

# Synthesis and luminescent properties of monodisperse $\text{SiO}_2@ \text{SiO}_2: \text{Eu}(\text{DBM})_3\text{phen}$ microspheres with core-shell structure by sol-gel method

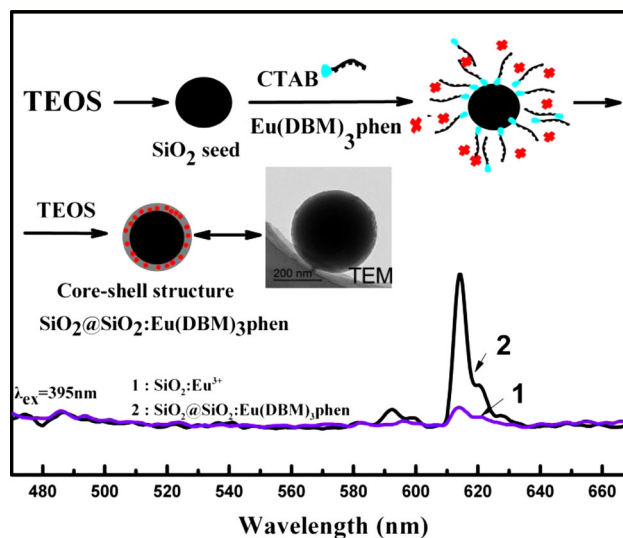
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**Abstract** Monodisperse core-shell structured  $\text{SiO}_2@ \text{SiO}_2: \text{Eu}(\text{DBM})_3\text{phen}$  microspheres, as a new kind of fluorescent particles, were fabricated by a seeded growth method in the presence of surfactant CTAB (DBM = dibenzoylmethane, phen = 1,10-phenanthroline, CTAB = cetyltrimethyl ammonium bromide). In this way, a thin mesoporous silica shell doped with  $\text{Eu}(\text{DBM})_3\text{phen}$  was grown on the prepared monodisperse silica colloids. The structures of the as-prepared phosphors were characterized by means of X-ray diffraction, Fourier transform infrared spectrum, thermal gravimetric analysis–differential scanning calorimetry, field emission scanning electron microscopy, transmission electron microscopy, energy-dispersive X-ray spectrometry, nitrogen adsorption-desorption techniques, and photoluminescence spectra. The experimental results show that the microsphere has a typical core-shell structure with a diameter of about 370 nm, consisting of the silica colloids core with about 340 nm in diameter and mesoporous silica shell doped with europium complexes with an average thickness of about 15 nm. The as-obtained core-shell microspheres exhibit a strong red emission peak originating from electric-dipole transition  $^5\text{D}_0 \rightarrow ^7\text{F}_2$  (610 nm) of  $\text{Eu}^{3+}$  ions under the excitation of 395 nm. Additionally, the photoluminescence intensity increases with the increasing of europium complexes concentration in mesoporous silica shell, and concentration quenching occurs when europium complexes concentration exceeds 7.0 mol%. The prepared microspheres may present potential applications in the fields

of optoelectronic devices, bio-imaging, medical diagnosis, and the packing structural functional composites.

**Graphical Abstract** Monodisperse core-shell structured  $\text{SiO}_2@ \text{SiO}_2: \text{Eu}(\text{DBM})_3\text{phen}$  microspheres with high luminescent performance were fabricated by a seeded growth method by means of surfactant CTAB.



**Keywords** Core-shell structure ·  $\text{SiO}_2@ \text{SiO}_2: \text{Eu}(\text{DBM})_3\text{phen}$  microspheres · Seeded growth method · Luminescence · Rare earth complexes

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## 1 Introduction

Rare earth (RE) composite materials played a critical role in the luminescent materials area, having been widely used in

novel photovoltaic materials, display technique, optical detection, and bio-medical imaging [1–5], as a result of their optical, electronic, and chemical features arising from the 4 f electrons of RE ions. For the past decades, a variety of researches have been employed for preparing RE composite materials which are mainly concentrated in the matrix of oxide [6, 7], sulfide [8], aluminate [9], silicate [10], and carbonate [11]. Among all kinds of these phosphors, silica-based fluorescent materials, especially the spherical silica-based phosphors are considered as excellent candidates that provide definite fluorescent emission signals, low light scattering, good photochemical stability and biocompatibility, making them suitable for drug tracing, biological label, display and microscopic analysis of composite materials [12–16]. However, it is difficult to prepare RE-doped monodispersed silica spheres by the traditional sol-gel method. The reason is that RE ions are usually rapidly precipitated in basic environments, leading to lanthanide incorporation inconsistent with silica particles, while the acid-catalyzed hydrolysis of Tetraethyl orthosilicate (TEOS) generally results in uncontrollable with sizes and size distributions of large particles [17]. In recent years, some methods have been utilized to prepare RE-doped silica spheres, Liu et al. [18] obtained europium doped silica microspheres using the W/O microencapsulation method with the assistance of the surfactant 3-aminopropyltriethoxysilane. Dood et al. [19] and Gong et al. [17] used alkali-catalyzed method to produce uniform sized SiO<sub>2</sub> core, then used acid-catalyzed method to grow SiO<sub>2</sub> shell doped with RE ions in the kernel growth. However, as the low fluorescence intensity of single RE ions doped materials, high doping content of RE ions is needed, which may be limited by the expensive price of RE ions and the occurrence of concentration quenching effect. So it is necessary to replace the RE ions with RE complexes.

RE complexes are ideal optical materials because of their excellent fluorescent radiation, narrow half-peak breadth, high color purity, and large Stokes shift. When the SiO<sub>2</sub> gel was doped with RE complexes, it can not only improve the stability and biocompatibility of the RE complexes but also get high fluorescence intensity [20]. Nevertheless, it is also difficult to get RE complex-doped monodispersed silica spheres by the Stöber method, as the RE complexes colloidal particles are easy to reunite and have poor solubility in the environment of alcohol-water system, which makes it is hard to incorporate the RE complexes into the microspheres. At present, methods for making spherical silica hybrid materials based on RE complexes mainly include chemical immobilization and forming core-shell structure. Zhang et al. [21] immobilized Lanthanide (III)-imidazole dicarboxylic acid complexes on colloidal mesoporous silica with diameter smaller than 100 nm by covalent bond grafting technique and obtained functionalized silica.

Zhao et al. [22–24] utilized the solubility difference of RE complexes and composite microspheres in acetone solution and employed the improved Stöber method to get the composite microspheres with core-shell structure. Yu et al. [25] fabricated sphere-shape Eu(DBM)<sub>3</sub>Phen@SiO<sub>2</sub> nanoparticles by employing a modified alkaline catalyzed hydrolysis and precipitation method and assistance of PVP. However, it is hard to obtain the particles with good sphericity and monodispersity prepared as described above, and the particle size are difficult to be controlled. What's more, it will consume large amounts of Re complexes when it acts as the core of the core-shell composites.

In this paper, we prepared monodisperse spherical silica particles, then a continuous europium complexes-doped mesoporous silica shell was coated on the surface of the silica core via a modified alkaline catalyzed hydrolysis and precipitation method. The core-shell structure monodisperse SiO<sub>2</sub>@SiO<sub>2</sub>:Eu(DBM)<sub>3</sub>phen microspheres were obtained. The microspheres have the following advantages: (1) the fluorescent signal has been enhanced a lot and photostability has been increased by encapsulating the hydrophobic europium complexes into hydrophilic silica shell by means of surfactant cetyltrimethyl ammonium bromide (CTAB); (2) the particles have good sphericity and homogeneous particle size for the monodisperse silica spheres acting as the seeds; (3) good sphericity and narrow diameter particles have been obtained by a modified sol-gel method in a low doping content of RE complexes which can reduce the cost of the phosphors to some extent; (4) the impact on hydroxyl activity of the surface of SiO<sub>2</sub> for the incorporation of RE complexes has been reduced as far as possible which can keep a high activity for the phosphors. The as-obtained core-shell structured phosphors show much stronger red light emission under ultraviolet light excitation compared with the phosphors doped with single RE ions, and it may have potential applications in the fields of optoelectronic devices, bioimaging, medical diagnosis, and study on the structure of functional composites. Furthermore, the possible formation mechanism of the architecture was discussed in detail.

## 2 Experimental

### 2.1 Materials and methods

TEOS, ammonium hydroxide (25 wt%), acetone, Cetyltrimethyl Ammonium Bromide (CTAB), 1,10-phenanthroline (phen) were obtained from Chengdu Kelong Chemical Co, Ltd. Eu(NO<sub>3</sub>)<sub>3</sub>·6H<sub>2</sub>O (99.99%) and dibenzoyl methane (DBM) were purchased from Aladdin Reagent Co. All chemicals were analytical grade reagents and used directly without further purification.

## 2.2 Synthesis of rear earth complexes Eu(DBM)<sub>3</sub>phen

The RE complexes Eu(DBM)<sub>3</sub>phen was synthesized as report [26], 1 mmol Eu(NO<sub>3</sub>)<sub>3</sub> was dissolved in ethanol to form Eu(NO<sub>3</sub>)<sub>3</sub> ethanol solution (solution a). 3 mmol dibenzoylmethane (DBM), 1 mmol phen and 1 mmol NaOH were dissolved in ethanol to form settled solution (solution b). Then solution a was dripped slowly to solution b, following by keeping stirring and adding NaOH ethanol solution to adjust the pH to 6–7, reacting for 2 h in a water bath at 50 °C, then aging for 3 h at room temperature. After filtrating, repeated washing with ethanol and drying in the air, the yellow powders of RE complexes were obtained.

## 2.3 Preparation of the silica spheres

Monodisperse silica spheres were obtained by the classical Stöber method [27]. As the typical synthesis process, 4 ml TEOS was dropped slowly into a mixture that consisted of 60 ml ethanol, 30 ml H<sub>2</sub>O, and 6 ml ammonia under stirring. After additional reaction for 30 min, the obtained white sol solution was centrifuged and washed with anhydrous ethanol and deionized water for several times, respectively. Finally, the samples were dried at 60 °C for 24 h in air.

## 2.4 Preparation of the SiO<sub>2</sub>@SiO<sub>2</sub>:Eu(DBM)<sub>3</sub>phen microspheres

The monodisperse SiO<sub>2</sub>@SiO<sub>2</sub>:Eu(DBM)<sub>3</sub>phen microspheres in a seeded growth way were synthesized in two steps. In the first step, silica seeds were prepared as above: 4 ml TEOS was rapidly dropped into a mixture that consisted of 60 ml ethanol, 30 ml H<sub>2</sub>O, and 6 ml ammonia under stirring, and then agitated them for 30 min. In the second step, 16 ml anhydrous ethanol, 7 ml distilled water, and 2 ml ammonia were added rapidly into the above mixture under continuous stirring. After additional agitation for 15 min at room temperature, 160 mg CTAB and different ratios of Eu(DBM)<sub>3</sub>phen acetone solution (the molar composition of Eu(DBM)<sub>3</sub>phen:TEOS (secondary addition) was from 1:100 to 9:100) were added slowly (the speed of dropping was about 0.2 ml/min) into the mixture which was sonicating by a sonicator. Then the mixture was sonicated for another 30 min at 600 W. Finally, some amount of TEOS was added slowly and agitated for 5 h to form the final sample. The samples were separated by centrifugation at 6000 r.p.m. for 5 min, washed with anhydrous ethanol, deionized water, and acetone for three times, respectively. The obtained samples were dried at 60 °C for 24 h in air. In order to remove residuary CTAB, the as-dried precursors were calcined at 350 °C for 3 h in nitrogen atmosphere. The heating rate was 20 °C/min.

## 2.5 Characterization methods

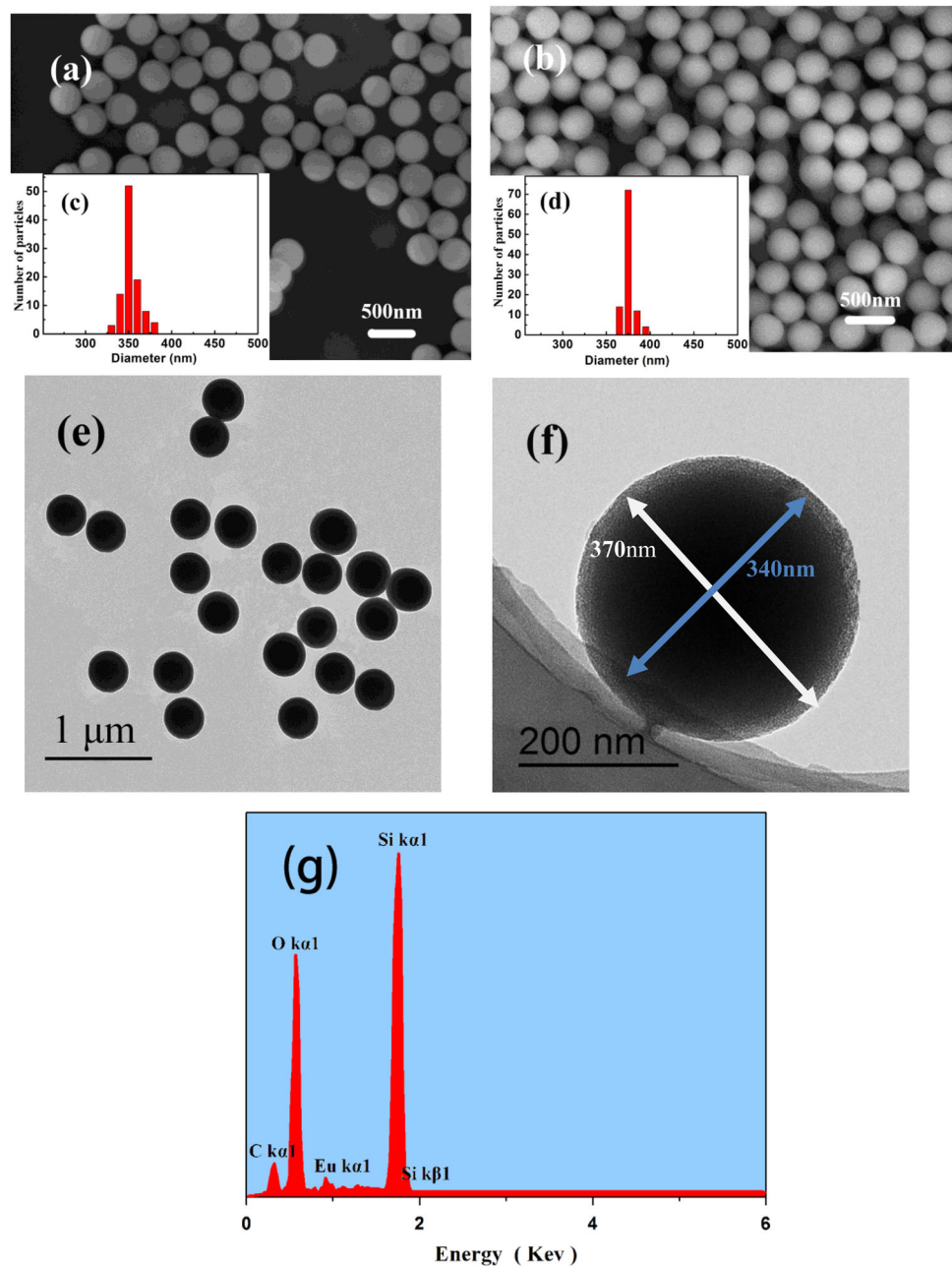
The crystal structure and phase purity of the as-prepared samples was examined by X-ray diffraction (XRD) (PANalytical B.V. X'Pert-PRO) using a XRD system with Cu-K $\alpha$  radiation ( $\lambda = 0.15406$  nm), working voltage of 35 kV and current of 60 mA. The morphology of the as-synthesized phosphors was investigated by using field emission scanning electron microscope (FESEM) (TESCAN MAIA3 SEM) equipped with an energy-dispersive spectrometer and transmission electron microscopy (TEM) (Zeiss Libra 200FE). The Photoluminescence (PL) excitation and PL emission spectra of rear earth complexes and SiO<sub>2</sub>@SiO<sub>2</sub>:Eu(DBM)<sub>3</sub>phen microsphere were measured at room temperature by using the luminescence spectrophotometer (Model F-4500, Hitachi) with a 150 W xenon lamp as the excitation source. Fourier transform infrared spectrum (FT-IR) spectra were measured with a Perkin-Elmer Spectrum One infrared spectrophotometer with the KBr pellet technique. Thermal gravimetric analysis and differential scanning calorimetry (TGA-DSC) data were recorded with a thermal analysis instrument (TGA/SDTA 851e, METTLER TOLEDO, Switzerland) at a heating rate of 20 °C/min in an nitrogen flow of 100 mL/min. Nitrogen adsorption-desorption isotherms were obtained at 77 K on a Micromeritics ASAP NOVA 3000 analyzer. All the measurements were performed at room temperature

## 3 Results and discussion

### 3.1 Morphology and structural analysis

The morphology and particle size distribution of the as-obtained products were investigate by SEM and TEM. Figures 1a–b show the FESEM of silica spheres synthesized by the Stöber method and core-shell SiO<sub>2</sub>@SiO<sub>2</sub>:Eu(DBM)<sub>3</sub>phen microspheres prepared by seeded growth method. As shown in Fig. 1a, the obtained silica spheres are well-dispersed and uniform with a diameter of ~350 nm, and their surfaces are very smooth. Similarly, the SiO<sub>2</sub>@SiO<sub>2</sub>:Eu(DBM)<sub>3</sub>phen particles formed in seeded growth way (shown in Fig. 1b) are still spherical and monodispersed, which are mainly because of the monodispersed silica spheres acting as as a growth intermediate and the secondary addition TEOS hydrolyzed into Si(OH)<sub>4</sub> depositing onto the surface of silica seeds [28]. While the average size of these particles become slightly larger than the silica spheres (as shown in Fig. 1c), and the particles are all non-aggregated with narrow size distribution (as shown in Fig. 1d) with an average size of ~370 nm, from which we can preliminarily conclude that a thin shell of silica which

**Fig. 1** The morphology and particle size distribution of the as-obtained products. **a** SEM image of silica spheres seeds; **b** SEM image of  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3\text{phen}$  microspheres; **c** histograms of size distribution of silica spheres seeds; **d** histograms of size distribution of  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3\text{phen}$  microspheres; **e** TEM photograph of  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3\text{phen}$  microspheres in low magnification; **f** TEM image in high magnification; **g** EDX spectrum of 5.0 mol% Eu ( $\text{DBM})_3\text{phen}$ -doped  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3\text{phen}$  microspheres



may be doped with  $\text{Eu}(\text{DBM})_3\text{phen}$  have been coated onto the surface of silica seeds by the assistance of CTAB.

To further understand the structure of  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3\text{phen}$  microspheres, TEM observations were also performed. Figures 1e–f shows the TEM images of different magnification of core-shell  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3\text{phen}$  microspheres prepared in seeded growth way, respectively. It can be observed from Fig. 1e and 1f that the  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3\text{phen}$  particles exhibit regular sphericity and obvious core-shell structure that is composed of the black  $\text{SiO}_2$  seeded cores with about 340 nm in diameter and gray shell with average thickness of about 15 nm. Moreover, we can find that the surface of  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3\text{phen}$

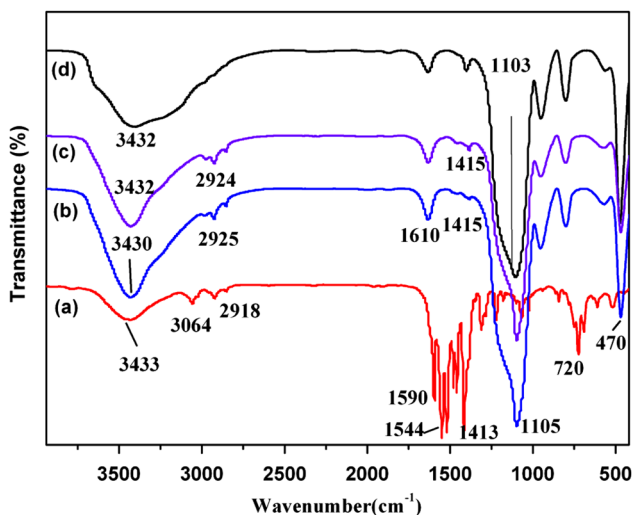
microspheres are rougher than that of pure silica spheres due to poor soakage of  $\text{Eu}(\text{DBM})_3\text{phen}$  fluorescent complexes on the surface of the  $\text{SiO}_2$  seeds. TEM indicated that the outer layer of the microspheres is a thin mesoporous silica shell as the bad compatibility between the inner silica layer and the outer silica layer which may caused by the CTAB act as template agent [29].

In order to study the element composition of the final product, the as-prepared  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3\text{phen}$  microspheres were analyzed by Energy-dispersive spectrometer and the results are shown in Fig. 1g. The results confirm the presence of carbon (C), silicon (Si), oxygen (O), and europium (Eu) elements in the microspheres, and there



are not any other peaks of impurity elements detected, indicating that the amorphous precursor compound has been converted to  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3$  and CTAB have been removed completely during the calcination process, which can give further support for the TEM analysis above.

The FT-IR spectra of pure earth complexes  $\text{Eu}(\text{DBM})_3\text{phen}$ , pure silica spheres core-shell  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3\text{phen}$  particles before and after calcination are shown in Fig. 2a–d, respectively. In the Fig. 2a, 1400–1600  $\text{cm}^{-1}$  absorption peak are stretching vibration absorption peak of C–O, C–C, and C–N, and 3064 and 2918  $\text{cm}^{-1}$  are the stretching vibration absorption peak of CH–CH unit of phenyl derivatives, which are all the characteristic absorption peaks of organic ligand  $\beta$ -diketones and phen, the peak of 1413  $\text{cm}^{-1}$  is the stretching vibration absorption peak of C=N which comes from phen, suggesting that earth complexes  $\text{Eu}(\text{DBM})_3\text{phen}$  are obtained successfully. For the Fig. 2b–d, there are the typical absorption bands of Si–O–Si (1105  $\text{cm}^{-1}$ , 805  $\text{cm}^{-1}$ ), Si–OH (470  $\text{cm}^{-1}$ ) and –OH

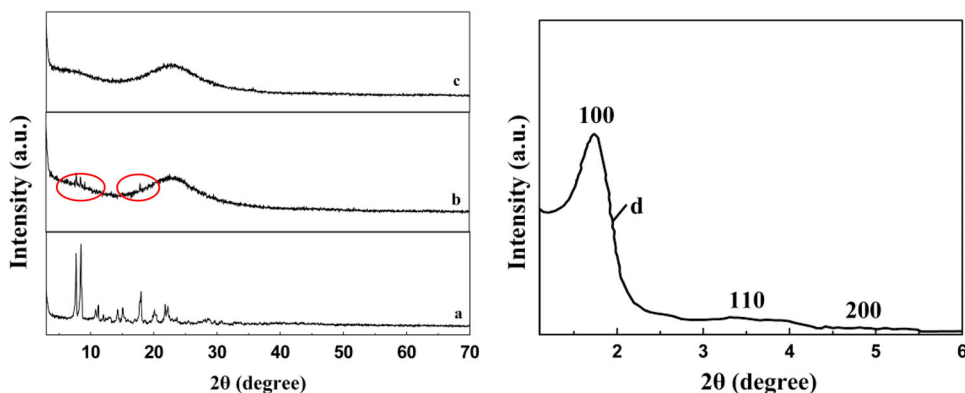


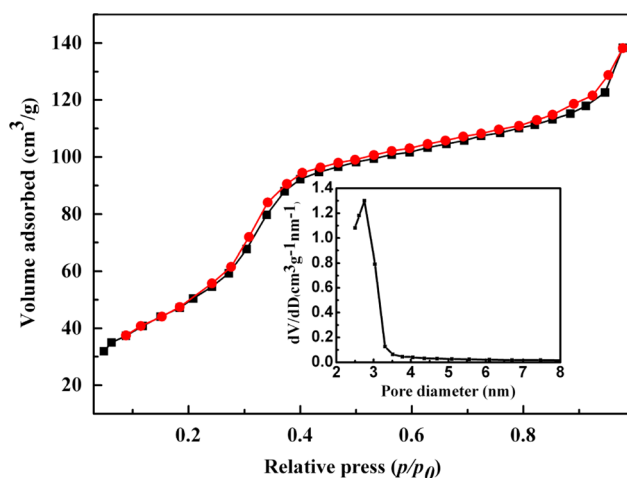
**Fig. 2** FT-IR spectra of pure earth complexes  $\text{Eu}(\text{DBM})_3\text{phen}$  a, 5.0 mol%  $\text{Eu}(\text{DBM})_3\text{phen}$ -doped core-shell  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3\text{phen}$  particles before b and after calcination c, and pure silica spheres d

vibration peak near 3430 and 1610  $\text{cm}^{-1}$ . Meanwhile, compared with Fig. 2d, it is easy to find in Fig. 2b and c that weak vibration peaks appear in 1400–1600  $\text{cm}^{-1}$  which are C=O, C=C, C=N stretching vibration peaks of  $\beta$ -diketone and phen, and the peaks of 2925  $\text{cm}^{-1}$  are the stretching vibration absorption peak of CH–CH unit of phenyl derivatives and CTAB, indicating that the surface of silica seeds are covered with a layer of  $\text{SiO}_2$  shell which is doped with  $\text{Eu}(\text{DBM})_3\text{phen}$  successfully. However, there are few differences between Fig. 2b and 2c, as the content of CTAB is low and the similarities of FT-IR spectra between CTAB and  $\beta$ -diketones. Through the FT-IR spectra of Fig. 2a–d, it can be indirectly indicated that the microspheres which are doped with RE complexes were obtained.

Figure 3 shows the X-ray diffraction patterns of the pure RE complexes, core-shell  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3\text{phen}$  particles formed in seed growth way and pure silica spheres prepared via base-catalyzed Stöber method, respectively. The peak positions of the  $\text{Eu}(\text{DBM})_3\text{phen}$  well match with the literatures [30, 31], the characteristic peaks of the complexes lie at  $2\theta \sim 8.2^\circ$ ,  $8.7^\circ$ , and  $17.4^\circ$  positions in which the first two are mainly due to  $\text{Eu}^{3+}$  ions and the later one is the new peak which may be generated due to increased cell size. For pure silica spheres and core-shell  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3\text{phen}$  particles, the position of band center are around  $23^\circ$ , which is the amorphous  $\text{SiO}_2$  characteristic pattern. Careful view shows that there are some weak peaks (marked with red circles in Fig. 3b) appeared in the XRD of  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3\text{phen}$  particles, which are attributed to RE complexes as shown in Fig. 3a, indicating that  $\text{Eu}(\text{DBM})_3\text{phen}$  have been incorporates into  $\text{SiO}_2$  shell successfully. The low-angle XRD pattern of  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3\text{phen}$  microspheres have been shown in Fig. 3d, the microspheres show a tense peak at  $2\theta = 1.89^\circ$ , revealing that the  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3\text{phen}$  microspheres have ordered hexagonal mesopore symmetry, which is due to the mesopore channels are perpendicular to the surface of silica seeds, ethanol can lower the hydrolysis

**Fig. 3** XRD patterns of the pure RE complexes a, 5.0 mol%  $\text{Eu}(\text{DBM})_3\text{phen}$ -doped core-shell  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3\text{phen}$  particles b, pure silica spheres c and low-angle XRD pattern of  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3\text{phen}$  microspheres d



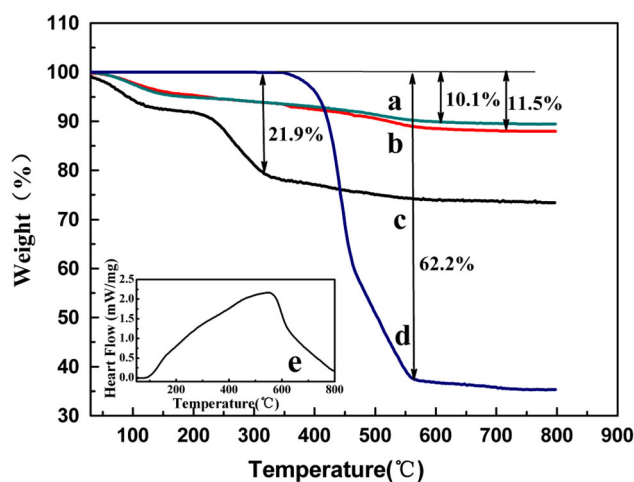


**Fig. 4**  $N_2$  adsorption-desorption isotherms and mesopore size distribution (the inset) of the synthesized  $SiO_2@SiO_2:Eu(DBM)_3phen$  microspheres. The adsorption branch is shown in black color and the desorption branch in red color

and condensation rate of TEOS and favor the coating and curving [32].

The nitrogen adsorption-desorption isotherms of  $SiO_2@SiO_2:Eu(DBM)_3phen$  microspheres and the corresponding pore size distribution analyses are shown in Fig. 4. The isotherms exhibit IV-type curve with an obvious hysteresis loop, indicating the mesoporous characteristics of the silica shell. The BET surface area and total pore volume are calculated to be  $269\text{ m}^2/\text{g}$  and  $0.274\text{ cm}^3/\text{g}$ , respectively. The mesopore size distribution exhibits a sharp peak at the mean value of  $2.9\text{ nm}$  (Fig. 4, inset), indicating the uniform mesopore of the silica shell.

The thermogravimetric analysis (TGA) curves of pure silica spheres, pure  $Eu(DBM)_3phen$ ,  $SiO_2@SiO_2:Eu(DBM)_3phen$  particles before and after calcination are shown in Fig. 5. And the corresponding DSC curve of  $SiO_2@SiO_2:Eu(DBM)_3phen$  after calcination is shown in Fig. 5e. The products were heated from  $30$  to  $800\text{ }^\circ\text{C}$  with the heating rate of  $20\text{ }^\circ\text{C}/\text{min}$  under  $N_2$  atmosphere. For pure RE complexes  $Eu(DBM)_3phen$ , It started to decompose at about  $380\text{ }^\circ\text{C}$  and the mass loss increased to about  $62.2\%$  when the temperature rose to  $600\text{ }^\circ\text{C}$  (in Fig. 5d), which are mainly resulted from the release of  $H_2O$ ,  $CO_2$ ,  $NO_2$ ,  $NH_3$  gases from the thermal decomposition of  $Eu(DBM)_3phen$ . For  $SiO_2@SiO_2:Eu(DBM)_3phen$  particles after calcination (in Fig. 5b), a small weight loss from  $30$  to  $150\text{ }^\circ\text{C}$  with endothermic peak was observed, due to the loss of the adsorbed water molecules. A large weight loss from  $250$  to  $600\text{ }^\circ\text{C}$  with exothermic peak attributed to the condensation among silanol groups was observed (the same as pure silica spheres in Fig. 5a) and the decomposition of  $Eu(DBM)_3phen$ . Meanwhile, a large weight loss from  $190$  to  $300\text{ }^\circ\text{C}$  appears in Fig. 5c, which caused by the



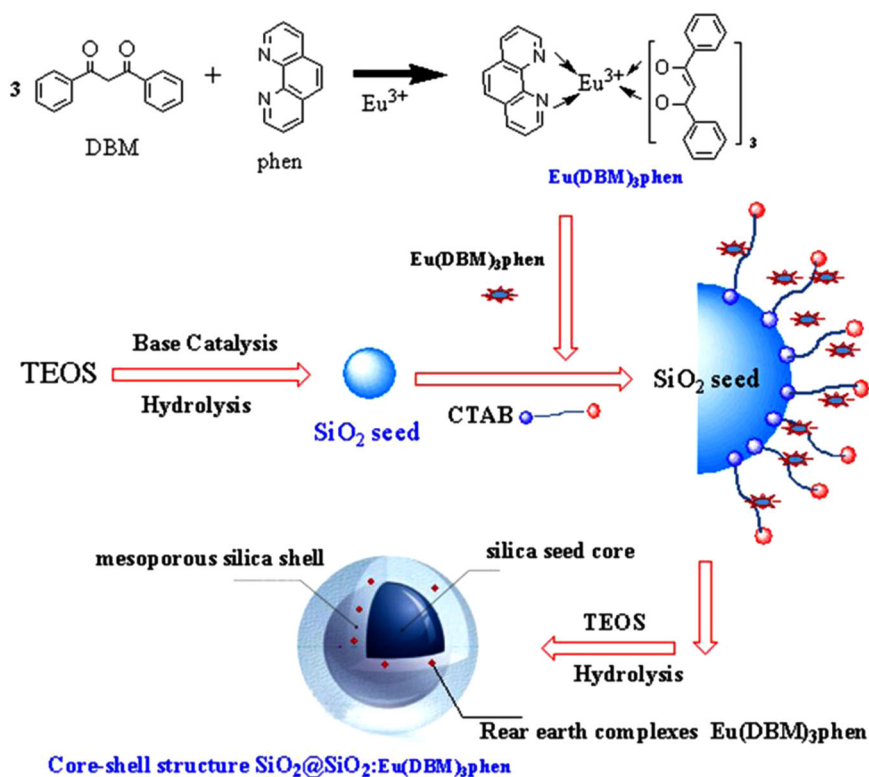
**Fig. 5** TGA curves of pure silica spheres **a**,  $5.0\text{ mol}\%$   $Eu(DBM)_3phen$ -doped core-shell  $SiO_2@SiO_2:Eu(DBM)_3phen$  particles (after calcination) **b**,  $5.0\text{ mol}\%$   $Eu(DBM)_3phen$ -doped core-shell  $SiO_2@SiO_2:Eu(DBM)_3phen$  particles (before calcination) **c** and pure  $Eu(DBM)_3phen$  **d**; Inset is the corresponding DSC curve of  $SiO_2@SiO_2:Eu(DBM)_3phen$  particles after calcination **e**

decomposition of CTAB, but it didn't appear in Fig. 5b, manifesting that the residuary CTAB have been removed completely after calcination. In addition, It's easy to find that the thermal stability of the RE complexes has been greatly improved by incorporating into  $SiO_2$  matrix. It may be caused by weak interactions between lanthanide complexes and the  $SiO_2$  matrix, such as van der Waals' contacts, hydrogen bonding, or electrostatic forces [33, 34]. And Fig. 5 gives a basis for the calcining process to remove the surfactant CTAB.

### 3.2 Formation process

On the basis of above experimental results, a possible formation process for the  $SiO_2@SiO_2:Eu(DBM)_3phen$  composite microspheres was proposed and shown in Fig. 6. Firstly, monodisperse silica spheres were formed via base-catalyzed hydrolysis of TEOS, and obviously there should be many hydroxyl groups with high activity on the surface of silica seeds. Secondly, as the surfactant, the hydrophilic end (marked with blue part) of CTAB connected with the surface of silica microspheres, and its hydrophobic end (marked with red part) connected with  $Eu(DBM)_3phen$ . With the secondary addition of TEOS and the occurrence of further hydrolysis reaction, the generated  $SiO_2$  along with RE complexes depositing on the surface of silica seeds and forming a mesoporous silica layer [35] which contains fluorescent  $Eu(DBM)_3phen$ . Finally, the residuary CTAB is removed by a calcining process and the final products  $SiO_2@SiO_2:Eu(DBM)_3phen$  composite microspheres were obtained.

**Fig. 6** The formation mechanism of SiO<sub>2</sub>@SiO<sub>2</sub>: Eu(DBM)<sub>3</sub>phen microspheres



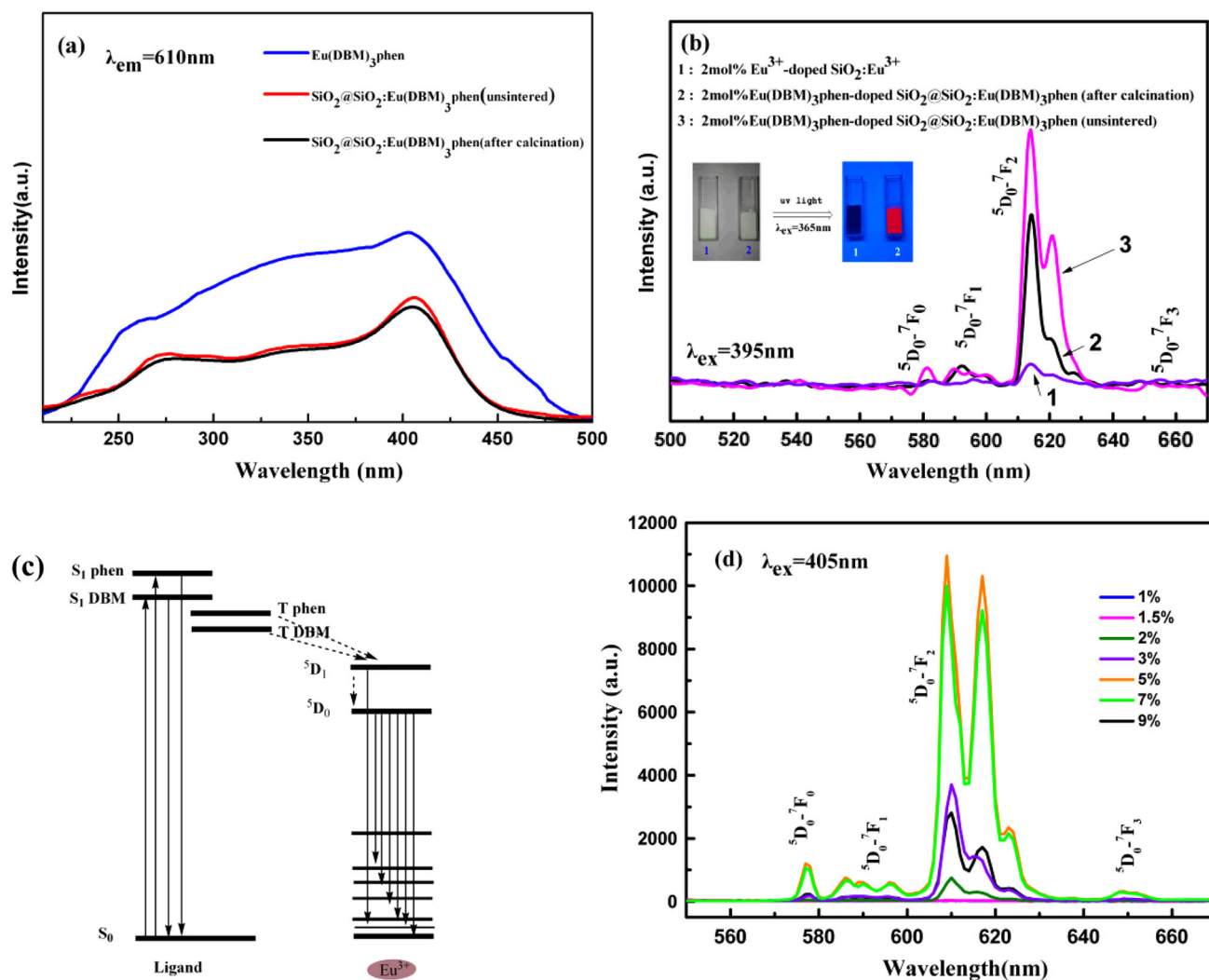
### 3.3 Luminescence properties

The photoluminescence properties of the as-obtained products are shown in Fig. 7. Figure 7a shows the excitation spectra of the pure Eu(DBM)<sub>3</sub>phen, core-shell SiO<sub>2</sub>@SiO<sub>2</sub>:Eu(DBM)<sub>3</sub>phen microspheres before and after calcination at 610 nm. It can be clearly observed from Fig. 7a that all excitation spectrums exhibit a very broad band ranging from 250 to 500 nm with the maximum excitation wavelength centered at about 405 nm which belongs to Eu<sup>3+</sup> ion f-f absorption transition from <sup>7</sup>F<sub>0</sub> → <sup>5</sup>L<sub>6</sub> [36]. Compared with the pure RE complexes, the maximum excitation wavelength of core-shell SiO<sub>2</sub>@SiO<sub>2</sub>:Eu(DBM)<sub>3</sub>phen microspheres shows a small blue shift changing from 403 to 410 nm, implying that different compositions may exist between the two samples caused by the silica matrix. In addition, there is little difference between the excitation spectra of SiO<sub>2</sub>@SiO<sub>2</sub>:Eu(DBM)<sub>3</sub>phen microspheres before and after calcination, indicating no thermal decomposition of Eu(DBM)<sub>3</sub>Phen in the calcining process.

As is shown in Fig. 7b, compared with silica spheres doped with Eu<sup>3+</sup> (prepared as report [17]), the core-shell SiO<sub>2</sub>@SiO<sub>2</sub>:Eu(DBM)<sub>3</sub>phen microspheres before and after calcination show much higher luminous intensity. The insets in Fig. 7b are photographs of the 2 mol% Eu<sup>3+</sup>-doped SiO<sub>2</sub>:Eu<sup>3+</sup> and 2 mol% Eu(DBM)<sub>3</sub>phen-doped core-shell SiO<sub>2</sub>@SiO<sub>2</sub>:Eu(DBM)<sub>3</sub>phen microspheres under UV light irradiation (λ<sub>ex</sub> = 365 nm), it can be observed that the core-

shell SiO<sub>2</sub>@SiO<sub>2</sub>:Eu(DBM)<sub>3</sub>phen microspheres emit strong red PL upon irradiation with ultraviolet light while the SiO<sub>2</sub>:Eu<sup>3+</sup> microspheres emit almost no light, which can be proved further by the emission spectra (λ<sub>ex</sub> = 395 nm) of Fig. 7b1–b2. It shows that the peak positions and spectral shapes of emission spectra are not influenced by replacing Eu<sup>3+</sup> ion with Eu(DBM)<sub>3</sub>phen, but the luminous intensity increase a lot, which is caused by the influence of organic ligands (the mechanism is shown in Fig. 7c): the DBM and phen should work as a “antenna” for the Eu<sup>3+</sup>, augmenting the number of photons converted into visible region [20]. Therefore, The RE complexes show much better luminous performance than the RE ions. And the luminous intensity of SiO<sub>2</sub>@SiO<sub>2</sub>:Eu(DBM)<sub>3</sub>phen microspheres before calcination is higher than the microspheres of after calcination (as shown in Fig. 7b3), which may caused by the “Fluorescence protector” effect provided by CTAB [37].

In order to investigate the effect of the dopant concentration on PL intensities, the concentrations of Eu(DBM)<sub>3</sub>phen dopants were varied from 1 to 9 mol%. Figure 7d shows the emission spectra of SiO<sub>2</sub>@SiO<sub>2</sub>:Eu(DBM)<sub>3</sub>phen with the RE complexes dopant concentration of 1.0, 1.5, 2.0, 3.0, 5.0, 7.0, and 9.0 mol%, respectively. These spectral shapes are almost same irrespective of the RE complexes concentration. Under the excitation of 405 nm, the emission spectrum consists of the <sup>5</sup>D<sub>0</sub> → <sup>7</sup>F<sub>J</sub> (J = 0, 1, 2, 3) (577, 595, 610, and 648 nm) transition of the Eu<sup>3+</sup>



**Fig. 7** The photoluminescence properties of the as-obtained products. **a** excitation spectra of pure  $\text{Eu}(\text{DBM})_3\text{phen}$  complexes, core-shell  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3\text{phen}$  microspheres before and after calcination; **b** emission spectra of 2.0 mol%  $\text{Eu}^{3+}$ -doped silica spheres (1), 2.0 mol%  $\text{Eu}(\text{DBM})_3\text{phen}$ -doped  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3\text{phen}$

microspheres before (2) and after calcination (3); **c** Schematic representation of photophysical processes of  $\text{Eu}(\text{DBM})_3\text{phen}$ ; **d** emission spectra of  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3\text{phen}$  microspheres with different RE complexes concentration

ions. The strongest emission peaking at 610 nm arises from the electric-dipole allowed  $^5\text{D}_0 \rightarrow ^7\text{F}_2$  transition of  $\text{Eu}^{3+}$  ions, which are caused by the lack of inversion symmetry at  $\text{Eu}^{3+}$  ions site [38]. The positions of the luminescence peaks in emission spectrum do not change significantly with increasing of the RE complexes concentration, but the intensity of the hypersensitive  $^5\text{D}_0 \rightarrow ^7\text{F}_2$  peak shows a linear increase with the concentration of RE complexes increasing about to 7 mol% where it starts to bend towards an asymptotic behavior, then decreases with more addition of RE complexes, which is regarded as the typical concentration quenching effect [39], and it is due to the interface effects of silicon hydroxyls and space steric hindrance that hinder the energy transfer between the activator ions.

## 4 Conclusions

Core-shell structural  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3\text{phen}$  microspheres have been successfully prepared via a seeded growth method which can enhance the fluorescent signal and increase the photostability by encapsulating the hydrophobic europium complexes into hydrophilic silica shell by means of surfactant CTAB. The samples are well dispersed and uniform spheres with the diameter of  $\sim 370$  nm, which include the core diameter of  $\sim 340$  nm and the shell thickness of  $\sim 15$  nm. The formation process of  $\text{SiO}_2@\text{SiO}_2:\text{Eu}(\text{DBM})_3\text{phen}$  microspheres is proposed in detail. The as-obtained microspheres show a much stronger red emission corresponding to the  $^5\text{D}_0 \rightarrow ^7\text{F}_2$  transition of the  $\text{Eu}^{3+}$  ions under ultraviolet light excitation  $\lambda_{\text{ex}} = 405$



nm compared with  $\text{SiO}_2:\text{Eu}^{3+}$ , and the PL intensity for the 450–750 nm emission increases with increasing of Eu (DBM)<sub>3</sub>phen's concentration and reaches the maximum when the concentration of Eu(DBM)<sub>3</sub>phen is 7.0 mol%. This method is simple and can reduce the usage of expensive RE ions. The as-obtained microspheres may present potential applications in the fields of optoelectronic devices, bioimaging, medical diagnosis, and study on the structure of functional composites.

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#### Compliance with ethical standards

**Conflict of interest** The authors declare that they have no competing interests.

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