



# Fabrication of antiseptic, conductive and robust polyvinyl alcohol/chitosan composite hydrogels

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Received: 3 November 2019 / Accepted: 10 August 2020 / Published online: 17 August 2020  
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## Abstract

Fabricating robust and multi-functional hydrogels is of great importance and challenge. In this work, chitosan (CS) and polyvinyl alcohol (PVA) were used to design antiseptic, conductive and robust hydrogels by a two-step method. Chemical structures of gels, the degree of crystallinity, the state of water in hydrogels, as well as their microstructures were characterized via a combination of FT-IR, XRD, DSC and SEM. Segment lengths of cross-linking points were calculated from elastic rubber theory. Their mechanical properties were evaluated on the electronic testing machine. It was shown that the tensile strength and elongation at break of single PVA hydrogel were only 200 kPa and 135%, respectively, due to the heterogenous structure with pore sizes between 1.5 ~ 8.2  $\mu\text{m}$ . By introducing CS into PVA matrix followed with soaking in a saturated NaCl solution, the network became homogeneous with a pore size of 0.5 ~ 1.1  $\mu\text{m}$ . Moreover, free water changed to bond water, and frictions between polymer chains increased because of hydrophobic associations and entanglements of CS segments. As a result, the tensile stress and strain increased to 3800 kPa and 270%, respectively. The gel also exhibited antiseptic property, electrical conductivity and swelling-resistant properties. The strength after reaching swell equilibrium was 3400 kPa, much higher than most gels at swollen states. This gel might find applications in bionic cartilage, sensors, food preservation and wearable devices.

**Keywords** PVA · Chitosan · Robust · Antiseptic · Conductive

## Introduction

As three-dimensional networks of polymer chains swelling in water, hydrogels were considered as one of the best candidates in drug delivery, tissue engineering, soft electronics and wound dressings, because of their structural similarities to

human tissues and organs [1–4]. Conventional polyacrylamide(PAM) gels are usually fragile arose from the heterogeneity cross-linking network and low density of polymer chains [5]. Therefore, developing tough and functional gels is urgently necessary while difficult. To date, dual-network, nano composited, topological cross-linking and micellar cross-linked hydrogels were successful strategies to improve the mechanics of traditional PAM gels [6–9]. However, biocompatibility and functionalities of PAM remain a concern. Therefore, strong, biocompatible and functional gels are highly expected.

Generally, gels based on alginate, cellulose, chitosan, polyvinyl alcohol (PVA) and other biocompatible materials are safe and biocompatible [10–17]. Among them, FDA-approved PVA received much attention [18, 19]. They were usually prepared by freezing-thawing (F-T) methods and radiation cross-linking [20, 21]. However, crystalline sizes by F-T circles increase irregularly, while the network structure is uneven for the later [20], giving rise to fragile PVA gels. The gel stress from three F-T methods was reported to be less than 500 kPa [21]. Compared with tissue, these weak gels also lack conductivity and other functions, which greatly retard their

**Electronic supplementary material** The online version of this article (<https://doi.org/10.1007/s10965-020-02247-6>) contains supplementary material, which is available to authorized users.

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applications [22]. Consequently, fabricating robust and functional PVA gels are urgent.

To enhance gels, physical interactions including hydrogen bondings, hydrophobic associations, chain entanglements and ionic bonds were adopted, which were essential to endow gels with ability to dissipate external loads efficiently [23–26]. For example, Yang and coworkers achieved a 56-fold increase in tensile stress via coordination bonds between chitosan and sodium citrate [25]. After forming stronger hydrogen bondings between PVA and tannic acid, Jin group prepared gels with tensile strength close to 10.0 MPa [26]. Despite these achievements, their functionalities are not sufficient [26]. Another great limitation of present tough gels is that they will absorb lots of water and become fragile when immersing in water [27]. Considering the majority content of the human body is water, hydrogels with anti-swelling ability are of promising applications in tissue engineering. However, such gels are still difficult to fabricate.

Herein, synergistic physical interactions, instead of chemical cross-linkers, were adopted to achieve this goal. As a natural polysaccharide prepared by deacetylating chitin, chitosan is non-cytotoxic, pH-sensitive and antiseptic, which has shown promise as an ideal candidate for drug delivery [28–32]. It contains amine, ether and hydroxyl groups, making it possible to form H-bonds with PVA. Furthermore, its solubility was dependant on ionic strength and pH values [28], providing potential methods to induce micro-phase separation. To this end, commercially available PVA and CS were applied to design functional gels. As shown in Fig. 1, the gel was prepared by three F-T repeating followed with soaking in a 26.5 wt% NaCl solution to induce hydrophobic association of CS chains, serving as additional cross-linking points to decrease the dangling ends and loops of PVA chains. This method results in gels with exciting mechanics (strength = 3800 kPa, strain = 270%, and toughness = 5900 kJ/m<sup>3</sup>), excellent antiseptic property, electrical conductivity and anti-

swelling ability, which might find application in cartilage repair, artificial skin and sensors.

## Experimental section

### Materials

Poly(vinyl alcohol) (PVA,  $M_n = 74,800$  g/mol, degree of alcoholysis is 99%) and sodium chloride (NaCl, AR) were provided by Aladdin Reagent (Shanghai) Co., Ltd. Chitosan (CS, degree of deacetylation is higher than 90%,  $M_n = 4000$  g/mol) was provided by TCI Shanghai. All agents were used without purification.

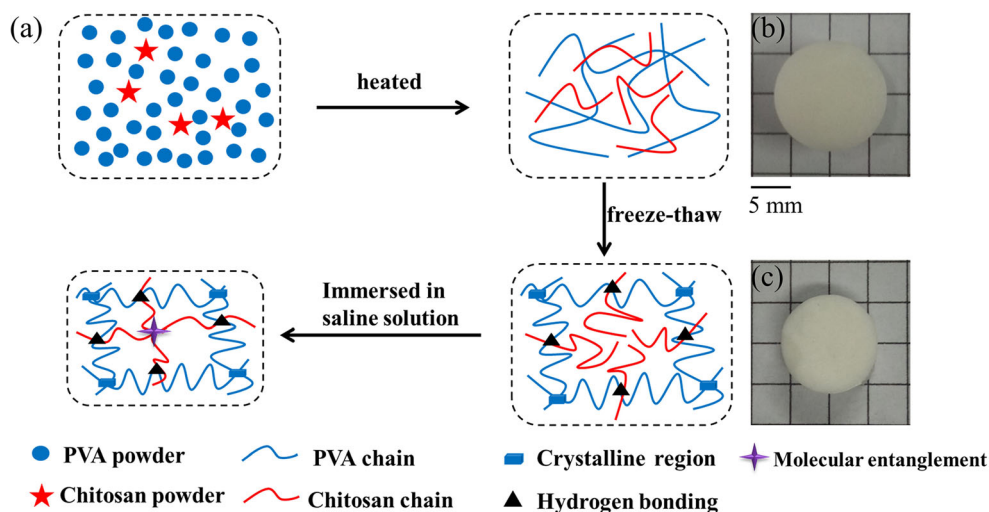
### Synthesis of PVA-CS gels

The PVA-CS-X-S gel was fabricated by simple mixing and soaking. Firstly, PVA and CS powders were dissolved in hot water to form a transparent solution with the PVA mass fraction of 10%. After injecting into molds, they were put in a refrigerator ( $-20$  °C) for 8 h and then stood at room temperature for 4 h. Through three Freezing-Thawing cycles, the PVA-CS-X gels were obtained, where X was the mass ratio of chitosan to poly (vinyl alcohol). These gels were immersed in a 26.5 wt% NaCl solution to achieve the resulted PVA-CS-X-S gels.

### Characterization and instruments

FTIR analysis was carried out using the KBr method within 4000–500 cm<sup>-1</sup> wavenumber (Nicolet Avatar 380 spectrometer). SEM photographs were taken by the ZEISS EVO-18 electron microscopy, and freeze-dried gels were painted with gold to enhance conductivity. DSC experiments were performed on a TA Instruments (model Q20) in the  $N_2$

**Fig. 1** a Synthetic mechanism of the PVA-CS-S hydrogel; transformation process of the b PVA-CS-5 hydrogel into c hybrid PVA-CS-5-S hydrogel



atmosphere from  $-40\text{ }^{\circ}\text{C}$  to  $40\text{ }^{\circ}\text{C}$  for 80 min. XRD was recorded on Rigaku SmartLab X-ray diffractometer(CuK $\alpha$  radiation) from  $5^{\circ}$  to  $80^{\circ}$  ( $2\theta$ ) at a scan rate of  $5\text{ }(^{\circ})/\text{min}$ . Water contents ( $X_n$ ) were estimated by the Eq. (1) [33]:

$$X_n = \frac{Q_{endo}}{Q_f} \times 100\% \tag{1}$$

Where  $Q_{endo}$  and  $Q_f$  are the fusion enthalpy from DSC curves and that from literature ( $333.5\text{ J/g}$ ) [33].

The crystallinity of PVA ( $\chi_c$ ) was calculated by the Eq. (2) [34]:

$$\chi_c = \frac{A_1}{A_2} \times 100\% \tag{2}$$

Where  $A_1$  and  $A_2$  are integral areas in the  $2\theta$  range of  $18^{\circ} \sim 21^{\circ}$  and  $5^{\circ} \sim 80^{\circ}$ , respectively.

### Mechanical properties

Mechanical properties of gels were recorded by a HZ-1003B electronic tensile instrument. Dumbbell-shaped gels were deformed under a tensile rate of  $100\text{ mm/min}$ . The strength ( $\sigma_b$ ), strain ( $\epsilon$ ), elastic modulus ( $E$ ) and toughness ( $T$ ) were calculated by the fracture strength, elongation at break, slope of linear range ( $5 \sim 10\%$ ), and integral area of tensile curves, respectively [35]. To estimate the dissipated energy ( $U_{hys}$ ), samples were stretched to certain strains and then released [11].

### The component content and cross-linking density of the hydrogel

Water content ( $W_{water}$ ) and polymer content ( $W_{gel}$ ) were estimated by Eqs. (3) and (4) [36]:

$$W_{water} = \frac{W_w - W_d}{W_w} \times 100\% \tag{3}$$

$$W_{gel} = \frac{W_G \times C_G}{W_w} \times 100\% \tag{4}$$

Where  $W_w$ ,  $W_d$ ,  $W_G$  and  $C_G$  are the wet weight, dry weight, polymer weight and polymer mass ratio, respectively [36].

The cross-linking density ( $\nu_0$ ) as well as chain lengths between adjacent cross-links ( $M_c$ ) were estimated by Eqs. (5) and (6) [37, 38] within the elastic range ( $\lambda = 2$ ):

$$\sigma = \nu_0 kT (\lambda - \lambda^{-2}) \tag{5}$$

$$M_c = \rho N_A / \nu_0 \tag{6}$$

Where  $\sigma$ ,  $\lambda$ ,  $k$  and  $T$  are the tensile strength, tensile strain, Boltzmann constant, and Kelvin temperature, respectively.

### The gel fraction ( $F_g$ ) and mesh size ( $\xi$ )

The gel fraction ( $F_g$ ) was investigated by a gravimetric method [39, 40]. Typically, freeze-dried samples were weighed ( $W_o$ ) and immersed in DI water at RT to remove the soluble linear polymers. Then they were dried at  $105\text{ }^{\circ}\text{C}$  for 24 h and weighed again ( $W_i$ ). The  $F_g$  was evaluated by Eq. (7) [39, 40]:

$$F_g = \frac{W_i}{W_o} \times 100\% \tag{7}$$

The mesh size ( $\xi$ ) was estimated by Eq. (8) [41]:

$$\frac{3kT}{E} = \frac{4}{3} \pi \left(\frac{\xi}{2}\right)^3 \tag{8}$$

Where  $E$ ,  $k$  and  $T$  are the elastic modulus, Boltzmann constant and Kelvin temperature, respectively.

### Antiseptic property

The antiseptic properties of gels were measured by the observation method [42]. PVA and PVA-CS-5-S gels were stored into glass bakerys at  $25\text{ }^{\circ}\text{C}$  under a humidity of  $30 \sim 35\%$  and covered by a film. After 15 days, two samples were taken out and their surfaces were investigated by a digital camera.

### Conductivity

Sample conductivities were investigated by recording the brightness of a green LED lamp from the gel-connected circuit [42].

## Result and discussion

### Preparation and characterization

The gel was obtained by mixing PVA and CS powders in hot water, followed by three F-T circles and soaking in the sodium chloride aqueous solution (Fig. 1). The interpenetrating network was firstly obtained after three F-T cycles to form crystalline microdomains of PVA, with CS chains embedded in the sticky PVA matrix. They were further soaked in a saturated NaCl solution to induce the micro-phase separation of CS chains. By doing so, the cross-linking density might increase with a decrease in dangling ends and loops of polymer chains [25]. Besides, the reversible break/reformation of multiple physical interactions (hydrophobic association, crystalline region, and hydrogen bonding) was effective to dissipate energy, which was important to achieve tough gels. Gel structures were firstly confirmed by FTIR. As shown in Fig. 2, CS displayed typical absorptions at

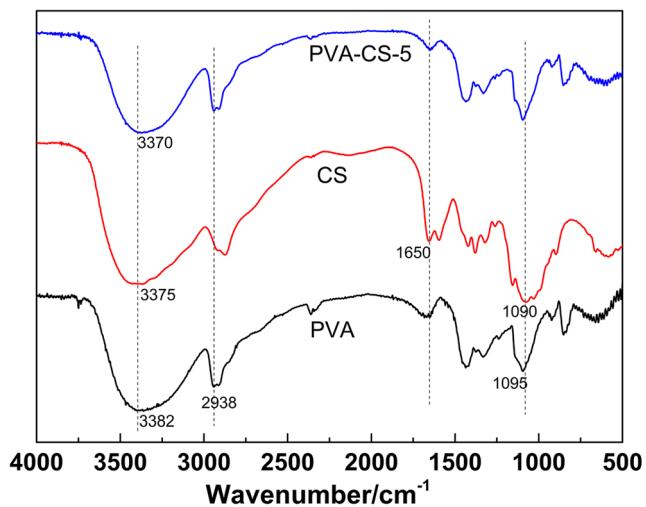


Fig. 2 FTIR spectra of PVA, CS and PVA-CS-5 gel

3375, 1650 and 1090  $\text{cm}^{-1}$  from the vibration of N-H, C=O and C-O bonds, respectively [43]. The peaks at 3382, 2938 and 1095  $\text{cm}^{-1}$  were associated with the vibration of O-H, C-H and C-O in PVA [44]. Peaks in PVA-CS-5 were well assigned from raw materials, while the O-H stretching at 3382  $\text{cm}^{-1}$  decreased to 3370  $\text{cm}^{-1}$ , suggesting that stronger hydrogen bonds were formed [45].

### Mechanical performances

Mechanical performances of PVA-CS-X gels were evaluated firstly by altering CS contents. As displayed in Fig. 3, tensile strength ( $\sigma_b$ ) and strain ( $\epsilon$ ) of single PVA gel were 200 kPa, 135%, respectively, consistent with literature reports [21]. For PVA-CS-5, the  $\sigma_b$  and  $\epsilon$  reached 900 kPa and 160%, respectively. They all

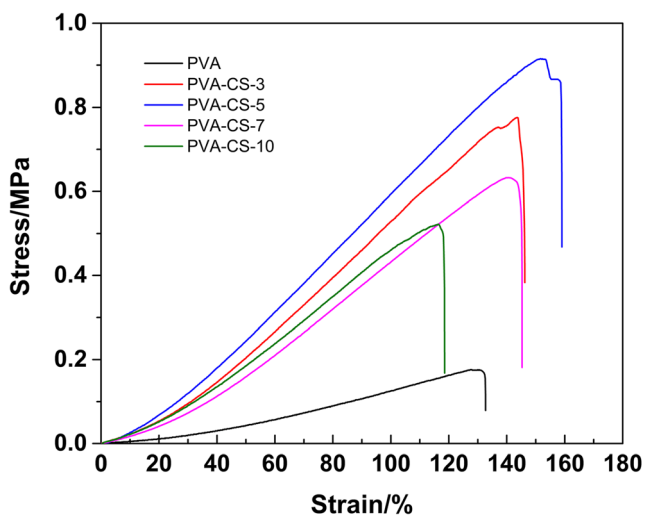


Fig. 3 Stress-strain curves of PVA-CS-X gels with various CS contents

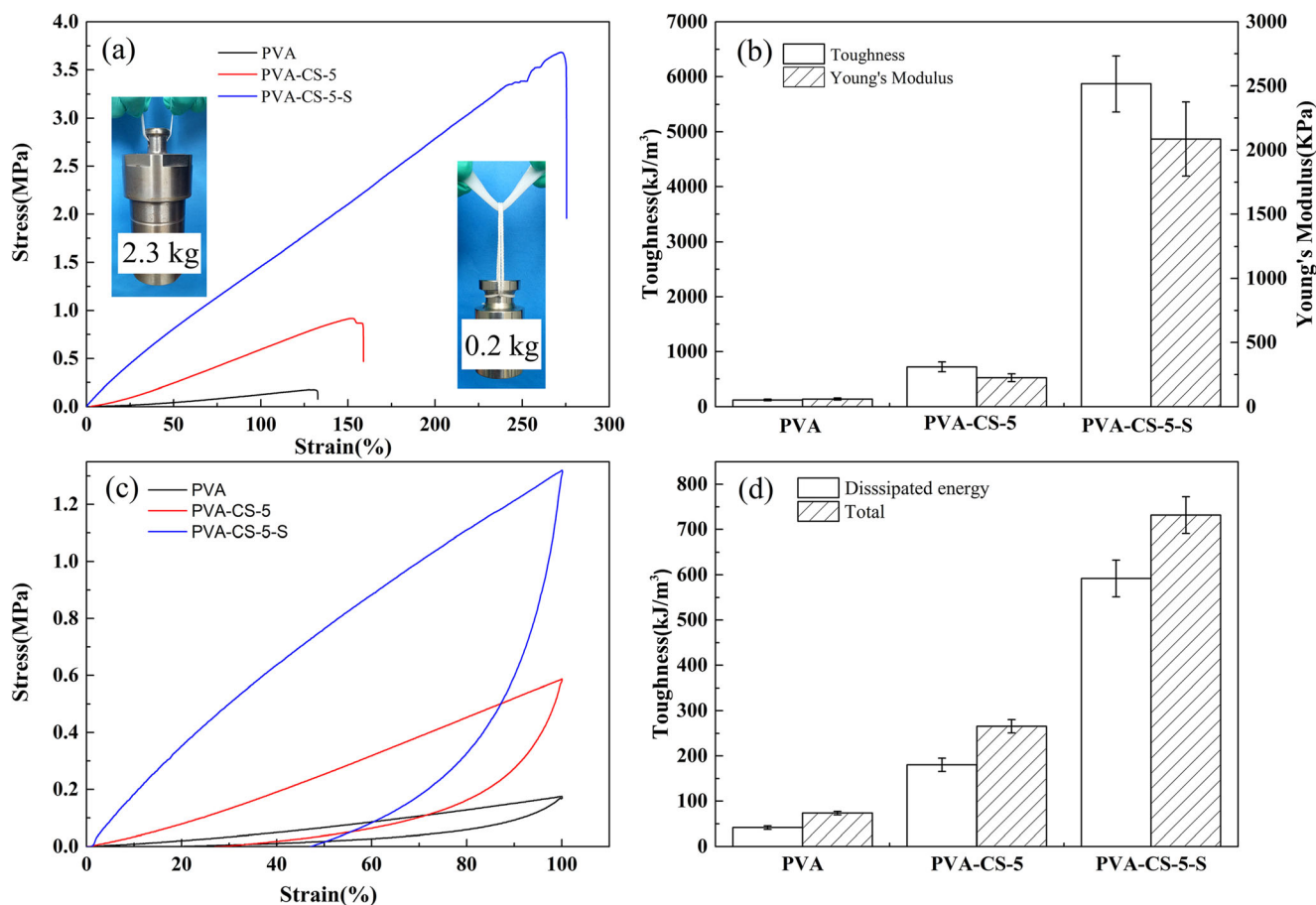
decreased when CS contents increased to 7 wt% and 10 wt%. The cross-linking density was inadequate at lower CS contents, whereas it might be over cross-linked at higher CS contents [46]. Therefore, optimal mechanics of precursor gels were obtained for PVA-CS-5, which was further enhanced by soaking in NaCl solution to form additional joint knots.

Pioneering works revealed that the energy dissipation mechanism, cross-linking densities and network parameters were essential to fabricate tough gels [23–26]. For example, the tensile strength of a PAM/CS composite gel increased 56 times when coordination bonds were formed between cationic amino groups in CS and anionic citrate groups from solution [25]. The solubility of CS in water was also determined by pH values and ionic strength [47]. Therefore, we immersed the PVA-CS-5 precursor gels into a saturated NaCl aqueous solution to induce micro phase separation of CS. To achieve a relatively homogenous network, the soaking time was investigated firstly, and optimally mechanical properties were achieved at 50 min for the PVA-CS-5 gel. As displayed in Fig. 4a, the PVA-CS-5 gel could hold a 0.2 kg load, whereas the PVA-CS-5-S gel could bear a 2.3 kg load, indicating the soaking strategy is successful. Then we systematically investigated its mechanical performances. As illustrated in Fig. 4b, the  $\sigma_b$ ,  $E$  and  $T$  of the resulted gel increased to 3800 kPa, 2100 kPa, 5900  $\text{kJ/m}^3$ , which were 3.2, 8.3 and 7.2 times higher than that of the precursor gel, respectively.

To reveal the structure-property relationship and to shed light on the energy dissipation mechanism, the hysteresis test was performed [48, 49]. As displayed in Fig. 4c and d, three gels all exhibited hysteresis loops, indicating energy dissipation mechanism. However, the dissipated energy of PVA-CS-5-S (592.1  $\text{kJ/m}^3$ ) was much higher than the other two gels, suggesting that chain entanglements of CS after immersion could distribute external load efficiently. Moreover, the ratio of the dissipated energy to the total energy was 80.9%, further confirming the effective energy dissipation mechanism. To summarize, chain entanglements of CS in the sticky PVA matrix could significantly toughen the gel.

### Property-structure relationship

Water states and the corresponding contents, and network parameters were also calculated via DSC, SEM, XRD and theoretical estimation. Solid contents were firstly obtained from the weighing method [36]. As shown in Fig. 5a, water content decreased from 89.1 wt% for PVA-CS-5 to 59.2 wt% for PVA-CS-5-S, accompanied by an increase in polymer contents from 10.9 wt% to 17.7 wt%. A higher polymer mass fraction would lead to a stronger

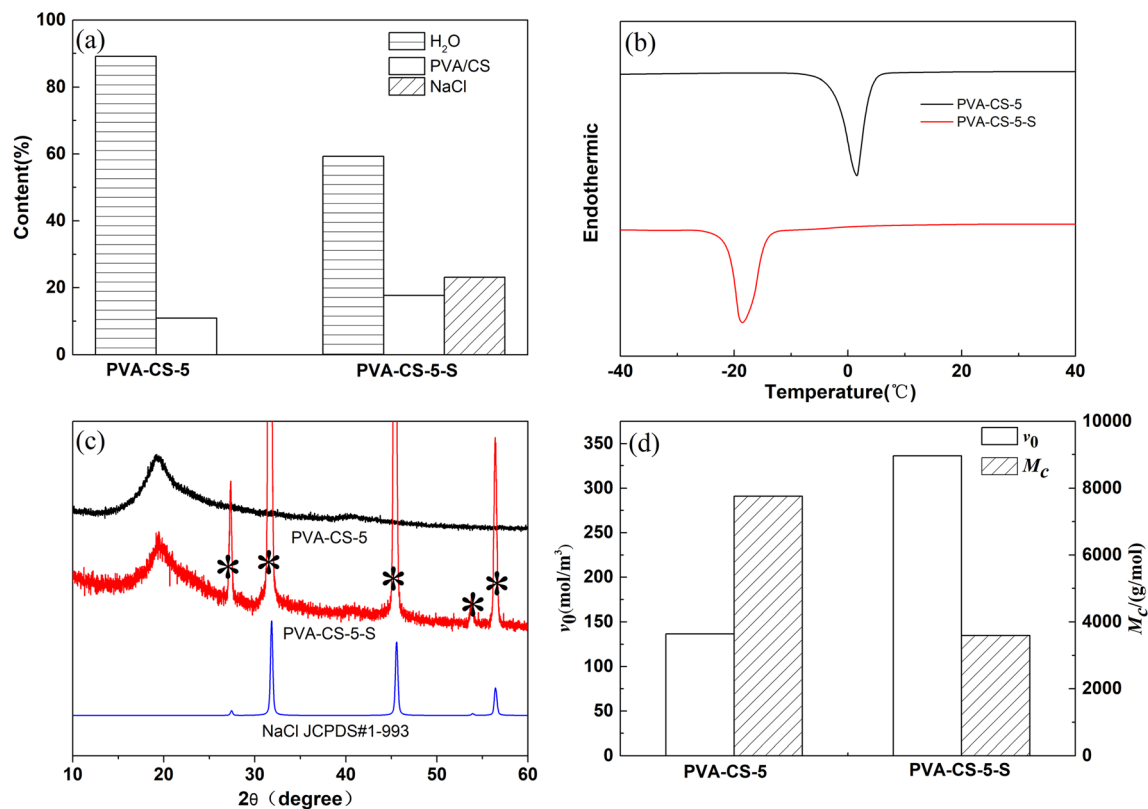


**Fig. 4** a Stress-strain curves, b toughness, young's modulus, c the loading-unloading curves and d the dissipated and total toughness of three gels

gel [50]. Contents of free water (FW), freezable bound water (FBW) as well as non-freezable bound water (NBW) were also determined from DSC curves [51, 52]. As displayed in Fig. 5b, the PVA-CS-5 gel exhibited an endothermic peak from FW around 1.5 °C. Based on Eq. (1) [33], contents for FW and NBW were 62.8 wt% and 26.3 wt%, respectively. After immersion in NaCl, the endothermic peak decreased from 1.5 °C to -18.4 °C. The contents for FW, FBW and NBW were 0, 33.3 wt% and 25.9 wt%, respectively. Generally, the decline of FW content would increase internal frictions of polymers [42], resulting in better mechanical performances. The crystallinity of PVA was analyzed by XRD. As shown in Fig. 5c, both samples shown strong characteristic diffraction peaks in the  $2\theta$  range 18~21°, corresponding to the 101 crystal plane of PVA. The crystallinity of PVA reduced slightly from 14.8% to 12.3% after soaking. The effective cross-linking density ( $\nu_0$ ) of the gel and molecular weight of cross-linking points ( $M_c$ ) were also calculated from Eqs. (5) and (6). As displayed in Fig. 5d, the  $\nu_0$  and  $M_c$  for the former were only 136.6 mol/m<sup>3</sup> and 7757 g/mol, while became

336.1 mol/m<sup>3</sup> and 3594 g/mol for the later. A rise in  $\nu_0$  (or a decline in  $M_c$ ) would result in a stronger gel [37]. In summary, free water content and crystallinity of PVA decreased while the frictions between polymer chains and cross-linking density increased after soaking in NaCl. Consequently, the gel was enhanced.

Network parameters were also revealed by SEM. As displayed in Fig. 6, they all exhibited typical spongy structures [53]. However, single PVA gel (Fig. 6a) had uneven heterogenous structures with pore sizes ranging from 1.5 to 8.2 μm. The network became denser after mixing with CS (Fig. 6b). After immersion into the saturated NaCl solution, the pore size decreased to ~1.0 μm (Fig. 6c), suggesting a densest network. To further clarify this, gel fraction ( $F_g$ ) of two gels were estimated, assuming that polymers were in Gaussian distribution in the gel network [41]. The  $F_g$  for PVA-CS-5 and PVA-CS-5-S were 58.6% and 87.2%, respectively. Higher  $F_g$  indicated fewer defects (loops and dangling chains), consistent with previous calculation in Fig. 5d. Meanwhile, the mesh size of network decreased from 98 Å to 48 Å. The results indicated that the gel became more compact and uniform

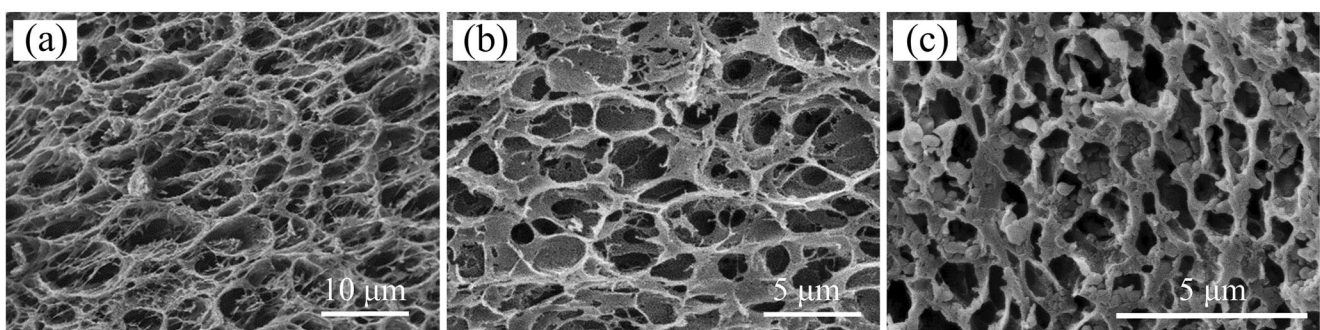


**Fig. 5** a The mass contents of each component, b DSC curves, c XRD patterns (peaks from NaCl was marked with star); d  $v_0$  and  $M_c$  of two gels upon soaking

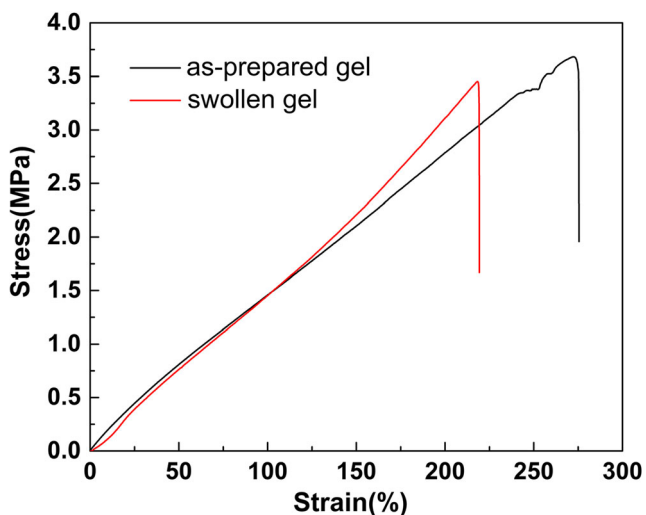
after soaking, consistent with SEM images. Generally, a rise in gel fraction and a decline in mesh size would result in a stronger gel [41].

Considering the potential application of our gel is bone repair, and the human body is a water-rich environment, we further examined the tensile properties of PVA-CS-5-S gel when reaching a swelling equilibrium in DI water. As displayed in Fig. 7, the  $\sigma_b$  and  $\epsilon$  of the swollen gel were 3400 kPa and 218%. Compared with the as-prepared gel, the retentions of stress and strain were as high as 89.5% and 80.7%, respectively, indicating good swelling-resistant ability. The mechanical

properties of our gel at the swollen state are among the top values of physical gels in literature [10, 11, 23, 44]. More importantly, the stress is much higher than traditional PAM gels at swollen states. For instance, the  $\sigma_b$  of a traditional PAM gel was only 20 kPa [27]. The excellent anti-swelling ability was probably due to the fewer defects and even distribution of knots in the gel network [27]. Another contribution might be the multiple interactions in our system including crystallinity of PVA and hydrophobic association of CS chains, which were rather stable at room temperature and non-acidic water, respectively [20, 28, 32].



**Fig. 6** SEM photographs of a PVA, b PVA-CS-5 and c PVA-CS-5-S gels



**Fig. 7** Mechanical properties of the as-prepared (black) and swollen (red) PVA-CS-5-S gel

### Multi-functionalities of the obtained gel

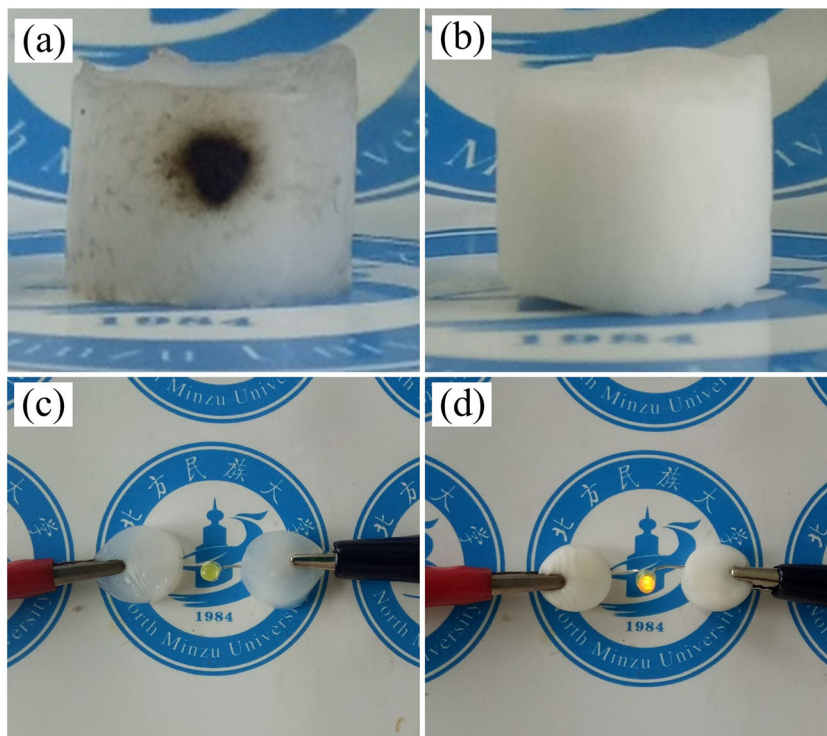
As a natural polysaccharide, chitosan was non-cytotoxic, pH-sensitive and antiseptic, which has shown promise in drug delivery for several years [28–32]. We were curious about the functions of this gel. We then investigated the antiseptic property as well as the conductivity of PVA and PVA-CS-5-S gels. As Fig. 8a displayed, after 15 days culture, obvious mold growth was observed at the surface of PVA gel, while nothing

appeared on the surface of PVA-CS-5-S gel, indicating good antiseptic properties. Later, the conductivities of two samples were compared. As illustrated in Fig. 8c and d, the LED bulb was shining when PVA-CS-5-S gel was assembled in the circuit, while it was dim when PVA gel was adopted, suggesting potential applications in sensing elements and wearable devices. Furthermore, the NaCl concentration was around 0.07 mmol per adult body weight (ABW) (supposing ABW = 60 kg) [54, 55], less than 205 mmol/kg ABW, which might induce hypernatremia [56], indicating our gel maintained the biocompatibility of PVA and CS.

### Conclusion

In this work, an antiseptic, conductive, swelling-resistant and robust gel was fabricated from two commercially available materials, PVA and CS. Because of the synergistic interactions including chain entanglements of CS, hydrogen bonds between PVA and CS, and crystalline domains of PVA, the resulted gels exhibited excellent mechanical performances, superior to most physical gels. Besides, it was antiseptic, conductive and swelling-resistant. The tensile strength of our gel was 3400 kPa even reaching a swelling equilibrium in water, much stronger than most water-rich gels. The robust and multi-functional material would have potential applications in tissue engineering and sensors.

**Fig. 8** The antiseptic property of **a** PVA gel and **b** PVA-CS-5-S gel, the conductive properties of **c** PVA gel and **d** PVA-CS-5-S gel



**Acknowledgements** This work was supported by the Specialized Research Fund in Ningxia Higher Education Institutions (NGY2018-165), Natural Science Foundation of Ningxia Province (2020AAC03205), and Natural Science Foundation of China (21464001).

## Compliance with ethical standards

**Competing interest** The authors declare no competing financial interest.

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