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# Nitrocellulose-based hybrid materials with T7-POSS as a modifier: effective reinforcement for thermal stability, combustion safety, and mechanical properties

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Abstract Novel nitrocellulose (NC)-based hybrid materials with self-synthesized heptaphenyltricycloheptasiloxane trihydroxy silanol  $(T_7$ -POSS) as a modifier were prepared using a "one-step" chemical cross-linking process. To comprehensively demonstrate the superiority of the modifier, hybrid materials with different contents of  $T<sub>7</sub>$ -POSS were assessed. The gel content was measured, and the chemical structure and composition of the  $T<sub>7</sub>$ -POSS-NC hybrid materials were characterized by attenuated total reflection Fourier transform infrared spectroscopy (ATR-FTIR) and X-ray photoelectron spectroscopy (XPS). Thermogravimetric analysis (TGA) results showed that the thermal stability of the  $T_7$ -POSS-NC hybrid materials increased with the  $T<sub>7</sub>$ -POSS content. Typically, when 12.3 wt.%  $T_7$ -POSS was incorporated into the NC, the decomposition temperature based on 50% weight loss (T<sub>50%</sub>) was delayed from 183.3 °C to 243.5 °C, the maximum weight loss rate (WLR $_{\text{max}}$ ) decreased markedly from 432.9%/min to 1.3%/min, and the char residues increased from 1.4% to 26.0%. The scanning electron microscopy (SEM) results of the char residues indicated that the introduction of  $T<sub>7</sub>$ -POSS led to the formation of a sufficient and compact char layer. Notably, the incorporation of  $T<sub>7</sub>$ -POSS improved not only the combustion safety according to micro-scale combustion calorimeter (MCC) results but also the mechanical properties due to the formation of cross-linking networks and the good distribution of  $T_7$ -POSS particles, which was confirmed by SEM and energy-dispersive spectroscopy (EDS).

 $\boxtimes$  Jianwei Hao hjw@bit.edu.cn Keywords Nitrocellulose . POSS . Cross-linking . Combustion safety . Thermal stability

## Introduction

Nitrocellulose (NC) with low nitrogen content, produced from the esterification of natural cellulose through the addition of nitric acid, has been used in different applications [\[1\]](#page-8-0). For example, it forms a tough thermoplastic film on photographic film, and it has also been applied in the coating, printing ink, adhesive, and artificial leather fields [[2,](#page-8-0) [3\]](#page-8-0). Unfortunately, the poor thermal stability and the rapid burning of NC (the rate of combustion of a NC film is approximately 15 times faster than that of wood [[4](#page-9-0)]) limit its use in many desirable applications. As previously reported, NC with  $12.1 \sim 12.3\%$  nitrogen content showed a very steep loss in weight, amounting to approximately 99% of the original weight [[5\]](#page-9-0), far more than cellulose, which puts its applications in jeopardy. The poor mechanical properties of NC also restrict its widespread use [[6,](#page-9-0) [7](#page-9-0)].

Some studies have reported various strategies to improve the thermal stability of NC with high nitrogen content, includ-ing adding different stabilizers [\[8](#page-9-0)], mixing in  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles [\[9\]](#page-9-0) or graphene oxide water solution [\[10\]](#page-9-0), and washing with industrial water and inorganic salts [[11\]](#page-9-0). However, to the best of our knowledge, there has been no report in the open literature on improving both the thermal stability and combustion safety of NC with low nitrogen content.

Hybrid materials, including both blends in which inorganic phases are physically dispersed in organic matrices, and networks in which the inorganic and organic components are linked through strong chemical bonding, have attracted attention for their potential in combining excellent thermal stability

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<span id="page-1-0"></span>**Scheme 1** Synthetic route of  $T_7$ -POSS



[\[12,](#page-9-0) [13](#page-9-0)] and mechanical properties [[14](#page-9-0)–[16](#page-9-0)]. Recently, polyhedral oligomeric silsesquioxane (POSS), which has an organic-inorganic structure, was introduced into polymer matrices to construct POSS-containing hybrid materials. For example, Bershtein V. synthesized a series of CER/ECH-POSS hybrid materials with varying POSS contents [[17\]](#page-9-0).

Many methods have been used to prepare cross-linked hybrid materials; these methods include the sol-gel process, high-energy radiation cross-linking, and chemical crosslinking [[18](#page-9-0)–[20](#page-9-0)]. Among these methods, chemical crosslinking is one of the most effective method because it is not very time- or energy- consuming, and can be performed without radiation.

Thus, in this work, to increase the thermal stability and improve the combustion safety as well as the mechanical properties of NC, heptaphenyltricycloheptasiloxane trihydroxy silanol  $(T_7$ -POSS), which has three -OH groups in each POSS unit, was synthesized and used as a modifier to prepare novel cross-linked  $T<sub>7</sub>$ -POSS-NC hybrid materials via a "one-step" chemical method. Then, the thermal decomposition characteristics, combustion safety and mechanical properties of the samples were systematically evaluated.

# Experimental

## **Materials**

NC (nitrogen content  $\sim$ 11.91 wt.%) was kindly provided by Baoding BaoFeng Nitrocellulose Co., Ltd. (Hebei, China). All NC samples were pre-treated in a freeze dryer at −80 °C for 48 h to remove residual water; the samples were then stored in a desiccator before use. Phenyltrimethoxysilane, 97% (Alfa

Aesar) was used without further treatment. Dibutyltin dilaurate (DBTL) as a catalyst and isophorone diisocyanate (IPDI) as a cross-linker were purchased from Alfa Aesar. Tetrahydrofuran (THF) and N, N-dimethylformamide (DMF) were further purified by distillation over  $CaH<sub>2</sub>$  and stored in a sealed vessel with a molecular sieve of 4 A. All other reagents were obtained from Beijing Chemical Works and used as received.

# Preparation

#### Synthesis of  $T_7$ -POSS

Heptaphenyltricycloheptasiloxane trisodium silanolate  $(T<sub>7</sub>-ONa)$  was synthesized according to the literature [[21](#page-9-0)]. Typically, phenyltrimethoxysilane (10.0 g, 0.05 mol), deionized water (1.15 g, 0.06 mol), THF (80 mL) and sodium hydroxide (0.85 g, 0.02 mol) were charged into a three-necked flask equipped with a reflux condenser and a magnetic stirrer. The mixture was refluxed in an oil bath at 70 °C for 5 h and then cooled to room temperature for an additional 15 h with vigorous stirring. After filtration, the precipitate was collected and washed with THF three times. The resulting white powder was dried in vacuum at 50 °C for 24 h to obtain 6.98 g of the products with a yield of 97%.

 $T_7$ -POSS was obtained as follows [\[22\]](#page-9-0): 5 g  $T_7$ -ONa and 50 mL trichloromethane were added into a separatory funnel, and several droplets of acetic acid were added. After vigorous shaking and washing with deionized water more than three times, the organic phase was precipitated into 400 mL hexane to obtain



Samples	NC(2)		IPDI (g) $T_7$ -POSS (g)	$T_7$ -POSS $(wt, \%)$
NC control	6	$\theta$	$\theta$	$\theta$
4.5 wt.% $T7$ -POSS-NC		0.4	0.3	4.5
8.6 wt.% $T_7$ -POSS-NC		0.4	0.6	8.6
12.3 wt.% $T_7$ -POSS-NC		0.4	0.9	12.3

<span id="page-2-0"></span>**Table 1** Preparation of NC control and  $T<sub>7</sub>$ -POSS-NC hybrid materials

white powders with a yield of 89%. The synthetic route of T<sub>7</sub>-POSS is shown in Scheme [1.](#page-1-0)

#### Preparation of  $T<sub>7</sub>$ -POSS-NC hybrid materials

The synthetic route of the  $T<sub>7</sub>$ -POSS-NC hybrid materials is shown in Scheme [2.](#page-1-0) The general preparation procedures of the  $T_7$ -POSS-NC hybrid materials are as follows. NC (6 g) was charged into a beaker with anhydrous DMF (20 mL) and stirred at ambient temperature for 1 h by mechanical stirrer. After the formation of a NC homogeneous solution, an anhydrous DMF solution (10 mL) of  $T<sub>7</sub>$ -POSS (with different contents) and IPDI (0.4 g) were added into the flask with vigorous stirring for 20 min, and then the bubbles were removed by ultrasonication for approximately 10 min. Subsequently, the mixture was poured into a PTFE casting mold with dimensions of 100 mm  $\times$  100 mm to obtain hybrid films of  $\sim$ 1 mm thickness at specific conditions (dried for 2 d at room temperature and for 2 h at 70 °C in an oven). Finally, the films were put in a vacuum for 48 h to remove any residual solvent. From this process, a series of  $T<sub>7</sub>$ -POSS-NC hybrid materials were obtained, following the detailed preparation presented in Table 1.

## Characterization methods

#### Structural characterization

T7-POSS Fourier transform infrared spectroscopy (FTIR) was performed in the range of 4000  $\sim$  400 cm<sup>-1</sup> with a Tensor-27

Fig. 1 (a) FTIR and (b)  $^{29}$  Si NMR spectra of  $T_7$ -POSS



(Bruker) using the KBr disk method at room temperature. <sup>29</sup> Si nuclear magnetic resonance  $(^{29}$  Si NMR) spectra were measured at 400 MHz using an AV400 (Bruker) in CDCl<sub>3</sub> at 25 °C, and

tetramethylsilane (TMS) was used as an internal standard. An electrospray mass spectrometer (ES-MS) was applied with UPLC-Q-Tof-MS (Waters) using CH<sub>3</sub>OH as the solvent.

 $T<sub>7</sub>$ -POSS-NC hybrid materials The samples were cut into strips, weighed  $(W_1)$ , and then immersed in tetrahydrofuran (THF) in stoppered bottles for 72 h (solvent was replaced every 24 h) to ensure that the uncross-linked components were extracted completely. The swelled samples were then removed and dried under vacuum (60 °C) to constant weights ( $W_2$ ). The gel content was calculated according to Eq. (1) [[23](#page-9-0)]:

$$
Gel content = \frac{W_2}{W_1} \times 100\% \tag{1}
$$

Each sample was tested at least three times. The gel content efficiency was calculated according to Eq. (2):

$$
Gel content efficiency = \frac{Gel content}{Weight percentage of T7 - POSS} \times 100\% \tag{2}
$$



Attenuated total reflection Fourier transform infrared (ATR-FTIR) spectra were collected in the range of

<span id="page-3-0"></span>**Table 2** Gel content and gel content efficiency of NC control and  $T_7$ -POSS-NC hybrid materials

Samples	Gel content $(\%)$	Gel content efficiency <sup>a</sup>
NC control	$\theta$	
4.5 wt.% $T7$ -POSS-NC	$40.6 \pm 2.4$	9.0
8.6 wt.% $T_7$ -POSS-NC	$60.7 \pm 3.1$	7.1
12.3 wt.% $T_7$ -POSS-NC	$65.5 \pm 3.8$	5.3

 $a$ : T<sub>7</sub>-POSS contribution per unit mass to the gel content

 $4000 \sim 400 \text{ cm}^{-1}$  using a Nicolet 6700 FTIR spectrometer at room temperature; the spectra were collected at 32 scans with a spectral resolution of  $4 \text{ cm}^{-1}$ . The elemental compositions of the surfaces were determined by X-ray photoelectron spectroscopy (XPS) (PHI Quantera II SXM) at 25 W under a vacuum environment less than 2.6 × 10<sup>-6</sup> Pa using an Al K<sub>α</sub> X-ray source. The pass energy is 26.0 eV, and the step length is 1 eV. The weight content of Si (Si %) for each sample was calculated according to Eq. (3):

$$
Si\% = \frac{C_A(Si) \times M(Si)}{C_A(C)M(C) + C_A(O)M(O) + C_A(N)M(N) + C_A(Si)M(Si)} \times 100\%
$$
\n(3)

where  $C_A$  (C),  $C_A$  (O),  $C_A$  (N) and  $C_A$  (Si) represent the atomic concentrations  $(C_A)$  of the C, N, O and Si elements on the surface of the  $T<sub>7</sub>$ -POSS-NC hybrid materials, respectively, and M(C), M (O), M (N) and M (Si) are their relative atomic masses (M).

#### Thermal stability

Thermogravimetric analysis (TGA) was performed using a TG209 (NETZSCH) with a continuous flow of a  $N_2$  atmosphere. Samples (5  $\sim$  10 mg) were heated from 50 °C to 500 °C at a rate of 10 °C/min. The following parameters were determined: initial decomposition temperature based on 5%



Fig. 3 ATR-FTIR spectra of NC control  $(A)$ , 4.5 wt.%  $T<sub>7</sub>$ -POSS-NC  $(B)$ , 8.6 wt.%  $T_7$ -POSS-NC (C) and 12.3 wt.%  $T_7$ -POSS-NC (D)

weight loss  $(T_{5\%})$ , middle decomposition temperature based on 50% weight loss  $(T_{50\%})$ , maximum weight loss rate (WLR<sub>max</sub>) and char residues under 500 °C (CR<sub>500 °C</sub>).

#### Combustion safety

The combustion safety was evaluated using a microscale combustion calorimeter (MCC) (FAA-PCFC) according to ASTM D7309. The samples were cut into strips ( $\sim$  2 mg) and heated from 135 °C to 180 °C at a constant heating rate of 0.5  $\degree$ C/s in a pyrolysis chamber under a nitrogen gas stream (80 cm<sup>3</sup>/min). The pyrolysis products were purged from the pyrolysis chamber by the nitrogen gas stream and then mixed with an oxygen gas stream  $(20 \text{ cm}^3/\text{min})$  prior to entering a 900 °C combustion furnace for the no flaming combustion (oxidation).

#### Mechanical properties

Rectangular tensile bars measuring  $50 \text{ mm} \times 10 \text{ mm} \times 0.1 \text{ mm}$ were obtained using a fresh razor blade. Uniaxial tensile tests (INSTRON1185) were carried out at 25 °C with a crosshead speed of 5 mm/min. The modulus of each sample was determined by linearly fitting the elastic portion of the stress-strain curve, and the results were the averages of 5 measurements.

## Morphology and distribution

The morphology of the char residues after TGA testing was determined by scanning electron microscopy (SEM) (Hitachi S-4700). The samples were quenched and fractured in liquid nitrogen in preparation for the examination of the  $T<sub>7</sub>$ -POSS distribution by SEM and energy-dispersive spectroscopy (EDS). Prior to the imaging, the surfaces of the samples were sputter-coated with gold.

**Table 3** Elemental compositions of the surface for NC control and  $T<sub>7</sub>$ POSS-NC hybrid materials

Samples	Atomic conc. $(\%)$			Weight content	
	$\mathbf{C}$	N	$\Omega$	Si	of Si $(\%)$
NC control 70.12 1.73 28.15 0				$\sqrt{ }$	
4.5 wt.% T <sub>7</sub> -POSS-NC 87.07 2.63 9.05 1.25 2.77					
8.6 wt.% $T_7$ -POSS-NC 71.65 5.41 20.26 2.69 5.64					
12.3 wt.% T <sub>7</sub> -POSS-NC $63.73$ 4.34 28.93 3.00 6.12					

<span id="page-4-0"></span>



# Results and discussion

## Structure of  $T_7$ -POSS

The FTIR spectrum of synthesized  $T<sub>7</sub>$ -POSS is presented in Fig. [1\(](#page-2-0)a). The absorption peaks at 3073 cm<sup>-1</sup>, 1768– 1964 cm<sup>-1</sup>, 1591 cm<sup>-1</sup> and 1432 cm<sup>-1</sup> are attributed to Si-Ph vibrations, and the absorption peaks at 891  $cm^{-1}$  and 3234  $\text{cm}^{-1}$  are attributed to the stretching vibrations of Si-OH. Peaks at 1134–1101 cm−<sup>1</sup> that correspond to Si-O-Si vibrations also appear in the spectrum, which indicates that the reaction occurred. The <sup>29</sup> Si NMR spectrum of  $T_7$ -POSS in CDCl<sub>3</sub> (ppm) is presented in Fig. [1](#page-2-0)(b). The signal at  $-77$  ppm is assigned to Si-O-Si-O-Si-OH (peak a), the signal appearing at −78 ppm corresponds to Si-O-Si-OH (peak b) and the signal at −69 ppm is attributed to Si-OH (peak c). ES-MS was set to record the m/z of the micro molecules, and a significant mo-lecular ion peak at m/z = 9[2](#page-2-0)9.7 (Fig. 2) was obtained for  $T_7$ -POSS. All of these data indicate that the  $T_7$ -POSS was successfully synthesized.

# Structure and composition of NC control and  $T_7$ -POSS-NC hybrid materials

Gel content of  $T<sub>\tau</sub>$ -POSS-NC hybrid materials

The gel content represents the weight fraction of the polymers participating in the cross-linked network. It can be seen that

Table 4  $TGA$  and DTG data of NC control and  $T_7$ -POSS-NC hybrid materials

Samples	$T_{5\%}$ (°C)	$T_{50\%}$ $(^{\circ}C)$	$WLR_{\text{max}}$ $(\% / \text{min})$	$CR_{500}$ °C (%)
NC control	172.9	183.3	432.9	1.4
4.5 wt.% $T7$ -POSS-NC	160.6	179.6	59.5	9.7
8.6 wt.% $T7$ -POSS-NC	159.5	208.9	1.4	20.3
12.3 wt.% $T_7$ -POSS-NC	158.3	243.5	1.3	26.0

the gel content increases with the increasing  $T<sub>7</sub>$ -POSS content, suggesting that  $T_7$ -POSS can promote the formation of a cross-linked structure. To analyze the  $T<sub>7</sub>$ -POSS contribution per unit mass to the gel content, the gel content efficiency is evaluated, and the related data are shown in Table [2.](#page-3-0) The degree of increase of gel content and the gel content efficiency decrease gradually. This result may be attributed to the nearsaturation of cross-linked point under the circumstance of the same IPDI contents. That is, additional unreacted  $T_7$ -POSS particles will be in the presence of the physical blending state.

## Structure of NC control and  $T<sub>7</sub>$ -POSS-NC hybrid materials

ATR-FTIR spectral analysis was conducted to characterize the structures of NC control and  $T<sub>7</sub>$ -POSS-NC hybrid materials (see Fig. [3](#page-3-0)). It should be noted that 4.5 wt.%  $T_7$ -POSS-NC, 8.6 wt.%  $T_7$ -POSS-NC and 12.3 wt.%  $T_7$ -POSS-NC here have had the unreacted substances extracted. For the all samples, the two characteristic peaks at 1635 cm<sup>-1</sup> and 1285 cm<sup>-1</sup> are attributed to  $-NO<sub>2</sub>$  asymmetric and symmetric stretching vibrations, respectively, whereas the peak at 824  $cm^{-1}$  is related to the -ONO<sub>2</sub> vibrations of NC. 4.5 wt.% T<sub>7</sub>-POSS-NC, 8.6 wt.% T<sub>7</sub>-POSS-NC and 12.3 wt.%  $T_7$ -POSS-NC exhibit typical transmittance at 1715 cm<sup>-1</sup>, which is attributed to the  $C = O$  of the T7-POSS-NC hybrid materials. Moreover, new peaks near  $1600 \text{ cm}^{-1}$  are assigned to the urethane bond, suggesting that chemical reactions occurred between the hydroxyl groups  $(-OH)$  in T<sub>7</sub>-POSS and NC and the isocyanate group  $(-NCO)$  in IPDI. The successful cross-linking of  $T<sub>7</sub>$ -POSS onto NC can be further confirmed by the appearance of dominant Si-O-Si stretching vibrations at 1123 cm<sup>-1</sup> and 490 cm<sup>-1</sup> [\[24](#page-9-0), [25\]](#page-9-0).

# Composition of NC control and  $T<sub>\tau</sub>$ -POSS-NC hybrid materials

To further prove the successful cross-linking reaction, XPS was used to detect the atomic concentrations of the C, N, O and Si elements on the surface of  $T<sub>7</sub>$ -POSS-NC hybrid materials with the unreacted substances extracted and NC control. The

<span id="page-5-0"></span>

corresponding data are listed in Table [3.](#page-3-0) The calculated weight content of Si as determined by the atomic concentrations of the C, N, O and Si elements, indicates that  $T_7$ -POSS actively takes part in the cross-linking reaction. Additionally, the calculated weight content of Si increases with the  $T<sub>7</sub>$ -POSS content, although its degree of increase gradually decreases, which is consistent with the result of the gel content analysis above.

## Thermal stability

The thermal stability of NC control and  $T<sub>7</sub>$ -POSS-NC hybrid materials were evaluated by TGA under a nitrogen atmosphere. The TGA and DTG curves are shown in Fig. [4](#page-4-0), and the corre-sponding data are summarized in Table [4.](#page-4-0) The  $T_{5\%}$  values of 4.5 wt.% T<sub>7</sub>-POSS-NC, 8.6 wt.% T<sub>7</sub>-POSS-NC and 12.3 wt.%  $T<sub>7</sub>$ -POSS-NC are decreased slightly compared with that of NC control. As previously reported, the urethane bond is unstable and decomposes at elevated temperature because it will dissociate and re-associate simultaneously from approximately 160 °C [[26\]](#page-9-0). Thus, we suspect that the  $T_{5\%}$  values of 4.5 wt.%  $T_7$ -POSS-NC, 8.6 wt.%  $T_7$ -POSS-NC and 12.3 wt.%  $T_7$ -POSS-NC being lower than that of NC control is due to the partial decomposition of the urethane bonds in the  $T<sub>7</sub>$ -POSS-NC hybrid materials. Although the initial decomposition temperature decreases, the thermal stability of the  $T<sub>7</sub>$ -POSS-NC hybrid materials is improved significantly in the high temperature region. The  $T_{50\%}$  of NC control is 183.3 °C, and the value of 4.5 wt.%  $T_7$ -POSS-NC is not significantly different. This result may be explained by the cross-linking density of 4.5 wt.%  $T<sub>7</sub>$ -POSS-NC not being sufficiently high to form a very tight network, as will be shown in following illustration. The  $T_{50\%}$  values of 8.6 wt.%  $T_7$ -POSS-NC and 12.3 wt.%  $T_7$ -POSS-NC shift to temperatures approximately 25 °C and 60 °C higher, respectively, indicating that the thermal stability of the  $T<sub>7</sub>$ -POSS-NC hybrid materials at high temperature is improved. Additionally, NC control shows a single and rapid weight loss step with a WLR<sub>max</sub> value of 432.9%/min, and the char residues are only 1.4% at 500 °C. In the case of 4.5 wt.%  $T_7$ -POSS-NC, the WLR<sub>max</sub> is decreased to 59.5%/ min, and the char residues are increased to 9.7% at 500 °C. The WLR<sub>max</sub> values of 8.6 wt.% T<sub>7</sub>-POSS-NC and 12.3 wt.% T<sub>7</sub>-POSS-NC are decreased to 1.4%/min and 1.3%/min, respectively, and their char residues are increased to 20.3% and 26.0%, respectively. The increase in the char residue gives rise to the formation of more stable char layers, which may protect the materials from further decomposition and in turn increases the thermal stability [\[27\]](#page-9-0). The trend of the TGA results is consistent with the gel content and weight content of Si mentioned above, indicating that the formation of a compact cross-linked network is the key factor for the thermal properties of  $T<sub>7</sub>$ -POSS-NC hybrid materials [[28](#page-9-0)].

 $T_{5\%}$ : initial decomposition temperature based on 5% weight loss;  $T_{50\%}$ : middle decomposition temperature based on 50% weight loss;  $WLR_{max}$ : maximum weight loss rate;  $CR_{500}$  °C: char residue under 500 °C.

# Morphology of char residues

Fig. 5 presents digital photographs of the char residues after the TGA test for each sample. The char residues for NC control and 4.5 wt.%  $T_7$ -POSS-NC are sparse, and the black traces on internal surfaces of the two crucibles are caused by the gaseous pyrolysis products that come from the rapid pyrolysis of NC control and 4.5 wt.%  $T_7$ -POSS-NC. By contrast, the char residues gradually increased through the formation of a compact cross-linked network and nearly keep their original shape for 8.6 wt.% and 12.3 wt.%  $T_7$ -POSS-NC.







Fig. 7 HRR curves of NC control (A), 4.5 wt.%  $T_7$ -POSS-NC (B), 8.6 wt.%  $T_7$ -POSS-NC (C) and 12.3 wt.%  $T_7$ -POSS-NC (D)

To further clarify the morphology of the char residues after the TGA testing, the outer surface of the char residues for all samples was studied by SEM (Fig. [6](#page-5-0)). The char residues of NC control and 4.5 wt.%  $T_7$ -POSS-NC present loose structures with many holes, which make the char layers become fragmented after being heated. By contrast, the char residues of 8.6 wt.%  $T_7$ -POSS-NC and 12.3 wt.%  $T_7$ -POSS-NC are continuous, with a compact structure on the surface, which is mainly attributed to the formation of a cross-linked network. Additionally, the Si-O-Si structure of the  $T_7$ -POSS may provide a stable layer of protection on the NC matrix [\[29](#page-9-0)].

#### Combustion safety

The peak heat release rate (pHRR) of the materials is probably the most important parameter in determining its hazard in a fire [[30\]](#page-9-0). Therefore, MCC was performed to estimate the combustion safety of all samples. Fig. 7 presents the HRR curves of various samples versus temperature. The detailed data obtained from the MCC results, including the pHRR, heat release capacity (HRC) and sum heat release capacity (sHRC), are summarized in Table 5. NC control presents the highest flammability, with a pHRR value of 984.3 W/g, HRC value of

**Table 5** MCC parameters of NC control and  $T_7$ -POSS-NC hybrid materials

Samples			$pHRR (W/g)$ HRC $(J/g·K)$ sHRC $(J/g·K)$
NC control	984.3	1362	1241
4.5 wt.% $T_7$ -POSS-NC	965.2	1322	1200
8.6 wt.% $T_7$ -POSS-NC	901.6	1267	1150
12.3 wt.% $T_7$ -POSS-NC 746.9		1041	932

1362 J/g⋅K and sHRC value of 1241 J/g⋅K. Incorporating  $T_7$ -POSS into the NC sample reduces the pHRR, HRC and sHRC, especially for 12.3 wt.%  $T_7$ -POSS-NC, for which they remarkably decrease by 24.1%, 23.6% and 24.9%, respectively, suggesting that the combustion safety of 12.3 wt.%  $T_7$ -POSS-NC is clearly improved. Notably, the decreasing trend of the MCC results is not in conformance with that of the gel content efficiency above. This result may be because either the cross-linked network or unreacted  $T<sub>7</sub>$ -POSS particles contribute to the improvement of the combustion safety.

## Mechanical properties

Fig. [8](#page-7-0) presents the stress-strain curves, tensile strengths and tensile moduli of all samples. The related data are shown in Table [6.](#page-7-0) Both 4.5 wt.%  $T_7$ -POSS-NC and 8.6 wt.%  $T<sub>7</sub>$ -POSS-NC exhibit higher tensile strengths than that of NC control, which is attributed to the formation of cross-linking networks among NC,  $T_7$ -POSS and IPDI. It is noteworthy that the tensile strength of 12.3 wt.%  $T_7$ -POSS-NC is decreased, whereas its elongation at break is drastically increased from 3.84% for NC control to 8.78%. As we know,  $T_7$ -POSS acts both as a cross-linking agent and as a plasticizing agent in the polymer matrix [\[22\]](#page-9-0). When the content of  $T_7$ -POSS is low, cross-linking plays the leading role; by contrast, when the content is high, the plasticization of nanoparticles occupies the dominant position. In other words, the two main factors are in a competition state. On the one hand, restraining the fracture of materials depends on restricting the polymer chain slippage on relatively large spatial scales [[31\]](#page-9-0) because of the cross-linking, which is the main reason for the improvement of the tensile strength. On the other hand, the blended  $T_7$ -POSS nanoparticles increase the free volume [\[31\]](#page-9-0), which enhances the ductility and reduces the rigidity of the  $T_7$ -POSS-NC hybrid materials. Intuitively, the elongation at break increases but the tensile modulus decreases. In addition, the introduction of  $T_7$ -POSS weakens the interactions (e.g., hydrogen bonding) between NC molecules in the  $T<sub>7</sub>$ -POSS-NC hybrid materials, which is also the reason for the decrease in the tensile moduli of the  $T_7$ -POSS-NC hybrid materials.

Figure [9](#page-7-0) can be used to visually explain the mechanical results mentioned above. As depicted, the NC con-trol (Fig. [9a](#page-7-0)) is a linear polymer, whereas the  $T_7$ -POSS-NC hybrid materials present cross-linking networks. Moreover, the cross-linking degree of the  $T<sub>7</sub>$ -POSS-NC hybrid materials is proportional to the  $T_7$ -POSS content to some extent. For example, when 4.5 wt.% and 8.6 wt.%  $T_7$ -POSS are incorporated into the NC, the

<span id="page-7-0"></span>



**Table 6** Mechanical properties of NC control and  $T_7$ -POSS-NC hybrid materials

Samples	Tensile strength (MPa)	Elongation at break $(\% )$	Tensile modulus (MPa)
NC control	$42.9 \pm 1.2$	$3.84 \pm 0.32$	$1953 \pm 108$
4.5 wt.% $T_7$ -POSS-NC	$44.9 \pm 3.1$	$4.37 \pm 0.76$	$1828 \pm 281$
8.6 wt.% $T7$ -POSS-NC	$46.9 \pm 2.5$	$6.08 \pm 0.91$	$1553 \pm 156$
12.3 wt.% $T_7$ -POSS-NC	$40.6 \pm 4.0$	$8.78 \pm 0.87$	$1084 \pm 174$

degree of cross-linking is increased gradually (Fig. 9b, c). In this case, the cross-linking factor in the competition dominates, which gradually increases the tensile strength of the NC hybrid materials. When 12.3 wt.%  $T<sub>7</sub>$ -POSS is introduced into the NC (Fig. 9d), the crosslinking degree may become saturated, and the unreacted (blended)  $T_7$ -POSS nanoparticles, acting as plasticizers, effectively increase the elongation at break. As we know, the material brittleness is inversely proportional to the elongation at break [\[32\]](#page-9-0). Therefore, the plasticization of  $T<sub>7</sub>$ -POSS can be a solution for the ductility improvement and the brittleness reduction of NC-based materials.

# Distribution of  $T_7$ -POSS

The thermal and mechanical properties of the resultant hybrid materials are affected by the distribution of the POSS moieties in the polymer matrices. The morphologies of the fractured surfaces for all samples as deter-mined by SEM are shown in Fig. [10.](#page-8-0)  $T<sub>7</sub>$ -POSS with a dimension of approximately 2 μm is well distributed in the entire region according to the SEM and EDS Si maps, leading to improved thermal and mechanical properties of the  $T_7$ -POSS-NC hybrid materials. The weight percentage of Si of the fracture surface for 4.5 wt.%  $T_7$ -POSS-NC, 8.6 wt.%  $T_7$ -POSS-NC and 12.3 wt.%  $T_7$ -POSS-NC as detected by the EDS Si maps is 0.91%, 1.71% and 1.94%, respectively, and the increasing trend coincide with the results of the XPS. It is noteworthy that 12.3 wt.%  $T_7$ -POSS-NC shows a coarse fractured surface and plastic flow property, which is very different from 4.5 wt.%  $T_7$ -POSS-NC and 8.6 wt.%  $T_7$ -POSS-NC. This phenomenon can further confirm that the increasing of the elongation at break mentioned above is related to the

Fig. 9 Illustration of NC control (a), 4.5 wt.%  $T_7$ -POSS-NC (b), 8.6 wt.%  $T_7$ -POSS-NC (c) and 12.3 wt.%  $T_7$ -POSS-NC (d)



<span id="page-8-0"></span>Fig. 10 SEM image of fracture surface for NC control (a), 4.5 wt.%  $T_7$ -POSS-NC (b), 8.6 wt.%  $T_7$ -POSS-NC (c) and 12.3 wt.%  $T_7$ -POSS-NC (d)



unreacted  $T<sub>7</sub>$ -POSS nanoparticles that act as plasticizers in the  $T_7$ -POSS-NC hybrid materials.

# Conclusion

Novel  $T<sub>7</sub>$ -POSS-NC hybrid materials with superior thermal stability, high combustion safety and good mechanical properties have been successfully prepared, and the performances are comprehensively characterized. After extracting the unreacted substance, the gel content of the  $T<sub>7</sub>$ -POSS-NC hybrid materials increased from 40.6% to 65.5%. When 12.3 wt.%  $T_7$ -POSS was incorporated into the NC sample, the  $T_{50\%}$  shifted upward by 60.2 °C higher, the WLR<sub>max</sub> decreased by 99.7%, and the char residues were 18.6 times as heavy as that of NC control based on the TGA results. Meanwhile, the peak heat release rate, the heat release capacity and the sum heat release capacity were remarkably reduced by 24.1%, 23.6%, and 24.9%, respectively. These results clearly indicate the improvement of the thermal stability and combustion safety of  $T_7$ -POSS-NC hybrid materials, and the mechanical properties of the  $T<sub>7</sub>$ -POSS-NC hybrid materials were also improved to some extent. In detail, the tensile strength can be increased from 42.9 MPa to 46.9 MPa, and the elongation at break can be increased from 3.84% to 8.78% after incorporating different contents of  $T<sub>7</sub>$ POSS. In summary, the improvement in the thermal stability, combustion safety and mechanical properties for  $T<sub>7</sub>$ -POSS-NC hybrid materials are attributed to the formation of a cross-linked network, the incorporation of unreacted  $T<sub>7</sub>$ -POSS particles and the good distribution of  $T_7$ -POSS.

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