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Preparation and analysis of biodegradable polydioxanone/chitosan film

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Abstract

In this study, we investigated the miscibility and degradability of biodegradable polymers. Polydioxanone (PDO) was blended with chitosan, and the resulting films were prepared via film casting. The blends were analyzed according to composition ratio and in vitro biodegradation. The intermolecular interaction in the blend was verifed through spectroscopic analysis. The addition of chitosan was found to lead to a signifcant enhancement in the mechanical properties of the blend. Also, the chitosan embedded flm showed diferent morphologies and thermal properties from the pure PDO flm. The current study might show a possibility of employing PDO based biodegradable flm for biomedical applications.

Keywords Polydioxanone (PDO) · Chitosan · Miscibility · Biodegradability

Introduction

GBR(guided bone regeneration) is a surgical procedure that helps bone formation by using a barrier membrane when there is a lack of bone around the implant assembly $[1,$ [2](#page-9-1)]. Also, It is important to prevent the faster formation of dental epithelium than bone [[3\]](#page-9-2). The barrier membrane is divided into two main types depending on its absorbency: Non-absorbable membrane such as e-PTFE (expandedpolytetrafuoroethylene) is widely used but requires the secondary removal surgical procedure. On the other hand, since absorbable membrane such as collagen does not need additional surgery, infection risk is relatively low [\[4](#page-10-0)]. However, the absorbable membrane has quite low mechanical properties to maintain dimensional stability [\[5–](#page-10-1)[7\]](#page-10-2). It is necessary to increase its strength to prevent tearing during surgical procedure.

Biodegradable polyesters such as polylactic acid (PLA), polycaprolactone (PCL), polyglycolic acid (PGA), polyhydroxy butyrate (PHB) and polydioxanone (PDO attract a lot of attention in the medical industry [[8,](#page-10-3) [9](#page-10-4)]. For instance, PDO has excellent biodegradability and biocompatibility. However, its commercial applications are limited due to high cost

 \boxtimes Young Seok Song ysong@dankook.ac.kr and low strength. Thus far, a few studies have been done on the characteristics of PDO, especially biodegradability. Sabino et al. introduced the hydrolysis process of PDO and showed biodegradable behavior in PBS solution [\[10](#page-10-5)]. Some attempts have been made to improve the physical properties of PDO and control the degradation behavior. Bai et al. incorporated PPDO with PDLLA by the flm casting and analyzed the hydrolysis behavior according to the blend composition [[11\]](#page-10-6). Also, Ebrahimpour et al. investigated the miscibility of biodegradable PBS and PDO by using a viscometer [[12](#page-10-7)]. The blending method enables one to control the biodegradation of polymer as well as to improve physical properties of components [\[13\]](#page-10-8). Chitosan is derived by deacetylation of chitin, which is easy to obtain from crabs, lobsters, and shrimps [[14,](#page-10-9) [15](#page-10-10)]. Since chitosan has biodegradable, biocompatible, nontoxic, and relatively good mechanical properties among other natural materials, it has been applied to biodegradable blends [\[16](#page-10-11), [17\]](#page-10-12). Since chitosan can form the intermolecular interaction with other polymers due to the amino and hydroxyl groups, they have been used as a blending material to optimize physicochemical properties of polymeric parts [\[18](#page-10-13)].

In the current study, biodegradable PDO/chitosan flms were manufactured using the flm casting, and their mechanical, thermal and chemical properties were evaluated. In addition, the chemical characteristics of the blend was analyzed using fourier transform infrared (FT-IR) and nuclear magnetic resonance (NMR) spectroscopy. The biodegradability test was conducted for 12 weeks.

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Scheme 1 Schematic illustration of preparation of the PDO/chitosan (CS) films

Experimental

Materials

PDO (DO-013) was purchased from Meta-Biomed Co. (Korea). It has a melt fow index of 2 at 170 ℃. Chitosan supplied by Aldrich has a viscosity of 20-300cP in the acetic acid solution and a degree of deacetylation > 75%. $1,1,1,3,3,3$ -Hexafluoro-2-propanol (HFIP, $> 99\%$) and acetic acid (>99.5%) were provided by TGI (Tokyo, Japan) and SAMCHUN pure chemical Co. (Korea), respectively. Phosphate-Buffered Saline (PBS, 7.38 pH) was purchased from BIONEER (Korea).

Film preparation

PDO and chitosan were blended via the film casting method (Scheme [1\)](#page-1-0). First, PDO was dissolved in HFIP of 1% w/v concentration at 60 \degree C for 2 h, and then chitosan was dissolved in acetic acid-HFIP (1:15 v/v) at room temperature. The two solutions were mixed with a magnetic stirrer. The mixed solution was poured into a glass mold, and the solvent was evaporated at 60 ℃ for 3 days under vacuum. The blend flms (PDO/chitosan) were prepared according to the following composition: 100/0, 90/10, 80/20, and 70/30 w/w.

Fig. 1 FTIR spectra of the PDO/chitosan flms: **a** 100/0, **b** 70/30, and **c** 0/100 ratios

Fig. 2 CP/MAS ¹³C NMR spectra of the chitosan, PDO/ chitosan (CS), and PDO flms

Film characterization

The prepared specimens were used to measure mechanical properties such as tensile strength, Young's modulus and elongation at break at an extension speed of 5 mm/min with a universal testing machine (UTM, Instron). Dynamic mechanical analysis (DMA) was carried out to characterize the viscoelastic properties such as storage modulus and loss modulus while applying a frequency sweep of 0.01-100 Hz. Prior to the test, a pre-tension process was applied. The measurements were conducted using a rheometer (MCR302 Anton-Paar) with a UXF kit. Thermogravimetric analysis (TGA) of the blend flms was carried out with a thermogravimetric analyzer (TGA N-1000, Scinco). The samples were heated from room temperature to 600 °C at a heating rate of 10 °C/min under a nitrogen flow of 100 mL/min. To confrm the intermolecular interaction in the blend, the flm was scanned between 400 *cm*[−]*^l* and 4000 *cm*[−]¹ using

Fig. 3 a Storage modulus (E') and **b** loss modulus (E") of the PDO/chitosan flms as a function of the angular frequency

an FT-IR spectrometer (Spectrum Two, PerkinElmer). Also, a 500 MHz solid-state ¹³C NMR spectra were obtained at room temperature by using a Bruker Avance II spectrometer at room temperature. The CP/MAS mode was employed for all samples using a 4 mm probe. A spinning speed of 5 kHz and a pulse repetition delay of 5 s were used.

In vitro degradation

For the in vitro degradation, The PDO/chitosan blend films were immersed in phosphate buffer solution (PBS). Degradation test was carried out in an incubator at 37 ℃ for periods 12 weeks. For the test, the specimens were washed with distilled water to remove the hydrolysis medium and dried in a vacuum oven at room temperature. The flms were placed in a desiccator before the measurement. To analyze hydrolysis, the change in pH was measured with a pH meter (Model 700, EUTECH). The residue weight was calculated according to the following formula:

Residue weight (
$$
\%
$$
) = $\frac{W_r}{W_0} \times 100\%$ (1)

where W_0 and W_r are the initial and residual weights of the specimens, respectively. The morphological characteristics of the samples were analyzed according to the degree of degradability by using scanning electron microscopy (SEM, S-4700 Hitachi). The thermal properties of the samples were probed using a diferential scanning calorimeter (DSC4000,

Fig. 4 a TGA result and **b** DTG result of the PDO/chitosan flms as a function of the temperature

Fig. 5 Stress–strain curves of the PDO/chitosan flms

Perkin Elmer). The samples were heated from an ambient temperature to 150 ℃ at a heating rate of 10 ℃/min. The exothermic and endothermic peaks were ascribed to the melting point (T_m) and the crystallization temperature (T_c) in the second heating step, respectively. The following equation was employed to obtain the crystallinity of the blend flms.

$$
Dc_{\text{blend}}(\%) = \frac{\Delta H_m}{\Delta H_m^0} \times 100\tag{2}
$$

$$
Dc_{pdo}(\%) = \frac{Dc_{blend}}{W_{pdo}}
$$
 (3)

$$
W_{pdo}(w/w) = \frac{PDO}{PDO + CS}
$$
\n⁽⁴⁾

where ΔH_m is the enthalpy of melting per gram of blend, W_{pdo} is the weight fraction of PDO in the blend, and ΔH_m^0 is the enthalpy of melting for 100% crystalline PDO [\[19](#page-10-14)].

Table 1 Mechanical properties of the PDO/chitosan films

Results and discussion

Figure [1](#page-1-1) shows the FTIR spectra of the prepared PDO/ chitosan flms with blend ratios of 100/0, 70/30, and 0/100. The characteristic peaks of chitosan were observed at 3428 cm^{-1} , which implies $-NH_2$ and $-OH$ groups stretching vibration. The peaks at 1656 *cm*[−]¹ and 1596 *cm*[−]¹ indicate the $CONH_2$ and NH_2 groups, respectively [[20\]](#page-10-15). For the PDO flm, the strong and sharp absorption bands at 2924 cm^{-1} and 1746 cm^{-1} are assigned to the C-H and C = O stretching vibration, respectively. The peaks at 1216 *cm*[−]¹ and 1135 *cm*[−]¹ are ascribed to the C–O–C stretching vibration. In addition, the small peak around 3472 *cm*[−]¹ was observed, indicating the existence of the end O–H from macromolecular chains [[12](#page-10-7)]. For the PDO/chitosan flm with a blend ratio of 70/30, similar results were obtained, which supports the existence of PDO and chitosan in the blend flm. On the other hand, frequency shifts show a specifc interaction between the characteristic groups of

the pure polymers [[21](#page-10-16)]. The peak of stretching vibration of carbonyl in PDO (1746 *cm*[−]¹) was shifted to lower wave number (1740 *cm*[−]1) after blending with chitosan due to the formation of intermolecular hydrogen bonds between PDO and chitosan [\[22\]](#page-10-17). Also, the spectrum of PDO/chitosan flm shows the shifts from 3428 to 3433 *cm*[−]¹ was observed for the stretching vibration of $-NH_2$ and $-OH$ groups.

In order to further investigate the intermolecular interactions of the blend, the CP/MAS ¹³C NMR spectra of the chitosan, PDO, PDO/chitosan flms were obtained as shown in Fig. [2.](#page-2-0) The peaks of the $C=O$ and $C-O$ units of

Fig. 8 DSC results of the PDO/chitosan blends with respect to the degradation time: **a** 100/0, **b** 90/10, **c** 80/20, and **d** 70/30 ratios

PDO appeared at 172 and 66.25 ppm, respectively. In the case of the pure chitosan, the residual acetyl group appeared at 172.36 ppm, while the residual methyl group was determined at 22.62 ppm [[23\]](#page-10-18). It is known that the amino units linked to the C2 atoms of chitosan can form intermolecular hydrogen bonds with the carbonyls units [[24](#page-10-19), [25\]](#page-10-20). This interaction afects the electron density around the carbons, which causes the changes in chemical shifts [\[26\]](#page-10-21). The C2 resonance peak of the PDO/chitosan at δ = 56.91 ppm shows downfield shifts. From the results from FTIR spectra and solid-state 13 C NMR spectra, the existence of the intermolecular hydrogen bonds between CS and PDO was confrmed.

DMA tests can help ones to understand the long-term stability of the dispersion and the behavior of the polymer. Figure [3](#page-3-0) shows the storage modulus (E') and loss modulus (E") of the PDO/chitosan flm as a function of angular frequency. All the specimens showed that E' was higher than E", which indicates a dominant elastic solidlike behavior $[27]$ $[27]$. Both the storage and loss modulus increased as a function of angular frequency. Additionally, the modulus values of the flm blended with chitosan had higher than the pure PDO. This is because of the relatively large elastic feature of chitosan [[28](#page-10-23)]. These results were similar to the tensile test results of the PDO/chitosan blend flm.

The thermal behavior of the PDO/chitosan blend flm is demonstrated in Fig. [4.](#page-3-1) The initial degradation temperature of PDO flm was 220 ℃, and the decomposition temperature was found around 290 ℃. There are two degradation steps in chitosan [\[29\]](#page-10-24). For the case with a weight ratio of 70:30, the weight loss frstly started at 200 ℃ and became around 75% at 300 ℃. The maximum weight reduction rate

occurred at 280 ℃. Secondly, the specimen showed 85% weight reduction at 360 ℃. In the case of the blend films, as the content of the chitosan increased, the degradation tem perature decreased. The total weight losses of 90/10, 80/20, and 70/30 at 600 ℃ were found to be 94.7%, 93.2%, and 90.7%, respectively.

Mechanical strength is important for medical applica tions. Currently, the usage of PDO flm is quite limited due to its low strength [[30](#page-10-25)]. Mechanical properties including Young's modulus, tensile strength and elongation at break were analyzed to evaluate the addition effects of chitosan to the PDO flm. Figure [5](#page-4-0) shows that the blended flm had dramatically increased strength compared to the pure PDO flm. Also, the tensile strength increased with increasing the chitosan content. However, the elongation at break decreased with an increase in the chitosan fraction. Since chitosan acts as a reinforcing agent, the higher content lead to the higher rigidity [[31\]](#page-10-26). The decrease in the elon gation at break is related to an increase in the stifness of the blend flms as a result of the addition of chitosan [[32](#page-10-27)].

As listed in Table [1](#page-4-1), the addition of chitosan plays a key role in increasing the strength of the PDO flm. In the case of the blend flm with a 30 wt% chitosan fraction, the tensile strength and Young's modulus showed the highest values of 32.75 MPa and 1108 MPa, respectively. From this, it is inferred that chitosan can serve as a reinforcing phase in the blend matrix.

The percentage of residue weight for the PDO/CS blend flms is demonstrated in Fig. [6](#page-5-0). According to the hydrolysis mechanism of aliphatic polyesters such as PDO, once water penetrates into polymer it breaks the ester bond and creates short and soluble fragments of molecular chains. As a result, this results in the residue weight and decreases pH [\[33](#page-10-28)]. For the pure PDO flm, the largest mass reduction between 6 and 8 weeks was found. This implies that the soluble frag ments were produced after 6 weeks by hydrolysis. When the chitosan was added to the PDO, the residue weight was decreased. This result indicates that the presence of chitosan expedites the hydrolysis (Fig. [7\)](#page-5-1). After 12 weeks of hydroly sis, the pure PDO and the blend flm with 30 wt% chitosan contents showed the reductions in the pH values from 7.38 to 6.13 and 5.88, respectively.

The DSC thermograms of the PDO/chitosan flms are presented in Fig. [8.](#page-6-0) The blend film showed a single melting temperature, which indicates a good miscibility of the PDO and chitosan. As the hydrolysis progressed, both the melting temperature (T_m) and enthalpy of melting (ΔH_m) increased. This result is similar to the result reported by Sabino et al. [[10\]](#page-10-5). This is because the amorphous region is first attacked due to the hydrolysis, thus resulting in the scission of unstable ester chains. This led to the reduction in the chain entanglements and the increment in the crystallinity

Fig. 9 SEM images of the PDO100/CS0 blend with respect to the degradation time: **a** 0 w, **b** 4 w, **c** 8 w, and **d** 12 w

in the remaining segments. However, when the hydrolysis progressed further, the crystalline regions were attacked as well. As a result, the crystallinity of PDO first increased and then decreased after 8 weeks. In the case of the pure PDO, the melting temperature (T_m) and enthalpy of melting (ΔH_m) were 110.3 °C and 66.1 °C, respectively.

As can be seen from Table [2](#page-7-0), the melting temperatures of the PDO/chitosan was afected by the relatively low melt viscosity of chitosan. Specifically, T_m and ΔH_m in the blends decreased as the chitosan content increased. This implied that the presence of chitosan led to a decrease in the crystallinity of PDO since the movement of the polymer segments was impeded by chitosan [\[34\]](#page-10-29).

Figure [9](#page-8-0) shows the SEM images of the pure PDO flm with respect to the degradation time. The sample surface was smooth at the beginning $(0, w)$, but small holes were observed on the surface during the hydrolysis. After the eighth week of hydrolysis (Fig. [9c](#page-8-0)), the cracks appeared on the surface, which allow water to penetrate the matrix. These cracks became more and deeper over time and eventually were connected. In other words, the chain loss happens as time elapses.

The SEM images shown in Fig. [10](#page-9-3)a confrmed that there was no phase separation in the blend, which implies a good miscibility of the blend. This result is in agreement with the DSC result. For the hydrolysis, the PDO flm with 30 wt% chitosan did not show defects on the surface until 4 weeks (Fig. [10\)](#page-9-3). The surface was not smooth, and deformation occurred in the 8th week sample (Fig. [10c](#page-9-3)). Also, some holes and empty spaces were formed due to the swelling of chitosan powders [[35](#page-10-30)].

Fig. 10 SEM images of the PDO70/CS30 blend with respect to the degradation time: **a** 0 w, **b** 4 w, **c** 8 w, and **d** 12 w

Conclusions

In this study, the biodegradable PDO/chitosan flms were prepared via the flm casting. The thermal and mechanical properties of the blends were evaluated through DMA, TGA, DSC, and UTM analyses. Also, the miscibility of the blend was investigated experimentally. The FTIR and NMR results showed that the intermolecular hydrogen bonds exist between chitosan and PDO. In addition, the biodegradability of the sample was tested for 12 weeks in the PBS solution. It was found that chitosan plays an important role in enhancing the mechanical properties of the blend flms. The chitosan embedded flm showed signifcantly diferent thermal properties and morphologies from the pure PDO flm. We conclude that physicochemical properties of PDO flm can be controlled by blending with chitosan for biomedical applications such as GBR.

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Declarations

Conflict of interest The authors declare no conficts of fnancial or non-fnancial competing interest.

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