REVIEW PAPER



Studies on the Properties of Manganese Substituted Nickel Ferrite Nanoparticles

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Abstract Mn²⁺-substituted Ni ferrite nanoparticles were synthesized by sol-gel auto combustion method. The synthesized samples were annealed at 800 °C and characterization studies were carried out by XRD, VSM, electron paramagnetic resonance (EPR), field emission scanning electron microscopy (FE-SEM) and FT-IR spectroscopy. The XRD patterns revealed that Mn²⁺-substituted Ni ferrite crystallizes in cubic spinel phase and addition of α -Fe₂O₃ phase was also observed. The average crystallite sizes were found to be from 42 to 56 nm using a Scherer equation. The coercivity and remanent magnetization decreases when Mn^{2+} ion concentration is increased. The EPR spectrum shows the phase homogeneity of the samples. The FE-SEM images revealed that particles are both spherical in shape and particle sizes varied from 22 to 41 nm. The FT-IR spectrum confirmed the two main metal ion vibrations of nickel ferrite near 550 to 560 cm⁻¹ (A site) and 441 to 460 cm⁻¹ (B site).

Keywords Mn-Ni ferrite · Sol-gel combustion · Structural · Magnetic properties

1 Introduction

Spinel ferrites are extensively used for technological applications because of their special electrical and magnetic properties. Spinel ferrite compounds are used in the production of electronic sensors, memory devices, carriers of drug, magnetic resonance imaging devices and multilayer chip inductors [1, 2]. The magnetic properties of these ferrites can be changed by the substitutions of various kinds of metal ions (Mn²⁺) among divalent cations or by introducing a relatively small amount of rare-earth ions [3, 4]. The structural formula of ferrite is $A^{2+}B_2^{3+}O_4$, where A is a divalent metal ion (Ni²⁺, Zn²⁺, Mn²⁺, etc.) and B is a trivalent metal ion (Fe³⁺, Al³⁺, Sr³⁺, etc.) [5]. The properties of the ferrites can be tuned by varying the cation substitution and their distribution among tetrahedral site and octahedral site. In Mn²⁺-substituted NiFe₂O₄, the Ni²⁺ ions occupy octahedral (B) sites and Mn²⁺ ions occupy tetrahedral (A) and octahedral (B) sites. Ferrites containing Mn²⁺ ions have a tendency to form α -Fe₂O₃ phase when heat treated above 200 °C in air atmosphere [6]. The properties of ferrites are dependent on their structural parameters of particle size and shape, which can be modified in the synthesis processes. In spinel ferrites, the structural and magnetic properties are strongly dependent on cation distribution and method of preparation [7, 8]. Synthesis of ferrite materials includes coprecipitation method, solvothermal technique, ball milling method, hydrothermal processing, sol-gel method, etc. [9–13]. Among all the methods, the sol-gel combustion method is widely used for the synthesis of ferrite due to its advantages of preparation, composition flexibility, homogeneity, and low cost.

In the present work, the results obtained in the studies on $Mn_xNi_{1-x}Fe_2O_4$ (x = 0, 0.2, 0.4, 0.6) ferrite nanoparticles annealed at 800 °C are reported. Powder X-ray diffraction (XRD), vibrating sample magnetometer (VSM), electron paramagnetic resonance (EPR), field emission scanning electron microscopy (FE-SEM) and FT-IR spectroscopy techniques were employed for structural and magnetic characterizations. The structural and magnetic properties are discussed and compared with earlier reports.

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Fig. 1 XRD patterns of the $Mn_x Ni_{1-x} Fe_2 O_4$ samples (x = 0, 0.2, 0.4, 0.6)

2 Material and Methods

2.1 Synthesis of Mn²⁺-Substituted Ni Ferrite Nanoparticles

Mn²⁺-substituted Ni ferrites of chemical composition $Mn_xNi_{1-x}Fe_2O_4$ samples (x = 0, 0.2, 0.4, 0.6) were synthesized by a sol-gel combustion method at room temperature using analytical reagent grade iron nitrate Fe(NO₃)₃.9H₂O, nickel nitrate (Ni(NO₃)₂.6H₂O), manganese nitrate (Mn(NO₃)₂.6H₂O), tartaric acid and ammonia (NH₃) solution. An appropriate amount of each material in stoichiometric ratio was dissolved in de-ionized water. The mixed nitrate solution was heat treated at 60 °C the solution became dehydrated and then the gel formation took place. The gel was dried in the hot air oven at 120 °C for 8 h. The final powder was annealed at 800 °C using a muffle furnace at a heating rate of 5 °C/min. The structural and phase analyses of synthesized samples were done by XRD technique using an X'Pert Pro Pan analytical X-ray diffractometer operated at 45 kV and 30 mA, Cu Ka, wavelength 1.5406 Å. Room-temperature magnetization studies were performed using a VSM (Lake Shore model 7404) operated at a field of 15 kOe. The g value and peak-topeak line width (ΔH_{pp}) of the samples were calculated at room temperature using a Bruker BioSpin EMX Plus EPR spectrometer operated at 100 kHz modulation frequency and 9.86 GHz microwave frequency. The morphology and particle size measurements were characterized by FE-SEM (model JEOL/JSM 6701F). FT-IR spectra were recorded by a Perkin Elmer FT-IR spectrometer using KBr pellets, in the range of 4000–400 cm⁻¹.

3 Results and Discussion

3.1 Structure and Phase Analysis of Mn²⁺-Substituted Ni Ferrite Nanoparticles

The XRD patterns of Mn substituted Ni ferrite samples are shown in Fig. 1. The XRD pattern revealed that all the diffracted peaks are related to cubic spinel phase and the addition of small peaks as secondary phase (α -Fe₂O₃) is also observed in the Mn²⁺-substituted Ni ferrite samples. The prominent (*hkl*) planes (111), (220), (311), (400), (422), (511), and (440) are indexed in the pattern. The intensity of these peaks increases with an increase in the concentration of Mn²⁺ ion. The XRD patterns of Mn-Ni ferrite samples match with the Joint Committee of Powder Diffraction Standards (JCPDS) (Card No.: 75-0894) and all the experimental XRD planes perfectly match with earlier reported studies [14].

The average crystallite size has been calculated using the Scherer equation

$$D = \frac{0.9\lambda}{\beta\cos\theta}$$

where *D* is the average crystallite size, λ is the wavelength of the X-ray radiation used ($\lambda = 1.5406$ Å), β denotes the full width at half maximum (FWHM) measured in radians and θ is the Bragg angle [15]. The average crystallite size of synthesized samples is found to be from 42.4 to 56.7 nm. The crystallite size is slightly increased with an increasing level of Mn²⁺ concentration. The lattice constant (*a*) of synthesized ferrite samples was calculated by using the Bragg equation

$$d = \frac{a}{\sqrt{h^2 + k^2 + l^2}}$$

Table 1Structural propertiesof the Mn substituted Ni ferritesamples

Lattice constant (a) (Å) Volume (Å³) X-ray density (g/cm³) Crystallite size Samples (x)XRD (nm) FE-SEM (nm) NiFe₂O₄ 8.355 603 5.255 42.47 22 - 30600 42.48 22-32 Mn_{0.2}Ni_{0.8}Fe₂O₄ 8.473 5.069 608 42.49 23-33 Mn_{0.4}Ni_{0.6}Fe₂O₄ 8.439 5.169 Mn_{0.6}Ni_{0.4}Fe₂O₄ 8.451 583 5.193 56.71 22-41



Fig. 2 M-H plots of the Mn_xNi_{1-x}Fe₂O₄ samples (x = 0, 0.2, 0.4, 0.6)

where *d* is the inter-planar distance and *h*, *k*, and *l* are the Miller indices. The lattice constant values of synthesized Mn²⁺-substituted Ni ferrite samples are presented in Table 1. The present results indicate that Mn²⁺-substituted Ni ferrite samples have slightly higher lattice constant values. This may be due to the replacement of smaller ionic radius Ni²⁺ (0.69 Å) ion by larger ionic radius Mn²⁺ (0.83 Å) ion. This causes the lattice expansion. The X-ray density (ρ_x) was calculated using the equation

$$\rho_x = \frac{8M}{Na^3}$$

where 8 represents the number of atoms in a unit cell of spinel lattice, M is the molecular weight of the Mn substituted Ni ferrite samples, N is Avogadro's number (6.02252 $\times 10^{26}$ kmol⁻¹) and a is the lattice constant. The X-ray densities of the synthesized samples are found to be from 5.19 to 5.25 g/cm³. The X-ray density values were found to decrease with an increase in Mn²⁺ concentration. All structural parameters of Mn²⁺-substituted Ni ferrite samples are calculated and listed in Table 1. The results are in good



Fig. 3 M-H enlarged plots of the Mn_xNi_{1-x}Fe₂O₄ samples (x = 0, 0.2, 0.4, 0.6)

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 Table 2
 Magnetic properties of the Mn-substituted Ni ferrite samples

Composition (X)	Samples (X)	$H_{\rm c}$ (Oe)	$M_{\rm r}$ (emu/g)
0	NiFe ₂ O ₄	200	0.100
0.2	Mn _{0.2} Ni _{0.8} Fe ₂ O ₄	110	0.075
0.4	Mn _{0.4} Ni _{0.6} Fe ₂ O ₄	70	0.050
0.6	$Mn_{0.6}Ni_{0.4}Fe_2O_4$	60	0.025

agreement with the result reported by Tirupanyam et al. [16] and Hassan et al. [17].

3.2 VSM Analysis of Mn²⁺-Substituted Ni Ferrite Nanoparticles

Analysis of the magnetic behavior of the Mn_xNi_{1-x} ferrite samples was carried out using a vibrating sample magnetometer. Variation of magnetization (M, emu/g) with the applied field (H, Oe) at room temperature is shown in Figs. 2 and 3. The magnetic properties such as coercivity (H_c) and remanent magnetization (M_r) values are measured from the plotted hysteresis (M-H) loops and are listed in Table 2. At room temperature, the very narrow loops for all the samples indicate the soft magnetic nature [18, 19]. Nonsaturation of magnetization even at an applied field of 12.5 kOe reveals the presence of single-domain nanoparticles in superparamagnetic state. The coercivity of Ni ferrite is 200 Oe. The addition of Mn^{2+} ions decreases the coercivity. The decrease of H_c is due to the decrease of Ni²⁺ content. The slight changes in coercivity due to an increase of Mn²⁺ ions may be attributed to domain structure, critical diameter, lattice strain, and shape anisotropy of crystal. The results are in good agreement with the results reported in the literature [20-23].



Fig. 4 EPR spectra of the $Mn_x Ni_{1-x} Fe_2 O_4$ samples (x = 0.2, 0.4, 0.6)

Composition Samples Resonance g value Line width, field, $H_{\rm r}$ (G) $\Delta H_{\rm pp}$ (G) (X)1950 0.2 Mn_{0.2}Ni_{0.8}Fe₂O₄ 2250 3.131 0.4 Mn_{0.4}Ni_{0.6}Fe₂O₄ 2500 2.817 2500 0.6 Mn_{0.6}Ni_{0.4}Fe₂O₄ 3050 2.012 500

Table 3 EPR parameters of the Mn substituted Ni ferrite samples

3.3 EPR Spectra Analysis

Powder EPR spectral measurement is an essential investigation for the magnetic materials because at high frequencies the resonance occurs due to the interaction between electron spin and electromagnetic waves. The EPR spectra of synthesized Mn^{2+} -substituted Ni ferrite samples were recorded at room temperature and are shown in Fig. 4. For all the samples, the obtained spectra show single and broad resonance peaks. This reveals that isolated Fe³⁺, Ni²⁺, and Mn²⁺ ions do not exist and also reveals the phase homogeneity of the prepared samples [24]. The spectrum was analyzed to obtain values of the magnetic field at resonance position corresponding to zero signals (H_r), peak-to-peak line width (ΔH_{pp}) , and effective g value. The obtained values are presented in Table 3.

The g value was calculated using

$$g = \frac{h\upsilon}{\beta H_{\rm r}}$$

where *h* is Planck's constant (6.6234 × 10⁻³⁴ Js), υ is the microwave frequency (9.86 GHz), β is the Bohr magneton (9.274 × 10⁻²⁴ Am²) and *H*_r is the magnetic field at resonance. The observed values of resonance field (*H*_r) and *g* factor are in general dependent on the magnetic dipole interactions among nanoparticles and the super-exchange interaction between Ni²⁺, Mn²⁺, and Fe³⁺ ions through nonmagnetic O²⁻ ions [25, 26].

3.4 FE-SEM Analysis of Mn²⁺-Substituted Ni Ferrite Nanoparticles

Figure 5 shows the FE-SEM micrographs of synthesized $Mn_xNi_{1-x}Fe_2O_4$ nanoparticles. As seen from FE-SEM images, the samples consist of mostly spherical-shaped particles. FE-SEM images show that there is development of grain structure with annealing temperature.



Fig. 5 FE-SEM images of the $Mn_x Ni_{1-x}Fe_2O_4$ samples (x = 0, 0.2, 0.4, 0.6)



Fig. 6 FT-IR spectrum of the $Mn_xNi_{1-x}Fe_2O_4$ samples (x = 0, 0.2, 0.4, 0.6)

Mn²⁺-substituted Ni ferrite synthesized and annealed at 800 °C composed of agglomerated spherical particles.

Increasing the Mn^{2+} ion concentration, the powders show irregular microstructures with small spherical particles and size of the particle ranging from 22 to 41 nm [27]. Generally, the grain size increases with an increase in Mn^{2+} ion concentration. The particles are spherical in shape, and the size and structural morphology of the particles varied in different levels of Mn compositions.

3.5 FT-IR Spectroscopy Analysis

The FT-IR spectra of synthesized Mn substituted Ni ferrite samples are shown in Fig. 6. The FT-IR spectra revealed that the region of the spectrum, centered around at 3411 and 3408 cm⁻¹ represents the stretching vibration of a hydroxyl group of water molecules (–OH) present in the samples. The frequencies around 1592 and 1405 cm⁻¹ are assignable as a metal ferrite (M/Fe) complex present in the samples.

The two main metal-oxygen vibrations are observed at 550 to 560 cm⁻¹ and at 449 to 480 cm⁻¹ due to tetrahedral and octahedral sites, respectively. The band near 550 to 560 cm⁻¹ corresponds to an intrinsic stretching vibration of metal of the tetrahedral site (metal oxide), whereas the bands at 449 to 480 cm⁻¹ are assigned to octahedral sites (Fe–O). The results are in good agreement with the result reported by Koseoglu et al. [28].

4 Conclusion

Manganese substituted nickel ferrite nanoparticles were synthesized at room temperature by a sol-gel auto combustion method. The XRD pattern revealed that the Mn substituted Ni ferrite has a cubic spinel structure. The VSM analysis revealed that the Mn substituted Ni ferrite nanoparticles were in superparamagnetic state at room temperature. The coercivity and remanent magnetization values were found to be decreased. The *g* value decreases with an increase in Mn^{2+} ion concentration. The surface morphology of the ferrite samples is spherically shaped and size of the ferrites was found to be from 22 to 41 nm. The FT-IR spectrum confirmed that two main metal ion vibrations are observed in the spectrum. It is concluded that substituted manganese ion strongly affects the magnetic and structural properties of nickel ferrite.

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