

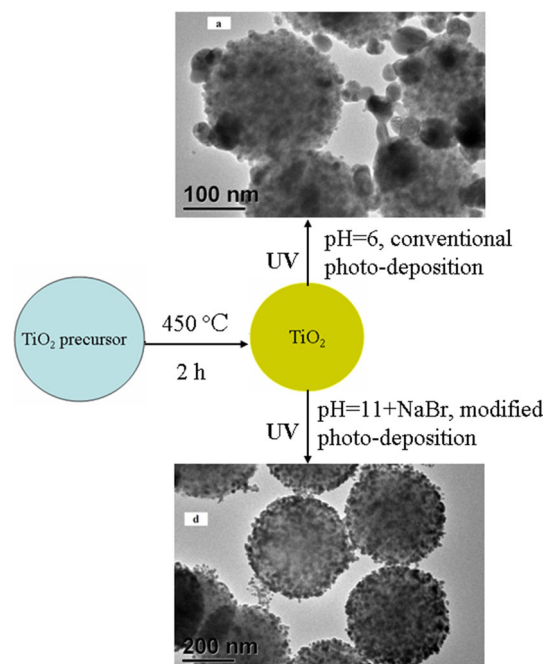
# Modified photodeposition of uniform Ag nanoparticles on TiO<sub>2</sub> with superior catalytic and antibacterial activities

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**Abstract** Photocatalytic reduction is a facile and effective route to deposit Ag nanoparticles (NPs) on TiO<sub>2</sub>. Nevertheless, it is difficult to produce Ag NPs that have small particle size and uniform distribution on TiO<sub>2</sub> to form core-shell structure TiO<sub>2</sub>@Ag composites by straightforward photodeposition. In this study, monodisperse TiO<sub>2</sub> spheres composed of anatase nanocrystals were directly employed as templates to deposit small and homogeneously dispersed Ag NPs by modified photodeposition procedure. To prove the feasibility of the innovative method, we evaluated the catalytic performance and bactericidal efficacy of the as-prepared samples based on the characterizations, since there is strong dependence between size and dispersion of Ag NPs and catalytic performance and inhibition of bacterial growth, where smaller and more highly dispersed Ag NPs displayed better antimicrobial and catalytic properties. The results indicate that the modified photodeposition route can effectively prevent TiO<sub>2</sub> spheres themselves from aggregating and Ag NPs from rapid overgrowing on the TiO<sub>2</sub> surface in the deposition process. The superior bactericidal and catalytic properties of the TiO<sub>2</sub>@Ag composites are largely attributed to the small and highly dispersed Ag NPs on TiO<sub>2</sub> spheres.

*Graphical Abstract*



**Keywords** Photodeposition · TiO<sub>2</sub>@Ag composites · Catalytic performance · Antibacterial activity

## 1 Introduction

Noble metal NPs exhibit completely new or improved properties based on specific characteristics such as size, distribution and morphology in comparison with their bulk counterparts [1–6]. Ag NPs have been studied intensively for their potential use in catalysis, biosensors, biomedicine and environmental filtration due to their unique optical and electrical properties and excellent antibacterial activity

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[7–12]. It is well known that the high surface energy of the Ag NPs makes them easily aggregate to big particles, which could deteriorate their unique chemical properties and induce loss of overall performance. Therefore, the stability is one of the most important factors for their practical applications in nanoscience and nanotechnology. To stabilize Ag NPs, an efficient measure is to immobilize highly dispersed Ag NPs on inexpensive substrates such as SiO<sub>2</sub>, Fe<sub>3</sub>O<sub>3</sub>, ZnO, ZrO<sub>2</sub>, CeO<sub>2</sub>, Al<sub>2</sub>O<sub>3</sub>, C materials and TiO<sub>2</sub> [13–31]. Particular attention has been focused on supporting and binding AgNPs onto TiO<sub>2</sub> surface to improve catalytic and photocatalytic performance, and antimicrobial activity due to the synergistic interaction between Ag NPs and TiO<sub>2</sub> support [20–28]. Moreover, the assembly of Ag NPs on the support could enhance their stability and easy separation from the reaction medium for recycle. Photodeposition is a facile and effective method to deposit various noble metal NPs on the surface of TiO<sub>2</sub> under UV light. Nevertheless, it is difficult to produce uniform Ag NPs that have very small size and are strongly anchored on the surface of TiO<sub>2</sub> by conventional photodeposition [29–31].

In this study, monodisperse TiO<sub>2</sub> spheres composed of anatase nanocrystals were directly employed as templates to deposit small and homogeneously dispersed Ag NPs by modified photodeposition procedure. To prove the feasibility of the innovative method, we evaluated the catalytic performance and bactericidal efficacy of the as-prepared samples based on the characterizations.

## 2 Experimental

### 2.1 Chemicals

Titanium tetraisopropoxide (TTIP, purity  $\geq 97\%$ ) is purchased from Sigma-Aldrich. AgNO<sub>3</sub> (purity  $\geq 99.9\%$ ), methylamine (MA, 40 % aqueous solution), rhodamine B (Rh B, purity  $\geq 99.0\%$ ), sodium borohydride ( $\geq 99.0\%$ ), methanol, ethanol and acetonitrile are of analytical grade and purchased from Shanghai Chemicals Co., Ltd., China, and used without any further purification. Gram-negative bacteria *Escherichia coli* (*E. coli*, ATCC 25922) and Gram-positive bacteria *Staphylococcus aureus* (*S. aureus*, ATCC 6538) were purchased from Guangdong Institute of Microbiology. Deionized water was used throughout the experiments except for in the antibacterial tests where sterilized H<sub>2</sub>O was used.

### 2.2 Preparation of TiO<sub>2</sub>, TiO<sub>2</sub>/Ag and TiO<sub>2</sub>@Ag composites

The monodisperse TiO<sub>2</sub> spheres were synthesized according to the reported procedures with some modifications

[32]. In a typical procedure, 6.0 mmol TTIP stabilized in 10 mL anhydrous ethanol was dropwise added into a pre-mixed solution of 50 mL ethanol and 40 mL acetonitrile containing 1.2 mmol MA and 24 mmol H<sub>2</sub>O. The solution turned milky within a few seconds, and the prepared suspension was stirred for 6 h to form the amorphous TiO<sub>2</sub> spheres. The white precipitate was harvested by centrifugation, followed by washing with ethanol five times and drying at 60 °C for 24 h. The dried powders were calcined at 450 °C for 2 h in air to remove organic components and transform the amorphous precursors into anatase TiO<sub>2</sub> spheres for further use.

Two different deposition processes, one is conventional photodeposition for TiO<sub>2</sub>/Ag composites and the other is modified photodeposition for TiO<sub>2</sub>@Ag composites, were conducted in a photochemical reactor as we adopted previously [33]. In the two cases, Ag-coated TiO<sub>2</sub> samples with Ag/Ti molar ratio = 8 % (the same depositing amount of Ag on TiO<sub>2</sub>) were prepared. For conventional photodeposition, suspension was prepared by ultrasonic mixing 100 mg (1.25 mmol) of TiO<sub>2</sub> and 15.8 mg (0.093 mmol) of AgNO<sub>3</sub> with 180 mL of ultrapure water and 20 mL of methanol in the Pyrex glass reactor. The suspension was purged with a stream of N<sub>2</sub> for 30 min to remove oxygen and then irradiated for 4 h at under N<sub>2</sub> and magnetic stirring. The black precipitate was recovered by centrifuging and washing five times, dried at 60 °C for 24 h.

The modified photodeposition was carried out in the same manner as described above, except for that the pH of the suspension was adjusted to 11 and a predetermined amount of NaBr was added into the suspension.

### 2.3 Characterization of samples

The XRD patterns were obtained on a Bruker D8 Advance X-ray diffractometer with Cu-K $\alpha$  radiation ( $\lambda = 0.15418$  nm). The observation of the morphology was performed on a JEOL JEM-2010 TEM at acceleration voltage 200 kV. The UV–visible absorption spectra were measured using a Hitachi U-3010 UV–Vis spectrophotometer. XPS study was carried out on a PHI-5702 multifunctional X-ray photoelectron spectroscopy, using MgK $\alpha$  radiation as the excitation source and the binding energy of contaminated carbon (C1s 284.6 eV) as the reference, at a pass energy of 29.4 eV and a resolution of  $\pm 0.2$  eV in the binding energy position.

### 2.4 Catalytic performance

The catalytic degradation of Rh B was carried out in a quartz cuvette with an optical path length of 1 cm at room temperature. 1.0 mL of aqueous Rh B solution

( $2 \times 10^{-5}$  mol/L) was mixed with 1.0 mL of a fresh  $\text{NaBH}_4$  solution (0.01 mol/L). 0.2 mL of aqueous dispersion of  $\text{TiO}_2$ ,  $\text{TiO}_2/\text{Ag}$  and  $\text{TiO}_2@\text{Ag}$  composites (0.03 %) was added separately. The color of the mixture was fading with the experiment proceeding, indicating a gradual reduction of Rh B. The catalytic performance was studied by monitoring the variation in optical density at the wavelength of the absorbance maximum ( $\lambda_{\text{max}}$ ) of the dye with UV-Vis.

## 2.5 Antibacterial activity

The antibacterial activity of the as-prepared samples against *E. coli* and *S. aureus* were evaluated using paper disk diffusion assay [19] and spread-plate method on LB agar spread onto a Petri plate [21, 34], respectively, with some modifications. The disk diffusion assay was performed by placing a 7-mm filter paper disk saturated with different concentrations of pure  $\text{TiO}_2$ ,  $\text{TiO}_2/\text{Ag}$  and  $\text{TiO}_2@\text{Ag}$  composite spheres (0.25, 0.5, 0.75 and 1.0 g/mL, respectively) onto agar plates (4 per plate) seeded with  $10^5$  CFU (colony forming units) of *E. coli* and *S. aureus*, respectively. After incubation at 37 °C for 24 h, the diameters of the inhibition zones were observed and measured. The paper disk saturated with sterilized water was used as the control. To use spread-plate method, the tested bacterial suspension (an aqueous suspension of bacterial/antimicrobial agents) was prepared with  $1 \times 10^3$  CFU/mL of *E. coli* and *S. aureus* containing 0.25 mg/mL  $\text{TiO}_2$ ,  $\text{TiO}_2/\text{Ag}$  and  $\text{TiO}_2@\text{Ag}$  composites, respectively. Out of each sample, 0.1 mL of the suspension was spread over LB agar plates, which were then incubated at 37 °C for 24 h in the dark. Control experiment with only bacterial suspension without antimicrobial agents was conducted to quantify the CFU of bacterial strain in the blank sample.

## 3 Results and discussion

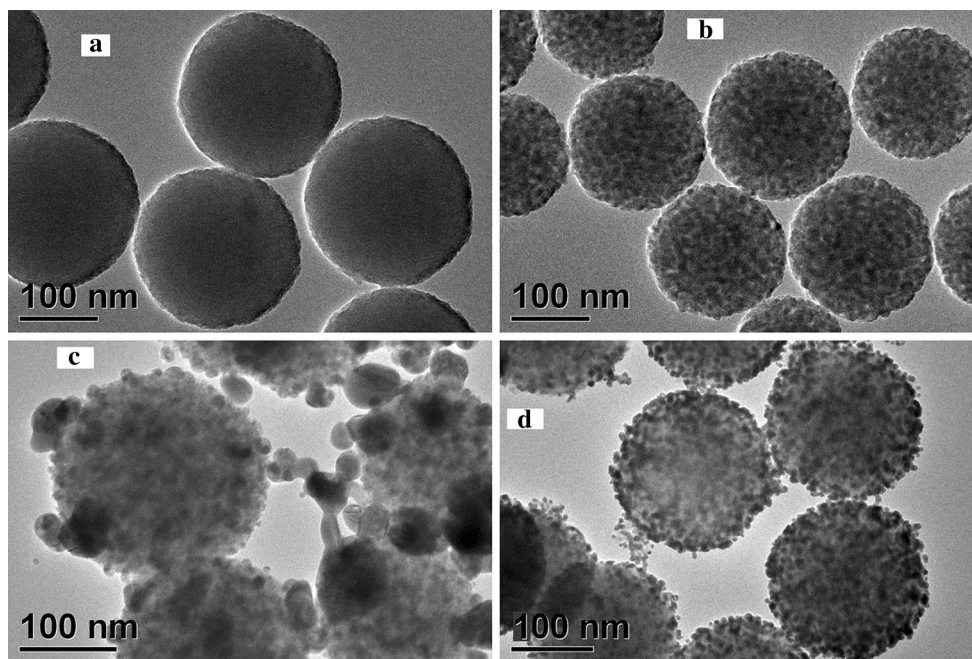
### 3.1 Characterization

Figure 1 shows typical TEM images of the amorphous  $\text{TiO}_2$  precursors and the anatase  $\text{TiO}_2$  spheres. It can be seen that amorphous  $\text{TiO}_2$  spheres are highly monodispersed with average size of 210 nm and possess very smooth surfaces without obvious granular feature (Fig. 1a). After heat treatment at 450 °C for 2 h, the anatase  $\text{TiO}_2$  spheres still keep the original monodisperse morphology, the average size of the particles was reduced to 185 nm (15 % shrinkage), and comparatively rough surfaces with granular feature were produced (Fig. 1b).

A simple suspension of anatase  $\text{TiO}_2$  spheres containing  $\text{AgNO}_3$ , without pH adjustment (pH 6) and other chemical

additives, was straightforwardly exposed to ultraviolet irradiation. The representative TEM image of the as-prepared sample is shown in Fig. 1c. It is found that the sizes of the Ag NPs deposited on the  $\text{TiO}_2$  are large and heterogeneously dispersed, and  $\text{TiO}_2$  spheres tend to have a certain degree of aggregation. No real core-shell structure  $\text{TiO}_2@\text{Ag}$  has been formed, and we designate it as  $\text{TiO}_2/\text{Ag}$ . In the case,  $\text{Ag}^+$  adsorbed onto the exposed  $\text{TiO}_2$  surfaces is photochemically reduced to initially form metal nucleus centers, which can act as the reservoir of photoinduced electrons to accelerate the photoreduction of  $\text{Ag}^+$  ions. The Ag deposits with large size and wide size distribution should be attributed to their rapid overgrowth on the original  $\text{TiO}_2$  particles. On the other hand, the pH employed is near the isoelectric point (IEP = 6.25) of  $\text{TiO}_2$  [35, 36] and might cause the aggregation of the  $\text{TiO}_2$  particles [30]. The stability of  $\text{TiO}_2$  spheres in suspension is influenced by pH values. If the pH value of suspension is higher or lower than the IEP of  $\text{TiO}_2$ , the  $\text{TiO}_2$  surface will be negatively or positively charged, respectively. The further the pH value is from the IEP, the more stable and the better dispersed  $\text{TiO}_2$  particles will be. Oppositely, if close to or equal to the IEP,  $\text{TiO}_2$  particles will be aggregated.

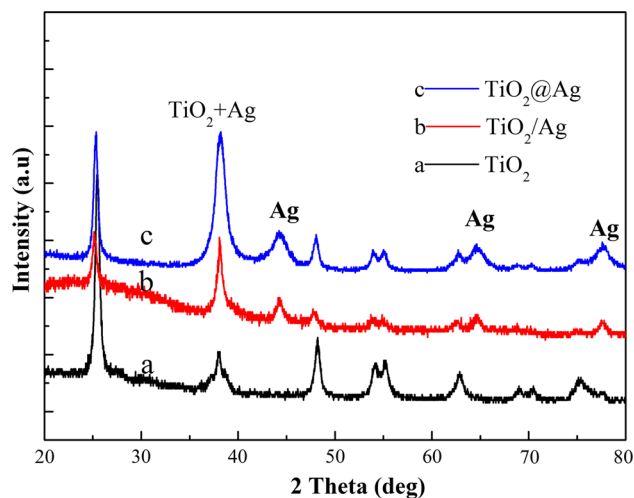
When the pH of the suspension was adjusted to 11 and a predetermined amount of NaBr was added into the suspension, most of  $\text{Ag}^+$  in the forms of insoluble  $\text{Ag}_2\text{O}$  and soluble  $\text{AgNO}_3$  can be transformed into insoluble  $\text{AgBr}$  due to the difference in solubility product constant  $K_{\text{sp}}$  [ $K_{\text{sp}}(\text{AgBr}) = 5.0 \times 10^{-13}$ ,  $K_{\text{sp}}(\text{AgOH}) = 2 \times 10^{-8}$ ]. As shown in Fig. 1d, the monodispersed Ag NPs with average size of  $8 \pm 1.5$  nm, uniform spherical shape and narrow size distribution are homogeneously deposited on  $\text{TiO}_2$  surfaces to form core-shell structured  $\text{TiO}_2@\text{Ag}$  composites. This is in perfect consistency with the hypothesis that lower concentration of free  $\text{Ag}^+$  on the  $\text{TiO}_2$  surface and in bulk solution is very effective in producing small and highly dispersed Ag NPs on  $\text{TiO}_2$ . Both nucleation kinetics and growth kinetics of metal will influence the morphology and size distribution of deposited Ag NPs. The simultaneous nucleation on different active sites and slow growth of these metal nuclei are preferred for the preparation of homogeneously dispersed metal NPs with sharp size distribution. It can be easily understood that if the growth of metal nucleus is too fast, minute diversity of nucleation rate on different active sites in the initial stage of irradiation will cause great growth difference in nucleus. The former deposited nucleus, acting as electron reservoir to inhibit the recombination of photoinduced electrons, can accelerate the growth of metal particles and become big more rapidly than those latter deposited nuclei, which broadens the resultant size distribution of metal particles. An anticipated way to weaken the influence of nucleation kinetics and growth kinetics of metal particles is to keep



**Fig. 1** TEM images of **a** amorphous  $\text{TiO}_2$  precursor spheres, **b** anatase  $\text{TiO}_2$  spheres, **c**  $\text{TiO}_2/\text{Ag}$  and **d**  $\text{TiO}_2@\text{Ag}$  composites

the local concentration of Ag ions on the  $\text{TiO}_2$  surface and the bulk concentration of Ag ions in the suspension extremely low. In addition, in this case, the pH value is far from the IEP of  $\text{TiO}_2$  to assure the  $\text{TiO}_2$  well dispersed which inhibited the aggregation of Ag NPs. Therefore, the procedure can effectively prevent  $\text{TiO}_2$  spheres themselves from aggregating and Ag NPs from rapid overgrowing on the  $\text{TiO}_2$  surface in the photodeposition process, thus resulting in more dispersed and smaller-sized Ag NPs on  $\text{TiO}_2$  compared to conventional photodeposition method.

To confirm the existed states of Ag coated on  $\text{TiO}_2$ , both the  $\text{TiO}_2/\text{Ag}$  and the  $\text{TiO}_2@\text{Ag}$  composites with the same Ag loading were characterized by XRD and XPS. Figure 2 shows the XRD patterns of  $\text{TiO}_2$  spheres,  $\text{TiO}_2/\text{Ag}$  and  $\text{TiO}_2@\text{Ag}$  composites, respectively. The powder XRD pattern of the pure  $\text{TiO}_2$  exhibits well-resolved diffraction peaks (Fig. 2a), corresponding to the reflections of anatase  $\text{TiO}_2$  (JCPDS card 21-1272). Compared to Fig. 2a, both the  $\text{TiO}_2/\text{Ag}$  (Fig. 2b) and  $\text{TiO}_2@\text{Ag}$  composites (Fig. 2c) exhibit four additional peaks at  $2\theta$  angles of  $38.1^\circ$ ,  $44.2^\circ$ ,  $64.4^\circ$  and  $77.4^\circ$  corresponding to the reflections of (111), (200), (220), (311) and (222) crystalline planes of face-centered cubic metallic silver (JCPDS card 04-0783), respectively. One can see the intensity of the corresponding anatase peaks was reduced when Ag NPs were deposited on  $\text{TiO}_2$ . Meanwhile, the peak intensity in the  $2\theta$  region of  $36^\circ$ – $40^\circ$  significantly enhances because both characteristic diffraction peaks of  $38.1^\circ$  of Ag crystals and  $38.6^\circ$  of anatase  $\text{TiO}_2$  overlap. However, it can be clearly seen that the XRD peaks of Ag crystals in the  $\text{TiO}_2@\text{Ag}$  composites



**Fig. 2** XRD patterns of **a** anatase  $\text{TiO}_2$  spheres, **b**  $\text{TiO}_2/\text{Ag}$  composites and **c**  $\text{TiO}_2@\text{Ag}$  composites

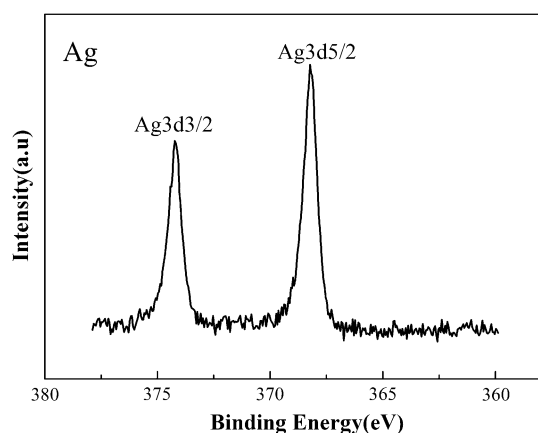
(Fig. 2c) are slightly wider than that in the  $\text{TiO}_2/\text{Ag}$  composites (Fig. 2b), indicating that the former has smaller Ag particle size than the latter. The results are consistent with TEM analysis. To further investigate the possible chemical status of Ag ( $\text{Ag}^0$  or  $\text{Ag}^+$ ) in  $\text{TiO}_2@\text{Ag}$  composites, the corresponding high-resolution XPS spectrum of Ag  $3d$  is shown in Fig. 3. The photoelectron peaks of both the Ag  $3d_{5/2}$  and Ag  $3d_{3/2}$  are basically symmetrical with binding energy (BE) of 368.3 and 374.2 eV, respectively, and the 6.0 eV difference between the BE of Ag  $3d_{5/2}$  and Ag  $3d_{3/2}$  peaks is also characteristic of metallic Ag  $3d$  states. The

results prove that Ag exists in form of metallic Ag in the composites [21–30]. Combined with the results of XRD analysis, it can be concluded that Ag particles are not oxidized in the photodeposition process under UV irradiation.

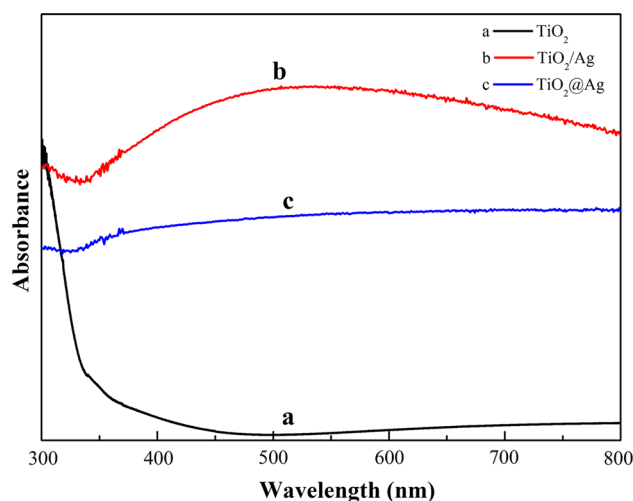
The UV–Vis absorption spectra of pure TiO<sub>2</sub>, TiO<sub>2</sub>/Ag and TiO<sub>2</sub>@Ag composites are illustrated in Fig. 4. It can be seen that neat TiO<sub>2</sub> spheres have a broad intense absorption below 400 nm. It is the characteristic absorption of TiO<sub>2</sub> corresponding to the charge transfer process from the valence band to conduction band in anatase TiO<sub>2</sub> [37–40]. The photodeposited Ag NPs cause significant changes to the absorption spectrum of TiO<sub>2</sub>, resulting in high absorbance from 400 nm to entire visible region, which is characteristic of surface plasmon absorption corresponding to deposited metallic Ag NPs on TiO<sub>2</sub> surfaces [21–29]. The absorbance in the visible region increased firstly and then decreased gradually from 400 to 800 nm for the TiO<sub>2</sub>/Ag composites, indicating that Ag NPs on TiO<sub>2</sub> are not evenly distributed [40]. In contrast, the TiO<sub>2</sub>@Ag composites prepared by the modified deposition route basically maintain constant absorption in the visible region, corresponding to uniformly distributed Ag NPs on TiO<sub>2</sub>.

### 3.2 Catalytic performance

The catalytic degradation of Rh B was chosen as a model reaction to investigate the catalytic performance of both the TiO<sub>2</sub>/Ag and the core–shell structure TiO<sub>2</sub>@Ag composites. Such a reaction catalyzed by Ag catalysts has been reported because this reaction can be rapidly and easily characterized [13, 41]. The absence of catalyst but in the presence of KBH<sub>4</sub> in the Rh B solution only results in a slight decrease in  $\lambda_{\max}$  after standing for 10 h (results not shown here), indicating that the reduction does not proceed without catalyst. Nevertheless, there was no obvious



**Fig. 3** High-resolution XPS spectrum of Ag 3d in TiO<sub>2</sub>@Ag composites

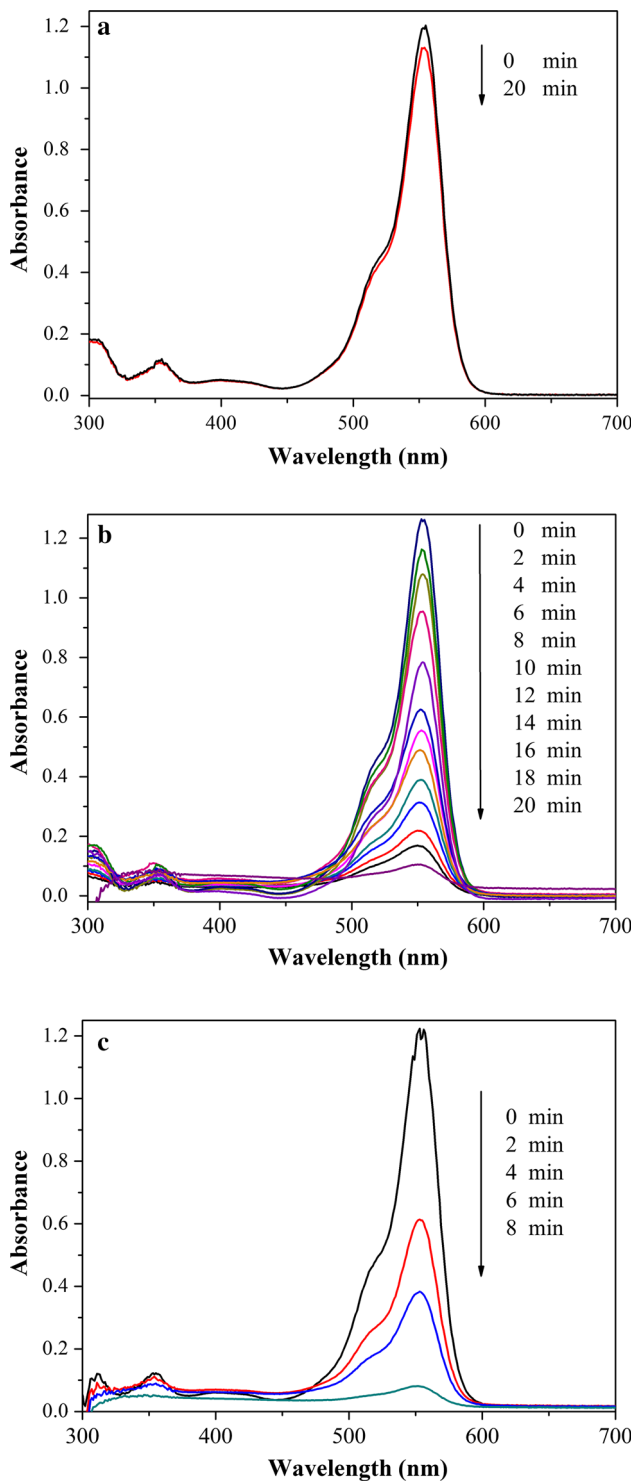


**Fig. 4** UV–Vis spectra of *a* anatase TiO<sub>2</sub> spheres, *b* TiO<sub>2</sub>/Ag composites and *c* TiO<sub>2</sub>@Ag composites

change in the intensity at 554 nm after 20 min when TiO<sub>2</sub> was added to the above mixture (Fig. 5a), indicating that it is inactive toward the catalytic degradation of RhB with bare TiO<sub>2</sub> spheres as catalyst [42]. Figure 5b shows the UV–Vis spectra monitoring the catalytic reduction of RhB, measured at different times using TiO<sub>2</sub>/Ag composites as the catalyst. The absorption peak at 554 nm gradually decreased in intensity and ultimately vanished as the reaction proceeded for 20 min, confirming that the reduction is mainly catalyzed by the Ag NPs on TiO<sub>2</sub>. Instead of TiO<sub>2</sub>/Ag, TiO<sub>2</sub>@Ag composites were added, the absorption intensity at 554 nm dramatically and quickly decreased and disappeared after 8 min (Fig. 5c). This indicates that the catalytic efficiency of TiO<sub>2</sub>@Ag composites is much higher than TiO<sub>2</sub>/Ag composites.

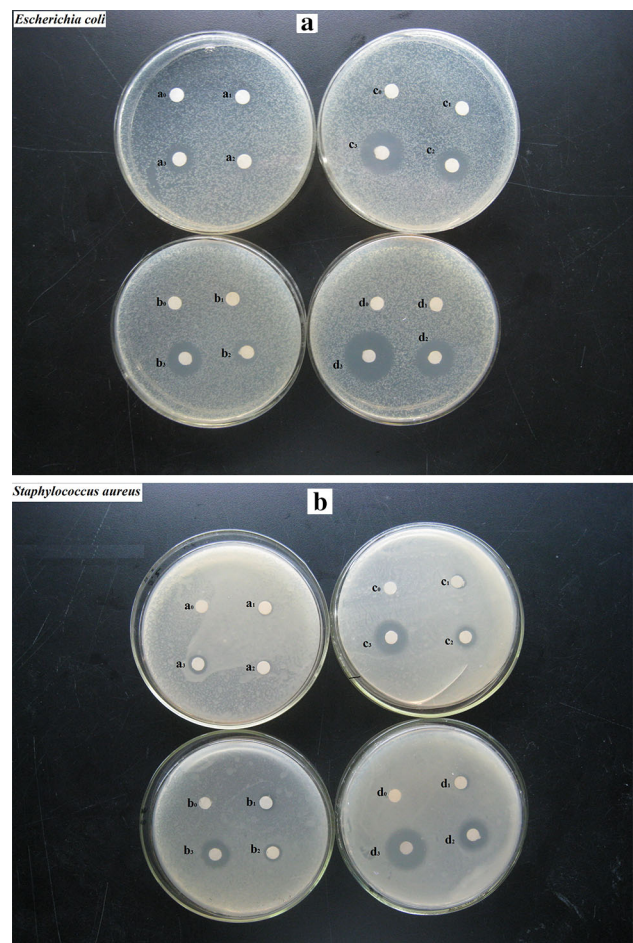
### 3.3 Antibacterial property

First, the antibacterial activity of the as-prepared samples was investigated by measuring the growth inhibition of the bacteria on solid agar plates through paper disk diffusion assay method. In the disk diffusion assay experiment, the paper disks were saturated with different concentrations of TiO<sub>2</sub> spheres, TiO<sub>2</sub>/Ag and TiO<sub>2</sub>@Ag composites, which were placed on LB agar plates and seeded with *E. coli* and *S. aureus*. Subsequently, these plates are incubated at 37 °C for 24 h, and then, the antibacterial effects are evaluated by measuring the diameter of inhibition zone. As shown in Fig. 6a, b, comparing with the blank control groups (a<sub>0</sub>–d<sub>0</sub>), the TiO<sub>2</sub> spheres (a<sub>1</sub>–d<sub>1</sub>) have almost no antibacterial activity against the both bacteria. When Ag NPs photodeposited on TiO<sub>2</sub>, the zones of inhibition can be clearly observed and the growth of the *E. coli* cells and *S. aureus* is significantly inhibited in the presence of both the



**Fig. 5** Evolution of the absorption spectra of Rh B solution (10 mg/L) in the presence of **a** pure TiO<sub>2</sub>, **b** TiO<sub>2</sub>/Ag composites and **c** TiO<sub>2</sub>@Ag composites

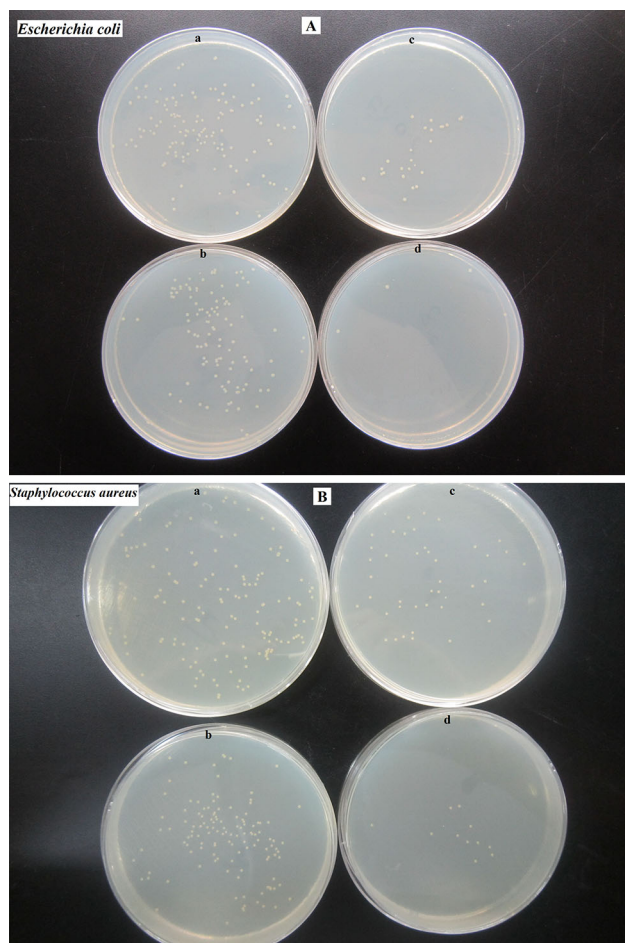
TiO<sub>2</sub>/Ag (a<sub>2</sub>–d<sub>2</sub>) and TiO<sub>2</sub>@Ag (a<sub>3</sub>–d<sub>3</sub>), even at very low concentration (0.2 mg/mL). Moreover, with their concentrations increasing gradually, the inhibition zones grow larger. However, the inhibition zone diameters of the TiO<sub>2</sub>/



**Fig. 6** Inhibition zones of the as-synthesized TiO<sub>2</sub> spheres, TiO<sub>2</sub>/Ag composites and TiO<sub>2</sub>@Ag composites against **a** *E. coli* and **b** *S. aureus*. Control groups saturated with sterilized water (a<sub>0</sub>–d<sub>0</sub>). Experimental groups saturated with 0.25, 0.5, 0.75 and 1.0 mg/mL of TiO<sub>2</sub> spheres (a<sub>1</sub>–d<sub>1</sub>), TiO<sub>2</sub>/Ag (a<sub>2</sub>–d<sub>2</sub>) and TiO<sub>2</sub>@Ag (a<sub>3</sub>–d<sub>3</sub>) composites, respectively

Ag spheres are obviously smaller than that of the TiO<sub>2</sub>@Ag composite spheres at the same concentration. The results reveal that TiO<sub>2</sub>@Ag composites exhibit higher antibacterial efficiency than TiO<sub>2</sub>/Ag.

The bactericidal activity was further confirmed using spread-plate method by the images of colonies on agar plates. Figure 7a, b presents the results of the antibacterial tests using 0.2 g/mL pure TiO<sub>2</sub>, TiO<sub>2</sub>/Ag and TiO<sub>2</sub>@Ag composites against *E. coli* and *S. aureus* on LB agar plates, respectively, after 24-h incubation. A comparison between the blank (Fig. 7Aa, Ba) and the TiO<sub>2</sub> (Fig. 7Ab, Bb) shows that pure TiO<sub>2</sub> particles exhibit extremely weak bacteriostatic action [21]. About 11 % *E. coli* and 24 % *S. aureus* survive in the presence of TiO<sub>2</sub>/Ag (Fig. 7Ac, 7Bc), and TiO<sub>2</sub>@Ag composites lead to almost complete suppression of *E. coli* and *S. aureus* growth (Fig. 7Ad, Bd). The antibacterial tests performed on the two Ag-modified



**Fig. 7** Spread-plate test results using 0.2 mg/mL of  $\text{TiO}_2$ ,  $\text{TiO}_2/\text{Ag}$  and  $\text{TiO}_2@\text{Ag}$  composites against **A** *E. coli* and **B** *S. aureus*. (a) Blank, (b) pure  $\text{TiO}_2$ , (c)  $\text{TiO}_2/\text{Ag}$  composites, (d)  $\text{TiO}_2@\text{Ag}$  composites

samples showed a much greater activity toward *E. coli* than *S. aureus* due to different cell wall structures and different antibacterial mechanisms of Ag NPs against different cells [34]. Several mechanisms for their bactericidal effects have been proposed, it is widely believed that Ag NPs are incorporated in the cell membrane, which causes leakage of intracellular substances and eventually causes cell death. Gram-negative strains such as *E. coli* have a thinner wall in comparison with Gram-positive ones (i.e., *S. aureus*), and it is difficult for Ag NPs to penetrate the thicker cell wall of *S. aureus* due to its higher resistance. However,  $\text{TiO}_2@\text{Ag}$  composites possess higher antibacterial efficiency against *E. coli* and *S. aureus* than  $\text{TiO}_2/\text{Ag}$ . The significantly different bactericidal activity between the two samples was largely attributed to the difference in the Ag particle size and distribution. The smaller Ag particle size and highly dispersed Ag NPs firmly attached onto  $\text{TiO}_2$  sphere surfaces should account for the higher antibacterial activity of  $\text{TiO}_2@\text{Ag}$ . The dependence on particle size and

distribution of antibacterial activity can be understood from three aspects: (1) Smaller particles have more surface atoms that are expected to be active upon contact with bacterial cells; (2) smaller particles also have a larger fraction of atoms on low-coordination and high-energy sites, which makes them more active than larger particles; (3) the number of Ag particles uniformly distributed on  $\text{TiO}_2$  sphere surfaces is more than un-uniformly distributed counterparts at the same silver load, and their exposed surface areas differ significantly as well. This unique core-shell structure of  $\text{TiO}_2@\text{Ag}$  composites could more effectively promote Ag/bacteria contact and thus enhanced bactericidal efficiency in attacking and destructing bacterial cell membranes.

## 4 Conclusions

Core-shell structure  $\text{TiO}_2@\text{Ag}$  composites have been synthesized using modified photodeposition route, i.e., the pH of the suspension of  $\text{TiO}_2$  and  $\text{AgNO}_3$  was adjusted to 11 and a certain amount of NaBr (equal molar to  $\text{AgNO}_3$ ) was added into the suspension to transform  $\text{Ag}^+$  into the corresponding AgBr. The highly dispersed  $\text{TiO}_2$  spheres can serve as a rigid scaffold to prevent the Ag NPs from aggregating during the reaction. A smaller particle size and more uniform particle distribution result in a larger surface-to-volume ratio and more exposed atoms on surface as potential active sites. This core-shell structural feature is favored for enhancing the catalytic and antibacterial properties compared with  $\text{TiO}_2/\text{Ag}$  composites in that the latter tend to suffer from the loss of activity due to aggregation. The present method is versatile, and other noble metals including Au, Pd, Pt and their bimetallic NPs can also be evenly loaded onto the surfaces of  $\text{TiO}_2$  spheres by the modified photodeposition procedure according to their respective properties.

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