



# Manganese oxides/N-doped carbon particles with high capacity retention for aqueous rechargeable zinc battery

Dong-Shuai Li · Shan Wu · Yi-Fan Wang · Ming Sun ·  
Wei-Liang Liu · Man-Man Ren · Fan-Gong Kong ·  
Shou-Juan Wang · Xin-Qiang Wang

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**Abstract** Manganese dioxide as the electrode for aqueous zinc-ion batteries (AZIBs) is influenced by the material dissolution. Herein,  $\beta$ -MnO<sub>2</sub>/N-doped carbon matrix (NCm) or Mn<sub>5</sub>O<sub>8</sub>/NCm composites were fabricated by effective synthesis process using polyaniline (PANI) as carbon/nitrogen sources. The conductive N-doped carbon layer was tied to  $\beta$ -MnO<sub>2</sub>, which increased the electrical conductivity of the  $\beta$ -MnO<sub>2</sub> nanorod. At current densities of 200 mA g<sup>-1</sup>, the  $\beta$ -MnO<sub>2</sub>/NCm electrode delivered a higher discharge capacity of 331 mAh g<sup>-1</sup> comparing with 185 mAh g<sup>-1</sup> for the pure  $\beta$ -MnO<sub>2</sub> electrode. Besides, the Mn<sub>5</sub>O<sub>8</sub>/NCm electrode could provide a discharge capacity of 266 mAh g<sup>-1</sup>. Therefore, the approach in this study may pave the way on preparing manganese oxides/NCm materials for AZIBs.

**Keywords** Manganese oxide · N-doped · Cathode materials · Capacity retention · Mild aqueous electrolyte · Energy storage

## Introduction

New electrical energy storage systems have drawn a tremendous amount of attention as the excellent method to overcome the shortcomings associated with the storage and use of renewable energy (Lund 2007; Jiang et al. 2012; Zhai et al. 2011; Hu et al. 2016). Due to high energy density and long cycling life, lithium-ion batteries have dominated the worldwide battery market for digital and mobile devices (Qiu et al. 2018a, b; Suo et al. 2015; Dunn et al. 2011; Chang et al. 2010; Zhang et al. 2018;

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D.-S. Li · S. Wu · Y.-F. Wang · M. Sun · W.-L. Liu (✉) ·  
M.-M. Ren  
School of Materials Science and Engineering, Key Laboratory of Amorphous and Polycrystalline Materials, Key Laboratory of Processing and Testing Technology of Glass Functional Ceramics of Shandong Province, Qilu University of Technology (Shandong Academy of Sciences), Daxue Road, Western University Science Park, Jinan 250353, People's Republic of China  
e-mail: wlliu@sdu.edu.cn

F.-G. Kong (✉) · S.-J. Wang  
State Key Laboratory of Biobased Material and Green Papermaking, Key Laboratory of Pulp & Paper Science and Technology of Shandong Province/Ministry of Education, Qilu University of Technology (Shandong Academy of Sciences), Jinan 250353, People's Republic of China  
e-mail: kfg@qlu.edu.cn

X.-Q. Wang  
State Key Laboratory of Crystal Materials, Shandong University, Jinan 250100, People's Republic of China

Zhang et al. 2018). However, the high cost and potential safety problems can limit their large-scale applications (Tarascon and Armand 2001; Wang et al. 2013a, b; Bruce et al. 2012; Hu et al. 2017; Wu et al. 2018a, b). Nonetheless, a long lifetime, as well as high energy density, remains a primary dispute (Cao et al. 2017).

During the past decade, a string of aqueous zinc-ion batteries (AZIBs) cathode that can achieve reversibly aqueous zinc-ion storage, for instance, tunnel-type  $\text{MnO}_2$  (Alfaruqi et al. 2015a, b; Zhao et al. 2018; He et al. 2017; Boeun Lee et al. 2016), Prussian blue analogues (Trocoli and La Mantia 2015; Zhang et al. 2015), and vanadium oxides (Xia et al. 2018a, b; Yan et al. 2018; Senguttuvan et al. 2016; Sambandam et al. 2018; Hu et al. 2018; Wei et al. 2018) have been reported. Xu et al. assembled a  $\text{MnO}_2/\text{ZnSO}_4$  or  $\text{Zn}(\text{NO}_3)_2/\text{Zn}$  system, which contributed a high discharge capacity of  $210 \text{ mAh g}^{-1}$  (Xu et al. 2012). Additionally, graphene scroll-coated  $\alpha\text{-MnO}_2$  nanowires have been studied as cathodes materials, which demonstrated high capacity, outstanding rate performance and distinguished stable storage (Wu et al. 2018a, b). Besides, the graphene oxide scrolls reduced the dissolution of  $\alpha\text{-MnO}_2$  and markedly improved the conductivity for AZIBs. Furthermore, the interplay between hydroxylated interphase on the surface and the unique bivalence structure of  $\text{Mn}_5\text{O}_8$  suppresses the gas evolution reactions and provides the facile pathway for ion transport via intra-/inter-layer defects of  $\text{Mn}_5\text{O}_8$  (Shan et al. 2016). Recently,  $\text{MnO}_x@\text{N-C}$  was used as the cathode and indicated that onion-like N-doped carbon and amorphous carbon shell contributed to a high reversibly capacity and cyclic stability for rechargeable aqueous zinc-ion batteries (Fu et al. 2018).

A series of approaches have been studied to enhance the electrochemical performances of  $\text{MnO}_2/\text{zinc salts}/\text{Zn}$  battery system. For instance, the electrical conductivity of the  $\text{MnO}_2$  electrode was improved by doping carbons (Hu et al. 2017; Zeng et al. 2017; Huang et al. 2018). Owing to its higher electronegativity (3.04) and smaller atomic diameter, N-doped carbon matrix is promising material and made a lot of gains (Fu et al. 2018). Herein, a special  $\beta\text{-MnO}_2/\text{N-doped carbon matrix (NCm)}$  composites and  $\text{Mn}_5\text{O}_8/\text{NCm}$  composites were effectively prepared throughout one-pot hydrothermal method, and polyaniline (PANI) by heating was used as the sources of carbon and nitrogen. This study provides insights necessary to design other metal oxides/NCm materials for AZIBs and other cost-effective and environmentally friendly battery systems.

## Experimental

### Materials synthesis

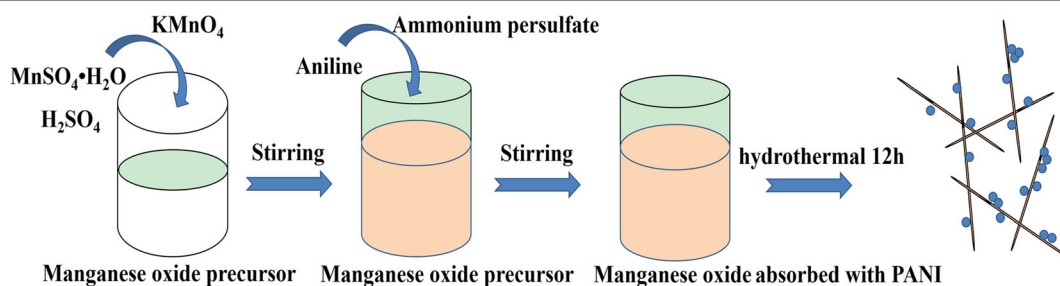
$\text{MnOOH}/\text{PANI}$  precursor was synthesized via a one-step hydrothermal method. First of all, 3 mmol of  $\text{MnSO}_4\cdot\text{H}_2\text{O}$  and  $0.5 \text{ mol L}^{-1}$  of  $\text{H}_2\text{SO}_4$  (2 mL) were transferred into 60 mL of deionized water, and magnetically agitated until a clear solution was obtained. Next, 0.1 mol  $\text{KMnO}_4$  (20 mL) aqueous solution was slowly dropped into the above mixture to form a dark violet solution. Then, the mixed solution was agitated for 2 h at room temperature. And then, 50  $\mu\text{L}$  of aniline was slowly dropped into the solution and continued to stir for 2 h. At last, 0.125 g of  $(\text{NH}_4)_2\text{S}_2\text{O}_8$  was added into the mixture. After agitating for another 30 min, the mixture was poured to 110 mL of Teflon-lined stainless steel autoclave and heated at  $120 \text{ }^\circ\text{C}$  for 12 h. The precursor was obtained by centrifugation after cooling, washed with deionized water and ethanol three times, and dried by vacuum drying oven. The  $\beta\text{-MnO}_2/\text{NCm}$  was prepared through heating at  $300 \text{ }^\circ\text{C}$  for 2 h in air atmosphere. The  $\text{Mn}_5\text{O}_8/\text{NCm}$  was prepared through heating at  $500 \text{ }^\circ\text{C}$  for 4 h in air atmosphere. The synthesis route of  $\text{MnOOH}/\text{PANI}$  precursor is shown in Scheme 1. Likewise, the  $\beta\text{-MnO}_2$  powders were also synthesized with a similar method without the polymerization with PANI.

### Materials characterization

The morphologies were characterized by scanning electron microscope (SEM, ZEISS SUPRATM 55). The prepared samples were characterized by X-ray diffraction (XRD, SHMADZUXRD-6100AS) with  $\text{Cu K}\alpha$  radiation ( $\lambda = 1.5418 \text{ \AA}$ ). Transmission electron microscope (TEM) and energy-dispersive X-ray spectroscopy (EDS) were obtained using Tecnai G2F30 S-Twin operated at 300 kV. X-ray photoelectron spectroscopy (XPS) measurements were performed on an ESCALAB 250 spectrometer with  $\text{Mg K}\alpha$  X-ray source.

### Electrochemical measurements

The 2032 coin-type cells were assembled in open air atmosphere to evaluate the electrochemical performance with a Neware BTS-4008 battery test system. The electrochemical properties were investigated using products as cathode, zinc foil as anode, filter paper as the



**Scheme 1** Schematic illustration of the fabrication process of MnOOH/PANI precursor

separator, and  $1 \text{ mol L}^{-1}$   $\text{ZnSO}_4$  aqueous solution as electrolyte. To prepare the cathode, the homemade products, acetylene black, and polyvinylidene fluoride were mixed in a weight ratio of 8/1/1, and the mixture was dispersed in N-methyl-2-pyrrolidone to form a stable homogeneous slurry. Then, the mixture was cast on stainless steel foil and dried overnight under vacuum conditions at  $110 \text{ }^\circ\text{C}$ . An active material loading of around  $1.0 \text{ mg cm}^{-2}$  was used, and the diameter of zinc foil was 12 mm. The simulated cells were cycled at room temperature within the voltage range of 1.00–1.85 V (vs.  $\text{Zn/Zn}^{2+}$ ). Cyclic voltammograms (CV) and electrochemical impedance spectroscopy (EIS) were performed on a CHI660E electrochemical workstation at room temperature. The EIS were tested at the frequency range 0.01 Hz–100 kHz with alternating-current voltage of 5 mV.

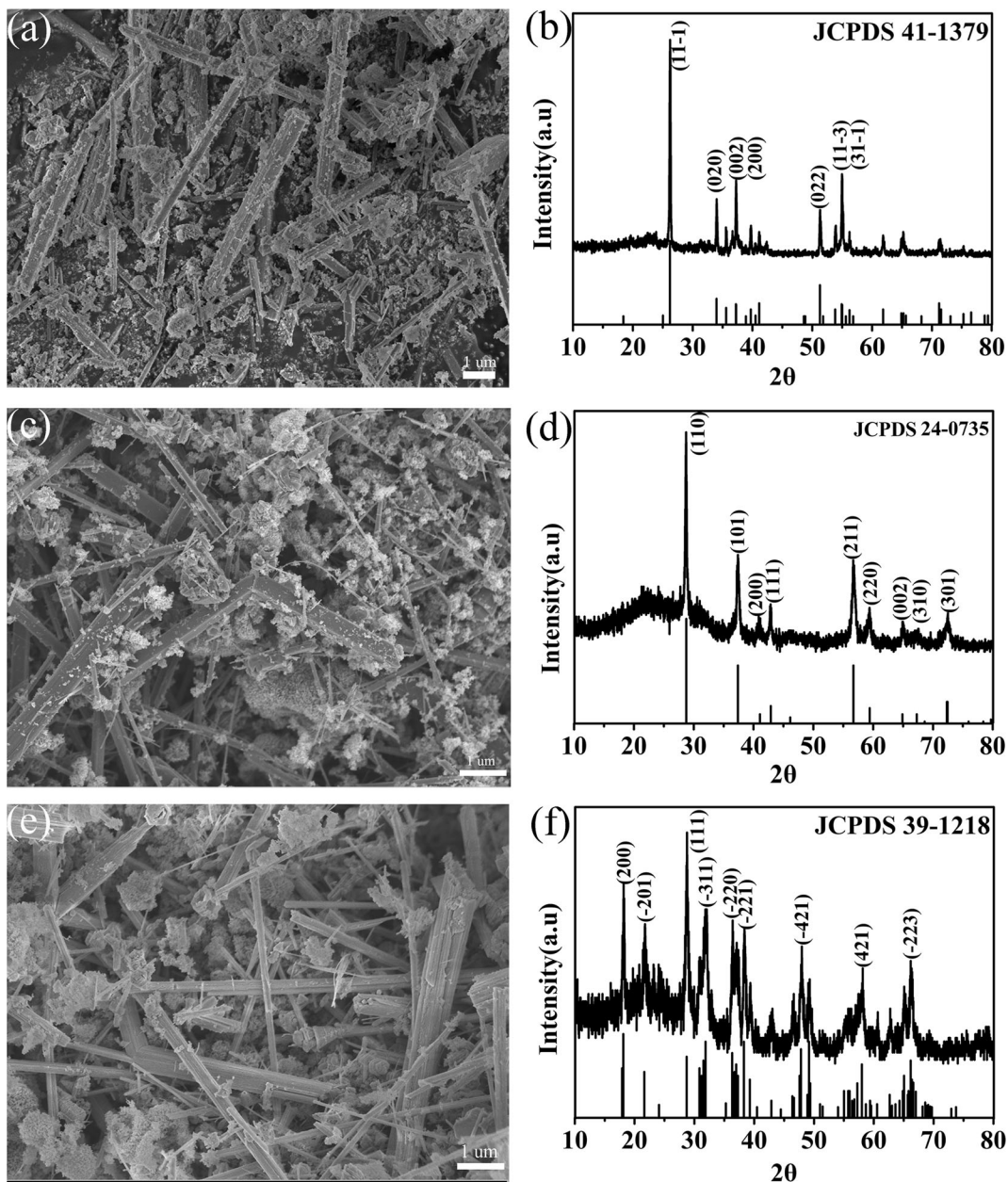
## Results and discussion

The SEM and XRD were carried out to identify the microstructure of the MnOOH/PANI intermediate,  $\beta\text{-MnO}_2/\text{NCm}$  composites, and  $\text{Mn}_5\text{O}_8/\text{NCm}$  composites. SEM image of the  $\text{MnO}_2$  (Supplementary Fig. S1) demonstrates a homogeneous one-dimensional nanorod structure. After reacting with PANI, the surface of the nanorod is coated with numerous PANI nanoparticle, and the size of MnOOH/PANI intermediate is 80–200 nm (Fig. 1a). The XRD pattern of the precursor in Fig. 1b exhibits that all peaks can be assigned to MnOOH (JCPDS 41-1379). Owing to PANI was amorphous, and PANI was not detected in the XRD pattern of MnOOH/PANI. The morphology of the precursor does not change after heat treatment at  $300 \text{ }^\circ\text{C}$  (Fig. 1c) or  $500 \text{ }^\circ\text{C}$  (Fig. 1e) for 4 h in air during the process of the formation of  $\text{MnO}_2$  and  $\text{Mn}_5\text{O}_8$ . The XRD results of the products suggest that all peaks can be assigned to  $\beta\text{-MnO}_2$

(JCPDS 24-0735, Fig. 1d) and  $\text{Mn}_5\text{O}_8$  (JCPDS 39-1218, Fig. 1f) during calcination, respectively.

The TEM provides more details about the  $\beta\text{-MnO}_2/\text{NCm}$  composites. Figure 2a demonstrates that N-doped carbon nanoparticles are tied to  $\beta\text{-MnO}_2$  nanorod, which is corresponded with Fig. 1c. The HAADF-STEM and EDX analysis were used to further confirm the element distribution of the  $\beta\text{-MnO}_2/\text{NCm}$  composites (Fig. 2b–g). The EDX mapping of the  $\beta\text{-MnO}_2/\text{NCm}$  composites shows the homogeneous distribution of C (Fig. 2d), Mn (Fig. 2e), O (Fig. 2f), and N (Fig. 2g), respectively. The EDX results imply that C and N element are the homogeneous distribution of  $\beta\text{-MnO}_2/\text{NCm}$ . The survey on XPS spectra of  $\beta\text{-MnO}_2/\text{NCm}$  composites indicates the existence of C, Mn, O, and N elements in Fig. 2h, which is in accordance with the EDX mapping result (Fig. 2b–g). The typical high-resolution spectrum of C 1s in Fig. 2i includes four peaks, which locate at 284.4, 285.1, 286.4, and 288.2 eV in consistent with C–C, C–N, C–O, and O=C–O, respectively (Li et al. 2017; Ren et al. 2017). From the spectrum of Mn 2p (Fig. 2j) reveals two peaks (642.2 and 653.9 eV), which are in accordance with the characteristic Mn  $2p_{3/2}$  and Mn  $2p_{1/2}$  binding energies of  $\text{MnO}_2$ , with the spin energy separation of 11.7 eV (Liu et al. 2010; Thirupathi and Smirniotis 2012). The spectrum of O 1s is presented in Fig. 2k, in which two peaks at 529.8 and 531 eV are ascribed to Mn–O–Mn and Mn–O–H, respectively (Wang et al. 2013a, b). The XPS survey spectrum of N 1s is shown in Supplementary Fig. S2.

The TEM image of  $\text{Mn}_5\text{O}_8/\text{NCm}$  composites is shown in Fig. 3a. The element distribution of  $\text{Mn}_5\text{O}_8/\text{NCm}$  was observed with HAADF-STEM and EDX analysis (Fig. 3b–g). EDX results show the homogeneous distribution of C, Mn, O, and N element in  $\text{Mn}_5\text{O}_8/\text{NCm}$  (Fig. 3d–g). The  $\text{Mn}_5\text{O}_8/\text{NCm}$  has been analyzed by XPS as shown in Fig. 3h, and the XPS result is in accordance with the EDX mapping result

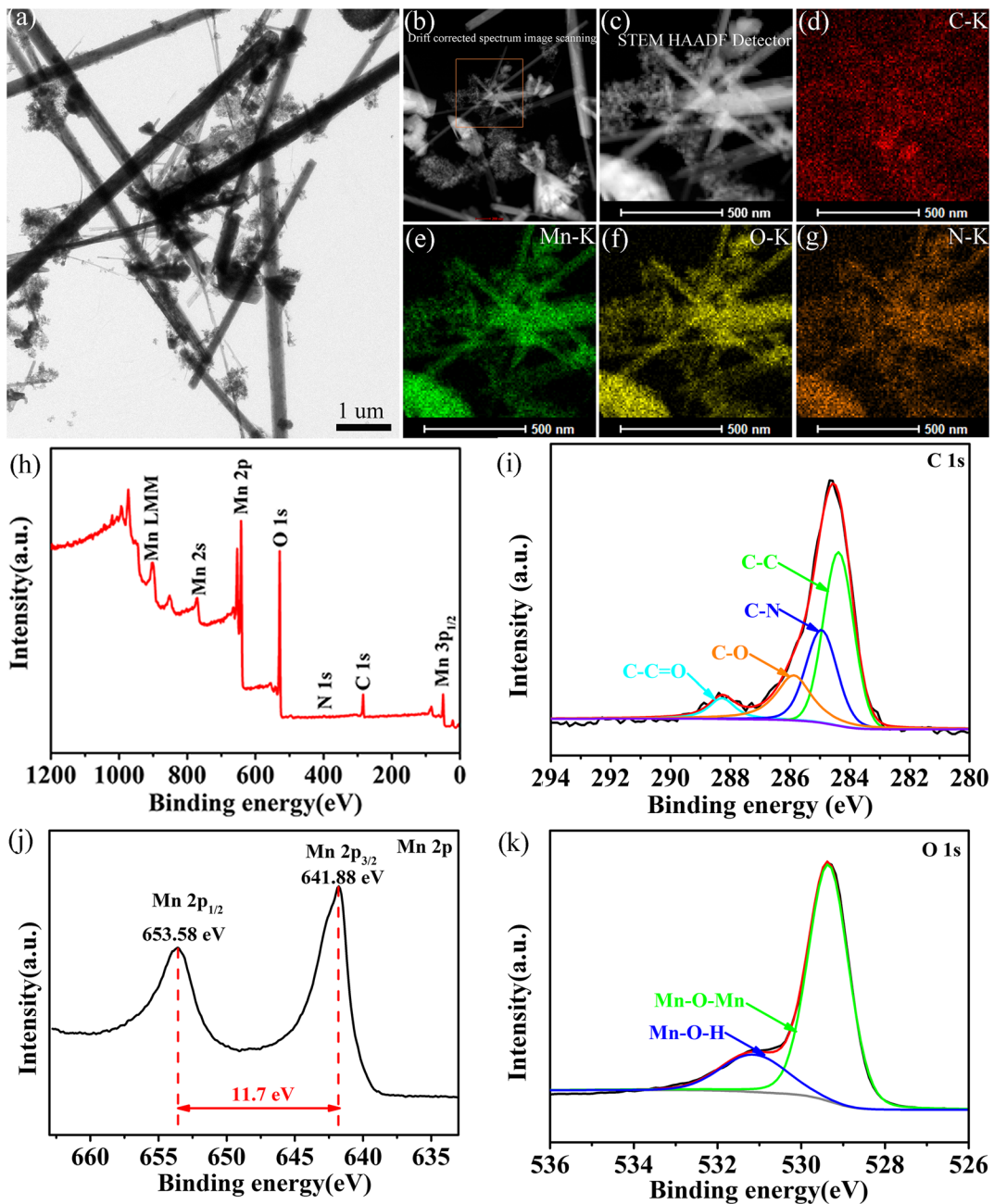


**Fig. 1** SEM images (a) and XRD pattern (b) of the MnOOH/PANI precursor. SEM images (c) and XRD (d) pattern of the  $\beta$ -MnO<sub>2</sub>/NCm composites. SEM images (e) and XRD (f) pattern of the Mn<sub>5</sub>O<sub>8</sub>/NCm composites

(Fig. 3b–g). The C 1s spectrum of Mn<sub>5</sub>O<sub>8</sub>/NCm composites is shown in Fig. 3i and four peaks at 285.9 eV, 285.4 eV, 285.7 eV, and 284.4 eV (Ren et al. 2017; Qiu et al. 2018a, b). As recorded in Fig. 3j, the spectrum of Mn 2p shows two peaks at 642.1 and 653.9 eV which ascribe to Mn 2p<sub>3/2</sub> and Mn 2p<sub>1/2</sub>, respectively (Liu et al. 2015). The O 1s photoelectron spectrum (Fig. 3k) of the Mn<sub>5</sub>O<sub>8</sub>/NCm nanorod shows two peaks at about 530.0

and 531.8 eV (Gao et al. 2010). The high-resolution spectrum of N1s is shown in Supplementary Fig. S3.

To research the electrochemical performance of the  $\beta$ -MnO<sub>2</sub>/NCm and Mn<sub>5</sub>O<sub>8</sub>/NCm, CV and galvanostatic discharge/charge measurement were executed. Figure 4a shows the CV profiles of  $\beta$ -MnO<sub>2</sub>/NCm composites cathodes. The CV curves of  $\beta$ -MnO<sub>2</sub>/NCm reduction/oxidation peaks located at 1.22/1.38

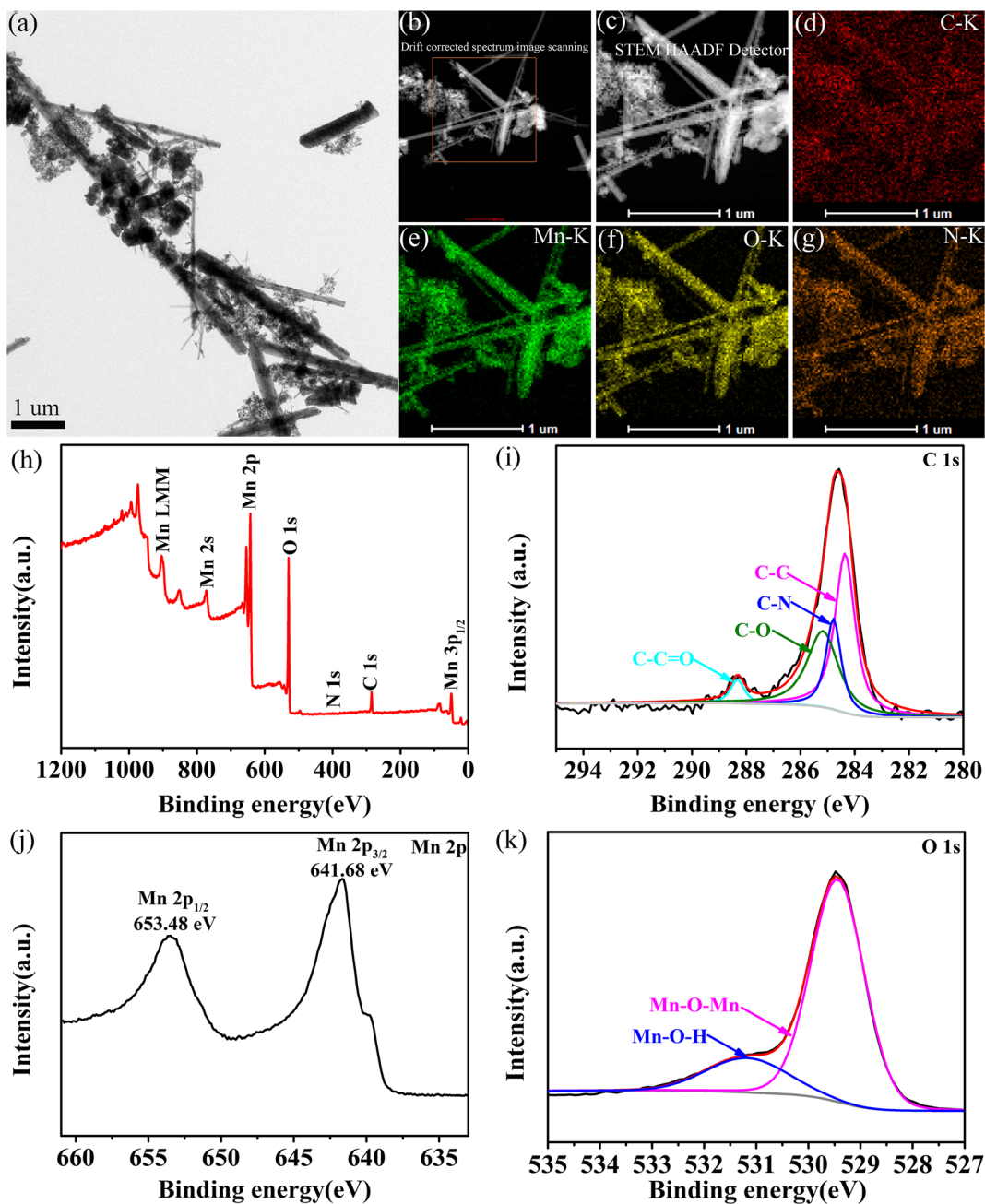


**Fig. 2** TEM image of  $\beta$ -MnO<sub>2</sub>/NCm composites (a). Drift corrected spectrum image scanning (b) and HAADF-STEM image of  $\beta$ -MnO<sub>2</sub>/NCm composites (c). EDX elemental mapping images

of C (d) Mn (e), O (f), and N (g). XPS wide spectrum of  $\beta$ -MnO<sub>2</sub>/NCm composites (h), C 1s (i), Mn 2p (j), and O 1s (k) in the composites

and 1.56 V, and the  $\beta$ -MnO<sub>2</sub> peaks located at 1.23/1.39 and 1.62 V (Supplementary Fig. S4), indicating a similar redox behavior. The peaks at 1.22 and 1.38 V for  $\beta$ -MnO<sub>2</sub>/NCm composites can be ascribed to the insertion of H<sup>+</sup> and Zn<sup>2+</sup> into the  $\beta$ -MnO<sub>2</sub>/NCm composites, corresponded with the reduction of Mn<sup>4+</sup> to Mn<sup>3+</sup>

(Huang et al. 2018). In contrast, the peak at 1.56 V for  $\beta$ -MnO<sub>2</sub>/NCm composites corresponds to the extraction of Zn<sup>2+</sup>, which involves the reversible oxidation of Mn<sup>3+</sup> to Mn<sup>4+</sup> (Alfaruqi et al. 2015a, b; Islam et al. 2017a, b). The electrochemical properties of the Mn<sub>5</sub>O<sub>8</sub>/NCm composites cathodes are assessed using

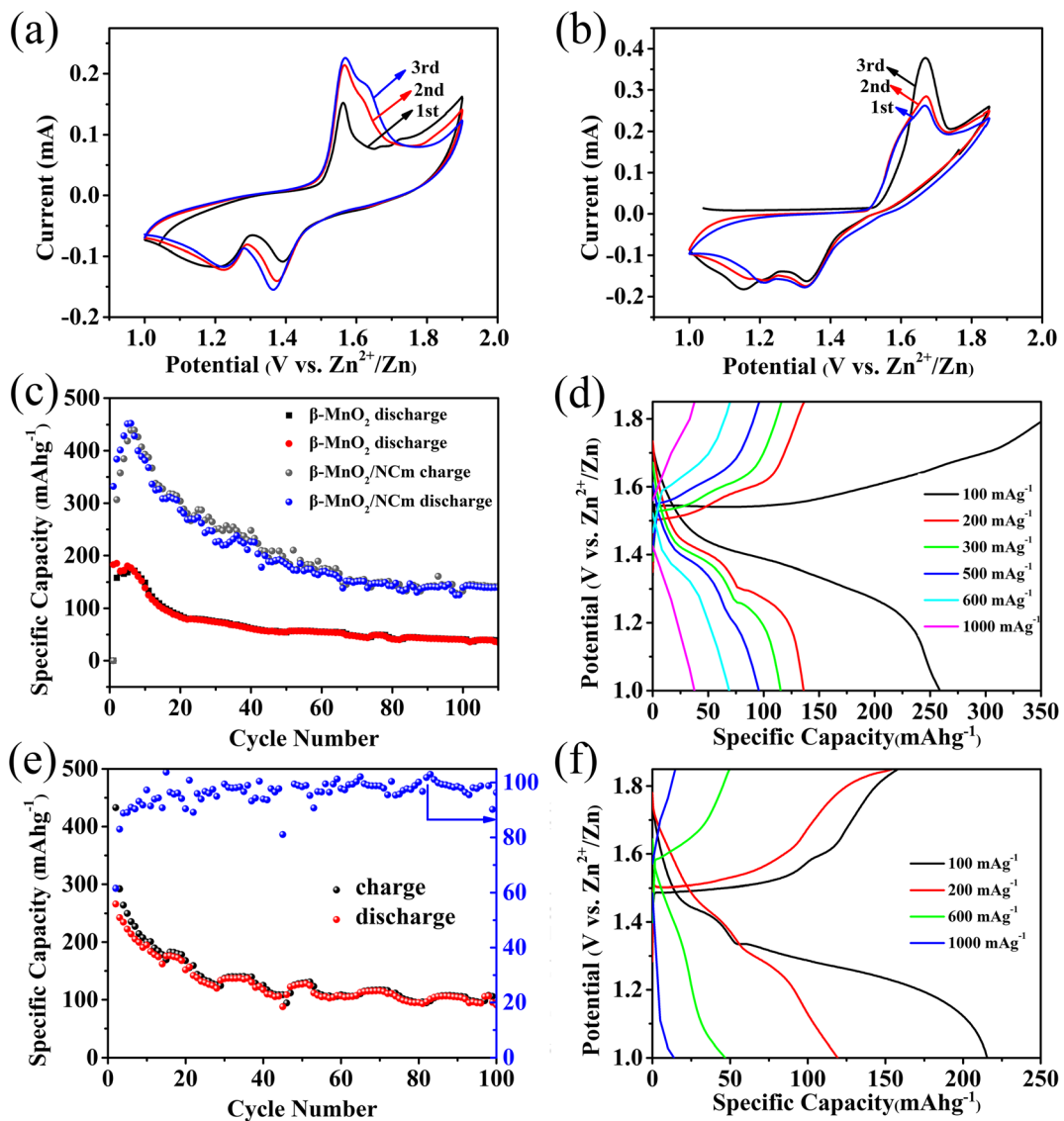


**Fig. 3** TEM image of  $\text{Mn}_5\text{O}_8/\text{NCm}$  composites (a). Drift corrected spectrum image scanning (b) and HAADF-STEM image of  $\text{Mn}_5\text{O}_8/\text{NCm}$  composites (c). EDX elemental mapping images

of C (d) Mn (e), O (f), and N (g). XPS wide spectrum of  $\text{Mn}_5\text{O}_8/\text{NCm}$  composites (h), C 1s (i), Mn 2p (j), and O 1s (k) in the composites

CV as shown in Fig. 4b. The CV curves of  $\text{Mn}_5\text{O}_8/\text{NCm}$  ( $\text{Mn}^{2+}_2\text{Mn}^{4+}_3\text{O}_8$ ) show that sharp anodic peak centered at about 1.67 V and two cathodic peaks at 1.20 and 1.33 V are related to extraction/insertion of  $\text{Zn}^{2+}$  and  $\text{H}^+$  (Hao et al. 2018). Figure 4c shows that the discharge/charge profiles of the  $\beta\text{-MnO}_2/\text{NCm}$  composite

electrode compare with that obtained for the  $\beta\text{-MnO}_2$  electrode at  $200 \text{ mA g}^{-1}$ . The  $\beta\text{-MnO}_2/\text{NCm}$  electrode delivered a higher discharge capacity of  $331 \text{ mAh g}^{-1}$  comparing with  $185 \text{ mAh g}^{-1}$  for the pure  $\beta\text{-MnO}_2$  electrode in a 2 M  $\text{ZnSO}_4$  aqueous electrolyte (Alfaruqi et al. 2017). After 100th cycles, the capacity

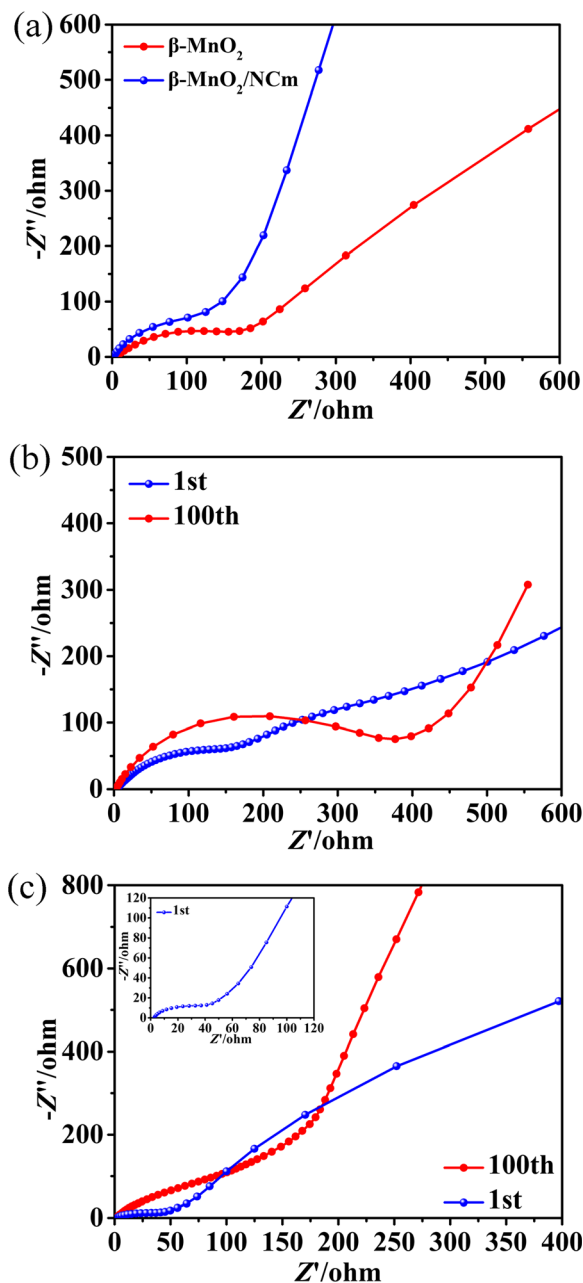


**Fig. 4** CV curves of the  $\beta$ - $\text{MnO}_2/\text{NCm}$  (a). Cycling performance of the  $\beta$ - $\text{MnO}_2/\text{NCm}$  at  $200 \text{ mA g}^{-1}$  (c). Rate performance of  $\beta$ - $\text{MnO}_2/\text{NCm}$  at different current densities (d). CV curves of the

$\text{Mn}_5\text{O}_8/\text{NCm}$  (b). Cycling performance of the  $\text{Mn}_5\text{O}_8/\text{NCm}$  at  $200 \text{ mA g}^{-1}$  (e). Rate performance of  $\text{Mn}_5\text{O}_8/\text{NCm}$  at different current densities (f)

of  $143.7 \text{ mAh g}^{-1}$  is retained for the  $\beta$ - $\text{MnO}_2/\text{NCm}$  electrode than that of the bare  $\beta$ - $\text{MnO}_2$  electrode ( $40.7 \text{ mAh g}^{-1}$ ) in a  $2 \text{ M ZnSO}_4$  aqueous electrolyte. The N-doped carbon-coated samples can accommodate more  $\text{Zn}^{2+}$  ions than the uncoated  $\text{MnO}_2$  (Islam et al. 2017a, b). The cycling performance of bare  $\beta$ - $\text{MnO}_2$  electrode, shown in Supplementary Fig. S5, indicates that the capacity of the bare  $\beta$ - $\text{MnO}_2$  electrode ( $120.5 \text{ mAh g}^{-1}$ ) with  $0.1 \text{ M MnSO}_4$  additive in a  $2 \text{ M ZnSO}_4$  aqueous electrolyte at  $200 \text{ mA g}^{-1}$  is slightly lower than that of reported in literature ( $\sim 135 \text{ mAh g}^{-1}$ ,

Islam et al. 2017a, b). The N-doped carbon and the additional  $\text{MnSO}_4$  in electrolyte can improve of the capacity in the  $\beta$ - $\text{MnO}_2$  electrode. After the initial cycle, the increase in capacity is owing to the activation of electrode (Islam et al. 2017a). The  $\beta$ - $\text{MnO}_2/\text{NCm}$  and  $\beta$ - $\text{MnO}_2$  electrode show a gradual fading in capacity which results from manganese dissolution in electrolyte (Alfaruqi et al. 2015a, b). The electrochemical property is further corroborated in the rate performances (Fig. 4d). The cell shows an excellent rate capability, achieving high capacities of 258, 136, 115, 95, 68, and



**Fig. 5** Nyquist plots of  $\beta\text{-MnO}_2/\text{NCm}$  hybrid composites and pure  $\beta\text{-MnO}_2$  electrodes (a). Nyquist plots of the  $\beta\text{-MnO}_2/\text{NCm}$  composites before the first cycle and after 100th cycles (b). Nyquist plots of the  $\text{Mn}_5\text{O}_8/\text{NCm}$  composites before the first cycle and after 100th cycles (c)

37 mAh  $\text{g}^{-1}$  at 100, 200, 300, 500, 600, and 1000 mA  $\text{g}^{-1}$ , respectively. The rate performances of  $\beta\text{-MnO}_2/\text{NCm}$  electrode are lower than the additional  $\text{MnSO}_4$  in electrolyte. The results above indicate that manganese oxides/N-doped carbon matrix composites

are promising for a high-performance and cost-effective battery system.

Figure 5 illustrates the EIS results of  $\beta\text{-MnO}_2/\text{NCm}$  composites and  $\text{Mn}_5\text{O}_8/\text{NCm}$  composites. The spectrum contains a semicircle and straight sloping line which correspond to the Faradaic reaction and the charge transfer resistance impedance ( $R_{ct}$ ), respectively (Alfaruqi et al. 2017). The diameter of the semicircle for the  $\beta\text{-MnO}_2/\text{NCm}$  cathode is smaller than that of the  $\beta\text{-MnO}_2$  cathode, suggesting lower charge-transfer impedance in  $\beta\text{-MnO}_2/\text{NCm}$  composites (Fig. 5a). The lower charge-transfer impedance of  $\beta\text{-MnO}_2/\text{NCm}$  is ascribed to the co-effect of N-doping and carbon material which increase the conductivity significantly (Ren et al. 2017). EIS results demonstrate the steeper slope of  $\beta\text{-MnO}_2/\text{NCm}$ , suggesting rapid  $\text{Zn}^{2+}$  diffusion in  $\beta\text{-MnO}_2/\text{NCm}$  electrode. The calculated  $R_{ct}$  value of the  $\beta\text{-MnO}_2/\text{NCm}$  electrodes was 142.7  $\Omega$  and increased to 377  $\Omega$  after 100 cycles (Fig. 5b). Figure 5c illustrates that the calculated  $R_{ct}$  value (40  $\Omega$ ) of the  $\text{Mn}_5\text{O}_8/\text{NCm}$  was lower than that of the  $\beta\text{-MnO}_2/\text{NCm}$ . After 100 cycles, the calculated  $R_{ct}$  value of the  $\text{Mn}_5\text{O}_8/\text{NCm}$  electrodes was 167  $\Omega$ . During the consistent discharge/charge processes, the pathways for  $\text{Zn}^{2+}$  insertion/extraction in manganese oxides/NCm composites deconstructed, which may result in the mild raise of the resistances (Wu et al. 2018a, b).

## Conclusions

In summary, the metal oxides/NCm composites electrodes were synthesized for use as cathodes in AZIB via a one-pot hydrothermal method combining with heat treatment. The  $\beta\text{-MnO}_2/\text{NCm}$  electrode registered a higher capacity than the  $\beta\text{-MnO}_2$  electrode. Ascribing to the N-doped carbon matrix, the  $\beta\text{-MnO}_2/\text{NCm}$  composites exhibited high discharge capacity and stable cycle performance for AZIB. By the similar process of preparation,  $\text{Mn}_5\text{O}_8/\text{NCm}$  composites can deliver 266 mAh  $\text{g}^{-1}$  at current densities of 200 mA  $\text{g}^{-1}$ . Compare to  $\beta\text{-MnO}_2/\text{NCm}$  composites,  $\text{Mn}_5\text{O}_8/\text{NCm}$  composites have superior electrical conductivity and lower reversible capacities. The synthesis process showed in this paper provides an effective and facile way to develop other oxides/NCm composites for AZIBs and other cost-effective and environmentally friendly battery systems.



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#### Compliance with ethical standards

**Conflict of interest** The authors declare that they have no conflict of interest.

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