EFFECT OF TEMPERATURE AND PRESSURE ON CORROSION AND HYDROGENATION OF STEEL IN CHLORIDE-ACETATE ENVIRONMENT WITH DIFFERENT CONCENTRATIONS OF HYDROGEN SULFIDE AND CARBON DIOXIDE

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Low-carbon pipe steel's corrosion rate and hydrogenation in a chloride-acetate solution with CO_2/H_2S mixtures at different temperatures and pressures were investigated. The rate of corrosion and hydrogenation depend on the H_2S concentration and the properties of corrosion films. In a solution with a pressure ratio of $P_{CO_2}: P_{H_2S} = 30:1$ and at the beginning of exposure at a pressure ratio of 3:1 corrosion slows down due to the formation of $Fe_{1+x}S$ mackinawite film. Over time, mackinawite transforms into hexagonal FeS troilite with an acicular structure, and the corrosion rate increases in approx. 2 times. The corrosion rate of steel at 60°C and 5 MPa in a solution with a pressure ratio of $P_{CO_2}: P_{H_2S} = 3:1$ was twice as low as at 20°C and 0.1 MPa. A dense layer of cubic iron sulfide FeS crystals was formed on the surface, which reduces corrosion. The adsorption of hydrogen by the steel reduced in approx. 15 times as the temperature increases.

Keywords: corrosion rate, hydrogenation, 17G1S-U steel, chloride-acetate solution, hydrogen sulfide, carbon dioxide, temperature, pressure.

Introduction

Natural gas from gas-bearing stratums, accompanying gases, and gases from gas-condensate fields contain carbon dioxide and hydrogen sulfide impurities, which are corrosive-active in the presence of moisture. Dissolved CO₂ accelerates cathodic reactions and affects the mechanism of steel hydrogenation. Hydrogen sulfide causes corrosion and steel hydrogenation, which is a precondition for their accelerated corrosionmechanical fracture [1–4]. In the presence of CO₂ and H₂S in the environment, films containing iron carbonates and sulfides are formed on the steel surface. Depending on the chemical composition and density, films can have different effects on corrosion, hydrogen absorption, and embrittlement of steels [5–7].

The purpose of the study is to investigate the effect of H_2S and CO_2 concentration in a chloride-acetate solution on the corrosion rate and hydrogenation of 17G1S-U steel at different temperatures and pressures.

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Research Method

Ferrite-pearlite 17G1S-U steel was studied in a solution of 5% NaCl + 0.5% CH₃COOH, pH 2.7 [2]. CO₂ or a mixture of CO₂ with H₂S continuously passing through this solution at a ratio of partial pressures of 30:1 and 3:1 to obtain the hydrogen sulfide concentration in the solution of 100 and 500 mg/dm³, respectively. Tests were carried out at a temperature of 20°C and a pressure of 0.1 MPa for 720 h and in an autoclave at 60°C and 5 MPa for 510 h (this exposure was chosen to ensure the stability of the environment's chemical composition). The corrosion rate $K_m \left(g/(m^2 \times h)\right)$ was calculated according to the formula $K_m = (\Delta m)/(S \times \tau)$, where Δm is the change in sample weight before and after the experiment; S is its area; τ is the exposure time to a corrosive environment. The hydrogen concentration in the samples was determined using a LECO DH 603 analyzer. The microstructure, chemical composition, and thickness of corrosion products on steel were studied on an EVO 40XVP scanning electron microscope with a micro-X-ray spectral analysis system, using an INCA ENERGY 350 energy dispersive spectrometer.

Results and Discussion

At a temperature of 20°C and a pressure of 0.1 MPa, the initial corrosion rate of steel is the lowest in a solution saturated with $CO_2 \left(0.4 \text{ g/}(\text{m}^2 \times \text{h}) \right)$. In the presence of hydrogen sulfide in the environment it is 2.3–3 times higher (Fig. 1). The corrosion rate slows down over time in the solution containing CO_2 and H_2S at a ratio of partial pressures of 30:1 and is halved after 720 h. A slight slowdown of corrosion is observed at the beginning of exposure to the solution saturated with carbon dioxide, as well as to the solution with the ratio $P_{CO_2}: P_{H_2S} = 3:1$, but after 240–300 h corrosion in these environments increases, reaching 0.79 and 2.16 $g/(m^2 \times h)$ after 720 h, respectively (Fig. 1*a*).

The concentration of absorbed hydrogen does not exceed 0.8 ppm after steel exposure to a solution saturated with carbon dioxide (Fig. 1*b*). Hydrogen absorption is approx. 4 ppm at the ratio of partial pressures $P_{CO_2}: P_{H_2S} = 30:1$. The hydrogen content in steel reaches approx. 7.9 ppm with an increase in the concentration of hydrogen sulfide ($P_{CO_2}: P_{H_2S} = 3:1$).

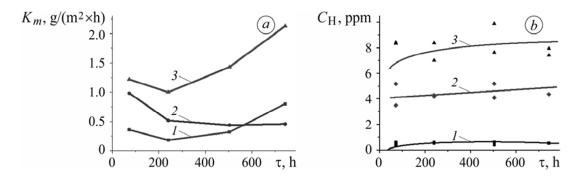


Fig. 1. Influence of H_2S and CO_2 on (a) corrosion rate and (b) hydrogenation of 17G1S-U steel in a chloride-acetate solution at a temperature of 20°C and a pressure of 0.1 MPa: (l) CO_2 (saturated); (2) $P_{CO_2} : P_{H_2S} = 30:1$; (3) $P_{CO_2} : P_{H_2S} = 3:1$.

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Iron carbonates are not formed in a chloride-acetate solution with pH 2.7, saturated with carbon dioxide. The corrosion rate increases over time due to the formation of local pits which increase the area of the corroding surface [2]. In the presence of hydrogen sulfide in the solution, the rate of general corrosion at the initial stage decreases because of sulfide films formation on the surface. At the ratio of partial pressures $P_{CO_2} : P_{H_2S} = 30:1$ the corrosion rate is the lowest, as a dense film of mackinawite (Fe_{1+x}S), which inhibits corrosion and acts as a diffusion barrier for the environment components [1, 2], is formed (Fig. 2*a*). At the ratio $P_{CO_2} : P_{H_2S} = 3:1$ the

rate of steel corrosion slows down at the beginning of exposure, but increases after some time, what is associated with the transformation of mackinawite and the formation of loose films of troilite FeS [1, 2] with an acicular structure (Fig. 2b). These results confirm that the morphology and protective properties of sulfide films on the steel surface depend on the content of hydrogen sulfide in the solution.

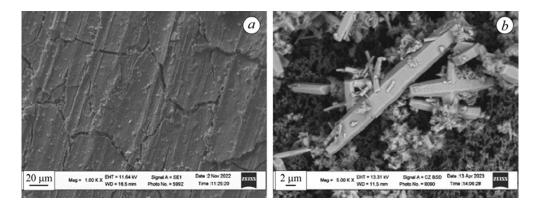


Fig. 2. Microstructure of the steel surface after exposure to a solution of 5% NaCl + 0.5% CH₃COOH at the ratio of the partial pressure (a) $P_{CO_2}: P_{H_2S} = 30:1$ and (b) $3:1; t = 20^{\circ}C; P = 0.1$ MPa.

The effect of temperature and pressure on the corrosion rate and hydrogenation of 17G1S-U steel in a chloride-acetate solution at different concentrations of CO₂ and H₂S is illustrated by the diagrams in Fig. 3. At a temperature of 60°C and a pressure of 5 MPa, the corrosion rate of steel after exposure for 510 h to a solution saturated with carbon dioxide and at the ratio of $P_{CO_2}: P_{H_2S} = 30:1$ increases by approx. 17 and approx. 13%, respectively. At a pressure ratio $P_{CO_2}: P_{H_2S} = 3:1$ corrosion slows down by half (Fig. 3*a*).

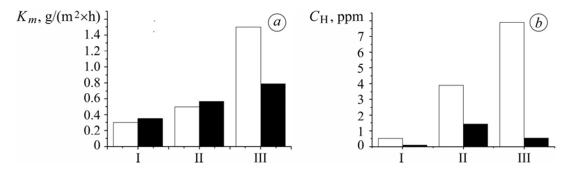


Fig. 3. Influence of H₂S and CO₂ partial pressures on (*a*) corrosion rate and (*b*) hydrogen absorption after exposure of steel to a chloride-acetate solution for 510 h at different temperatures and pressures: (I) CO₂ (saturated); (II) $P_{CO_2} : P_{H_2S} = 30:1$; (III) $P_{CO_2} : P_{H_2S} = 3:1$ [(\Box) P = 0.1 MPa, $t = 20^{\circ}$ C; (\blacksquare) P = 5.0 MPa, $t = 60^{\circ}$ C].

The hydrogenation of steel was significantly reduced with the increase in temperature and pressure. It is negligible in a solution saturated with CO₂, at the ratio $P_{CO_2}: P_{H_2S} = 30:1$ the concentration of absorbed hydrogen decreases in approx. 2.7 times, at the ratio $P_{CO_2}: P_{H_2S} = 3:1$ it decreases in approx. 15 times compared to the results obtained at 0.1 MPa and 20°C (Fig. 3*b*).

With the increase in temperature and pressure corrosion products transform. After tests in a solution with a pressure ratio of P_{CO_2} : $P_{H_2S} = 30:1$, a dense layer of grains with a size of 1 to 100 µm was revealed on the surface of the steel (Fig. 4). Small crystals have an octahedral shape and a chemical composition which corresponds to cubic iron sulfide FeS (50.58 at.% S and 49.42 at.% Fe) [1]. As the exposure time increases the size of grains increases and a dense barrier layer, which protects the steel from corrosion is formed. No pits and cracks were found under the corrosion products (Fig. 4*c*).

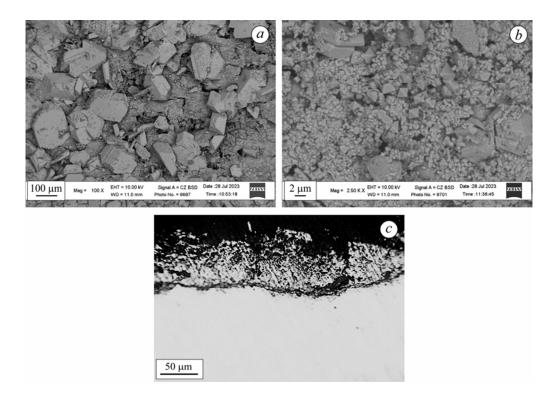


Fig. 4. Microstructure (a, b) and (c) cross-section of corrosion products on the steel surface after exposure to a chloride-acetate solution at the ratio of the partial pressure $P_{CO_2}: P_{H,S} = 30:1$, $t = 60^{\circ}C$ and P = 5 MPa.

Sulfide ions adsorbed on the steel surface can inhibit the recombination of hydrogen ions, which promotes their diffusion into the metal. As the temperature increases the mobility of sulfide ions increases and their adsorption slows down. As a result, the formation of hydrogen molecules accelerates and the steel's hydrogenation slows down. Probably, an increase in pressure affects the adsorption capacity of sulfide ions in the solution less than the temperature.

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CONCLUSIONS

The corrosion rate of 17G1S-U steel in a chloride-acetate solution saturated with CO₂ is lower than in the presence of hydrogen sulfide but increases over time due to the absence of protective carbonate films on the surface. The corrosion rate and hydrogenation of steel were determined by the hydrogen sulfide concentration in the environment. Corrosion slowed down at the ratio of partial pressures $P_{CO_2}: P_{H_2S} = 30:1$ and at the beginning of exposure to a solution with ratio $P_{CO_2}: P_{H_2S} = 3:1$ due formation of the dense films of mackinawite (Fe_{1+x}S). Over time mackinawite transformed into hexagonal FeS troilite with an acicular structure, and the corrosion rate doubled. Hydrogen absorption was approx. 4 ppm at a ratio of $P_{CO_2}: P_{H_2S} = 30:1$, and it increased to approx. 7.9 ppm at $P_{CO_2}: P_{H_2S} = 3:1$. The corrosion rate was reduced by half in a solution with $P_{CO_2}: P_{H_2S} = 3:1$ at $t = 60^{\circ}$ C and P = 5 MPa compared to $t = 20^{\circ}$ C and P = 0.1 MPa. A dense layer of FeS cubic iron sulfide crystals, which slowed down corrosion and practically eliminated hydrogenation, was formed on the steel surface.

Author contributions. All authors have contributed equally to the work, authors also read and approved the final manuscript.

Conflict of interest. Authors V. I. Pokhmurskii and M. S. Khoma are the Editorial Board Members of Materials Science (translated from Fizyko-Khimichna Mekhanika Materialiv). The paper was handled by another Editorial Board Member and has undergone a rigorous peer review process. Authors V. I. Pokhmurskii and M. S. Khoma were not involved in the journal's peer review or decisions related to this manuscript.

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