  $\text{TiO}_2$  NFs show the absorbance onset > 400 nm due to possible better light harvesting as a consequence of porous nature of  $\text{TiO}_2$  NFs which tend to increase light scattering effect inside the  $\text{TiO}_2$  NFs (Bijarbooneh et al. 2013). On the other side, the enhancement in the absorption covers a wide range from 400 to 1100 nm compared with porous  $\text{TiO}_2$  NFs confirm the complete adsorption of  $\text{Ag}_2\text{Se}$  QDs on  $\text{TiO}_2$  NF-based photoanode. The corresponding optical bandgap was calculated by Tauc plot as shown in Fig. 5b and found the decreased bandgap



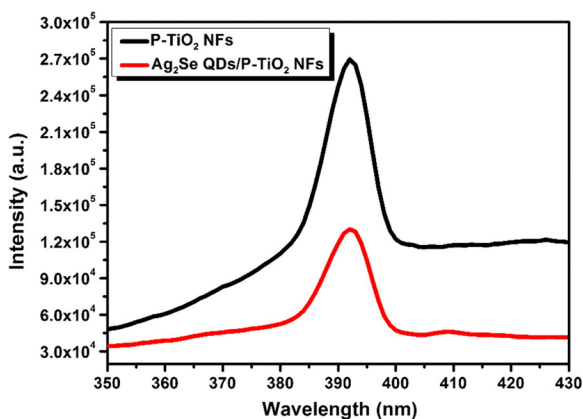
**Fig. 4** XRD pattern of  $\text{Ag}_2\text{Se}$  QD-sensitized porous  $\text{TiO}_2$  NF-based photoanode



**Fig. 5** **a** UV-Vis-NIR absorption spectra and **b** Tauc plots of porous TiO<sub>2</sub> NFs and Ag<sub>2</sub>Se QD-sensitized porous TiO<sub>2</sub> NF-based photoanodes

of 0.91 eV for Ag<sub>2</sub>Se QD-sensitized porous TiO<sub>2</sub> NF-based photoanode. Therefore, it remarkably improved absorption suggesting Ag<sub>2</sub>Se QDs as a promising sensitizer for QDSC.

It can be seen from Fig. 6 that the separation of photogenerated electron holes and their kinetics was investigated by photoluminescence (PL) studies. The photoluminescence emission peak occurs at 392 nm for porous TiO<sub>2</sub> NFs when excited by 325-nm wavelength attributes the indirect band-to-band transition of the TiO<sub>2</sub> (Abazovic et al. 2006). Ag<sub>2</sub>Se QD-sensitized porous TiO<sub>2</sub> NF photoanode shows emission at the same wavelength as that of TiO<sub>2</sub> NFs but with the significant quenching in the intensity of the emission peak. The quenching of



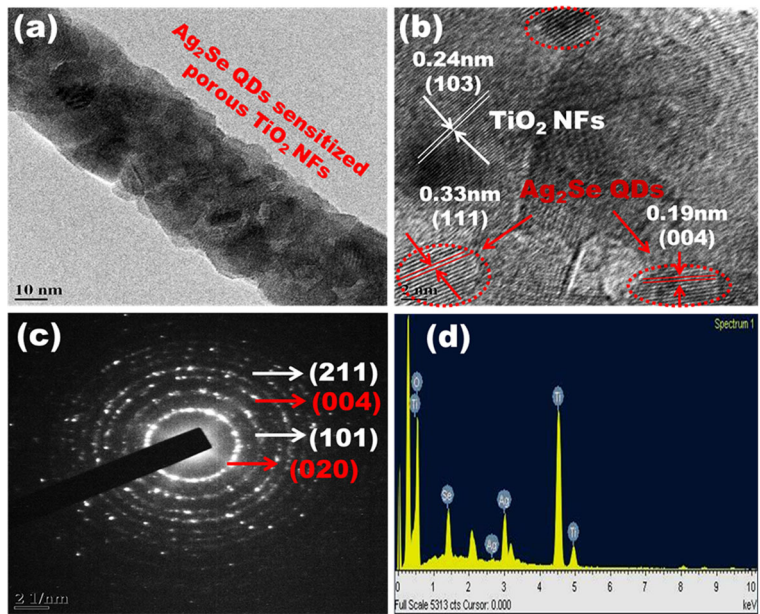
**Fig. 6** Emission spectra of porous TiO<sub>2</sub> NFs and Ag<sub>2</sub>Se QD-sensitized porous TiO<sub>2</sub> NFs

the intensity indicates the reduced electron-hole recombination and efficient electron transfer from the CB of Ag<sub>2</sub>Se QDs to the CB of TiO<sub>2</sub> NFs which forms a hybrid nanostructure.

Thus, the PL spectra proved the charge separation by increasing the driving force of extracted electrons, and it suggests the higher conduction band of Ag<sub>2</sub>Se QDs due to quantum confinement effect (Gao et al. 2009; Cao et al. 2015). Therefore, these extracted electrons would improve the photocurrent efficiency of the QDSC.

Figure 7a and b show the HR-TEM image of Ag<sub>2</sub>Se QD-sensitized porous TiO<sub>2</sub> NFs in which porous TiO<sub>2</sub> NFs can be seen with the diameter ~ 95 nm composed of a network of TiO<sub>2</sub> nanoparticle size of < 10 nm. Here, it is clear that Ag<sub>2</sub>Se QDs (about ~5 nm) are tethered over the porous TiO<sub>2</sub> NFs. The observed lattice spacings of 0.323 nm and 0.210 nm of Ag<sub>2</sub>Se QDs that correspond to the planes (112) and (204), respectively, confirm the adsorption of Ag<sub>2</sub>Se QDs over the TiO<sub>2</sub> NFs. The diffraction rings in the selected electron diffraction are shown in Fig. 7c, and they match well with (211) and (101) planes of TiO<sub>2</sub> and (004) and (020) planes of Ag<sub>2</sub>Se QDs, respectively. The compositional EDX analysis was carried out on Ag<sub>2</sub>Se QD-sensitized TiO<sub>2</sub> NF-based photoanode which is shown in Fig. 7d. It suggests that the Ag<sub>2</sub>Se QDs is composed of Ag and Se elements and is homogeneously distributed on the TiO<sub>2</sub> substrate.

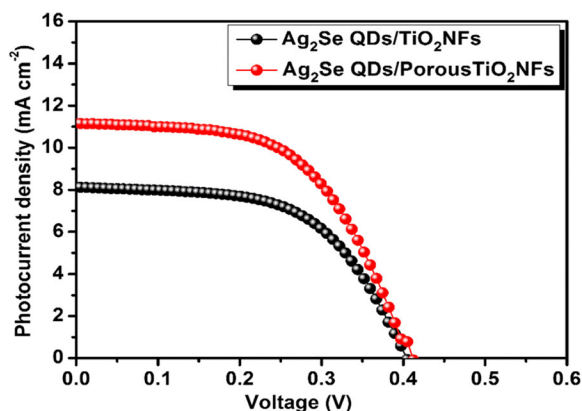
**Fig. 7** **a, b** HR-TEM images of Ag<sub>2</sub>Se QD-sensitized porous TiO<sub>2</sub> NFs. **c** SAED pattern. **d** EDX spectrum of Ag<sub>2</sub>Se QD-sensitized porous TiO<sub>2</sub> NFs



Photovoltaic performance of QDSC

The photovoltaic performance of fabricated QDSC is evaluated from the photocurrent-voltage (J-V) characteristic measurements which is shown in Fig. 8. Its average photovoltaic parameters from the three tested parallel fabricated QDSCs are given in Table 1. It can be seen from the table that the photovoltaic performance of Ag<sub>2</sub>Se QD-sensitized porous TiO<sub>2</sub> NF-based QDSC is increased to 2.50% from 1.79% of conventional TiO<sub>2</sub> NF-based QDSC. This enhancement in the performance is a consequent result of a wide solar absorption from

visible to near-infrared region by Ag<sub>2</sub>Se QDs that are loaded over the porous TiO<sub>2</sub> NFs. The substantial improvement in the  $J_{sc}$  (11.12 mA cm<sup>-2</sup>) is due to the absorption of narrow size of Ag<sub>2</sub>Se QDs that could easily penetrate the porous TiO<sub>2</sub> photoanode and lead to more photogenerated electrons which increase the electron injection to the TiO<sub>2</sub> NFs. It can also be seen that  $V_{oc}$  and FF of Ag<sub>2</sub>Se QDs sensitized porous TiO<sub>2</sub> NFs photoanode-based QDSC got improvement than that of the conventional TiO<sub>2</sub> photoanode-based QDSC. This is due to the large quantity of Ag<sub>2</sub>Se QDs loaded on the porous TiO<sub>2</sub> NFs that helps to elevate the Fermi level of Ag<sub>2</sub>Se QDs towards more negative than the TiO<sub>2</sub> NFs. The porous network of TiO<sub>2</sub> NFs offer proper interaction with the polysulfide electrolyte to penetrate inside the TiO<sub>2</sub> NFs. The counter electrode, Cu<sub>2</sub>S, is necessary to reduce the oxidized Ag<sub>2</sub>Se QDs and prevent the recombination of electrons. Thus, the power conversion efficiency of QDSC obtained using Ag<sub>2</sub>Se



**Fig. 8** J-V characteristic curves of Ag<sub>2</sub>Se QD-sensitized conventional TiO<sub>2</sub> NFs and porous TiO<sub>2</sub> NF-based QDSCs

**Table 1** Photovoltaic parameters of fabricated QDSCs based on Ag<sub>2</sub>Se QDs/TiO<sub>2</sub> NFs and Ag<sub>2</sub>Se QDs/P-TiO<sub>2</sub> NFs as photoanodes

QDSCs	$J_{sc}$ (mA cm <sup>-2</sup> )	$V_{oc}$ (V)	FF	$\eta\%$
Ag <sub>2</sub> Se QDs/TiO <sub>2</sub> NFs	8.10	0.402	0.552	1.79
Ag <sub>2</sub> Se QDs/P-TiO <sub>2</sub> NFs	11.12	0.410	0.550	2.50

**Table 2** Comparison of photovoltaic performance of QDSC based on prepared Ag<sub>2</sub>Se/TiO<sub>2</sub> with reported Ag<sub>2</sub>S/TiO<sub>2</sub>-based QDSC

QDSCs	$J_{sc}$ (mA cm <sup>-2</sup> )	$V_{oc}$ (V)	FF	$\eta\%$	Ref.
Ag <sub>2</sub> S QDs/TiO <sub>2</sub> NRs	4.25	0.29	0.330	0.41	(Li et al. 2014)
Ag <sub>2</sub> S QDs/TiO <sub>2</sub> NRs	2.30	0.248	0.350	0.78	(Pawar et al. 2017)
Ag <sub>2</sub> S QDs/TiO <sub>2</sub> NPs	1.43	0.520	0.470	0.35	(Badawi et al. 2018)
Ag <sub>2</sub> Se QDs/P-TiO <sub>2</sub> NFs	11.12	0.410	0.550	2.50	This work

QD-sensitized porous TiO<sub>2</sub> NFs as the photoanode has significant improvement than that obtained using the conventional TiO<sub>2</sub> NFs and the other reported Ag<sub>2</sub>S/TiO<sub>2</sub>-based QDSCs (Table 2).

### Electrochemical AC impedance studies of QDSC

To further understand the electron transport dynamics and charge recombination inside the fabricated QDSC, the impedance measurements were carried out and obtained fitted parameters which are given in Table 3. Figure 9 shows the Nyquist plots obtained for Ag<sub>2</sub>Se-sensitized conventional TiO<sub>2</sub> NFs and porous TiO<sub>2</sub> NF-based QDSCs. These plots illustrate the recombination mechanism occurred between Cu<sub>2</sub>S/polysulfide electrolyte ( $R_1$ ) and TiO<sub>2</sub> NFs/Ag<sub>2</sub>Se QDs/polysulfide electrolyte ( $R_2$ ) interfaces in QDSCs. In accordance with the fitted parameters values, the  $R_2$  interface resistance of 147.56  $\Omega$  has to exhibit the potential of Ag<sub>2</sub>Se QDs in reducing the electron back recombination with the redox couple of electrolyte as Ag<sub>2</sub>Se QDs adsorbed sufficiently on the TiO<sub>2</sub> NFs that have reduced grain boundaries for the fast electron transfer without interruption. The resistance at  $R_2$  interface was found to be reduced for Ag<sub>2</sub>Se QD-sensitized porous TiO<sub>2</sub> NF-based QDSC that is also in agreement with the improved  $V_{oc}$  of 0.410 V in photovoltaic performance. The  $R_1$  (1.18  $\Omega$ ) of Ag<sub>2</sub>Se QD-sensitized porous TiO<sub>2</sub> NF-based QDSC is comparatively less resistance than that of conventional TiO<sub>2</sub> NF-based QDSC. This is due to the porous nature of TiO<sub>2</sub> NFs that provide sufficient surface

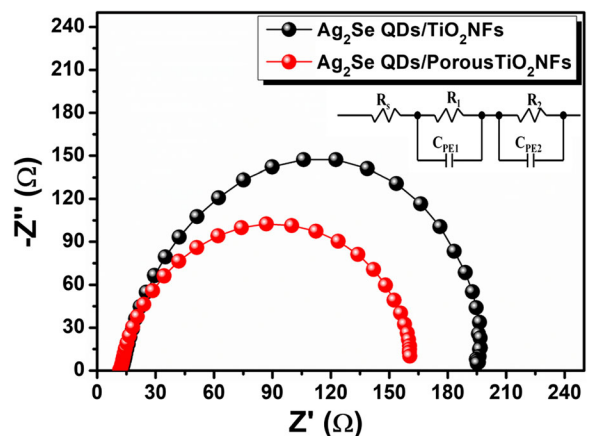
**Table 3** Simulated electrochemical impedance parameters of QDSCs based on Ag<sub>2</sub>Se QDs/TiO<sub>2</sub> NFs and Ag<sub>2</sub>Se QDs/P-TiO<sub>2</sub> NFs as photoanodes

QDSCs	$R_s$ ( $\Omega$ )	$R_1$ ( $\Omega$ )	$R_2$ ( $\Omega$ )
Ag <sub>2</sub> Se QDs/TiO <sub>2</sub> NFs	12.37	1.37	182.3
Ag <sub>2</sub> Se QDs/P-TiO <sub>2</sub> NFs	11.62	1.18	147.56

area to make good contact of polysulfide electrolyte into the TiO<sub>2</sub> NFs to speed up the redox process (Chen et al. 2013).

### Conclusion

In summary, Ag<sub>2</sub>Se QDs were successfully synthesized by a one-pot hot injection method with an average size of ~5 nm. The prepared Ag<sub>2</sub>Se QDs were sensitized on the porous TiO<sub>2</sub> NFs by a direct ex situ method. The optical studies have shown the beneficial scope of Ag<sub>2</sub>Se/P-TiO<sub>2</sub> to use as photoanode material for QDSC. This is due to the high surface area of porous TiO<sub>2</sub> NFs and the maximum loading of Ag<sub>2</sub>Se QDs. The obtained power conversion efficiency of 2.50% for QDSC fabricated with Ag<sub>2</sub>Se QD-sensitized porous TiO<sub>2</sub> NFs brings out that the porous nature of TiO<sub>2</sub> NFs and Ag<sub>2</sub>Se QDs is the cause to improve the photocurrent density of QDSC. The combination of a wide solar spectrum light harvester, Ag<sub>2</sub>Se QDs as the sensitizer with porous TiO<sub>2</sub> NFs as the photoanode, would open more opportunities in QDSC research.

**Fig. 9** Nyquist plots of Ag<sub>2</sub>Se QD-sensitized conventional TiO<sub>2</sub> NFs and porous TiO<sub>2</sub> NF-based QDSCs



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#### Compliance with ethical standards

**Conflict of interest** The authors declare that they have no conflicts of interest.

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