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Uniform Fe₃O₄–PANi/PS composite spheres with conductive and magnetic properties and their hollow spheres

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Abstract Core-shell multifunctional composite spheres consisting of Fe₃O₄-polyaniline (PANi) shell and polystyrene (PS) core were fabricated using core-shell-structured sulfonated PS spheres (with uniform diameter of 250 nm) as templates. PANi was doped in situ by sulfonic acid resulting the composite spheres are well conductive. Dissolved with solvent. PS cores were removed from the coreshell composite spheres and hollow Fe₃O₄-PANi spheres were obtained. Removing the PANi and PS components by calcinations produced hollow Fe₃O₄ spheres. The cavity size of the hollow spheres was uniformly approximate to 190 nm and the shell thickness was 30 nm. The cavity size and the shell thickness can be synchronously controlled by varying the sulfonation time of the PS templates. The shell

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Z. He Tianjin Medical College, Tianjin 344000, People's Republic of China thickness in size range was of 20–86 nm when the sulfonation time was changed from 1 to 4 h. These resulting spheres could be arranged in order by self-assembly of the templates. Both the Fe_3O_4 –PANi/PS composite spheres and the hollow Fe_3O_4 spheres exhibit a super-paramagnetic behavior. Scanning electron microscopy, transmission electron microscopy, Fourier transform infrared spectroscopy, and X-ray powder scattering were used to characterize these as-prepared spheres.

Keywords Hollow spheres · Composite spheres · Polyaniline · Multifunction · Super-paramagnetic · Nanomaterial

Introduction

Nano-engineering of particle surfaces and hollow spheres has received considerable scientific and technological interest in recent years (Caruso 2000, 2001, 2003). The fabricated core–shell particles and hollow spheres could find many useful applications in catalysis, optics, electronics, drug delivery, and biosensing. Among all of the synthesis protocols, the LbL-colloidal templating approach has been demonstrated to be extremely effective in the fabrication of various uniform core–shell particles and hollow spheres (Caruso 2002). More and more interests are focused on the preparation and study of core–shell colloids mainly because these particles have the novel and enhanced properties, such as electrical, magnetic, mechanical, catalytic, and optical.

At the same time, multifunctionality of micro- or nanospheres of conducting polymers is an important subject in the field of material science. Conducting polymers have received especial attention because of their excellent electronic properties, with tunable conductivities covering the whole range from insulator to metal, mechanical properties, and the processing advantages of polymers (Hohnholz et al. 2005). Various approaches have been designed to realize the multifunctionality of micro- or nanostructured conducting polymers in order to match the requirements of applications in optics, electronics, mechanics, membranes, protective coatings, catalysis, sensors, biology, and others (e.g., Arshady et al. 1999; Nadian and Lindblom 2002; Xu et al. 2005; Smith et al. 2003; Cao et al. 2001; Porter et al. 1997; Zhang et al. 2006). Moreover, magnetic nanoparticles also have attracted an increasing interest in the fields of basic science and technological applications in electronic, optoelectronic, and spintronic devices (Fannin et al. 2002; Soeya et al. 2002; Sorenson et al. 2002). With this increased interest comes the realization that the shape of the particles takes a center stage in determining their magnetic properties. Much attention has been paid to the fabrication of polymer microspheres coated with magnetic nanoparticles, which may find application in diverse areas, ranging from photonics to fillers and pigments to microencapsulation (Caruso 2001; Tartaj et al. 2003). However, composite materials made from magnetic nanosize particles and conducting polymers are still the most interesting and challenging areas of research in recent times. The fabrication of uniform composite spheres and hollow spheres with both conductive and magnetic properties has rarely been reported so far.

Very recently, sulfonated polystyrene microspheres have been used as template to produce hollow polyaniline and hollow polypyrrole microspheres. The template particles were prepared by simply sulfonation of PS microspheres. Aniline or pyrrole was oxidatively polymerized on the surface of sulfonated PS (Yang et al. 2005). In this article, we report an extension of this chemical modification method to prepare conductively magnetic Fe_3O_4 – PANi/PS composite core–shell spheres and their corresponding hollow spheres. The shell of the as-prepared composite magnetic sphere is Fe_3O_4 -PANi, and the core consists of polymer PS. The super-paramagnetic properties of these nanoparticles are of great interest for biomedical applications. And the conducting polymers have good tunable conductivity and chemically active surface. Nanocomposites made from magnetic nanoparticles and conducting polymers possess the good properties of both of them. Such multifunctionality spheres may find theoretic and practical application in electronics and magnetism. Furthermore, the resulting spheres could be arranged in order by self-assembly of the templates. The hollow magnetic sphere arrays have uniform cavity size and shell thickness, which were difficultly reachable by other methods.

Experimental section

Preparation of sulfonated PS template spheres

Monodisperse polystyrene particles with uniform diameter of approximately 250 nm were prepared by seed-emulsion polymerization. The PS spheres were immersed in concentrated sulfuric acid and stirred at 45 °C for 2 h. After the products of sulfonated PS spheres were repeatedly washed with ethanol and water, they were separated by centrifugation at 8,000 r/min. To control the shell thickness of sulfonated PS spheres, the reaction time was varied from 1 to 4 h.

Preparation of PANi/PS composite spheres

After 1.0 g of sulfonated PS template spheres was dispersed in water, 0.279 g of aniline (Beijing Chem. Co., distilled under reduced pressure before using) was added with stirring for 30 min; then 0.03 mol of ammonium persulfate ($(NH_4)_2S_2O_8$, APS, A.R., Beijing 3rd Chemical Reagents Factory) was added, followed by an oxidative polymerization at room temperature for 24 h. The green products of PANi/PS composite spheres were purified by centrifugation and dried under vacuum at ambient temperature.

Preparation of Fe₃O₄-PANi/PS composite spheres

Of freeze-dried PANi/PS composite spheres, 0.5 g was immersed in 20 mL ammonium hydroxide (2 M) for 12 h. It was observed that the color of the PANi/PS

spheres turned to dark blue from green. After saturating with ammonium hydroxide, the PANi/PS spheres were re-dispersed in 50 mL water; 3 mL aqueous mixture of ferrous chloride (0.5 M)/ferric chloride (1 M) (vol:vol = 1:1) were added with stirring under a nitrogen atmosphere for another 2 h. The product was kept for another 24 h and then isolated by centrifugation (6000 r/min).

Preparation of hollow spheres

Two common methods were used to prepare hollow spheres: (1) the PS cores were dissolved by solvent extraction using *N*,*N*-dimethylformamide (DMF) to produce hollow PANi/PS spheres and (2) the Fe₃O₄/ PANi/PS composite spheres were calcinated at 450 °C for 3 h to yield hollow Fe₃O₄ spheres.

Characterization

Very dilute dispersion of the products in ethanol was spread on carbon-coated copper grids for transmission electron microscopy (TEM) characterization (JEOL 100CX operating at 100 kV). Scanning electron microscopy (SEM) measurements were performed with a HITACHI S-4300 instrument operated at an accelerating voltage of 15 kV. The samples were ambient dried and vacuum sputtered with Pt about average size 3 nm. The infrared spectra were recorded by a BRUKER EQUINOX 55 FT-IR using KBr disks. A Wide-angle X-ray powder scattering (Rigaku D/max-2500) was used to characterize the crystalline phase of these materials. A Perkin-Elmer TGA 7 was used to determine the inorganic content of the composite spheres. Elemental analysis was carried out using a Flash EA-1112 apparatus. A vibratingsample magnetometer (VSM JDM-13) was used to characterize the magnetic property of the magnetite spheres. A Keithley 196 System DMM digital multimeter was used to measure the electrical conductivity of the compressed pieces of dry powders using a standard four-probe method at room temperature.

Results and discussion

As shown in Scheme 1, the core-shell sulfonated PS spheres were prepared by sulfonation of the PS spheres with concentrated sulfuric acid. The sulfonation process occurred synchronously through the whole PS spheres' surfaces, ensuring the sulfonation thickness was uniform. The thickness of the sulfonated PS spheres is tunable by varying the reaction time and temperature. In this article, the reaction temperature was at 45 °C and the sulfonation time was 2 h, ensuring a suitable sulfonation thickness. The hydrophility of the PS spheres' surfaces increasing as the sulfonation processing, and thus the sulfonated spheres were easily dispersed in water. As the sulfonic acid is a good dopant for aniline polymerization, aniline monomer can be easily absorbed in the sulfonated shell and in situ polymerize therein, resulting in the formation of PANi/PS composite spheres. The sulfonic acid group and the PANi (Zhang and Wan 2003) in the shell can be favorably bind metal ions by ion exchange facilitating formation of magnetite nanocrystallite, resulting in the formation of Fe₃O₄–PANi/PS composite spheres. After removing the PS cores by solvent extraction using DMF, composite hollow Fe₃O₄-PANi





Fig. 1 Morphologies of the sulfonated PS template spheres: (a) SEM; (b) TEM

spheres were produced. On the condition that the Fe_3O_4 -PANi/PS composite spheres were calcinated at high temperature, all the organic components were removed, resulting in the formation of inorganic hollow Fe_3O_4 spheres.

Figure 1a shows a typical SEM image of the sulfonated PS spheres. The diameters of these spheres were determined to be uniformly approximate to 250 nm from the microphotograph. The corresponding TEM image of the sulfonated PS spheres (Fig. 1b), with distinct difference from that of the mother template PS spheres (Fig. s1), reveals these spheres were core-shell structure. The thickness of the shell and the diameter of the core are approximately 30 and 190 nm, respectively. The existence of sulfonic group was confirmed by FTIR spectra. As shown in Fig. s2, the characteristic peaks of 1008, 1042, and 1183 cm^{-1} are attributed to the vibration of sulfonic group. Anline monomer can be preferentially absorbed in the shell because of its electric charge. Furthermore, the absorbed aniline is prior to polymerizing in the shell since the sulfonic acid is a good dopant. After the aniline monomer was absorbed and polymerized in the gel shell, PANi/PS composite capsules formed.

The chemical structure of PANi was confirmed by FTIR spectra (Fig. s2). The characteristic peaks



Fig. 2 SEM images of the PANi/PS spheres (a) and the Fe₃O₄–PANi/PS composite spheres (b)

around 1,304 and 1,585 cm⁻¹ are attributed to the stretching modes of C–N and C=N bonds, respectively. This indicates the presence of leucoemeraldine and pernigraniline state. As shown in Fig. 2a, PANi was clearly found on the surface of the spheres. The spheres were connected to each other forming coalescent necks, and they were arranged in order by self-assembly. Elemental analysis indicated the content of PANi was 21%. The dry powder of the PANi/PS capsules posses a conductivity of 9.1×10^{-2} S cm⁻¹ at room temperature, indicating the PANi was continuous in the gel shell.

The PANi/PS composite spheres were then used for preparation of Fe₃O₄-PANi/PS multifunctional spheres. The PANi/PS spheres turned blue from black green when immersed in ammonium hydroxide. This observed phenomenon indicates the PANi was de-doped by ammonium hydroxide. Along with the adding of aqueous mixture of ferrous chloride (0.5 M)/ferric chloride (1 M) (vol:vol = 1:1), the target precursor is being absorbed in the spheres' shell and forming iron oxide therein. As shown in Fig. 2b, the resulting composite spheres have rough surface consisting of 10-20 nm nanoparticles. The conductivity of Fe₃O₄-PANi/PS composite spheres is $3.61 \times 10^{-4} \text{ S cm}^{-1}$, very lower than that of the PANi/PS spheres (0.091 S cm^{-1}), which was mainly caused by the de-doping process of ammonium hydroxide and the potential blocking of conductive path by the Fe₃O₄ nanoparticles (Zhang and Wan 2003).

In order to resume the conductivity of the composite spheres, these spheres were immersed in sulfonic acid with low concentration $(1 \times 10^{-3} \text{ mol L}^{-1})$ and treated for 10 min. Following the redoping process of protonic acid, the conductivity of Fe₃O₄–PANi/PS composite capsules enhanced about 10 times. The morphology and the magnetism of the composite spheres have no distinct influence in that the treat time was short and the acid concentration was very low.

When dissolved the PS core from the composite spheres by DMF, composite hollow Fe_3O_4 -PANi spheres formed. Mono-disperse hollow sphere could be obtained by ultrasonic dispersion. The typical TEM image of a mono-disperse hollow Fe_3O_4 -PANi sphere (Fig. 3a) shows its shell thickness was about 30 nm; the PANi and Fe_3O_4 was interconnected. However, if both the PS and PANi were removed



Fig. 3 TEM images of representative hollow spheres: (a) hollow Fe_3O_4 -PANi, the templates were sulfonated for 2 h; (b) hollow Fe_3O_4 , the templates were sulfonated for 2 h;

from the composite spheres by calcinations, hollow inorganic Fe_3O_4 spheres were prepared. Figure 3b is a corresponding typical TEM image of mono-disperse hollow Fe_3O_4 sphere. The thickness of the shell has no obvious change from that of Fe_3O_4 –PANi indicated Fe_3O_4 was overall in the shell of the composite Fe_3O_4 –PANi/PS spheres. Mesopores were clearly observed due to the removing of organic components. These mesopores are very important for application of such mesoporous hollow sphere materials in combining other components into magnetic composite materials.

Furthermore, it is very attractive that the shell thickness of the spheres could be tuned by control the sulfonation time of PS. For example, when the sulfonation time of PS was reduced to 1 h, the shell thickness of the hollow Fe₃O₄ decreased to about 20 nm (Fig. 3c). However, if prolonged the sulfonation time of PS to 4 h, the shell thickness was increased to about 82 nm, and the cavity diameter was 86 nm (Fig. 3d). As a result, the conductivity of the Fe₃O₄-PANi/PS composite spheres deriving from 4 h sulfonated templates increased to 8.23×10^{-4} S cm⁻¹. The series of experiments by varying the sulfonation time demonstrate that we can obtain monodisperse core-shell gel particles with tunable gel thickness and controllable core size. This provided a way to adjust the conductivity of these spheres through varying the PANi content.

As the colloidal crystals could be formed by the self-assembly of mono-disperse particles (Bartlett et al. 2001), PANi/PS spheres could be arranged in ordered arrays (Fig. 2a). Ordered arrays of these spheres were prepared by drying the capsule dispersion in an ethanol/water mixture. After the removal of the polystyrene cores, either by dissolution or by calcination, the ordered structure was retained. Figure 4 shows the ordered arrays of the hollow Fe_3O_4 spheres. The ordered structure possesses

(c) hollow Fe₃O₄, the templates were sulfonated for 1 h; (d) hollow Fe₃O₄–PANi, the templates were sulfonated for 4 h



Fig. 4 Morphologies of the ordered arrays for hollow Fe_3O_4 spheres: (a) SEM; (b) TEM

promising applications in information storage and photonic crystals (Vickreva et al. 2000; Göltner 1999).

Powder X-ray diffraction patterns of the hollow Fe_3O_4 spheres and the hollow Fe_3O_4 /PANi are shown in Fig. 5. The diffraction peaks of the magnetic particles synthesized in this study were measured to be $2\theta = 30.19$, 35.50, 43.14, 53.58, 57.08, 62.83, 74.41. These data suggested the magnetic particles are pure Fe_3O_4 crystalline without any impurity phases (Wang et al. 2004). Obviously, the broad peak is due to PANi and the sharp peaks are attributed to Fe_3O_4 . No distinct peak shift or additional peak is found. That suggests there is no strong interaction between PANi and Fe_3O_4 in the composite hollow Fe_3O_4 /PANi spheres.

Figure 6 shows the dependence of the magnetization on the applied magnetic fields for hollow Fe_3O_4 and PANi–Fe₃O₄ (20 wt%) spheres at room temperature. In case of hollow Fe₃O₄ spheres, saturated magnetization (Ms) were measured to be 46.2 emu/g. No hysteresis loop (i.e., remnant magnetization, Mr = 0 and coercive force, Hc = 0) was observed, indicating a super-paramagnetic behavior. As for hollow PANi/Fe₃O₄ spheres, Ms = 10, Mr = 0 and Hc = 0, respectively, indicating a super-paramagnetic behavior, too. And its super-paramagnetic behavior is similar to that of hollow Fe₃O₄ spheres. So, it is reasonable to believe that Fe₃O₄ in the Fe₃O₄/

Fig. 5 X-ray scatter patterns of (a) hollow Fe₃O₄ spheres; (b) hollow Fe₃O₄-PANi spheres



20

40

 2θ (degrees)

0

-500

533

80

70

J Nanopart Res (2009) 11:923-929

533

80

60



311

40

50

 2θ (degrees)

60

2000

1500

1000 220

500

0

30

Intensity (a. u.)

Fig. 6 The dependence of the magnetization on the applied magnetic fields for (a) hollow Fe₃O₄ spheres; (b) Fe₃O₄-PANi/ PS composite spheres

PANi/PS composite spheres contribute to the superparamagnetic behavior. Furthermore, the Ms of the hollow PANi/Fe₃O₄ spheres is much lower than that of the Fe₃O₄ hollow spheres. It is because the content of the Fe_3O_4 in the composite spheres is wt 20%; nevertheless, there is no other component in the hollow Fe₃O₄ spheres. It is rational to believe that we can adjust the Ms by changing the content of Fe₃O₄ in the hollow PANi/Fe₃O₄ spheres, just as to change the conductivity of these spheres by varying the content of PANi.

Conclusions

Fe₃O₄-PANi/PS multifunctional composite spheres and their corresponding hollow spheres with uniform cavity size (190 nm) and shell thickness (30 nm) are prepared using core-shell-structured sulfonated PS spheres as templates. The cavity size and the shell thickness can be synchronously controlled by varying the sulfonation time of the PS templates. The shell thickness in size was of 20 nm when decreasing the sulfonation time to 1 h, and it was 86 nm when the sulfonation time was increased to 4 h. Conducting PANi was in situ doped without external additives, and thus the composite spheres show good conductivity. The composite and hollow spheres exhibit a super-paramagnetic behavior (i.e., no hysteresis loop). Ordered arrays of these spheres were successfully prepared by self-assembly. The template method is simple and effective. It can be extended to prepare other composite spheres and hollow spheres with varied composition. The multifunctionality spheres and hollow spheres may find many applications such as in electronics, controlled delivery, and microwave absorption.

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