

# **Physicochemical processes and thermochemical parameters**  of GdFeO<sub>3</sub> formation from amorphous hydroxides: decisive role **of carbonate impurities**

**Yamen Albadi1,2  [·](http://orcid.org/0000-0002-8214-5291) Anastasia K. Bachina3 · Vadim I. Popkov3**

Received: 3 May 2023 / Accepted: 30 September 2023 / Published online: 21 November 2023© Akadémiai Kiadó, Budapest, Hungary 2023

## **Abstract**

The mechanism of formation of nanocrystalline gadolinium orthoferrite  $(GdFeO<sub>3</sub>)$  during the heat treatment of gadolinium and iron(III) hydroxides synthesized by ultrasound-assisted co-precipitation was studied. The obtained samples were investigated by energy-dispersive X-ray spectroscopy, simultaneous thermal analysis via coupled diferential scanning calorimetry and thermogravimetric analysis (DSC–TGA), powder X-ray difraction (PXRD) and Fourier-transform infrared (FTIR) spectroscopy. The DSC–TGA results confirm that the formation of  $GdFeO<sub>3</sub>$  occurs after the complete decomposition of gadolinium oxycarbonate derivatives. The PXRD results indicate that  $GdFeO<sub>3</sub>$  is formed as a result of the reaction of amorphous iron(III) oxide (*am*-Fe<sub>2</sub>O<sub>3</sub>) at temperatures of 675–700 °C with amorphous gadolinium oxide (*am*-Gd<sub>2</sub>O<sub>3</sub>) (primary carbonate-independent pathway) and at temperatures of 725–775 °C with cubic gadolinium oxide ( $c$ -Gd<sub>2</sub>O<sub>3</sub>) transformed from the hexagonal gadolinium oxide  $(h-Gd<sub>2</sub>O<sub>3</sub>)$  which results from the decomposition of gadolinium oxycarbonate derivatives at temperatures of 675–725 °C (secondary carbonate-dependent pathway). The FTIR results are consistent with the assumption that gadolinium oxycarbonate derivatives decompose with the formation of  $h$ -Gd<sub>2</sub>O<sub>3</sub> in the last-mentioned temperature range. The enthalpy of the reaction of formation of nanocrystalline  $GdFeO<sub>3</sub>$  defined from the DSC–TGA data is equal to −16.89 ± 0.36 kJ mol<sup>-1</sup>. The activation energy for the formation of nanocrystalline GdFeO<sub>3</sub> obtained from the DSC data is equal to  $1193.62 \pm 112.05$  kJ mol<sup>-1</sup>,  $1202.27 \pm 112.06$  kJ mol<sup>-1</sup> and  $1151.08 \pm 106.53$  kJ mol<sup>-1</sup> according to the Kissinger, Augis–Bennett/Boswell and Flynn–Wall–Ozawa methods, respectively. Also, based on the DSC data, the true onset temperature of the formation of nanocrystalline GdFeO<sub>3</sub> was found to be ~756 °C.

**Keywords** Gadolinium orthoferrite · GdFeO<sub>3</sub> · Co-precipitation · Gadolinium oxycarbonates · Enthalpy of formation · Activation energy

# **Introduction**

Rare-earth orthoferrites (*RE*FeO<sub>3</sub>, where *RE* is a rare-earth element: Sc, Y, La–Lu) are a class of chemical compounds with an orthorhombic perovskite structure that have been actively studied due to their magnetic, optical and electrical properties  $[1-7]$  $[1-7]$  $[1-7]$ . One of these compounds is gadolinium orthoferrite (GdFeO<sub>3</sub>) which has been proposed for

 $\boxtimes$  Yamen Albadi albadi.yamen@technolog.edu.ru

various applications such as catalysis [[8](#page-13-2)], photocatalysis [[9\]](#page-13-3), electrocatalysis [[10\]](#page-13-4), luminescence [[11](#page-13-5)], gas sensing [[12\]](#page-13-6) and as a contrast agent for magnetic resonance imaging (MRI)  $[13-16]$  $[13-16]$  $[13-16]$ . In the synthesis of GdFeO<sub>3</sub> nanocrystals, various methods have been employed, including coprecipitation [[17\]](#page-14-1), hydrothermal method [[18\]](#page-14-2), solution combustion [\[19\]](#page-14-3), sol–gel [[20\]](#page-14-4), microwave method [[21](#page-14-5)], sonochemical method [[22\]](#page-14-6) and using heterobimetallic [[23](#page-14-7)] and metal–organic [[24](#page-14-8)] precursors. The co-precipitation method, which we preferred in our previous works [[25](#page-14-9)[–30\]](#page-14-10), makes it possible to synthesize nanocrystalline  $GdFeO<sub>3</sub>$  with minimal foreign chemical impurities since the co-precipitated gadolinium and iron(III) hydroxides are dehydrated during heat treatment with the formation of the corresponding oxides, which then form the compound of interest. However, due to the active sorption of

<sup>&</sup>lt;sup>1</sup> Saint Petersburg State Institute of Technology, Saint Petersburg, Russian Federation 190013

<sup>&</sup>lt;sup>2</sup> Al-Baath University, Homs, Syrian Arab Republic 77

<sup>&</sup>lt;sup>3</sup> Ioffe Institute, Saint Petersburg, Russian Federation 194021

carbon oxide  $(CO<sub>2</sub>)$  from the ambient air into the reagent solutions during their preparation or onto the surface of co-precipitated hydroxides during their washing and drying, gadolinium carbonate can be formed. Such partial carbonatization of co-precipitated hydroxides may lead to sample inhomogeneity in terms of phase composition, and as a result, to the formation of another phase along with  $GdFeO<sub>3</sub>$  after heat treatment [[25](#page-14-9), [26,](#page-14-11) [30](#page-14-10)]. Therefore, it is of scientifc and practical interest to study and understand the role of carbonates in the formation mechanism of nanocrystalline  $GdFeO<sub>3</sub>$  by co-precipitation.

In the literature, several publications have investigated the mechanism of formation of  $GdFeO<sub>3</sub>$  nanocrystals by the co-precipitation method  $[31-33]$  $[31-33]$  $[31-33]$ . Prakash et al.  $[31]$ investigated the formation of nanocrystalline  $GdFeO<sub>3</sub>$  by sintering a synthesized sample of co-precipitated hydroxides at temperatures of 600–1200 °C. They found that gadolinium oxide  $(\text{Gd}_2\text{O}_3)$  was present in the sintered products up to a temperature of 1000 °C, above which single-phase  $GdFeO<sub>3</sub>$  nanocrystals were obtained; therefore, they consider the temperature of 1200 °C to be optimal for sintering. The authors reported the presence of carbonates in the synthesized sample, but they did not indicate their role in the formation of GdFeO<sub>3</sub>. Tugova and Karpov [[32\]](#page-14-14) studied the mechanism of formation of nanocrystalline  $GdFeO<sub>3</sub>$ upon the calcination of co-precipitated hydroxides at temperatures of 500–900 °C. They established that the products heat treated at temperatures of 500–600 °C contain gadolinium dioxymonocarbonate  $(\text{Gd}_2\text{O}_2\text{CO}_3)$  which decomposes at temperatures of 600–700 °C to form cubic  $Gd_2O_3$ , and starting from a temperature of 700  $\degree$ C, GdFeO<sub>3</sub> is formed. Based on these results, they concluded that the formation of  $GdFeO<sub>3</sub>$  nanocrystals occurs through two mechanisms: the frst mechanism is related to the simultaneous dehydration of co-precipitated hydroxides, while the other is associated with both dehydration and decarbonation, which leads to an increase in the temperature of  $GdFeO<sub>3</sub>$  synthesis. The work of Popkov et al. [[33](#page-14-13)] confrmed the results of the previous one  $\lceil 32 \rceil$  that the formation of  $GdFeO<sub>3</sub>$  nanocrystals proceeds via two routes: from the co-precipitated hydroxides along the main route and from the products of partial carbonatization along the side one.

In this paper, we continue the work of the aforementioned colleagues, focus on simultaneous thermal analysis of the precursor and study in detail the mechanism of formation of  $GdFeO<sub>3</sub>$  nanocrystals synthesized employing the ultrasound-assisted co-precipitation method, which apparently, maintains sample homogeneity by enhancing difusion and mixing processes [[30\]](#page-14-10). The content of carbonates in the products of heat treatment of the synthesized sample of co-precipitated hydroxides at diferent temperatures is estimated. In addition, the true onset temperature of the formation of nanocrystalline  $GdFeO<sub>3</sub>$  is found, and the enthalpy

and the activation energy for this formation reaction are also defned.

## **Materials and methods**

#### **Synthesis of samples**

In this work, gadolinium and iron(III) hydroxides were synthesized by direct co-precipitation accompanied by ultrasonication. After that, samples of the obtained co-precipitated hydroxides were heated at diferent temperatures in the air.

An aqueous solution of stoichiometric amounts of gadolinium nitrate hexahydrate "Gd(NO<sub>3</sub>)<sub>3</sub>·6H<sub>2</sub>O" (puriss., 99.9%) and iron(III) nitrate nonahydrate "Fe(NO<sub>3</sub>)<sub>3</sub>.9H<sub>2</sub>O" (pur., 98.00%) with a concentration of 0.01 mol  $L^{-1}$  for each salt was used as a source of gadolinium and iron(III) ions. An aqueous solution of sodium hydroxide (NaOH) (p.a.) with a concentration of 1 mol  $L^{-1}$  was used as a precipitating medium. Gadolinium and iron(III) nitrates react with sodium hydroxide to form water-soluble sodium nitrate (NaNO<sub>3</sub>) and a precipitate of gadolinium hydroxide " $Gd(OH)3$ " and iron(III) hydroxide "Fe(OH)<sub>3</sub>" (exchange reaction). The equation for this reaction can be written as follows:

 $Gd(NO_3)_{3}(aq) + Fe(NO_3)_{3}(aq) + 6NaOH(aq) = Gd(OH)_{3}($ s) + Fe(OH)<sub>3</sub>(s) + 6NaNO<sub>3</sub>(aq).

#### **Co‑precipitation of gadolinium and iron(III) hydroxides**

For the synthesis of co-precipitated hydroxides, 100 mL of 1 mol  $L^{-1}$  sodium hydroxide solution was added dropwise to 100 mL of 0.01 mol  $L^{-1}$  gadolinium and iron(III) nitrates solution, which was constantly stirred with a magnetic stirrer and irradiated with ultrasound by immersing the ultrasonic horn of a "Venpan type UD-20 automatic" ultrasonic disintegrator into it throughout the entire co-precipitation process. The synthesis was repeated two more times, each time with 100 mL of each reagent solution. Then, the resulting precipitates with the supernatants from the three syntheses were mixed using a magnetic stirrer. After settling, most of the supernatant was decanted, and the co-precipitated hydroxides were separated from the remainder by centrifugation. Thereafter, the obtained precipitate was washed three times with distilled water to remove residual sodium hydroxide and sodium nitrate. The washed co-precipitated hydroxides were separated from water by centrifugation and then dried at a temperature of 45 °C.

#### **Heat treatment of co‑precipitated hydroxides**

Samples of the synthesized co-precipitated hydroxides with a mass of about 50 mg were heated at diferent temperatures in the range of 650–850 °C with an interval of 25 °C (650, 675, 700, 725, 750, 775, 800, 825 and 850 °C) for 4 h in the air.

## **Characterization of co‑precipitated hydroxides**

#### <span id="page-2-1"></span>**Energy‑dispersive X‑ray spectroscopy**

The content of gadolinium and iron in the synthesized sample of co-precipitated hydroxides was determined by energydispersive X-ray spectroscopy (EDXS) using a "TESCAN VEGA3" scanning electron microscope (TESCAN ORSAY HOLDING, Brno-Kohoutovice, Czech Republic) with an accelerating voltage of 30.0 kV coupled with an "x-act" silicon drift detector (Oxford Instruments, Abingdon, Oxfordshire, UK).

## <span id="page-2-0"></span>**Diferential scanning calorimetry and thermogravimetric analysis**

The synthesized sample of co-precipitated hydroxides was investigated by diferential scanning calorimetry and thermogravimetric analysis (DSC–TGA) using a "NETZSCH STA 449 F3" simultaneous thermal analyzer (NETZSCH-Gerätebau GmbH, Selb, Germany) up to 900 °C at three different heating rates (10, 20 and 30  $\textdegree$ C min<sup>-1</sup>) in an inert atmosphere of argon. The sample masses were 11.160, 11.568 and 11.183 mg at heating rates of 10, 20 and 30 °C min−1, respectively. The DSC and TGA curves of the reference substances (potassium chromate " $K_2CrO_4$ " and barium carbonate "BaCO<sub>3</sub>") were obtained using the same simultaneous thermal analyzer at the same three heating rates. The sample mass of  $K_2CrO_4$  was 15.677 mg at a heating rate of 10 °C min−1 and 16.130 mg at heating rates of 20 and 30  $^{\circ}$ C min<sup>-1</sup>, while the sample mass of BaCO<sub>3</sub> was 11.933 mg at all heating rates. All measurements were carried out in platinum crucibles (DSC/TG pan Pt). The processing of the DSC–TGA data was carried out using the "NETZSCH Proteus Thermal Analysis" software.

#### **Characterization of heat‑treated products**

#### **Powder X‑ray difraction**

The powder X-ray diffraction (PXRD) patterns were recorded using a ''Rigaku SmartLab 3'' diffractometer (Rigaku Corporation, Tokyo, Japan) in the Bragg angle range of 15–61° at a scan speed of 0.2 deg min−1 and a step width of 0.01°. The processing of the PXRD data was carried out using the ''SmartLab Studio II'' software. The mass fractions of the crystalline phases were estimated by the Rietveld refnement method using the crystal structure data of the Inorganic Crystal Structure Database (ICSD) structures with the identifers of ICSD 27278, ICSD 162247 and ICSD 150677 for orthorhombic GdFeO<sub>3</sub>, hexagonal Gd<sub>2</sub>O<sub>3</sub> and cubic  $Gd_2O_3$ , correspondently. The relative mass fractions of the X-ray amorphous phases in the products that underwent heat treatment at temperatures up to 775 °C were estimated by normalizing the obtained difractograms of these products, determining the integrated intensity of the amorphous halo on the normalized difraction patterns in the Bragg angle range of  $\sim$  15.2–27.2° and recalculating the resulting values as percentages of the maximum. The estimated results of mass fractions of the X-ray amorphous and crystalline phases in each heat-treated product were combined by subtracting the integrated intensity of the amorphous halo from 100% and recalculating the mass fractions of crystalline phases as fractions of the obtained value.

#### **Fourier‑transform infrared spectroscopy**

The Fourier-transform infrared (FTIR) spectra in the wavenumber range of 400–4000  $cm^{-1}$  were obtained by Kubelka–Munk transformation of difuse refectance data acquired using a "Bruker INVENIO-S" FTIR spectrometer (Bruker Optics, Ettlingen, Germany) with a "PIKE" difuse refectance attachment (PIKE Technologies, Madison, WI, USA). The relative mass percentages of gadolinium oxycarbonate derivatives in the products that underwent heat treatment at temperatures up to 800 °C were estimated by normalizing the obtained FTIR spectra of these products, determining the integrated intensity of the asymmetric and symmetric stretching vibrations of carbonate groups  $(CO_3^{2-})$ on the normalized FTIR spectra in the wavenumber range of~1260–1625 cm<sup>-1</sup> and recalculating the resulting values as percentages of the maximum.

# **Results and discussion**

#### **Energy‑dispersive X‑ray spectroscopy**

According to the EDX spectrum presented in Fig. [1,](#page-3-0) the synthesized sample of co-precipitated hydroxides is practically chemically pure. The carbon detected in the investigated sample is related to the presence of carbonate groups, the appearance of which, as we mentioned earlier, is a consequence of the active sorption of  $CO<sub>2</sub>$  from the ambient air into the reagent solutions during their preparation or onto the surface of co-precipitated hydroxides during their washing and drying. Based on the EDXS results, the atomic ratio of gadolinium to iron (Gd:Fe) is  $(49.92 \pm 0.03)\%$ :(50  $.08 \pm 0.03$ %; therefore, the average Gd:Fe atomic ratio is about 1.000:1.003, which is very close to the atomic ratio of 1:1 in  $GdFeO<sub>3</sub>$ , i.e., the content of gadolinium and iron in the synthesized sample corresponds to the stoichiometry of the compound of interest. It is worth noting that the



<span id="page-3-0"></span>**Fig. 1** EDX spectrum of the co-precipitated hydroxides. The inset (pie chart) shows the average Gd:Fe atomic ratio in the sample

Gd:Fe atomic ratio is preserved during the synthesis since neither gadolinium nor iron is included in the composition of compounds that are volatile during the heat treatment of co-precipitated hydroxides.

## **Diferential scanning calorimetry and thermogravimetric analysis**

As was shown earlier [[25](#page-14-9), [27,](#page-14-15) [29,](#page-14-16) [33](#page-14-13)], upon heat treatment, gadolinium and iron(III) hydroxides undergo a series of physicochemical processes before forming  $GdFeO<sub>3</sub>$ . The DSC and TGA curves of the synthesized



sample of co-precipitated hydroxides at diferent heating rates (10–30 °C min<sup>-1</sup>) are presented in Fig. [2](#page-3-1)a and b, respectively. The frst derivatives of the TGA curves (the DTG curves) are presented in supplementary Fig. S1. On the obtained DSC curves (Fig. [2a](#page-3-1)), 7 thermal effects can be detected: 6 endothermic (effects indicated by downward pointing arrows 1–6) and one exothermic (efect indicated by upward pointing arrow 8), whereas on the presented DTG curves (Fig. S1), 7 endothermic effects are observed (effects  $1-7$ ). These effects are associated with the evaporation of the physically adsorbed water on the sample surface (endothermic effect  $1:- 53-63 °C$  by DSC, ~ 51-58 °C by DTG); the dehydration of iron(III) hydroxide to form iron(III) oxide (endothermic effect  $2: \sim 126-163$  °C by DSC,  $\sim$  132–156 °C by DTG); the dehydration of gadolinium hydroxide in three stages: frstly to form gadolinium oxide sesquihydrate (endothermic efect 3: ~ 226–247 °C by DSC,  $\sim$  223–246 °C by DTG), secondly to form gadolinium oxyhydroxide (endothermic efect 4: ~ 294–319 °C by DSC,  $\sim$  299–322 °C by DTG), and thirdly, to gadolinium oxide (endothermic effect 6:  $\sim$  414–452 °C by DSC,  $\sim$  420–442 °C by DTG); the first stage of decomposition of gadolinium carbonate to form an oxycarbonate derivative (endothermic effect  $5: \sim 371-384$  °C by DSC,  $\sim$  369–392 °C by DTG); the last stage of decomposition of gadolinium oxycarbonate derivative to form gadolinium oxide (endothermic effect  $7: \sim 756-758$  °C by DTG), and fnally, the formation of gadolinium orthoferrite from gadolinium and iron(III) oxides (exothermic effect 8:  $\sim$  764–772 °C by DSC). All these results are summarized in Table [1](#page-4-0). As can be seen from Table [1](#page-4-0), the temperatures of the maximum thermal and mass loss efects



<span id="page-3-1"></span>**Fig. 2** DSC (**a**) and TGA (**b**) curves of the co-precipitated hydroxides at diferent heating rates. The inset in (**b**) shows the exothermic efect of  $GdFeO<sub>3</sub>$  formation (*t* is the temperature,  $t_{\text{max}}$  is the temperature

 $\mathcal{D}$  Springer

of the maximum thermal efect). The arrows pointing down indicate endothermic effects (effects  $1-6$ ), while the arrows pointing up indicate exothermic effects (effect 8)

<span id="page-4-0"></span>**Table 1** Thermal (DSC) and mass loss (DTG) effects of the co-precipitated hydroxides at diferent heating rates



 $t_{\text{max}}$  is the temperature of the maximum thermal or mass loss effect,  $\beta$  is the heating rate

<sup>a</sup>The values of x in gadolinium oxycarbonate derivatives are defined later in this paper (see Section ["Car](#page-11-0)[bonate content in gadolinium oxycarbonate derivatives"](#page-11-0))

<sup>b</sup>These temperatures could not be accurately determined due to stage overlap and low intensity

generally tend to increase with increasing heating rate. According to the TGA curves (Fig. [2b](#page-3-1)), the total mass loss is about 21.69%, 21.75% and 20.58% at a heating rate of 10, 20 and 30  $^{\circ}$ C min<sup>-1</sup>, correspondently. These mass losses are related to the release of water (effects 1–4 and 6) and  $CO<sub>2</sub>$  (effects 5 and 7). The results obtained are in good agreement with the previous ones [\[32,](#page-14-14) [33](#page-14-13)], which confrm that the formation of  $GdFeO<sub>3</sub>$  occurs after the mass loss is completed, i.e., after the complete decomposition of gadolinium oxycarbonate derivatives.

# **Enthalpy of the reaction of formation of nanocrystalline GdFeO3**

To found the enthalpy of formation of nanocrystalline  $GdFeO<sub>3</sub>$  based on the DSC–TGA results, two reference substances were used:  $K_2$ CrO<sub>4</sub> and BaCO<sub>3</sub>. Potassium chromate and barium carbonate undergo certain physicochemical transformations at temperatures below and above the formation temperature of  $GdFeO<sub>3</sub>$ , respectively. According to the data used to calibrate the thermal analyzer, at a heating rate of 10 °C min<sup>-1</sup>, K<sub>2</sub>CrO<sub>4</sub> transforms from orthorhombic to hexagonal form in an endothermic process [\[34\]](#page-14-17) at a temperature of 668.0 °C with an enthalpy of 35.000 J  $g^{-1}$ , while  $BaCO<sub>3</sub>$  melts in an endothermic process at a temperature of 808.0 °C with an enthalpy of 94.900 J  $g^{-1}$ .

At each heating rate  $(\beta)$ , the onset temperature ( $t_{onset}$ ) and peak area  $(A)$  of the endothermic effects of the reference substances and the exothermic effect of  $GdFeO<sub>3</sub>$  formation were determined from the DSC curves, whereas the values of sample mass at each onset temperature ( $m_{onset}$ ) were

determined from the TGA curves. Peak areas in µV °C were calculated from their values in  $\mu$ V s given by the software using the heating rate according to Eq. [1.](#page-4-1)

<span id="page-4-1"></span>
$$
A[\mu V^{\circ}C] = A[\mu V s] \cdot \frac{\beta [^{\circ}C \min^{-1}]}{60[s \min^{-1}]} \tag{1}
$$

Then, the "specifc" peak areas (*A\**) were calculated by dividing the obtained peak areas by the corresponding sample masses at onset temperature as shown in Eq. [2](#page-4-2).

<span id="page-4-2"></span>
$$
A^* \left[ \mu V^{\circ} C \, m g^{-1} \right] = \frac{A \left[ \mu V^{\circ} C \right]}{m_{\text{onset}} \left[ m g \right]}
$$
 (2)

After that, the known enthalpy values associated with the endothermic effects of the reference substances were divided by the corresponding "specifc" peak areas as shown in Eq. [3.](#page-4-3)

<span id="page-4-3"></span>
$$
\Delta H/A^* \left[ J \mu V^{-1} {}^{\circ}C^{-1} \right] = \frac{\Delta H \left[ J g^{-1} \right] \cdot 10^{-3} \left[ g \, mg^{-1} \right]}{A^* \left[ \mu V {}^{\circ}C \, mg^{-1} \right]}
$$
(3)

Then, the resulting values of Δ*H*/*A\** were plotted as a function of  $t_{onset}$  at each heating rate (Fig. [3](#page-5-0)), where each dependence is a straight line drawn between two reference points: one for  $K_2CrO_4$  and the other for BaCO<sub>3</sub>.

Based on the equations of the straight lines obtained, the values of  $\Delta H/A$ <sup>\*</sup> for GdFeO<sub>3</sub> at each heating rate were calculated using the corresponding onset temperatures of the exothermic effect of  $GdFeO<sub>3</sub>$  formation. All these data are summarized in Table [2.](#page-5-1)



<span id="page-5-0"></span>**Fig. 3** Dependences of the ratio of enthalpy to the "specifc" peak area on the onset temperature of the thermal efect in the temperature range~663–805 °C at diferent heating rates

Thereafter, multiplying the resulting values of Δ*H*/*A\** for  $GdFeO<sub>3</sub>$  at each heating rate by the corresponding "specific" peak areas calculated earlier by Eq. [2](#page-4-2), the enthalpy of the reaction of formation of nanocrystalline  $GdFeO<sub>3</sub>$  can be estimated (Eq. [4\)](#page-5-2).

$$
\Delta H\left[J\,\mathrm{g}^{-1}\right] = \Delta H/A^* \left[J\mu V^{-1} \,{}^{\circ}\mathrm{C}^{-1}\right] \cdot A^* \left[\mu V \,{}^{\circ}\mathrm{Cmg}^{-1}\right] \cdot 10^3 \left[\mathrm{mg}\,\mathrm{g}^{-1}\right] \tag{4}
$$

According to Eq. [4,](#page-5-2) the enthalpy of the reaction of formation of nanocrystalline  $GdFeO<sub>3</sub>$  is in the range from about – 63.36 to – 66.54 J  $g^{-1}$  or from about – 16.54 to − 17.37 kJ mol<sup>-1</sup>. In other words, it is approximately equal to−64.67 ± 1.36 J g<sup>-1</sup> or − 16.89 ± 0.36 kJ mol<sup>-1</sup>. This value is much less than the enthalpy of formation of  $GdFeO<sub>3</sub>$  at 298.15 K (−1351.432 ± 8.368 kJ mol<sup>-1</sup> [[35\]](#page-14-18)), which can be explained by the fact that the standard enthalpy of formation

corresponds to the formation of  $GdFeO<sub>3</sub>$  from its constituent elements (Gd, Fe and O) in their standard states, while in this work, the enthalpy of formation of  $GdFeO<sub>3</sub>$  from gadolinium and iron(III) oxides was estimated. Furthermore, the calculated value does not correspond to bulk  $GdFeO<sub>3</sub>$ , but to nanosized  $GdFeO<sub>3</sub>$ ; therefore, it should be smaller.

## **Activation energy for the formation**  of nanocrystalline GdFeO<sub>3</sub>

To estimate the activation energy for the formation of nanocrystalline  $GdFeO<sub>3</sub>$  based on the DSC data, three kinetic methods were applied: the Kissinger method [\[36](#page-14-19)], the Augis**–**Bennett/Boswell method [[37,](#page-14-20) [38](#page-14-21)] and the Flynn**–**Wall**–**Ozawa (FWO) method [[39](#page-14-22), [40](#page-14-23)]. These non-isothermal kinetic methods depend on the relationship between the heating rate and the absolute temperature at which the reaction reaches its maximum value [[41\]](#page-14-24). The mathematical formulas for these methods can be written as shown in Eqs. [5–](#page-5-3)[7.](#page-5-4)

#### **Kissinger method**

<span id="page-5-3"></span>
$$
\ln\left(\frac{\beta}{T_{\text{max}}^2}\right) = -\frac{E_a}{RT_{\text{max}}} + \ln\left(\frac{AR}{E_a}\right) \tag{5}
$$

<span id="page-5-2"></span>**Augis–Bennett/Boswell method**

<span id="page-5-5"></span>
$$
\ln\left(\frac{\beta}{T_{\text{max}}}\right) = -\frac{E_a}{RT_{\text{max}}} + \text{const.}
$$
 (6)

#### **Flynn–Wall–Ozawa method**

<span id="page-5-4"></span>
$$
\log \beta = -0.4567 \frac{E_a}{RT_{\text{max}}} + \text{const.} \ln \beta = -1.052 \frac{E_a}{RT_{\text{max}}} + \text{const.}
$$
\n(7)

<span id="page-5-1"></span>**Table 2** DSC–TGA data used in the estimation of the enthalpy of the reaction of formation of nanocrystalline GdFeO<sub>3</sub>

$\beta$ /°C min <sup>-1</sup>	Substance	$t_{onset}$ <sup>o</sup> C	$m_{\text{onset}}$ /mg	$A/\mu V$ s	$A/\mu V$ °C	$A^*/\mu V$ °C mg <sup>-1</sup>	$\Delta H/A^*$ /J $\mu$ V <sup>-1</sup> °C <sup>-1</sup>
10	$K_2$ CrO <sub>4</sub>	664	15.610	322.2	53.70	3.44	$1.02 \times 10^{-2}$
	BaCO <sub>3</sub>	805	11.804	482.3	80.38	6.81	$1.39 \times 10^{-2}$
	GdFeO <sub>3</sub>	759	8.754	$-261.8$	$-43.63$	$-4.98$	$1.27 \times 10^{-2}$
20	$K_2$ CrO <sub>4</sub>	664	16.330	345.2	115.1	7.05	$4.97 \times 10^{-3}$
	BaCO <sub>3</sub>	805	12.469	476.2	158.7	12.73	$7.45 \times 10^{-3}$
	GdFeO <sub>3</sub>	765	9.064	$-268.1$	$-89.37$	$-9.86$	$6.75 \times 10^{-3}$
30	$K_2$ CrO <sub>4</sub>	663	16.720	346.9	173.5	10.37	$3.37 \times 10^{-3}$
	BaCO <sub>3</sub>	805	12.394	474.6	237.3	19.15	$4.96 \times 10^{-3}$
	GdFeO <sub>3</sub>	768	8.905	$-251.3$	$-125.7$	$-14.11$	$4.54 \times 10^{-3}$

<span id="page-6-0"></span>**Fig. 4** Dependences of  $ln(\beta/T_{\text{max}}^2)$  (Kissinger method), ln(*β*/*T*<sub>max</sub>) (Augis–Bennett/ Boswell method) and ln *β* (FWO method) on the reciprocal absolute temperature of the maximum exothermic efect of  $GdFeO<sub>3</sub>$  formation on the DSC curves.  $R^2$  is the coefficient of determination



In these equations,  $\beta$  is the heating rate,  $T_{\text{max}}$  is the absolute temperature at which the reaction reaches its maximum value, i.e., the peak of the thermal effect (in our work,  $T_{\text{max}}$  is the absolute temperature of the maximum exothermic efect of GdFeO<sub>3</sub> formation on the corresponding DSC curve),  $E_a$ is the activation energy, *A* is the pre-exponential factor and *R* is the molar gas constant.

Based on the Kissinger method (Eq. [5\)](#page-5-3), plotting  $\ln(\beta/T_{\text{max}}^2)$ against 1/*T*max results in a straight line with a slope of−*E*a/*R* and a *y*-intercept of ln(*AR*/*E*a). Similarly, in accordance with the Augis–Bennett/Boswell method (Eq. [6](#page-5-5)), plotting ln( $\beta/T_{\text{max}}$ ) versus  $1/T_{\text{max}}$  gives a straight line with a slope of−*E*a/*R*, whereas according to the FWO method (Eq. [7](#page-5-4)), plotting of ln  $\beta$  against  $1/T_{\text{max}}$  gives a straight line with a

slope of – 1.052*E*<sub>a</sub>/*R*. Figure [4](#page-6-0) shows these straight lines obtained using the previously determined temperatures of the maximum exothermic effect of  $GdFeO<sub>3</sub>$  formation on the DSC curves at different heating rates (Table [1](#page-4-0), effect 8). In the frst two methods, the activation energy for the formation of nanocrystalline  $GdFeO<sub>3</sub>$  can be calculated by multiplying the resulting slopes by−*R*, while in the third method, the obtained slope should be multiplied by−*R* and divided by 1.052. The DSC data used and the calculation results are summarized in Table [3.](#page-6-1)

According to the calculation results, the activation energy for the formation of nanocrystalline  $GdFeO<sub>3</sub>$ is approximately equal to  $1193.62 \pm 112.05$  kJ mol<sup>-1</sup>,  $1202.27 \pm 112.06$  kJ mol<sup>-1</sup> and  $1151.08 \pm 106.53$  kJ mol<sup>-1</sup>

$\beta$ /°C min <sup>-1</sup>	$t_{\text{max}}$ /°C	$T_{\text{max}}$ /K	$1/T_{\rm max}/K^{-1}$	Kissinger $\frac{\ln \beta/T_{\text{max}}^2}{K^{-1} \text{ min}^{-1}}$	Augis-Bennett/Boswell $\ln \beta/T_{\text{max}}/\text{min}^{-1}$	Flynn-Wall-Ozawa $\ln \beta / K$ min <sup>-1</sup>
10	764	1037	$9.64 \times 10^{-4}$	$-11.59$	$-4.642$	2.303
20	770	1043	$9.59 \times 10^{-4}$	$-10.90$	$-3.954$	2.996
30	772	1045	$9.56 \times 10^{-4}$	$-10.50$	$-3.551$	3.401
			$R^2$	0.9913	0.9914	0.9915
			$E_{\rm a}/\mathrm{kJ}$ mol <sup>-1</sup>	1193.62	1202.27	1151.08
			$\Delta E_{\rm s}$ /kJ mol <sup>-1</sup>	112.05	112.06	106.53
			$\Delta E_{\rm a}/E_{\rm a}/\%$	9.39	9.32	9.25

<span id="page-6-1"></span>**Table 3** DSC data used in the estimation of activation energy for the formation of nanocrystalline  $GdFeO<sub>3</sub>$  and the results obtained

 $R^2$  is the coefficient of determination,  $\Delta E_a$  is the standard error of the calculated activation energy,  $\Delta E_a$  is the relative standard error of the calculated activation energy

Y. Albadi et al.

based on the Kissinger, Augis–Bennett/Boswell and FWO methods, correspondently, which are quite close. The relative standard error of the estimated values ranges from about 9.25 to 9.39%. This high value of activation energy is due to the high temperature required for the complete decomposition of gadolinium oxycarbonate derivatives before the formation of  $GdFeO<sub>3</sub>$ , as discussed earlier (See Section "[Diferential scanning calorimetry and thermogravimet](#page-2-0)[ric analysis"](#page-2-0)). In accordance with the Kissinger method, the pre-exponential factor has an average value of about  $1.62 \times 10^{60}$  min<sup>-1</sup>.



<span id="page-7-0"></span>**Fig. 5** Dependence of the onset temperature of the formation of nanocrystalline  $GdFeO<sub>3</sub>$  on the heating rate

**True onset temperature of the formation**  of nanocrystalline GdFeO<sub>3</sub>

The values of the onset temperature of the exothermic effect of  $GdFeO<sub>3</sub>$  formation previously determined from the DSC curves (Table [2\)](#page-5-1) are found to be about 759, 765 and 768 °C at a heating rate of 10, 20 and 30 °C min−1, respectively. Therefore, the temperature at which nanocrystalline  $GdFeO<sub>3</sub>$  begins to form increases with increasing heating rate. To found the true onset temperature of the formation of nanocrystalline GdFeO<sub>3</sub>, i.e., at a heating rate of 0  $\rm{°C \, min}^{-1}$ , the dependence of  $t_{\text{onset}}$  on  $\beta$  was plotted, and as a result, a straight line was obtained (Fig. [5\)](#page-7-0). By extrapolating the resulting straight line to the *y*-axis (where  $\beta = 0$  °C min<sup>-1</sup>), the true onset temperature of the formation of nanocrystalline GdFeO<sub>3</sub> was found to be about 756 °C, which is quite close to the temperature of 750 °C used in our previous works [\[25](#page-14-9)–[30\]](#page-14-10).

## <span id="page-7-2"></span>**Powder X‑ray difraction**

The difractograms of the products of heat treatment of the co-precipitated gadolinium and iron(III) hydroxides in the temperature range of 650–850 °C for 4 h are presented in Fig. [6](#page-7-1). According to these difraction patterns, the heattreated product obtained at a temperature of 650 °C is X-ray amorphous, whereas the products that underwent heat treatment at a temperature of 800 °C and higher are pure-phase crystalline samples and consist of orthorhombic gadolinium orthoferrite ( $o$ -GdFeO<sub>3</sub>) (ICSD 27278, space group

<span id="page-7-1"></span>**Fig. 6** Powder X-ray difraction patterns of the heat-treated products of the co-precipitated hydroxides. The peak bars indicate the positions of the refections of the ICSD structures used in the Rietveld refnement. The Miller indices presented belong to  $o$ -GdFeO<sub>3</sub>



62: *Pbnm*). On the other hand, the X-ray phase analysis of the heat-treated products obtained in the temperature range of 675–775 °C shows that, in addition to  $o$ -GdFeO<sub>3</sub>, they also contain other crystalline phases that, apparently, can be associated with hexagonal gadolinium oxide  $(h-Gd<sub>2</sub>O<sub>3</sub>)$ (ICSD 162247, space group 164: *P*-3*m*1) and cubic gadolinium oxide  $(c\text{-Gd}_2O_3)$  (ICSD 150677, space group 206: *Ia*-3). Along with these three crystalline phases, the lastmentioned products also contain X-ray amorphous phases, as seen from the amorphous halo in the Bragg angle range of  $\sim$  15.2–27.2°. These phases include amorphous iron(III) oxide (am-Fe<sub>2</sub>O<sub>3</sub>), amorphous gadolinium oxide (am- $Gd_2O_3$ ) and/or amorphous gadolinium oxycarbonate derivatives " $am-Gd_2O_{3-x}(CO_3)$ ". The estimated results of mass fractions of the X-ray amorphous and crystalline phases in all heat-treated products are presented in Fig. [7.](#page-8-0)

In accordance with Fig. [7,](#page-8-0) with an increase in the treatment temperature from 650 to 700  $^{\circ}$ C, the mass fraction of X-ray amorphous phases sharply decreases (from 100% at 650 to ~ 31% at 700 °C), while the mass fraction of  $o$ -GdFeO<sub>3</sub> sharply increases (from 0 at 650 to  $\sim 61\%$ ) at 700 $\degree$ C). This can be explained by the formation of  $o$ -GdFeO<sub>3</sub> as a result of the reaction of  $am$ -Fe<sub>2</sub>O<sub>3</sub> and  $am$ -Gd<sub>2</sub>O<sub>3</sub> at temperatures of 675–700 °C. This primary pathway of  $GdFeO<sub>3</sub>$  formation is independent of the decomposition of gadolinium oxycarbonate derivatives.

On the other hand, the mass fractions of  $h$ -Gd<sub>2</sub>O<sub>3</sub> and  $c$ -Gd<sub>2</sub>O<sub>3</sub> progressively increase with an increase in the treatment temperature up to 725 °C (reaching a maximum of  $\sim$  13% for *h*-Gd<sub>2</sub>O<sub>3</sub> and  $\sim$  1.5% for *c*-Gd<sub>2</sub>O<sub>3</sub>), above which they begin to decrease gradually and completely disappear at a temperature of 800 °C. It is well known



<span id="page-8-0"></span>**Fig. 7** Mass fractions of the X-ray amorphous phases " $am$ -Fe<sub>2</sub>O<sub>3</sub>,  $am-Gd_2O_3$ ,  $am-Gd_2O_{3-x}(CO_3)$ <sup>r</sup>, and crystalline phases ( $o-GdFeO_3$ ,  $h\text{-}Gd_2O_3$ ,  $c\text{-}Gd_2O_3$ ) in the heat-treated products of the co-precipitated hydroxides

that under normal conditions, gadolinium oxide may exist in cubic (C-type, space group 206: *Ia*-3) or monoclinic (B-type, space group 12: *C*2/*m*) forms, whereas the hexagonal form (A-type, space group 164: *P*-3*m*1), at ambient pressure, is only stable at very high temperatures (more than 2443 K/ $\sim$ 2170 °C) [[42,](#page-14-25) [43\]](#page-14-26). However, the formation of  $h$ -Gd<sub>2</sub>O<sub>3</sub> in the products heat treated in the temperature range of 675–775 °C is most likely related to the presence of gadolinium oxycarbonate derivatives, which, if they were crystalline, could have a hexagonal form that is more stable than other crystalline forms, e.g., the hexagonal form of  $Gd_2O_2CO_3$  is more stable than its tetragonal and monoclinic forms [[44\]](#page-14-27). Based on this assumption and the results obtained, in the temperature range of 675–725 °C, gadolinium oxycarbonate derivatives progressively decompose with the formation of  $CO<sub>2</sub>$ and  $h$ -Gd<sub>2</sub>O<sub>3</sub>, which then transforms to the form stable in the synthesis conditions,  $c$ -Gd<sub>2</sub>O<sub>3</sub>, which, at temperatures of 725–775 °C, reacts with  $am\text{-Fe}_2\text{O}_3$  to form  $o\text{-GdFeO}_3$ . This is the secondary pathway of  $GdFeO<sub>3</sub>$  formation that depends on the decomposition of gadolinium oxycarbonate derivatives. It should be noted that the formation of  $o$ -GdFeO<sub>3</sub> as a result of the reaction of  $am$ -Fe<sub>2</sub>O<sub>3</sub> and  $h\text{-}Gd_2O_3$ , i.e., without the transformation of  $h\text{-}Gd_2O_3$  to  $c$ -Gd<sub>2</sub>O<sub>3</sub>, may be possible, but there is no clear evidence for this, especially since the mass fraction of  $c$ -Gd<sub>2</sub>O<sub>3</sub> is less than the mass fraction of  $h$ -Gd<sub>2</sub>O<sub>3</sub> in all heat-treated products containing both of these phases, which indicates that the  $GdFeO<sub>3</sub>$  formation along this pathway occurs with the participation of  $c$ -Gd<sub>2</sub>O<sub>3</sub> consumed. Besides, since the last stage of decomposition of gadolinium oxycarbonate derivatives proceeds at a temperature of ~ 756–758, as fol-lows from the DTG curves (Table [1,](#page-4-0) effect 7),  $h$ -Gd<sub>2</sub>O<sub>3</sub> can be formed even above the temperature of 725 °C; however, we cannot observe this on the difractograms of the products heat treated at temperatures of 750–775 °C, most likely because above the temperature of 725 °C,  $h$ -Gd<sub>2</sub>O<sub>3</sub> transforms to  $c$ -Gd<sub>2</sub>O<sub>3</sub> which then reacts with  $am$ -Fe<sub>2</sub>O<sub>3</sub> faster than  $h$ -Gd<sub>2</sub>O<sub>3</sub> is produced from the decomposition of gadolinium oxycarbonates derivatives. Similarly, the  $GdFeO<sub>3</sub>$  formation along this pathway can begin below the temperature of 725 °C, but in this case, the reaction of  $c$ -Gd<sub>2</sub>O<sub>3</sub> (and perhaps *h*-Gd<sub>2</sub>O<sub>3</sub>) with *am*-Fe<sub>2</sub>O<sub>3</sub> is slower than the production of  $h$ -Gd<sub>2</sub>O<sub>3</sub> from the decomposition of gadolinium oxycarbonate derivatives. The obtained results are in good agreement with the results of DSC, according to which the exothermic effect of  $GdFeO<sub>3</sub>$  formation is observed at a temperature of ~764–772 °C (Table [1,](#page-4-0) effect 8).

Figure [8](#page-9-0) illustrates the proposed mechanism for the formation of nanocrystalline  $GdFeO<sub>3</sub>$  synthesized, i.e., by ultrasound-assisted co-precipitation, as follows from the DSC–TGA and PXRD data.

<span id="page-9-0"></span>**Fig. 8** Mechanism of formation of nanocrystalline GdFeO<sub>3</sub> via heat treatment of co-precipitated hydroxides



<span id="page-9-1"></span>**Fig. 9** FTIR spectra of the heattreated products of the co-precipitated hydroxides  $(M = Gd^{3+})$ ,  $Fe<sup>3+</sup>$ ). The absorption bands corresponding to atmospheric  $CO<sub>2</sub>$  in the wavenumber range of~2275–2400 cm−1 were omitted to facilitate comparison

# **Fourier‑transform infrared spectroscopy**

The FTIR spectra of the products of heat treatment of the synthesized sample of co-precipitated gadolinium and iron(III) hydroxides in the temperature range of 650–850 °C for 4 h are presented in Fig. [9](#page-9-1). The wavenumbers of the absorption bands that correspond to the main vibrations in the obtained FTIR spectra of the products that underwent heat treatment in the temperature range of 650–775 °C and their assignment are summarized in Table [4](#page-10-0).

<span id="page-10-0"></span>**Table 4** Assignment of the absorption bands corresponding to the main vibrations in the FTIR spectra of the products heat treated up to 775 °C



The broad absorption band in the region of high wavenumbers centered at ~ 3256–3385  $cm^{-1}$  (band 1) is a consequence of the overlapping of asymmetric and symmetric stretching vibrations of water "*ν*(H<sub>2</sub>O)", which is physically adsorbed on the surface of the samples during their storage and/or preparation for investigation [[28\]](#page-14-28). The absorption band of the bending vibration of physically adsorbed water " $\delta$ (H<sub>2</sub>O)" is observed at ~ 1636–1637 cm<sup>-1</sup> (band 2). The bands observed at ~ 1507–1508 cm<sup>-1</sup> (band 3) and at ~1391–1398 cm<sup>-1</sup> (band 4) can be attributed to the asymmetric " $\nu_{\text{as}}(\text{CO}_3^2)$ " and symmetric " $\nu_{\text{s}}(\text{CO}_3^2)$ " stretching vibrations of carbonate groups, respectively, whereas the bands observed at ~ 1045–1088 cm<sup>-1</sup> (band 5), at ~849–859 cm<sup>-1</sup> (band 6) and at ~664–670 cm<sup>-1</sup> (band 7) may be assigned to the symmetric stretching vibration of the C–O bond " $\nu_s(C-O)$ ", the out-of-plane bending " $\pi$ (CO<sub>3</sub><sup>2-</sup>)" and rocking " $\rho$ (CO<sub>3</sub><sup>2-</sup>)" vibrations of carbonate groups, correspondently [[45](#page-14-29)]. The band observed at ~ 564–570 cm<sup>-1</sup> (band 8) can be attributed to the stretching vibrations of the Gd–O bond "*ν*(Gd–O)" and the Fe–O bond "*ν*(Fe–O)" in the Gd–O–Fe and Fe–O–Fe systems [[46](#page-14-30)], while the band observed at ~ 435–444 cm<sup>-1</sup> (band 9) may be assigned to the O–Fe–O bending vibration " $\delta$ (O–Fe–O)" in octahedral FeO<sub>6</sub> groups [[31](#page-14-12)]. The last-mentioned absorption band (band 9) is practically absent in the FTIR spectrum of the product heat treated at a temperature of 650  $\degree$ C, which is X-ray amorphous as followed from the PXRD results (see Section "[Powder](#page-7-2) [X-ray difraction"](#page-7-2)).

As can be seen from Fig. [9](#page-9-1), the intensity of the asymmetric and symmetric stretching vibrations of carbonate groups (Table [4](#page-10-0), bands 3 and 4) on the normalized FTIR



<span id="page-10-1"></span>**Fig. 10** Relative mass percentages of gadolinium oxycarbonate derivatives in the products heat treated at temperatures up to 800 °C according to the FTIR data (purple) and relative mass fractions of  $h$ -Gd<sub>2</sub>O<sub>3</sub> in these products as followed from the PXRD results (orange). "Relative (rel.)" here means that the values are calculated relative to the maximum

spectra in the wavenumber range of ~  $1260-1625$  cm<sup>-1</sup> gradually decreases with an increase in heating temperature to become practically negligible at a temperature of 800 °C. The relative mass percentages of gadolinium oxycarbonate derivatives in the products that underwent heat treatment at temperatures up to 800 °C, estimated based on the integrated intensity of these vibrations, are presented in Fig. [10.](#page-10-1) The mass fractions of  $h$ -Gd<sub>2</sub>O<sub>3</sub> in these heattreated products, previously estimated using the PXRD

data, were recalculated as percentages of the maximum and presented in the same fgure for comparison.

The results presented in Fig. [10](#page-10-1) are in good agreement with our assumption that gadolinium oxycarbonate derivatives progressively decompose with the formation of  $h$ -Gd<sub>2</sub>O<sub>3</sub> up to the temperature of 725 °C, above which the mass fraction of  $h$ -Gd<sub>2</sub>O<sub>3</sub> begins to decrease due to the formation of  $o$ -GdFeO<sub>3</sub> along with the secondary pathway, i.e., with the participation of  $c$ -Gd<sub>2</sub>O<sub>3</sub> (see Section "[Pow](#page-7-2)[der X-ray difraction](#page-7-2)"). Above the temperature of 725 °C, gadolinium oxycarbonate derivatives continue to decompose gradually, but, as we discussed earlier,  $h$ -Gd<sub>2</sub>O<sub>3</sub> then transforms to  $c$ -Gd<sub>2</sub>O<sub>3</sub> which reacts with  $am$ -Fe<sub>2</sub>O<sub>3</sub> faster than  $h$ -Gd<sub>2</sub>O<sub>3</sub> is produced; therefore, we observe not an increase but a decrease in the mass fraction of  $h$ -Gd<sub>2</sub>O<sub>3</sub>. At a temperature of 800 °C,  $h$ -Gd<sub>2</sub>O<sub>3</sub> completely disappears and gadolinium oxycarbonate derivatives become practically negligible.

# <span id="page-11-0"></span>**Carbonate content in gadolinium oxycarbonate derivatives**

To estimate the carbonate content in gadolinium oxycarbonate derivatives, i.e., to estimate the values of *x* in the formula  $Gd_2O_{3-x}(CO_3)_x$  at different temperatures, we used the TGA data at temperatures of 700, 725 and 750 °C at diferent heating rates (10–30 °C min<sup>-1</sup>) as well as the PXRD data for the products that underwent heat treatment at these temperatures, which include only *am*-Fe<sub>2</sub>O<sub>3</sub> and *am*-Gd<sub>2</sub>O<sub>3-*x*</sub>(CO<sub>3</sub>)<sub>*x*</sub> as amorphous phases. As discussed earlier, at temperatures of 675–700 °C,  $am\text{-Fe}_2\text{O}_3$  reacts with  $am\text{-Gd}_2\text{O}_3$  to form *o*-GdFeO<sub>3</sub> along the primary pathway (see Section "[Powder](#page-7-2)" [X-ray difraction](#page-7-2)"); therefore, the product heat treated at a temperature of 675 °C still contains  $am-Gd<sub>2</sub>O<sub>3</sub>$ , the presence of which complicates the calculation (the unnormalized difraction pattern of this product shown in Fig. S2 confirms that, at a treatment temperature of 675 °C,  $o$ -GdFeO<sub>3</sub> is not fully crystallized in comparison with the products heat treated at higher temperatures). As for the product heat treated at a temperature of 775 °C, the relative mass percentage of gadolinium oxycarbonate derivatives is rather small (Fig.  $10$ ), which may also affect the calculation results.

According to the EDXS results, the average Gd:Fe atomic ratio is about 1.000:1.003, which is maintained during heat treatment (see Section "[Energy-dispersive X-ray spectros](#page-2-1)[copy](#page-2-1)"); therefore, the average Gd:Fe molar ratio can be considered equal to 1:1 in all heat-treated products, which means that the amounts in moles (*n*) of gadolinium and iron are equal (Eq. [8\)](#page-11-1).

<span id="page-11-1"></span>
$$
n_{\rm Gd} = n_{\rm Fe} \tag{8}
$$

The amounts in moles of gadolinium and iron in the selected heat-treated products can be determined from Eqs. [9](#page-11-2) and [10](#page-11-3), respectively.

$$
n_{\text{Gd}} = n_{o-\text{GdFeO}_3} + 2n_{h-\text{Gd}_2\text{O}_3} + 2n_{c-\text{Gd}_2\text{O}_3} + 2n_{\text{Gd}_2\text{O}_{3-x}(\text{CO}_3)_x}
$$
(9)

<span id="page-11-3"></span><span id="page-11-2"></span>
$$
n_{\text{Fe}} = n_{o-\text{GdFeO}_3} + 2n_{am-\text{Fe}_2\text{O}_3}
$$
 (10)

Substituting Eqs. [9](#page-11-2) and [10](#page-11-3) in Eq. [8](#page-11-1) and rearranging, we can determine the amount in moles of  $Gd_2O_{3-r}(CO_3)$ <sub>x</sub> (Eq. [11\)](#page-11-4).oxycarbonate derivatives, one mole

<span id="page-11-4"></span>
$$
n_{\text{Gd}_2\text{O}_{3-x}}(\text{CO}_3)_x = n_{am-\text{Fe}_2\text{O}_3} - n_{h-\text{Gd}_2\text{O}_3} - n_{c-\text{Gd}_2\text{O}_3} \tag{11}
$$

Since the amount of a substance in moles is equal to its mass (*m*) divided by its molar mass (*M*), Eq. [11](#page-11-4) can be written as shown in Eq. [12.](#page-11-5)

<span id="page-11-5"></span>
$$
\frac{m_{\text{Gd}_2\text{O}_{3-x}}(\text{CO}_3)_x}{M_{\text{Gd}_2\text{O}_{3-x}}(\text{CO}_3)_x} = \frac{m_{am-\text{Fe}_2\text{O}_3}}{M_{\text{Fe}_2\text{O}_3}} - \frac{m_{h-\text{Gd}_2\text{O}_3}}{M_{\text{Gd}_2\text{O}_3}} - \frac{m_{c-\text{Gd}_2\text{O}_3}}{M_{\text{Gd}_2\text{O}_3}} \tag{12}
$$

The molar mass of gadolinium oxycarbonate derivatives can be determined as a function of *x* from Eq. [13.](#page-11-6)

<span id="page-11-6"></span>
$$
M_{\text{Gd}_2\text{O}_{3-x}}(\text{CO}_3)_x = M_{\text{Gd}_2\text{O}_3} + x \cdot M_{\text{CO}_2}
$$
\n(13)

Dividing both sides of Eq. [12](#page-11-5) by the mass of the sample at a temperature of  $t$  ( $m_t$ ) and considering that the mass of a phase divided by the sample mass at a certain temperature is equal to the mass fraction of that phase (*w*) at that temperature, Eq. [12](#page-11-5) is converted to Eq. [14](#page-11-7).

<span id="page-11-7"></span>
$$
\frac{w_{\text{Gd}_2\text{O}_{3-x}}(\text{CO}_3)_x}{M_{\text{Gd}_2\text{O}_3} + x \cdot M_{\text{CO}_2}} = \frac{w_{am} - \text{Fe}_2\text{O}_3}{M_{\text{Fe}_2\text{O}_3}} - \frac{w_{h} - \text{Gd}_2\text{O}_3}{M_{\text{Gd}_2\text{O}_3}} - \frac{w_{c} - \text{Gd}_2\text{O}_3}{M_{\text{Gd}_2\text{O}_3}}
$$
(14)

The mass fraction of  $am\text{-}\mathrm{Fe}_2\mathrm{O}_3$  in the selected samples can be determined from the estimated mass fraction of amorphous phases ( $w_{\text{amorph}}$ ) by subtracting the mass fraction of the other amorphous phase,  $Gd_2O_{3-x}(CO_3)$ <sub>x</sub>, as shown in Eq. [15](#page-11-8).

<span id="page-11-8"></span>
$$
w_{am-Fe_2O_3} = w_{amorph} - w_{Gd_2O_{3-x}}(CO_3)_x
$$
\n(15)

Substituting Eq. [15](#page-11-8) in Eq. [14](#page-11-7) and rearranging, we can determine the mass fraction of gadolinium oxycarbonate derivatives in the investigated samples as a function of *x* (Eq. [16\)](#page-12-0).

<span id="page-12-4"></span>**Table 5** TGA and PXRD data used in the estimation of carbonate content in gadolinium oxycarbonate derivatives and the results obtained



$$
w_{\text{Gd}_2\text{O}_{3-x}(\text{CO}_3)_x} = \frac{[w_{\text{amorph}} \cdot M_{\text{Gd}_2\text{O}_3} - (w_{h-\text{Gd}_2\text{O}_3} + w_{c-\text{Gd}_2\text{O}_3}) \cdot M_{\text{Fe}_2\text{O}_3}] \cdot (M_{\text{Gd}_2\text{O}_3} + x \cdot M_{\text{CO}_2})}{M_{\text{Gd}_2\text{O}_3} \cdot (M_{\text{Fe}_2\text{O}_3} + M_{\text{Gd}_2\text{O}_3} + x \cdot M_{\text{CO}_2})}
$$
(16)

In accordance with the decomposition equation for gadolinium oxycarbonate derivatives, one mole of  $Gd_2O_{3-x}(CO_3)_x$ decomposes with the release of  $x$  mole of  $CO<sub>2</sub>$ , and a mass of  $m(\text{Gd}_2\text{O}_{3-x}(\text{CO}_3)_x)$  decomposes with the release of a mass of  $m(CO<sub>2</sub>)$ ; therefore, using chemical arithmetic, we can write Eq. [17](#page-12-1).

$$
m_{\text{Gd}_2\text{O}_{3-x}}(\text{CO}_3)_x \cdot x \cdot M_{\text{CO}_2} = m_{\text{CO}_2} \cdot M_{\text{Gd}_2\text{O}_{3-x}}(\text{CO}_3)_x \tag{17}
$$

Since the mass of  $Gd_2O_{3-x}(CO_3)_x$  at a certain temperature, as mentioned above, is equal to its mass fraction multiplied



<span id="page-12-5"></span>**Fig. 11** Dependences of *x* in the formula of gadolinium oxycarbonate derivatives " $Gd_2O_{3-x}(CO_3)_x$ " on temperature (*t*) at different heating rates. The inset shows the extrapolation of these dependences to the temperature range in which the obtained DSC and DTG curves showed the frst stage of gadolinium carbonate decomposition with the formation of an oxycarbonate derivative.  $R^2$  is the coefficient of determination

<span id="page-12-0"></span>by the sample mass at that temperature, Eq. [17](#page-12-1) can be written as shown in Eq. [18.](#page-12-2)

<span id="page-12-2"></span>
$$
w_{\text{Gd}_2\text{O}_{3-x}(\text{CO}_3)_x} \cdot m_t \cdot x \cdot M_{\text{CO}_2} = m_{\text{CO}_2} \cdot \left( M_{\text{Gd}_2\text{O}_3} + x \cdot M_{\text{CO}_2} \right)
$$
\n(18)

Substituting Eq. [16](#page-12-0) in Eq. [18,](#page-12-2) rearranging and solving for *x*, we obtain the required equation (Eq. [19](#page-12-3)).

<span id="page-12-3"></span><span id="page-12-1"></span>
$$
x = \frac{m_{\text{CO}_2} \cdot M_{\text{Gd}_2\text{O}_3} \cdot (M_{\text{Fe}_2\text{O}_3} + M_{\text{Gd}_2\text{O}_3})}{M_{\text{CO}_2} \cdot [M_{\text{Gd}_2\text{O}_3} \cdot (m_t \cdot w_{\text{amorph}} - m_{\text{CO}_2}) - m_t \cdot M_{\text{Fe}_2\text{O}_3} \cdot (w_{h-\text{Gd}_2\text{O}_3} + w_{c-\text{Gd}_2\text{O}_3})]} \tag{19}
$$

Using Eq. [19](#page-12-3), the carbonate content in gadolinium oxycarbonate derivatives in the investigated samples can be estimated. At diferent heating rates (10–30 °C min−1), the sample mass at a temperature of  $t$  ( $m_t$ ), where  $t = 700, 725$ and 750 °C, was determined from the TGA curves (Fig. [2](#page-3-1)b). In addition, the mass of  $CO<sub>2</sub>$  included in the sample at a temperature of  $t$  " $m(CO<sub>2</sub>)$ ", i.e., its mass that will be released upon the complete decomposition of gadolinium oxycarbonate derivatives in this sample, was also determined using the TGA data as the diference between the sample mass at each temperature  $(m_t)$  and the sample mass at the end (completion) of mass loss on the TGA curve  $(m_{end})$ . The mass fractions of the amorphous phases ( $w_{\text{amorph}}$ ), *h*-Gd<sub>2</sub>O<sub>3</sub> "*w*(*h*- $Gd_2O_3$ " and  $c$ - $Gd_2O_3$  "w( $c$ - $Gd_2O_3$ )" at each investigated temperature were previously estimated based on the PXRD data (Fig. [7\)](#page-8-0). The TGA and PXRD data used and the calculation results are summarized in Table [5](#page-12-4). The dependences of *x* in  $Gd_2O_{3-x}(CO_3)_x$  on temperature at different heating rates are presented in Fig. [11.](#page-12-5)

As follows from Fig. [11](#page-12-5), the obtained dependences can be described by linear equations with coefficients of determination in the range of 0.9363–0.9999. The extrapolation of these dependences to the temperature of 775 °C shows that *x* at this temperature is estimated at  $\sim 0.00-0.22$  with an average value of  $\sim 0.12$ , which is in good agreement with the PXRD and FTIR results. On the other hand, extrapolating the resulting dependences to the temperature range in which the obtained DSC and DTG curves showed the frst stage of gadolinium carbonate decomposition with the formation of an oxycarbonate derivative (Table [1](#page-4-0), effect 5), the *x* value in this derivative may be estimated at  $\sim 1.56-2.33$ with an average value of  $\sim$  1.91. This result is very close to the literature data [\[33](#page-14-13)], according to which, at a temperature of ~363–370 °C, gadolinium carbonate " $Gd_2(CO_3)$ " decomposes to gadolinium monooxydicarbonate " $Gd_2O(CO_3)_2$ " with the release of  $CO<sub>2</sub>$ .

# **Conclusions**

In this work, a complex and detailed experimental investigation of the mechanism of formation of  $GdFeO<sub>3</sub>$ nanocrystals was performed, which shows that the co-precipitated gadolinium and iron(III) hydroxides upon heat treatment undergo a series of physicochemical processes including the dehydration to the corresponding oxides and the decomposition of gadolinium oxycarbonate derivatives prior to the formation of  $GdFeO<sub>3</sub>$ . The PXRD results indicated that the  $GdFeO<sub>3</sub>$  formation occurs via two pathways: a primary carbonate-independent pathway involving  $am-Gd<sub>2</sub>O<sub>3</sub>$  and a secondary carbonate-dependent pathway involving  $h\text{-}Gd_2O_3$  and  $c\text{-}Gd_2O_3$ . The FTIR results showed that gadolinium oxycarbonate derivatives gradually decompose with increasing heating temperature to become practically negligible at a temperature of 800 °C. The resulting linear equations for the dependence of *x* in  $Gd_2O_{3-x}(CO_3)$ <sub>x</sub> on temperature predict that gadolinium carbonate at a temperature of  $\sim$  266–426 °C decomposes to  $Gd_2O(CO_3)$ <sub>2</sub> which, at a temperature of ~553–601 °C, decomposes to  $Gd_2O_2CO_3$ , which, in turn, decomposes to Gd<sub>2</sub>O<sub>3</sub> at a temperature of ~ 776–839 °C.

**Supplementary Information** The online version contains supplementary material available at<https://doi.org/10.1007/s10973-023-12647-7>.

**Acknowledgements** The study was partially performed using the equipment of the Engineering Center of Saint Petersburg State Institute of Technology.

**Authors' contributions** Y. Albadi and V. I. Popkov contributed to conceptualization, methodology, validation and formal analysis; Y. Albadi and A. K. Bachina did investigation; A. K. Bachina and V. I. Popkov helped in resources; Y. Albadi did data curation, writing—original draft, visualization and writing—review & editing; V. I. Popkov contributed to writing—review & editing, supervision, project administration and funding acquisition.

#### **Declarations**

**Conflict of interest** The authors do not have relevant fnancial or nonfnancial interests to disclose.

# **References**

- <span id="page-13-0"></span>1. Bamzai KK, Bhat M. Electrical and magnetic properties of some rare earth orthoferrites (RFeO<sub>3</sub> where R = Y, Ho, Er) systems. Integr Ferroelectr. 2014;158:108–22.
- 2. Zhou Z, Guo L, Yang H, Liu Q, Ye F. Hydrothermal synthesis and magnetic properties of multiferroic rare-earth orthoferrites. J Alloys Compd. 2014;583:21–31.
- 3. Ramu N, Muralidharan R, Meera K, Jeong YH. Tailoring the magnetic and magnetoelectric properties of rare earth orthoferrites for room temperature applications. RSC Adv. 2016;6:72295–9.
- 4. Nakhaei M, Sanavi KD. Study on structural, magnetic and electrical properties of  $\text{ReFeO}_3$  ( $\text{Re} =$  La, Pr, Nd, Sm & Gd) orthoferrites. Physica B Condens Matter. 2021;612: 412899.
- 5. Warshi MK, Mishra V, Sagdeo A, Mishra V, Kumar R, Sagdeo PR. Structural, optical and electronic properties of  $\text{RFeO}_3$ . Ceram Int. 2018;44:8344–9.
- 6. Wang Z-Q, Lan Y-S, Zeng Z-Y, Chen X-R, Chen Q-F. Magnetic structures and optical properties of rare-earth orthoferrites RFeO<sub>3</sub> (R = Ho, Er, Tm and Lu). Solid State Commun. 2019;288:10–7.
- <span id="page-13-1"></span>7. Sultan K, Samad R, Islam SAU, Habib MZ, Ikram M. Efect of rare earth ions  $(R = Pr, Eu$  and  $Ho)$  on the structural and electrical properties of orthoferrites. J Electron Mater. 2019;48:6003–7.
- <span id="page-13-2"></span>8. Yafarova LV, Chislova IV, Zvereva IA, Kryuchkova TA, Kost VV, Sheshko TF. Sol–gel synthesis and investigation of catalysts on the basis of perovskite-type oxides  $GdMO<sub>3</sub>$  (M = Fe, Co). J Solgel Sci Technol. 2019;92:264–72. [https://doi.org/10.1007/](https://doi.org/10.1007/s10971-019-05013-3) [s10971-019-05013-3](https://doi.org/10.1007/s10971-019-05013-3).
- <span id="page-13-3"></span>9. Niu X, Li H, Liu G. Preparation, characterization and photocatalytic properties of  $REFeO<sub>3</sub>$  (RE=Sm, Eu, Gd). J Mol Catal A Chem. 2005;232:89–93.
- <span id="page-13-4"></span>10. Li L, Wang X, Lan Y, Gu W, Zhang S. Synthesis, photocatalytic and electrocatalytic activities of wormlike  $GdFeO<sub>3</sub>$  nanoparticles by a glycol-assisted sol-gel process. Ind Eng Chem Res. 2013;52:9130–6. [https://doi.org/10.1021/ie400940g.](https://doi.org/10.1021/ie400940g)
- <span id="page-13-5"></span>11. Zhang Y, Zheng A, Yang X, He H, Fan Y, Yao C. Cubic GdFeO<sub>3</sub> particle by a simple hydrothermal synthesis route and its photoluminescence and magnetic properties. CrystEngComm. 2012;14:8432.
- <span id="page-13-6"></span>12. Niu X, Du W, Du W. Preparation, characterization and gassensing properties of rare earth mixed oxides. Sens Actuators B Chem. 2004;99:399–404.
- <span id="page-13-7"></span>13. Söderlind F, Fortin MA, Petoral RM Jr, Klasson A, Veres T, Engström M, et al. Colloidal synthesis and characterization of ultrasmall perovskite GdFeO<sub>3</sub> nanocrystals. Nanotechnology. 2008;19: 085608.
- 14. Pinho SLC, Amaral JS, Wattiaux A, Duttine M, Delville M-H, Geraldes CFGC. Synthesis and characterization of rare-earth orthoferrite  $LnFeO<sub>3</sub>$  nanoparticles for bioimaging. Eur J Inorg Chem. 2018;2018:3570–8. [https://doi.org/10.1002/ejic.20180](https://doi.org/10.1002/ejic.201800468) [0468.](https://doi.org/10.1002/ejic.201800468)
- 15. Athar T, Vishwakarma SK, Bardia A, Khan AA. Super paramagnetic iron oxide and gadolinium ( $FeGdO<sub>3</sub>$ ) nanopowder synthesized by hydrolytic approach passes high level of biocompatibility and MRI-based dual contrast property for competent molecular imaging and therapeutic interventions. Biomed

Phys Eng Express. 2016;2:025010. [https://doi.org/10.1088/](https://doi.org/10.1088/2057-1976/2/2/025010) [2057-1976/2/2/025010](https://doi.org/10.1088/2057-1976/2/2/025010).

- <span id="page-14-0"></span>16. Deka S, Saxena V, Hasan A, Chandra P, Pandey LM. Synthesis, characterization and in vitro analysis of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>–GdFeO<sub>3</sub> biphasic materials as therapeutic agent for magnetic hyperthermia applications. Mater Sci Eng C. 2018;92:932–41.
- <span id="page-14-1"></span>17. Abiev RSh, Almjasheva OV, Popkov VI, Proskurina OV. Microreactor synthesis of nanosized particles: the role of micromixing, aggregation, and separation processes in heterogeneous nucleation. Chem Eng Res Des. 2022;178:73–94.
- <span id="page-14-2"></span>18. Mariyappan V, Keerthi M, Chen S-M, Jeyapragasam T. Nanostructured perovskite type gadolinium orthoferrite decorated RGO nanocomposite for the detection of nitrofurantoin in human urine and river water samples. J Colloid Interface Sci. 2021;600:537–49.
- <span id="page-14-3"></span>19. Ateia EE, Hussein B, Singh C, Okasha N. Study of physical properties of Co substituted  $GdFeO<sub>3</sub>$  orthoferrites and evaluation of their antibacterial activity. J Inorg Organomet Polym Mater. 2020;30:4320–8.
- <span id="page-14-4"></span>20. Santhosh BS, Yashas SR, Kumara Swamy N, Shivaraju HP. Application of non-hierarchical gadolinium ortho-ferrite nanostructure for LED-driven photocatalytic mineralization of doxycycline hydrochloride. J Mater Sci: Mater Electron. 2022;33:11676–86.
- <span id="page-14-5"></span>21. Tang P, Hu Y, Lin T, Jiang Z, Tang C. Preparation of nanocrystalline  $GdFeO<sub>3</sub>$  by microwave method and its visible-light photocatalytic activity. Integr Ferroelectr. 2014;153:73–8. [https://doi.org/](https://doi.org/10.1080/10584587.2014.902720) [10.1080/10584587.2014.902720.](https://doi.org/10.1080/10584587.2014.902720)
- <span id="page-14-6"></span>22. Sivakumar M, Gedanken A, Bhattacharya D, Brukental I, Yeshurun Y, Zhong W, et al. Sonochemical synthesis of nanocrystalline rare earth orthoferrites using  $Fe(CO)_{5}$  precursor. Chem Mater. 2004;16:3623–32. [https://doi.org/10.1021/cm049345x.](https://doi.org/10.1021/cm049345x)
- <span id="page-14-7"></span>23. Mathur S, Shen H, Lecerf N, Kjekshus A, Fjellvåg H, Goya GF. Nanocrystalline orthoferrite  $GdFeO<sub>3</sub>$  from a novel heterobimetallic precursor. Adv Mater. 2002;14:1405–9.
- <span id="page-14-8"></span>24. Lone IH, Khan H, Jain AK, Ahmed J, Ramanujachary KV, Ahmad T. Metal-organic precursor synthesis, structural characterization, and multiferroic properties of  $GdFeO<sub>3</sub>$  nanoparticles. ACS Omega. 2022;7:33908–15.
- <span id="page-14-9"></span>25. Albadi Y, Martinson KD, Shvidchenko AV, Buryanenko IV, Semenov VG, Popkov VI. Synthesis of  $GdFeO<sub>3</sub>$  nanoparticles via low-temperature reverse co-precipitation: the efect of strong agglomeration on the magnetic behavior. Nanosyst Phys Chem Math. 2020;11:252–9.
- <span id="page-14-11"></span>26. Albadi Y, Sirotkin AA, Semenov VG, Abiev RS, Popkov VI. Synthesis of superparamagnetic  $GdFeO<sub>3</sub>$  nanoparticles using a free impinging-jets microreactor. Russ Chem Bull. 2020;69:1290–5. [https://doi.org/10.1007/s11172-020-2900-x.](https://doi.org/10.1007/s11172-020-2900-x)
- <span id="page-14-15"></span>27. Popkov VI, Albadi Y. The efect of co-precipitation temperature on the crystallite size and aggregation/agglomeration of  $GdFeO<sub>3</sub>$ nanoparticles. Nanosyst Phys Chem Math. 2021;12:224–31.
- <span id="page-14-28"></span>28. Albadi Y, Ivanova MS, Grunin LY, Martinson KD, Chebanenko MI, Izotova SG, et al. The infuence of co-precipitation technique on the structure, morphology and dual-modal proton relaxivity of GdFeO<sub>3</sub> nanoparticles. Inorganics (Basel). 2021;9:39.
- <span id="page-14-16"></span>29. Albadi Y, Abiev RS, Sirotkin AA, Martinson KD, Chebanenko MI, Nevedomskiy VN, et al. Physicochemical and hydrodynamic aspects of  $GdFeO<sub>3</sub>$  production using a free impinging-jets method. Chem. Eng. Process.: Process Intensif. 2021;166: 108473.
- <span id="page-14-10"></span>30. Albadi Y, Ivanova MS, Grunin LY, Makarin RA, Komlev AS, Chebanenko MI, et al. Ultrasound-assisted co-precipitation synthesis of  $GdFeO<sub>3</sub>$  nanoparticles: structure, magnetic and MRI contrast properties. Phys Chem Chem Phys. 2022;24:29014–23.
- <span id="page-14-12"></span>31. Prakash BJ, Rudramadevi BH, Buddhudu S. Analysis of ferroelectric, dielectric and magnetic properties of  $GdFeO<sub>3</sub>$  nanoparticles. Ferroelectr Lett Sect. 2014;41:110–22. [https://doi.org/10.1080/](https://doi.org/10.1080/07315171.2014.956020) [07315171.2014.956020.](https://doi.org/10.1080/07315171.2014.956020)
- <span id="page-14-14"></span>32. Tugova EA, Karpov ON. Nanocrystalline perovskite-like oxides formation in  $Ln_2O_2-Fe_2O_2-H_2O$  (Ln = La, Gd) systems. Nanosyst Phys Chem Math. 2014;5:854–60.
- <span id="page-14-13"></span>33. Popkov VI, Tugova EA, Bachina AK, Almyasheva OV. The formation of nanocrystalline orthoferrites of rare-earth elements  $XFeO<sub>3</sub>$  (X = Y, La, Gd) via heat treatment of coprecipitated hydroxides. Russ J Gen Chem. 2017;87:2516–24. [https://doi.org/](https://doi.org/10.1134/S1070363217110020) [10.1134/S1070363217110020.](https://doi.org/10.1134/S1070363217110020)
- <span id="page-14-17"></span>34. Natarajan M, Secco EA. Anisotropic conductivity and phase transformation studies in potassium chromate crystals. Can J Chem. 1974;52:2436–8.
- <span id="page-14-18"></span>35. Glushko VP, editor. Thermal constants of substances. handbook in 10 Issues. Issue 8. Part 1. Tables of accepted values. Moscow: USSR Academy of Sciences. All-Union Institute of Scientifc and Technical Information. Institute for High Temperatures; 1978.
- <span id="page-14-19"></span>36. Kissinger HE. Reaction kinetics in diferential thermal analysis. Anal Chem. 1957;29:1702–6.
- <span id="page-14-20"></span>37. Augis JA, Bennett JE. Calculation of the Avrami parameters for heterogeneous solid state reactions using a modifcation of the Kissinger method. J Therm Anal. 1978;13:283–92.
- <span id="page-14-21"></span>38. Boswell PG. On the calculation of activation energies using a modifed Kissinger method. J Therm Anal. 1980;18:353–8.
- <span id="page-14-22"></span>39. Ozawa T. A new method of analyzing thermogravimetric data. Bull Chem Soc Jpn. 1965;38:1881–6.
- <span id="page-14-23"></span>40. Flynn JH, Wall LA. A quick, direct method for the determination of activation energy from thermogravimetric data. J Polym Sci B. 1966;4:323–8.
- <span id="page-14-24"></span>41. Zhang X. Applications of kinetic methods in thermal analysis: a review. Eng Sci. 2021;14:1–13.
- <span id="page-14-25"></span>42. Zinkevich M. Thermodynamics of rare earth sesquioxides. Prog Mater Sci. 2007;52:597–647.
- <span id="page-14-26"></span>43. Zhang FX, Lang M, Wang JW, Becker U, Ewing RC. Structural phase transitions of cubic  $Gd<sub>2</sub>O<sub>3</sub>$  at high pressures. Phys Rev B. 2008;78: 064114.
- <span id="page-14-27"></span>44. Ge W, Li Z, Lei Z, Chen T, Fu Z, Peng R, et al. Synthesis of hexagonal phase  $Gd_2O_2CO_3$ :  $Yb^{3+}$ , Er<sup>3+</sup>upconversion nanoparticles via  $SiO<sub>2</sub>$  coating and  $Nd<sup>3+</sup>$  doping. CrystEngComm. 2015;17:5702–9.
- <span id="page-14-29"></span>45. Gaspar RDL, Mazali IO, Sigoli FA. Particle size tailoring and luminescence of europium(III)-doped gadolinium oxide obtained by the modifed homogeneous precipitation method: Dielectric constant and counter anion efects. Colloids Surf A Physicochem Eng Asp. 2010;367:155–60.
- <span id="page-14-30"></span>46. Sai Vandana C, Hemalatha Rudramadevi B. Effect of  $Cu^{2+}$  substitution on the structural, magnetic and electrical properties of gadolinium orthoferrite. Mater Res Express. 2018;5(4): 046101. <https://doi.org/10.1088/2053-1591/aab7a8>

**Publisher's Note** Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional afliations.

Springer Nature or its licensor (e.g. a society or other partner) holds exclusive rights to this article under a publishing agreement with the author(s) or other rightsholder(s); author self-archiving of the accepted manuscript version of this article is solely governed by the terms of such publishing agreement and applicable law.