

Structure, optical and magnetic properties of LaFeO₃ nanoparticles prepared by polymerized complex method

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Abstract This work reports the study the structure, optical and magnetic properties of LaFeO₃ nanoparticles synthesized by the polymerized complex method. The LaFeO₃ nanoparticles were successfully obtained from calcination of the precursor at different temperatures from 750 to 1,050 °C in air for 2 h. The calcined LaFeO₃ nanoparticles were characterized by X-ray diffraction (XRD), transmission electron microscopy (TEM), UV–Visible spectroscopy, X-ray photoelectron spectroscopy (XPS), X-ray absorption near edge spectroscopy (XANES) and vibrating sample magnetometry. The XRD and TEM results showed that all LaFeO₃ samples had a single phase nature with the orthorhombic structure. The estimated crystallite sizes were in the range of 44.5 ± 2.4 – 74.1 ± 4.9 nm. UV–Vis spectra showed strong UV and Vis absorption with small band gap energy. The valence states of Fe ions were in the Fe³⁺ and Fe⁴⁺ state, as confirmed by XPS and XANES results. The weak ferromagnetic behavior with specific saturation magnetization of 0.1 emu/g at 10 kOe was obtained for the small particle of 44.5 ± 2.4 nm. The uncompensated spins at the surface was proposed as playing a part in the magnetic properties of small sized LaFeO₃.

Keywords LaFeO₃ · Nanoparticles · Magnetic properties · X-ray photoelectron spectroscopy (XPS) · X-ray absorption near edge spectroscopy (XANES)

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1 Introduction

LaFeO₃-type perovskite is one of the most important materials and has attracted attention because of their wide applications, such as electrodes materials for fuel cells, catalysts, chemical sensors, optoelectronic devices [1–5], etc. The substitution with various transition metal (TM) ions in the composition site is observed to improve the electrical/magnetic properties. In earlier report, (Pb_{0.8}La_{0.2})(Ti_{0.8}Fe_{0.2})O₃ sample originates from combining A-site (lone pair) ferroelectricity with B-site magnetic order among substitution at A-site and B-site. Results of detailed dielectric, ferroelectric, and magnetic studies are reported to show that the broad transitions in dielectric spectra correspond to the magnetic ordering in the sample [6]. Therefore, the incorporation of divalent or trivalent cations into the La or Fe sub-lattices has been major investigated as alternatives [6–8].

Recently, the magnetic properties of LaFeO₃ have been extensively studied but the magnetic study of LaFeO₃ nanoparticles is rare [9]. Antiferromagnetic nanoparticles always show unusual magnetic properties due to the finite-size effects, surface anisotropy effects, interface effects, shape anisotropy effect and so on [10–12]. In addition, the introduction of Ba²⁺ ions into La³⁺ site causes a decrease in the number of covalence bonds leads to the limitation of displacement in octahedron [13]. Moreover, mixed valences of Fe³⁺/Fe³⁺/Fe⁴⁺ by oxygen non-stoichiometry of the materials were expected for improvement the magnetic property. Therefore, nano-size of LaFeO₃ system has been major investigated as an alternative. Various types of LaFeO₃ nanoparticles can be synthesized by many methods such as sol–gel [14–16], co-precipitation [17], ball milling [18], sonochemical [19], and hydrothermal [20].

Polymerized complex (PC) method has been synthesized on polyester network between citric acid (CA) and

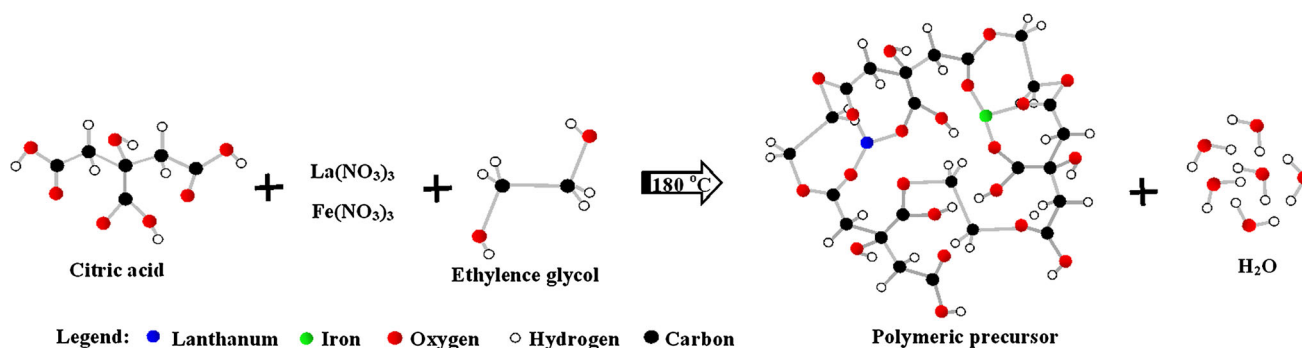


Fig. 1 Schematic representation of polymeric precursor in PC method of LaFeO_3 nanoparticles

ethylene glycol (EG) solution. This solution plays an important role to balance the difference in individual behavior of metal ions in solution. It results in a better distribution of ions and prevents the separation of components at later process stages. Initially, the metal ions are introduced into the solution. During the heating reaction, the complexes and production of polymer gel were formed above $100\text{ }^\circ\text{C}$ and then the oxidation and pyrolysis of the polymer matrix begin also above $400\text{ }^\circ\text{C}$. This step leads to forming amorphous oxide or carbonate precursor [21]. Therefore, the phase-pure with high-quality LaFeO_3 nanocrystalline has been successfully prepared at lower temperature by PC method [22, 23]. This method is suitable in recent years for synthesizing magnetic nanoparticles due to the magnetic properties are usually strongly dependent on the particle size.

In this study, the PC method is used to synthesize LaFeO_3 nanoparticles and the effect of the particles size on the magnetic properties of LaFeO_3 was investigated and discussed. The synthesized LaFeO_3 nanoparticles were characterized by thermogravimetric-differential thermogravimetric (TG-DTG), X-ray diffraction (XRD), transmission electron microscopy (TEM), UV-Visible spectroscopy (UV-Vis), X-ray photoelectron spectroscopy (XPS), and X-ray absorption spectroscopy (XAS). The magnetic properties of LaFeO_3 nanoparticles were investigated using a vibrating sample magnetometer (VSM) at room temperature (RT).

2 Experimental

In the preparation of LaFeO_3 nanoparticles, $\text{LaN}_3\text{O}_9 \cdot 6\text{H}_2\text{O}$ (Aldrich), $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$ (Kanto) and $\text{C}_6\text{H}_8\text{O}_7$ (Analar Normapur) were dissolved in EG of 180 ml to promote the polyesterification. The lanthanum (III) and iron (III) ions are bound by the strong ionic bonds between the metallic ions and carboxylate ions in a polymeric chain or between the polymeric chains. The polymeric precursor compound is schematically represented in Fig. 1. The precursor solution

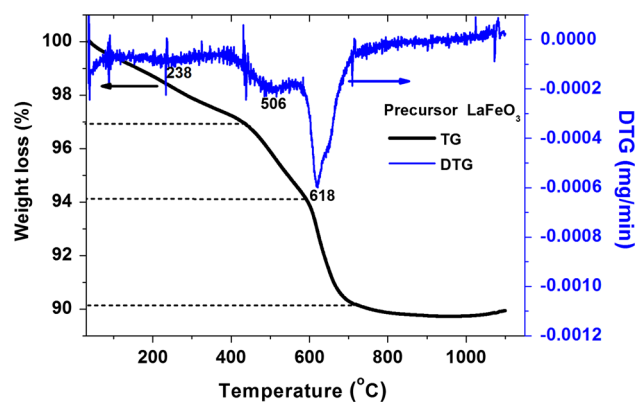


Fig. 2 TG-DTG curve of LaFeO_3 precursor

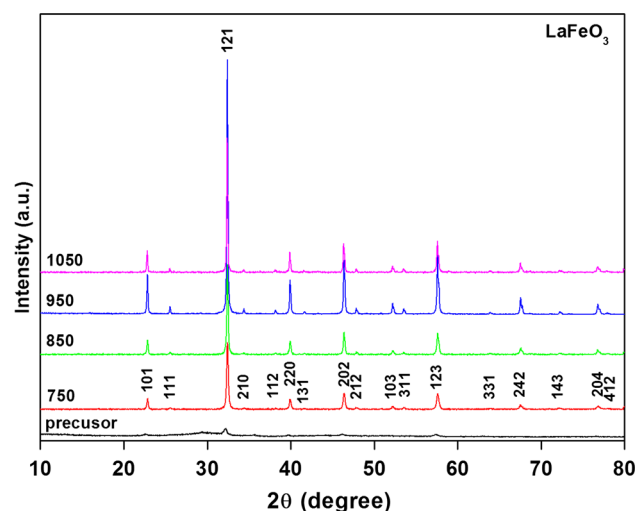


Fig. 3 XRD patterns of precursor and LaFeO_3 samples calcined at various temperatures for 2 h in air

was stirred at RT for 6 h and then heated at $180\text{ }^\circ\text{C}$ until dried. Then the final product was pre-calcined in a furnace at $400\text{ }^\circ\text{C}$ for 2 h, leading to the partial decomposition of the polymeric gel. The precursor was calcined at 750, 850, 950 and $1,050\text{ }^\circ\text{C}$ for 2 h in air to obtain single phase.

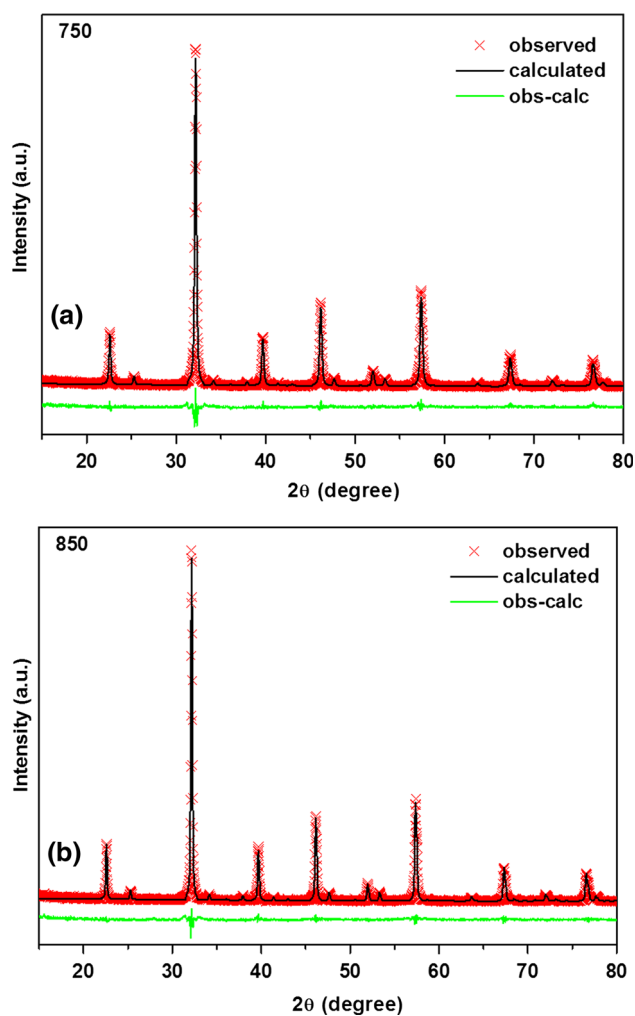


Fig. 4 Refined XRD patterns of LaFeO₃ samples calcined at **a** 750 °C and **b** 850 °C. The experimental data are indicated by the *dots* and the calculated data by the *solid line* overlaying them and the *bottom curve* shows the difference between the experimental and calculated data

The prepared samples were characterized using TG-DTG, XRD, TEM, UV–Vis, XPS, XANES, and VSM. Thermogravimetric-differential thermogravimetric analysis (Mettler Toledo Stare System TG-DSC), and X-ray diffractometer (XRD) using Cu K α radiation with $\lambda = 0.154184$ nm (Bruker D2 Phaser, Germany) were used to study the phases of the LaFeO₃ samples. The morphology and crystal structure of the

samples were performed using TEM (FEI, TECNAI G² 20, Netherlands). UV–Vis was performed using a UV-3101PC UV–Vis-NIR scanning spectrometer (Shimadzu, Japan). XPS analysis was performed by AXIS Ultra DKD (Kotros analytical Ltd, Manchester, United Kingdom). The samples were excited with X-ray hybrid mode of monochromatic Al K $\alpha_{1,2}$ radiation at 1.4 keV. XANES spectra of Fe K edge was studied using X-ray absorption near edge spectroscopy in transmission mode at the BL5.2 line at Siam Photon (Synchrotron Research Institute) in Nakhon Ratchasima, Thailand. The magnetic measurements were performed at RT using a vibrating sample magnetometer (VSM, Versa Lab, Quantum Design, USA).

3 Results and discussion

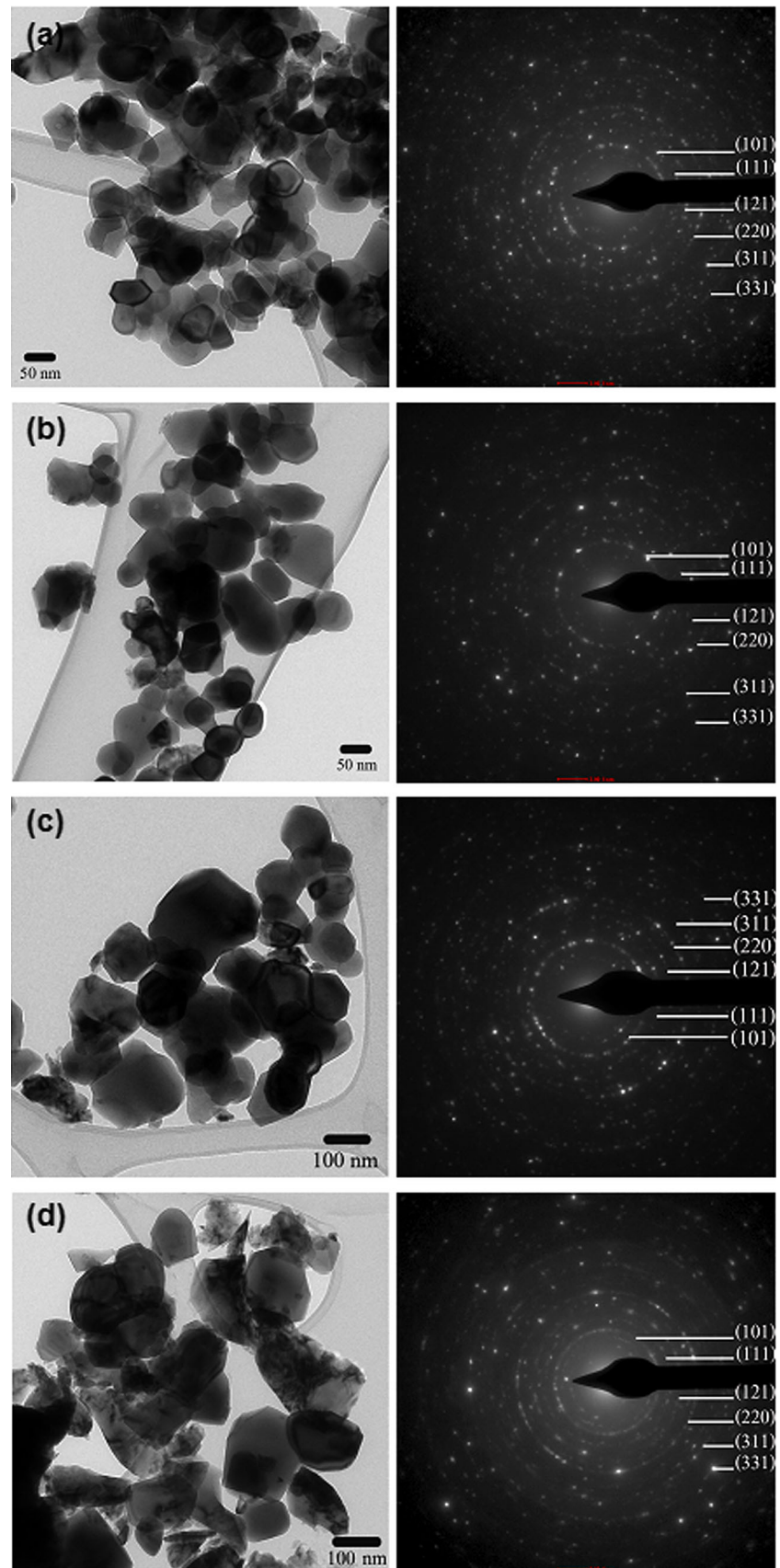
The thermal decomposition and crystallization temperature of the LaFeO₃ precursor were obtained by TG-DTG analysis as shown in Fig. 2. The TG curve showed three steps of weight loss between 30 and 1,100 °C. The weight of the sample remained constant at around 700 °C until the temperature up to 1,100 °C, indicates that the reaction is complete and no evidence of a phase transition is present in the sample. On the DTG curve, three peaks were observed at $\sim 238, 506$ and 618 °C, suggesting that the thermal events related to the burn-out of moisture and trapped solvent (water and carbon dioxide), and nitrates corresponding reaction of LaCO₃OH into La₂O₂CO₃, and La₂O₂CO₃ with Fe₂O₃ into LaFeO₃ [24].

Figure 3 shows the XRD patterns of the precursor and LaFeO₃ samples calcined at various temperatures. The precursor showed amorphous with some weak peak of orthorhombic phase. With increasing the temperature to 750 °C, the diffraction peaks become stronger and sharper, indicating crystallinity of LaFeO₃ becomes better during the calcination process. The calcined samples exhibited peaks consistent with the orthorhombic structure of LaFeO₃ in the standard from JCPDS 88-0641 and no diffraction peaks corresponding to impurity phases were observed. The Rietveld analysis confirmed a single phase orthorhombic structure (space group Pnma (62)) without any impurity phase. Figure 4a, b show the Rietveld refined plot of the samples

Table 1 Crystallite sizes from XRD line broadening, lattice parameter (*a*, *b* and *c*) calculated from Rietveld refined XRD patterns, and magnetization *M* of LaFeO₃ nanoparticles calcined in air at 750, 850, 950 and 1,050 °C for 2 h

Sample	Rietveld refinement					χ^2	R _{ex} (%)	<i>M</i> (emu/g)
	Crystallite size (nm)	Lattice parameter (nm)						
		<i>a</i>	<i>b</i>	<i>c</i>				
750	44.5 ± 2.4	0.5562(6)	0.7856(1)	0.5556(9)	1.190	11.38	0.10	
850	55.6 ± 2.9	0.5563(4)	0.7854(6)	0.5555(5)	1.367	11.78	0.08	
950	66.2 ± 4.0	0.5565(3)	0.7859(5)	0.5559(3)	1.964	9.81	0.08	
1,050	74.1 ± 4.9	0.5563(2)	0.7854(3)	0.5557(2)	1.823	10.78	0.09	

Fig. 5 TEM images with SAED patterns of the LaFeO_3 samples calcined at **a** 750 °C, **b** 850 °C, **c** 950 °C and **d** 1,050 °C



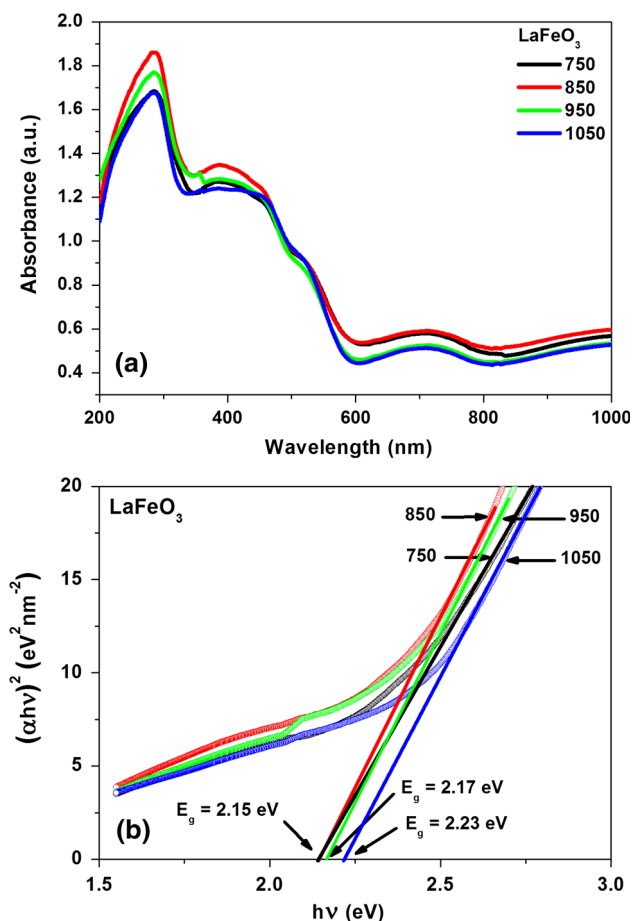


Fig. 6 **a** Room-temperature optical absorbance spectra of the LaFeO_3 samples **b** plot of $(\alpha hv)^2$ as a function of photon energy for LaFeO_3 nanoparticles

calcined at 750 and 850 °C, respectively. The final values for the quality factors and convergence are listed in Table 1. The average crystallite sizes of all samples were calculated from X-ray line broadening of the peaks at (101), (121), (220), (202), and (123) planes using Scherrer's equation. The crystallite sizes were obtained to be 44.5 ± 2.4 , 55.6 ± 2.9 , 66.2 ± 4.0 , and 74.1 ± 4.0 nm for the samples calcined at 750, 850, 950 and 1,050 °C, respectively, which increased with the increase in the calcination temperature. The values of the lattice parameter (a , b , c) were in good agreement with that of orthorhombic LaFeO_3 (JCPDS 88-0641) with $a = 0.5564$ nm, $b = 0.7855$ nm, $c = 0.5556$ nm.

The TEM images with selected area electron diffraction (SAED) patterns of the LaFeO_3 samples are shown in Fig. 5. The samples showed small nanoparticles with particle size below 100 nm except for the sample calcined at 1,050 °C. Particle size of the sample 950 °C-calcined clearly shows that particles become larger as the calcination temperature increases, which agree with the XRD results. The SAED patterns of the samples showed spotty

ring patterns indicative of a polycrystalline structure, which agree with the XRD results.

Figure 6a shows the UV–Vis absorption spectra of the LaFeO_3 nanoparticles. All samples showed a strong absorption in the ultraviolet (~ 200 – 400 nm) and visible light region (~ 400 – 800 nm). This absorption is interesting because LaFeO_3 could be developed a new visible light photocatalyst. The direct band gap energy (E_g) was determined by fitting the absorption data to the direct transition as equation $\alpha hv = A(hv - E_g)^{1/2}$, where α is the optical absorption coefficient, hv is the photon energy, E_g is the direct band gap, and A is a constant [25]. The extrapolation of the linear portions of the curves toward absorption equal to zero ($y = 0$) gives E_g for direct transitions (see Fig. 6b). The estimated direct band gaps of all samples were in the range of 2.15–2.23 eV. These band gaps are closed to the values reported in the literature for LaFeO_3 . For examples, Parida et al. [15] and Yang et al. [26] have reported direct band gap value of 2.1 eV for LaFeO_3 nanoparticles synthesized by sol–gel auto-combustion method. Saad et al. [16] have reported a direct band gap value of 2.15 eV for LaFeO_3 nanoparticles prepared by sol–gel combustion method. These small band gaps of LaFeO_3 are interesting for application in photocatalytic, sensor materials and electrode material in solid oxide fuel cells (SOFCs) [15, 27–30].

The valence states of La, and Fe in the prepared samples were investigated by XPS, which is more sensitive to surface. The XPS spectra of La3d, Fe2p, and O1s of LaFeO_3 nanoparticles for samples calcined at 850 and 1,050 °C were measured at RT as shown in Fig. 7a–c for the sample calcined at 850 °C, and Fig. 7d–f for the sample calcined at 1,050 °C. Figure 7a, d show that the peak position at approximately 833.5–833.6 and 850.3–850.6 eV are assigned to La 3d_{5/2} and La 3d_{3/2} [15, 31], respectively, indicating that the La ions are in the La³⁺ ions. The Fe peak of the samples included two components of Fe³⁺ (709.9–710.0 and 723.2–723.5 eV) and Fe⁴⁺ (711.6–711.7 and 724.9–725.2 eV) [15, 32] as shown in Fig. 7b, e. This result indicates that the Fe ions in our LaFeO_3 samples are in a mixed of the Fe³⁺ and Fe⁴⁺ valence state. For O ions in LaFeO_3 , the samples showed similar peak position of crystal lattice oxygen (O_L) at approximately 529.2–529.3 eV and hydroxyl oxygen (O_H) at approximately 531.3–531.6 eV, indicating that it is attributed to the contribution of La–O and Fe–O in LaFeO_3 crystal lattice for the O_L signal [31, 32] as shown in Fig. 7c, f.

To confirm the valence states of La and Fe in the prepared LaFeO_3 samples, we performed experiments by measuring the XANES. The XANES spectra at Fe K-edge were measured in transmission mode at RT. Figure 8 shows edge energies of the Fe foil, Fe₃O₄ (Fe²⁺, Fe³⁺) standard, Fe₂O₃ (Fe³⁺) standard, and LaFeO_3 samples calcined at different temperature for comparison. The shift

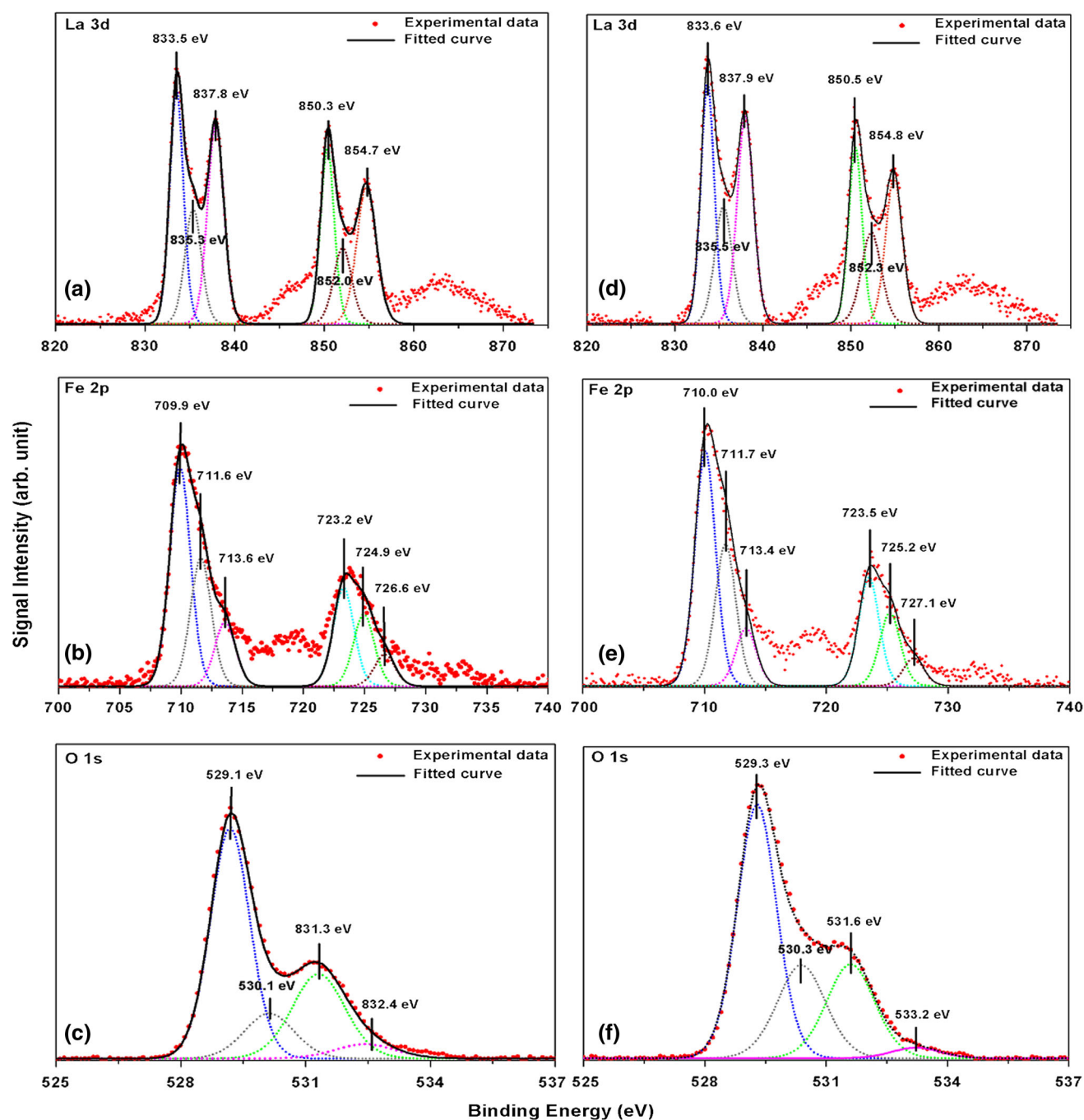


Fig. 7 XPS spectra of La3d, Fe2p and O1s for the LaFeO₃ samples calcined at 850 and 1,050 °C

of the edge position can be used to determine the valence state. From Fig. 8, edge position of Fe₃O₄ (Fe²⁺, Fe³⁺) standard is approximately 7,124 eV, while Fe₂O₃ (Fe³⁺) standard is approximately 7,125 eV. In addition, edge position of FeO (Fe²⁺) standard (as not shown) is approximately 7,120 eV. These can be used simply as a fingerprint of phases and valence state. It is seen that the edge positions of all the samples were upper than those of

the Fe³⁺ at approximately 7,126 eV. Thus, this result indicated that most of the Fe ions in our samples are in a mixed valence state of Fe³⁺ and Fe⁴⁺, which agree with the XPS results.

The magnetization curves obtained from VSM measurements for all LaFeO₃ samples measured at RT with magnetic field (H) in the range of ±10 kOe are shown in Fig. 9. All the LaFeO₃ samples exhibit weak ferromagnetic

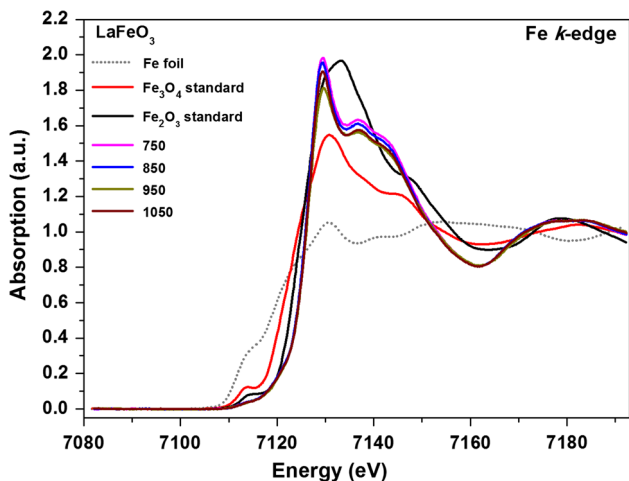


Fig. 8 XANES spectra at the Fe K absorption edge of the Fe foil, Fe₃O₄ (Fe²⁺, Fe³⁺) standard, Fe₂O₃ (Fe³⁺) standard and LaFeO₃ samples calcined at different temperatures for comparison

behaviour with the highest magnetization (M) at 10 kOe of ~1.0 emu/g, H_c of ~25–125 Oe. The magnetizations of the LaFeO₃ samples in this work are summarized in Table 1. It can be seen that nanoparticles of LaFeO₃ show an increase of M with decreasing nanoparticle size as

44.5 ± 2.4 nm in size. This result is of great interest because bulk LaFeO₃ is antiferromagnetic behavior due to the superexchange interactions between these neighboring Fe³⁺ ions of Fe³⁺–O²⁻–Fe³⁺. The origin of ferromagnetism in our LaFeO₃ nanoparticles could be a result of several reasons. The first possibility is because of the large fraction of uncompensated spins from the surfaces of the nanocrystals and the canted internal spin. This behavior has been also observed in other perovskites LaFeO₃ samples as well as in BiFeO₃ [33, 34] and YFeO₃ [35]. Therefore, for ferromagnetism in perovskites oxides, the small crystallite size is also important consideration because of the increase in surface area. The second is the double exchange (DE) interaction, which is often reported for several oxides. In this work, the XPS and XANES results show the evidence of Fe³⁺ and Fe⁴⁺ ions in LaFeO₃, which can be attributed to the mechanism of Fe³⁺–O²⁻–Fe⁴⁺ ions for DE interaction. Therefore the RT-FM in these samples is suggested as being a result of the FM according to the DE. Finally, ferromagnetism may be originated by TM clusters such as Fe, FeO or Fe₂O₃. Fe is a well-known ferromagnetic material, as FeO (T_N ~ 200 K) and Fe₂O₃ (T_C ~ 240 K) show antiferromagnetic or weak ferromagnetic behavior at low temperature [36]. Our FM results are not from the presence of TM metal clusters because if TM clusters exist

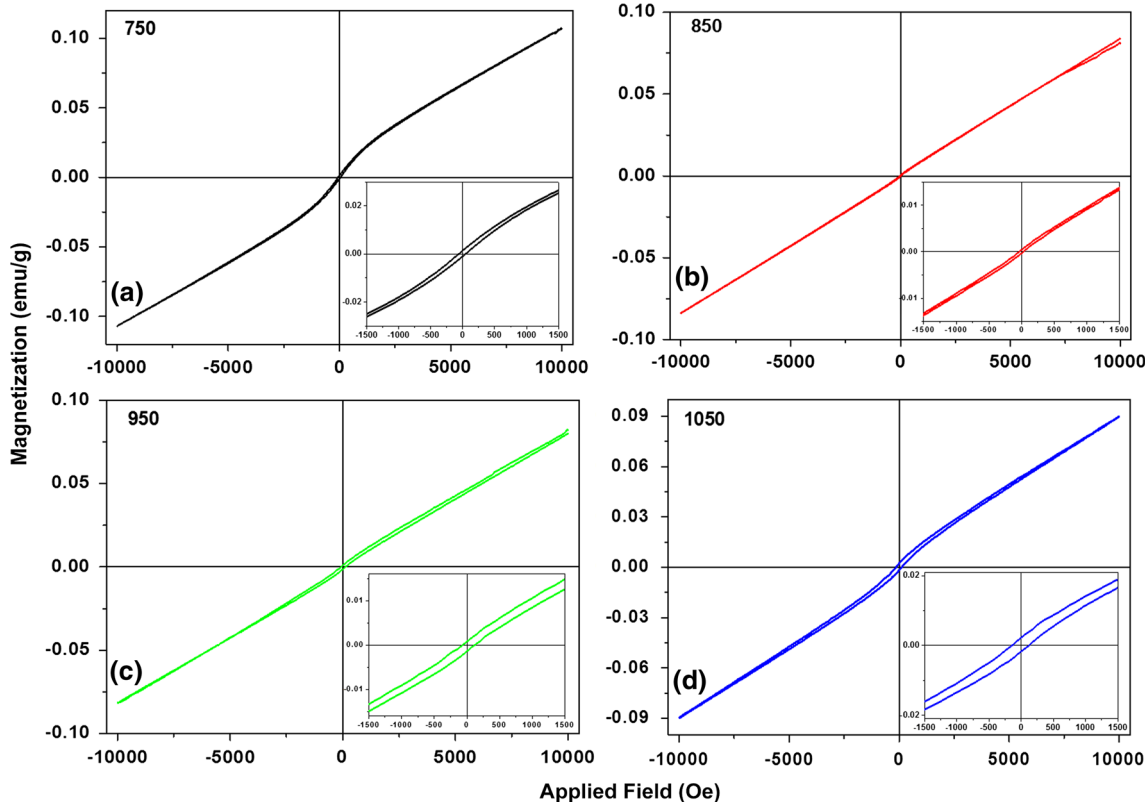


Fig. 9 M–H curves at ±10 kOe measured at RT of the LaFeO₃ samples

the magnetic moment would be proportional to the amount of TM concentration. In comparison to other works, the M value of 0.1 emu/g in this work is lower than the value of 0.38 emu/g for LaFeO_3 nanoparticles (~ 21.9 nm) synthesized by sol–gel reported by Saad et al. [16], the value of 0.44 emu/g for LaFeO_3 nanoparticles (~ 50 nm) synthesized by milling method reported by Thuy et al. [18], and the value of 0.9 emu/g for LaFeO_3 nanofibers (~ 20 nm) synthesized by electrospinning reported by Lee et al. [37]. The difference in M of LaFeO_3 materials depends on the preparation conditions or size of nanoparticles or surface of the nanoparticles. However, further work is needed to study on the bond length and bond angle between Fe and O, and this will be of great interest to understand the ferromagnetic coupling on the DE mechanism.

4 Conclusions

LaFeO_3 nanoparticles have been successfully prepared by PC method. Structural characterization showed that the structure had a single phase of orthorhombic LaFeO_3 . The nanoparticles consisted of the particles of approximately 44.5 ± 2.4 – 74.1 ± 4.9 nm. UV–Vis spectra showed the prepared LaFeO_3 nanoparticles having a strong UV-light absorption. XPS and XANES spectra further confirmed the main composition of the Fe ions were in mixed valence states of Fe^{3+} and Fe^{4+} . Study of magnetic properties at RT showed that LaFeO_3 samples with the smallest particle of 44.5 ± 2.4 nm exhibited soft ferromagnetic behavior with magnetization at 10 kOe of ~ 0.1 emu/g. This behavior indicated that ferromagnetism is due to the uncompensated spins at the surface and the canted internal spin, which is the nature of size-induced magnetism on nanoparticles.

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