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Visible light-driven removal of Rhodamine B using indium-doped zinc oxide prepared by sol-gel method

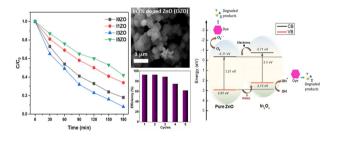
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Abstract

Industrial dye contamination in wastewater poses significant environmental challenges, necessitating the development of efficient photocatalysts for degradation. In this work, we investigate the In doping effect in the photocatalytic activity of zinc oxide (ZnO) nanoparticles for effective RhB degradation. Indium-doped ZnO nanoparticles were synthesized via sol–gel method and x-ray diffraction (XRD) analysis revealed a wurtzite hexagonal structure, with the crystallite size being varying from 65 nm to 53 nm with the introduction of In content. XPS measurements on the 3% In-doped ZnO sample revealed distinct core level spectra for In 3d, Zn 2p, and O 1s regions, confirming the presence of indium, zinc, and oxygen. Brunauer–Emmett–Teller (BET) analysis revealed increased surface area and pore size, with specific surface areas escalating from 0.9 m²/g for pure ZnO to 10.1 m²/g for 3% indium-doped ZnO under visible light irradiation due to the effect of the presence of In, which causing light absorption enhancement, narrow the band gap and improve charge carrier separation. These findings underscore the potential of indium-doped ZnO nanoparticles as efficient and sustainable photocatalysts for wastewater treatment, offering a promising avenue to address environmental challenges associated with industrial dye-contaminated effluents.

Graphical Abstract



Keywords Indium doping · Nanoparticle characterization · Photocatalysis · Rhodamine B degradation · Wastewater treatment · Zinc oxide

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Highlights

- Sol-gel synthesis of Indium-doped ZnO offers a scalable method for effective photocatalysts in wastewater treatment.
- Achieved 93% Rhodamine B degradation with Indium-doped ZnO nanoparticles under visible light.
- XRD analysis showed a reduction in crystallite size from 65 nm (pure ZnO) to 53 nm (3% In-doped ZnO).
- BET surface area increased from 0.9 m²/g (pure ZnO) to 10.1 m²/g (3% In-doped ZnO).
- UV-Vis spectroscopy indicated a reduced band gap in Indium doped ZnO (3.19 eV).
- TGA analysis highlighted improved thermal stability in Indium doped ZnO nanoparticles.

1 Introduction

The treatment of wastewater generated from industrial processes is imperative to mitigate its adverse impact on the environment, living organisms, and human health [1, 2]. This concern is particularly pronounced in industries such as textiles, paper, cosmetics, and printing, where xanthene dyes are extensively utilized [3, 4]. These dyes, including Rhodamine B (RhB), pose significant environmental hazards due to their water solubility and carcinogenic nature. When discharged into water bodies, these dyes can disrupt aquatic ecosystems and endanger human health through bioaccumulation in the food chain. Various techniques have been employed for wastewater treatment, ranging from conventional methods like sedimentation and filtration to advanced processes such as chemical oxidation and biological treatment [5]. Among these, Advanced Oxidation Processes (AOPs), particularly heterogeneous photocatalysis, have emerged as promising methods for degrading organic pollutants under UV/Vis light [6]. This process involves the generation of highly reactive hydroxyl radicals (OH) that can effectively oxidize organic contaminants, transforming them into harmless byproducts like water and carbon dioxide [7].

Zinc oxide (ZnO), known for its wide band gap and high quantum efficiency, has gained attention for its photocatalytic applications. However, despite its intrinsic photocatalytic properties, ZnO suffers from limitations such as rapid charge carrier recombination, which reduces its overall efficiency [8]. To address this challenge, efforts have focused on enhancing ZnO's efficacy through various means, including metal ion doping, surface modification, and nanostructuring. Metal ion doping, in particular, has shown promise in improving ZnO's photocatalytic performance by modifying its electronic band structure and reducing charge carrier recombination rates [9, 10].

In recent years, visible-light-driven photocatalysts have garnered attention for their eco-friendly potential. Unlike traditional photocatalysts, which are primarily activated by UV light, visible-light-driven photocatalysts can harness solar energy more efficiently, making them cost-effective and sustainable alternatives for wastewater treatment [11, 12]. Various strategies, including transition metal-ion doping, rare earth metal doping, co-doping, polymer modification, hetero-junction formation, and dye sensitization, have been explored to activate photocatalysts under visible light [13, 14]. Dye sensitization, in particular, offers a promising approach for enhancing the photocatalytic activity of semiconductor materials by harnessing the lightabsorbing properties of organic dyes [15].

Dye sensitization involves the continuous adsorption of dye molecules on the semiconductor surface, followed by their degradation upon excitation by visible light. This process facilitates electron transfer from the dye molecules to the semiconductor, initiating the photocatalytic reaction for organic pollutant degradation [16, 17]. Moreover, dye sensitization can extend the absorption range of photocatalysts into the visible region of the electromagnetic spectrum, enabling more efficient utilization of solar energy for wastewater treatment. Thus, the integration of dye sensitization with photocatalysis holds significant potential for developing sustainable and effective solutions for addressing environmental challenges associated with industrial wastewater containing xanthene dyes [18, 19].

Ways to further enhance the photodegradation properties of pollutants in water have been explored by many researchers who have used doping with alkaline-earth, transition, and post-transition metal elements. For instance, doping ZnO with Ca, was studied by Alejandro et al. [20] and it was demonstrated that it enhances the removal of tetracyclines (10 ppm) by sonophotocatalysis under visible light, achieving over 99% mineralization in 90 min for Ca-doped ZnO at 2%. Benamara et al. [21] revealed that Al doping enhances the removal of cyanotoxins in ZnO nanoparticles with visible LED irradiation. Ersöz et al. [22] showed that doping ZnO with Ag, a post-transition metal element, enhances the removal of RhB, achieving ~66% decomposition after 120 min under UV light irradiation. Doping ZnO with posttransition metal elements enhances photocatalytic dye removal via bandgap narrowing, reduced electron-hole recombination, improved charge separation, increased surface area, and synergistic effects, enhancing efficiency in degrading organic pollutants like RhB [23, 24]. In this study, we explored In as a dopant in ZnO nanoparticles. Indium doping was chosen because of its ability to introduce intermediate energy levels within the ZnO bandgap and improve charge separation efficiency during photocatalysis. Therefore, we have investigated the photo-degradation of RhB by using different indium doping content in ZnO nanoparticles.

2 Experimental details

2.1 Nanoparticles preparation

Indium doped zinc oxide (IZO) nanoparticles were synthesized via the sol-gel method, utilizing 16 g of zinc acetate dehydrate [Zn(CH₃COO)₂·2H₂O; 99%] as the precursor in 112 mL of methanol. The synthesis process involved magnetic stirring at room temperature for 10 min. Subsequently, the appropriate quantity of indium chloride (InCl₃) was added to achieve [In]/[Zn] ratios of 0.01, 0.03, and 0.05. After an additional 15 min of magnetic stirring, the solution was transferred to an autoclave and subjected to drying under supercritical conditions of ethyl alcohol (EtOH). The resulting nanopowders were then subjected to heat treatment in a furnace at 400 °C for 2 h in an air environment. The synthesized samples were assigned unique codes based on the nominal indium loading of each sample: I0ZO, I1ZO, I3ZO, and I5ZO.

2.2 Characterization

The following analytical techniques were employed for comprehensive characterization: Thermogravimetric Analysis (TGA) spectroscopy, X-ray Diffraction (XRD), Scanning Electron Microscopy (SEM) coupled with Energy-Dispersive X-ray Spectroscopy (EDX), Fourier-Transform Infrared (FTIR) spectroscopy, Brunauer–Emmett–Teller (BET) measurement, and Photoluminescence (PL) measurements. These methods collectively provide precise information about crystal structure, surface morphology, particle size, chemical element compositions, surface area, pore sizes, and more.

The X-ray diffraction patterns were captured within the 2 θ range spanning from 20° to 70°, utilizing a Bruker Philips X-Pert diffractometer with Ni β -filtered Cu-K α radiation (1.54178 Å radiation). The average crystallite size (d) was determined using Scherrer's [25–27]

$$d = \frac{0.9\lambda}{B\cos\theta_B} \tag{1}$$

Where λ represents the X-ray wavelength, θ_B is the maximum of the Bragg diffraction peak (in radians), and B is the full width at half maximum (FWHM) of the (101) XRD peak.

Surface images of the samples were obtained using an FEI Inspect S instrument, coupled with an Oxford INCA PentaFETx3 EDX spectrometer. The instrument featured a resolution of 137 eV at 5.9 keV (Mn K α_1) and was equipped with a nitrogen-cooled Si(Li) detector. Spectral data were collected at a working distance of 10 mm, an acceleration voltage of 20 kV, counting times of 60 s, and an approximate count rate of 3000 counts/s (cp).

BET measurements were conducted using N₂-physisorption at 77 K with a pressure transducer. Prior to the N₂ physisorption analyses, the samples underwent degassing at 150 °C for 12 h under a flow of N₂.

Photoluminescence (PL) measurements were performed on a NanoLog modular spectrofluorometer Horiba, employing a Xe lamp as the excitation light source at room temperature. Excitation occurred at a wavelength of 325 nm, and emissions were recorded between 350 and 750 nm. The optical absorbance spectra were measured using a Shimadzu UV-3101 PC UV-Vis-NIR spectrophotometer, covering a wavelength range of 200 nm to 800 nm.

X-ray photoelectron spectroscopy (XPS) measurements were performed for the I3ZO sample on a PHI Quantum 2000. Prior to the measurement, the nanopowder was fixed on adhesive carbon tape. The excitation energy was 1486.7 eV (monochromatic Al K α_1), while operating the X-ray source at a power of 50 W and a voltage of 15 kV. The pressure during the measurement was on the order of ~2 × 10⁻⁹ Torr. A dual beam (electron and ion) neutralizer was used for charge compensation. Charge referencing was performed by aligning the C1s emission line from the carbon tape to 284.5 eV corresponding to the C 1 s core level binding energy of graphite [28]. The In/Zn ratio was calculated from the In 3d_{5/2} and Zn 2p_{3/2} peaks after Shirley background subtraction and applying the appropriate corrected sensitivity factors.

2.3 Photocatalysis experiment

100 mL of 0.2μ M RhB dye solution was prepared and kept in the dark to avoid any disturbance before applying the catalyst. 20 mg of catalyst was then mixed with dye solution in the dark and kept for some time to attain the equilibrium. This mixture was irradiated by visible light using Heber photoreactor. At each 30 min interval of irradiation, the absorption spectrum was recorded to analyze the RhB degradation. To check the stability of the sample the same procedure was repeated by recovering the used catalyst each time and adding it to fresh dye solution for phototdegradation.

3 Results and discussion

3.1 SEM and EDS analysis of in-doped ZnO

The morphology of In-doped ZnO samples, annealed at 400 °C for 2 h, was examined through scanning electron microscopy (SEM). Figure 1a–c depict SEM images corresponding to 11ZO, 13ZO, and 15ZO, respectively. The SEM analysis reveals the presence of round agglomerates with dimensions ranging between 1 and 5 μ m, exhibiting

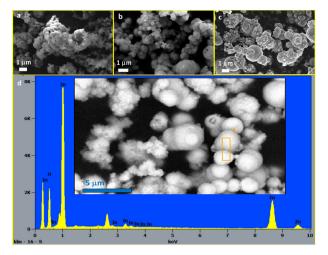


Fig. 1 SEM images of (a) I1ZO, (b) I3ZO, and (c) I5ZO samples (scale 1 µm). d EDS spectrum of I5ZO sample

spherical shapes and featuring a rough, porous, fine-grained microstructure. It was observed that the average smallest agglomerate was achieved for sample I3ZO.

To further ascertain the elemental composition within the In-doped ZnO structure, energy-dispersive X-ray spectroscopy (EDS) measurements were performed on the I5ZO sample, as illustrated in Fig. 1d. The EDS analysis indicates a significant presence of oxygen and zinc elements, with a relatively lower concentration of indium elements. This observation underscores the essential elements required for our structural composition, emphasizing the incorporation of indium into the ZnO matrix.

3.2 Microstructural analysis of annealed In-doped ZnO samples

The microstructure of the annealed In-doped ZnO (IZO) samples was examined through X-ray diffraction (XRD), and the obtained spectra are presented in Fig. 2. The discerned peaks were unequivocally indexed to the wurtzite hexagonal-shaped ZnO with the space group P63mc, in accordance with the Joint Committee on Powder Diffraction Standards (JCPDS) card file no 01-073-8765 [29]. These peaks correspond to the (100), (002), (101), (102), (110), and (103) planes of ZnO [30, 31]. At higher dopant concentrations (I3ZO and I5ZO samples) two additional peaks, attributed to a secondary phase, correspond to the (222) and (440) planes of the cubic bixbyite crystal structure of In_2O_3 , within the space group Ia-3 (JCPDS) card file 06-0416) [32].

The calculated lattice parameters, derived from the (100) and (002) planes, closely align with typical wurtzite ZnO values [33], yielding a = 3.251 Å and c = 5.203 Å. Utilizing the Scherrer equation, the average crystallite size (d) was estimated from the full width at half maximum (FWHM) of

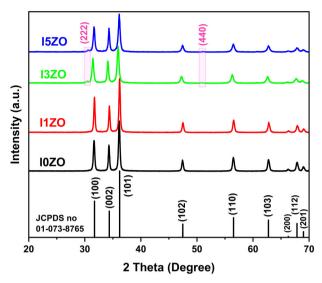


Fig. 2 XRD patterns of I0ZO, I1ZO, I3ZO and I5ZO samples

the diffraction peak. For the pure ZnO, the average crystallite size was 65 nm, while for 11ZO, 13ZO, and 15ZO samples, sizes of 53 nm, 56 nm, and 54 nm, respectively, were determined. These results provide valuable insights into the microstructural evolution induced by indium doping in ZnO.

3.3 Room-temperature photoluminescence analysis

Figure 3 presents the room-temperature photoluminescence (PL) spectrum captured across the 350-800 nm range for both pure and In-doped ZnO samples. From Fig. 3, it can be observed that, the pure ZnO sample exhibit a sharp and less intense emission peak at 390 nm which correspond to the near band edge (NBE) peak responsible for the recombination of free excitons within ZnO [34] and a broad and high intense emission peak at around 540 nm which correspond to the deep level emission (DLE) responsible for transitions within the bandgap. The substantial increase in the NBE peak intensity for the In-doped ZnO sample is due to improved crystalline quality and reduced defect concentration from In doping. In³⁺ ions minimize structural defects, resulting in a more ordered crystal structure and higher radiative recombination probability [35, 36]. Additionally, In³⁺ ions introduce shallow donor levels, increasing the concentration of free carriers that enhance NBE emission. This combination of reduced defects and increased free carriers leads to the observed increase in NBE peak intensity [37]. These defects create localized electronic states within the bandgap, which can trap and recombine charge carriers generated by photoexcitation. The intensity of the DLE peak appears least for I3ZO as shown in Fig. 3. This decrease can be attributed to In^{3+} ions reducing the concentration of oxygen defects responsible

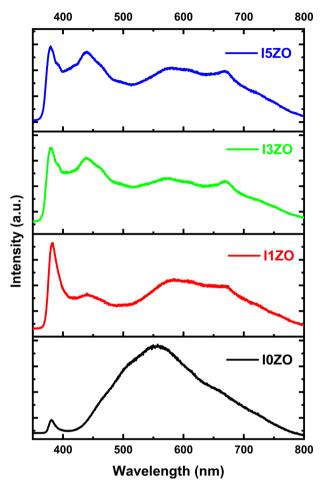


Fig. 3 PL spectra of IZO samples annealed at 400 °C for 2 h in air

for this emission peak by occupying lattice sites otherwise available for defect formation [38]. As acceptor dopants, In^{3+} ions help maintain charge neutrality, decreasing the need for oxygen vacancies. Additionally, In doping enhances ZnO's crystalline quality by minimizing structural defects, resulting in a more ordered crystal structure. Experimental studies show that In-doped ZnO has fewer oxygen vacancies compared to undoped ZnO due to In^{3+} ions during growth or annealing [39, 40].

3.4 UV-Visible absorbance spectra of the prepared IZO samples

To explore the optical characteristics and band gap, absorbance spectra of the samples were assessed using a UV-Vis spectrophotometer. The absorption spectra of pure and 3% In-doped ZnO nanoparticles in the UV and visible ranges are depicted in the Fig. 4. The spectra indicate pronounced absorption in the UV range (200–380 nm) and comparatively lower absorption in the visible range. Notably, absorbance varies with increasing doping

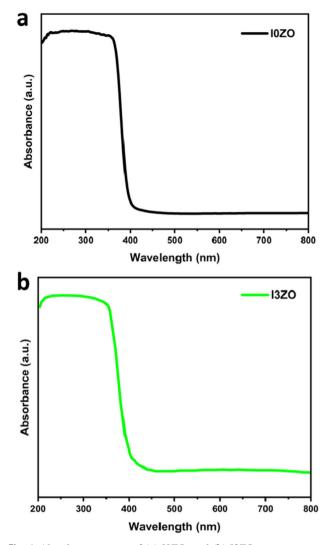


Fig. 4 Absorbance spectra of (a) I0ZO, and (b) I3ZO

concentration in the visible range. Utilizing a classical Tauc approach, the band gap (E_{σ}) of the synthesized samples was determined [41]. The doped ZnO sample exhibited a slightly reduced gap energy (3.19 eV) compared to pure ZnO (3.21 eV), likely due to disorder induced by defects in the ZnO matrix. Furthermore, the band gap energy decreased slightly with rising indium concentration. The diminished band gap energy of Indoped ZnO nanoparticles holds promise for enhancing their photocatalytic activity, particularly in RhB photodegradation. A smaller band gap facilitates better utilization of visible light, leading to increased electron-hole pair generation and subsequent formation of reactive oxygen species responsible for organic pollutant degradation. Additionally, indium dopants introduce defect states within the band gap, serving as trapping sites for photogenerated charge carriers, thereby prolonging their lifetime and enhancing photocatalytic efficiency [42].

3.5 Fourier transform Infrared spectroscopy (FTIR) spectra of IZO NPs

The FTIR spectra in Fig. 5 depict a comparison between pure ZnO nanoparticles and those incorporating indium doping. Across all spectra, a consistent broadband is

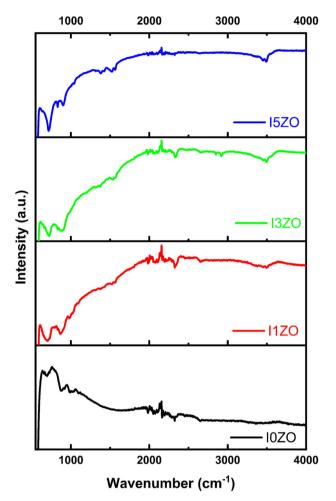


Fig. 5 FTIR spectra of pure and In-doped ZnO

observed, centered at 3415 cm^{-1} , indicating O–H stretching [43]. Additionally, the band within the 866–875 cm⁻¹ range is associated with C–H bending, and the band at 695–673 cm⁻¹ is attributed to M–O–M (M=Zn, In) interactions. The introduction of In results in the emergence of two new bands at 1525 and 1357 cm⁻¹, suggesting the formation of carbonate species [44].

3.6 X-ray photoelectron spectroscopy (XPS) on Indoped ZnO

For the 3% In-doped ZnO sample, XPS measurements were performed. The corresponding core level spectra of the In 3d, Zn 2p, and O 1s regions are displayed in Fig. 6. An intense C 1s signal from the graphitic substrate (carbon tape) was observed, shown in Fig. S1 (see supporting information). The In 3d and Zn 2p core levels show both a doublet occurring due to spin-orbit-coupling, while the two peaks visible in the O 1s spectrum are attributed to two different oxygen species. The relative intensity of the high binding energy component in the O 1s spectra is increasing for lower substrate coverage with the I3ZO powder and still present for the bare substrate. Therefore, the high binding energy component in the O 1s spectra can be assigned to a substrate-related oxygen species, rather than to an often discussed different core hole screening state [45, 46]. The low binding energy component lays well in the region of reported O 1s peak positions for ZnO $(540.4 \pm 0.5 \text{ eV} [47])$, 540.4 [48]) and In_2O_3 (540.5 ± 0.5 eV [47]). Considering the In 3d spectrum, the binding energy of both doublet peaks are in line with the mean value calculated from the NIST XPS database for In_2O_3 (In $3d_{5/2}$: $444.8 \pm 0.6 \text{ eV}$) [47]. Other In(III) chalcogenides and pnictogens as well as some In(I) halide compounds show similar In 3d line positions [48], while the respective values of the metallic In(0) are reported roughly 1 eV lower $(443.8 \pm 0.3 \text{ eV} [47])$. In case of the Zn 2p, the binding energy agree with the

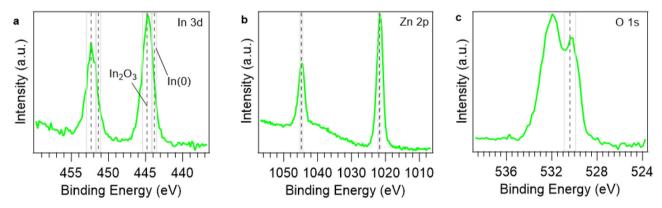


Fig. 6 XPS core level spectra of the (**a**) In 3d, (**b**) Zn 2p, and (**c**) O 1s regions. The dashed lines and gray regions are mean values and the respective standard deviation of the binding energies reported in the

NIST database [47]. In case of the Zn 2p peaks, the regions of metal and oxide are overlapping. The O 1s reference values are based on the reported values for ZnO

ranges of both Zn(II) in ZnO and metallic Zn(0) from literature (Zn $2p_{3/2}$: $1021.8 \pm 0.4 \text{ eV}$ and $1021.7 \pm 0.3 \text{ eV}$ [46]). Therefore, an unambiguous differentiation of Zn oxidation states from the binding energy alone is difficult. While chemical state analysis based on the Zn 2p core level emission is often ambiguous, the Zn LMM Auger emission exhibits much more pronounced chemical shifts [49]. When considering the Zn LMM Auger line (see supporting information), its position relative to the Zn $2p_{3/2}$ matches with Zn(II) rather than with the one for Zn(0) [50]. To conclude, for both In and Zn, there are no indications for metallic species but both elements show typical spectra expected for ZnO and In₂O₃. Addressing the indication of carbonate species from the FTIR, a second, low intensity peak in the C 1s spectrum is present at ~4 eV higher binding energies than the graphitic peak, which could be attributed to carbonates (O-C=O species: 3.8-4.3 eV higher [45]). Since a similar peak is also observed in the spectrum of the bare substrate, a final assignment to a powder related carbonate cannot be made with certainty. From the In $3d_{5/2}$ and Zn $2p_{3/2}$ peaks, a In/Zn ratio of 0.08 ± 0.01 was obtained for the I3ZO sample.

3.7 BET spectra of IZO nanoparticles

Figure 7 displays the Brunauer-Emmet-Teller (BET) adsorption-desorption isotherms for pure ZnO and In3%-doped ZnO samples synthesized through the sol-gel method. As per IUPAC classification, the registered isotherms exhibit type II characteristics. The presence of In3% doping results in clustered particles with a mesoporous architecture and an unrestricted monolayer-multilayer adsorption, as evidenced by an H_3 -type hysteresis loop [51]. The incorporation of In leads to a substantial increase in the surface areas of the ZnO nanoparticles compared to the pure ZnO counterparts. Specifically, the projected specific surface area (SBET) experiences a nearly 10 fold increase, escalating from $0.9 \text{ m}^2 \text{ g}^{-1}$ for pure ZnO to 10.1 m² g⁻¹ for the 3 percent In-doped sample. The insets in Fig. 7a, b depict the Barret-Joyner-Halenda (BJH) pore size distribution for IOZO and I3ZO samples, respectively. All pore sizes fall within the mesoporous range, with average diameters less than 50 nm, consistent with typical mesoporous material characteristics. A noteworthy augmentation in the average pore diameter is observed, progressing from 7.4 nm for pure ZnO to 36.9 nm for the In_{3%}-doped ZnO samples.

3.8 Thermogravimetric (TG) analysis

Thermogravimetric analysis (TGA) was employed to assess the thermal stability of ZnO and In3%-doped ZnO nanoparticles (NPs) by monitoring the weight loss of volatile components in the arranged samples. Figure 8 presents the TGA thermograms for IOZO and I3ZO NPs. The distinct

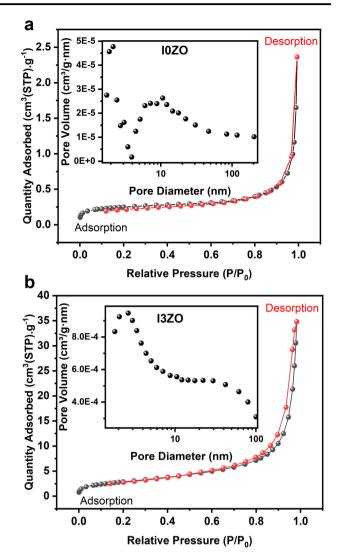


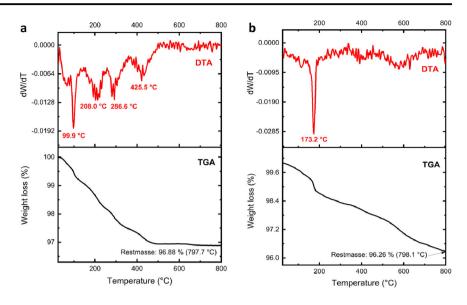
Fig. 7 Nitrogen adsorption-desorption isotherms with pore size distribution (inset) of (a) the pure ZnO and (b) the $In_{3\%}$ doped ZnO

TGA regions with percentage weight loss are depicted in Fig. 8.

In Fig. 8a, the initial region of ZnO NPs, observed up to 200 °C, exhibited a weight loss of ~1.3% attributed to moisture evaporation and the disintegration of volatile organic compounds. This observation was further confirmed by the peak intensity of the derivative thermogravimetric (DTG) curve at 99.9 °C. Similarly, in Fig. 8b, the first region of the TGA curve for I3ZO NPs, up to 200 °C, with a DTG intensity at 173.2 °C, was linked to the evaporation of surface moisture on I3ZO NPs, resulting in a weight loss of about 1.5%.

The TGA region between 200–800 °C for pure ZnO NPs exhibited sharp peaks in the DTG curve at 208 °C, 286.6 °C, and 425.5 °C. These peaks were associated with the thermal decomposition of volatile functional groups from various phytochemicals acting as reducing or stabilizing agents in the biogenesis of ZnO NPs, leading to a

Fig. 8 Thermogravimetric (TG) analysis with TDA curves of (a) the I0ZO and the (b) I3ZO samples



weight loss of 1.82%. In contrast, for the I3ZO sample, a weight loss of 2.24% was observed in the TGA curve between 200–800 °C.

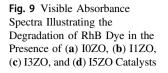
The weight loss observed between 200-500 °C in the TGA curve for both samples was attributed to the decomposition of lower molecular weight phytochemicals [52]. Additionally, in the TGA curve region between 500-800 °C, it is believed that the weight loss resulted from the thermal decomposition of higher molecular weight phytochemicals serving as capping or stabilizing agents [53]. In addition, the presence of carbon in the structure of In-doped ZnO nanoparticles (I3ZO NPs) compared to pure ZnO nanoparticles (IOZO NPs) can be attributed to the higher weight loss observed in the TGA analysis for I3ZO NPs. The TGA results indicate that I3ZO NPs exhibit a higher weight loss (3.74%) compared to IOZO NPs (3.12%), suggesting a greater presence of organic carbon components in the In-doped sample [54]. The presence of these organic carbon components can potentially influence the photocatalytic activity of the nanoparticles, including the photodegradation of RhB. Carbon-based materials are known to enhance the photocatalytic performance of semiconductors like ZnO by acting as sensitizers, facilitating charge separation, and promoting the formation of reactive oxygen species [55]. However, the specific effect of the carbon components present in the In-doped ZnO nanoparticles on the photodegradation of RhB requires further investigation.

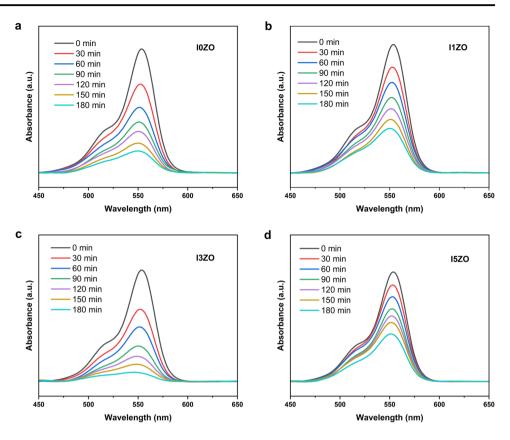
3.9 RhB degradation by In doped ZnO photocatalysts

The absorbance spectra of the RhB solutions containing I0ZO, I1ZO, I3ZO, and I5ZO are presented in the Fig. 9a–d, respectively. The decrease in dye concentration was detected with the increase in time, and ultimately, the dye was

degraded after 180 min. It is due to the fact that the nanoparticles acted as catalysts to degrade the dye molecules. These results of RhB degradation have shown 82% degradation by using pure zinc oxide nanoparticles as represented in Fig. 10a. After the doping of ZnO with In, the degradation of dye was found to be increased by 93%, which is much higher than undoped nanoparticles, particularly for the sample I3ZO.

The concentration report C/C₀ is crucial for understanding the reaction kinetics. It can be depicted through the equation: Ln $(A_t/A_0) = -kt$ [56, 57], here, k stands for the pseudo-first-order rate constant, t denotes the reaction time, A_0 represents the initial concentration of RhB at t = 0, and At signifies the concentration at time t, which can be derived from the absorbance of the peak at 550 nm. The constant k can be accurately determined directly from the slope of the straight line resulting from plots of $\ln (A_t/A_0)$ versus reaction time, presented in Fig. 10b. The values of k for I0ZO, I1ZO, I3ZO, and I5ZO were 9.36×10^{-3} , 5.86×10^{-3} , 13.26×10^{-3} , and $4.56 \times 10^{-3} \text{ min}^{-1}$, respectively. The value of k indicates how fast the dye degradation reaction proceeds. A higher value of k implies a faster rate of reaction, and hence it is clear that I3ZO is more efficient in facilitating the degradation of the dye molecule. This can be attributed to the large specific surface area and pore diameter of the I3ZO catalyst as evident from BET studies. The large surface area provides more active sites allow for more reactant molecules to come into contact with the catalyst surface, leading to enhanced catalytic activity and the large pore diameter facilitates the diffusion of reactant molecules to and from the active sites within the catalyst structure enhancing the reaction kinetics. In addition to that, the reduced recombination rate of charge carriers in the presence of I3ZO as evident from PL studies also contributes to the high efficiency of I3ZO catalyst.





A schematic diagram of the underlying mechanism of the photocatalytic degradation is shown in Fig. 11. The valence band (VB) and conduction band (CB) values are calculated from Mulliken electronegativity approach. Under visible light irradiation, the photocatalysts become excited, generating electrons and holes.

$$ZnO + h\vartheta \to e_{CB}^- + h_{VB}^+$$
 (2)

Since the CB edge potential of In_2O_3 is more negative than that of ZnO, the electrons from In_2O_3 can easily transfer to the CB of ZnO [58]. This transfer facilitates the efficient separation of photogenerated electron-hole pairs, reducing recombination losses. The transferred electrons in the CB of ZnO then interact with molecular oxygen adsorbed on the catalyst surface, reducing it to superoxide anion radicals.

$$e^- + O_2 \to \dot{O}_2^- \tag{3}$$

Similarly, The VB edge potential of ZnO is more positive than that of In_2O_3 . This potential difference drives the photogenerated holes to transfer from the VB of ZnO to the VB of In_2O_3 . The holes in the VB of In_2O_3 then directly participate in the degradation of the dye molecule. All of these highly reactive species engage in the photocatalytic degradation of dye molecules adsorbed on the photocatalyst surface to form smaller by-products [59]. Therefore, the application of indium-doped ZnO emerges as promising strategy for the effective charge separation and thereby enhancing the efficiency of photocatalytic degradation processes, particularly in the treatment of industrial dye-contaminated effluents.

The reusability of the catalyst was analyzed through multiple cycles of experiment and the efficiency for each cycle is shown in Fig. 12.

The I3ZO catalyst demonstrates excellent reusability, as established through five continuous cycles of experiments. In the first two cycles, there is no significant change in efficiency. A slight decrease in efficiency is observed in the third cycle, followed by a gradual decline in subsequent cycles. Notably, even after five cycles, the catalyst retains more than 60% of its initial efficiency, which is commendable compared to other reported catalysts [60, 61]. The observed decrease in efficiency can be attributed to the weight loss during the recovery of samples after each cycle and the partial removal of adsorbed dye molecules from the catalyst surface.

Based on Table 1, we conducted a comparative study of our prepared sample (I3ZO) against other relevant works in the literature. In-doped ZnO synthesized via the sol–gel method demonstrates superior performance in the photocatalytic degradation of RhB. It achieves a 93% removal rate at a concentration of 0.2 mM within 180 min, with a high rate

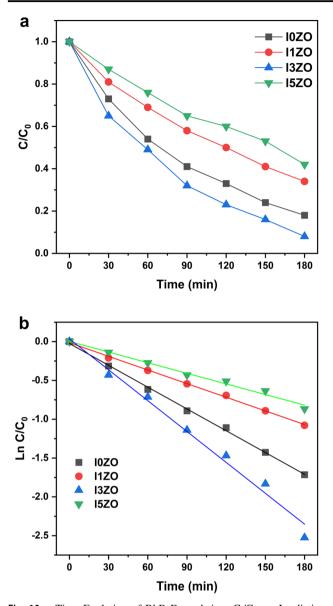


Fig. 10 a Time Evolution of RhB Degradation: C_t/C_0 vs. Irradiation Time Plots for All Samples, and (b) In (C_t/C_0) vs. Irradiation Time Demonstrating First-Order Rate Kinetics with Linear Fitting

constant of 13.26×10^3 min⁻¹ under visible light illumination. This efficiency surpasses other materials such as Fe/Cd codoped ZnO (76% in 140 min) [60], Bi₂WO₆/FTO (94% in 240 min) [61], and Cu₂O/TiO₂ (78% in 180 min) [62]. Additionally, compared to ZnO nanoparticles synthesized via laser methods (rate constant of 2.85×10^3 min⁻¹) [63] and N–TiO₂/FTO (64% in 240 min) [64], In-doped ZnO not only achieves a higher removal rate but also does so more rapidly and efficiently against some materials. Furthermore, other materials such as N-ZnO/FTO (43% in 160 min) [65], ZnO/ Ag (38% in 300 min) [66], and WO₃/TiO₂/FTO (58.7% in 160 min) [67] show significantly lower performance metrics. The ability of In-doped ZnO to operate effectively under visible light conditions makes it particularly advantageous for

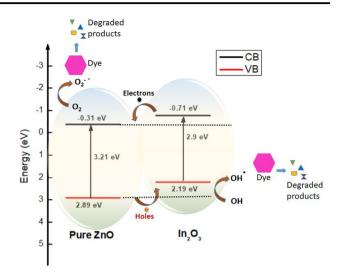


Fig. 11 Photocatalytic dye degradation mechanism

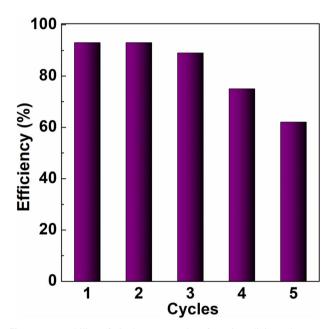


Fig. 12 Reusability of I3ZO nanopowders for mineralizing RhB

practical environmental applications. This comparative analysis clearly highlights In-doped ZnO's powerful and rapid photocatalytic capabilities, positioning it as a leading material for the photocatalytic degradation of organic pollutants.

4 Conclusion

In this work, the photocatalytic degradation of RhB under visible light irradiation by using indium-doped zinc oxide nanoparticles was investigated. Regardless of the In content ranging from 3 to 5%, indium doping induces secondary phase formation, altering ZnO microstructure, while for pure ZnO and 1% In doping, the wurtzite

Table 1	Comparative	Performance of	Photocatalytic	Materials for	Rhodamine B	Degradation

Material	Synthesis	RhB concentration	Percentage removal	Time (min)	Rate constant (min ⁻¹)	Light illumination	References
Fe/Cd co-doped ZnO	Sol-gel	0.05 g/L	76%	140	9.16×10^{-3}	visible light	[60]
ZnO nanoparticles	Laser-Synthesized	5 mg/L		180	2.85×10^{-3}	UV light	[63]
Bi ₂ WO ₆ /FTO	Spray pyrolysis	30 mg/L	94%	240	$1.08 \times 10^{-4} \mathrm{s}^{-1}$	visible light	[61]
N-ZnO/FTO	Electrodeposition	1 mM	43%	160	0.023	visible light	[65]
N-TiO ₂ /FTO	Chemical spray pyrolysis	0.5 mM	64%	240		UV-Visible light	[68]
N-TiO ₂	Ion implantation	80 mg/L	43.2%	80	0.011	Visible light	[64]
ZnO/Ag	Chemical bath deposition	5 mg/L	38%	300		Visible light	[66]
WO ₃ /TiO ₂ / FTO	Spray pyrolysis	1 mM	58.7%	160	5.48×10^{-7} s ⁻¹	sunlight	[<mark>67</mark>]
Bi ₂ WO ₆ /WO ₃ / TiO ₂	Hydrothermal	5 mg/L	66.07%	120		visible light	[69]
Cu ₂ O/TiO ₂	SILAR	5 mg/L	78%	180		visible light	[<mark>61</mark>]
In doped ZnO	Sol-gel	0.2 mM	93%	180	13.26×10^{-3}	visible light	This work

structure is preserved. XPS analysis confirmed the presence of indium, zinc, and oxygen in the 3% In-doped ZnO sample, with no indications of metallic species, and typical spectra expected for ZnO and In_2O_3 . However, with increasing the In content from 0 to 3%, we observed an enhanced surface area from 0.9 m²/g to 10.1 m²/g, which is responsible for the improvement of the photocatalytic efficiency. Therefore, In doping in ZnO nanoparticles proves to be a promising approach for sustainable photocatalysts for wastewater treatment, addressing environmental challenges posed by dyecontaminated effluents.

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Compliance with ethical standards

Conflict of interest The authors declare no competing interests.

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