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Synthesis of mesoporous $SnO₂$ nanomaterials with selective gas-sensing properties

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Abstract Using SBA-15/KIT-5/KIT-6 as the hard templates, the mesoporous $SnO₂$ nanomaterials with different structures were synthesized by nanocasting. X-ray diffraction, transmission electron microscopy, and nitrogen adsorption isotherms were used to testify their structure characteristics. These mesoporous $SnO₂$ nanomaterials showed high specific surface areas $(57-96 \text{ m}^2 \text{ g}^{-1})$ and pore volume $(0.17-0.27 \text{ cm}^3 \text{ g}^{-1})$. The nanopore of these templates makes the nanosize particle of the final mesoporous $SnO₂$ nanomaterials (4–9 nm) at last. The sensing properties of acetone, ethyl alcohol and methyl alcohol were investigated. The response of $SnO₂-15$, $SnO₂-5$, and $SnO₂$ -6 are 17.0, 19.5, and 16.1, respectively as the concentration of ethyl alcohol on 200 ppm. The sensitivity of $SnO₂ - 5$ is 28.2 as the concentration of acetone was increased to 200 ppm. With the large surface area, high pore volume, and nanosized particles (close to $2 L = 6$ nm of $SnO₂$, the $SnO₂-5$ show four fold enhancement in sensitivity compared to commercial $SnO₂$ powder and low detection limit (even at 200 ppb). The surface area and particle size play a significant party in the gas response. With the large surface area and smallest particle size, $SnO₂ - 5$ shows the highest sensitivity of all. These mesoporous nanomaterials show well potential application on the gas response.

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L. Li e-mail: leilei82701334@163.com **Keywords** $SnO₂ \cdot Mesoporous \cdot Replicated$ synthesis - Gas sensor

1 Introduction

Since the successful synthesis of MCMs in 1992 [[1,](#page-9-0) [2](#page-9-0)], mesoporous materials have attracted much attentions due to their highly ordered structure, tunable pore size, high pore volume, and large surface area [\[3–7](#page-9-0)]. These materials have been widely used in adsorption, separation, nanoreactors, drug carriers, and catalysis etc. Owing to the special structure advantages, mesoporous silica materials were usually used as the host to disperse the guest components to improve their properties on various practical applications. Li and Kawi [\[8\]](#page-9-0) utilized MCM-41 as a support to mechanically mix with $SnO₂$, and the sensitivity of the resulting sensor to H_2 could be enhanced compared with that of pure $SnO₂$ gas sensor.

Considering the advantage of mesoporous structure materials, the mesoporous materials of carbon, organic compound, metal oxide, and metal sulfide also have been synthesized, followed the mesoporous silica materials. Ryoo's group reported the preparation of the ordered mesoporous carbon using highly ordered mesoporous silica [\[9](#page-9-0)]. With the excellent optical and electric properties, the synthesis of mesoporous semiconductor metal oxide has become research foucs on catalysis, electrics, and gas sensor etc.

Generally, there are two ways to synthesize the mesoporous metal oxide, including soft template and hard template methods. For the soft template method, the surfactant was used as the template by solvent evaporation method. After calcinations to remove the surfactant, the mesoporous structures are disordered and collapsed easily

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[\[10](#page-9-0)]. For the hard template method, mesoporous carbon materials were usually for mesoporous metal oxide because the carbon can be removed under calcinations and the metal oxide framework remained. But the hydrophobic property of carbon always makes it difficult for mesoporous carbon material to be filled with the metal precursor completely. Huang et al. [\[11\]](#page-9-0) used SBA-15 as template to successfully synthesize In_2O_3 nanorods mesoporous materials with diameter of about 7 nm. Wagner et al. [[12\]](#page-9-0) synthesized mesoporous ZnO successfully.

SnO2, an n-type semiconductor with a wide band-gap $(Eg = 3.6 \text{ eV})$, has been widely used in solar cells $[13-16]$, catalysis $[17–19]$ $[17–19]$ $[17–19]$, and gas sensors $[20–29]$ $[20–29]$. To increase the efficiencies for these applications [\[30–32](#page-10-0)], mesoporous structure $SnO₂$ is required to provide an enhanced the surface area and pore volume. Most previous efforts [\[18](#page-9-0), [19,](#page-10-0) [25–29,](#page-10-0) [33](#page-10-0)] are focused on the study of structure and componentdependent properties. Therefore, it is meaningful to study the gas-sensing properties of mesoporous $SnO₂$.

In this paper, SBA-15, KIT-5, and KIT-6 were employed as the hard templates to replicate the mesoporous $SnO₂$ nanomaterials with different structure. These mesoporous nanomaterials were characterized by power X-ray diffraction (XRD), N_2 adsorption and desorption, and transmission electron microscope (TEM), and Raman. The sensing properties of these mesoporous $SnO₂$ nanomaterials to ethanol, methanol and acetone were detected in detail. And sensor mechanisms were discussed as well.

2 Experimental

2.1 Chemicals

Poly(ethylene glycol)–block–poly(propylene glycol)–block– poly(ethylene glycol) triblock copolymer Pluronic P123 $(Mw = 5,800, \quad EO_{20}PO_{70}EO_{20})$ was purchased from Aldrich Corp. Pluronic F127 triblock copolymer [(HO(CH2- CH_2O ₁₀₆–(CH₂CH(CH₃) O)₇₀–(CH₂CH₂O)₁₀₆H, EO₁₀₆ $PO_{70}EO_{106}$] was purchased from Sigma. SnCl₄:5H₂O was purchased from Guoyao Chemical Co. HCl was purchased from Guangfu Industry of Fine Chemicals Institute, Tianjin. H2SO4, was purchased from Fuchen Chemical Co., Tianjin. Tetraethyl orthosilicate (TEOS) was purchased from Tiantai Chemical Co., Tianjin. $C_4H_{10}O$ was purchased from Guangfu Industry of Fine Chemicals Institute, Tianjin. NaOH was purchased from Dalu Chemical Co., Tianjin. Commercial $SnO₂$ powder was purchased from Guangfu Industry of Fine Chemicals Institute, Tianjin.

All chemicals were analytical reagents and used as received without any further treatment. Deionized water was used in all experiments.

2.2 Preparation of template

SBA-15 was synthesized by using a non-ionic oligomeric alkyl-ethylene oxide surfactant (Pluronic 123) as the structure directing agent [[34\]](#page-10-0). 4 g of P123 was dissolved in 30 g of deionized water with stirring. After the polymer had been dissolved, 120 g of 2 M HCl solution, and 8.5 g of tetraethyl orthosilicate (TEOS) was added under stirring at room temperature. After the system was further stirred at 35 \degree C for 20 h, the mixture was transferred into an oven at 100 \degree C for 2 days. And then, filtrated and washed by plenty of deionized water and dried. Calcinations at 550 \degree C for 5 h in air remove the surfactant (SBA-15).

KIT-6 [\[35](#page-10-0)] was synthesized by using P123 as the structure directing agent. 6 g of P123 was dissolved in 217 g of deionized water and 10 ml HCl solution, and then 6 g of butanol ($C_4H_{10}O$) was added under stirring at 35 °C. After stirring 1 h, 12.9 g of tetraethyl orthosilicate (TEOS) was added. The mixture was left under stirring for 24 h. Then the mixture was transferred into an oven at 100 $^{\circ}$ C for 1 day. Upon cooling down to room temperature, filtrated, and washed by HCl and C_2H_5OH , dried. Then calcinations at 550 \degree C for 5 h in air remove the template.

KIT-5 [\[36](#page-10-0)] was synthesized by using F127 as the structure directing agent. 5 g of F127 was dissolved in 240 g of deionized water and 10.5 ml HCl solution. Then 24 g of TEOS was added under stirring at 45° C. After stirring 24 h, the mixture was transferred into an oven at 100 \degree C for 1 day. Upon cooling down to room temperature, filtrated, and washed by plenty of deionized water, dried. Then calcinations at 550 \degree C for 5 h in air remove the template.

2.3 Synthesis of mesoporous $SnO₂$ nanomaterials

 $SnCl₂2H₂O$ was dissolved in 10 g of ethanol. This solution was mixed with 1 g of the template and stirred at room temperature until all of the solution was adsorbed. After drying at room temperature, this process (impregnation and drying) was repeated to obtain the $SnO₂$ precursor/SiO₂. The composite was further annealed at 500 \degree C for 3 h and the silica template was removed from the $SnO₂/SiO₂$ composite by treating with dilute 2 M NaOH solution for 6 h and washing with deionized water and ethanol several times. The mesoporous $SnO₂$ nanomaterials replicated from SBA-15, KIT-5, and KIT-6 are named as $SnO₂-15$, $SnO₂-5$, and $SnO₂-6$, respectively.

2.4 Measurement of gas sensing properties

The as-synthesized sample was mixed with deionized water to form the paste. The paste was coated on a ceramic tube on which a pair of gold electrodes, and then a Ni–Cr

data acqisition system

heating wire was inserted in the tube to form a side-heated gas sensor.

Gas sensing properties were measured using a static test system which included a test chamber and a data acquisition/processing system. Dry air was used as both a reference gas and a diluting gas to obtain desired concentrations of target gases. The target gas was injected into the test chamber by a syringe through a rubber plug. When the response reached a constant value, the sensor was taken out to recover in dry air and the experiments were repeated. The gas response was defined as Ra/Rg , where Ra and Rg are the resistances of the sensor upon exposure to dry air and target gas, respectively. The response time was specified as the time to rise to 90 % of the equilibrium value of sensor resistance after target gas was injected. The recovery time was defined as the time to fall to 10 % of the final resistance value after the removal of target gas [[37](#page-10-0)]. In Fig. 1 A schematic diagram of the experimental setup this paper, we selected 1 L volume jars for response and Fig. 1 A schematic diagram of the experimental setup

Fig. 2 a Small-angle XRD patterns of a SBA-15, b KIT-5 and c KIT-6; b Small-angle XRD patterns of a SnO₂-15, b SnO₂-5 and c SnO₂-6 and C Wide-angle XRD patterns of a $SnO₂-15$, b $SnO₂-5$ and c $SnO₂-6$

recovery time measurement. And we put the sensor into the jars after the gas concentration reach equilibrium and remove it into air jar after test. A schematic diagram of the experimental setup are shown in Fig. [1](#page-2-0).

2.5 Characterization

X-Ray Diffraction data were collected on a SIEMENSD 5005 diffractometer with Cu Ka, using a radiation at 40 kV and 30 mA. The nitrogen adsorption/desorption, surface areas, and median pore diameters were measured using a Micromeritics ASAP 2010 M sorptometer. Before measured at 77 K, the samples were degassed at 473 K for 12 h. Specific surface areas and pore size distributions were calculated using the Brunauer–Emmett–Teller (BET) and Barrett–Joyner–Halenda (BJH) models from the adsorption branches, respectively. TEM images were recorded on JEOL 3010. The Raman spectrum was measured on a VERTEX-70 BRUKER RAM II spectrophotometer with an Nd:YAG laser at the excitation wavelength of 1,064 nm.

3 Results and discussion

3.1 Structural

Figure [2](#page-2-0)A shows small-angle XRD patterns of SBA-15, KIT-5, and KIT-6. As shown in Fig. [2a](#page-2-0), it exhibits three

Table 1 Textural properties of the ordered mesoporous SnO₂ nanomaterials

	Sample $S_{BET} m^2 g^{-1}$ Pore volume $cm^3 g^{-1}$ Pore diameter nm	
$SnO2-15$ 96	0.21	3.7
$SnO2-5$ 77	0.27	3.8
$SnO2-6$ 57	0.17	3.5

well-resolved peaks corresponding to the (100), (110), and (200) typical diffractions of 2D-hexagonal symmetry [\[38](#page-10-0)]. Figures [2](#page-2-0)b and [2c](#page-2-0) display the cubic $Im \overline{3}m$ and Ia $\overline{3}d$ structured KIT-5 and KIT-6, both possess highly ordered mesoporous structures. Figure [2](#page-2-0)B shows small-angle XRD pattern of $SnO₂$ samples. The (100), (110), and (200) diffractions are observed, implying $SnO₂-15$ replicated the structure of SBA-15. While for $SnO₂-5$ and $SnO₂-6$, the peak intensity is weaker, exhibiting the less ordered mesoporous structure. Figure [2](#page-2-0)C reveals the wide-angle XRD patterns of $SnO₂-15$, $SnO₂-5$, and $SnO₂-6$ composites. The peaks at 26.8°, 34.1°, 38.6°, and 51.9° is ascribed to the typical (110), (101), (200), and (211) diffractions of rutile structure of $SnO₂$ (JCPDS 41-1445). No diffraction peaks from any other impurities are detected.

The nitrogen adsorption isotherms of mesoporous $SnO₂$ nanomaterials are shown in Fig. 3a. All sample show the typical IV isotherm curve, revealing the mesoporous structure of them. This is consistent with the small angle XRD pattern as described above. Figure 3b shows the BJH pore diameter distribution of these samples, demonstrating the uniform pore distribution of them. As calculated by the BJH method from the desorption branch of the nitrogen isotherm, the corresponding pore diameter is 3.7, 3.8, 3.5 nm for $SnO₂-15$, $SnO₂-5$ and $SnO₂-6$, respectively. The structure features of all the nanomaterials are summarized in Table 1. The BET surface areas and pore volumes of them are 96 m^2g^{-1} and 0.21 cm^3g^{-1} , 77 m^2g^{-1} and 0.27 cm³g⁻¹, and 57 m²g⁻¹ and 0.17 cm³g⁻¹, respectively. With the mesoporous structure, all the samples have the large surface area and high pore volume compared with other common $SnO₂$ materials [\[39](#page-10-0)].

The detailed microstructures of the as-synthesized mesoporous $SnO₂$ nanomaterials were investigated by TEM. The TEM images of $SnO₂-15$, $SnO₂-5$ and $SnO₂-6$ are shown in Fig. [4](#page-4-0)a–i. From the Fig. [4](#page-4-0)a, b, the typical stripe-

Fig. 3 Nitrogen adsorption–desorption isotherms a and pore diameter distribution curves b of mesoporous $SnO₂$ nanomaterials

like and hexagonally arranged TEM images, viewed from the $[100]$ and $[110]$ directions, are observed in SnO₂-15. And their particle size is about 7–9 nm. From Fig. 4d, e, g, h, $SnO₂-5$ and $SnO₂-6$ show the less ordered structure with the particles size of 4–6 nm and 5–7 nm, respectively. After the removing of the template, the mesoporous structure damage in $SnO₂-5$ and $SnO₂-6$. A high resolution TEM images (Fig. 4c, f, i) show the clearly lattice fringes with d spacing values are calculated to be about 0.34 and 0.26 nm, corresponding to (110) and (101) planes of $SnO₂$ nanomaterials.

Raman scattering is a useful tool for the characterization of nanosized nanomaterials and a qualitative probe of the presence of lattice defects in solids [[40\]](#page-10-0). Meanwhile, the Raman spectrums of mesoporous $SnO₂$ nanomaterials are shown in Fig. 5. As in Fig. 5, three Raman active peaks located at 631, 777 and 475 cm^{-1} are consistent with the

Fig. 5 Raman spectrum of a $SnO₂-15$, b $SnO₂-5$, and c $SnO₂-6$

Fig. 4 TEM images of $a-c$ SnO₂-15, $d-f$ SnO₂-5 and $g-i$ SnO₂-6

 A_{1g} , B_{2g} and E_{g} vibration modes respectively. Besides, we also observed other Raman peaks of 354 (N₁) and 575 (N₂) cm^{-1} , which can be assigned to the surface defects of the SnO₂ crystals $[41]$ $[41]$.

3.2 Gas sensitivity properties

The gas sensing properties of the as-prepared mesoporous SnO2 nanomaterials for ethanol were studied. Firstly, the

Fig. 6 a Sensor sensitivity versus calcining temperature of the synthesized mesoporous SnO₂ nanomaterials on 50 ppm ethanol and b Sensor sensitivity versus operating current of the mesoporous $SnO₂$ nanomaterials on 10 ppm ethanol

Fig. 7 Response of a SnO₂-15, b SnO₂-5 and c SnO₂-6 sensors to 200 ppb–200 ppm ethanol

influence of calcining temperature on the sensing responses towards 50 ppm ethanol was investigated. The sensitivity varied with the calcining temperature is shown in Fig. [6](#page-5-0)a. From the curve, it can be seen that with the increase of the calcining temperature, the sensitivity increases. When the calcining temperature increases to 400 \degree C, the sensitivity reaches 7.0. Further increase of calcining temperature induces the sensitivity decrease obviously. Secondly, the effect of operating temperature on sensing responses towards 10 ppm ethanol was also investigated (Fig. [6](#page-5-0)b). From Fig. [6b](#page-5-0), the sensitivity increases with the increase of operating temperature, until at 240 $^{\circ}$ C it reaches its maximum of 5.2. When exceed 240 \degree C, the sensitivity decrease obviously. From the above investigations, the calcinations at 400 \degree C and the operating temperature at 240 \degree C was used for all the samples in the follow experiments.

Figure [7](#page-5-0) shows the sensitivity of $SnO₂-15$, $SnO₂-5$, and $SnO₂-6$ as the function of ethanol concentration. From the curves, it can be seen that the sensitivity of the sensor undergoes a drastic rise upon the injection of ethanol and drops to its initial state after the sensor was exposed to air. As can be seen in Fig. [7a](#page-5-0), the sensitivity of $SnO₂-15$ on 200 ppb ethanol is 1.2, the sensitivity increased to 17.0 as the concentration of ethanol was increased to 200 ppm. As shown in Fig. [7b](#page-5-0), the sensitivity of $SnO₂-5$ on 200 ppb ethanol is 1.3, the sensitivity increased to 19.5 as the concentration of ethanol was increased to 200 ppm. Whereas, the sensitivity of $SnO₂-6$ (Fig. [7](#page-5-0)c) on 200 ppb ethanol is 0.9, the sensitivity increased to 16.1 as the concentration of ethanol was increased to 200 ppm.

From the above investigation, $SnO₂-5$ showed the highest sensitivity of all. Two main reasons are ascribed. One is the grain size effect of sample for sensitivity. It is known that, when $D \le 2 L$ (L is the thickness of the depletion layer), the gas sensing perform better. The $2 L$ of $SnO₂$ $(L = 3 \text{ nm})$ is 6 nm. From Fig. [4](#page-4-0), the average particle size followed SnO_2 -5 (4–6 nm) \lt SnO₂-6 (5–7 nm) \lt SnO₂-15 (7–9 nm). On the other hand, the surface area also plays an important party in the sensitivity. From Table [1,](#page-3-0) we can found that $SnO₂-15$ shows the largest surface area

Fig. 8 Response of SnO₂-5 sensor to a 200 ppb–200 ppm methanol and b 200 ppb–200 ppm acetone, c Response of SnO₂-5 sensor to different concentrations of methanol, ethanol and acetone

(96 m²g⁻¹), later is SnO₂-5 (77 m²g⁻¹), and SnO₂-6 possesses the lowest surface area $(57 \text{ m}^2 \text{g}^{-1})$ among all. With smallest particle size $(4–6 \text{ nm})$ and moderate surface area, $SnO₂ - 5$ shows the highest sensitivity. Although the grain size of $SnO₂-15$ is biggest, the largest surface area also makes $SnO₂-15$ show the excellent sensitive. From the above investigation, the mutual effect of grain size and the surface area determine the final sensitivity.

The response of $SnO₂-5$ to different concentrations of methanol and acetone are shown in Fig. [8](#page-6-0)a, b. The sensitivity increases with the increase of methanol and acetone concentration. The response slowly increases with increasing methanol and acetone concentration below 10 ppm. Above 10 ppm, the response rapidly increases with the methanol concentrations. As can be seen in Fig. [8](#page-6-0)a, the sensitivity of $SnO₂-5$ to 200 ppb methanol is 1.0, the sensitivity increased to 14.7 as the concentration of methanol was increased to 200 ppm. As shown in Fig. [8b](#page-6-0), the sensitivity of $SnO₂-5$ is 1.3 at 200 ppb acetone, the sensitivity increased to 28.2 as the concentration of acetone

was increased to 200 ppm. As shown in Fig. [8](#page-6-0)c, the sensor exhibits the highest sensitivity to acetone than the others, implying a good response to acetone, three main reasons are ascribed.

1. Increases oxygen vacancies resulting $CH₃COCH₃$ amount, and thus greater reduced resistance of the material, significantly improving the sensitivity of the gas material on the $CH₃COCH₃$.

$$
C_2H_5OH + 6O^-(ads) \longrightarrow 2CO_2 + 3H_2O + 6e^-
$$

\n
$$
CH_3COCH_3(g) + 8O^-(ads) \longrightarrow 3CO_2 + 3H_2O + 8e^-
$$

\n
$$
CH_3OH + 3O^-(ads) \longrightarrow CO_2 + 2H_2O + 3e^-
$$

- 2. Good for gas sensing properties of acetone may be attributed to acetone gas contains two strong electronwithdrawing group, easier for SnO₂ adsorption,
- 3. Different required for gas distribution volume of liquid.

Fig. 9 Response and recovery *curves* of a $SnO₂-15$, b $SnO₂-5$ and c $SnO₂-6$ to 50 ppm ethanol

Fig. 10 Sensitivities of the SnO₂-5 and commercial SnO₂ powder based sensors to acetone, ethanol and methanol, respectively

As the molecular weight, density, we configured with 100 ppm ethanol, methanol, acetone, the desired volume of liquid, respectively: 5.2 , 2.37 , 6.62 μ L. Acetone is relatively large volume of liquid required.

In addition, the response and recovery time were also investigated for the mesoporous $SnO₂$ nanomaterials based sensors. The response and recovery times were 10.8 and 7.3 s for $SnO₂ - 15$, 8.7 and 6.5 s for $SnO₂ - 5$, 12.6 and 8.8 s for $SnO₂-6$, respectively, as shown in Fig. [9.](#page-7-0) It known that, KIT-5 and KIT-6 show the typical 3D structure. $SnO₂-5$ and $SnO₂$ -6 templated from KIT-5 and KIT-6 also show the 3D structure characters. From the data, $SnO₂$ -5 shows the fastest response due to their 3D structure, which facilitates the diffusion of gas molecules. Although $SnO₂-6$ also possesses 3D porous structure, the low surface area limits the response.

In order to evaluate the performance of these mesoporous $SnO₂$ on sensors, commercial $SnO₂$ powder was used for contrast (Fig. 10). From Fig. 10a, the response of commercial $SnO₂$ powder is 5.9 (acetone 200 ppm),

whereas $SnO₂$ -5 sensor is 28.2. As can be seen in Fig. 10b, at ethanol concentration of 200 ppm, the response of commercial $SnO₂$ powder is 5.3, whereas $SnO₂$ -5 sensor is 19.5. As shown in Fig. 10c, the response is 4.3 and 14.7 for commercial $SnO₂$ and $SnO₂-5$ at 200 ppm methanol. With the special mesoporous structure, show the $SnO₂-5$ exhibits the obviously enhanced sensitivity compared with commercial SnO₂ powder.

The long-time stability of this sensor was investigated by repeating the test after aging. Figure [11](#page-9-0) shows the sensitivity of the sensor to 200 ppm methanol, ethanol and acetone within 100 days. As shown in Fig. [11,](#page-9-0) the mesoporous $SnO₂$ nanomaterials shows the well stability for the three gases after 100 days.

3.3 Gas sensor mechanism

The sensing mechanism of sensors can be explained as follows.

Fig. 11 Stability of $SnO₂-5$ to 200 ppm methanol, ethanol and acetone

N-type semiconductor adsorbed oxygen molecules in the air, the oxygen molecules trapping electron in the surface of the semiconductor, thereby forming oxygen anions (O^-, O^{2-}) , this results in an increase in resistance semiconductor surface. When the measured gas in contact with the semiconductor surface and react with the adsorbed oxygen, electron captured by oxygen to return to the semiconductor, then resistance is reduced, such significant changes in the resistance of the semiconductor on the realization of the gas detection.

The related reactions are expressed as follows:

$$
O_2(g) \longleftrightarrow O_2(ads) \tag{1}
$$

$$
O_2(ads) + e^- \longleftrightarrow O_2^-(ads) \tag{2}
$$

$$
O_2^-(ads) + e^- \longleftrightarrow 2O^-(ads) \tag{3}
$$

$$
O^{-}(ads) + e^{-} \longrightarrow O_{2}^{-}(ads)
$$
\n(4)

$$
C_2H_5OH + 6O^-(ads) \longrightarrow 2CO_2 + 3H_2O + 6e^-
$$
 (5)

 $CH_3COCH_3(g) + 8O^- \longrightarrow 3CO_2 + 3H_2O + 8e^-$ (6)

$$
CH3OH + 3O-(ads) \longrightarrow CO2 + 2H2O + 3e- (7)
$$

4 Conclusions

Mesoporous $SnO₂$ nanomaterials were synthesized by replicate hard templates SBA-15/KIT-5/KIT-6, respectively. These mesoporous $SnO₂$ nanomaterials showed high BET surface areas and pore volume. The corresponding BET surface areas and pore volumes for $SnO₂-15$, $SnO₂-5$ and SnO₂-6 are 96 m^2g^{-1} and 0.21 cm^3g^{-1} , 77 m^2g^{-1} and 0.27 cm³g⁻¹, and 57 m²g⁻¹ and 0.17 cm³g⁻¹, respectively. In particular, $SnO₂$ nanoparticles show a strong sizeinduced enhancement of gas sensing performance. The response of $SnO₂-15$, $SnO₂-5$, and $SnO₂-6$ are 17.0, 19.5, 16.1, respectively as the concentration of ethanol on 200 ppm. The $SnO₂ - 5$ exhibit a four fold enhancement in sensitivity compared to commercial $SnO₂$ powder on acetone. With the large surface area and small particle size, $SnO₂ - 5$ showed the best response of all. The mesoporous SnO2 nanomaterials exhibit low detection limit (ppb level), a complete and fast recovery to its initial state, and longterm sensing stability, this could be ascribed to the grain size, the surface areas, and the porous structure. With the large surface area, and particle size, these mesoporous $SnO₂$ nanomaterials have potential application on catalysis, adsorption and batteries.

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