

# Preparation and Magnetic Properties of La-Substituted Strontium Hexaferrite by Microwave-Assisted Sol-Gel Method

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**Abstract** Sr<sub>1-x</sub>La<sub>x</sub>Fe<sub>12</sub>O<sub>19</sub> ( $x = 0, 0.15, 0.25, 0.5$ ) hexaferrites were prepared by microwave-assisted sol-gel method. The thermal decomposition process, structural, and magnetic properties of the products were studied by thermal differential scanning calorimeter (DSC), thermogravimetry (TG), X-ray diffraction (XRD), and vibrating sample magnetometer (VSM). The phase of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> appeared at  $x = 0.25$  and  $x = 0.5$ . The coercivity of La<sup>3+</sup>-substituted strontium hexaferrites is improved to 5960.2 Oe at  $x = 0.25$ .

**Keywords** Microwave-assisted calcination · M-strontium hexaferrites · Sol-gel · La doping

## 1 Introduction

M-strontium hexaferrites have been widely used in communication, electromagnetic wave absorbers, and magnetic recording medium for their good chemical stability, high magnetization, excellent corrosion resistivity, and low manufacturing cost [1–5]. The magnetization and intrinsic magnetic properties of SrFe<sub>12</sub>O<sub>19</sub> hexaferrites result from the unfinished offset of Fe<sup>3+</sup> magnetic moment and the partial

substitution for Sr<sup>2+</sup> sites in different crystallographic sites, respectively [6, 7]. The rare earth elements, such as La<sup>3+</sup> [6, 7], usually were used to replace the Sr<sup>2+</sup> of crystallographic sites to modify the coercivity. Many previous studies focused on the improvement of the magnetic characteristics by the preparation of thin films and submicron particles [8–10]. Sol-gel auto-combustion is an advanced preparation method of ultrafine power because of cheap precursor, low reaction temperature, and accurate control of material composition. Moreover, the process of calcination has great effects on magnetic properties. Microwave-assisted synthesis has been paid more attention since the first discussion in 1986 [11–14]. The heating energy in the process of microwave-assisted calcination derives from the interaction between materials and microwave, instead of originating from external sources, which can improve the diffusion coefficient, reduce calcination temperature, shorten reaction time, and endow materials with some new characteristics [15]. In this work, the structural, magnetic properties of La-substituted strontium hexaferrites have been studied by employing microwave-assisted sol-gel method.

## 2 Experimental

The analytically pure of Sr(NO<sub>3</sub>)<sub>2</sub> · 3H<sub>2</sub>O, La(NO<sub>3</sub>)<sub>3</sub> · 6H<sub>2</sub>O, Fe(NO<sub>3</sub>)<sub>3</sub> · 9H<sub>2</sub>O and citric acid with stoichiometric amounts were dissolved in a certain amount of deionized water. After the solution was stirred homogeneously at 80 °C, the pH value was adjusted to 6 through aqua ammonia solution. The heating and stirring process continued until it formed a viscous dark-green sol. Subsequently, the sol was dried at 120 °C to be brown gel. The self-combustion process took place at 175 °C and the gray fluffy precursor formed. The precursor was calcined at 800 °C

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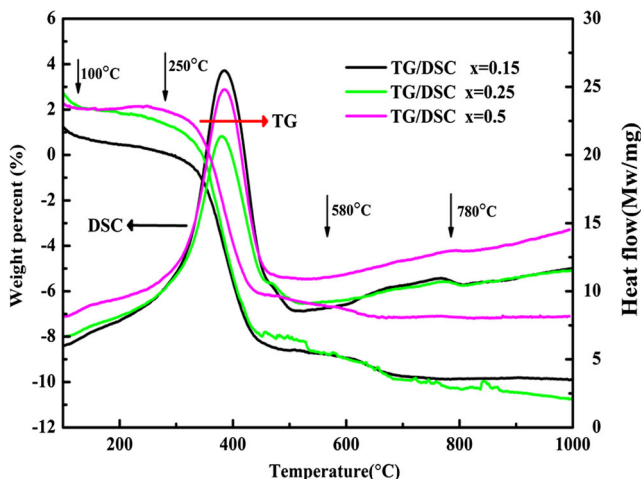
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for 80 min in microwave furnace with a heating rate of 40 °C/min. The Setaram labsys TG/DSC was applied to detect the phase transformation of precursor with a heating rate 20 °C/min. The crystal structure of calcined samples was detected by D/max-2000 X-ray diffraction with Cu Ka radiation. The phase volume percent was calculated on the basis of the highest diffraction peaks intensity of the different phases from the XRD patterns. The magnetic properties were measured via the vibrating sample magnetometer (VSM) with a maximum applied magnetic field of 1.8 T at room temperature.

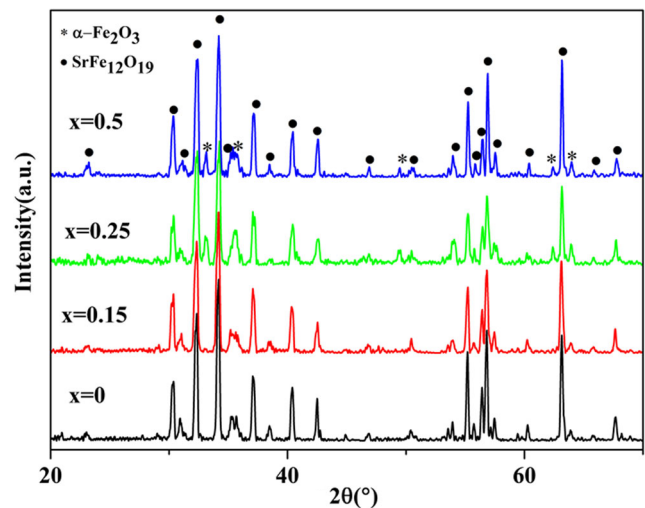
### 3 Results and Discussion

Figure 1 illustrates the TG/DSC curves of the  $\text{Sr}_{1-x}\text{La}_x\text{Fe}_{12}\text{O}_{19}$  ( $x = 0.15, 0.25, 0.5$ ) precursors. The weight loss appears at 100 °C deriving from the evaporation of free water and bound water in the precursors. A precipitous exothermic peak is observed from 250 to 425 °C, which can be attributed to the citrates decomposition and the formation of metal oxides. Weight loss also occurs from 580 to 800 °C, accompanied with a minor exothermic peak at about 780 °C, which suggests the formation of M-strontium ferrite [16, 17]. Those curves show that the substitution of  $\text{La}^{3+}$  almost has no effect on the phase transition temperature of the precursors. According to our previous study [16, 18], the precursors are calcined at 800 °C for 80 min in microwave furnace.

Figure 2 shows the XRD patterns of  $\text{Sr}_{1-x}\text{La}_x\text{Fe}_{12}\text{O}_{19}$  ( $x = 0, 0.15, 0.25, 0.5$ ) powders with microwave-assisted calcination at 800 °C for 80 min. A strong diffraction peak of  $\text{SrFe}_{12}\text{O}_{19}$  phase is detected in all the samples. The single  $\text{SrFe}_{12}\text{O}_{19}$  hexaferrite has been identified with the  $\text{La}^{3+}$



**Fig. 1** TG/DSC curves of  $\text{Sr}_{1-x}\text{La}_x\text{Fe}_{12}\text{O}_{19}$  ( $x = 0.15, 0.25, 0.5$ ) precursors



**Fig. 2** X-ray diffraction patterns of the  $\text{Sr}_{1-x}\text{La}_x\text{Fe}_{12}\text{O}_{19}$  ( $x = 0, 0.15, 0.25, 0.5$ ) powders

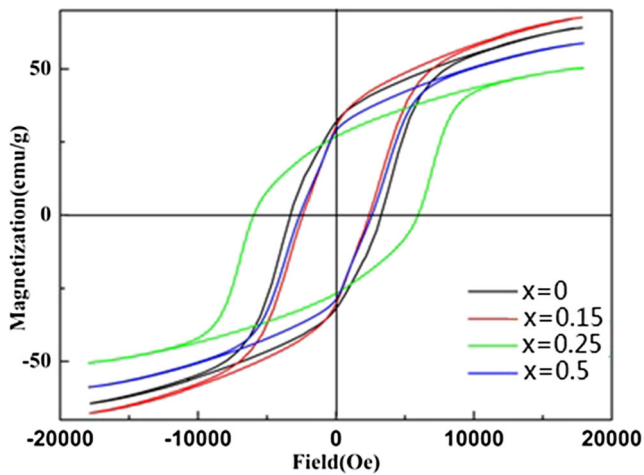
substitution at  $x = 0.15$ . The diffraction peaks of  $\alpha\text{-Fe}_2\text{O}_3$  phases are observed in the samples with  $x = 0.25$  and  $0.5$ , respectively. Comparing the ionic radius of  $\text{Sr}^{2+}$  (1.27 Å) with  $\text{La}^{3+}$  (1.22 Å), the  $\text{Sr}^{2+}$  substitution by smaller ionic radius  $\text{La}^{3+}$  could reduce the cell volume. Table 1 lists the lattice constants of  $\text{Sr}_{1-x}\text{La}_x\text{Fe}_{12}\text{O}_{19}$  ( $x = 0, 0.15, 0.25, 0.5$ ) and the volume percentage of  $\alpha\text{-Fe}_2\text{O}_3$ . The lattice constant  $a$  and  $c$  are calculated by Formula (1), where  $d_{hkl}$  represents the crystal distance and  $(h k l)$  is the Miller indices.

$$d_{(hkl)} = \left( \frac{4}{3} \frac{h^2 + hk + k^2}{a^2} + \frac{l^2}{c^2} \right)^{-\frac{1}{2}} \quad (1)$$

It indicates that the lattice constant  $c$  of strontium ferrites unsubstituted with  $\text{La}^{3+}$  is bigger than that of substituted samples, because the smaller ionic radius of  $\text{La}^{3+}$  would reduce the distance along  $c$ -axis of the stacking fault, while the lattice constant  $a$  has a different change. The Mössbauer spectra analysis has figured out that the substitution of  $\text{Sr}^{2+}$  by  $\text{La}^{3+}$  is accompanied with the change of  $\text{Fe}^{3+}$  (0.64 Å) to  $\text{Fe}^{2+}$  (0.75 Å) at  $2a$  or  $4f_2$  crystallographic sites to keep the electrical neutrality of strontium hexaferrite [19]. The  $\text{Fe}^{2+}$  ions have a large ionic radius than  $\text{Fe}^{3+}$  ions, which results in the enlargement of the lattice constant  $a$ . The volume percentage of  $\alpha\text{-Fe}_2\text{O}_3$  decrease to 10.5 % after reaching to

**Table 1** Lattice constants of  $\text{Sr}_{1-x}\text{La}_x\text{Fe}_{12}\text{O}_{19}$  ( $x = 0, 0.15, 0.25, 0.5$ ) and the volume percentage of  $\alpha\text{-Fe}_2\text{O}_3$

$x$	0	0.15	0.25	0.5
$a$ (Å)	5.883	5.882	5.872	5.879
$c$ (Å)	23.065	23.055	23.012	22.964
$\alpha\text{-Fe}_2\text{O}_3$	–	–	21.1 %	10.5 %



**Fig. 3** The hysteresis loops of  $\text{Sr}_{1-x}\text{La}_x\text{Fe}_{12}\text{O}_{19}$  ( $x = 0, 0.15, 0.25, 0.5$ ) powders at room temperature

21.1 % at  $x = 0.25$ , which is different from the substitution by  $\text{Pr}^{3+}$  and  $\text{Dy}^{3+}$  [16].

Figure 3 shows the hysteresis loops of  $\text{Sr}_{1-x}\text{La}_x\text{Fe}_{12}\text{O}_{19}$  ( $x = 0, 0.15, 0.25, 0.5$ ) powders at room temperature. The saturation magnetization ( $\sigma_s$ ) increases slightly with the  $\text{La}^{3+}$  doping content improvement, accompanied with the slight decrease in coercivity. As reported, some reasons have been discussed for the effects of the  $\text{La}^3$  substitution. When a divalent  $\text{Sr}^{2+}$  is replaced by a trivalent  $\text{La}^{3+}$ ,  $\text{Fe}^3$  would be converted to  $\text{Fe}^{2+}$  to maintain the electrical neutrality. The emergence of  $\text{Fe}^{2+}$  could cause the decrease of net magnetic moment in  $\text{Sr}_{1-x}\text{La}_x\text{Fe}_{12}\text{O}_{19}$  for the valence change of  $\text{Fe}^{3+}$  to  $\text{Fe}^{2+}$ , the spin canting deviating from the collinear to non-collinear arrangement, and the weaker super-exchange of  $\text{Fe}^{3+}-\text{O}^{2-}-\text{Fe}^{2+}$  [6, 20]. Those combined effects could cause the degradation in  $\sigma_s$ . The substitution of  $\text{Sr}^{2+}$  by  $\text{La}^{3+}$  with smaller ionic radius leads to the reduction of lattice constant  $c$  [20, 21], as shown in Table 1, which causes the decrease of the  $\text{Fe}-\text{O}$  distance that is parallel to the  $c$ -axis. The superexchange interaction of  $\text{Fe}^{3+}-\text{O}^{2-}-\text{Fe}^{3+}$  increases at the  $12k$  and  $2b$  site, corresponding with the improvement of  $\sigma_s$ . For the sample with  $\text{La}^{3+}$  substitution at  $x = 0.15$ , the stronger superexchange interaction of  $\text{Fe}^{3+}-\text{O}^{2-}-\text{Fe}^3$  could be dominant factor for the slight increase of  $\sigma_s$ . As listed in Table 2, the ratio of  $\sigma_r/\sigma_s$  is 0.447 with  $\text{La}^{3+}$  substitution at  $x = 0.15$ , showing its magnetic multi-domain structure [22]. The magnetization of samples with multi-domain structure is mainly

**Table 2** The grain size and magnetic properties of  $\text{Sr}_{1-x}\text{La}_x\text{Fe}_{12}\text{O}_{19}$  ( $x = 0, 0.15, 0.25, 0.5$ )

$x$	0	0.15	0.25	0.5
Grain size (nm)	38.3	31.7	29.5	34.2
$\sigma_r/\sigma_s$	0.497	0.447	0.532	0.493

determined by the domain wall displacement. The movement of multi-domain walls is much easier than flipping the magnetization of an entire particle coherently which is the magnetization mechanism of single domain, so the multi-domain grain has lower coercivity. For the sample at  $x = 0.25$ , the  $\sigma_s$  reduces to 50.449 emu/g. The decrease of  $\sigma_s$  and  $\sigma_r$  could be attributed to the presence of the  $\alpha\text{-Fe}_2\text{O}_3$  phase. The content of  $\alpha\text{-Fe}_2\text{O}_3$  is about 21.1 % and it dilutes the magnetization values. The ratio of  $\sigma_r/\sigma_s$  is 0.532 and a remanence enhancement effect is observed. The exchange coupling effects take place between the hard magnetic  $\text{SrFe}_{12}\text{O}_{19}$  phase and soft magnetic  $\alpha\text{-Fe}_2\text{O}_3$  phase and the smooth hysteresis loop also verify the effect. This coupling could hide the domain rotation and block the displacement of domain walls. The grain size calculated by scherrer formula at  $x = 0.25$  for  $\text{La}^{3+}$  substitution has only 29.5 nm, resulting in higher coercivity. The ratio of  $\sigma_r/\sigma_s$  is 0.493 at  $x = 0.5$  and the grain has no interaction when the value ratio of  $\sigma_r/\sigma_s$  below the 0.5. The  $\sigma_s$  and  $\sigma_r$  increased at  $x = 0.5$  because of the lower phase composition of  $\alpha\text{-Fe}_2\text{O}_3$  (Table 1).

### 4 Conclusion

$\text{Sr}_{1-x}\text{La}_x\text{Fe}_{12}\text{O}_{19}$  ( $x = 0, 0.15, 0.25, 0.5$ ) hexaferrites were prepared by microwave-assisted sol-gel method. The grain size of strontium ferrite reduces when the  $\text{La}^{3+}$  doping content below  $x = 0.25$ . The substitution of  $\text{La}^{3+}$  can remarkably improve the coercivity to 5960.2 Oe at  $x = 0.25$ . This derived from the fine grain and exchange coupling between the higher volume percentage of  $\alpha\text{-Fe}_2\text{O}_3$  and the strontium ferrite with signal domain.

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