

# CAM sol–gel synthesized $\text{LiMPO}_4$ (M=Co, Ni) cathodes for rechargeable lithium batteries

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**Abstract** The emerging category cathode candidates such as  $\text{LiCoPO}_4$  and  $\text{LiNiPO}_4$  were synthesized at 800 °C using Citric acid assisted modified sol–gel (CAM sol–gel) method and examined for possible lithium intercalation behavior. Compound formation temperature is confirmed from thermogravimetry and differential thermal analysis (TG/DTA). Powder X-ray diffraction (PXRD) pattern evidenced the absence of undesirable peaks and confirmed the formation of phase pure  $\text{LiMPO}_4$  (M=Co, Ni) compounds with an orthorhombic structure and finer crystallite size. Presence of nanosized particles as observed from TEM image of  $\text{LiCoPO}_4$  and the presence of preferred local cation environment as understood from FT–IR studies are the added advantages of CAM sol–gel synthesis. Further, Cyclic voltametry (CV) and Impedance spectroscopy (EIS) studies performed on the synthesized  $\text{LiCoPO}_4$  and  $\text{LiNiPO}_4$  cathodes revealed excellent reversibility and structural stability of CAM sol–gel synthesized cathodes, especially upon storage as well as during cycling.

**Keywords** CAM sol–gel method ·  $\text{LiCoPO}_4$  ·  $\text{LiNiPO}_4$  · XRD · Cyclic reversibility

## 1 Introduction

Rechargeable lithium batteries which are commonly used in consumer electronics possess salient features such as high energy density, less weight, no memory effect, negligible self discharge, etc., due to which the same are popular in defense, automotive, and aerospace applications. Among the various components of lithium-ion batteries, certain cathodes viz.,  $\text{LiMO}_2$  layered oxide such as cobalt oxide or nickel oxide, spinel oxide such  $\text{LiMn}_2\text{O}_4$  type and a  $\text{LiMPO}_4$  type olivine such as lithium iron phosphate have been accepted as potential cathode candidates. Hereagain, layered  $\text{LiCoO}_2$ , though known as a successful commercial material in small size batteries, suffers from lack of safety upon overcharge [1] and the possibility of realizing lower deliverable capacity compared to higher theoretical capacity.  $\text{LiNiO}_2$  on the other hand is limited by its lower thermal stability, off-stoichiometry problem, and the requirement of stringent synthesis conditions. Similarly, the spinel  $\text{LiMn}_2\text{O}_4$  has no degree of freedom for opening the interstitial space of the close packed oxide-ion array in which the  $\text{Li}^+$  ions generally move [2] and suffers from Jahn-Teller distortion also, due to known reasons.

Regarding phosphate olivines, research is focused on the  $\text{LiFePO}_4$  cathode in the recent years, due to its inherent advantages such as safety, economic viability, eco benign nature, and abundant availability. But the poor electronic conductivity and stringent synthesis methodology of  $\text{LiFePO}_4$  are the major hampering issues against the commercialization of the same [3, 4]. To address the same, several attempts are being reported on the reduction of

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particle size [5], carbon nanopainting, otherwise known as coating or co-synthesizing of compounds with carbon [6–8], doping of polyvalent metal cations [9], etc., in order to improve the electrochemical properties of native  $\text{LiFePO}_4$ . However, the synthesis of  $\text{LiFePO}_4$  despite having reached the commercial status, still poses issues on the amount, type, and the mode of carbon addition to realize  $\text{LiFePO}_4$  with appreciable energy as well as power density required for the heavy electrical vehicles (HEV) applications. As a result, research on other possible olivine cathodes with transition metals like Co or Ni to form  $\text{LiCoPO}_4$  and  $\text{LiNiPO}_4$ , respectively, has gaining importance. Towards this direction, the present work is planned on the synthesis and characterization of  $\text{LiCoPO}_4$  and  $\text{LiNiPO}_4$  cathodes, especially through the less studied solution assisted method, namely citric acid assisted modified sol gel (CAM sol–gel) method.

It is well-known that the rate capability of lithium-ion batteries could be enhanced by way of reducing the particle size of electrode materials [10]. Also, it is well reported that the synthesis procedure plays a vital role in reducing the particle size and the related improvement in terms of physical as well as electrochemical properties of synthesized cathode materials [11]. Hence, CAM sol–gel method exploited so far against the synthesis of oxide cathode materials and known for producing meso structured compounds has been chosen for present study to synthesize newer category  $\text{LiMPO}_4$  ( $M=\text{Co}, \text{Ni}$ ) compounds [12].

Herein, the currently adopted CAM sol–gel method, based on solution approach ensures the uniform distribution and an atomic level mixing of reactants and gelling agents. Such an intimate mixing of precursors render phase pure final products with nanoparticle size, which is the significance of CAM sol–gel method. In this method, both the acrylamide and  $N,N'$ -methylene-bis-acrylamide act as gelling agents, whereas citric acid with a molecular formula of  $\text{C}_6\text{H}_8\text{O}_7 \cdot \text{H}_2\text{O}$  acts as a fuel for the formation of bond between the transition metal ions and phosphate polyanion as described schematically in Fig. 1. Originally, the carboxylic acid group that is present in the complexing agent citric acid forms a chemical bond with the metal ions, which upon evaporation of solvent leads to the formation of viscous gel like paste precursor. It is presumed that the metal ions are trapped inside the so called viscous gel to ensure atomic level mixing, which is the key point of the CAM sol–gel method.

Hence, CAM sol–gel synthesized cathode candidates are expected to exhibit enhanced battery performance, due to much shorter diffusion distances of the intercalating  $\text{Li}^+$  ions. Interestingly, CAM sol–gel method of the present study has resulted in the formation of nanosized  $\text{LiCoPO}_4$  and  $\text{LiNiPO}_4$  powders with desirable physical and electrochemical characteristics, which is in accordance with the focal theme of the present work.

## 2 Experimental details

### 2.1 Synthesis procedure

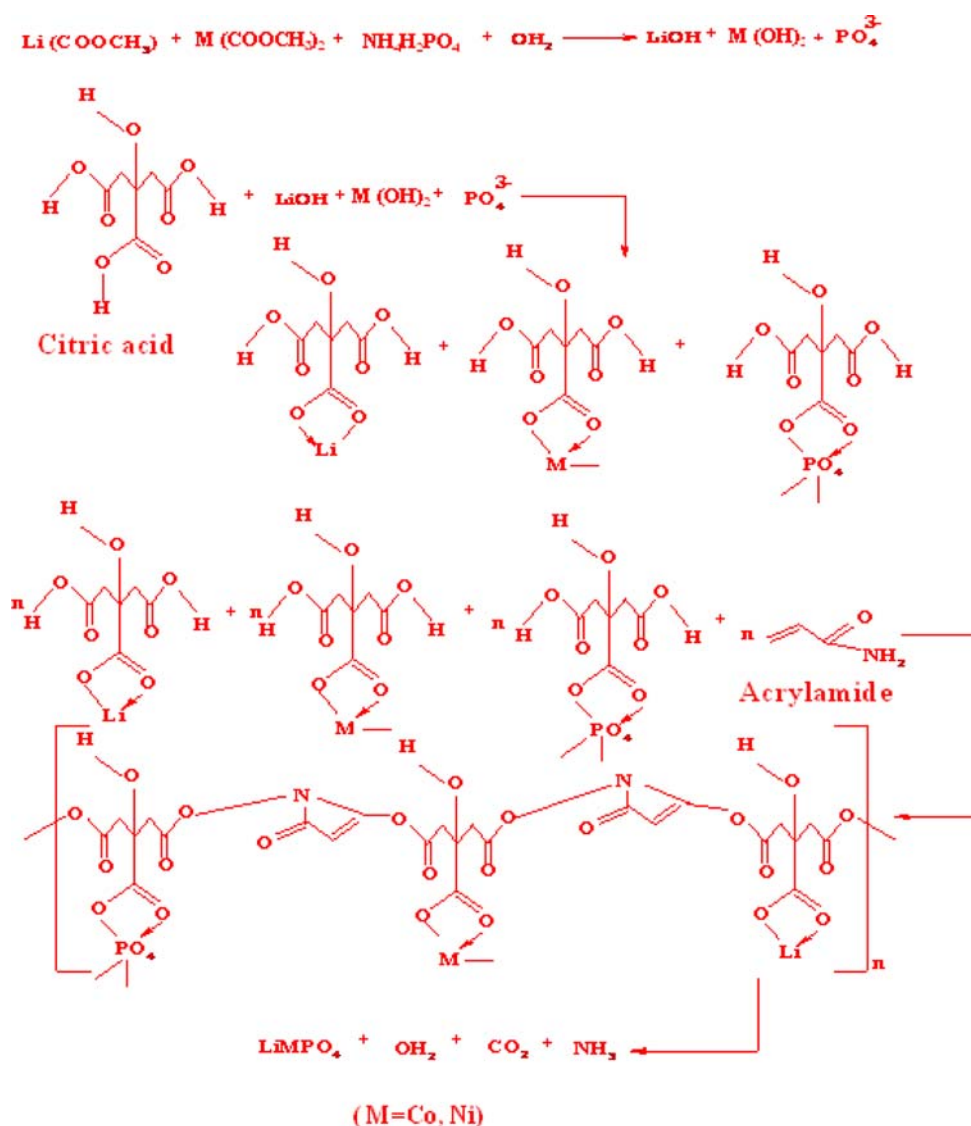
$\text{LiCoPO}_4$  and  $\text{LiNiPO}_4$  cathode powders were synthesized from stoichiometric ratios of the starting materials viz.,  $\text{LiOOCCH}_3 \cdot 2\text{H}_2\text{O}$  (Lithium acetate, Merck),  $(\text{CH}_3\text{COO})_2\text{Co} \cdot 4\text{H}_2\text{O}/(\text{CH}_3\text{COO})_2\text{Ni} \cdot 4\text{H}_2\text{O}$  (Cobalt acetate/Nickel acetate, Merck) and  $\text{NH}_4\text{H}_2\text{PO}_4$  (Ammonium dihydrogen phosphate, Merck). Primarily, the reactants were dissolved in hot water with stirring to get a homogeneous solution. To the solution, calculated amount of citric acid, acryl amide, and finally  $N,N'$ -methylene-bis-acrylamide were added and the process of stirring and heating were continued. At this point, it is quite interesting to note that as soon as the  $N,N'$ -methylene-bis-acrylamide was added, the viscosity of the homogeneous solution increased drastically and the same upon further heating and stirring has formed a transparent gel. Further, it is understood that citric acid acts as not only as a chelating agent but also as a fuel to reduce the heat energy supplied and the added acryl amide acts as a buffering agent to adjust the pH required for the transparent gel formation. Similarly, the added  $N,N'$ -methylene-bis-acrylamide facilitates the rapid formation of transparent gel, via. formation of meta stable chelated complex. Representative schematic diagram with a mechanism involved in the synthesis of  $\text{LiMPO}_4$  ( $M=\text{Co}, \text{Ni}$ ) using CAM sol–gel method is displayed in Fig. 1.

The thus formed gels were dried at  $110^\circ\text{C}$  for about 12 h in a hot air oven initially and the dried samples were further ground and furnace-heated to  $300^\circ\text{C}$  for about 6 h. The resultant powders were also ground again and fired further at  $800^\circ\text{C}$  for about 5 h using alumina crucibles. Herein, both the rate of heating and cooling in the furnace heating were maintained at  $10^\circ\text{C}/\text{min}$  to avoid surface cracking of the particles and to ensure the presence of uniformly distributed particles of nanosize. The ultrafine powders of  $\text{LiCoPO}_4$  and  $\text{LiNiPO}_4$  obtained after  $800^\circ\text{C}$  were collected and subjected to systematic physical as well as electrochemical characterizations.

### 2.2 Physical and electrochemical characterizations

Compound formation temperature was confirmed by thermogravimetry and differential thermal analysis (TG/DTA) with a thermo balance model STA 409 PC in the temperature range  $25\text{--}800^\circ\text{C}$ , using alumina crucibles, under air with a heating rate of  $20^\circ\text{C}/\text{min}$ . Phase characterization was done by powder X-ray diffraction technique on a Philips 1830 X-ray diffractometer using Ni-filtered  $\text{CuK}\alpha$  radiation ( $\lambda = 1.5406\text{\AA}$ ) in the  $2\theta$  range  $10\text{--}90^\circ$  at a scan rate of  $0.04^\circ/\text{s}$ . Fourier transform infrared spectroscopic (FT–IR) study was performed on a Perkin–Elmer paragon

**Fig. 1** Chemical reaction mechanism involved in the CAM sol-gel method to synthesize  $\text{LiMPO}_4$  ( $M=\text{Co}, \text{Ni}$ ) compounds



–500 FTIR spectrophotometer using a pellet containing a mixture of KBr and the active materials in the region  $400\text{--}2,000\text{ cm}^{-1}$ . Electron paramagnetic resonance (EPR) measurements were carried out using a Bruker instrument. Room temperature electrochemical studies such as cyclic voltammetry (CV) and Electrochemical Impedance Spectroscopy (EIS) measurements were performed using an Auto lab electrochemical workstation, wherein CV scan rate was fixed as  $1\text{ mV/s}$  and impedance analysis was done over a frequency range of  $100\text{ kHz--}0.1\text{ Hz}$ .

### 2.3 Electrode preparation and coin cell fabrication

To prepare cathode electrode, the synthesized powder was first mixed intimately with super P carbon black (additive) and polyvinylidene fluoride (binder) in 80:10:10 ratio, respectively, and the same was treated with *N*-Methylpyrrolidin-2-one (solvent) to form a slurry. The slurry was cast

uniformly on a thin aluminum foil ( $20\text{ }\mu\text{m}$ ) and then dried for about 2 h at  $110\text{ }^\circ\text{C}$ . The dried foil was hot pressed (under 5 ton pressure) and the circular electrodes were punched out from the coated foil. Using such cathodes, 2016 coin cells were assembled in an Argon-filled glove box and crimp sealed prior to electrochemical studies. CV and EIS characterizations were carried out on such type of 2016 coin cells consisting of lithium anode, synthesized  $\text{LiMPO}_4$  ( $M=\text{Co}, \text{Ni}$ ) cathode and a non-aqueous electrolyte containing 1 M  $\text{LiPF}_6$  dissolved in 1:1 v/v EC:PC with a cell guard separator.

## 3 Results and discussion

### 3.1 Thermal analysis

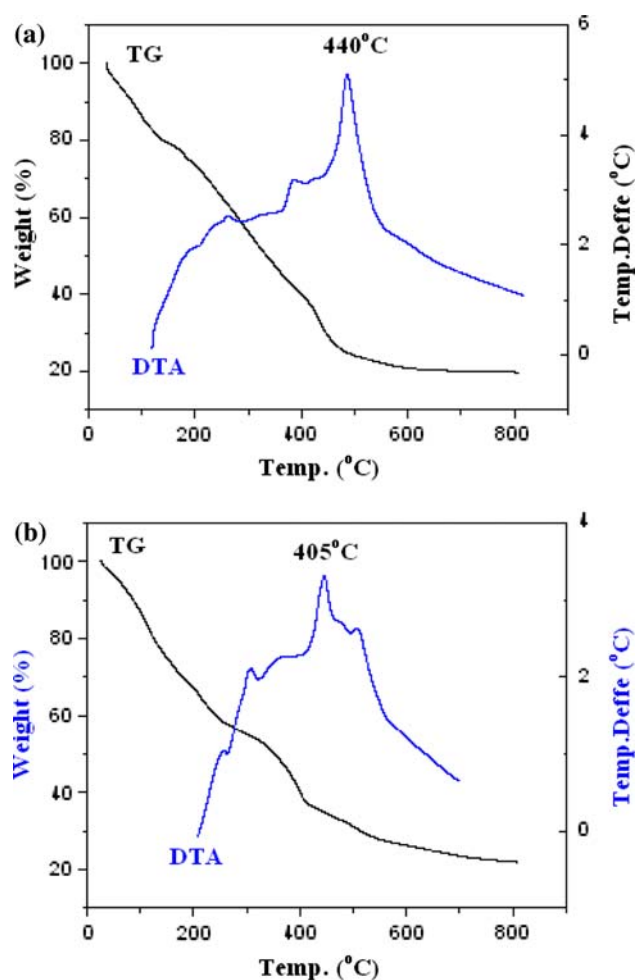
Figures 2a and b show the thermogravimetric and differential thermal analysis (TG/DTA) scans of the mixtures

containing the fine ground precursors of  $\text{LiCoPO}_4$  and  $\text{LiNiPO}_4$ , respectively. In the case of  $\text{LiCoPO}_4$ , TG curve shows a gradual and an initial weight loss up to about  $450^\circ\text{C}$  due to the initial evaporation of water (below  $200^\circ\text{C}$ ) and a subsequent decomposition of precursors ( $200\text{--}450^\circ\text{C}$ ) from the in-situ formed meta stable complex. In other words, the decomposition of citric acid and amides to release  $\text{CO}_2$  and  $\text{NH}_3$  followed by the final decomposition of chelated metal acetate are obvious from the larger exothermic peak observed at  $440^\circ\text{C}$  in the DTA curve. Herein, the compound formation process starts at a temperature as low as  $325^\circ\text{C}$  and the same gets completed around  $570^\circ\text{C}$  only. Since no significant weight loss has been observed especially beyond  $570^\circ\text{C}$ , it is deduced that the compound formation process gets completed well within  $570^\circ\text{C}$  itself. Such a thermal stability of  $\text{LiCoPO}_4$  beyond  $600^\circ\text{C}$  has already been reported by Chen et al. [13] and the current observations related to the TG/DTA behavior of chosen category precursors of  $\text{LiCoPO}_4$  are also in good agreement with the same. Hence, it is confirmed from this study that the complete compound formation of  $\text{LiCoPO}_4$  is taking place well below  $600^\circ\text{C}$  itself. However, the sample of the present study was heat treated further to  $800^\circ\text{C}$  in order to get phase pure and better crystalline product.

Figure 2b shows that the precursors of  $\text{LiNiPO}_4$  also exhibits a continuous weight loss up to  $550^\circ\text{C}$  due to the evaporation of water and the decomposition aided evaporation of gases like  $\text{CO}_2$  and  $\text{NH}_3$ , as in the case of  $\text{LiCoPO}_4$ . Similarly, it is obvious from Fig. 2b that the compound formation process starts at  $350^\circ\text{C}$  and gets completed at  $550^\circ\text{C}$ . After  $550^\circ\text{C}$ , negligible weight loss in TG curve has been observed which confirms the complete formation of phase pure  $\text{LiNiPO}_4$  compound at  $550^\circ\text{C}$  itself. Despite the same, as in the previous case, the powder of  $\text{LiNiPO}_4$  has also been heat treated further at  $800^\circ\text{C}$  for about 5 h, for reasons of realizing better purity and crystallinity.

### 3.2 Phase characterization

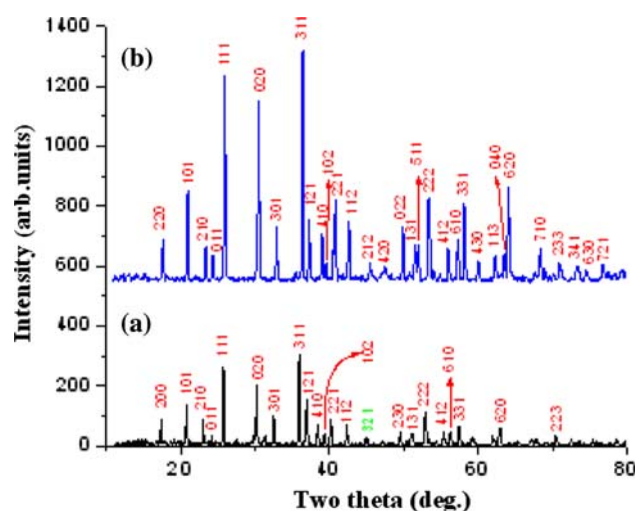
The XRD patterns of  $\text{LiCoPO}_4$  and  $\text{LiNiPO}_4$  compounds synthesized by CAM sol–gel method are shown in Figs. 3a and b, respectively. The existence of well defined and highly intense peaks with the absence of unknown Bragg peaks confirms the presence of crystalline and phase pure products of desired category. The miller indices (hkl) of all the peaks corresponding to those of  $\text{LiCoPO}_4$  and  $\text{LiNiPO}_4$  are indexed as per the JCPDS file numbers 89–6192 and 88–1297, respectively. From search match analysis, it is deduced that both the compounds possess orthorhombic structure with Pnma space group. Further, lattice parameter



**Fig. 2** a TG/DTA behavior of  $\text{LiCoPO}_4$  and b TG/DTA behavior of  $\text{LiNiPO}_4$

(a, b, c) values were calculated using iterative least square analysis and the corresponding cell volume values have also been calculated for  $\text{LiCoPO}_4$  and  $\text{LiNiPO}_4$  individually (Table 1).

Similarly, the average crystallite size ( $D$ ) has been calculated using Scherer's formula [14], which is in favor of significantly reduced nanometric crystallite size of  $\text{LiCoPO}_4$  and  $\text{LiNiPO}_4$  ( $<50\text{ nm}$ ). Herein, it is believed that the presence of nanometric crystallites of  $\text{LiCoPO}_4$  and  $\text{LiNiPO}_4$  are due to the impact of the CAM sol–gel method. In addition, Table 1 infers that the calculated stoke's strain values are also found to be very small, which is an indication for the presence of strain free crystal lattice in the CAM sol–gel synthesized phosphate olivines. As, it is deduced qualitatively from XRD (using Scherrer's formula) that the olivines synthesized using CAM sol–gel method are found to possess nanometric crystallite size, the same has been intended to verify from TEM studies also.



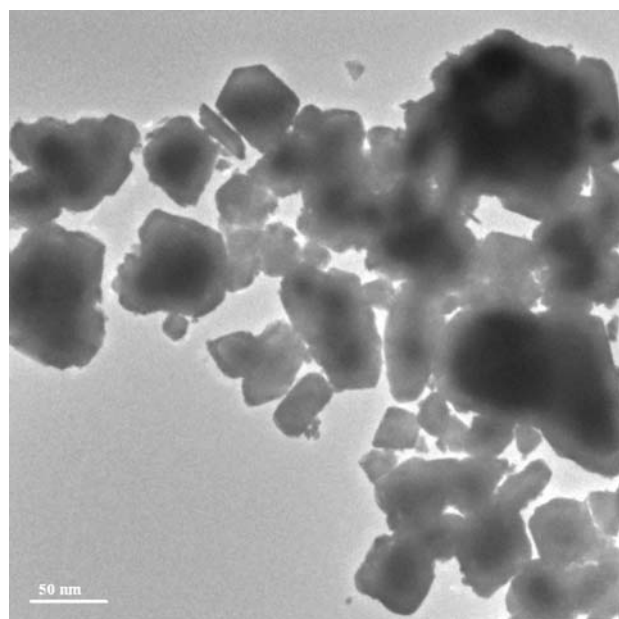
**Fig. 3** PXRD pattern of (a) LiCoPO<sub>4</sub> and (b) LiNiPO<sub>4</sub> synthesized by CAM sol-gel method

### 3.3 Transmittance electron microscope (TEM) study

In order to examine the actual size of the particles produced by CAM sol-gel method, TEM image was recorded for LiCoPO<sub>4</sub> and the result is displayed in Fig. 4. Interestingly, TEM picture of LiCoPO<sub>4</sub> exhibits the presence of 50 nm particles, thus substantiating the qualitative results derived from XRD. Hence, it is confirmed from TEM studies also that CAM sol-gel method is effective in reducing the size of the particles significantly, which is desirable for lithium inserting cathode materials.

### 3.4 Local cation environment by FTIR

Generally, Fourier transform infrared (FTIR) spectroscopy is sensitive to the short range environment of oxygen coordination around the cation in an oxide lattice. With regard to the factor group analysis and molecular vibration model [15] suggested to understand the structure of phospho-olivines, it is obvious from the current study also that the fundamental frequencies of PO<sub>4</sub><sup>3-</sup> polyanions are split in to many components ( $\nu_1$ – $\nu_4$ ) due to the correlation effect which in turn is induced by the coupling of M–O bonds. As a result, the vibration spectra of LiCoPO<sub>4</sub> and LiNiPO<sub>4</sub> are dominated by the fundamental vibrations of PO<sub>4</sub><sup>3-</sup> polyanions, which is not unusual [15]. Figure 5a shows four fundamental vibrations of PO<sub>4</sub><sup>3-</sup> such as an asymmetric



**Fig. 4** TEM image of LiCoPO<sub>4</sub> synthesized by CAM sol-gel method

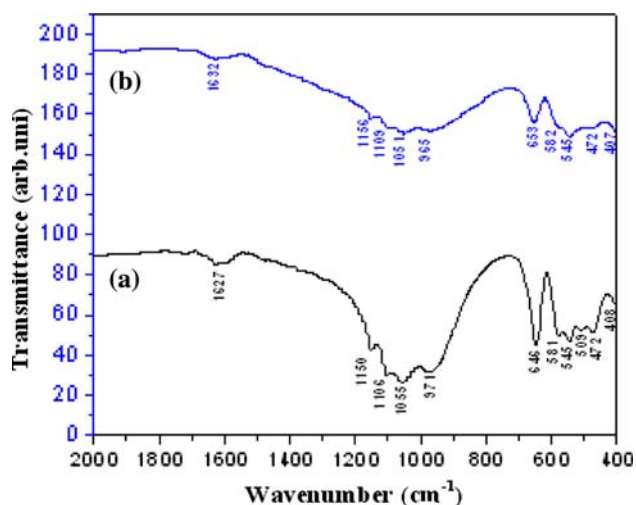
stretching mode at  $\nu_1$ –971 cm<sup>-1</sup>, a doublet at around  $\nu$ –472 cm<sup>-1</sup> and two triplets viz.,  $\nu_3$  and  $\nu_4$  in the region 1,055–1,106 cm<sup>-1</sup> and at 646 cm<sup>-1</sup>. Further, the observed peaks at 509 cm<sup>-1</sup> and 581 cm<sup>-1</sup> region are attributed to the asymmetric stretching modes of CoO<sub>6</sub> octahedra [16], thus confirming the presence of LiCoPO<sub>4</sub>.

In the case of LiNiPO<sub>4</sub> (Fig. 5b), the fundamental vibrations include an asymmetric stretching mode at  $\nu$ –965 cm<sup>-1</sup>, a doublet around  $\nu$ –472 cm<sup>-1</sup> and two triplets  $\nu_3$  and  $\nu_4$  in the regions 1,051–1,109 cm<sup>-1</sup> and 582–653 cm<sup>-1</sup>. Besides the fundamental vibrations of PO<sub>4</sub><sup>3-</sup> polyanions, the peak observed around 545 cm<sup>-1</sup> may be ascribed to the asymmetric stretching of Ni–O bonds in NiO<sub>6</sub> octahedra [17] and the other peak at 408 cm<sup>-1</sup> may be attributed to the bending modes of O–Ni–O bonds. Hence, the structure of LiNiPO<sub>4</sub> could be understood.

Basically  $\nu_1$  and  $\nu_3$  modes involve the symmetric and antisymmetric stretching vibration of the P–O bonds, where as  $\nu_2$  and  $\nu_4$  modes involve mainly O–P–O symmetric and antisymmetric bending modes with a small contribution of P vibration [18]. Also the FTIR spectra obtained in the present study are comparable with those of the results obtained in some of the earlier studies on phosphate based compounds [19], thus confirming the fact that all the PO<sub>4</sub><sup>3-</sup> tetrahedra possess stable chemical bonds

**Table 1** Physical parameters of LiCoPO<sub>4</sub> and LiNiPO<sub>4</sub> compounds

Compound	Lattice constants (Å°)			Cell volume (Å <sup>3</sup> )	Avg. crystallite size (nm)	Stoke strain
	a	b	c			
LiCoPO <sub>4</sub>	10.2	5.91	4.73	285.13	48.9	$2.5 \times 10^{-3}$
LiNiPO <sub>4</sub>	10.03	5.85	4.69	275.19	40.6	$3.1 \times 10^{-3}$

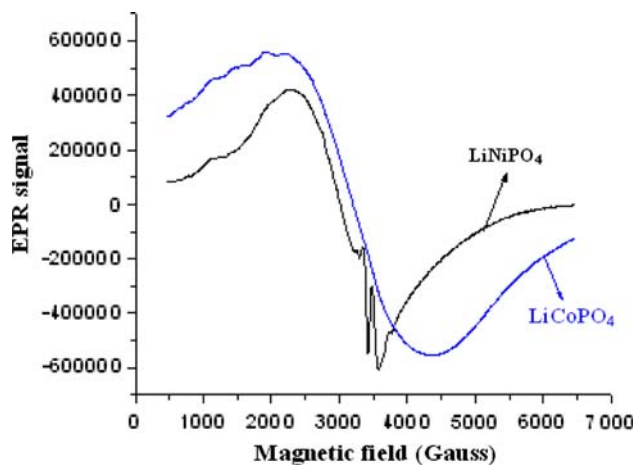


**Fig. 5** FTIR Spectra of (a) LiCoPO<sub>4</sub> and (b) LiNiPO<sub>4</sub>

and thereby producing LiCoPO<sub>4</sub> and LiNiPO<sub>4</sub> compounds with preferred local cation environment.

### 3.5 Room temperature EPR studies

In order to understand the possible existence of difference in electronic structure and the related magnetic properties of LiMPO<sub>4</sub> (M=Co, Ni) cathodes [20], EPR study was carried out for LiCoPO<sub>4</sub> and LiNiPO<sub>4</sub> individually and the results of the same are displayed in Fig. 6. Herein, the frequency of the microwave field is 9.9778 GHz and the frequency of the ac modulation magnetic field is 100 kHz. Under such conditions, the EPR signals are found to get centered at 3275G (g=2.16) and 3384G (g=2.06) for LiCoPO<sub>4</sub> and LiNiPO<sub>4</sub>, respectively. LiCoPO<sub>4</sub> shows a broad signal that may be attributed to the insufficient electronic spin density for the transition to take place at room temperature. But LiNiPO<sub>4</sub> shows a finer signal along



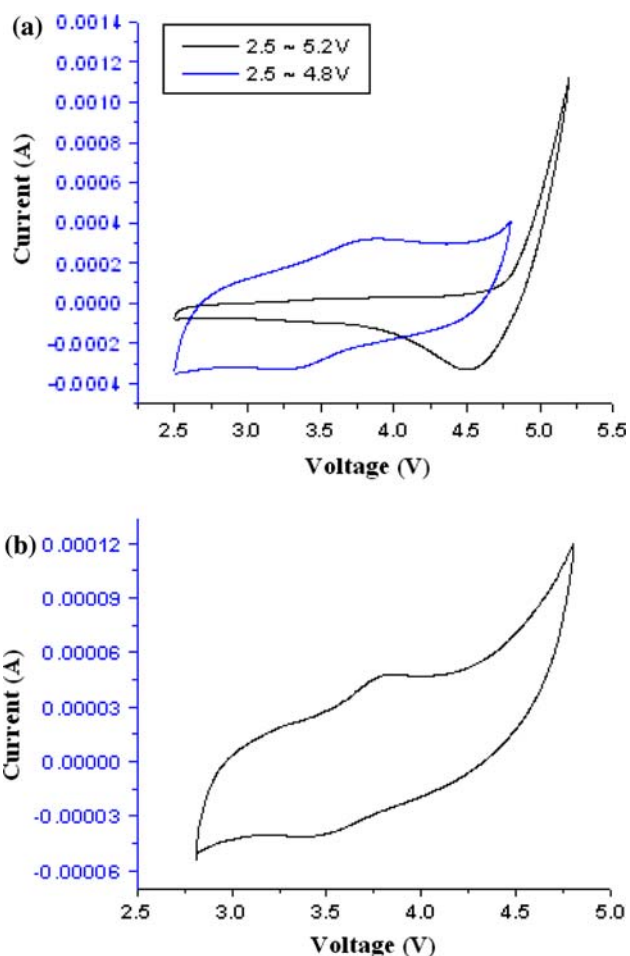
**Fig. 6** EPR spectrums of LiCoPO<sub>4</sub> and LiNiPO<sub>4</sub>

with a hyperfine splitting due to the interaction of unpaired electron with the neighboring nuclei. Thus, the presence of low-spin Ni<sup>2+</sup> and high-spin Co<sup>2+</sup> pertinent to LiNiPO<sub>4</sub> and LiCoPO<sub>4</sub> compounds could be understood.

### 3.6 Electrochemical properties

#### 3.6.1 Cyclic voltametry studies

Figures 7a and b show the representative cyclic voltammograms of LiCoPO<sub>4</sub> and LiNiPO<sub>4</sub> compounds, recorded at room temperature with the scan rate of 1 mV/sec vs Li/Li<sup>+</sup>. Figure 7a shows two representative cyclic voltammograms recorded in the potential range 2.5–4.8 V and 2.5–5.2 V individually. It is interesting to note that when LiCoPO<sub>4</sub> is cycled between 2.5 and 4.8 V, the presence of prominent anodic peak (de-intercalation) at 3.8 V and a corresponding cathodic peak (intercalation) at 3.3 V are seen. On the other hand, when LiCoPO<sub>4</sub> cathode is cycled in the extended potential range of 2.5–5.2 V, a predominant



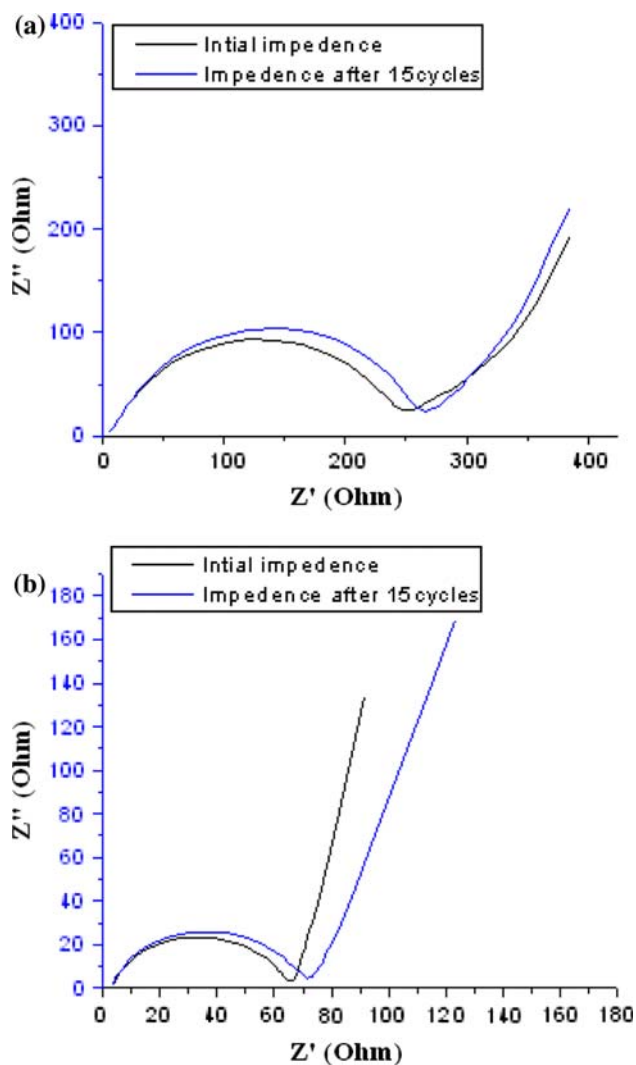
**Fig. 7** Representative cyclic voltammogram of (a) LiCoPO<sub>4</sub> and (b) LiNiPO<sub>4</sub> cathodes sintered at 800 °C (5 h)

current rise is observed around 4.8 V, that may be considered as the anodic peak along with a prominent cathodic peak at 4.5 V, which is in accordance with the reported results [21, 22].

Figure 7b shows the presence of an anodic peak at 3.82 V and a corresponding cathodic peak at 3.4 V due to  $\text{LiNiPO}_4$  cathode cycled between 2.8 and 4.8 V. Generally it is reported that  $\text{LiNiPO}_4$  prepared under air without the deployment of argon atmosphere and the addition of conducting carbon may not exhibit significant redox peaks, especially in the potential range 3.0–5.0 V [21, 23]. On the contrary,  $\text{LiNiPO}_4$  synthesized using CAM sol-gel method and sintered in air through the present study has produced redox peaks at 3.8 and 3.5 V (Fig. 6b), which is noteworthy. Hence, it is claimed from the present study that CAM sol-gel method, especially due to the perfect tuning and monitoring of heating sequences along with the usage of select category gelling and chelating agents has resulted in the well formed cyclic voltamogram with the presence of the redox couple of  $\text{Ni}^{3+}/\text{Ni}^{2+}$  at the respective positions. Therefore, it is demonstrated from the study that the synthesis of  $\text{LiNiPO}_4$  could be exempted from the requirement of stringent synthesis conditions that involves the deployment of argon atmosphere and carbon coating to the precursors mix provided suitable synthesis approach like CAM sol-gel method is followed. Because, it is believed that the CAM sol-gel method, by virtue of availability of excess of carbon would result in the in-situ addition of conducting carbon, thus exempting the need for stringent synthesis condition, which is the significance of select category CAM sol-gel method.

### 3.6.2 Impedance measurements

The EIS results of the as fabricated coin cells and cells after completing 25 CV cycles are displayed for  $\text{LiCoPO}_4$  (Fig. 8a) and  $\text{LiNiPO}_4$  (Fig. 8b). It is evident from Fig. 8a that the initial ( $R_1$ ) as well as final internal resistance value ( $R_{25}$ ) of the cell containing  $\text{LiCoPO}_4$  cathode is found to differ by an insignificant value of 12  $\Omega$  only, i.e., such a small increase in internal resistance is negligible compared to the magnitude of initial (256  $\Omega$ ) and final (268  $\Omega$ ) impedance values of  $\text{LiCoPO}_4$  cathode. Similarly, the difference in the internal resistance values of  $\text{LiNiPO}_4$  cathode in the as fabricated ( $R_1 = 65 \Omega$ ) and in the cell after 25 cycles ( $R_{25} = 73 \Omega$ ) is also meager ( $\Delta R = 8 \Omega$ ), thereby substantiating the fact that the CAM sol-gel synthesized  $\text{LiCoPO}_4$  and  $\text{LiNiPO}_4$  cathodes possess good storage as well as cycling stability, which is noteworthy. However, efforts to minimize the internal resistance of the cell upon cycling is underway. Further, it is understood from the study that the appreciable electrochemical behavior with a good structural stability of synthesized cathodes is



**Fig. 8** a EIS behavior of  $\text{LiCoPO}_4$  and b EIS behavior of  $\text{LiNiPO}_4$

resulting from the synthesis methodology driven size reduced particles of phase pure  $\text{LiCoPO}_4$  and  $\text{LiNiPO}_4$  cathodes, which is highlight of the CAM sol-gel methodology.

## 4 Conclusions

$\text{LiMPO}_4$  (M=Co, Ni) category phospho-olivine cathode materials have been synthesized by adopting CAM sol-gel method. The chosen method is found to be advantageous in producing nanocrystalline  $\text{LiCoPO}_4$  and  $\text{LiNiPO}_4$  products with preferred physical and electrochemical properties required for rechargeable lithium-ion batteries. XRD results confirm the formation of above said compounds with an orthorhombic crystal lattice structure. TEM and FTIR results are in favor of nanometric particle size and preferred local cation environment, respectively. Further,

the cycleability and structural stability of the compounds are found to be excellent, thus recommended that  $\text{LiMPO}_4$  ( $M=\text{Co, Ni}$ ) cathodes could be considered as next generation lithium insertion electrode materials. Particularly, the possibility of synthesizing  $\text{LiNiPO}_4$  at ambient synthesis conditions using CAM sol–gel method has been demonstrated through the study, thus excluding the reported category requirements of stringent synthesis conditions to prepare electrochemically active  $\text{LiNiPO}_4$  cathode.

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