

Layer-by-layer thin films of polyaniline alternated with natural rubber and their potential application as a chemical sensor

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Abstract Self-assembled films of polyaniline (PAni) alternated with natural rubber (NR) were successfully produced by layer-by-layer self-assembly technique (LbL), by using the latex of natural rubber obtained from the *Hevea-brasiliensis* tree. The centrifuged NR latex was used without any previous treatment. The influence of pH on the adsorption kinetics and film growth was evaluated. The films presented a typical linear growth, showing a constant adsorption of material in each deposition step. The FTIR spectroscopy study confirmed the presence of rubber chains (cis-1,4-poly-isoprene) on the LbL films. The morphological characterizations showed great roughness and a spherical shape of the NR particles as a predominant characteristic on the film's surface. Finally, the potential use of the films as a chemical sensor was identified in our tests.

Keywords Self-assembly · Polyaniline · Natural rubber · Thin film · Chemical sensor

Introduction

The science of thin films has an essential role in technology, due to the versatility of applications and unique electrical,

optical and thermal properties exhibited at nanoscale. In the last few years, many studies have been performed by using polyaniline (PAni) as a sensitive material for many applications [1–3], and natural rubber (NR) has been investigated for its potential for biomedical applications [4, 5]. Recently, the search for biopolymers to develop thin films for sensor applications has aimed to replace synthetic materials, making sensors more versatile, biodegradable, and non-toxic, with lower processing costs [1].

In this scenario, the layer-by-layer self-assembly technique (LbL) [6] is one of the most common and inexpensive techniques employed to produce thin films, which has a good control of thickness and molecular architecture. Different mechanisms can be used to build up multilayer films; basically, the LbL technique consists of the alternate adsorption of molecular layers via electrostatic interaction, but can also be driven by hydrogen bonding or bio-specific interactions [7].

Polyaniline is a conductive polymer that has been studied extensively, owing to its excellent chemical stability in a doped state, solubility, low cost, easy processing, and doping when compared with other polymers [8], and it can also be employed as a sensitive material for sensor applications [8–10]. Thin LbL films of polyaniline alternated with a large variety of other polymers have already been produced [9–14], including synthetic rubber for pressure sensor materials [15], but, to our knowledge, this is the first report on the fabrication of LbL films of PAni alternated with NR evaluated as a chemical sensor. Natural rubber is constituted by spherical particles ranging from 5 nm to 3 μm, surrounded by proteins, lipids, and long chains of fatty acid, which confers a negative charge to the particles; this configuration is stable only for pHs higher than 5.0. The basic constituents of NRs are polymer chains of cis-1,4-poly-isoprene with a high molecular weight [16]. The main

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advantage of NR is that it presents elasticity, flexibility, resilience and it is a common natural polymer that is easy to handle. It can be an inexpensive and easy alternative for producing thick or thin LbL films with interesting morphological and mechanical properties [5].

In this work, LbL films of PANi alternated with NR latex were produced by employing PANi doped and undoped, and the results showed that in both cases, the film presented similar physical-chemical characteristics but the morphology was quite different. Its potential application as a chemical sensor was evaluated for different analytes, by using capacitance measurement.

Experimental details

Polyaniline synthesis and solution preparation

Polyaniline was chemically synthesized with ammonium peroxydisulfate $(\text{NH}_4)_2\text{S}_2\text{O}_8$ in an aqueous HCl solution (1.0 M), as described elsewhere [17]. The monomer to oxidant ratio was kept at 4:1 with polymerization carried out at 0 °C for about 3 h to yield the polymer in the emeraldine base form (EB). Dedoping was performed by treatment with a 0.1 M ammonium hydroxide solution for 16 h at room temperature. The polymer was then dried under a dynamic vacuum for 24 h at room temperature.

The PANi solutions were prepared by the dissolution of PANi in N,N-Dimethylacetamide (DMAc) under constant stirring for 16 h; the final solution was filtered to remove undissolved PANi. Five parts of deionized water in volume was added to the final solution before being used in the deposition process. The presence of water in the solution facilitates the deposition process [18] and this ratio of solvent to water was determined by a solution stability study [19]. Finally, the concentration of the solution was adjusted by using Lambert-Beer's law [20], by measuring the solution absorption in the UV-vis region.

Natural rubber and solution preparation

Natural rubber (NR) latex was collected from *Hevea brasiliensis* trees (Clone RIMM 600) [Experimental Farm of the University of São Paulo State (UNESP), campus de Ilha Solteira, Brazil] and stabilized in a commercial solution of ammonium hydroxide. The latex was centrifuged at 12,500 rpm for 20 min at 4 °C to remove impurities and other non-rubber constituents [21]. The latex was diluted in deionized water and the pH adjusted to 10 by using an ammonium hydroxide solution 1 M. The final solution was held at a low temperature (~5 °C) until being used. The NR solution,

prepared for the deposition process, was adjusted with a concentration of 1.0 g/L and pH around 7.0.

Film deposition and characterization

The film deposition was performed by the self-assembly technique on glass substrates, previously cleaned by the RCA method [22]. The first layer of the film was formed by immersing the substrate into the PANi (0.8 g/L) solution followed by the adsorption of the natural rubber particles; at each deposition step, the film was washed in deionized water (with the same pH as the previous solution) and dried with nitrogen pressured gas. Films with 10 bilayers were obtained by repeating the previous procedures.

The isosbestic points of the PANi were determined by UV-vis spectroscopy of the PANi/DMAc/H₂O solution at different pHs. The adsorption kinetics [23] were carried out by measuring the absorbance of the films at 464 nm (an isosbestic point). The amount of material deposited, Γ ($\mu\text{g}/\text{cm}^2$), in each step was estimated according to the Lambert-Beer law $A = c\epsilon b$, where ϵ (m^2/g) is the molar extinction coefficient, and c (g/L) and b (mm) are the solution concentration and path length, respectively [20].

UV-vis and Infrared spectroscopy (FTIR) analyses were performed using a Carry 50 spectrophotometer and a NEXUS 670 spectrophotometer, respectively. The FTIR spectra of pure PANi (KBr pellets containing about 2% wt. of PANi powder) and NR (casting the NR/toluol solution on the disk surface of KBr) were measured by using the transmittance method. The PANi/NR LbL thin film was deposited on a zinc selenite (ZnSe) substrate with 10 bilayers for the FTIR transmittance measurement.

The film's morphology was studied in a $10 \times 10 \mu\text{m}$ area with Atomic Force Microscopy (AFM), from Digital Instruments Nanoscope III, by using the intermittent mode, and also by Scanning Electron Microscopy with a field emission gun (FEG-SEM) from Zeiss supra 35, the images were obtained by using a SE detector, beam acceleration at 25 kV and magnification around 50 kX.

Tests as a potential chemical sensor

To determine the potential of the films as a chemical sensor, tests were performed by using PANi/NR (undoped) thin films with 10 bilayers deposited on a glass substrate with interdigitalized gold electrodes. The circuit employed in this study contained 50 pairs of electrodes; their dimensions were standardized at 10 μm of length, distanced by 10 μm and with 0.1 μm of thickness.

The sensitivity of the films as sensors was studied by monitoring changes in their capacitance after being immersed for 5 min in different analyte solutions that simulated different tastes. The tests were performed by using different analytes:

saccharose (sweet), caffeine (bitter), HCl (acid) and NaCl (salt). First, the thin film's response was measured in pure deionized water, as reference. Then the thin film's capacitance was measured in the frequency range of 0.01–10,000 kHz by using an impedance analyzer (Hewlett Packard 4192A). After being used in a specific analyte, the thin film was washed in deionized water for 5 min and dried with nitrogen before being used again.

Results and discussion

The spectra of PANi in the DMAc/H₂O (1:5) solution at different states of oxidation (or pHs) are shown in Fig. 1. Two absorption bands can be observed for pH 5.0 (320 and 600 nm), which are characteristic of PANi emeraldine base form [24]. For pH 2.5, the shoulder at 400 nm and the free carrier tail above 600 nm, related to the doping process, are assigned as polaron bands [24]. The free carrier tail is indicative of secondary doping effects, which result in a more extended chain conformation, allowing a more delocalized polaronic band [24]. The points that intersect the spectra are known as isosbestic points; in this case, there are 3 isosbestic points at 365, 464, and 673 nm. The isosbestic point at 464 nm was used to evaluate the amount of deposited material in the PANi/NR film in each self-assembled step.

The adsorption kinetics study was performed by monitoring the film's absorption at 464 nm in each step of the deposition. The amount of adsorbed material on the glass substrate was estimated by using Beer's law. The mechanism of adsorption for the PANi chains at lower pHs (under pH 4, doped PANi) is usually driven by electrostatic

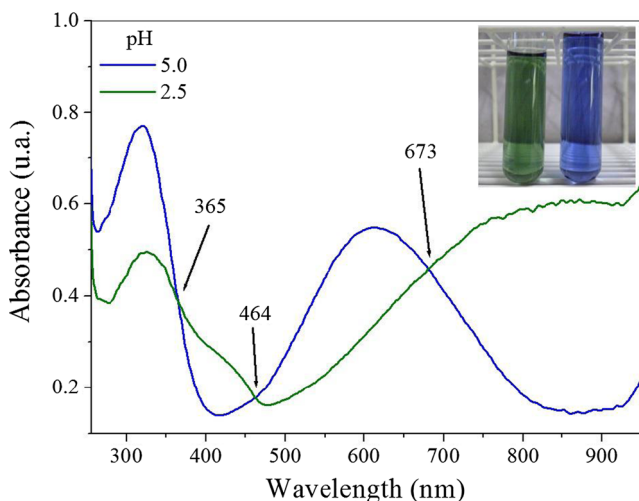


Fig. 1 UV-vis spectra of PANi/DMAc/H₂O at different pHs, as indicated. In the inset is a photograph of the PANi solution with pH 2.5 (green color) and pH 5.0 (blue color)

interactions, because the polymeric chains acquire positively charged sites; consequently, the polymeric chains exhibit a more elongated conformation due to the electrostatic repulsion between the charged sites [18, 25]. However, above pH 4, the undoped PANi does not exhibit charged sites; consequently, the polymeric chains are in a more compact conformation, and the adsorption mechanism can be driven by secondary interactions or even adsorbed as a neutral material [26–28]. Thus, the PANi pH solution is a decisive parameter that provides a charged or uncharged polymeric chain and modifies the mechanism parameters. On the other hand, the pH of the NR solution is directly related to the proteic membrane's stability, which is responsible for the spherical form of the NR particles and its negative charge characteristic, and at a lower pH can disrupt the proteic membrane, causing coagulation [16, 29, 30]. Therefore, evaluating the LbL film fabrications as functions of the pH solutions is an important step for producing films with better quality.

Figure 2 shows the results obtained for the adsorption kinetics of PANi's first layer, with the PANi concentration of 0.8 g/L and pHs of 2.5 and 5.0. The proper choice of these pHs is correlated to different mechanisms of adsorption; that could be related to charged and uncharged PANi chains, usually, electrostatic interaction (doped PANi – pH 2.5) and secondary interactions (undoped PANi – pH 5.0), respectively [18, 23, 26].

In Fig. 2, the adsorption process of PANi's first layers occurs at two different steps. The two-step process is in accordance with results obtained by Raposo et al. [23], the first one should correspond to the polymer molecules being driven toward the substrate due to electrostatic interaction, while the second one correspond to the rearrangement of polymer

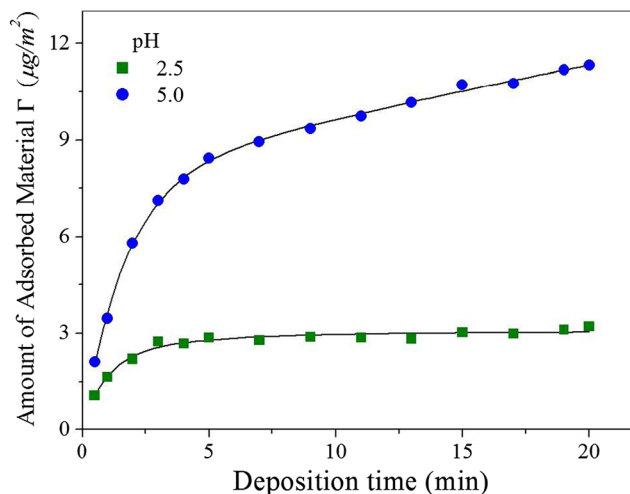


Fig. 2 Study of the adsorption kinetics of PANi's first layer, on a glass substrate, at different pHs

molecules on the substrate. The results can be fit if two exponential functions are superimposed, eq. (1).

$$A = k_1 \left(1 - e^{-\frac{t}{\tau_1}}\right) + k_2 \left(1 - e^{-\left(\frac{t}{\tau_2}\right)^n}\right) \quad (1)$$

where A is the absorbance, k_1 , k_2 and n are constants, and τ_1 and τ_2 are the characteristic times. Table 1 shows the parameters used for fitting the data of Fig. 2.

The first exponential corresponds to a first-order kinetics, with characteristic times (τ) around 1 - 2 s, the adsorption process in this case is relatively fast, occurring in the first few seconds of the immersion of the substrate within the PANi solution, indicating good interaction between the polymer and the substrate [25]. The second exponential can be associated by a so-called Johnson-Mehl-Avrami function with t^n in the exponential and with characteristic times (τ) around 200 - 350 s. As evidenced by the results, Fig. 2, in the doped PANi (2.5 pH) with a positively charged chain, the adsorption process occurs more quickly than the undoped PANi, and reaches the equilibrium plateau before 3 min of immersion; after that the adsorption process became self-limited by electrostatic repulsive interactions between the adsorbed chains and the chains in the solution, showing final amount of adsorbed material 2,94 $\mu\text{g}/\text{m}^2$ [23, 31]. Otherwise, the undoped PANi adsorption process is not self-limited and the amount of adsorbed material is much higher than in the doped PANi, showing final amount of adsorbed material 9,62 $\mu\text{g}/\text{m}^2$. In both cases, the adsorption process occurs faster, before 5 min, and usually the adsorption will cease due to the mechanical instability of the adsorbed layer [25, 31].

The initial tests of producing LbL films with PANi (5.0 pH) alternated with NR revealed that 5 min of immersion was not enough to adsorb a homogeneous layers. After further testing, we found that when we used 10 min of adsorption time in a single step with the same solution, linear growth was observed, Fig. 3. The LbL films produced with doped PANi (2.5 pH) showed linear growth with 3 min of immersion time; during the growth process, the water molecules can attached to the substrate surface, reducing the number of sites [25] for PANi/NR adsorption. The charged polymeric chain was able to overcome the diffusion of the water molecules and was more quickly adsorbed on the substrate surface, forming a homogenous monolayer that covered the entire surface before the

Table 1 Parameters used in eq. 1 for fitting the experimental data of Fig. 2

Ph	k_1	τ_1 (s)	k_2	τ_2 (s)	n
2,5	0,00275	1,0797	0,02817	224,48	1,76
5,0	0,00649	1,7057	0,02687	287,84	0,61

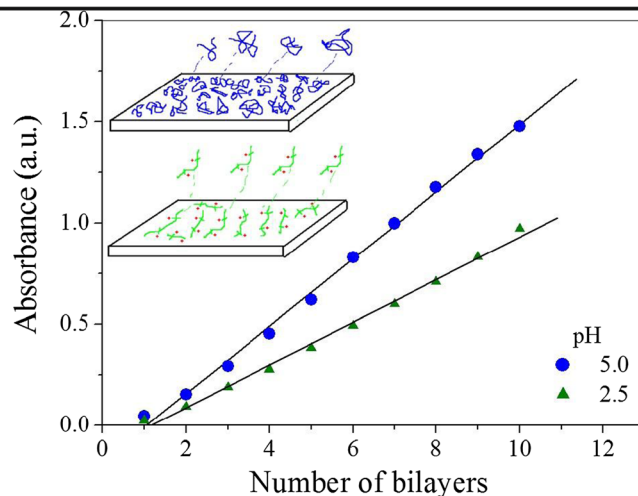


Fig. 3 Study of film PANi/NR growth at different pHs

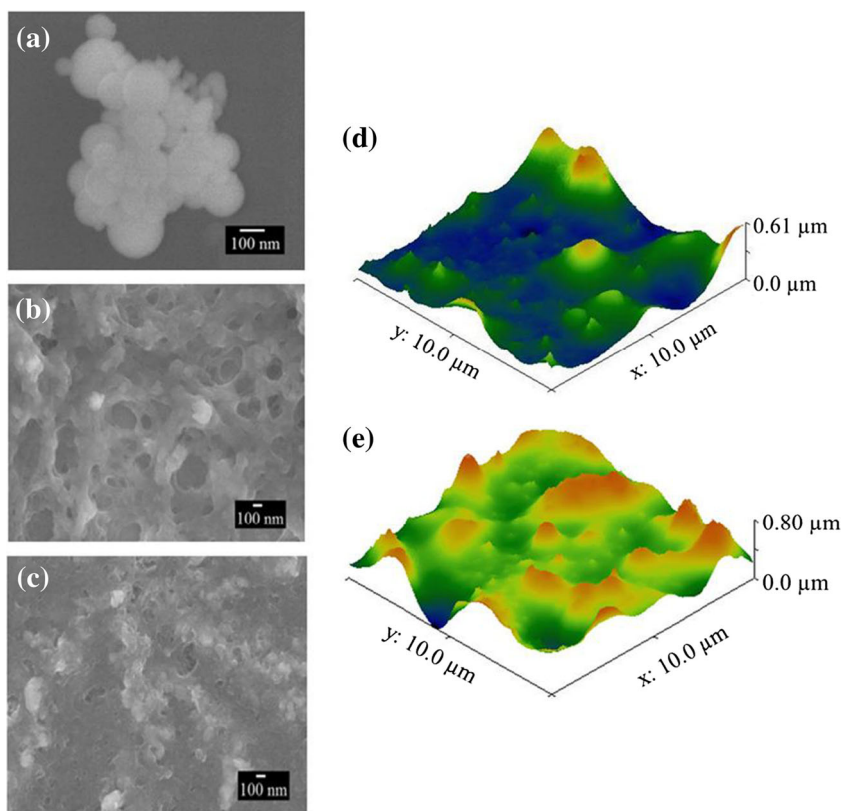
reactive sites on the substrate surface became inactive due to the attachment of water molecules.

The pH dependence in the material's adsorption was evidenced by our film growth studies, as shown in Fig. 3. The immersion time was standardized at 10 min for both pHs analyzed. The NR latex was employed with pH 7.0 and a concentration of 1.0 g/L. The amount of adsorbed material increased linearly with the number of bilayers for both LbL films, being higher at higher pHs. After 10 bilayers, some or even no additional material was adsorbed, probably due to the high amount of material present in the film. The system became self-limited by the mechanical instability of the layers or even electrostatic repulsion, as discussed by the literature [25, 31].

The amount of material adsorbed at 5.0 pH was much higher than that at 2.5 pH. The undoped PANi attach to the substrate as a neutral polymer and there is no electrical repulsion between the chains that exhibit a more compact coil conformation, which enables a dense film formation, and consequently a large amount of material will be deposited. The adsorption mechanism for PANi deposited at pH 2.5 is driven by electrostatic interaction, exhibiting an extended coil conformation [28, 32] and occupying more space, and due to the electrostatic repulsion between the chains, less material is necessary to cover the substrate surface, as illustrated in the scheme inset in Fig. 3.

The morphology of the natural rubber was investigated by using FEG-SEM microscopy. For that, highly diluted NR latex at pH 7.0 was cast on a silicon substrate, dried at room temperature and coated with a thin layer of gold. As shown in Fig. 4(a), we clearly see a spherical shape of NR with various diameter sizes. This result is in accordance with those found in the literature [29]. The monolayers of PANi self-assembled films were also investigated with undoped PANi, Fig. 4(b), and doped PANi, Fig. 4(c). The undoped PANi films were adsorbed in a compact conformation, with a huge porous and holes formed across the surface. The doped PANi showed

Fig. 4 FEG-SEM images and the respective chemical structures of (a) natural rubber particles deposited by casting; the first layer of (b) PANi undoped and (c) PANi doped deposited by self-assembly technique. AFM images of self-assembled LbL thin films PANi/NR with 10 bilayers (d) PANi pH 5.0 and (e) PANi pH 2.0



a more homogeneous structure, with few defects in a dense film formation.

The PANi/NR LbL thin films with 10 bilayers, obtained from PANi at pH 5.0 (Fig. 4(d)) and pH 2.5 (Fig. 4(e)), were characterized by AFM. A quite irregular surface with evidence of globular particles, probably due to NR particles, were observed in the films. The RMS roughness value was calculated in a single area of 100.0 μm² by Gwyddion software analysis. The results for the PANi/NR LbL films deposited at pH 5.0 and pH 2.5 showed RMS roughness around 87.7 nm and 107 nm, respectively.

The LbL film roughness decreased with the increasing amount of adsorbed material. The adsorption of a single layer of undoped PANi showed a large number of pores (Fig. 4(b)) compared to doped PANi (Fig. 4(c)). These pores can be filled with NR rubber particles, resulting in a LbL film with smaller roughness.

The FTIR spectra obtained for the PANi/NR LbL thin films and the cast films of undoped PANi and natural rubber are shown in Fig. 5 and the assignment of principal vibration modes are summarized in Table 2. The results obtained for PANi (Fig. 5(c)) and for natural rubber (Fig. 5(b)) are in accordance with the literature [30, 32]. The characteristic absorption peaks of PANi (1587, 1495, and 1305 cm⁻¹) and NR (2962, 2927, 2852, and 1450 cm⁻¹) can also be found in the spectra of the PANi/NR LbL thin films (Fig. 5(a)). The spectrum was a simple overlap of the spectrums of its constituents,

indicating that there was no chemical interaction between NR and PANi.

The LbL thin films were evaluated as chemical sensors by monitoring their electrical properties (capacitance vs.

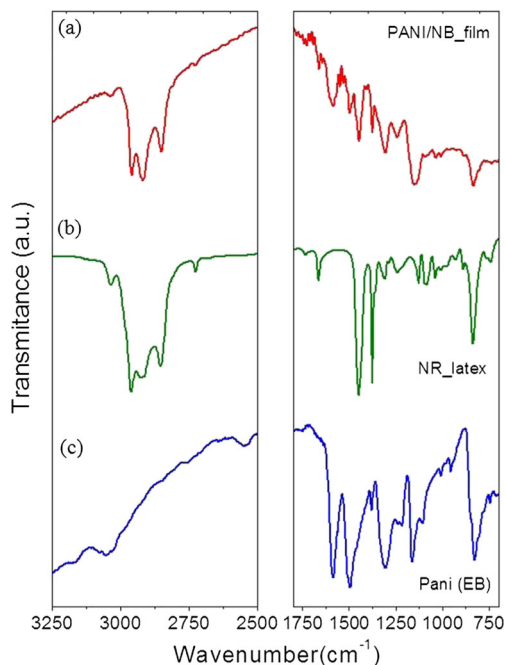


Fig. 5 FTIR spectra of (a) PANi (undoped)/NR LbL thin film with 10 bilayers, (b) natural rubber, and (c) undoped PANi

Table 2 Assignments of FTIR wavenumbers identified in the samples [30, 33]

Vibrational Assignment	Wavenumber (cm ⁻¹)		
	PAni	LbL film	NR
= CH stretching		3035	3033
C-H stretching of CH ₃		2962	2962
C-H stretching of CH ₂		2927	2927
C-H stretching of CH ₃ and CH ₂		2854	2855
N = Q = N stretching	1587	1585	
C = C stretching		1664	1664
N-B-N stretching	1495	1496	
C-H bending of CH ₂		1448	1450
C-H bending of CH ₃		1375	1375
N-H bending	1303	1305	
B-NH ⁺ -B or Q ^a = NH ⁺ -B	1162	1169	
C-H bending		1126	1127
C = CH wagging		836	837
C-H bending	829	829	

frequency) when immersed in different analytes; a typical response is shown in the inset of Fig. 6 (a). The observed results for capacitance values are shown for PAni (undoped)/NR LbL thin film with 10 bilayers in Fig. 6.

The most expressive changes were observed in the analytes with free ions in an aqueous solution, such as H⁺ and Cl⁻ in an HCl or NaCl solution. These ions must interact directly with the polyaniline chains by inducing conformational changes and the doping process. Otherwise, the caffeine and saccharose solutions must interact with the thin film through physical interactions, so detection must occur due to changes in the thin film's capacitance. The measurements also showed a good distinctions between the concentrations from 1 to 50 mM, Fig. 6(b), revealing a potential application as a chemical sensor for the LbL PAni/NR films.

Conclusion

The viability of processing self-assembly LbL films from natural rubber alternated with polyaniline on glass substrates by different mechanisms of adsorption was demonstrated. The growth of the films was successfully observed for 10 min of immersion and until 10 bilayers. The films fabricated with PAni doped and undoped presented similar physical-chemical characteristics, but their morphologies were quite different. The film's morphology was strongly influenced by the presence of NR particles and can be adjusted by changing the PAni solution's pH. Finally, the potential use of the film as a sensor was evaluated, and significant changes were observed in the film's response to different analytes at very low concentrations.

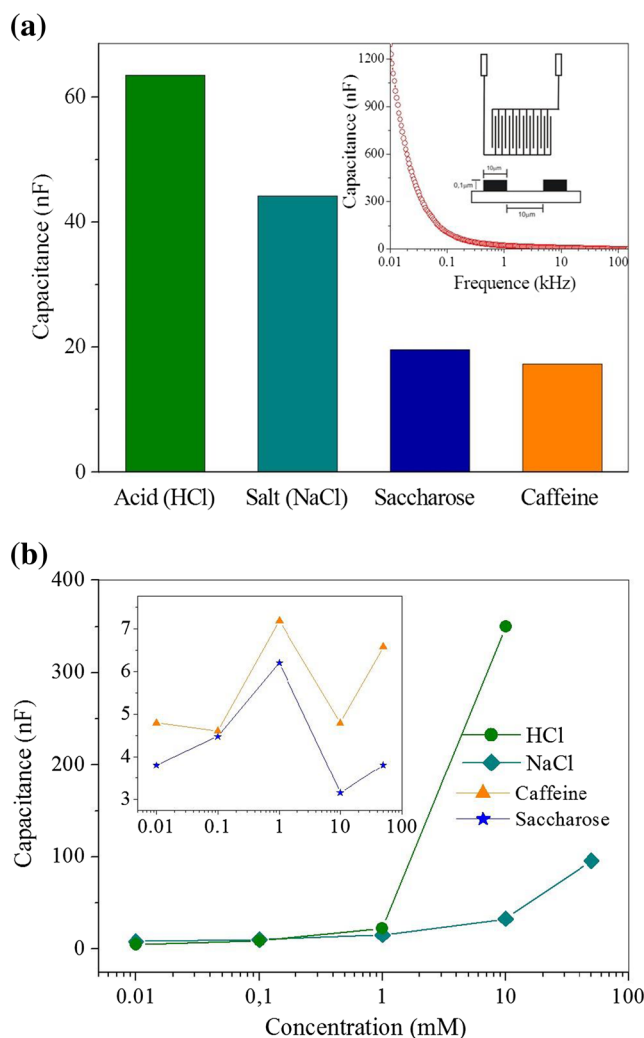


Fig. 6 Capacitance value measured at 1 kHz for different analytes with a solution concentration of 1 mM. The inset shows the capacitance measurement in HCl solution, and the electrode configuration

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