ORIGINAL PAPER

# **One-Step Hydrothermal-Electrochemical Route to Carbon-Stabilized** *γ* **-Fe<sub>2</sub>O<sub>3</sub> Powders**

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**Abstract** Carbon coated maghemite ( $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>) particles with nanoscale sizes were synthesized by an inexpensive and environmental friendly hydrothermal electrochemical method in a one-step process. Glucose and ferric citrate were used as the carbon and iron source, respectively. Transmission electron spectroscopy (TEM) analysis indicated that a carbon layer was coated on the surfaces of the individual  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles prepared at 180 °C. The composition and phase structure of as-prepared materials were characterized by Raman and Fourier transform infrared spectroscopy (IR). Electromagnetic properties of the carbon/maghemite complex materials were measured using vibration sample magnetometer (VSM). The saturation of as prepared  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>/C nanocomposition was 31.2 emu/g.

**Keywords** Carbon · *γ* -Fe2O3 · Complex · Hydrothermal electrochemical method

## **1 Introduction**

Nanoscale magnetic iron oxides offer a high potential application in several areas such as electronics, optoelectronics, medicine, magnetic storage, and biotechnologies [\[1](#page-3-0)[–4\]](#page-3-1). Nanosized materials are known to take on peculiar properties compared to the bulk material. For instance, iron

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oxide magnetic nanoparticles have received considerable attention concerning the cleanup of environmental contaminants because of their small particle size, high surface area, catalytic activity, low cost, and ease of preparation [\[5\]](#page-3-2). However, iron nanoparticles are sometimes unstable, and their stabilization particularly in terms of aggregation and oxidation in air is a crucial point to be solved. These deficiencies could be avoided by coating ferromagnetic nanoparticles with different stable and nonmagnetic materials [\[6–](#page-3-3)[10\]](#page-3-4). Among them, carbon materials have attracted more interest than other materials due to the high electrical conductivity, low cost, broad chemical, and physical stability [\[9,](#page-3-5) [10\]](#page-3-4). Various techniques have been developed for synthesis of such nanocomposite structures, including arc techniques [\[9\]](#page-3-5), laser pyrolysis method [\[11\]](#page-3-6), catalytic chemical vapor deposition (CVD) [\[12\]](#page-3-7), and magnetron and ion-beam sputtering [\[13\]](#page-3-8). However, one-step method has not been reported for the synthesis of  $C/\gamma$ -Fe<sub>2</sub>O<sub>3</sub> complex materials from solution so far.

In this paper, we reported the preparation of  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> core carbon shell obtained by hydrothermal electrochemical method in one step. *γ* -Fe<sub>2</sub>O<sub>3</sub>/C materials were synthesized under mild aqueous conditions. The  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>/C obtained this way could be dispersed well in aqueous solution, and there were functional groups on its surface, which facilitated further modification in future applications.

# **2 Experimental**

Ta substrates with  $10 \times 10 \times 0.1$  mm<sup>3</sup> dimensions of 99.9 % purity were mechanically polished and degreased with acetone using an ultrasonic cleaner. Ferric citrate and glucose were of reagent grade and were used without any further purification. To form a transparent ferric citrate solution,

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1.22 g ferric citrate (FeC $_6$ H<sub>5</sub>O<sub>7</sub>, 0.005 mol) was dissolved in 100 mL distilled water Then, 1.0 g glucose  $(C_6H_{12}O_6,$ 0.005 mol) was added into the solution and stirred.

The detailed description of the deposition facility and the growth method has been given elsewhere [\[14\]](#page-3-9). A typical preparation process was carried out under galvanostatic conditions with a constant current density 0.001 mA/cm2 for 20 h. And the temperature of bath was maintained at 180 ◦C or 170 ◦C. After each experiment, black powders were obtained. The powders were washed with water, ultrasonically in ethanol, and air-dried prior to characterization.

Raman (Labram HR 800, Jobin-Yvon) and IR spectroscopy (Nicolet Nexus 670) were employed to characterize the structure and bond parameters. The morphology of the samples was examined by TEM (JEM-1200EX). And the magnetic properties (*M*–*H* curve) were measured using VSM (HHZ15) at room temperature.

# **3 Results and Discussion**

#### 3.1 FTIR Spectroscopy

Figure [1a](#page-1-0), b showed the FTIR spectra of as-prepared samples prepared at 180 °C (sample A) and 170 °C (sample B), respectively. FTIR spectra revealed the coexistence of *γ* - Fe<sub>2</sub>O<sub>3</sub> and carbon, in which 580 and 630 cm<sup>-1</sup> was assigned to maghemite [\[15\]](#page-3-10), and 2927, 2854, 1627, 1562, 1387 and 1113 cm<sup>-1</sup> to carbon [\[16–](#page-3-11)[21\]](#page-3-12). The absorption bands at 2927, 2854, and 1387 cm<sup>-1</sup> were corresponding to hydrogen bonded  $sp^3$  carbon [\[16,](#page-3-11) [17\]](#page-3-13), 1627 cm<sup>-1</sup> to C=O bonds [\[18\]](#page-3-14), 1562 cm<sup>-1</sup> to C=C bonds [\[19\]](#page-3-15), and 1113 cm<sup>-1</sup> to C=C–H or C–O–C groups [\[20,](#page-3-16) [21\]](#page-3-12). Among them, the relative intensity of Fe–O, C=C–H, and  $CH_3$  bonds in sample A was stronger and the peak at  $1562 \text{ cm}^{-1}$  was only found

<span id="page-1-1"></span>

**Fig. 2** Raman spectra of samples A at different laser power

in sample A, indicating that the hydrothermal electrochemical reaction was more exhaustive at 180 ◦C. Beside the main products, the peaks at 3435 and 2373 cm<sup>-1</sup> implied the existence of residual hydroxyl groups [\[22\]](#page-3-17) and byproduct of  $CO_3^{2-}$  [\[23\]](#page-3-18).

# 3.2 Raman Spectroscopy

The Raman spectra were searched by exciting the sample with the visible light (488 nm) of an argon ion laser. Figure [2](#page-1-1) showed the Raman spectra of samples A under different excitation energies. As shown in Fig. [2,](#page-1-1) the laser power has no clear effect on the sample, indicating no sample degradation by laser irradiation. The displayed spectrum revealed the presence of  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> at 378, 524, and 678 cm<sup>-1</sup> [\[24\]](#page-3-19) and as an other phase carbon at 1361, 1589, and 1428 cm<sup>-1</sup> [\[25,](#page-3-20) [26\]](#page-3-21). The peak around 1361, 1589 and 1428 cm<sup>-1</sup> was assigned to D, G and  $sp^3$ -CH<sub>n</sub>, respectively [\[25,](#page-3-20) [26\]](#page-3-21).

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**Fig. 1** FTIR spectra of samples

<span id="page-1-2"></span>

**Fig. 3** Raman spectra of sample B at different laser radiation power

#### <span id="page-2-0"></span>**Fig. 4** TEM image of sample A

 $50<sub>nn</sub>$  $50<sub>nn</sub>$  $20 \text{ nm}$  $20 \text{ nm}$ 

<span id="page-2-1"></span>**Fig. 5** TEM image of sample B

The Raman spectrum also showed a band at  $1045 \text{ cm}^{-1}$ owing to the presence of  $CO_3^{2-}$  [\[23\]](#page-3-18).

The Raman spectra of sample B are shown in Fig. [3.](#page-1-2) The spectrum recorded with 1 mW showed the same behavior as sample A, but when the laser power was raised to 5 mW, new bands showed up. The new bands at 179, 218, and 282 cm<sup>-1</sup> were characteristic of hematite  $[27]$ . Moreover, the intensity of the D and G peak strongly decreased with increasing laser power. It indicated that unstable carbon prepared at 170 °C gradation by laser irradiation, the carbon could not afford effective protection, and maghemite was assumed to transform into hematite during high-power laser radiation.

## 3.3 TEM Analysis

Figures [4](#page-2-0) and [5](#page-2-1) showed the TEM images of sample A and sample B, respectively. According to Fig. [4,](#page-2-0) dispersible carbon-stabilized  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles with average particle size of 10 nm could be obtained at 180 ◦C. However, coarse aggregates consisting of tens of  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> particles were observed in Fig. [5.](#page-2-1)

## 3.4 VSM Analysis

Figure [6](#page-2-2) showed the magnetization curves of the sample A and sample B at room temperature. The corresponding saturation magnetizations strengths  $(M<sub>s</sub>)$  were 31.2 and

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**Fig. 6** Magnetization hysteresis loops of the samples

13.6 emu/g, respectively. These values were much lower than 76 emu/g of the corresponding bulk magnetite [\[28\]](#page-3-23), which could be attributed to the nanosize of the  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> particles and the presence of carbon [\[29,](#page-3-24) [30\]](#page-3-25).

The saturation magnetization for sample B was lower than that of sample A. This might be attributed to high proportion of non-crystal form of carbon, OH− and nonmagnetic iron oxide.

# **4 Conclusion**

The  $C/\gamma$ -Fe<sub>2</sub>O<sub>3</sub> powders with high magnetization (31.2) emu/g) was obtained by a hydrothermal electrochemical technique at 180 °C one step. The dispersible  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles with average particle size of 10 nm were stabilized by carbon. The carbon could afford an effective protection against maghemite transforming into nonmagnetic hematite.

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