

# Recent progress in all-solid-state quantum dot-sensitized TiO<sub>2</sub> nanotube array solar cells

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**Abstract** All-solid-state quantum dot-sensitized TiO<sub>2</sub> nanotube array solar cells have been drawing great attention to solar energy conversion, which break through restrictions in traditional solar cells, such as the high recombination at interfaces of porous TiO<sub>2</sub> films/sensitizers/hole conductors/counter electrodes, instability of dyes, and leakage of solution electrolyte, and so the novel solar cells exhibit promising applications in the future. In this Minireview article, the assembling of solar cells including the preparation of TiO<sub>2</sub> nanotube array photoanodes, quantum dot preparation and sensitization on photoanodes, filling of hole conductors in TiO<sub>2</sub> nanotubes, and selection of counter electrodes are overviewed, and the development course of all-solid-state quantum dot-sensitized TiO<sub>2</sub> nanotube array solar cells in recent years are summarized in detail. Moreover, the influences of TiO<sub>2</sub> nanotube array photoanodes, quantum dots, solid electrolyte, and counter electrodes on photon-to-current efficiencies of solar cells are summarized. In addition, current problems of solid-state quantum dot-sensitized TiO<sub>2</sub> nanotube array solar cells are

analyzed, and the corresponding improvements, such as multisensitizers and passivation layers, are proposed to improve the photoelectric conversion efficiency. Finally, this Minireview provides a perspective for the future development of this novel solar cell.

**Keywords** Solar cells · TiO<sub>2</sub> nanotube arrays · Quantum dots · Hole transporting materials · Counter electrodes · Energy conversion

## Introduction

In recent years, because of the shortage of fossil fuels and the huge demand of energy, new energy has attracted intense attention due to its friendly environment and sustainability, and solar energy is no doubt the most dazzling one. As the third solar cell, quantum dot-sensitized solar cells show many advantages of low cost, simple fabrication progress, tunable band gap and multiple exciton generation effect for solar light harvesting, and excellent persistence compared with silicon-based solar cells and dye-sensitized solar cells (Kamat 2008; Lee and Lo 2009; Guan et al. 2011; Liu et al. 2012b; Speirs et al. 2015). TiO<sub>2</sub> porous materials are mainly used as photoanode materials in quantum dot-sensitized solar cells due to their high BET for quantum dot adsorption; however, their inherent disadvantages limit the improvement of

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photon-to-current efficiencies. Firstly, porous materials are composed by  $\text{TiO}_2$  nanoparticles and mesoporous in an irregular arrangement way, and the defects and grain boundaries in porous materials cause the high resistance, which is unfavorable for the rapid transfer of photogenerated charge carriers. Secondly, the high recombination reduces the power conversion efficiency (PCE) of solar cells. Fortunately, the recent development of  $\text{TiO}_2$  nanotube arrays ( $\text{TiO}_2$  NTs) overcomes these restrictions, and the well-oriented tubular structures could provide excellent electron percolation pathways for vertical charge transfer. In addition, regular tubes in  $\text{TiO}_2$  NTs could effectively adsorb quantum dots (QDs), inhibit the recombination of electron-hole pairs, and then higher PCE could be obtained (Ruby and Sohrab 2011; Su and Zhou 2011; Yang et al. 2015). Ip et al. (2012) prepared quantum dot-sensitized  $\text{TiO}_2$  nanotube array solar cells (QDSTNASC), and PCE of the QDSTNASC achieved 7 %, which attracted immense attention of scientists. However, due to the volatilization, corrosion, leakage, and complex packaging process of liquid electrolyte, liquid-state solar cells show disadvantages of the short lifetime and instability. Based on the above restrictions, the preparation of solid-state QDSTNASC is required.

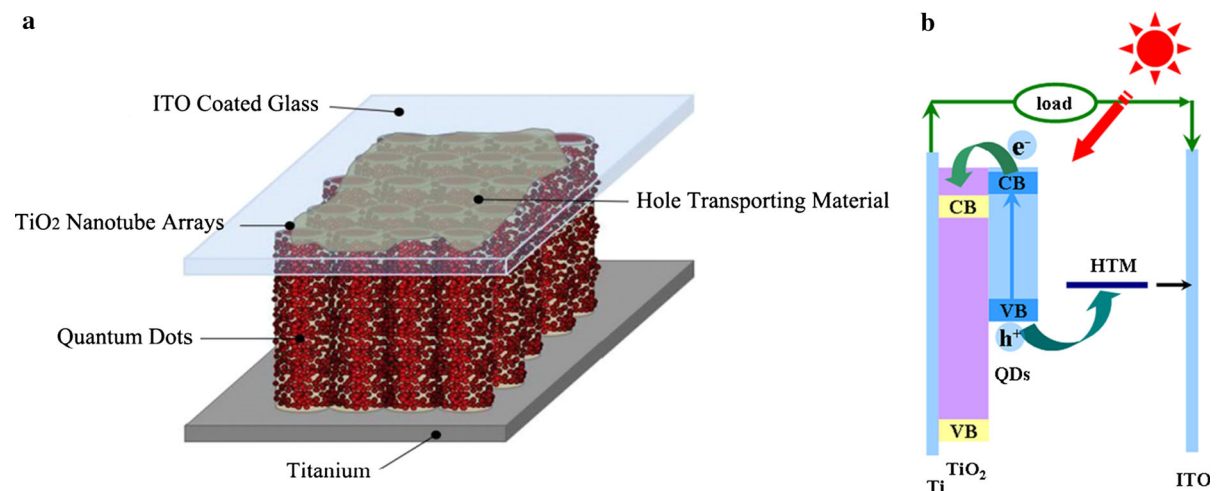
The composition and work mechanism of solid-state QDSTNASC are shown in Fig. 1. Generally, the solid-state QDSTNASC are composed by five parts. (1) Ti substrate or ITO glass; (2)  $\text{TiO}_2$  NTs serving as the backbone of photoelectrodes; (3) The QDs

adsorbed on the surface of  $\text{TiO}_2$  NTs as sensitizers; (4) Hole conductor materials; (5) Counter electrodes. Figure 1b displays the mechanism of solid-state QDSTNASC. Under visible light irradiation, the excited electrons jump into the conduction band (CB) of QDs, leaving holes in the valance band (VB). The photoelectrons transfer rapidly to the CB of  $\text{TiO}_2$  nanotubes, and transfer along nanotube walls to Ti substrates. Otherwise, holes in the VB of QDs pass through hole conductor materials to the counter electrode.

In this paper, we focus on recent progresses in solid-state QDSTNASC, and summarized influences of  $\text{TiO}_2$  NT photoanodes, QDs, modification ways of QDs on  $\text{TiO}_2$  NTs, solid electrolyte, and counter electrodes on the photoelectrochemical (PEC) performance of the QDSTNASC. In addition, the current problem and prospect for future development of the novel solar cell are also proposed.

### Preparation of QD-sensitized photoelectrodes

Photoelectrode materials are the hardcore part of solar cells, and they are composed of photoanodes and QDs sensitizers. The photoelectrode materials play significant roles in solar absorption and transfer of photogenerated electrons, which greatly influence the final PCE. Recently, many investigations on the novel QDSTNASC using  $\text{TiO}_2$  NTs as photoanodes and QDs as sensitizers were reported (Smith et al. 2012; Li et al.



**Fig. 1** The configuration (a) and operation principle (b) of the solid-state quantum dot-sensitized  $\text{TiO}_2$  nanotube array solar cell

2015; Liao et al. 2012), and the PCE were gradually improved.

### Preparation of TiO<sub>2</sub> NT photoanodes

TiO<sub>2</sub> NTs as photoanode materials caused people's intense interest, and investigations about influences of preparation methods, nanotube lengths, and diameters on PCE of solar cells were studied in the previous report (Mor et al. 2006). In this paper, two photoanodes of TiO<sub>2</sub> NTs aligned on Ti substrates or ITO glass were contrasted in detail, respectively.

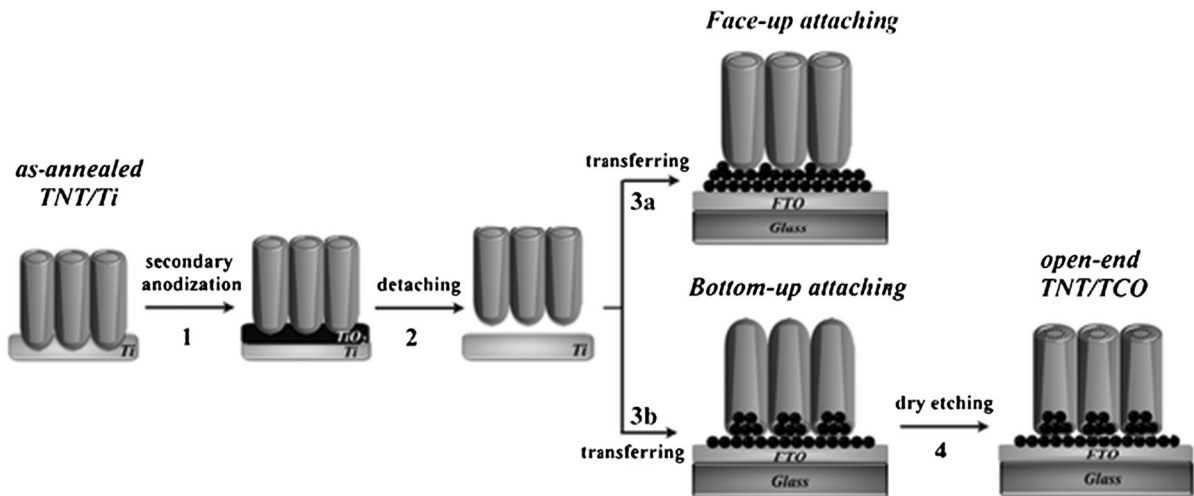
TiO<sub>2</sub> NTs are prepared using the anodization of Ti foils, and the obtained samples could be used as a simple photoanode. The combination of TiO<sub>2</sub> NTs and Ti substrates is tight, which effectively increases the transfer efficiency of electrons. Albu et al. (2010) summarized the formation mechanism of TiO<sub>2</sub> NTs grown on Ti substrates, and investigated influences of anodization parameters, such as voltage, electrolyte composition, electrolyte pH and temperature on the morphology and PEC property of TiO<sub>2</sub> NTs. TiO<sub>2</sub> NTs with different lengths and diameters could be prepared by changing of experimental parameters. However, many disadvantages also exist in this kind of photoanode. Firstly, barrier layers are formed between TiO<sub>2</sub> nanotubes and Ti substrates, and barrier layers inhibit the effective transfer of electrons, increase the recombination chance of electrons and holes, and decrease the PCE. Secondly, due to the light-tight trait of Ti substrates, solar cells only could be irradiated from top of TiO<sub>2</sub> nanotubes which couldn't utilize solar light from the opposed direction. On the contrary, TiO<sub>2</sub> NTs on ITO glass could overcome this restriction, and two sides of the photoanode could be irradiated. Wang and Lin (2009) sputtered a thin layer of Ti on ITO glass, and then the Ti layer was anodized to prepare TiO<sub>2</sub> NTs. The anodization time is hard to be controlled, and a thin barrier layer is still inevitable. Currently, the common method to prepare TiO<sub>2</sub> NT photoanodes with top and back irradiation was described that the free-standing TiO<sub>2</sub> NTs were transferred onto ITO glass, and combined with the substrate by TiO<sub>2</sub> sol. The free-standing TiO<sub>2</sub> NTs could be mainly prepared by four methods including mechanical or ultrasonic forced stripping, electrochemical extraction, chemical dissolution separation, and selective dissolution separation. The detailed stripping and through-hole methods were reported in

the previous review (Liu et al. 2012a). The selective dissolution separation method was an effective way to prepare intact free-standing TiO<sub>2</sub> NTs. The detailed progress is given as following: First, the TiO<sub>2</sub> NTs with proper lengths were prepared by anodization of Ti foils. The TiO<sub>2</sub> NTs were annealed at 250 °C for 2 h, and then anodized for 10 min. The samples were immersed into H<sub>2</sub>O<sub>2</sub> solution for 5 min to prepare through-hole TiO<sub>2</sub> NTs. At last, the combination of TiO<sub>2</sub> NTs and ITO glass was carried out by the hydrolysis of TiCl<sub>4</sub>. The preparation progress of the through-hole TiO<sub>2</sub> NT photoanode is displayed in Fig. 2 (Li et al. 2011b). Hsiao et al. (2011) investigated transfer kinetics of photogenerated electrons on the wall of the through-hole TiO<sub>2</sub> NT solar cells irradiated from top and back surfaces, and the results indicated that electrons could be easily trapped from the back irradiation because of massive defects and crystal boundaries, and the PCE were also lower than that of top irradiation.

### QDs-sensitized TiO<sub>2</sub> NTs

#### *Matchability of QDs with TiO<sub>2</sub> NTs*

Recently, various QDs including noble metals, TiO<sub>2</sub> nanoparticles, and semiconductors with narrow band gaps, were explored to sensitize TiO<sub>2</sub> NTs. Wet-chemical and electrochemical deposition methods were mainly applied to prepare noble-metal QDs, such as Au (Zhang et al. 2013), Ag (Chang et al. 2013b), and Pt (Wang et al. 2012d), on the surface of TiO<sub>2</sub> NTs. The formed Schottky junction on the interface of metal QDs and TiO<sub>2</sub> NTs plays significant roles in the following aspects: (1) The rapid transfer rate of photogenerated electrons on the surface of metals, (2) Visible light absorption of noble metals, such as the absorption peak of Ag nanoparticles at 410 nm, (3) Plasma resonance absorption produced by the interaction of metal nanoparticles and TiO<sub>2</sub> nanotubes. In addition, the investigations indicated that the deposition of TiO<sub>2</sub> nanoparticles on the surface of TiO<sub>2</sub> NTs also significantly improved the PCE. TiO<sub>2</sub> nanoparticle sensitizers were prepared by the hydrolysis of TiCl<sub>4</sub> on the surface of TiO<sub>2</sub> NTs, which increased the BET and PCE of solar cells. Prof. Roy et al. (2009) successfully prepared TiO<sub>2</sub> nanoparticles with diameters of 3 nm on TiO<sub>2</sub> NTs by the hydrolysis of TiCl<sub>4</sub>, and the prepared TiO<sub>2</sub>

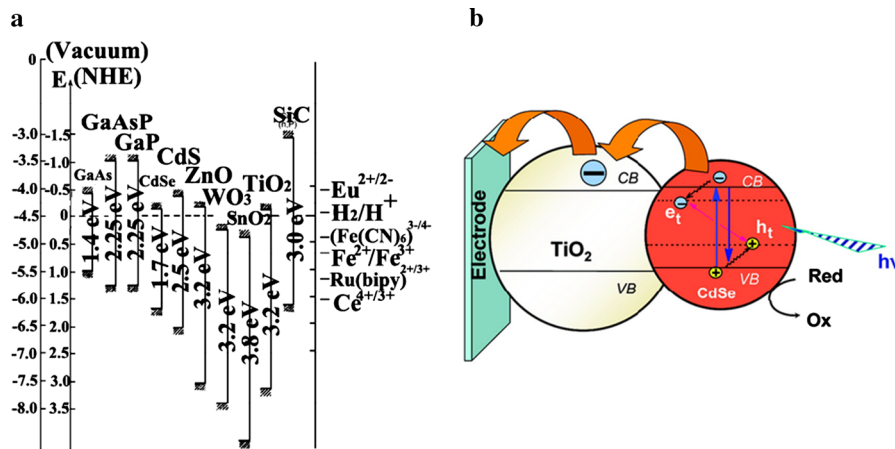


**Fig. 2** Schematic representation of the detachment and transfer of the ordered 1D nanotube arrays for front-illuminated TiO<sub>2</sub> NT solar cells (Li et al. 2011b)

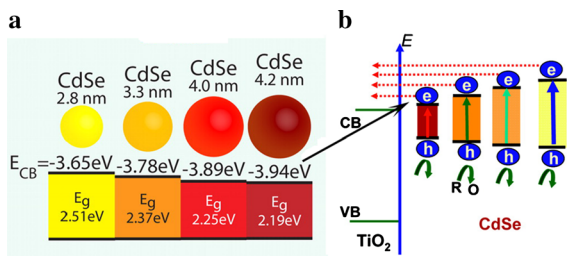
nanoparticles were annealed to form mixed anatase and rutile phases, which increased the PCE from 1.9 to 3.8 %.

The visible light absorption intensities of TiO<sub>2</sub> NT photoanodes sensitized by the above sensitizers are weak due to their inherent energy band structures. Fortunately, the sensitization by semiconductor QDs with narrow band gaps could overcome this restriction. The matchability of QDs with TiO<sub>2</sub> NTs is mainly based on the following three aspects: (1) The band gaps should match well with solar spectrum, and 1.5 eV is the optimal band gap, which could cover the whole visible light region; (2) The CB bottom should be more negative than that of TiO<sub>2</sub>, and photogenerated electrons in QDs could transfer rapidly to that of TiO<sub>2</sub>; (3) The transform of electrons should be rapid and effective, which could reduce the recombination rate of photogenerated electron–hole pairs. Figure 3a shows band gaps and energy band structures of many semiconductors, and only a few ones match these criteria. As shown in Fig. 3b, CdS and CdSe QDs are suitable sensitizers, and the type-II band structure could be formed on the surface of QDs and TiO<sub>2</sub> NTs. The stepwise energy band structures are favor to the transfer of charge carriers, and reduce recombination chances of electron–hole pairs. However, the band gaps and CB edges of QDs aren't commonly immovable, which could be influenced by changing of QDs crystal sizes. Just as CdSe (Chu et al. 2012) and PbS (Ratanatawanate et al. 2008) QDs, decreasing crystal

sizes, the influences of quantum confinement and small-size effects become intense, and band gaps and energy band structures of these QDs are correspondingly changed. As shown in Fig. 4, when sizes of CdSe QDs decreased from 4.2 to 2.8 nm, their visible light absorption regions also changed, the solution color changed from dark red to bright yellow, thus, the CB edge became more negative, and the band gap and energy band structure of QDs could be adjusted. This phenomenon could overcome the restrictions of the poor matchability of semiconductors with TiO<sub>2</sub>, just like the reported PbS QD-sensitized TiO<sub>2</sub> NT solar cells (Kang et al. 2011a). Currently, CdS (Wang et al. 2012b), CdSe (Zhang et al. 2009), CdTe (Wang et al. 2013b), PbS (Chalita et al. 2009), PbSe (Schaller and Klimov 2004), InP (Zeng et al. 2003), Bi<sub>2</sub>S<sub>3</sub> (Li et al. 2012), and Sb<sub>2</sub>S<sub>3</sub> (Xie et al. 2013) QDs were used as sensitizers in solar cells, and sizes of QDs were adjusted to optimize band gaps and energy band structures, which could achieve the optimal solar light absorption and electron transfer. Among these QDs, CdS and CdSe QDs were the most effective sensitizers. Baker and Kamat (2009) prepared CdS QDs with a diameter of 2 nm to sensitize TiO<sub>2</sub> NTs by a successive ionic layer adsorption and reaction (SILAR) method, and the open voltage was 1.15 V, short photocurrent was 6.78 mA/cm<sup>2</sup>, quantum efficiency was 55 % at 517 nm, much higher than that of CdS QD-sensitized TiO<sub>2</sub> porous layer photoelectrodes. Xie et al. (2010b) prepared CdS QDs with a diameter of



**Fig. 3** a Band gaps and energy band structures of several semiconductors, b schematic diagram of excited electron transfer in the TiO<sub>2</sub> NTs sensitized by CdSe QDs (Hagfeldt and Grätzel 1995)



**Fig. 4** Diagram of the relative electronic band gaps and energy band structures of CdSe QDs with different size (Tvrđy et al. 2012)

2 nm on inner walls of TiO<sub>2</sub> NTs by a new ultrasonic-assisted chemical precipitation method, and the open voltage was 1.05 V, higher than 0.98 V of the photoelectrode prepared by a traditional wet-chemical method. The photocurrent density of the as-prepared photoelectrodes improved up to 2.06 mA/cm<sup>2</sup> from 1.43 mA/cm<sup>2</sup>. Si et al. (2008) sensitized TiO<sub>2</sub> NT photoelectrodes by physical adsorption of the prior prepared CdSe QDs, and the influences of QD sizes on the visible light response and photocurrent densities were investigated. The results indicated the photocurrent density of solar cells sensitized by CdSe QDs with diameters of 4.5 nm was 1.5 mA/cm<sup>2</sup>, however, the photocurrent of solar cells sensitized by CdSe QDs with diameters of 3 nm was improved to 1.8 mA/cm<sup>2</sup>.

The band gaps of these QDs such as CdS and CdSe nanoparticles are nearly changeless, therefore, the visible light response regions of QDs are limited in the scope of their energy bands. Although the visible light

absorption region could be slightly adjusted by changing their sizes, the extension is extremely limited. To further extend the visible light response of these QD sensitizers, ternary direct band gap I–III–VI<sub>2</sub> QDs such as CuInS<sub>2</sub> (Chen et al. 2011a), CuInSe<sub>2</sub> (Nguyen et al. 2012), ZnCdSe<sub>2</sub> (Zheng et al. 2011), ZnCdSe<sub>2</sub> (Kang et al. 2012), and CuAlS<sub>2</sub> (Huang et al. 2010a) have many advantages of non-toxicity, low cost, stable properties, and regulatable band gaps and energy band structures by adjusting two metal element ratios. Prof. Chang et al. (2011) deposited CuS and In<sub>2</sub>S<sub>3</sub> layers on the surface of TiO<sub>2</sub> NTs by a SILAR method, and then CuInS<sub>2</sub> QDs were formed by the post annealing treatment. The influences of Cu/In mole ratios on the visible light response and PCE were investigated in detail, and the transient photocurrent density achieved 300 μA/cm<sup>2</sup>.

Same QDs with various diameters were deposited on TiO<sub>2</sub> NTs to maximally absorb solar light. The QDs with small diameters are distributed on the top of tubes to absorb UV and visible light with short wavelengths, and QDs with large diameters are distributed on bottom parts of nanotubes to absorb visible light with long wavelengths. The cap structures could absorb the maximum solar light, and the efficiency of solar cells is significantly improved. Moreover, different QDs with various band gaps on TiO<sub>2</sub> NTs could form heterojunction, and synergistic effect could induce the visible light absorption and PCE. Taking CdS and CdSe QDs for example, the band gap of CdSe QDs is 1.74 eV, much narrower than that of CdS (2.4 eV),

and visible light with long wavelengths could be absorbed and the utilization efficiency of solar light is high (Li et al. 2014). However, the electron transfer rate of CdSe QDs is much lower than that of CdS, and the electron injection rate and compatibility between CdSe and TiO<sub>2</sub> NTs are also lower than that of CdS QDs. Therefore, combination of two QDs will obtain the synergistic enhancement in the visible light absorption and electron transfer efficiency (Shin et al. 2010). Prof. Cheng et al. (2012) and Wang et al. (2013a) prepared the co-sensitization of CdS and CdSe QDs on TiO<sub>2</sub> NTs by a spray pyrolysis method and SILAR method, respectively. The heterojunctions were formed between CdS QDs and CdSe QDs, and the co-sensitization not only improved the visible light absorption, but also inhibited the recombination of electrons and holes, which greatly improved the PCE of solar cells.

#### *The sensitization of TiO<sub>2</sub> NTs by QDs with novel microstructures*

The QDs with novel microstructures could further extend the response region to visible light or near-infrared light, accelerate the transfer rate of photo-generated electrons, improve the effective connection of QDs and TiO<sub>2</sub> NTs, reduce the combination of electron-hole pairs, and finally improve the photon-to-current conversion efficiency of solar cells.

- (1) *Heterojunction QDs* In recent years, the PCE of single QD-sensitized TiO<sub>2</sub> NT photoelectrodes has been limited by inherent photoelectrochemical performances of these QDs. Co-sensitization including two or multiple QDs was reported to sensitize TiO<sub>2</sub> NTs for further synergistic enhancement of PEC performances (Lee et al. 2009a; Lai et al. 2012). Therefore, material scientists are trying their best to improve the visible light response, electron transfer, and photocurrent density of the photoelectrodes co-sensitized by two or multiple QDs. However, high recombination is still observed in the simple hybrid of various sensitizers due to their poor interfaces, and electrons may be trapped by grain boundaries at the interface of sensitizers. Fortunately, the synergistic effect of co-sensitizer heterojunctions could effectively promote the transfer of the photogenerated electrons. For

example, Chang et al. (2013a) successfully prepared solar cells composed of PbS/CdS colloidal QD heterojunctions, and the solar cell exhibited an average PCE of 3.5 %, which opened up the possibility of novel solar cells sensitized by heterojunction QDs.

- (2) *QDs with core-shell structures* The combination and separation are two contradictory progresses of electron transfer in solar cells. The introduction of core-shell structures into QDs provides a new way to optimize photoelectrodes via adjusting energy band structures and reducing the recombination of electrons and holes. The heterojunction is formed at the interface of core-shell QDs, and hierarchical energy bands of core-shell QDs could facilitate the transfer of charge carriers. Approaches for core-shell QDs synthesis can be broadly divided into two categories: “top-down” and “bottom-up”, and the bottom-up approach has proven more suitable (Chaudhuri and Paria 2012). The common preparation technology of core-shell QDs is deposition of shell materials on the pre-existing core materials by hydrolysis or wet-chemical deposition. This technology should be based on similar properties such as lattice matching in core and shell materials. LBL assembly technique also could be applied to deposit shell materials with negative charge on the surface of core materials, and the adjustment of the core-shell thickness and composition by reaction time and LBL cycle experimental parameters could optimize PEC properties of photoelectrodes. Our group (Wang et al. 2015) successfully prepared CdS@CdSe and CdS@CdTe core-shell QDs on TiO<sub>2</sub> NTs by the ionic exchange method using CdS QDs as sacrificial templates. The new photoelectrodes significantly enhanced the visible light response, and the photocurrent densities of TiO<sub>2</sub> NTs/CdS@CdSe and TiO<sub>2</sub> NTs/CdS@CdTe solar cells achieved 22.6 and 14.7 mA/cm<sup>2</sup>, respectively. Apart from in situ deposition, the core-shell QDs could be deposited on the surface of TiO<sub>2</sub> NTs by a coupling agent when the direct combination of QDs and TiO<sub>2</sub> NTs is different due to the huge surface differences. Prof. Wang et al. (2013a) prepared CdTe QDs by an injecting method, and sensitized CdTe with

CdSe layers to prepare CdTe@CdSe core-shell QDs with diameters of 4.9 nm via a SILAR method. Then the as-prepared core-shell QDs were deposited on the surface of TiO<sub>2</sub> NTs by MPA coupling agents, and the PCE of this core-shell solar cell achieved 6.76 %.

- (3) *Hierarchical QDs* Hierarchically structured QDs with various morphologies have attracted immense attention, and the hierarchical structure could significantly improve optoelectronic performances of solar cells. The hierarchical branching frameworks are promising for designing high-performance PEC electrodes owing to their effective optical paths for the efficient light harvesting and high-quality conducting channels for the rapid carrier separation and transportation, as well as large surface areas for fast interfacial carrier transfer and electrochemical reactions. Zhou et al. (2011) prepared novel hierarchical Cu<sub>2</sub>ZnSnS<sub>4</sub> particles by a solvothermal method, and investigated the influences of hierarchical structures on the performance of photoelectrodes. The results indicated that hierarchical sensitizers significantly promoted the absorption coefficient in the visible light region, electronic transmission, and photocurrent density. Prof. Bierman and Jin (2009) reviewed the investigation of hierarchical nanowire structures formed by branching nanowires for solar conversion devices, and the applications in solar energy harvesting and conversion are summarized in detail.

### *Sensitization methods of QDs on TiO<sub>2</sub> NTs*

Novel QDs attracted intense attention in applications of TiO<sub>2</sub> NT photoelectrodes due to their high transfer efficiencies of photogenerated electrons, and the sensitization methods of QDs on TiO<sub>2</sub> NTs also play significant roles in the final PCE. Taking Ag QDs for example, Ag QDs have been deposited on the surface of TiO<sub>2</sub> NTs by many methods, such as UV reduction (Chen et al. 2013), electrochemical deposition (Huang et al. 2011b; Lai et al. 2010), and wet-chemical reduction (Macak et al. 2007). Among these methods, UV reduction and electrochemical deposition methods are popular ways to prepare well-distributed Ag nanoparticles with the uniform morphology and narrow size distribution. In the UV reduction progress,

UV light with high energy could reduce the Ag<sup>+</sup> ions adsorbed on TiO<sub>2</sub> NTs into Ag QDs with the uniform distribution and small crystal size. Prof. Paramasivam et al. (2007) successfully prepared Ag QDs by a UV reduction method, and the detailed experimental progress are listed as following: The TiO<sub>2</sub> NTs prepared by anodization of Ti foils were immersed into 1 M AgNO<sub>3</sub> solution, and then UV irradiated for 0.5 h, so Ag QDs were adsorbed on top and inner surfaces of TiO<sub>2</sub> nanotubes. Li et al. (2010) and Xie et al. (2010a) prepared TiO<sub>2</sub> NTs sensitized by Ag QDs by constant voltage and pulse electrodeposition methods, respectively, and the sizes of Ag QDs could be adjusted by changing experimental parameters. In addition, chemical reducing agents were also used to prepare Ag QDs, but the violent reaction rate using strong reducing agents caused the agglomeration of Ag QDs. Prof. Liang et al. (2011) and Wang et al. (2012c) prepared Ag QDs on the surface of TiO<sub>2</sub> NTs by chemical bath deposition and SILAR methods using NaBH<sub>4</sub> as a reducing agent. The sensitization of noble-metal QDs on TiO<sub>2</sub> NTs could significantly reduce the recombination of photogenerated electrons and holes, however, the visible light harvesting still showed at a low level. Therefore, compared with semiconductor sensitizers, the low PCE of noble-metal QDs-sensitized TiO<sub>2</sub> NT photoelectrodes limit their potential applications.

Chalcogenide semiconductor QDs could be deposited on TiO<sub>2</sub> NTs by the in situ growth and *non*-in situ growth. Electrochemical deposition, chemical bath deposition, and SILAR deposition are in situ growth styles. The in situ growth progress includes the progress of the nucleation and growth of QDs, and this style shows high coverage and tight combination of QDs and TiO<sub>2</sub> NTs. The disadvantage of the in situ style is the uncontrollable growth of these QDs sensitizers. The external energy (voltage and current) in the electrochemical deposition progress drastically induces the nucleation and growth of QDs, and the rapid growth rate may cause the agglomeration on TiO<sub>2</sub> NT surface, which blocks the inner sensitization (Liu et al. 2011; Huang et al. 2011a). Chemical bath deposition and SILAR methods are mainly used in the sensitization of chalcogenide semiconductor QDs on TiO<sub>2</sub> NTs. The adsorbed S<sup>2-</sup> ions could react with metal ions to form chalcogenide semiconductor QDs on the inner surface of TiO<sub>2</sub> NTs, and this method has been extended to the selenide and telluride. Sun et al.

(2008) and Hossain et al. (2011) prepared CdS and CdSe QDs-sensitized TiO<sub>2</sub> NTs by a chemical bath deposition method, and their PCE achieved 4.15 and 1.56 %, respectively. Prof. Cheng et al. (2011) and Kang et al. (2011b) prepared CdS and PbS QDs-sensitized TiO<sub>2</sub> NT photoelectrodes by a SILAR method, respectively, and the solar cells showed high PCE. In addition, our group (Zhong et al. 2014) developed a novel photoreduction method to prepare chalcogenide QDs for the sensitization of TiO<sub>2</sub> NTs. Photodeposition technique is recognized as an effective in situ growth method to achieve tight interfacial connection between semiconductor QDs and TiO<sub>2</sub> NTs, and the TiO<sub>2</sub> NTs/semiconductors heteronanojunction is formed by the photocatalytic redox property of TiO<sub>2</sub> NTs. These metal sulfide QDs can be photodeposited not only on external surfaces but also on inner surfaces of mesopores. We successfully achieved TiO<sub>2</sub> NTs sensitized by CdS (Zhong et al. 2014) and PbS (Qiao et al. 2014) QDs by this method, and their visible light photocurrent densities achieved 6.1 and 9.49 mA cm<sup>-2</sup>, respectively. Other binary and tribasic chalcogenide QDs such as Sb<sub>2</sub>S<sub>3</sub> and CuInS<sub>2</sub> are being investigated to further improve the PCE of solar cells.

The *non*-in situ growth styles including physical adsorption and chemical adsorption by organic coupling agents are also exploited to deposit QDs on TiO<sub>2</sub> NTs. In the *non*-in situ growth, the pre-synthesized QDs were deposited on the surface of TiO<sub>2</sub> NTs by Van der Waals forces or coupling agents. The lower QDs coverage of solar cells is the main disadvantage of physical adsorption, and the lower than 14 % coverage results in low PCE. Organic coupling agents in chemical adsorption could induce the top and inner sensitization of TiO<sub>2</sub> NTs, and sizes of the adsorbed QDs could be accurately adjusted. However, the combination between QDs and TiO<sub>2</sub> NTs is weak, and some organic molecules also inhibit the transport of photogenerated electrons. Two kinds of organic coupling agents are commonly applied to sensitize TiO<sub>2</sub> NTs. Mercaptoacetic acid is a small molecule coupling agent with short molecule chains. The carboxyl group in one end of molecules could adsorb TiO<sub>2</sub> NTs, and sulfhydryl group in the other end could adsorb these QDs (Gao et al. 2009). L-cysteine is another small molecule coupling agent that has carboxyl group and amidogen in two ends of molecule chains. Mercaptoacetic acid is extensively used in

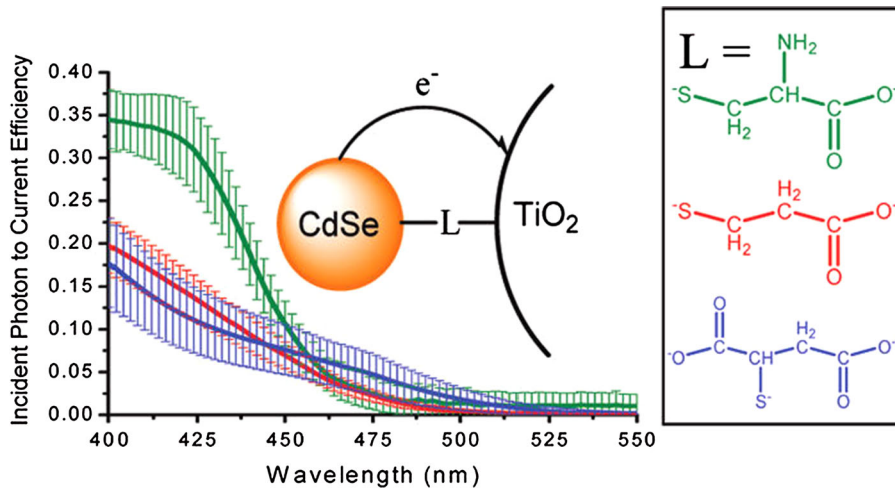
previous investigations, but the electron injection efficiency of QDs sensitized by L-cysteine is more promising. The adsorption mechanism of organic coupling agents is shown in Fig. 5 (Nevins et al. 2011). Prof. Nevins et al. (2011) investigated the adsorption efficiencies of L-cysteine, mercaptoacetic acid, and mercaptosuccinic acid on CdSe QDs, and the PCE of TiO<sub>2</sub> NTs/CdSe photoelectrodes sensitized by L-cysteine is 1.2 and 6-fold higher than that of photoelectrodes sensitized by mercaptoacetic acid and mercaptosuccinic acid, respectively. The high PEC performance of solar cells sensitized by L-cysteine is attributed to the higher electron transfer rate and the low recombination rate of electron-hole pairs.

Organic coupling agents in chemical adsorption are mainly used to deposit chalcogenide semiconductors on TiO<sub>2</sub> NTs, but metallic oxide QDs with narrow band gaps are generally prepared by an electrochemical deposition method (Wen et al. 2011). Generally, metallic oxide QDs are prepared by annealing treatment of hydroxide precursors produced during the preparation progress of metal cations in alkaline environment. The hydroxide precursors are flocculent precipitates prepared by wet-chemical methods, which easily block nanotubes and few QDs could be deposited on the inner surface of TiO<sub>2</sub> NTs. Compared with traditional wet-chemical methods, the electrochemical energy provided from an electrochemical workstation could induce crystal nucleations on inner walls of TiO<sub>2</sub> NTs, and this method demonstrates tremendous advantages in the preparation of metallic oxide semiconductors. Li et al. (2011a) and Zhang et al. (2011) prepared Cu<sub>2</sub>O QDs with octahedron structures to sensitize TiO<sub>2</sub> NTs by pulse electrodeposition and constant voltage methods, which greatly improved photocurrent densities of the QDSTNASC, and the detailed mechanism and PEC results are given in Fig. 6.

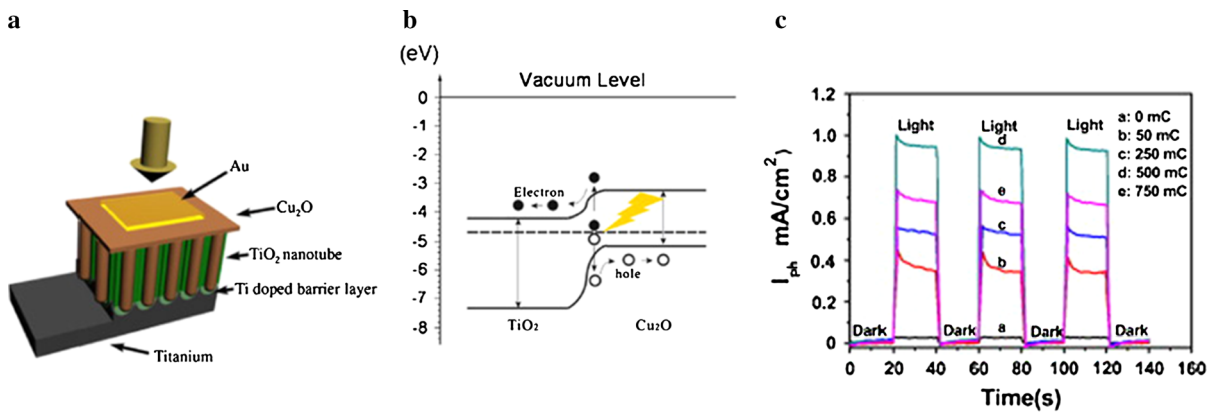
### Selection of hole conductors

The overall PCE of most solid-state solar cells is much lower than that of liquid-state solar cells, and the further improvement is difficult to break through. The main obstacle is ascribed to hole conductors, especially the low filling ratio of these hole conductors in nanotubes. Contrary to liquid electrolyte, solid-state





**Fig. 5** The incident photon-to-current efficiency of TiO<sub>2</sub> NTs sensitized by CdSe using cysteine, 3-mercaptopropionate, and mercaptosuccinate as coupling agents (Nevins et al. 2011)

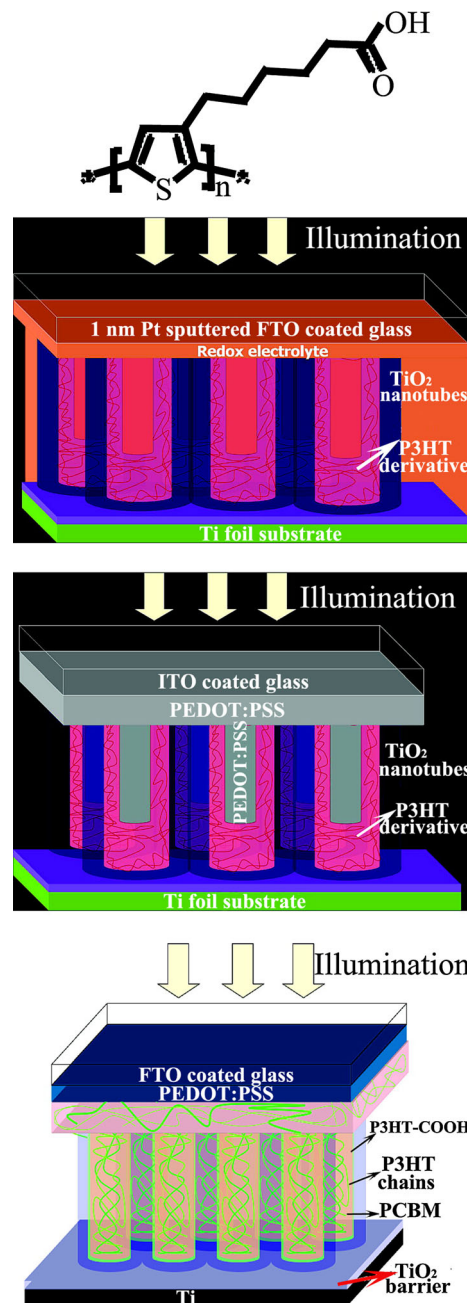


**Fig. 6** Schematic diagram (a), energy band diagram (b), and photocurrent density profiles (c) of Cu<sub>2</sub>O/TiO<sub>2</sub> NTs solar cells (Li et al. 2011a; Zhang et al. 2011)

hole conductors show the low filling ratio and weak connection with QDs, the poor interface connection reversely inhibits the effective transfer of photogenerated charge carriers, and the low transfer efficiency correspondingly causes the intense recombination of electron/hole pairs. Therefore, the superior hole conductors and interface connections are the presupposition of outstanding solid-state solar cells with high PCE. There are four criteria for the selection of hole conductors: (1) VB edges of hole conductors should match well with that of QDs, which could be in favor of the hole flow on the interface of QDs/hole conductors; (2) Hole conductors should be amorphous solids, and the reason is that the crystal of hole

conductors may inhibit the complete filling in TiO<sub>2</sub> NTs; (3) The hole transfer rate in hole conductors should be high; (4) Hole conductors should show high stability and low absorption in the visible light region. The effective filling and interface connection between hole conductors and TiO<sub>2</sub> NTs play key influences on the PEC of solar cells, and the transfer rate of hole conductors is just a secondary consideration. There are decades for investigations of the solid-state electrolyte, and all these hole conductors could be roughly classified into three kinds: (1) Inorganic salt electrolyte, such as CuI, CuBr, CuSCN, NiO, CsSnI<sub>3</sub>, and etc.; (2) Small organic molecule electrolyte, such as Spiro-OMeTAD; (3) Macromolecule electrolyte, such

as polyaniline, polypyrrole, polythiophene, and etc. CuI and CuSCN are the most popular inorganic salt electrolyte. They have the advantage of rapid hole transport rates, but also have disadvantages of the crystallization and instability which are main reasons for the low PCE of solid-state solar cells. Chen et al. (2011a) prepared CdSe QDs-sensitized TiO<sub>2</sub> porous membrane as the photoelectrode, and CuSCN was selected as the hole conductor. The solid-state solar cells showed low PCE, and its photocurrent density was only 6.5  $\mu\text{A}/\text{cm}^2$ . The reason was attributed that the low transfer rate of holes and incompact interface connection with TiO<sub>2</sub> nanoparticles caused low PEC properties. CsSnI<sub>3</sub> is a new hole conductor prepared in recent years, and the PCE of solid-state solar cells using SnF<sub>2</sub>-doped CsSnI<sub>3</sub> as the hole conductor achieved 10.2 %, which approached the level of liquid electrolyte (Chung et al. 2012a, b). In addition, the solid-state solar cell showed high stability of effective running for 25 years. Unfortunately, the super high cost limited investigations only in the level of theoretical research in laboratories. Spiro-OMeTAD is recognized as the optimal solid-state hole conductor, and it has advantages of high hole transfer rate, excellent wettability, no polymerization, and simple spin-coating performance in tubes, so solar cells using this organic electrolyte showed higher PCE than others. Using Spiro-OMeTAD as the hole conductor, Prof. Chi et al. (2011) prepared solid-state CdS/CdSe QDs-sensitized TiO<sub>2</sub> solar cells, and the PCE achieved 0.88 %. However, the high cost similarly goes against the extensive applications. Due to the high hole transfer rate (550 S/cm), thermal stability, and low cost, polythiophene is recognized to be the most prospective hole conductor. As shown in Fig. 7, Shankar et al. (2007) prepared solid-state solar cells by physical spin coating of polythiophene on the surface of TiO<sub>2</sub> NTs, but the filling ratio is still low. To improve the filling ratio, the in situ polymerization is explored to be complete filling by monomer polymerization in nanotubes. Chemical polymerization, electrochemistry polymerization, and UV light polymerization are the main in situ polymerization styles of polythiophene in TiO<sub>2</sub> NTs. The investigations indicated that solar cells filled by polythiophene polymerized by UV light-assisted polymerization methods showed high PCE. Prof. Lim et al. (2012) prepared polythiophene hole conductors by UV polymerization methods. Polythiophene was then doped by



**Fig. 7** Schematic diagram of TiO<sub>2</sub> nanotube arrays coated with self-assembled carboxylated P3HT (Shankar et al. 2007)

I<sup>-</sup> ions, and PCE of solar cells improved 27.52 % more than that without doping treatment. The VB edges could be adjusted by the addition of Li<sup>+</sup> ions and other small molecules, which could increase the filling efficiency and transfer rate of hole carriers, and it is an effective route to decrease the recombination of

electrons/holes and improve the PCE of solid-state solar cells. Compared with chemical polymerization and UV light polymerization, the in situ electropolymerization could assist the orientated growth of hole conductors from the bottom of TiO<sub>2</sub> nanotubular electrodes, and finally achieve the complete filling of mesoporous structures. For example, poly-3,4-ethylenedioxythiophene (PEDOT) conducting polymers were electrodeposited in TiO<sub>2</sub> NTs, and electropolymerization typically leads to the deposition of PEDOT in tubes and spaces among tubes, which constructed compact PEDOT/TiO<sub>2</sub> NTs interfaces for hole transfer (Kowalski et al. 2013).

### Selection of counter electrodes

Counter electrodes (CE) are the key composition of solar cells, and they have significant roles to play in collecting holes in the circuit. In the early period of solar cells, PCE are so low that the influence of CE on the PCE is faint, and the investigation about the preparation and design of CE is scarce. However, with the improvement of PCE and commercial applications of solar cells, especially emergence of solid-state solar cells, the investigations of CE materials increased enormously, and so many papers reported investigations about the preparation and modification of various CE materials. The CE materials could be divided into the following four categories: (1) Pt and Au noble-metal counter electrodes; (2) Cu<sub>2</sub>S and its derivative counter electrodes; (3) Carbon counter electrodes; (4) Polymer and other organic counter electrodes.

#### Pt, Au noble-metal counter electrodes

Noble-metal counter electrodes mainly consist of Pt, Au, and Ag noble metals, and Pt layers are the most extensive and optimal CE materials. Several methods, such as magnetron sputtering (Dao et al. 2011) and thermal decomposition (Wang et al. 2005), were applied to prepare Pt CE on ITO/FTO glass with high electrochemical performances. However, the high cost and complex preparation technologies of Pt and Au CE materials greatly limited the large-scale application in solar cells. Therefore, people made great efforts to explore the substitutes with low cost and high efficiency.

#### Cu<sub>2</sub>S and its derivative counter electrodes

Pt and other noble metals like Au show low catalytic activities, which are generally applied in dye-sensitized solar cells, and they aren't suitable for the QDSTNASC. Alternative CE materials, such as CoS, Cu<sub>2</sub>S, NiS, and PbS have been investigated in recent years. Cu<sub>2</sub>S has been demonstrated to be a promising CE material for solar cells owing to its low cost, simple fabrication, and broad variety of materials, as well as excellent catalytic activity. Based on Cu<sub>2</sub>S CE materials, Xu et al. (2012b) investigated the derivatives of Cu<sub>2</sub>S, and the new CE materials of monodisperse tetragonal Cu<sub>2</sub>SnS<sub>3</sub> and rhombohedral Cu<sub>1.8</sub>S hierarchical microspheres were prepared via a simple solvothermal approach. The new CE promoted significant enhancement of PCE. The Cu<sub>2</sub>ZnSnS<sub>4</sub> compound is another promising candidate for low-cost photovoltaic materials due to the abundances of constituting elements, low toxicity, suitable energy band gap (~1.5 eV), and high absorption coefficient (~10<sup>4</sup> cm<sup>-1</sup>). Xu et al. (2012a) prepared Cu<sub>2</sub>ZnSnS<sub>4</sub> hierarchical microspheres as an effective CE material for QDSSC, and the PCE of this solar cell achieves 3.73 % under AM 1.5G illumination with an intensity of 100 mW cm<sup>-2</sup>, higher than that (2.27 %) of solar cells using noble-metal Pt-coated FTO glass substrates as CE. Moreover, other chalcogenide semiconductor CE materials also showed excellent electrochemical properties. Tachan et al. (2011) prepared PbS as CE materials to significantly improve the PCE.

#### C counter electrodes

Compared with other CE, carbon CE materials have advantages of excellent thermal stability, well corrosion resistance, high photocatalytic property and electroconductivity, as well as extensive sources and low cost. Carbon black, graphite, and porous carbon as traditional carbon CE materials have been intensely studied because of their high BET and electroconductivity to improve the PCE of solar cells. Therefore, the replacement of Pt with low-cost carbonaceous materials could facilitate the commercialization of solar cells. In recent years, the burgeoning CNT, carbon nanofibre, and grapheme as novel CE materials cause more and more attention. Because of their high BET and hole transfer properties, carbon CE materials

greatly improve the PCE of solar cells. Screen painting is the main method to prepare carbon CE, but the complicated preparation progress and weak adhesive attraction are obvious drawbacks to limit their popularization. Electrospun carbon nanofibers have been explored as an electrocatalyst and low-cost alternative to Pt CE. The results indicated that solar cells using carbon nanofibers as CE showed high fill factor and PCE (Joshi et al. 2010). Graphene, which is an atomic planar sheet of hexagonally arrayed  $sp^2$  carbon atoms, has been demonstrated to be a promising CE material due to its outstanding conductivity and high electrocatalytic activity. The fabrication and performance of graphene-based CE were discussed and reviewed in previous paper (Wang and Hu 2012).

### Polymer counter electrodes

Conductive polymers have been widely applied as CE materials in solar cells. The low cost and simple preparation technology of polymer electrodes are natural advantages as substitutes of Pt CE. In addition, the band gap and electroconductivity could be further adjusted by doping and modification of polymer CE materials. Polythiophene CE materials could be prepared by electrochemical polymerization on a FTO substrate, and the solar cells constructed by polythiophene CE showed a higher conversion efficiency (7.88 %) compared with that (7.77 %) of sputtered-Pt CE (Lee et al. 2009b).

Recently, people have paid more attention to the composite CE (Huang et al. 2010b), such as C/CoS (Lin et al. 2011), Pt/C (Li et al. 2009), C/conducting polymers (Sun et al. 2010), and so on. Taking advantage of the merit of single CE material, composite CE materials show higher photoelectrochemical performances than single CE materials, which is the development trend of CE materials in the further studies. Radich et al. (2011) prepared reduced graphene oxide (RGO)/Cu<sub>2</sub>S composite CE, and the sandwich-like CdSe QDSSC constructed by RGO-Cu<sub>2</sub>S composite CE exhibited PCE of 4.4 %, much higher than that of solar cells constructed by Pt CE.

### Conclusions

This Minireview highlights the latest progress in constituent parts of the solid-state QDSTNASC,

including the preparation of TiO<sub>2</sub> NT photoanodes, matchability and modification methods of QDs on TiO<sub>2</sub> NTs, and filling of hole conductors, as well as selection and exploitation of counter electrodes. TiO<sub>2</sub> NTs attached on ITO glass as the photoanode showed higher PCE than that on Ti substrates, and these QDs prepared by chemical bath deposition and SILAR methods on the surface of TiO<sub>2</sub> NTs showed rapid electron transfer rate, compact connection, and high PCE. The capillary force caused by small tube diameters is the main obstacle for the complete filling and effective connection between TiO<sub>2</sub> NTs and hole conductors. The filling of hole conductors in TiO<sub>2</sub> NTs using inorganic salt and macromolecules as hole conductors were reviewed. Various CE materials, including noble-metal CE, Cu<sub>2</sub>S CE, C CE, and polymer CE, were overviewed in detail to investigate the influences of CE materials on PCE.

### Problems and perspective

Looking forward into the future of this exciting field, it is important to keep several questions in mind as well. In recent years, the PCE of QDSTNASC were gradually improved, but still showed extremely low efficiencies, and it was very difficult to achieve a great improvement. However, the PCE of dye-sensitized solar cells and perovskite solar cells exceed 12 and 20 %, respectively. To further improve the PCE of QDSTNASC, many efforts such as enhancing the visible light harvesting and reducing the recombination of electron-hole pairs should be taken. The photoelectrodes should be developed from weak visible light absorption of a single QD to combinatorial modification of multiple QDs or QDs/dyes, and the synergistic sensitization would achieve the broader visible light response until covering the whole solar spectrum. In addition, we should further investigate the kinetic of the carrier transfer and recombination, improve the energy band structure matchability of QDs and TiO<sub>2</sub> NTs, enhance the effective filling and connection of solid electrolyte in the nanotubes, and inhibit the recombination of electrons and holes. For example, passivation layers, such as ZnS, Al<sub>2</sub>O<sub>3</sub>, and MgO films, could be covered on surfaces of QDs and TiO<sub>2</sub> NTs, which would form buffer layers between electron conductors and hole conductors, significantly decrease the photocorrosion of QDs, increase the

stability and connection of QDs and TiO<sub>2</sub> NTs, and finally improve the PCE of the solid-state QDSTNASC. This Minireview is written to provide effective strategies for optimizing the solid-state QDSTNASC and to stimulate future research interests in the development of novel solid-state solar cells with new nanostructures for photoelectric conversion.

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