

White-light-controlled resistance switching in $\text{TiO}_2/\alpha\text{-Fe}_2\text{O}_3$ composite nanorods array

B. Sun · Q. L. Li · W. X. Zhao · H. W. Li ·
L. J. Wei · P. Chen

Received: 1 January 2014 / Accepted: 26 March 2014 / Published online: 11 April 2014
© Springer Science+Business Media Dordrecht 2014

Abstract White-light-controlled resistance switching in $\text{TiO}_2/\alpha\text{-Fe}_2\text{O}_3$ composite nanorods array grown on fluorine-doped tin oxide substrate by hydrothermal process is investigated. The average length of $\text{TiO}_2/\alpha\text{-Fe}_2\text{O}_3$ nanorods is about 3.5 μm , and the average diameter is about 250 nm. The sizes of the $\alpha\text{-Fe}_2\text{O}_3$ particles are in the range of 30 ~ 70 nm. The current–voltage characteristics of the composite nanorods array show a good rectifying property and bipolar resistive-switching behavior, and the resistive-switching behavior can be regulated by white-light illumination at room temperature. This study is helpful for exploring the multifunctional materials and their applications in nonvolatile multistate memory devices.

Keywords $\text{TiO}_2/\alpha\text{-Fe}_2\text{O}_3$ composite nanorods array · White-light-controlled · Resistance switching · Nanocomposites

Introduction

It is undeniable that resistance switching is one of the most promising candidates for the next generation of nonvolatile computer memories, which has high operational speed, high storage density, and low power consumption (Meijer 2008). Currently, a large variety of solid-state materials exhibit the resistance switching effect, including organic materials (Stewart et al. 2004; Ma et al. 2003), binary oxides (Liu et al. 2009; Schindler et al. 2009), amorphous Si (Jo et al. 2009), carbon-based materials (Li et al. 2008; Zhuge et al. 2010; He et al. 2009), and complex perovskite oxides such as $\text{Pr}_{1-x}\text{Ca}_x\text{MnO}_3$ (Liu et al. 2000), $\text{La}_{1-x}\text{Ca}_x\text{MnO}_3$ (Hasan et al. 2008), and $\text{La}_2\text{CuO}_{4+x}$ (Hamaguchi et al. 2006). Recently, a light-controlled resistance switching was observed in $\text{Pd}/\text{Al}_2\text{O}_3/\text{SiO}_2$ film and ZnO nanorod (Ungureanu et al. 2012; Park et al. 2012, 2013; Bera et al. 2013), which are added to the light as extra control parameter for the resistance switching. The light-controlled resistance switching provides the potential for nonvolatile light-controlled memory applications.

In recent years, $\alpha\text{-Fe}_2\text{O}_3$ has received increasing attention due to their extensive applications as magnetic materials, catalysts, pigments, gas sensors, optical and electromagnetic devices, drug delivery, tissue repairing engineering, and electromagnetic devices. $\alpha\text{-Fe}_2\text{O}_3$ nanostructures with various morphologies have been successfully synthesized (Bean and Livingston 1959; Faust et al. 1989; Hyeon 2003;

B. Sun · W. X. Zhao · H. W. Li · L. J. Wei ·
P. Chen (✉)
School of Physics Science and Technology, Southwest
University, Chongqing 400715, China
e-mail: pchen@swu.edu.cn

B. Sun · W. X. Zhao
Institute for Clean Energy & Advanced Materials
(ICEAM), Southwest University, Chongqing 400715,
China

Q. L. Li
The Southwest University Hospital, Southwest University,
Chongqing 400715, China

Chen et al. 2005), and among these structures, one-dimensional α -Fe₂O₃ possesses interesting physical properties, such as good light-harvesting and charge transport properties, and α -Fe₂O₃ with band gap of 1.9–2.2 eV, which could absorb most of visible light, is a semiconductor material (Khan and Aikusa 1999; Beermann et al. 2002). At the same time, semiconductor-based photocatalysis has attracted extensive interest for basic and applied chemical utilization of solar energy (Burschka et al. 2013; Crossland Edward et al. 2013; Hodes 2013; Mitchinson 2008; Gratzel 2003), and one of the most commonly used materials is anatase TiO₂ (band gap 3.1–3.2 eV) (Adachi et al. 2012; Wu et al. 2011; Cao et al. 2011). Rutile TiO₂ has been proven to be comparable to anatase TiO₂ in dye-sensitized solar cells (DSSCs) with additional advantages in visible light including better chemical stability, photochemical activity, and higher refractive index (Liu and Aydil 2009; Diwald et al. 2004; Wang et al. 2007), and the rutile TiO₂ is a key material for water dissociation (Schaub et al. 2001). In addition, for the one-dimensional nature of α -Fe₂O₃ and TiO₂, abundant availability, and the inexpensive and non-toxic nature of both the oxides make this composite material an exciting one for various applications (Liu and Gao 2006).

To the best of our knowledge, although the photocatalytic properties of individual TiO₂ have been intensively investigated (Jang et al. 2001; Beydoun and Amal 1999; Watson et al. 2004; Chu et al. 2008), the resistance switching characteristics of TiO₂/Fe₂O₃ composite materials have not yet been reported. In this paper, we present a white-light-controlled resistance switching behavior in TiO₂/ α -Fe₂O₃ composite nanorods array at room temperature.

Experimental

Preparation of TiO₂/ α -Fe₂O₃ composite nanorods array

TiO₂/ α -Fe₂O₃ composite nanorods array grown on the FTO substrate was prepared by hydrothermal process (Fig. 1a). All chemicals used in this work were of analytic reagent grade and commercially available, and used without further purification. We introduce the synthesis steps of the most preferred one in our experiments, and the detailed experimental procedures

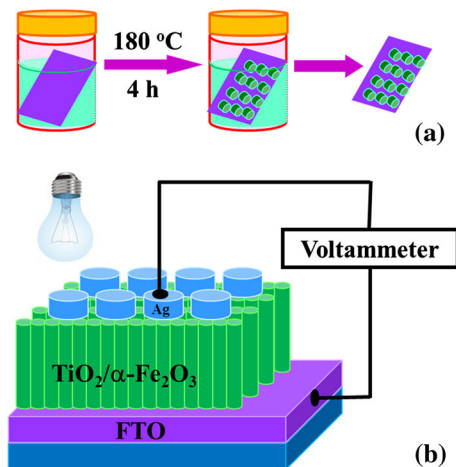


Fig. 1 **a** The preparation process of TiO₂/ α -Fe₂O₃ composite nanorods grown on FTO substrate, **b** the experimental test circuit

are as follows: deionized water (7 mL) was mixed with hydrochloric acid (6.5 mL, 36.5–38 wt%) and stirred for 5 min; then, 0.2 mL titanium(IV) isopropoxide (TIP; 97 %, Sigma) (the TIP is liquid reagent) and 1.35 g FeCl₃·6H₂O (Sigma) were dissolved into the above solution and stirred for a few minutes; and then the mixture solution was transferred to a 50 mL Teflon-lined stainless steel autoclave. In addition, Fluorine-doped tin oxide (FTO)-coated glass substrates (NSG, 14 Ω per square) were cleaned prior to ultrasonic using acetone, ethanol, and deionized water, and subsequently dried in air. Then, the clean FTO substrate was put into the above mixture solution, and the conductive surface was down. The autoclave was put in an oven at a temperature of 180 °C for 4 h. After the autoclave was cooled to room temperature, the FTO substrate was rinsed with deionized water and subsequently annealed at 450 °C for 2.5 h in air.

Characterization

Microstructure of TiO₂/ α -Fe₂O₃ composite nanorods was characterized by X-ray diffraction (XRD, Shimadzu XRD-7000 X-ray diffractometer) with Cu K α radiation. Surface morphology of TiO₂/ α -Fe₂O₃ composite nanorods grown on FTO substrate was characterized using scanning electron microscopy (SEM, JSM-6510). The size, morphology, and the energy dispersive X-ray (EDX) analysis of the TiO₂/ α -Fe₂O₃ composite nanorods were observed by transmission

electron microscopy (JEM-2100) at an acceleration voltage of 200 kV.

Figure 1b shows the schematic diagram of the test circuit. We used ordinary filament lamp with various power densities as light source, and the wavelength white-light is in the range of 400 ~ 760 nm. Ag and FTO are top electrode and bottom electrode, respectively. The Ag electrodes with area about 4 mm² were prepared by silver glue. Electric characterizations were tested using the electrochemical workstation CHI-660D. The resistance switching properties of the samples were examined in the dark and under white-light illumination.

Results and discussion

The crystalline compositions of the samples were characterized by XRD patterns. According to previous reports in the literature (Liu and Aydil 2009), the peak of FTO substrate is obvious. Therefore, in order to make diffraction peaks of TiO₂/ α -Fe₂O₃ clearer, we also present the XRD pattern of the pure FTO substrate without TiO₂/ α -Fe₂O₃ composite nanorods (Fig. 2a(A)). Figure 2a(B) shows the XRD pattern of TiO₂/ α -Fe₂O₃/FTO. We can see that the crystallizations of TiO₂ and α -Fe₂O₃ are vary sufficiently without any other impurity phase. TiO₂ exhibits rutile phase. So, we can obtain a conclusion that the sample is TiO₂/ α -Fe₂O₃ composite nanorods. The composition of TiO₂/ α -Fe₂O₃ is further confirmed by elemental analysis carried out and observed from energy-dispersive X-ray spectra

(EDS). The EDX data in Fig. 2b confirm that the elements of composition nanowire are Ti, Fe and O without any other impurities. The Fe/Ti ratio in the nanorod is about 30 %. We also prepared the TiO₂/ α -Fe₂O₃ nanorods array with other Fe/Ti ratio. But the white-light-regulated resistance switching for the TiO₂/ α -Fe₂O₃ nanorods array with other Fe/Ti ratio is not obvious. Therefore, we just report the results of the TiO₂/ α -Fe₂O₃ nanorods array with Fe/Ti ratio 30 %.

Figure 3a, b shows the SEM image of TiO₂/ α -Fe₂O₃ composite nanorods array grown on FTO substrate. We can see that the as-prepared sample consists of vertically and uniform nanorods (Fig. 3a). The typical cross-sectional SEM image of TiO₂/ α -Fe₂O₃ composite nanorods array is shown in Fig. 3b, which shows that the average length of TiO₂/ α -Fe₂O₃ nanorods is about 3.5 μ m, and the diameter is about 250 nm (The inset to Fig. 3b). Figure 3c shows the TEM image of an individual TiO₂/ α -Fe₂O₃ composite nanorod, which shows that the nanorod is composed of TiO₂ and α -Fe₂O₃, and α -Fe₂O₃ is particle with diameter in the range of 30–70 nm. Figure 3d shows the high-resolution TEM (HRTEM) of TiO₂/ α -Fe₂O₃ composite nanorods. The fringes with a spacing of 0.33 nm correspond to (101) planes of TiO₂, and the fringes with a spacing of 0.3 nm correspond to (300) planes of α -Fe₂O₃, and the inset shows the SAED pattern of the region without α -Fe₂O₃ particles in Fig. 3c, which shows that the TiO₂ of TiO₂/ α -Fe₂O₃ composite nanorod is single-crystalline structure.

Figure 4a shows the current–voltage (I–V) curves of Ag/[TiO₂/ α -Fe₂O₃]/FTO in the dark and under

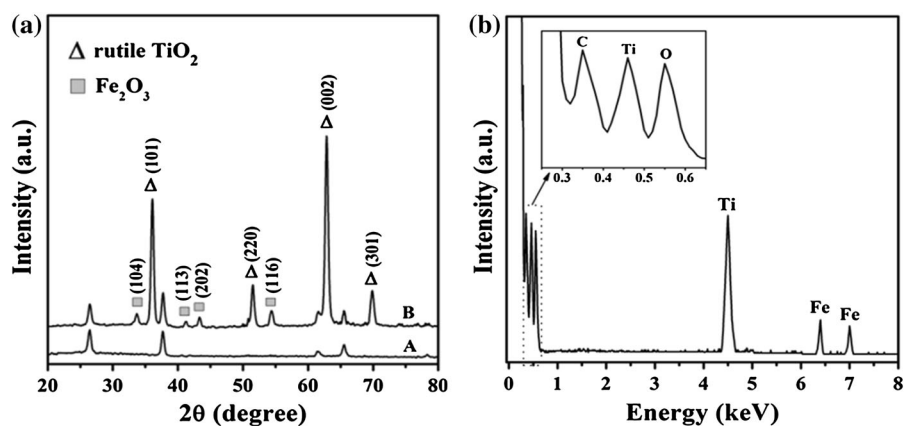


Fig. 2 **a** A The XRD of the FTO substrate. B The XRD of TiO₂/ α -Fe₂O₃/FTO. **b** The EDX spectrum of TiO₂/ α -Fe₂O₃ composite nanorods, which shows the Fe/Ti ratio in the nanorods as 30 % nearly

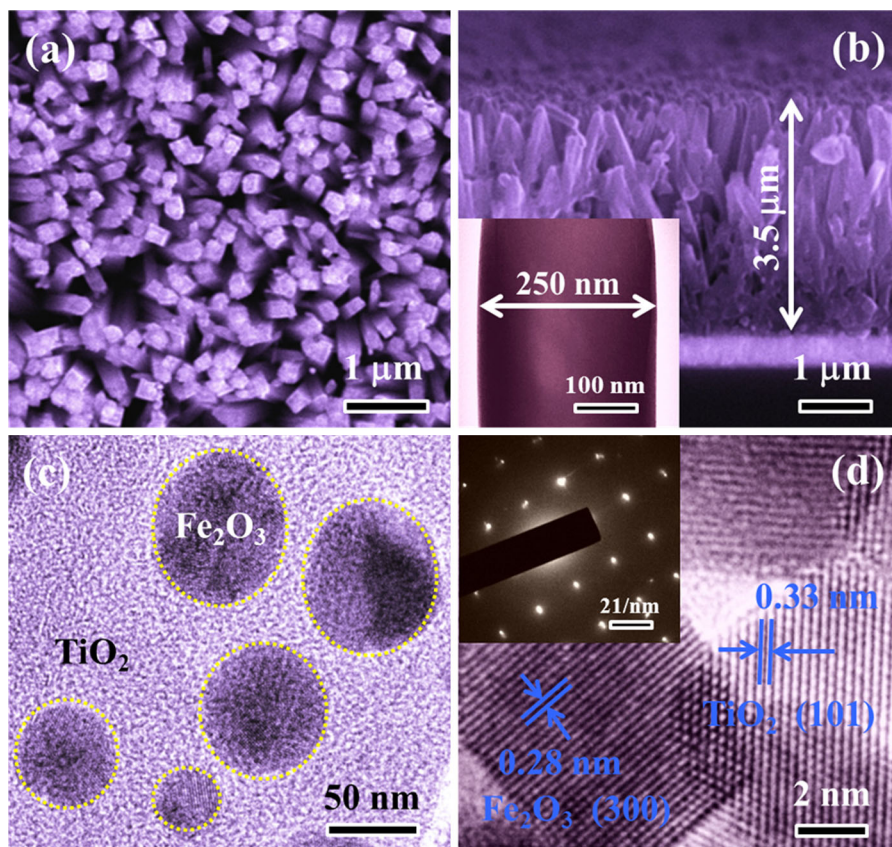


Fig. 3 **a, b** Scanning electron microscopy (SEM) image of $\text{TiO}_2/\alpha\text{-Fe}_2\text{O}_3$ composite nanorods grown on FTO substrate, the inset shows a single nanorod. **c** TEM image of individual $\text{TiO}_2/\alpha\text{-Fe}_2\text{O}_3$ composite nanorod. **d** The HRTEM image of $\text{TiO}_2/\alpha\text{-Fe}_2\text{O}_3$

$\alpha\text{-Fe}_2\text{O}_3$ composite nanorods. The fringes with a spacing of 0.33 nm correspond to (101) planes of TiO_2 , and the fringes with a spacing of 0.28 nm correspond to (300) planes of $\alpha\text{-Fe}_2\text{O}_3$. The inset is the corresponding SAED pattern

white-light illumination with various power densities at room temperature, which all exhibit asymmetric behavior with significant hysteresis. The arrows in the figure denote the sweeping direction of voltage. The asymmetric behavior of I – V curve demonstrates that a Schottky barrier is formed at the interface between TiO_2 and FTO, which was studied in previous work (Yang et al. 2014). The obvious bipolar resistive-switching behaviors are observed, which exhibit the rapid conversion and good reproducibility. The bipolar resistive-switching effect should result from the trapped and detrapped charge in the Schottky-like depletion layer (Ungureanu et al. 2012; Jang et al. 2006; Won et al. 2008; Jeong et al. 2007; Park et al. 2010). The inset of (Fig. 4a) shows a large resistance switching effects. The resistive-switching phenomenon becomes more obvious with the white-light power density increasing from 50 to 200 mW/cm^2 .

Therefore, the resistive-switching effect can be controlled by white-light with various power densities at room temperature, which demonstrates that $\text{TiO}_2/\alpha\text{-Fe}_2\text{O}_3$ nanorod array is a potential candidate for multilevel light-controlled memory applications. It is elucidated that the resistance switching behavior is activated by the modulation of trapped electrons in the active layer under illumination conditions, as the total number of electrons is increased by joining the photogenerated current (Ungureanu et al. 2012; Park et al. 2012, 2013; Bera et al. 2013).

To evaluate the resistive-switching characteristics of $\text{TiO}_2/\alpha\text{-Fe}_2\text{O}_3$ composite nanorods array, the resistance–time curves in the dark and under white-light illumination with a positive bias of 50 mV are tested and shown in Fig. 4b. It is obvious that the resistance is about 40 $\text{M}\Omega$ in the dark and 100 $\text{k}\Omega$ under white-light illumination with power density of 50 mW/cm^2 ,

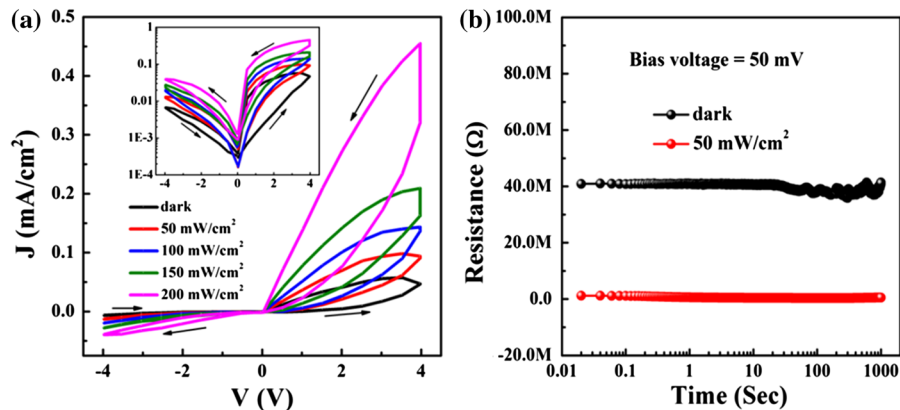


Fig. 4 **a** Current–Voltage curves in the dark and under white-light illumination with various power density at room temperature. The inset is the resistance switching effects, **b** Resistance–Time curves in the dark and under white-light illumination

and the resistive-switching ratio is up to approximately three orders of magnitude. This adequately illustrates that the current density is greatly changed by white-light illumination. According to the above results, the steady light-controlled resistive-switching behavior in $\text{TiO}_2/\alpha\text{-Fe}_2\text{O}_3$ composite nanorods provides the potential for nonvolatile light-controlled memory applications.

Conclusion

In this article, $\text{TiO}_2/\alpha\text{-Fe}_2\text{O}_3$ composite nanorods array with the resistance switching properties has been fabricated. These composite nanorods exhibit white-light-controlled resistance switching effect at room temperature. This work will shed light on the application of oxide composite materials in memory devices.

Acknowledgments This work was supported by the National Science Foundation of China (Grant No. 51372209).

References

- Adachi M et al (2012) Shape control of highly crystallized titania nanorods based on formation mechanism. *J Mater Res* 27(2):431–439. doi:10.1557/jmr.2011.353
- Bean CP, Livingston JD (1959) The anisotropy of very small cobalt particles. *J Appl Phys* 20(2–3):298–302. doi:10.1051/jphysrad:01959002002-3029800
- Beermann N, Vayssieres L, Linquist ES, Hagfeldt A (2002) Synthesis of $\text{Fe}_2\text{O}_3/\text{TiO}_2$ nanorod–nanotube arrays by filling TiO_2 nanotubes with Fe. *J Electrochem Soc* 147(24):56–61. doi:10.1088/0957-4484/19/31/315601
- Bera A, Peng H, Lourembam J, Shen Y, Sun XW, Wu T (2013) A versatile light-switchable nanorod memory: wurzite ZnO on perovskite SrTiO_3 . *Adv Funct Mater* 23:4977–4984. doi:10.1002/adfm.201300509
- Beydoun D, Amal R, Low G, McEvoy S (1999) Role of nanoparticles in photocatalysis. *J Nanopart Res* 1(4):439–458. doi:10.1023/A:1010044830871
- Burschka J et al (2013) Sequential deposition as a route to high-performance perovskite-sensitized solar cells. *Nature* 499:316. doi:10.1038/nature12340
- Cao CB, Li JL, Wang X, Song XP, Sun ZQ (2011) Current characterization and growth mechanism of anodic titania nanotube arrays. *J Mater Res* 26:437. doi:10.1557/jmr.2010.33
- Chen J, Xu L, Li WY, Gou X (2005) Alpha- Fe_2O_3 nanotubes in gas sensor and lithium-ion battery applications. *Adv Mater* 17(5):582–586. doi:10.1002/adma.200401101
- Chu D, Yuan X, Qin G, Xu M, Zheng P, Lu J, Zha L (2008) Efficient carbon-doped nanostructured TiO_2 (anatase) film for photoelectrochemical solar cells. *J Nanopart Res* 10(2):357–363. doi:10.1007/s11051-007-9241-7
- Crossland Edward JW et al (2013) Mesoporous TiO_2 single crystals delivering enhanced mobility and optoelectronic device performance. *Nature* 495:215–219. doi:10.1038/nature11936
- Diwald O, Thompson TL, Zubkov T, Goralski EG, Walck SD, Yates JT Jr (2004) Photochemical activity of nitrogen-doped rutile TiO_2 (110) in visible light. *J Chem Phys* 108:6004–6008. doi:10.1021/jp031267y
- Faust BC, Hoffmann MR, Bahnemann DW (1989) Photocatalytic oxidation of sulfur-dioxide in aqueous suspensions of alpha- Fe_2O_3 . *J Chem Phys* 93(17):6371–6381. doi:10.1021/j100354a021
- Gratzel M (2003) Applied physics-solar cells to dye for. *Nature* 421:586–587. doi:10.1038/421586a
- Hamaguchi M, Aoyama K, Asanuma S, Uesu Y, Katsufuji T (2006) Electric-field-induced resistance switching universally observed in transition-metal-oxide thin films. *Appl Phys Lett* 88(14):142508. doi:10.1063/1.2193328
- Hasan M, Dong R, Choi HJ, Lee DS, Seong DJ, Pyun MB, Hwang H (2008) Uniform resistive switching with a thin reactive metal interface layer in metal-La(0.7)Ca(0.3)MnO(3)-metal

- heterostructures. *Appl Phys Lett* 92(20):202102. doi:[10.1063/1.2932148](https://doi.org/10.1063/1.2932148)
- He CL et al (2009) Nonvolatile resistive switching in graphene oxide thin films. *Appl Phys Lett* 95(23):232101. doi:[10.1063/1.3271177](https://doi.org/10.1063/1.3271177)
- Hodes G (2013) Perovskite-based solar cells. *Science* 342:317–318. doi:[10.1126/science.1245473](https://doi.org/10.1126/science.1245473)
- Hyeon T (2003) Chemical synthesis of magnetic nanoparticles. *Chem Commun* 8:927–934. doi:[10.1039/b207789b](https://doi.org/10.1039/b207789b)
- Jang HD, Kim S-K, Kim S-J (2001) Effect of particle size and phase composition of titanium dioxide nanoparticles on the photocatalytic properties. *J Nanopart Res* 3(2–3):141–147. doi:[10.1023/A:1017948330363](https://doi.org/10.1023/A:1017948330363)
- Jang JS, Kim D, Seong TY (2006) Schottky barrier characteristics of Pt contacts to n-type InGaN. *J Appl Phys* 99(7):073704. doi:[10.1063/1.2187274](https://doi.org/10.1063/1.2187274)
- Jeong DS, Schroeder H, Waser R (2007) Coexistence of bipolar and unipolar resistive switching behaviors in a Pt/TiO₂/Pt stack. *Electrochem Solid-State Lett* 10(8):G51–G53. doi:[10.1149/1.2742989](https://doi.org/10.1149/1.2742989)
- Jo SH, Kim KH, Lu W (2009) Programmable resistance switching in nanoscale two-terminal devices. *Nano Lett* 9(1):496–500. doi:[10.1021/nl803669s](https://doi.org/10.1021/nl803669s)
- Khan SUM, Akikusa J (1999) Efficient photochemical water splitting by a chemically modified n-TiO₂. *J Chem Phys* 103(718):4–9. doi:[10.1126/science.1075035](https://doi.org/10.1126/science.1075035)
- Li YB, Sinitskii A, Tour JM (2008) Electronic two-terminal bistable graphitic memories. *Nat Mater* 7(12):966–971. doi:[10.1038/nmat2331](https://doi.org/10.1038/nmat2331)
- Liu B, Aydil ES (2009) Growth of oriented single-crystalline rutile TiO₂ nanorods on transparent conducting substrates for dye-sensitized solar cells. *J Am Chem Soc* 131:3985–3990. doi:[10.1021/ja8078972](https://doi.org/10.1021/ja8078972)
- Liu H, Gao L (2006) Preparation and properties of nanocrystalline alpha-Fe₂O₃-sensitized TiO₂ nanosheets as a visible light photocatalyst. *J Am Ceram Soc* 89(1):370–373. doi:[10.1111/j.1551-2916.2005.00686.x](https://doi.org/10.1111/j.1551-2916.2005.00686.x)
- Liu SQ, Wu NJ, Ignatiev A (2000) Electric-pulse-induced reversible resistance change effect in magnetoresistive films. *Appl Phys Lett* 76(19):2749–2751. doi:[10.1063/1.126464](https://doi.org/10.1063/1.126464)
- Liu M, Abid Z, Wang W, He XL, Liu Q, Guan WH (2009) Multilevel resistive switching with ionic and metallic filaments. *Appl Phys Lett* 94(23):233106. doi:[10.1063/1.3151822](https://doi.org/10.1063/1.3151822)
- Ma LP, Pyo S, Ouyang J, Xu Q, Yang Y (2003) Nonvolatile electrical bistability of organic/metal-nanocluster/organic system. *Appl Phys Lett* 82(9):1419–1421. doi:[10.1063/1.1556555](https://doi.org/10.1063/1.1556555)
- Meijer GI (2008) Materials science—who wins the nonvolatile memory race? *Science* 319:1625–1626. doi:[10.1126/science.1153909](https://doi.org/10.1126/science.1153909)
- Mitchinson A (2008) Materials science—solar cells go round the bend. *Nature* 455:744. doi:[10.1038/455744a](https://doi.org/10.1038/455744a)
- Park WY, Kim GH, Seok JY, Kim KM, Song SJ, Lee MH, Hwang CS (2010) A Pt/TiO₂/Ti Schottky-type selection diode for alleviating the sneak current in resistance switching memory arrays. *Nanotechnology* 21(195201):4. doi:[10.1088/0957-4484/21/19/195201](https://doi.org/10.1088/0957-4484/21/19/195201)
- Park J, Lee S, Yong K (2012) Photo-stimulated resistive switching of ZnO nanorods. *Nanotechnology* 23:385707. doi:[10.1088/0957-4484/23/38/385707](https://doi.org/10.1088/0957-4484/23/38/385707)
- Park J, Lee S, Lee J, Yong K (2013) A light incident angle switchable ZnO nanorod memristor: reversible switching behavior between two non-volatile memory devices. *Adv Mater* 25:6423–6429. doi:[10.1002/adma.201303017](https://doi.org/10.1002/adma.201303017)
- Schaub R, Thostrup P, Lopez N, Lægsgaard E, Stensgaard I, Nørskov JK, Besenbacher F (2001) Oxygen vacancies as active sites for water dissociation on rutile TiO₂ (110). *Phys Rev Lett* 87(26):266104. doi:[10.1103/PhysRevLett.87.266104](https://doi.org/10.1103/PhysRevLett.87.266104)
- Schindler C, Staikov G, Waser R (2009) Electrode kinetics of Cu-SiO₂-based resistive switching cells: overcoming the voltage-time dilemma of electrochemical metallization memories. *Appl Phys Lett* 94(7):072109. doi:[10.1063/1.3077310](https://doi.org/10.1063/1.3077310)
- Stewart DR, Chen Y, Williams RS, Jeppesen JO, Nielsen KA, Stoddart F (2004) Molecule-independent electrical switching in Pt/organic monolayer/Ti devices Molecule-independent electrical switching in Pt/organic monolayer/Ti devices. *Nano Lett* 4(1):133–136. doi:[10.1021/nl034795u](https://doi.org/10.1021/nl034795u)
- Ungureanu M, Zazpe R, Golmar F, Stoliar P, Llopis R, Casanova F, Hueso LE (2012) A light-controlled resistive switching memory. *Adv Mater* 24(18):2496–2500. doi:[10.1002/adma.201200382](https://doi.org/10.1002/adma.201200382)
- Wang Y, Zhang L, Deng K, Chen X, Zou Z (2007) Low temperature synthesis and photocatalytic activity of rutile TiO₂ nanorod superstructures. *J Phys Chem C* 111:2709–2714. doi:[10.1021/jp066519k](https://doi.org/10.1021/jp066519k)
- Watson S, Beydoun D, Scott J, Amal R (2004) Preparation of nanosized crystalline TiO₂ particles at low temperature for photocatalysis. *J Nanopart Res* 6(2–3):193–207. doi:[10.1023/B:NANO.0000034623.33083.71](https://doi.org/10.1023/B:NANO.0000034623.33083.71)
- Won S, Go S, Lee K, Lee J (2008) Resistive switching properties of Pt/TiO₂/n⁺-Si ReRAM for nonvolatile memory application. *Electron Mater Lett* 4(1):29–33
- Wu JS et al (2011) Growth of rutile TiO₂ nanorods on anatase TiO₂ thin films on Si-based substrates. *J Mater Res* 26:1646. doi:[10.1557/jmr.2011.190](https://doi.org/10.1557/jmr.2011.190)
- Yang M, Ding B, Lee J-K (2014) Surface electrochemical properties of niobium-doped titanium dioxide nanorods and their effect on carrier collection efficiency of dye sensitized solar cells. *J Power Sources* 245:301–307. doi:[10.1016/j.jpowsour.2013.06.016](https://doi.org/10.1016/j.jpowsour.2013.06.016)
- Zhuge F, Dai W, He CL, Wang AY, Liu YW, Li M, Wu YH, Cui P, Li RW (2010) Nonvolatile resistive switching memory based on amorphous carbon. *Appl Phys Lett* 96(16):163505. doi:[10.1063/1.3406121](https://doi.org/10.1063/1.3406121)