ORIGINAL PAPER: SOL-GEL AND HYBRID MATERIALS FOR OPTICAL, PHOTONIC AND OPTOELECTRONIC APPLICATIONS



# Phase formation and luminescent properties of $Ca_{12}AI_{14}O_{33}$ :Eu, Nd prepared by sol-gel method

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# Abstract

Calcium aluminate  $Ca_{12}Al_{14}O_{33}$  and phosphors based on it were obtained using sol–gel technology. Complex studies were used to identify the dynamics of phase and structural transformations in the synthesis of  $Ca_{12}Al_{14}O_{33}$ . The main stages of the calcium aluminate formation were determined using thermal analysis, IR spectroscopy and X-ray phase analysis. Solid-phase interaction in the formation of  $Ca_{12}Al_{14}O_{33}$  includes the steps of obtaining calcium carbonate and amorphous alumina (800 °C), aluminates of composition  $Ca_{3}Al_{2}O_{6}$  and  $CaAl_{2}O_{4}$  (900–1000 °C), and  $Ca_{5}Al_{6}O_{14}$  and  $Ca_{4}Al_{6}O_{13}$  (1000–1100 °C).  $Ca_{12}Al_{14}O_{33}$  begins to form at 1100 °C. The graphic image of the structure was constructed according to the obtained atomic coordinates using the ReX Powder diffraction program. Luminescent properties of the phosphors of the composition  $Ca_{12-(0.05+x)}Al_{14}O_{33}$ :Nd<sub>x</sub>Eu<sub>0.05</sub> (x = 0.025-0.1) were investigated. It was determined that the highest intensity of luminescence is achieved at x = 0.05. With a further increase in the content of neodyum, there was a concentration quenching of the luminescence.

## **Graphical Abstract**



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#### Highlights

- Calcium aluminate Ca<sub>12</sub>Al<sub>14</sub>O<sub>33</sub> and phosphors based on it was obtained using sol-gel technology.
- Based on thermal analysis, IR spectroscopy and X-ray phase analysis, scheme of intermediate products transformation during synthesis was proposed.
- The highest intensity of luminescent was reached at a neodymium(III) content of 0.05 mol.

Keywords Calcium aluminate · Sol-gel · Phosphors · Europium · Neodymium.

# 1 Introduction

Until now, white LEDs, which were obtained by combining a blue LED chip and a yellow phosphor based on an yttrium aluminum garnet, had been popular. In recent years, there has been a tendency to replace them with LEDs made on the basis of a combination of ultraviolet LED chips and trichromatic crystal phosphors. Red phosphor shows lower luminescence efficiency compared with blue and green phosphor among trichromatic phosphors. Therefore, at present, the development of red phosphors, which have good photoluminescent characteristics, high thermal and chemical stability, are efficiently excited by UV radiation is important today [1–3].

Alkaline earth metal aluminates have a diverse chemical composition and numerous crystalline structures [4–8]. They are used as matrices for obtaining effective crystal phosphors. Ions of trivalent europium are used as an activator ion. They emit an intense red light corresponding to the transition  ${}^{5}D_{0} \rightarrow {}^{7}F_{2}$  when they are in a non-centrosymmetric environment [9]. The phase composition, crystal structure, and surface morphology of the luminescent matrix strongly depend on the calcination temperature and the ratio of Ca<sup>2+</sup>: Al<sup>3+</sup> in the initial reaction mixture. This has a significant effect on the properties of the final product. It is known that phosphors based on alkaline earth metal aluminates, with the same M<sup>2+</sup>:Al<sup>3+</sup> ratio in equivalent structures, have a radiation intensity that increases in the series Ca  $\rightarrow$  Ba  $\rightarrow$  Sr  $\rightarrow$  Mg-containing phosphors [10, 11].

One of the most promising substances for obtaining effective red phosphors are cubic calcium aluminates of the composition  $Ca_3Al_2O_6$  and  $Ca_{12}Al_{14}O_{33}$ , activated by europium (III) ions [12–16]. The crystal structure of  $Ca_{12}Al_{14}O_{33}$  consists of hollow formations with the inclusion of free oxygen ions. It is prompting suggestions that intensity of the emission of the europium ion will increase as a result of co-activation of neodymium (III) ions [17].

There are many ways to obtain aluminate systems, including the method of solid-phase synthesis, the method of chemical coprecipitation, the method of combustion, sol-gel method. At present sol-gel method is widely used. It is one of the most effective methods for obtaining materials with different properties [18–21]. Sol-gel method mainly differs from the others, it provides a homogeneous mixing

of the starting components at the molecular level. It is achieved by dissolving of salts and oxides of the starting substances. This method allows us to achieve a reduction in energy costs. The product turns out high degree of purity at all stages of synthesis. Products are characterized by monophasic crystalline structure, strictly stoichiometric composition, absence of extraneous phases [22–26].

The purpose of the work was to study the phase formation and the formation of the structure of calcium aluminate of the composition  $Ca_{12}Al_{14}O_{33}$  and the preparation of phosphors based on it activated by europium (III) and neodymium (III) ions.

## 2 Experimental

#### 2.1 Materials preparation

Calcium nitrate tetrahydrate (Vekton, Russia), aluminum nitrate nonahydrate (LLC "NPF Nevsky chemist", Russia) and citric acid monohydrate (LLC "CITROBEL", Russia) was used as initial materials to synthesize  $Ca_{12}Al_{14}O_{33}$ . A molar ratio of Ca:Al:H<sub>4</sub>Cit was 12:14:26. The luminescent materials  $Ca_{12-(0.05+x)}Al_{14}O_{33}$ :Nd<sub>x</sub>Eu<sub>0.05</sub> were obtained by introducing an ions of europium and neodymium. The molar ratio of the initial reagents is shown in Table 1. A certain amount of Eu<sub>2</sub>O<sub>3</sub> and Nd<sub>2</sub>O<sub>3</sub> was dissolved in the nitric acid (BIOMEDHIM, Russia). Salts were dissolved in the minimum amount of distilled water. The resulting clear solutions were mixed together and stirred for 1.5 h using a magnetic stirrer. Yellowish solution was subjected to heat treatment at 130 °C in the drying oven (SNOL 58/350, Lithuania) for 5 h. The resulting xerogel was ground

Table 1 The molar ratio of the initial reagents

Sample	Amount of reagents, mol					
	Nd <sup>3+</sup>	$Ca^{2+}$	Eu <sup>3+</sup>	Al <sup>3+</sup>	C <sub>6</sub> H <sub>8</sub> O <sub>7</sub> ·H <sub>2</sub> O	
1	0	11.95	0.5	14	26	
2	0.025	11.925	0.5	14	26	
3	0.05	11.90	0.5	14	26	
4	0.075	11.875	0.5	14	26	
5	0.1	11.85	0.5	14	26	





and calcined in the muffle furnace (SNOL 6/1300, Lithuania) at different temperature (from 200 to  $1200 \,^{\circ}\text{C}$ ) for 3 h. Scheme of the synthesis is shown in Fig. 1.

#### 2.2 Materials characterization

For researching of processes which proceed at calcium aluminate formation we used a complex of physical and chemical research techniques:

thermal analysis was done using a NETZSCH STA 449C thermoanalyser ("NETZSCH" Germany) at a rate of  $5^{\circ}$ /min in air, the temperature range was from 25 to 1000 °C;

XRD was done using Rigaku MiniFlex 600 (Japan) (Cu K $\alpha$  radiation  $\lambda = 1.5406$  Å) at room temperature with the range of angles  $2\theta$  from 3 to 80°) at the scanning speed of 2°/min. The phase identification was carried out using PDF-2 database. The structure refinement was made by the Rietveld method using the ReX Powder diffraction program [27–29];

the crystal structure model is constructed using the VESTA program [30];

IR spectroscopy was done by IR spectrometer Agilent Technologies Cary 600 Series FTIR Spectrometer (made in USA). Decoding of IR spectra was carried out according to the literature [8, 12, 31, 32];

the luminescent properties were investigated using a spectrofluorometer Solar CM 2203 (Belarus).

# 3 Result and discussion

We defined the main stages of the synthesis process, the structure of prepared powders and the nature of the bonding using Thermal analysis, XRD and IR spectroscopy.



Fig. 2 TG (1), DTG (2), and DSC (3) curves of  $Ca_{12}Al_{14}O_{33}$  precursor

Analysis of TG and DSC curves allowed to see the main stages of formation of calcium aluminate in the temperature range from 25 to  $1000 \,^{\circ}$ C (Fig. 2).

The thermogram shows that some processes overlap each other. At the first stage there is a solvent remove from the surface of the sample, it is accompanied by a small endothermic effect and a small value of activation energy. The activation energy of this process is 58 kJ/mol. It indicates the course of the physical process. Activation energy calculated by Metzger–Horowitz method [25].

At the temperature ranges from 205 to 450 °C there is a sharp decrease of mass of the sample. It is accompanied by a large exothermic effect. The activation energy is 195 kJ/mol. It indicates the course of chemical processes, such as the decomposition of the organic framework. At further temperature increase there is a decomposition of a polymeric complex and oxidation of organic matters. It is accompanied by some exothermic effects.

At temperatures above 700, aluminates are formed. These processes are accompanied by high values of activation energy. The activation energy is about 700 kJ/mol.

We received samples at a temperature of 130, 300, 450, 600, 800, 900, 1000, 1100, and 1200 °C in order to describe in detail the processes that occur at each stage and investigated the received samples using IR spectrometric and X-ray phase analyzes.

Figure 3 shows the IR spectra of  $Ca_{12}Al_{14}O_{33}$  precursors. According to the data of IR spectroscopy, we can say that all the M–O bonds are formed during the maturation process of the solution. In the future, we observe only the growth of the polymer.

At low temperatures from 130 to  $300 \text{ }^{\circ}\text{C}$  in the regions from 1410 to  $1340 \text{ cm}^{-1}$  and from 860 to  $800 \text{ cm}^{-1}$ , there are characteristic bands belonging to the nitrate ion. In



Fig. 3 IR spectra of Ca<sub>12</sub>Al<sub>14</sub>O<sub>33</sub> precursors

samples obtained up to 500 °C, there are bands in the range from 1610 to  $1550 \text{ cm}^{-1}$ . They are associated with stretching vibrations of the carboxyl group in citrate complexes. With increasing the temperature, the vibrations of the carboxyl group and the vibrations of the C–O bond in the carboxyl group disappear. It indicates the decomposition of citrate complexes.

When the initial solution passes into the gel, there is characteristic fluctuation of the O–C–O in  $1074 \text{ cm}^{-1}$ . It is due to the polycondensation process. At temperatures above 900, we can see bands in the low-frequency region from 900 to 500 cm<sup>-1</sup> related to vibrations of the Ca–O–Al bond. This bond forms the basis of the calcium aluminate crystal lattice.

In the samples obtained during annealing in the temperature range from 600 to 800 °C there are bands in the region of  $1450-1410 \text{ cm}^{-1}$ , corresponding to the carbonate ion. X-ray phase data confirm the presence of CaCO<sub>3</sub> in the samples.

The strong broad band absorptions at around 800 cm<sup>-1</sup> are attributed to [AlO<sub>4</sub>] tetrahedra stretching vibration, which are composed by two absorption peaks at 850 cm<sup>-1</sup> for "condensed" AlO<sub>4</sub> and at 773 cm<sup>-1</sup> for "isolated" AlO<sub>4</sub> tetrahedra [33]. Those results demonstrate there are two [AlO<sub>4</sub>] tetrahedral structures in the lattice, which is accordance with the obtained structure in the Ca<sub>12</sub>Al<sub>14</sub>O<sub>33</sub> unit cell [34]. The peaks located 617 cm<sup>-1</sup> and 574 cm<sup>-1</sup> should derive from characteristic vibration absorption of Al–O bonds.

X-ray phase analysis data of the  $Ca_{12}Al_{14}O_{33}$  precursors calcined at 800–1200 °C are presented in Table 2 and in Figs. 4 and 5.

Synthesis temperature, °C	Crystalline phases	Space group	ICDD (PDF-2/ Release 2012	Quantitative content, % (mass.)
800	CaCO <sub>3</sub>	R3c	01-086-2341	100
900	Ca <sub>3</sub> Al <sub>2</sub> O <sub>6</sub>	$Pa\overline{3}$	00-038-1429	50.7
	CaAl <sub>2</sub> O <sub>4</sub>	$P2_1/n$	00-053-0191	49.3
1000	Ca <sub>3</sub> Al <sub>2</sub> O <sub>6</sub>	$Pa\overline{3}$	00-038-1429	34.0
	Ca <sub>5</sub> Al <sub>6</sub> O <sub>14</sub>	$Cmc2_1$	01-070-0801	26.3
	Ca <sub>4</sub> Al <sub>6</sub> O <sub>13</sub>	I4 <del>3</del> m	01-072-0755	20.1
	CaAl <sub>2</sub> O <sub>4</sub>	$P2_1/n$	00-053-0191	19.6
1100	Ca <sub>5</sub> Al <sub>6</sub> O <sub>14</sub>	$Cmc2_1$	01-070-0801	79.2
	Ca <sub>4</sub> Al <sub>6</sub> O <sub>13</sub>	I4 <del>3</del> m	01-072-0755	6.2
	Ca <sub>3</sub> Al <sub>2</sub> O <sub>6</sub>	$Pa\overline{3}$	00-038-1429	7.2
	Ca <sub>12</sub> Al <sub>14</sub> O <sub>33</sub>	$I4\overline{3}d$	01-078-2976	7.4
1200	Ca12Al14O33	$I4\overline{3}d$	01-078-2976	97.0
	Ca <sub>5</sub> Al <sub>6</sub> O <sub>14</sub>	$Cmc2_1$	01-070-0801	2.7
	$Ca_4Al_6O_{13}$	I4 <del>3</del> m	01-072-0755	0.3

Until the temperature of 700 °C, the samples are X-ray amorphous. Heating the precursors at 800 °C for 3 h produces a mixture of crystalline calcium carbonate and amorphous alumina. Noncrystalline alumina is undetect by diffraction methods, therefore evidence of its existence is indirect. Heating at 900 °C gives a mixture of Ca<sub>3</sub>Al<sub>2</sub>O<sub>6</sub> and CaAl<sub>2</sub>O<sub>4</sub>. Mass ratio of aluminates is practically equal. Increasing calcination temperature to 1000 °C, the content of the phases Ca<sub>3</sub>Al<sub>2</sub>O<sub>6</sub> and CaAl<sub>2</sub>O<sub>4</sub> significantly decreases. New phases Ca<sub>5</sub>Al<sub>6</sub>O<sub>14</sub> and Ca<sub>4</sub>Al<sub>6</sub>O<sub>13</sub> appear. The peaks of these aluminates are clearly visible on X-ray diffraction patterns. Ca<sub>12</sub>Al<sub>14</sub>O<sub>33</sub> begins to form at 1100 °C, but the main phase is  $Ca_5Al_6O_{14}$ . The content of  $Ca_4Al_6O_{13}$ and Ca<sub>3</sub>Al<sub>2</sub>O<sub>6</sub> at this temperature significantly decreases. At 1200 °C, Ca<sub>12</sub>Al<sub>14</sub>O<sub>33</sub> becomes the main phase. An insignificant content of Ca5Al6O14 and Ca4Al6O13 is also observed. These intermediates of the synthesis also have been discovered by other researchers [25]. Based on the



Fig. 4 XRD patterns of  $Ca_{12}Al_{14}O_{33}$  precursors calcined at various temperature

**Fig. 5** Graphic representation of the results of structure refinement using the Rietveld method for a sample obtained at 1200 °C

experiment, the following transformation scheme was proposed:

(1) The precursors decompose to  $CaCO_3$  and  $Al_2O_3$ .

(2) Calcium carbonate is decomposed to CaO and  $CO_2$ . It is confirmed by the results of thermal analysis (we can see endothermic effect on the DSC curve and mass reduction on the TG curve):

$$CaCO_3 \rightarrow CaO + CO_2$$

(3) At 900 °C, monoaluminate and tricalcium aluminate are formed:

$$CaO + Al_2O_3 \rightarrow CaAl_2O_4$$
  
 $3CaO + Al_2O_3 \rightarrow Ca_3Al_2O_6$ 

(4) At 1000 °C, monoaluminate and tricalcium aluminate are converted to  $Ca_5Al_6O_{14}$  and  $Ca_4Al_6O_{13}$ . The following transformations are possible:

$$\begin{split} & Ca_3Al_2O_6+CaO+2Al_2O_3 \rightarrow Ca_4Al_6O_{13}\\ & CaAl_2O_4+4CaO+2Al_2O_3 \rightarrow Ca_5Al_6O_{14} \end{split}$$

The formation of these aluminates is most likely due to the similarity of the structures  $Ca_3Al_2O_6$  and  $Ca_4Al_6O_{13}$ , and  $CaAl_2O_4$  and  $Ca_5Al_6O_{14}$ . Although it is possible that other aluminates are formed:

$$\begin{split} & \text{CaAl}_2\text{O}_4 + 3\text{CaO} + 2\text{Al}_2\text{O}_3 \rightarrow \text{Ca}_4\text{Al}_6\text{O}_{13} \\ & \text{Ca}_3\text{Al}_2\text{O}_6 + 2\text{CaO} + 2\text{Al}_2\text{O}_3 \rightarrow \text{Ca}_5\text{Al}_6\text{O}_{14} \end{split}$$

(5) At 1100 °C, Ca<sub>5</sub>Al<sub>6</sub>O<sub>14</sub> becomes the main phase:  $Ca_4Al_6O_{13} + CaO \rightarrow Ca_5Al_6O_{14}$ 



It is possibly:

$$CaAl_2O_4 + 4CaO + 2Al_2O_3 \rightarrow Ca_5Al_6O_{14}$$
$$Ca_3Al_2O_6 + 2CaO + 2Al_2O_3 \rightarrow Ca_5Al_6O_{14}$$

The main product begins to form:

$$2Ca_5Al_6O_{14} + 2CaO + Al_2O_3 \rightarrow Ca_{12}Al_{14}O_{33}$$

It is possibly:

 $Ca_4Al_6O_{13} + 8CaO + 4Al_2O_3 \rightarrow Ca_{12}Al_{14}O_{33}$ 

 $Ca_3Al_2O_6 + 9CaO + 6Al_2O_3 \rightarrow Ca_{12}Al_{14}O_{33}$ 

 $CaAl_2O_4 + 11CaO + 6Al_2O_3 \rightarrow Ca_{12}Al_{14}O_{33}$ 

 $2Ca_4Al_6O_{13}+Ca_3Al_2O_6+CaO\rightarrow Ca_{12}Al_{14}O_{33}$ 

(6) At 1200 °C, cubic  $Ca_{12}Al_{14}O_{33}$  is formed.

 $\begin{aligned} &2\mathrm{Ca}_5\mathrm{Al}_6\mathrm{O}_{14}+2\mathrm{Ca}\mathrm{O}+\mathrm{Al}_2\mathrm{O}_3\to\mathrm{Ca}_{12}\mathrm{Al}_{14}\mathrm{O}_{33}\\ &\mathrm{Ca}_4\mathrm{Al}_6\mathrm{O}_{13}+8\mathrm{Ca}\mathrm{O}+4\mathrm{Al}_2\mathrm{O}_3\to\mathrm{Ca}_{12}\mathrm{Al}_{14}\mathrm{O}_{33}\\ &\mathrm{Ca}_3\mathrm{Al}_2\mathrm{O}_6+9\mathrm{Ca}\mathrm{O}+6\mathrm{Al}_2\mathrm{O}_3\to\mathrm{Ca}_{12}\mathrm{Al}_{14}\mathrm{O}_{33} \end{aligned}$ 

 $2Ca_4Al_6O_{13}+Ca_3Al_2O_6+CaO\rightarrow Ca_{12}Al_{14}O_{33}$ 

 $Ca_4Al_6O_{13}+Ca_5Al_6O_{14}+Ca_3Al_2O_6\rightarrow Ca_{12}Al_{14}O_{33}$ 

With increasing synthesis temperature, the following scheme is implemented:

$$\begin{array}{l} CaAl_2O_4(Ca_3Al_2O_6) \rightarrow Ca_4Al_6O_{13}(Ca_5Al_6O_{14}) \\ \\ \rightarrow Ca_5Al_6O_{14} \rightarrow Ca_{12}Al_{14}O_{33} \end{array}$$

The structure refinement was made by the Rietveld method using the ReX Powder diffraction program. The calculation results showed that the space group I4 $\overline{3}$ d, a = 12.043 Å, Rp = 0.168136, Rexp = 0.168202, and Rwp = 0.223990. The graphic image of the structure was constructed according to the obtained atomic coordinates.

 $Ca_{12}Al_{14}O_{33}$  has a complex structure composed of hollow spheres (Fig. 6). It possesses a framework composed of aluminum-oxygen tetrahedra [AlO<sub>4</sub>]. Tetrahedra are interconnected through oxygen atoms and partially through bridging calcium cations, forming eight-membered rings. The structure includes one type of calcium atoms (Ca1), two types of aluminum atoms (Al1 and Al2) in a tetrahedral environment of oxygen and three types of oxygen (O1, O2, and O3). They are distinguished by their surroundings. O1 are coordinated by the aluminum atom Al2; O1 and O2 are



Fig. 6 The crystal structure of  $Ca_{12}Al_{14}O_{33}$  (the blue ones are Ca atoms, the green ones are Al atoms, the red ones are bound O atoms, the white ones are unbound O atoms)



Fig. 7 Spectra of excitation of phosphors  $Ca_{12-(0.05+x)}Al_{14}O_{33}$ :  $Nd_xEu_{0.05}$  ( $\lambda_{em} = 618 \text{ nm}$ ) (n(Nd<sup>3+</sup>), mol: 1-0; 2-0.025; 3-0.05; 4-0.075; 5-0.1

coordinated by the aluminum atom Al1. They form aluminum-oxygen tetrahedra, herewith tetrahedron containing O2 is distorted. O3 form anions of unbound oxygen. Such anions  $O2^{2-}$  are located inside hollow spheres.

Phosphors based on  $Ca_{12}Al_{14}O_{33}$ , activated by europium (III) and neodymium (III) ions were obtained at a temperature of 1200 °C.

We obtained excitation (Fig. 7) and photoluminescence (Fig. 8) spectra for samples with different content of neodymium in  $Ca_{12}Al_{14}O_{33}$  (n(Nd<sup>3+</sup>): 1—0; 2—0.025; 3— 0.05; 4—0.075; 5—0.1 mol).

An intense band with  $\lambda_{max} = 250-260$  nm is observed in the short-wave region of the excitation spectrum. It indicates the transfer of excitation energy from the O<sup>2-</sup> ion to



Fig. 8 Spectra of radiation of phosphors  $Ca_{12-(0.05+x)}Al_{14}O_{33}$ :  $Nd_xEu_{0.05}$  ( $\lambda_{ex} = 254$  nm) (n(Nd<sup>3+</sup>), mol: 1-0; 2-0.025; 3-0.05; 4-0.075; 5-0.1

the Eu<sup>3+</sup> ion. Therefore, we used an excitation wavelength of 254 nm in obtaining luminescence spectra. In the luminescence spectra of phosphors containing europium(III) and neodymium(III) ions, several narrow bands with a wavelength of 618 nm are observed. It corresponds to the red region of the spectrum and indicates the occurrence of europium(III) ion in the structure of calcium aluminate in the trivalent state. The luminescence spectrum consists of characteristic narrow emission lines of the Eu<sup>3+</sup> ion, corresponding to the transitions of electrons inside the 4f shell. The most intense is the induced electric dipole transition  ${}^{5}D_{0}-{}^{7}F_{2}$  with a maximum at a wavelength of 618 nm, then  ${}^{5}D_{0}-{}^{7}F_{4}$  with a wavelength of 704 nm. The magnetic dipole transition  ${}^{5}D_{0}-{}^{7}F_{1}$  has the lowest intensity ( $\lambda_{max} = 595$  nm) [17].

The presence of neodymium(III) ions in the structure of calcium aluminate affects the radiation intensity of luminescence centers, which include europium(III). The highest intensity is reached at a neodymium content of 0.05 mol. With increasing concentration of the Nd<sup>3+</sup> ion, concentration quenching occurs and the radiation intensity decreases.

## 4 Conclusion

Calcium aluminate with the composition  $Ca_{12}Al_{14}O_{33}$  and phosphors based on it, containing europium(III) (0.05 mol) and neodymium(III) (from 0.025 to 0.1 mol) were obtained using sol–gel technology.

The main stages of the formation of calcium aluminate were determined using thermal analysis, IR spectroscopy and X-ray phase analysis. Solid-phase interaction in the formation of  $Ca_{12}Al_{14}O_{33}$  includes the steps of obtaining calcium carbonate and amorphous alumina (800 °C),

aluminates of composition  $Ca_3Al_2O_6$  and  $CaAl_2O_4$  (900–1000 °C), and  $Ca_5Al_6O_{14}$  and  $Ca_4Al_6O_{13}$  (1000–1100 °C).  $Ca_{12}Al_{14}O_{33}$  begins to form at 1100 °C.

The graphic image of the structure was constructed according to the obtained atomic coordinates using the ReX Powder diffraction program.

Luminescent properties of the phosphors of the composition  $Ca_{12-(0.05+x)}Al_{14}O_{33}:Nd_xEu_{0.05}$  (x = 0.025-0.1) were investigated. It was determined that the highest intensity of luminescence is achieved at x = 0.05. With a further increase in the content of neodyum there is a concentration quenching of the luminescence.

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#### **Compliance with ethical standards**

**Conflict of interest** The authors declare that they have no conflict of interest.

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## References

- Li H, Noh HM, Moon BK, Choi BC, Jeong JH, Jang K, Lee HS, Yi SS (2013) Wide-band excited Y<sub>6</sub>(W, Mo)<sub>0.5</sub>O<sub>12</sub>:Eu red phosphor for white light emitting diode: structure evolution, photoluminescence properties, and energy transfer mechanisms involved. Inorg Chem 52(19):11210–11217
- Li YT, Liu XH (2014) Structure and luminescence properties of Ba<sub>3</sub>WO<sub>6</sub>:Eu<sup>3+</sup> nanowire phosphors obtained by conventional solid-state reaction method. Opt Mater 38:211–216
- Jin C, Ma HX, Liu QB, Li X, Liu PF (2014) Photoluminescence properties of the high-brightness Eu<sup>3+</sup>-doped KNaCa<sub>2</sub>(PO<sub>4</sub>)<sub>2</sub> phosphors. Spectrochim Acta A 122:767–771
- Singh V, Borkotoky S, Murali A, Rao JL, Gundu Rao TK, Dhoble SJ (2015) Electron paramagnetic resonance and photoluminescence investigation on ultraviolet-emitting gadoliniumion-doped CaAl<sub>12</sub>O<sub>19</sub> phosphors. Spectrochim Acta A 139:1–6
- Puchalska M, Sobczyk M, Targowska J, Watras A, Zych E (2016) Infrared and cooperative luminescence in Yb<sup>3+</sup> doped calcium aluminate CaAl<sub>4</sub>O<sub>7</sub>. J Lumin 143:503–509
- Pakhnutova EA, Mishenina LN, Selyunina LA, Belyaninova TV, Slizhov YG (2018) Nitrate-citrate sol-gel synthesis of hydrated calcium aluminate and sorption materials on its basis. Russ J Appl Chem 91(6):899–907
- Genga G, Myersa RJ, Yud YS, Shapirod DA, Winarskie R, Levitzf PE, Kilcoyned DAL, Paulo JM (2018) Synchrotron X-ray nanotomographic and spectromicroscopic study of thericalcium aluminate hydration in the presence of gypsum. Cement and Concrete Research 111:130–137
- Pan X, Zhang D, Wu Y, Yu H (2018) Synthesis and characterization of calcium aluminate compounds from gehlenite by high-temperature solid-state reaction. Ceram Int 44:13544– 13550
- 9. Hussain A, Mehmood S, Rasool MN, Aryal, Rulis P, Ching WY (2016) Electronic structure, mechanical, and optical properties of

CaO·Al\_2O\_3 system: a first principles approach. Indian J Phys  $8{:}917{-}929$ 

- Park K, Pi JW, Hakeem DA (2016) Effect of alkaline metal ions on the photoluminescence properties of Eu<sup>3+</sup>-doped Ca<sub>3</sub>Al<sub>2</sub>O<sub>6</sub> phosphors. J Rare Earths 34(12):1193–1198
- Liang CJ, Siao HY (2016) Luminescence behaviors of Eu- and Dycodoped alkaline earth metal aluminate phosphors through potassium carbonate coprecipitation. Mater Chem Phys 177:429–436
- 12. Liang CJ, Siao HY (2017) Effects of calcining temperatures of Eu<sup>2+</sup> and Dy<sup>3+</sup> ion-codoped calcia-alumina binary compounds on their phase transition and luminescence properties. Mater Chem Phys 193:117–128
- 13. Yadav RV, Yadav RS, Bahadur A, Rai SB (2016) Down shifting and quantum cutting from  $Eu^{3+}$ ,  $Yb^{3+}$  co-doped  $Ca_{12}Al_{14}O_{33}$ phosphor: a dual mode emitting material. RSC Adv 6:9049–9056
- 14. Rashad MM, Mostafa AG, Rayan DA (2016) Structural and optical properties of nanocrystalline mayenite  $Ca_{12}Al_{14}O_{33}$  powders synthesized using a novel route. J Mater Sci: Mater Electron 27:2614–2623
- 15. Gao X, Lei L, Lv C, Sun Y, Zheng H, Cui Y (2008) Preparation and photoluminescence property of a loose powder Ca<sub>3</sub>Al<sub>2</sub>O<sub>6</sub>:Eu<sup>3+</sup> by calcination of a layered double hydroxide precursor. J Solid State Chem 181:1776–1781
- Barros BS, de Oliveira RS, Kulesza J, Melo VRM, Melo DMA, Alves JrS (2015) Reddish-orange Ca<sub>3-x</sub>Al<sub>2</sub>O<sub>6</sub>:<sub>x</sub>Eu<sup>3+</sup> nanophosphors: Fast synthesis and photophysical properties. J Phys Chem Solids 78:90–94
- Zhang J, Zhangb Z, Wanga T, Hao W (2003) Preparation and characterization of a new long afterglow indigo phosphor Ca<sub>12</sub>Al<sub>14</sub>O<sub>33</sub>:Nd,Eu. Mater Lett 57:4315–4318
- Bakhmetyev VV, Minakova TS, Mjakin SV, Sychov MM, Ringuede A (2016) Synthesis and surface characterization of nanosized Y<sub>2</sub>O<sub>3</sub>:Eu and YAG:Eu luminescent phosphors which are useful in photodynamic therapy of cancer. Eur J Nanomed 8 (4):173–184
- Mal'chik AG, Kuznetsova SA, Kryuchkova SO, Kozik VV (2017) Sol-gel synthesis of Ta<sub>2</sub>O<sub>5</sub>–SiO<sub>2</sub> composites from tantalum(V) chloride and tetraethyl orthosilicate in ethanol. Inorg Mater 53 (9):994–1003
- Agliullin MR, Grigor'eva NG, Danilova IG, Magaev OV, Vodyankina OV (2015) Template-free sol-gel synthesis of catalytically active mesoporous aluminosilicates. Kinet Catal 56(4):501–508

- Mishenina LN, Selyunina LA, Botvina TM (2015) Synthesis and characteristics of CaAl<sub>2</sub>O<sub>4</sub>:Eu<sup>3+</sup> phosphor prepared by zol-gel method. Key Eng Mater 670:95–100
- Aitasalo T, Hölsä J, Jungner H et al. (2002) Comparison of sol-gel and solid-state prepared Eu<sup>2+</sup> doped calcium aluminates. J Mater Sci 20(1):15–20
- Mandić V, Kurajica S (2015) The influence of solvents on sol–gel derived calcium aluminate. Mater Sci Semicond Process https:// doi.org/10.1016/j.mssp.2015.01.004
- Choi SW, Hong H (2010) Size and morphology control by planetary ball milling in CaAl<sub>2</sub>O<sub>4</sub>:Eu<sup>2+</sup> phosphors prepared by Pechini method and their luminescence properties. J Mater Sci Eng B 171:69–72
- Yuan X, Xu YB, He Y (2007) Synthesis of Ca<sub>3</sub>Al<sub>2</sub>O<sub>6</sub> via citric acid precursor. J Mater Sci Eng, A 447:142–145
- Botvina TM, Botvin VV, Mishenina LN, Selyunina LA (2018) Synthesis of Calcium Aluminate-Based Luminophores by the Citrate Nitrate Sol–Gel Process. Russ J Inorg Chem 63(10):1262–1267
- Bortolotti M, Lonardelli I (2013) ReX.Cell: a user-friendly program for powder diffraction indexing. J Appl Cryst 46:259–261
- Boultif A, Louër D (2004) Powder pattern indexing with the dichotomy method. J Appl Cryst 37:724–731
- Visser JW (1969) A fully automatic program for finding the unit cell from powder data. J Appl Cryst 2:89–95
- Momma K, Izumi F (2011) VESTA 3 for three-dimensional visualization of crystal, volumetric and morphology data. J Appl Crystallogr 44:1272–1276
- Wang R, Yang H, Lu Y, Kanamori K, Nakanishi K, Guo X (2017) Synthesis, reduction, and electrical properties of macroporous monolithic mayenite electrides with high porosity. ACS Omega 2:8148–8155
- 32. Liu1 R, Yan Y, Ma Ch (2018) Self-propagating combustion synthesis, luminescent properties and photocatalytic activities of pure  $Ca_{12}Al_{14}O_{33}$ :Tb<sup>3+</sup>(Sm<sup>3+</sup>). https://doi.org/10.3389/fchem. 2018.00069
- Tolkacheva AS, Shkerin SN, Plaksin SV, Vovkotrub EG, Bulanin KM, Kochedykov VA, Ordinartsev DP, Gyrdasova OI, Molchanova NG (2011) Synthesis of dense ceramics of single-phase mayenite (Ca<sub>12</sub>Al<sub>14</sub>O<sub>32</sub>)O. Russ J Appl Chem 84(6):907–911
- 34. Boysen H, Lerch M, Stysb A, Senyshyn A (2007) Structure and oxygen mobility in mayenite ( $Ca_{12}Al_{14}O_{33}$ ): a high-temperature neutron powder diffraction study. Acta Cryst B63:675–682