

Synthesis and microstructure of Co/Ni: MgGa₂O₄ nanoparticles

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Abstract This report discusses the preparation and microstructure of Co/Ni co-doped MgGa₂O₄ nanoparticles. The nanoparticles with the size of 20–55 nm were synthesized by sol-gel method. The phase and crystallinity were confirmed by X-ray powder diffraction (XRD) pattern. The particle size was estimated according to XRD data and transmission electron microscopy. The electronic structure was studied using X-ray photoelectron spectroscopy (XPS). The XPS studies showed that Ga³⁺ ions possess tetrahedral and octahedral sites of spinel structure and the inverse degree (two times of the fraction of tetrahedral Ga³⁺ ions) has increased with the increase of the doping concentration of Co²⁺ and Ni²⁺ ions. For Co/Ni co-doped MgGa₂O₄, two broad absorption bands of 350–500 and 550–700 nm were observed in the absorption spectra. The broad band at 350–500 nm was assigned to the combination of the absorption of octahedral Co²⁺ and Ni²⁺ ions, whereas the absorption band at 550–700 nm is mainly due to tetrahedrally coordinated Co²⁺ ions and octahedrally coordinated Ni²⁺ ions.

Keywords Sol-gel method · X-ray photoelectron spectroscopy · Electronic structure · Optical properties · Nanostructure

Introduction

Magnesium gallate (MgGa₂O₄) is an electroluminescent material and has highly been attracted for applications in vacuum fluorescent and field emission displays (Costa et al. 2009; Moon et al. 2008; Pedro et al. 2015) because of its good chemical stability, low excitation voltage, and high purity of luminescence (Liu et al. 2013; Choi et al. 2010). It belongs to AB₂O₄ spinel-type oxides, where A and B are divalent and trivalent metal cations, respectively. The oxygen ions in AB₂O₄ arrange in close-packed cubic structure, forming tetrahedral and octahedral vacancies which are occupied by A and B cations. MgGa₂O₄ is known to be mixed spinel structure (Jiang et al. 2012; da Silva et al. 2013). The distribution of these cations in tetrahedral and octahedral sites classifies the spinel-type oxide material (AB₂O₄) into three types: (I) normal spinel: A and B ions occupy the tetrahedral and octahedral sites, respectively; (II) inverse spinel: a half of B ions occupy the tetrahedral position and the other half of B and all of A ions occupy the octahedral position; and (III) mixed spinel: A and B ions occupy the two different sites randomly. The distribution of A and B cations can be controlled by different preparation methods, type of impurities, and its concentration that will affect the material color, absorption, and emission properties (Liu et al. 2015; Yu et al. 2002; Kumar et al. 2004; Tsai et al. 2004).

Recently, MgGa₂O₄ doped with transition metal ions has attracted much attention because of strong emission and absorption properties in the visible region (Song et al. 2014; da Silva et al. 2010; Li et al. 2009; Yu and Liu

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2011; Sun et al. 2016; Lv et al. 2009; Costa et al. 2012). The introduction of transition metal ions can tune the optical properties of spinel oxide materials. For example, Co- and Ni-doped MgGa_2O_4 materials show emissions at 685 nm (red) and 590 nm, respectively. At present, much work has been reported about the luminescence properties of MgGa_2O_4 doped with single active ions such as Cr^{3+} , Mn^{2+} , Ni^{2+} , Eu^{3+} , and Pr^{3+} (Pedro et al. 2015; Lv et al. 2009; Costa et al. 2012; Suzuki et al. 2008; Suzuki et al. 2005; Lv et al. 2010). Though many reports on optical properties, electronic structure of MgGa_2O_4 nanoparticles has seldom been studied (Yu et al. 2002; Kumar et al. 2004). To the best of our knowledge, the co-doping of Co and Ni in MgGa_2O_4 nanoparticle has not been reported. In this work, Co/Ni co-doped MgGa_2O_4 nanoparticles have been synthesized by the sol-gel method and the microstructure were probed through different analytical methods such as X-ray powder diffraction, TEM, and X-ray photoelectron spectroscopy (XPS). The influence of co-doping of transition elements (Co^{2+} and Ni^{2+}) on the structure and optical properties were discussed.

Experimental

Polycrystalline Co/Ni co-doped MgGa_2O_4 nanoparticles were prepared by citric acid-assisted sol-gel method. $\text{Co}(\text{CH}_3\text{COO})_2 \cdot 4\text{H}_2\text{O}$ (AR, Shanghai Hengxin Chemical Reagent Factory), $\text{Ni}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ (AR, Aladdin Chemistry Co.), $\text{Mg}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ (AR, Tianjin Kermel Chemical Reagent Center), and $\text{Ga}(\text{NO}_3)_3 \cdot x\text{H}_2\text{O}$ (AR, Aladdin Chemistry Co.) were used as starting materials. Calculated amount of these raw materials were dissolved in deionized water. Co and Ni concentrations (mol%) were varied to produce a range of $\text{Co}_x\text{Ni}_y\text{Mg}_{1-x-y}\text{Ga}_2\text{O}_4$ ($0.05 \leq x \leq 0.3$, $0.01 \leq y \leq 0.3$) stoichiometries. Citric acid was added into the above solution as a chelating agent. The molar ratio of metal ions to citric acid was 1:2; the pH of the solution was 3.0. The final solution was stirred for 1 h using a magnetic stirrer and then heated until formation of highly viscous gels. The gels were dried in an oven at 150 °C for 8 h followed by calcination at 800 °C for 5 h. The obtained nanoparticles were used for all the characterization studies.

Powder X-ray diffraction (XRD) data of the doped MgGa_2O_4 nanoparticles were collected on a Rigaku D/Max-rA diffractometer using a $\text{Cu K}\alpha$ radiation ($\lambda = 0.15418$ nm) and with a graphite monochromator. The samples were step scanned in steps of 0.02° (2θ)

using a count time of 0.1 s/step. Transmission electron micrographs were recorded using a JEM-2100 microscope. The samples were ultrasonicated in ethanol and then, a droplet of the suspension was deposited and dried on the sample support. The X-ray photoelectron spectra (XPS) were measured using a Thermofisher ESCALAB 250 spectrometer with monochromatized $\text{Al K}\alpha$ radiation under ultrahigh vacuum ($< 10^{-7}$ Pa). The binding energies were calibrated using C1s peak (284.6 eV) as a reference. All the peaks were deconvoluted after background subtraction using a mixed Gaussian–Lorentzian function. Diffuse absorbance spectra were recorded on a Shimadzu UV-2550 spectrophotometer with a step of 1 nm using an integration sphere at room temperature. BaSO_4 was used as reference.

Results and discussion

Synthesis of nanocrystals

The XRD patterns of the $\text{Co}_x\text{Ni}_y\text{Mg}_{1-x-y}\text{Ga}_2\text{O}_4$ nanoparticles annealed at 800 °C are presented in Fig. 1. The peaks indexed as (220), (311), (222), (400), (422), (511), and (440) crystal planes in the XRD patterns are assigned to cubic MgGa_2O_4 spinel phase (JCPDS No. 10-113). The intensity of the diffraction peaks increased, and the full width at half-maximum (FWHM) decreased with the increase of Co and Ni concentrations, which indicates the improved crystallinity. It can be seen that a weak diffraction peak due to MgO impurity phase (marked as “*” in

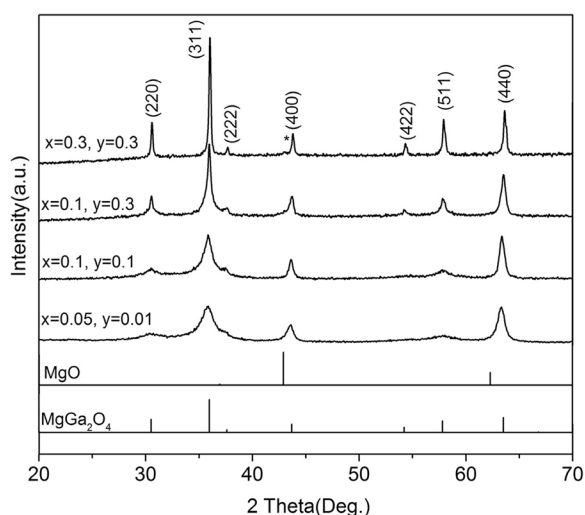


Fig. 1 XRD patterns of the $\text{Co}_x\text{Ni}_y\text{Mg}_{1-x-y}\text{Ga}_2\text{O}_4$ nanoparticles annealed at 800 °C

Table 1 Particle size of $\text{Co}_x\text{Ni}_y\text{Mg}_{1-x-y}\text{Ga}_2\text{O}_4$ nanoparticles obtained from XRD and TEM

Composition (x, y)	Particle size (nm)	
	XRD	TEM
0.05, 0.01	13	15
0.1, 0.1	16	20
0.1, 0.3	22	25
0.3, 0.3	50	55

Fig. 1) appeared when $x = 0.1$ and $y = 0.3$, and the intensity of the diffraction peak of MgO impurity increased with the increase of Co and Ni concentrations.

The average particle size of the samples was calculated by means of the Scherrer formula $D = 0.9\lambda / (\beta \cos\theta)$, where λ is the X-ray wavelength (0.15418 nm), β is the FWHM, and θ is the diffraction angle of the high intense peak. The size of the Co/Ni-doped MgGa_2O_4 nanoparticle is about 13 nm when $x = 0.05$ and $y = 0.01$, and the size increases to 50 nm when $x = 0.3$ and $y = 0.3$ (Table 1). The particle size increases dramatically when Ni content increases from 0.1 to 0.3. It may be that the doped Ni^{2+} ions occupy the position of Ga^{3+} ions, and the ionic radius of Ni^{2+} (69 pm) is larger than that of Ga^{3+} (62 pm), which makes the size of the particle increase.

The TEM images of the $\text{Co}_x\text{Ni}_y\text{Mg}_{1-x-y}\text{Ga}_2\text{O}_4$ nanoparticles are shown in Fig. 2. It can be seen from

Fig. 2 TEM images of $\text{Co}_x\text{Ni}_y\text{Mg}_{1-x-y}\text{Ga}_2\text{O}_4$ nanoparticles annealed at 800 °C: **a** $x = 0.05, y = 0.01$; **b** $x = 0.1, y = 0.1$; **c** $x = 0.1, y = 0.3$; **d** $x = 0.3, y = 0.3$

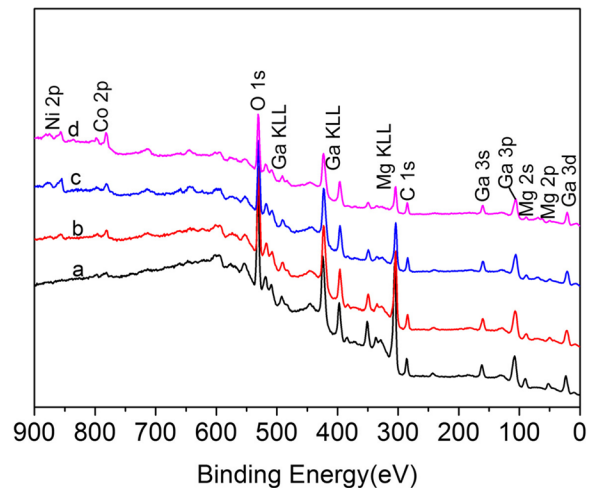
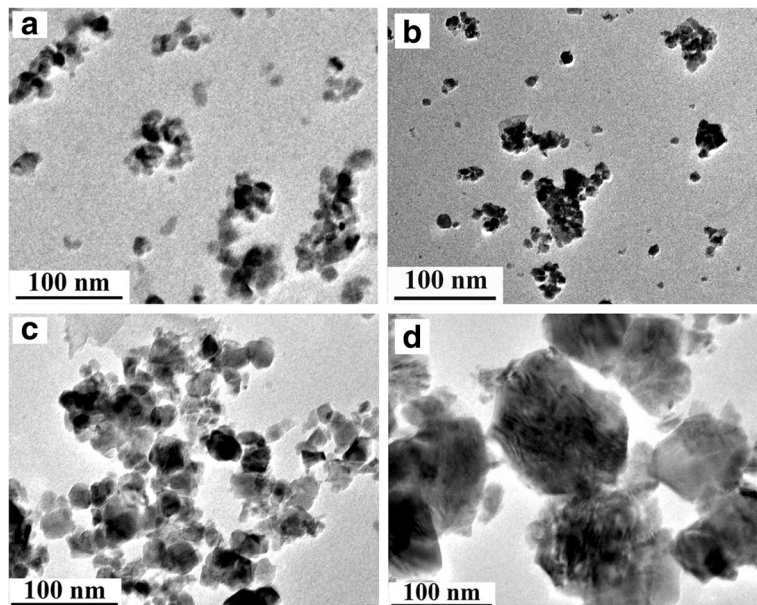


Fig. 3 XPS survey spectra of $\text{Co}_x\text{Ni}_y\text{Mg}_{1-x-y}\text{Ga}_2\text{O}_4$ nanoparticles annealed at 800 °C: (a) $x = 0.05, y = 0.01$; (b) $x = 0.1, y = 0.1$; (c) $x = 0.1, y = 0.3$; (d) 0.3, 0.3

the figure that the particle shape is not regular. The particle size is in the range 15–55 nm and the size increases with the increase of dopant concentration. The TEM results are consistent with those of XRD (Table 1).

Electronic structure

The composition and the electronic structure of as-prepared Co/Ni co-doped MgGa_2O_4 nanoparticles were

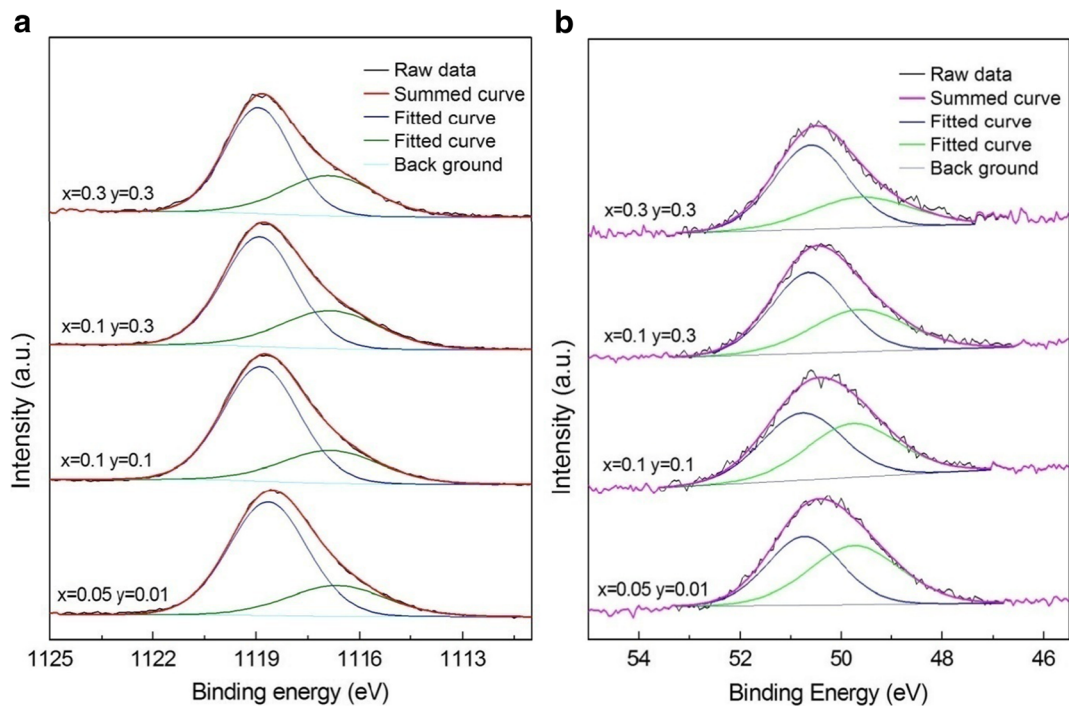


Fig. 4 XPS core level spectra of $\text{Co}_x\text{Ni}_y\text{Mg}_{1-x-y}\text{Ga}_2\text{O}_4$ nanoparticles annealed at 800 °C: **a** Ga 2p_{3/2}; **b** Mg 2p

determined through XPS analysis. Figure 3 shows the XPS survey spectra of the $\text{Co}_x\text{Ni}_y\text{Mg}_{1-x-y}\text{Ga}_2\text{O}_4$ nanoparticles. The XPS core level spectra of Ga 2p, Mg 2p, Co 2p, and Ni 2p were collected as shown in

Figs. 4 and 5. The XPS survey spectra revealed that no other elements were detected except contaminated carbon. The C 1s peak at 284.6 eV were used for calibration.

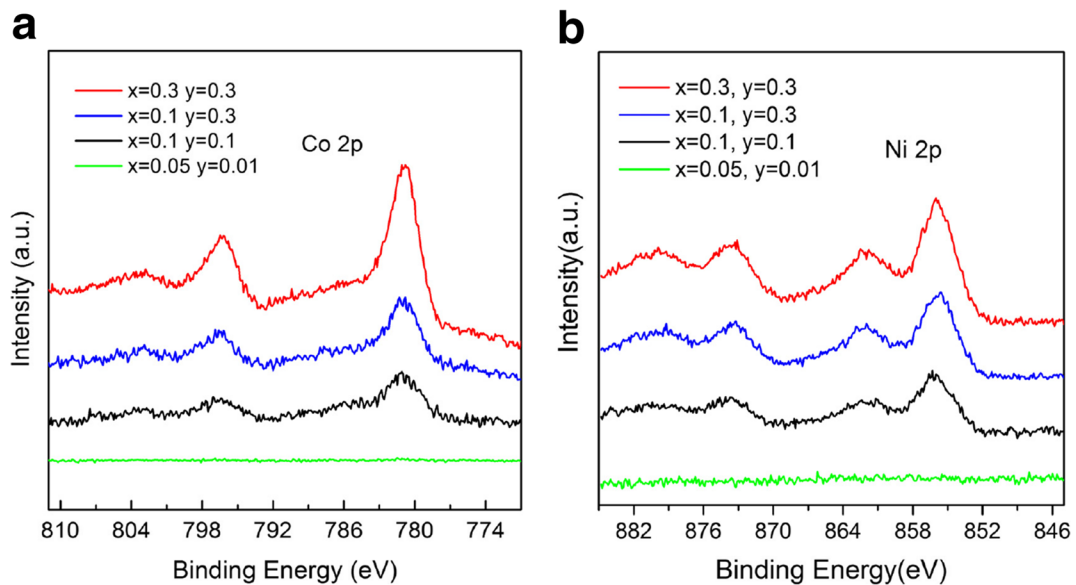


Fig. 5 XPS core level spectra of $\text{Co}_x\text{Ni}_y\text{Mg}_{1-x-y}\text{Ga}_2\text{O}_4$ nanoparticles annealed at 800 °C: **a** Co 2p; **b** Ni 2p

Table 2 XPS data for Ga 2p_{3/2} spectra of Co_xNi_yMg_{1-x-y}Ga₂O₄ nanoparticles annealed at 800 °C

Composition (x, y)	Binding energy (± 0.1 eV)	Area percent (± 0.5%)	FWHM (± 0.1 eV)	Fraction of tetrahedral Ga ³⁺
0.05, 0.01	1118.6	74.7	2.8	0.25
	1116.6	25.3	3.5	
0.1, 0.1	1118.8	73.0	2.7	0.27
	1116.8	27.0	3.5	
0.1, 0.3	1118.9	68.5	2.6	0.31
	1116.8	31.5	3.5	
0.3, 0.3	1118.9	65.4	2.4	0.35
	1116.8	34.6	3.2	

Figure 4a shows the Ga 2p_{3/2} spectra of MgGa₂O₄ nanoparticles with different concentrations of Co and Ni dopants. It can be found that all the spectra are asymmetric, indicating more than one chemical state of Ga³⁺ ions. After deconvolution, Ga 2p_{3/2} spectra are divided into two peaks, indicating the Ga³⁺ ions occupy two different coordination sites. The compositional details and peak parameters of Ga³⁺ are listed in Table 2. The peaks located at 1118.8 eV and 1116.8 eV are respectively assigned to octahedral and tetrahedral Ga³⁺ ions. When x = 0.05 and y = 0.01, the inversion degree (two times of the fraction of tetrahedral Ga³⁺ ion) is 0.5, and the value increases with the increase of the doping concentration. When both x and y are equal to 0.3, the inversion parameter is increased to 0.70.

The Mg 2p XPS spectra (Fig. 4b) are also not symmetric and consist of two peaks. The peak at 50.6 eV can be assigned to Mg²⁺ ion in octahedral site, and the peak at 49.7 eV can be ascribed to Mg²⁺ ion in tetrahedral site. These values are consistent with the literature reports (Ono et al. 2001; Zakaznova-Herzog et al. 2006). The fraction of octahedral Mg²⁺ ion is 0.48 when

Table 3 XPS data for Mg 2p spectra of Co_xNi_yMg_{1-x-y}Ga₂O₄ nanoparticles annealed at 800 °C

Composition (x, y)	Binding energy (± 0.1 eV)	Area percent (± 0.5%)	FWHM (± 0.1 eV)	Fraction of octahedral Mg ²⁺
0.05, 0.01	50.7	48.4	1.8	0.48
	49.7	51.6	2.2	
0.1, 0.1	50.7	52.6	1.9	0.53
	49.7	47.4	2.2	
0.1, 0.3	50.6	60.3	1.8	0.60
	49.6	39.7	2.3	
0.3, 0.3	50.6	66.7	1.8	0.67
	49.6	33.3	2.4	

x = 0.05 and y = 0.01, and this value increases to 0.67 when x = 0.3 and y = 0.3. The compositional details and peak parameters of Mg²⁺ are listed in Table 3.

Figure 5a shows the Co 2p XPS spectra of Co_xNi_yMg_{1-x-y}Ga₂O₄ nanoparticles with different concentrations of Co and Ni dopants. The peak around 781 eV is ascribed to high-spin nature of divalent cobalt, and this value is consistent with Co²⁺ in other materials (Stelmachowski et al. 2014; Sharma et al. 2008). The FWHM is relatively wide probably because of the Co²⁺ ions occupy two different sites in the nanoparticles. The Ni 2p XPS spectra are displayed in Fig. 5b. The peak at 855.4 eV associated with the satellite peak at 861.6 eV is assigned to Ni 2p_{3/2}, which is consistent with the literatures (Zhang et al. 2009; Bennet et al. 2016; Mittal et al. 2006). The compositional details and peak parameters of Co 2p_{3/2} and Ni 2p_{3/2} are listed in Table 4.

Optical properties

To study the relationship between the structure and optical properties of Co_xNi_yMg_{1-x-y}Ga₂O₄ nanoparticles, the absorption spectra were recorded as presented in Fig. 6. The Ni-doped MgGa₂O₄ shows two absorption peaks at 370 and 650 nm; this can be assigned to the transition of ³A_{2g}(³F) → ³T_{1g}(³P) and ³A_{2g}(³F) → ³T_{1g}(³F) of Ni²⁺ in the octahedral sites, respectively (Suzuki

Table 4 Binding energies of Co 2p_{3/2} and Ni 2p_{3/2} core levels (eV) of Co_xNi_yMg_{1-x-y}Ga₂O₄ nanoparticles annealed at 800 °C

Composition (x, y)	Co 2p _{3/2} (± 0.1 eV)	Ni 2p _{3/2} (± 0.1 eV)
0.1, 0.1	781.0	855.4
0.1, 0.3	780.9	855.2
0.3, 0.3	780.8	855.1

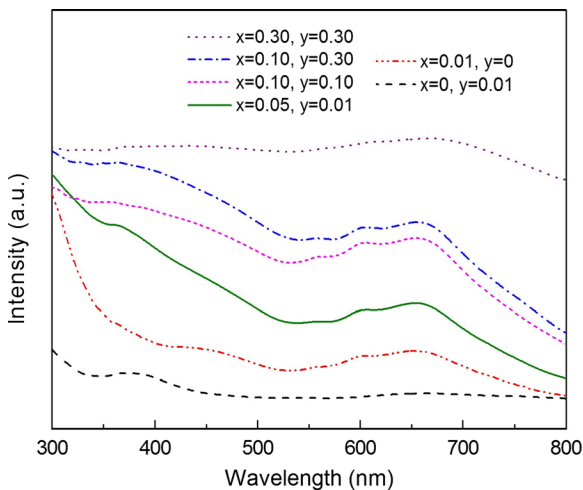


Fig. 6 Absorption spectra of $\text{Co}_x\text{Ni}_y\text{Mg}_{1-x-y}\text{Ga}_2\text{O}_4$ nanoparticles with different doping concentrations

et al. 2008; Suzuki et al. 2010). Similarly, for Co-doped sample, two broad bands at around 400~500 and 550~700 nm were observed. The absorption band at 400~500 nm is assigned to Co^{2+} ions in octahedral sites, and the absorption peaks at 550, 600, and 650 nm are assigned to the spin-allowed of ${}^4\text{A}_2({}^4\text{F}) \rightarrow {}^2\text{T}_2({}^2\text{G})$, ${}^4\text{A}_2({}^4\text{F}) \rightarrow {}^2\text{T}_1({}^2\text{G})$, and ${}^4\text{A}_2({}^4\text{F}) \rightarrow {}^4\text{T}_1({}^4\text{P})$ transition of Co^{2+} in tetrahedral sites, respectively (Chen et al. 2002; Gaudon et al. 2009). For Co/Ni co-doped MgGa_2O_4 , the broad band at 350~500 nm is the combination of the absorption of octahedral Co^{2+} and Ni^{2+} ions, while the band at 550~700 nm is mainly due to tetrahedral Co^{2+} and octahedral Ni^{2+} ions. The absorption intensity of the peaks increases with increasing concentration of Co^{2+} and Ni^{2+} ions.

Conclusion

$\text{Co}_x\text{Ni}_y\text{Mg}_{1-x-y}\text{Ga}_2\text{O}_4$ nanoparticles with the size of 15~55 nm were synthesized by citric acid-assisted sol-gel method. The powder XRD pattern confirmed the synthesized MgGa_2O_4 phase and particles size were increased with increase of dopant concentrations. The XPS spectra indicate that Ga^{3+} and Mg^{2+} ions occupy both the tetrahedral and octahedral sites. The fraction of tetrahedral Ga^{3+} ions increases with increasing doping concentration, which is consistent with the change of the proportion of octahedral Mg^{2+} ions. The inverse parameter increases with the increase of the doping concentrations. Two broad absorption bands at 350~500 and

550~700 nm were observed for Co/Ni co-doped MgGa_2O_4 . The absorption is attributed to the combination of octahedral or tetrahedral Co^{2+} and octahedral Ni^{2+} ions. The intensity of the absorption bands increases with the increase of dopant concentrations of Co and Ni.

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Compliance with ethical standards

Conflict of interest statement The authors declare that they have no conflict of interest.

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