### **REVIEW PAPER**



# **Advances towards development of industrially relevant short natural fiber reinforced and hybridized polypropylene composites for various industrial applications: a review**

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### **Abstract**

There has been an apparent signifcant increase in the application of composite materials in diverse industrial sectors lately with a focus on improving performance properties and processing techniques. Among several advances noticed, the most signifcant one for composite industry has been the reinforcement of environment friendly natural fber (NF) to the polymeric matrix for the fabrication of superior composites. This benign entry of NF in the composite industry has prominently decreased the carbon footprint and composite cost compared to the previously synthetic fber reinforced ones. Motivated by the decreased cost and latent positive environmental efects, various engineering sectors have already started using these composite materials extensively. The scientifc community has observed that hybridization of natural fbers with certain fllers helps to enhance the performance of these composites further. In line with this, the review aims to describe the developments related to the particulate hybridization with short industrially pertinent NF reinforced polypropylene (PP) composites. The efect of hybridization on the composite processability, interfacial interactions, mechanical properties, thermomechanical and thermal properties, and moisture absorption has been covered. The review also covers various challenges and prospects of NF/PP composites for application in the automotive, aerospace and other industries.

**Keywords** Short natural fber · Polypropylene composites · Sustainable development · Hybrid composites · Moisture mitigation

### **Abbreviations**



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# **Introduction**

Development of environment friendly and cost-efective composites materials has received signifcant interest globally as a result of the international rules and legislative pressure towards the safe guarding of natural assets [[1–](#page-15-0)[4\]](#page-15-1). This obligatory task has pushed the scientists engaged in diferent composite sectors to ideate and develop renewable or renewable inspired materials which can replace their traditional counterparts. For instance, in the composite industry, the reinforcement with the synthetic fbers is proposed to

be considerably replaced completely by natural fbers or in combination with them. Furthermore, carbon/glass fber use in some applications of automotive interiors has been replaced by plant-based fbers [[5,](#page-15-2) [6\]](#page-15-3). Plant based natural fber like sisal, hemp, kenaf, fax, ramie, jute, rice husk and many more have been reviewed by many researchers for their suitability in aircraft, automotive components, hat racks, furniture and other application [[7\]](#page-15-4). Fabrication and application of these natural fber reinforced composites have seen many transitions towards a sustainable approach since last few decades [\[3](#page-15-5), [4](#page-15-1)]. A comparable specifc mechanical property, low density, light weight, low cost, required aspect ratio, decent thermal, electrical and acoustic insulation properties, and most essentially minimal health hazard and energy consumption makes plant-based fbers an obvious choice instead of synthetic fibers  $[1, 2, 8, 9]$  $[1, 2, 8, 9]$  $[1, 2, 8, 9]$  $[1, 2, 8, 9]$  $[1, 2, 8, 9]$  $[1, 2, 8, 9]$  $[1, 2, 8, 9]$  $[1, 2, 8, 9]$ .

It is worth mentioning here that the evolution of new composite products in twenty-frst century would address the concerns for economy, ecology and equity (EEE) for a sustainable and eco-secure development [[10,](#page-15-9) [11\]](#page-15-10). Although synthetic fibers are light-weight and strong enough to accomplish specifc needs of the industry, but still they have failed when it comes to environmental accountability and economics. One such development has been the proposed use of fax fber instead of carbon, for the manufacturing of formula one racing car seats developed by Bcomp and McLaren as reported earlier by Maurya et al*.* [[12](#page-15-11)]. The authors reported that the use of fax instead of carbon fber ofers to reduce the carbon foot print drastically. Likewise, aluminum alloys and composite still make most of the aircraft's weight in modern aircraft. However, this industry has also adopted many polymer composites to fabricate

various parts. For instance, Boeing 787 and the Airbus A350 used carbon fber reinforced polymer composites for the structural components such as structural components such as the wing box, spars, stringers, and ribs [[13\]](#page-15-12).

Natural fbers possess very complex biological confguration and anatomy [\[14](#page-15-13)]. Figure [1](#page-1-0) demonstrates the structure of the natural fber and their various chemical components such as cellulose, lignin, and hemicellulose [\[15](#page-15-14)]. Cellulosic groups are fundamentally rigid and crystalline in nature, whereas lignin and hemicellulose exhibit amorphous regions encompassing cellulosic regions within them [\[16](#page-15-15)]. Conclusively, celluloses form the main components of natural fiber. A detailed chemical structure of the cellulose, hemicellulose and lignin has been depicted in Fig. [2](#page-2-0)a–c.

It can be observed that 2 of the -OH groups in cellulose can form intramolecular hydrogen bonding while others are available to interact with intermolecularly [\[14](#page-15-13)]. Furthermore, tensile strength of the natural fber depends on the fbril angle which is a measure of angle between fiber axis to the cellulose direction. Higher fbril angle leads to the lower tensile properties [\[17](#page-15-16)]. The presence of such groups makes natural fber hydrophilic in nature, while the base matrices employed are generally hydrophobic polymers. Hence, formulating performance composites the fller/fber-matrix compatibilization must be achieved between these two distinct phases [[18](#page-16-0)]. Based on the tensile strength and uses in various industrial applications, this paper is focused on industrially relevant short natural fber reinforced PP composites. Natural fbers such as sisal (SSL), hemp, kenaf (KNF), fax, ramie and jute have been selected as the industrial fbers for the current analysis. Table [1](#page-2-1) displays their various mechanical properties along with their chemical compositions.



#### <span id="page-1-0"></span>**Fig. 1** Anatomy of natural fiber



<span id="page-2-0"></span>**Fig. 2** Structure of **a** cellulose **b** hemicellulose **c** lignin

<span id="page-2-1"></span>**Table 1** Chemical composition and various mechanical properties of industrial relevant natural fbers [[19](#page-16-1)[–22\]](#page-16-2) (Sir will correct the table)



<span id="page-3-0"></span>



### **Polypropylene**

PP, a polyolefn and a commodity plastic, is used in several applications since its invention. It is cheaper and possesses an excellent processing quality up to 200 °C. Interestingly, PP has shown resistance against numerous chemical solvents, bases and acids [[23\]](#page-16-3). In the literature studies reported earlier, PP was found to have suitable application in automotive bumpers, wheel covers, chemical tanks, cable insulation, carpet fber and many more applications. Total fraction (32%) of the PP used in an automotive industry compared to other plastics is much higher [\[24\]](#page-16-4). Table [2](#page-3-0) gives most pertinent applications of various polymers and their composites employed in the automotive industry. Additionally Fig. [3](#page-3-1) shows the various components of a car made up from different plastics.

However, the use of above plastics was based on the specifc properties possessed by them during distinct applications. For instance, PP was used due to its better heat, hinge, and modulus properties compared to PE. Likewise, TPE provides better toughness and gloss properties, ABS gives better processability, high impact and mechanical strength, SMA offers excellent heat resistance compared to typical polystyrene, while PU possesses high impact strength under low temperature [\[36\]](#page-16-5). These plastics may or may not be replaced with newer materials. Nonetheless, PP was found

<span id="page-3-1"></span>

to be used in almost all the automotive components. Using natural fber reinforced PP composite instead of pristine PP might beneft the automakers from environmental and cost efectivity points of view. Hence, based on the above discussion the current study and review is limited to the selected industrial natural fber (Table [1\)](#page-2-1) reinforced polypropylene hybrid composites.

Recently, with the development of new technologies, polypropylene-based composites have become a potential replacement for the engineering plastics used in automotive components. Additionally, with the reinforcement of new rules and legislation imposed by the government, automotive companies are moving towards more sustainable options for engineering plastics. Although many companies have already developed bio composites for body panel doors, etc., of the automotive, drawbacks like moisture absorption, low mechanical properties, voids, and gaps in the composites are still signifcant issues. However, these issues were well handled by industry and academic experts to sort out the problem by combining a certain amount of particulate reinforcements with short natural fbers. The current review summarizes the recent development towards natural particulate hybridized natural fber polypropylene hybrid composites. To conclude, technologies developed to date, diferent scientifc literatures have been reviewed. Various mechanical, thermal, and thermomechanical properties of these hybrid composites have been studied and presented collectively to compare these propertiesfor the automotive/ aerospace applications. This review paper reports the role, development and effect of hybridization on the properties of short natural fber reinforced polypropylene composites. Furthermore, authors have also reported the various possible techniques of compatibilization among fber and base matrix, factors infuencing the mechanical, thermal and thermomechanical properties. However, the fabrication technique of the short natural fber reinforced polypropylene composites has been till now majorly limited to injection and compression molding.

### **Natural fiber polymer composite**

Natural fber polymer composites employ a specifc natural fber as reinforcement to a polymeric matrix. The mechanical properties of natural fber reinforced polymeric composites are majorly dependent on the extent of load transfer from the matrix to the reinforcing natural fiber  $[12, 41, 42]$  $[12, 41, 42]$  $[12, 41, 42]$  $[12, 41, 42]$  $[12, 41, 42]$  $[12, 41, 42]$ . This load transfer among the interfaces can be controlled by various factors. The chemical interactions between cellulose, hemicellulose and lignin of the natural fber with the polymeric matrix determines the mechanical strength of the composite. To increase these interactions various techniques such as fber treatment, use of compatibilizer, and adoption of combined fber treatment/compatibilizer have been developed in the past. For the fber treatment various physical, chemical and biological techniques have been developed for increasing the compatibilization between natural fber and polymer matrix [[6,](#page-15-3) [43–](#page-16-22)[45](#page-16-23)]. Table [3](#page-4-0) gives an illustration of such techniques employed based on their interaction with the fiber surface and reinforced polymer matrix versus the mechanical properties of composites obtained. The physical treatments such as plasma and corona make a very small chemical alteration at the natural fber surface, and hence, interaction at the interfaces is totally dependent on the mechanical interlocking. Likewise, chemical treatment mainly alters the chemistry at the fber surface to increase compatibilization through bonding/de-bonding. Chemical treatment may also form a new coating (silane) over the fber surface which demonstrates excellent compatibilization with base matrix and also resistance to moisture [\[46,](#page-16-24) [47\]](#page-16-25). For instance, alkalization of the natural fber to remove some part of lignin and hemicellulose. It is presumed that mechanical roughness created at the fber surface due to the lignin and impurity removal may enhance the mechanical interlocking with the polymer matrix. Likewise, it has also been observed that the formation of metallic-alkoxide may help in the same [\[48](#page-16-26)]. Furthermore, a biological treatment with specifc enzyme makes the fber surface more compatibilized to base polymer matrix and less vulnerable/detectable to the degrading fungi or microorganism searching for the nutrient medium. Additionally, an enzymatic pre-treatment of natural fber helps in decreasing moisture equilibrium and lowering of the decay rate [[49\]](#page-16-27).

Similar to fber treatment, compatibilizers also helps in compatibilizing the interactions between fber and matrix. The compatibilizer are bifunctional reagents which are utilized to improve interfacial interactions between natural fber and polymer matrix [\[50](#page-16-28)]. Compatibilizers try to lower the polarity of the natural fber by reducing the polarity of the natura fber thus resulting in phase similar to polymer [[50](#page-16-28)]. The typical compatibilizing agents that are used to increase interfacial interactions between fber and matrix

<span id="page-4-0"></span>**Table 3** Natural fber treatment and their composites with polypropylene

Natural fiber $(wt, \%)$	Treatment type	Coupling agent $(wt, \%)$	Tensile strength (MPa)	Flexural modulus (GPa)	Ref.
Sisal	Alkali-silane		$\sim$ 37	$-2.09$	[41]
Hemp	Alkali-silane	$PP-g-MA$	$-42$	٠	[58]
Hemp	Alkali	$PP-g-MA$	$-29$	~1.8	[62]
Kenaf	Alkali-silane		~1.57		[60]
Flax		$PP-g-MA$	$-27$		[61]
Ramie		$PP-g-MA$	67	2.7	[63]
Jute		$PP-g-MA$	~238	4.2	[64]

are m-phenylene bismaleimide modifed fbers, maleic anhydride grafted polypropylene (MAPP) and monofunctional isocyanates [[51,](#page-16-29) [52](#page-16-30)]. Likewise, Lu et al. [[51](#page-16-29)] also reported that dispersing agents can be classifed as compatibilizers since they reduce the interfacial energy at the interface by aggregation and agglomeration.

Coupling agents are another type of compatibilization where the formation of chemical bridges between fiber and matrix takes place. These bridges can be covalent bonding, hydrogen bonding or chain entanglement [[51](#page-16-29)]. Coupling agents form strong interfacial adhesion at the interface while compatibilizer do not possesses the same [\[50](#page-16-28)]. However, some compatibilizer such as compatibilizers based on maleic anhydride can be classifed under compatibilizer as well as coupling agent. These coupling agents form ester and hydrogen bonding at the fber and matrix surface interface. For instance, Maurya et al*.* [[41\]](#page-16-20), Gogoi et al*.* [[53\]](#page-16-31) and Kumar et al. [\[54](#page-16-32)]*.* used MA-SEBS, MA-PP and MA-EPR as a coupling agent for the fabrication of diferent natural fber reinforced polymer composites, separately. All the authors reported formation of ester (-COO) and hydrogen (H–H) bond at the fber and polymer matrix interface. However, use of compatibilizer or coupling agents have their own disadvantage. For instance, when hard and stif fbers are used with a strong coupling agent, they tend to decrease the notched impact strength of the composites. In this direction, Mihalic et al. [\[55](#page-17-6)] and Sain et al. [[56](#page-17-7)] reported that effective energy dissipation requires de-bonding or fber breakage during crack propagation in notched impact strength. Hence, a too-strong bond between fber and matrix may break the fber, resulting in lesser impact properties. However, fber pull out from the matrix during impact dissipates higher energy than fber breakage. Hence, if a coupling agent increases the tensile and fexural properties of the composites at the same time it also decreases the notched impact strength [\[55](#page-17-6), [56\]](#page-17-7). These were some potential disadvantages of using a coupling agent for preparing natural fber based polymeric composites.

However, all these processes have their own merits and demerits based on the effluent, water wastage, moisture absorption, lower thermal degradation stability and poor/ superior mechanical properties. For instance, the chemical treatment of the fbers is preferred over physical treatment since the latter one may damage the microstructure of the surface and subsequently the mechanical properties of the composites as well [\[57\]](#page-17-8). Table [3](#page-4-0) gives an illustration of chemical treatment process and coupling agents used along with their mechanical properties. The most optimized fiber to matrix ratio for tensile strength and fexural modulus of short natural fber reinforced polymer composites was reported to be 30:70 (NF:Matrix) by the literature [\[41,](#page-16-20) [58](#page-17-0)]. Additionally, tensile strength and fexural modulus are essential properties required for automotive applications, and hence, the composites in the Table [3](#page-4-0) report the use of 30–40 wt.% of natural fbers [[5](#page-15-2), [6](#page-15-3), [55,](#page-17-6) [59,](#page-17-9) [60](#page-17-2)]. It is also well proven and reported fact that alkali treated followed by silane coated short natural fber reinforced polymer composites demonstrated excellent mechanical properties [[41,](#page-16-20) [58](#page-17-0), [60](#page-17-2)]. Additionally, literature investigation have also recommended the use of 5 wt.% of any coupling agent to give optimum mechanical properties of composites [[61\]](#page-17-3).

#### **NFCs in automotive components**

Natural fber polymer composites are high in demand due to their desirable properties: good economics, anti-corrosion, environment friendly and light weight [[65\]](#page-17-10). Companies like Ford, Daimler Chrysler, Audi, VW, and Mercedes-Benz are already deploying the NFCs for various exterior and interior applications for designing their cars. Tables [4](#page-5-0) and [5](#page-6-0) lists details of the specifc fbers and polymers employed by them with the corresponding mechanical property estimates.

Researchers have also reported that by using these NFCs a reduction of  $\sim$  25% in weight can be achieved which could ultimately result in lesser fuel consumption [[69,](#page-17-11) [70\]](#page-17-12). It was also interpolated by scientists that a preferable use of NFCs over synthetic fiber composites may reduce the energy demand and greenhouse gas emissions by a factor of 9.6% and 18.6% respectively [\[71](#page-17-13)]. Table [5](#page-6-0) shows the various components fabricated from NFCs used in automotive, ballistic and construction industries.

Besides, possessing the aforementioned excellent properties NFCs have some serious concerns such as water/moisture absorption, poor fber and matrix adhesion, and defects

<span id="page-5-0"></span>**Table 4** Tensile strength and fexural modulus of various natural fber polymer composite used by various industries



 $\overline{a}$ 

<span id="page-6-0"></span>**Table 5** Application of natural fber polymer composite in various automotive, defense and construction industry



in the form of voids. Addition of natural fbers to respective polymers have also shown to decrease the impact and toughness properties of the composites. Recent developments in the area of such composite fabrication for the automotive sector have been well reviewed and reported recently [\[79](#page-17-17)].

#### **NFCs in aerospace industry**

Since, being light weight is the most essential priority for an aircraft or airbus after safety, hence, natural fber reinforced polymer composites have a wide scope in this sector. Also, high strength, impact resistance, tremendous vibration and thermal stability is also required. In this direction, ramie fber-based composite was studied to fabricate wing box of the aircraft [[80\]](#page-17-18). Theoretical studies show that ramie fberbased composite can replace 7000-series aluminum alloy and reduce weight by 12 to 14% [[80\]](#page-17-18). In another study, Kling et al*.* [[81\]](#page-17-19) tried to design wing box of the airbus using NFPCs however initial assessment shows that NFPCs were increasing the weight of the aircraft compared to carbon fber reinforced polymer composites and aluminum based wing box. Aviation industry is trying to implement environment friendly interiors of the aircraft. For instance, Boeing Research and Technology Europe (Spain), INvent GmbH (Germany), Aimplas (Spain), and Lineo (France) through the European project named "Cayley" aimed to use natural fber such as fax based polymer composites for the fabrication of interiors of the aircrafts [[82\]](#page-17-20). One possible development for this industry was reported by, De et al*.* [[83\]](#page-17-21) fabricated cotton fber reinforced polyurethane (PU) composites where PU was derived from the castor bean oil making composites fully naturally derived. Author stated that cotton fber-based PU composites were possessing enough tensile properties compared to glass fber reinforced polymer composites to be used in aircrafts interiors.

However, to some extent, using only natural fiber with polymer into the aircraft industry did not show desired properties. Hence, researchers have recommended the use of two or more fllers/synthetic fber with natural fber [\[13](#page-15-12)]. In this direction, most recent development was the use of nano fllers such as cellulose nano fber into the polymer composites. For instance, Bhatnagar and Sain [\[84](#page-17-22)] fabricated CNF reinforced composites for the possible application in the aerospace industry since composites were possessing high strength and light weight feature. In another study, Walsh et al*.* [[85\]](#page-17-23) used cork based sandwich structure with carbon fber face sheet. A comparative study on the bending stifness, acoustic, damping, and impact damage resistance of the cork sandwich composite with the structure made up of Rohacell<sup>®</sup> 110 IG, a synthetic foam commonly used as a core material in high-performance aerospace applications was done [\[85](#page-17-23)]. It was found that despite showing a little low bending stifness all the other properties of the composites performed better than the synthetic foam. Manan et al*.* [\[86](#page-17-24)] reported that sandwich structure with honeycomb core demonstrates high stifness at lower weight with high fexural rigidity. Author fabricated honeycomb core made up of unidirectional kenaf fber-based epoxy composite using a novel production process. Extending the research, Chok et al. [\[87](#page-17-25)] studies the low impact velocity damage to the radome application. Author prepared glass fber, kenaf fber and kenaf fber/glass fber hybrid composites and reported that kenaf based composites (including hybrid composite) were less prone towards the damage on the Radome structure of the aircraft. Authors reported that kenaf-based composite have lowest natural frequency reduction during impact, ranging between 1 and 18% from its original value [[87\]](#page-17-25). Likewise, Haris et al. [[88\]](#page-18-0) also investigated the same radome application by using kenaf, bamboo, pineapple, oil palm empty fruit bunch and banana composite, separately. Authors reported that kenaf was the most suitable natural fber for designing the radome application in the aircraft. Additionally, Ghori et al. [[89](#page-18-1)] reported that many aircraft parts can be designed using kenaf/glass fber composites, which show promising mechanical and rain erosion properties [[90\]](#page-18-2).

# **Drawbacks of NFCs**

#### **Moisture absorption**

Absorption of the moisture by the NFCs limits their application to nonstructural and interior automotive parts only. Hydrophilic nature of the natural fber induces it to absorb water at the poor interface of fber and matrix, leading to the development of cracks via crazing around these weak zones [\[72,](#page-17-26) [91\]](#page-18-3). Diffusion of water molecules into the composites may happen due to the micro or nano gaps/void in-between chains of the base matrix, capillary gaps among fber and matrix, and difusion through microcracks from the surface of the composites. Additionally, it has also been presumed that presence of moisture result in weakening of the interfacial region which ultimately turns into a the low stress transfer region [[92,](#page-18-4) [93](#page-18-5)]. Figure [4](#page-7-0) gives an illustration of the difusion and de-bonding process for the natural fber polymer composites.

Despite the aforementioned issues, various techniques like use of coupling agent, organic, inorganic or biological coating [[94\]](#page-18-6), and chemical or physical treatment of the fbers may reduce/inhibit the moisture absorption in composites. The researchers have also reported hybridization with synthetic fber/particulates to decrease the water uptake in the NFCs [[41](#page-16-20)]. A detailed discussion of the same is presented in the upcoming section.

Treatment of natural fber with alkali or silane makes them lesser susceptible towards moisture. For instance, treatment of natural fber with NaOH makes alkoxide on the fber surface ofering lesser number of available -OH groups for the moisture absorption. In this direction, Haameem et al*.* have reported that 10 wt.% alkali treated short napier grass fber absorbed ~ 46% lesser moisture compared to untreated one in the short napier grass fber reinforced polyester composite [\[92](#page-18-4)]. Furthermore, Munirah Abdullah and Ahmad used alkali and silane treated coconut fber to reinforce polyester matrix. In their study, authors reported that treatment of coconut fber with silane treatment is more efective compared to alkali treatment in moisture mitigation [\[95](#page-18-7)]. It was reported that silanes offer a stiffer interface which inhibits the plastic deformation. The chemistry of silane coating is dependent on the interaction between silanol (silane dissolved in alcohol-water mixture) and -OH group of the natural fiber [[41](#page-16-20), [96](#page-18-8), [97](#page-18-9)], where the, former gets chemisorbed on the fiber surface making a strong and stable covalent bond [[41,](#page-16-20) [96](#page-18-8)]. Another similar study, on moisture absorption reported by Wu et al. [[98\]](#page-18-10) involves use of polyethylene (PE) coated hemp fber for the reinforcement of the polyester matrix. PE coated hemp fber reinforced composites showed reduced tendency towards water absorption. It was also found that water absorption was along the edges of the composites [\[98](#page-18-10)]. It was reported by Mrad et al. [[99](#page-18-11)] that higher the fber loading, greater was the % moisture absorbed by the composites for a wood particle

<span id="page-7-0"></span>

reinforced high density PE matrix system. Panthapulakkal and Sain fabricated 40 wt.% hemp/glass fber reinforced PP hybrid composites and reported that hybridization of glass fber to the composite system reduced the water absorption behavior signifcantly [\[100\]](#page-18-12). Also, Akil et al. [\[101\]](#page-18-13) used three diferent mediums such as distilled water, sea water and acidic mediums for studying the moisture absorption behavior of the jute reinforced polyester composites. A key fnding from this research was that the presence of salt particles in the sea water reduced the water molecule absorption/ difusion into the composite samples. Some researchers have also reported that addition of particulate reinforcement/fller may reduce the water absorption by the NFCs. In this direction, Alamri and Low [\[102](#page-18-14)] use nano silicon carbide (SiC) for reinforcing recycled paper reinforced epoxy composites. The presence of SiC particles act as a barrier against the water molecule diffusion. Also, with the increasing wt.% percentage of the SiC particle to the composite, a decrease in the diffusion coefficient was reported by the authors  $[102]$  $[102]$ . Likewise, Maurya et al. reported lowest water absorption behavior for the 5 wt.% filled nano clay to the sisal fiber reinforced epoxy composites. However, the recent addition in this direction was the incorporation of fly ash to the sisal fber reinforced PP composites [\[41](#page-16-20)]. Authors fabricated alkali and silane treated sisal fber and fy ash (5–15 wt.%) flled PP composites. A key fnding from this research was that both alkali-silane treatment of fbers and addition of fy ash particles offered resistance against the diffusion of the water molecule to the polymeric composites without much compromising the mechanical properties [[41\]](#page-16-20). Hence, the most efficient and effective method for decreasing moisture uptake in NFCs is the use of alkali-silane treated natural fber with particulate reinforcements.

#### **Presence of voids**

Another major issue which occurs in the NFCs is the presence of voids. These voids might be exhibited as a result of unflled regions due to the entrapment of air or various volatiles within the composites during the processing [\[103,](#page-18-15) [104\]](#page-18-16). Other factors such as mold complexity, geometry, resin properties like viscosity and location into the composites may also increase/decrease void content. As per investigation reported by Ismail and Calabrese in separate studies, these voids may turn into micro-voids due to combination of two or more small voids which may result in the sudden failure of the material during load applications, and subsequently, reduce the mechanical characteristics [\[105,](#page-18-17) [106\]](#page-18-18). Furthermore, even the presence of a small amount  $(-1\%)$  of voids may reduce the mechanical properties of composites by 20% [\[104](#page-18-16)]. Literature article in this area have also suggested that composites containing fbers with lower density possessed reduced amount of voids compared to one consist of high density fbers [[107](#page-18-19)].

#### **Decrease in impact properties**

Impact strength of the NFCs is a very essential property for some specifc applications to avoid brittle fracture against a sudden loading [\[108\]](#page-18-20). It is captured as the energy required to break a sample against an instant force [[108](#page-18-20)]. Impact energy required to break a notched sample demonstrates much lower value in comparison to the un-notched one. This behavior is exhibited due to the predominant stress concentration region in the notched specimen and no energy required for pre-existing cracks (notch). However, in contrary to this, un-notched samples needs extra energy for crack initialization and propagation [\[109](#page-18-21)]. It is worth mentioning here that coupling agents play an important role in deciding the crack propagation. During crack propagation, efective energy dissipation needs to either pull out or de-bond natural fber from the base matrix [[55,](#page-17-6) [109,](#page-18-21) [110](#page-18-22)]. Hence, preventing a much higher interaction of the reinforced fber with the matrix with a very strong coupling agent may break the fber and that would need lesser energy in comparison to fber pull out on the application of sudden load [\[109,](#page-18-21) [110](#page-18-22)]. Conclusively, we can report that both an excellent and very poor interaction at fber/matrix interface may lead to reduced impact properties of the composites. This might lead to a catastrophic brittle failure in the former case and an easy pull out in later scenario [[109,](#page-18-21) [111](#page-18-23)]. Hence, as a result, an optimum interaction between the matrix and natural fber is essential for enhanced tensile, fexural and impact properties. In this direction, Sain et al. [\[109\]](#page-18-21) reported a reduction in the notched Izod impact strength of 40 wt.% kraft pulp and hemp fber reinforced PP composite separately compared to pristine PP. Authors fnd that addition of too much coupling agent lead to improved interaction between fber and matrix and hence, as a result, decreased impact strength of the composites were found. Ruksakulpiwat et al*.* [\[112\]](#page-18-24) fabricated vetiver grass reinforced PP composites. Authors reported a reduced impact toughness after the incorporation of vetiver grass in to the composites. With the increasing grass content, a decreasing trend in the impact properties of the composites were evident [\[112\]](#page-18-24). The reason behind decreased impact properties were due to the incorporation of alkali-silane treated vertiver grass fiber, which is stifer and brittle in nature compared to polymer matrix.

# **Particulate (micro/nano) reinforced polymer composites**

The development of novel composites with improved mechanical and thermal properties is the need of the hour. For instance, incorporation of micro- and nano-structured reinforcements has been invented already. These reinforcements offer better interaction with the base matrix compared to their pristine (normal size) counterparts. A good interaction among particulate reinforcements and the base polymer/matrix in the interfacial region leads to high strength composites. A good amount of literature articles support that, an excellent chemical, physical or physicochemical interaction exhibits substantial improvement in the mechanical properties by transferring load from base matrix to reinforcement. It has been observed that such reinforcements when dispersed uniformly inhibit the motion of the dislocation and impeded deformation behavior. Hence, as a result, the fabricated composites possess excellent mechanical performance compared to the base matrix.

Researchers have invented the micro and nano structured reinforcements of various shape and sizes. Furthermore, while micro-structured reinforcement provides better surface area for interactions, the nano structured reinforcements additionally offer better interactions at the interfacial region due to their ultra-fne form. Hence, as a result, the chain movements can be restricted efficiently. A larger specifc surface area of the reinforcement gives them the ability to interact with the base matrix in totality and hence, afects the mechanical properties, crystallinity, melting point, thermal degradation, mobility, charge distribution electrical and thermal conductivity, etc. Some nano materials such as multiwalled nanotubes, silica nano particles, and nano clay have affinity to interact with the base matrix quickly and even when used in very small amounts the mechanical properties of the polymeric composite can be enhanced to a great extent.

<span id="page-9-0"></span>**Table 6** Types of nano reinforcements in a polymer composite

Nano materials can be effective/non-effective depending on their shape and size. Table [6](#page-9-0) gives an insight of the diferent shape and sizes such as sheet, spherical or fbrous (rodlike) of diferent nanomaterials. Depending on the interaction capability, rod shaped ones are best suited followed by spherical and sheet-based nano materials for the reinforcement into a polymeric composite. Zhang et al*.* explained that, the shape of the nano materials afects the mobility of the chain and lamellas of the polymer in a polymer and reinforcement network. As a result, they effect both the diffusion coefficient of the polymers and the internal stress. Furthermore, larger shape reinforcements increase the void content and decrease the stress contribution, bead numbers (how many chains in a polymer) and mobility.

### **Drawbacks of particulate composites**

The above discussion enables to conclude that the incorporation of a small amount of nano reinforcement improves the desired properties without afecting the density and toughness of the composites. However, at higher loading content, an increase in defects and voids is obvious resulting in decreased mechanical properties. Another problem associated to the reinforcement of nano materials are the aggregation and agglomeration  $[113]$  $[113]$ . These two effects decrease the efective surface area available for their interaction with the polymer matrix. Physical aggregations are non-coherent and might be in any state and follow a defnite pattern, while agglomerations are rigid in nature due to the chemical interactions and strong van der Walls forces among the particles [\[114–](#page-18-26)[116\]](#page-18-27). However, in both the cases formation of clusters of nano particles takes place inside the matrix resulting in



weakening of the composites. Hence, the dispersion of the micro/nano reinforcements is the main concern during the fabrication of such composites [\[117](#page-18-28)[–119\]](#page-18-29).

# **Particulate reinforced natural fber PP hybrid composite (PNFC)**

Numerous articles have been reported on the fabrication of natural fbers and particulate fller reinforced hybrid polymer composites to tailor the mechanical properties of single fller composites. It has been believed that adoption of these hybrid composites will lead to lesser water absorption, higher strength and modulus compared to only natural fber reinforced polypropylene composites. The recent decades have seen increasing use of fllers such as  $\text{Al}_2\text{O}_3$ , SiO<sub>2</sub>, ZnO, TiO<sub>2</sub>, ZrO<sub>2</sub>, CaCO<sub>3</sub>, SiC, hollow glass microsphere, fy ash and other organic/inorganic particles with natural fber to formulate and reinforce polymer composites [[120](#page-18-30)]. These fllers were incorporated in the NFCs to overcome the aforementioned limitations. These fllers improved various mechanical properties of the composites by removing gaps and providing excellent packing of the composite materials. However, diferent properties of the composites were also afected by the dispersion of the fller, interfacial interaction between fller and base matrix, and distribution of the fllers [[120](#page-18-30)]. In the current survey, sisal fber, hemp fber, fax fber and kenaf fber along with incorporation of particulate reinforcement to the polypropylene hybrid composites have been reported. Table [7](#page-10-0) summarize the particulate hybridized natural fber reinforced polypropylene composites along with their mechanical properties.

# **Process description**

For the development of natural fber reinforced polymer composites, a number of processes have been developed. For instance, extrusion, injection, thermoforming, compression molding, cold pressing, heating, direct long fber reinforced thermoplastic, flament binding, foam molding, rotational molding, calendaring, co-extrusion are the famous techniques employed for the fabrication of natural fber reinforced polymer composites [[135](#page-19-0)]. However, only extrusion, injection and compression molding are in demand and extensively used for fabricating short fber reinforced thermoplastic polymer composites. Figure [5](#page-11-0) gives a brief description about the Extrusion/injection molding and while Fig. [6](#page-11-1) illustrate the schematic of compression molding.

## **Injection molding**

Injection molding is used for the mass production of the materials. Initially, the granules of the composite materials/ polymeric materials are fed in to the barrel of the machine,

<span id="page-10-0"></span>**Table 7** Tensile and fexural strength of natural fber reinforced polypropylene hybrid composite flled with particulate fller



<span id="page-11-0"></span>**Fig. 5** Illustration of extrusion/injection molding of natural fber reinforced polymer composites hybridized with particulate fllers



where it gets melted and get uniformly due to mixed heating and appropriate screw design. These molten materials are then injected through the nozzle of the sprue to the mold cavity followed by cooling and hardening at room temperature to get the desired shaped product. Barrel of the injection molding machine normally contains 6 zones: feeding zone (for submitting the material into the extruder), melting/compression zone (for melting and partial mixing of the material), venting zone (for discharging moisture and volatile), mixing zone (for providing proper mixing using screw), vacuum zone (to complete removal of the moisture and volatiles at required level) and metering zone (for building the required pressure to send the material into the die) [\[136\]](#page-19-13). Both the thermosets and thermoplastic polymers can be molded in the injection molding machine. Injection molding machines are available in both horizontal and vertical confgurations. However, irrespective of confgurations, operating an injection molding machine is very smooth. The primary function of an injection molding machine is to melt the material and inject the extruded melt in to a desired shape replica. Figure [5](#page-11-0) shows the manufacturing process of a natural fber reinforced polymer composites hybridized with particulate fllers particles.

### **Compression molding**

Compression molding has been used frequently in the past for the production of lighter weight component manufacturing. It is an old technique to place a mold between the two plates of hydraulic press. To fabricate the composite, components resins/composite granules can be charged in between the cavities of the preheated mold, where once the desired temperature is reached, pressure can be applied on the mold cavity to get the required shape of the composite components. The molds of the machine are kept closed until the desired curing has been achieved among the resin matrix specially for thermoset materials. The polymeric materials in the mold cavity change to molten state by the application of both heat and pressure. However, for the thermoplastic composites, temperature is needed to be cooled down before the ejection of the composite components. Besides, this cooling step is not mandatory for the thermoset plastic composites since they need to develop strong crosslinking under pressure during solidifcation. Both heating/ cooling and pressure of the compression molding machines decides the properties of the composites such as density, dimension and thickness [\[137](#page-19-14)]. Figure [6](#page-11-1) shows an insight into the processing of resins through compression molding.



<span id="page-11-1"></span>**Fig. 6** Illustration of compression molding of natural fber reinforced polymer composites hybridized with particulate fllers

#### **Mechanical properties of PNFCs**

Mechanical properties such as tensile, fexural and impact strength of the PNFCs have been evaluated by many researchers. Diferent kind of micro- and nano-structured fillers have been used with natural fiber. Impact of hybridization, interfacial adhesion and other property deciding factors have been discussed. In this direction, literatures shows that the addition of 3 wt.% of cellulose nano crystal to the 27 wt.% sisal fber reinforced PP composites enhanced its tensile and fexural properties drastically compared to pristine PP [[121\]](#page-18-31). The authors reported that the apparent increase in mechanical properties of the hybrid composites was due to the synergistic effect between cellulose nano crystal and sisal fbers. In another major development, Maurya et al*.* reported both signifcant and insignifcant increment in the mechanical properties compared to pristine PP through fabrication of micro- and nano-structured fy ash flled sisal fber reinforced PP hybrid composites [\[41](#page-16-20), [42](#page-16-21)]. The increment in the mechanical properties was exhibited due to the decrease in voids and interfacial gap between matrix and sisal fber as a result of incorporation of micro/nano fly ash [\[41](#page-16-20), [42\]](#page-16-21). The tensile and fexural modulus of all the composites increased drastically with the increment of extent of such reinforcements. This might be attributable due to the rigid nature of both hybridizing fller and reinforcing fber compared to pristine PP. Some other examples of natural fber reinforced and particulate fller flled polypropylene composites and their properties have been reported in Table [7](#page-10-0) [\[138\]](#page-19-15). When it comes to the impact strength of the composites, it has been reported that lesser the presence of voids and defects in the composite more is the impact strength of the composites. This is in good agreement with the data reported in our earlier studies  $[41]$  $[41]$ , wherein with the increment of fly ash to the sisal fber reinforced polypropylene hybrid composite, a signifcant increment in the notched Izod impact strength of the composites was evident. It was believed that the fy ash particles must have provided resistance against the propagation of the stress due to the impact load [\[41](#page-16-20)].

One important factor that many researchers miss or fail to address in their investigations is the decrease in the impact strength of the polymer composites due to natural fber reinforcement. This decrease in the impact properties of the composites was apparent due to the incorporation of relatively brittle natural fber to the polymers. During path propagation of the impact stress, these fbers fall in the path and are forced to be pulled out or broken as a result of excessive load [\[55](#page-17-6), [109](#page-18-21)]. Nonetheless, if there is a very strong interfacial adhesion with matrix then the fbers would be broken and if there exists a poor or negligible interfacial adhesion then the fbers may be pulled out. Hence, as a result, it would be right to say that an optimal interfacial interaction is needed for a desirable impact property.

Furthermore, incorporation of toughening agents like SEBS, SEBS-g-MA, SBS, EPR, EPDM, and POE (10–15 wt.%) is reported by many authors to compensate the effect of reduction in impact strength [[41](#page-16-20), [42,](#page-16-21) [139](#page-19-16)]. Also, in notched impact strength, stress propagation majorly depends on the development of crazes and shear bands. In the impact modifed polymers, the rubbery phase particles are entrapped into the polymer matrix. While the rubber particles may act as the center of stress concentration persuading crazes and shear bands, but, at the same time they may control initiation of new cracks and termination of old crazes before becoming destructive cracks. Additionally, formation and development of new shear zone needs a large amount of energy. The aforementioned reasons result is an enhanced impact strength of rubber toughened polymer matrix [[140](#page-19-17)].

### **Thermal properties of the PNFC**

The thermal characteristics of a natural fber reinforced polymer hybrid composites like, glass transition temperature  $(T<sub>e</sub>)$ , melting point  $(T<sub>m</sub>)$ , crystallization temperature  $(T<sub>c</sub>)$ , percentage crystallinity  $(X_c)$ , thermal degradation temperature and viscoelastic behavior depend on the characteristics of natural fber and fllers used to hybridize it [[141](#page-19-18), [142](#page-19-19)]. However, interactions among various functional groups of the reinforcements and base matrix at the interface synergistically change the thermal properties of the hybrid composites. The ensuing thermal properties of the composites are observed to be diferent from their original constituents used for the fabrication of composites. Upcoming sections describes how the thermal behavior of the natural fiber reinforced polypropylene hybrid composites may be evaluated and efectively described in terms of thermogravimetric analysis (TGA), diferential scanning calorimetry (DSC), and dynamic mechanical thermal analysis (DMA).

#### **Thermogravimetric analysis (TGA)**

TGA can help in demonstrating the thermal stability, percentage weight loss at any temperature and residual material left after complete degradation of the composites [\[143](#page-19-20)]. Degradation of natural fiber reinforced polypropylene composites occurs in three steps. The initial step includes the removal of moisture and volatiles which occurs up to 100 °C. The second stage encompasses the degradation of natural fber (lignin, cellulose and hemicellulose). The third and fnal stage consists of depolymerization of the polymers [[143](#page-19-20)]. Generally, the decomposition temperature at every stage of the NFCs lies between the decomposition temperatures of the natural fber and polymer. However, incorporation of specifc particles reinforcements may enhance the thermal stability of the composites compared to only natural fber reinforced polymer composites. Furthermore,

this stability may depend on the quality of interfacial region between the reinforcements and polymer matrix [[132\]](#page-19-10). In this regard, Boujmal et al.  $[132]$  fabricated alfa fiber/clay reinforced polypropylene hybrid composites. The authors reported that a composite consisting of 15 wt.% each of alfa fber and clay in PP matrix reported an increment of 21 °C in the degradation temperature at 15% wt. loss compared to only alfa fber reinforced PP composites. This enhancement in the thermal stability of the composites exhibited due to the incorporation of inorganic clay particles which are thermally more stable compared to alfa fber and PP matrix [\[132](#page-19-10)]. In another major development, kenaf fiber with graphene nanoplatelets (GNP) were reinforced into PP to fabricate a hybrid composite. TGA reports from the article demonstrate an improvement in the thermal degradation of the composites. The authors suggested that the high aspect ratio of the GNP offers barrier protection and inhibits gaseous molecule emission during thermal degradation [[130](#page-19-8)]. It was also reported by Idumah and Hassan that uniform distribution of the GNP slows down the oxygen supply by depositing a layer over the surface of the sample, and hence, the, thermal stability of the composites increases [\[129](#page-19-7)]. Other investigation of hybrid composites of PP reinforced by kenaf fber/magnesium hydroxide [[131\]](#page-19-9), pine cone fber/ clay [\[124](#page-19-2)], bagasse fber/calcium carbonate [\[125](#page-19-3)], sisal/cellulose nano crystal [\[121\]](#page-18-31), sisal fber/halloysite nanotubes [[122\]](#page-18-32) and coir fiber/coir shell [\[123\]](#page-19-1) separately, have also reported an enhanced thermal degradation temperature due to the hybridization with particulate fllers.

#### **Differential scanning calorimetry (DSC)**

Important physical properties such as  $T_g$ ,  $T_c$ ,  $T_m$ ,  $\Delta H$ , amd  $X_c$  can be evaluated from the thermogram of heat flow versus temperature curve. However, the  $T<sub>g</sub>$  values from DSC are not discussed here but elaborated *in* the next section from the DMA curves. While the pristine PP shows  $T_c$  at around ~ 107–112 °C [[41,](#page-16-20) [53](#page-16-31)], the addition of reinforcements and fllers provides new site for nucleation and growth of the polymeric crystals. These types of nucleation are termed as heterogeneous nucleation and come naturally in existence due to the presence of foreign materials in the polymeric matrix [\[42](#page-16-21), [139](#page-19-16), [144\]](#page-19-21). The heterogeneous nucleation resulting into the formation of spherulites and crystals in composites starts at relatively higher temperatures compared to homogeneous nucleation in pristine polymer/polypropylene during sample cooling. The presence of natural fbers and particulate fillers both affect the  $T_c$  of the composites and shifts it towards higher temperature in comparison to the pristine PP. This is in good agreement with the observation by Agarwal et al. where authors have reported a  $T_c$  of 129 °C for sisal fber/cellulose nano crystal reinforced PP  $(T_c = 122 \text{ °C})$  hybrid composites. Other articles based on PP hybrid composites which reported similar increase in  $T_c$  due to the presence of fber/particulate reinforcements are sisal fber/halloysite nanotubes [[122](#page-18-32)], bamboo fber/hollow glass microspheres [[53](#page-16-31)] and sisal fiber/fly ash [\[42](#page-16-21), [139\]](#page-19-16). Likewise, generally the  $T_m$  of the pristine PP lies between 160 to 166 °C [[42,](#page-16-21) [53](#page-16-31), [139](#page-19-16)]. However, an incorporation of the reinforcements and fllers to the PP matrix makes an insignifcant change to the  $T_m$ . For instance, an insignificant change in  $T_m$  of the sisal/ cellulose nano crystal reinforced PP composites was also reported earlier [[121\]](#page-18-31). Authors reported a  $T_m$  of 166.8 and 166.7 °C for both PP and sisal fiber/cellulose nano crystal reinforced PP composites. Melting heat of crystallization (∆*H*) was obtained from the area under the melting peaks of the composites. Using this information, the percentage crystallinity of the pristine PP and composites was calculated using the Eq. [1.](#page-13-0)

<span id="page-13-0"></span>
$$
X_c = \frac{\Delta H}{\Delta H_o} \tag{1}
$$

In the above equation,  $\Delta H$ <sub>o</sub> stands for the melting heat required for the 100% crystalline PP, which is constant and taken as 207.1 J/g [[144](#page-19-21)].

∆*H* and *X<sub>c</sub>* of 90.58 J/g and 43.33% were recorded using same methodology of the pristine PP and reported by Gogoi et al. [[53\]](#page-16-31) and Sui et al. [\[144\]](#page-19-21) earlier. However, with the incorporation of the reinforcements these values dropped drastically. The authors reported that with the increment of the reinforcements, obstacles were created during the ordered alignment of the polymeric chain amid crystallization [[144](#page-19-21)]. The same thermal behavior of decrement in the crystallinity was reported by Maurya et al. [\[42,](#page-16-21) [139\]](#page-19-16) and Agarwal et al. [\[121\]](#page-18-31), where both the authors fabricated sisal fber reinforced polypropylene composites using fy ash and cellulose nano crystal as hybridizing particulate material. Hence, conclusively we can say with the inclusion of the reinforcements/fllers in to the polymeric matrix, the degree of crystallization might be reduced as a result of restriction ofered by them towards an ordered alignment of polymer matrix.

#### **Dynamic mechanical thermal analysis (DMA)**

For the fber reinforced and particulate hybridized polymer composites, the DMA characteristics depend on the physical and chemical properties of the polymeric matrix and reinforcing components. Nevertheless, the fber-matrix and particulate-matrix interfacial interactions also play a pivotal role in deciding the DMA properties. Any smaller/larger change in the reinforcements quality and type might change the dynamic mechanical property of the composites. Such properties are generally evaluated at a heating rate of  $1-5$  °C/ min, where the frequency can be varied as per application.

Most of the polymeric composites use a three-point bending/ cantilever type of fexural samples for assessing the same.

**Storage modulus (E**′**)** The storage modulus of a material characterizes the ability of the material to store energy elastically. With the incorporation of reinforcements to the polymer matrix, *E′* of the composites records higher values compared to the pristine polymer. This statement is in accordance with the data reported by Maurya et al*.* [\[41,](#page-16-20) [142,](#page-19-19) [145](#page-19-22)] for fly ash hybridized sisal fiber reinforced polypropylene composites. Authors, stated that increment in the *E′* exhibited due to the excellent interfacial interactions among various fllers at the interfaces. Authors also depicted that insertion of nano/micro fy ash particles helped in decreasing interfacial gap between sisal fber and base matrix. However, with the increase in temperature *E′* of all the samples decreased due to the softening of the base matrix. Upadhyaya et al*.* reported the increased *E′* of hybrid composites compared to pristine PP due to the hybridization of wood four to the wheat husk reinforced PP composites [\[126\]](#page-19-4). Likewise, Agarwal et al*.* [\[121](#page-18-31)] reported *E′* of 3.9 and 4.7 GPa at -25 °C for PP and sisal/cellulose nano crystal reinforced PP hybrid composites. The increase in *E′* values of the composites was imparted due to the synergistic reinforcing nature of both the sisal fber and cellulose nano crystals. In another major development, Krishnaiah et al*.* [[122](#page-18-32)] reported an increment of  $\sim$  57% in the *E'* values of the sisal fber/halloysite reinforced PP hybrid composites compared to pristine PP. Moreover, some authors have also reported that storage moduli exhibits an underlying correlation with the tensile moduli of the composites [[122](#page-18-32)]. Likewise, Essabir et al. [[123](#page-19-1)] reported that reinforcement of the soft polymeric phase with rigid fllers improves its *E′* values. Authors stated that inclusion of 20 wt.% of coir fber/ particles restricted the polymeric chain movement resulting in a more rigid composite material. Hence, incorporation of comparatively rigid reinforcements/fllers results in the higher *E′* values of the composites. Additionally, building up an excellent interfacial interaction between reinforcements and matrix would lead to improved *E′* values.

**Loss modulus (E**″**)** *E″* is the measure of the energy dissipated per sinusoidal cycle of the deformation. In case of the hybrid composites, energy may be additionally lost due to the friction between fber–fber, fber-fller, and fller-fller. Hence, with the increment of reinforcements, a chance of energy absorption/dissipation and superior mechanical properties increases as reported by Krishnaiah for a sisal fber/halloysite nanotube reinforced PP composite system [\[53](#page-16-31), [122](#page-18-32)]. Likewise, Maurya et al*.* reported *E″* values of sisal fber/fy ash reinforced PP hybrid composites, their recyclates and 5 wt.% sisal fber reinforced to subsequent recylates. In all the three cases, authors reported an increment in the loss modulus of the composites with the increasing wt.% of sisal fber and fy ash [\[145](#page-19-22)]. However, Agarwal et al*.* reported that with increasing wt.% of fiber/fillers to the polymer matrix a decrease in the viscous response was recorded for a sisal fber/cellulose nano crystal/PP composite system [[121\]](#page-18-31). The authors stated that there might have been an adsorption of PP matrix over the surface of fiber/filler and the same effect might be contributing to the decrease in *E″* due to the increment of reinforcements.

**Loss factor or tan δ** Loss factor or  $tan \delta$  is the ratio of viscous (*E″*) to elastic response (*E′*). It is already well reported by many researchers that PP demonstrates its  $\alpha$  and  $\beta$  peaks corresponding to rotational and lamellar flow of the crystalline phase contributing to  $T_g$  of the PP [\[41](#page-16-20), [58,](#page-17-0) [121\]](#page-18-31). Thus,  $T_{g}$  of the pristine PP have been reported by several authors as at around -12 [\[121](#page-18-31)], 12–13 [\[58](#page-17-0)], and 26–27 °C [\[41](#page-16-20)]. However, with the increment of reinforcements all the composites were shown to yield a positive shift in the  $T<sub>g</sub>$  of the composites. It was believed that the presence of reinforcement fber/particles provided restriction towards the mobility of the molecular chains. Hence,  $T_g$  which is the measure of the polymer's softening point, shifted towards higher temperatures [\[41,](#page-16-20) [58](#page-17-0), [121](#page-18-31)]. Another interesting outcome from the *tan*  $\delta$  was the peak height of the curve at  $T<sub>g</sub>$ . Krishnaiah et al. [[122](#page-18-32)] fabricated sisal fber/halloysite nanotube reinforced PP hybrid composites and reported that with the increasing reinforcement content and interfacial interactions a decrease in the peak height of the  $tan \delta$  curve was observed. The findings from the above research were further confrmed by the Maurya et al*.* [[41](#page-16-20)] and Agarwal et al*.* [[121\]](#page-18-31) for sisal fber reinforced PP hybrid composites.

# **Conclusion**

The current study intends to present extensive details on the industry relevant natural fber reinforced polypropylene composites. Although the natural fber reinforced polymer composites reported by many researchers have enough strength to be used for automotive components, however, some major drawbacks like moisture absorption, void content, manufacturing defects and low interfacial interactions are still the concern. To resolve these issues various treatment methods have been reported in the recent articles. A proper treatment of the natural fbers may provide solution to these problems and open a new window of opportunities. Likewise, with the advancement in the science, a system of two or more type of reinforcements to the polymeric matrix referred to as hybrid systems, has been developed by researchers worldwide. This may help, to take the advantage of a possible synergistic efect between fbers and particulate type of reinforcements collectively

added to the polymeric matrix. The impact of various type of possible shapes of particulate fllers on properties has been presented in the current study. The prime job of many of these fllers is to reduce moisture absorption and fll the micro/nano gaps present in the composites. In some cases, these fllers also help in transferring the load from matrix to the reinforcing fber. These kind of hybrid composites can be deployed in automotive, aircrafts, construction, packaging, military, biomedical, furniture industry, etc.

### **Future scope**

This review illustrates that, natural fber reinforced particulate hybrid composites could be a potential alternative to carbon, glass and aramid fber reinforced composites with benign properties of light weight and low manufacturing cost. Likewise, based on the automotive and aerospace industry these NFPCs can be tailored by replacing or adding new generation of eco-friendly nano/micro fllers. NFPCs with hybridization from these fllers could be answer to the current industry problem. For instance, the natural fber's moisture absorption and low thermal stability is the well-known drawback associated with the NFPCs. These issues can be addressed by fabricating fller hybridized natural fber polymer composites due to the lesser voids and gap. These composites might be able to sustain the low impact damage and rain erosion into the components of the aircrafts. Likewise, there are very smaller number of literatures available based on the simulation and modelling of the natural fber reinforced polymer composites. By introducing precise simulation and modeling we can predict the expected properties and damage control of the NFPCs. Hence, an effort to experiment new set of natural fber-based composites can be reduced.

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**Data Availability** The data and articles that support the review of this study are available from the corresponding author upon reasonable request.

### **Declarations**

**Conflict of interest** The authors declare that there is no competing interests and funding related to the current work.

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