**ORIGINAL PAPER**



# **Unveiling the Potential of** *Borassus fabellifer's* **Leaves Derived ZnO Nanoparticles in Augmenting the Attributes of PLA‑Surface Modifed Nanocellulose Bio‑composite**

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# **Abstract**

The utilization of synthetic plastics has resulted in environmental and human health concerns. Consequently, there is a growing focus on developing eco-friendly alternatives like polylactic acid (PLA). Nonetheless, its drawbacks include slow crystallization, vulnerability to UV radiation, and absence of antibacterial properties. One potential strategy involves integrating nano-cellulose (NC) and metal oxide nanoparticles into PLA to enhance its antibacterial features and mechanical robustness. The NC was extracted from the leaves of *Borassus fabellifer* and further treated with lactic acid to obtain surface-modifed NC (SMNC). Besides, the Zinc oxide nanoparticles (ZnO NPs) were green-synthesized using the *B. fabellifer* leaves extract and further characterized through various analytical techniques such as SEM, XRD, and FTIR. The optimized bio-composite ratio was 89.5 mg of PLA, 3.5% (w% of PLA) of SMNC, and 7% (wt% of PLA) of ZnO NPs based on the appearance, water contact angle (WCA), and antimicrobial activities of the bio-composite flm. The PLA-SMNC-ZnO NPs flm was further characterized and compared with PLA and PLA-SMNC flms for its structural, mechanical, and thermal properties. The PLA-SMNC- ZnO NPs flm exhibited an improved tensile strength of 13.9 MPa and Young's modulus of 0.00689 GPa than PLA and PLA-SMNC. Also, the thermal stability of PLA-SMNC-ZnO NPs flm was 320 °C from TGA. Additionally, PLA-SMNC- ZnO NPs flm exhibited reduced water absorption and improved resistance to UV radiation. In conclusion, these fndings validate the potential of the ZnO NPs derived from *B. fabellifer's* leaves in improving the versatility of PLA.

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# **Graphical Abstract**



**Keywords** Polylactic acid · *Borassus fabellifer* · Green synthesis · Zinc oxide nanoparticles · Surface-modifed nanocellulose

# **Introduction**

Reducing dependency on conventional plastics while pursuing environment-friendly alternatives is critical in an increasingly eco-conscious world. Societies may promote a more sustainable future and mitigate the negative efects of conventional plastics on the environment by using biodegradable materials. Switching to "green" materials is essential since traditional plastics have a detrimental effect on the environment  $[1]$  $[1]$  $[1]$ . One of the most promising solutions is the creation and widespread use of biodegradable substances. These substances minimize the adverse efects of non-biodegradable plastics, such as pollution and resource depletion. As a result, biopolymers are being used as a practical substitute for plastics derived from petroleum [[2](#page-14-1), [3\]](#page-14-2). These biopolymers have several benefts over conventional plastics since they are made from sustainable biomass sources like maize, starch, and vegetable oil [\[4\]](#page-14-3). Primarily, biopolymers can consistently degrade over time owing to the action of microbes. This has an immense advantage since it lessens the number of plastic debris that ends up in landflls and ecosystems. Moreover, they can be developed to have good barrier properties against oxygen and moisture [[5](#page-14-4)], preserving food products' freshness and shelf life. Biopolymers such as polyhydroxybutyrate (PHB), polylactic acid (PLA), polycaprolactone (PCL), starch, cellulose, chitosan, etc., have been utilized in food packaging [\[6](#page-14-5)]. One of the most widely used bio-based and biodegradable polymers for food packaging is PLA which is recognized for having mechanical properties equivalent to polymers based on petroleum like polyethylene and polypropylene [[7](#page-14-6)]. Furthermore, PLA is considered to be "generally recognized as safe" (GRAS) [\[8](#page-14-7)], which confrms its suitability in food packaging applications. However, pure PLA flm is too brittle to be employed as a thin flm and has no antibacterial and antioxidant properties [[9](#page-14-8), [10\]](#page-14-9).

Nanofillers, such as cellulose, lignin, clay,  $TiO<sub>2</sub>$ , graphene, calcium carbonate silica, etc., have been blended with the PLA to enhance the mechanical and thermal characteristics of PLA [\[11–](#page-14-10)[13\]](#page-14-11). *Borassus fabellifer* is a versatile tree that belongs to the *Arecaceae* family and is widely distributed in the southern region of the Indian subcontinent. The proximate analysis of the *B. fabellifer* leaves exposed the high content of cellulose. The *B. flabellifer* leaves derived nanocellulose (NC) were reported for their excellent mechanical properties, such as high tensile strength, stifness, and UV stability when blending with the PLA matrix as a nanofller [[11](#page-14-10), [14](#page-14-12), [15\]](#page-14-13). However, NC's hydrophilic properties make distributing it evenly in the hydrophobic PLA matrix difficult. Its usefulness as a reinforcing material is hampered by hydroxyl groups on the NC surface, making it challenging to scatter uniformly in a nonpolar liquid. Therefore, before being further incorporated into PLA, NC is surface-modifed by acetylation or transesterifcation to improve its hydrophobicity [\[15](#page-14-13), [16\]](#page-14-14). Hence, the PLA matrix has been incorporated with surface-modifed nanocellulose (SMNC) to inherit salient features like high surface area-tovolume ratio, enhanced mechanical qualities, and biocompatibility. However, PLA-SMNC composite should possess substantial antimicrobial activity to improve its candidature for widespread use. Various chemical functionalizations with antibacterial groups or blends with metal nanoparticles (NPs) and metal oxides can demonstrate the antibacterial activity of NC [[17,](#page-14-15) [18\]](#page-14-16). Noble metallic NPs like Au, Ag, Pt, Pd, etc., and non-metallic oxides like  $ZnO$  and  $TiO<sub>2</sub>$  possess extensive antibacterial and photocatalytic activity [[19,](#page-14-17) [20\]](#page-14-18). In addition, zinc oxide nanoparticles (ZnO NPs) possess strong UV absorption capability, photocatalytic activity and good stability [\[19](#page-14-17)[–21\]](#page-14-19). Moreover, ZnO NPs are cheap and also easy to synthesize. Precipitation, thermal breakdown, hydrothermal synthesis, and physical vapor synthesis are some of the synthesis techniques used to produce ZnO NPs [\[21–](#page-14-19)[23\]](#page-14-20).

The distinguishing features of our study extend beyond the green synthesis of ZnO alone, with a primary focus on the uniqueness featured in the PLA-SMNC-ZnO NPs flm. Foremost, the synchronized utilization of *B. fabellifer* leaves for NC and eco-friendly green synthesis of ZnO NPs by valorizing its extract establishes an unprecedented harmony between these components, addressing inherent compatibility issues in PLA composites. This green synthesis not only contributes to the ecological integrity of the process but also imparts a heightened degree of safety to the ZnO component, aligning seamlessly with stringent standards for various applications. This integrated approach not only streamlines production but also ensures a synergistic interaction within the PLA matrix. The resulting PLA-SMNC-ZnO NP flm, characterized comprehensively for its structural, mechanical, and antimicrobial attributes, signifes a breakthrough in the realm of functional biopolymer composites, underpinned by a multifaceted approach to sustainability and innovation.

# **Materials and Methodology**

### **Green Synthesis of ZnO NPs**

The *B. fabellifer* leaves were collected from a local site. These leaves were washed multiple times with purifed water to eliminate any impurities. Subsequently, they were gently dried in a controlled hot air oven at 37–40 °C. The dried leaves were carefully processed into a fne powder using ball milling. 5 g of the dried powder was measured and placed into a 250 ml beaker; then, 100 ml of deionized water was added to the beaker and was heated in a water bath for 1 h, maintaining a temperature of 80 °C. The solution was fltered using Whatman No. 1 flter paper, and the resulting extract was carefully collected into another beaker.

Once cooled, this extract was stored for the subsequent green synthesis of ZnO NPs [\[24](#page-14-21)]. In a beaker, 0.1 M Zinc Nitrate was dissolved in 100 ml of distilled water and mixed with 20 ml of leaf extract to attain a 5:1 ratio. The pH of the solution was carefully adjusted to the desired range of 10–12 by adding 0.1 M NaOH. The mixture was placed on a magnetic stirrer and stirred at 1000 rpm for 2 h at 60 °C, and afterwards, the mixture was kept at 40 °C for 2 h. After cooling, the sample was sonicated for an hour before being centrifuged at 5000 rpm for 15 min. The supernatant was discarded, and the remaining pellet was washed with distilled water before being centrifuged at the same speed. The pellet was calcinated in a muffle furnace at 400  $^{\circ}$ C for 3 h [[25,](#page-14-22) [26\]](#page-14-23).

### **Characterization of ZnO NPs**

#### **UV−Visible Spectroscopy**

The optical properties and conformation of ZnO NPs were determined using a UV−Vis Spectrophotometer, Agilent Technologies. The sample was prepared by diluting the dried ZnO NPs with water, and distilled water was kept as blank for reference. The spectral analysis was performed within a wavelength of 200–800 nm [[27\]](#page-14-24).

#### **Scanning Electron Microscopy (SEM)**

SEM is widely used to examine the microstructure and chemistry of various materials. The surface morphology of nanoparticles was studied using Hi-Resolution SEM (Thermoscientifc Apreo S). The structure of nanoparticles was observed operating at a voltage of 20 kV.

### **Fourier Transform Infrared Spectroscopy (FTIR)**

FTIR spectroscopy relies on the absorption of infrared radiation when passing through a sample. The Agilent Technologies Cary 600 Series FTIR spectrophotometer was utilized to unravel the distinct characteristics of functional groups present in the ZnO NPs. The recorded spectrum of the powdered sample was examined using the KBr pellet method over a broad wavelength range  $(400–4000 \text{ cm}^{-1})$ , which identifies and differentiates between various molecules based on their unique molecular structures [\[27](#page-14-24)].

### **X‑ray Difraction (XRD)**

The X-ray difractogram (BRUKER USA D8 Advance) was used to examine the powdered ZnO NPs, using monochromatic Cu Ka radiation (1.5419Å) at a voltage of 40 kV and a current of 15mA [[28\]](#page-15-0). By accounting for all possible sample orientations, 2θ angles were obtained. These angles were then converted to d-spacings, enabling the defnite recognition of the material. The crystalline size (D) of the synthesized ZnO NPs was estimated using the Debye–Scherrer formula shown below:

$$
D = \frac{k\lambda}{\beta cos cos \theta} \tag{1}
$$

where  $\lambda$  is the X-ray wavelength of 1.5419 Å, k (value 0.9) is the shape factor (dimensionless), θ is the Bragg angle in radians, and β is the full width at half maximum (FWHM) in radian.

### **Zeta Potential**

Utilizing a zeta potential analyzer (Malvern/Nano ZS-90), the zeta potential of the nanoparticles was also measured. The zeta potential could be accurately determined by observing the direction and velocity of the particles in a known electric feld. To assess the charge of the nanoparticles, the dried ZnO NPs were frst diluted in water, which was later used to obtain accurate and reliable data on the particles [\[29\]](#page-15-1).

### **Optimization of Blending Ratio of Bio‑composite**

PLA (Green biostar P 1006), commercially available, was dissolved in chloroform and heated at 40–60 °C for 15–20 min. NC was extracted from the dried leaves of *B. fabellifer* as described by Arun et al. [[30\]](#page-15-2) and surface-modifed with lactic acid (LA). The aqueous solution of NC was blended with LA at a 15:1 ratio and 100 g of Stannous chloride (SnCl<sub>2</sub>). SnCl<sub>2</sub> is employed as a catalyst that facilitates the condensation of acid or acyl anhydrides with a cellulosic alcohol group, creating ester bonds (C–O=O), during surface modifcation of nanocellulose [[16](#page-14-14)]. Further, the solution was homogenized for 10 min at 5000 rpm, followed by ultrasonication for 1 h at 20% amplitude with a pulse of 10 s on and off. The treated NC was kept at 80 °C for 10 h. Afterwards, the surface-modifed NC (SMNC) solution was fltered and dried [[31\]](#page-15-3). The quantity of SMNC was kept constant at 3.5% (w/w) of the PLA in the composite preparation, which was optimized in our preliminary studies (data not shown). Varying concentrations of ZnO NPs (ranging from 1 to 7% (w/w) of the PLA) were added into separate beakers and were mixed thoroughly for 10 min. After reheating, the solution was poured onto a plate and left to air dry. The prepared flm was analyzed further for various properties [\[32,](#page-15-4) [33](#page-15-5)].

### **Water Contact Angle Measurement (WCA)**

The sessile drop method was used to calculate the diferent concentrations of PLA-SMNC-ZnO NPs composite flm's WCA. A 1 µL water droplet was deposited on the film surface to fnd the contact angle, and the angle between the droplet and the surface was measured [\[34](#page-15-6)].

### <span id="page-3-0"></span>**Antimicrobial Assay**

The antibacterial properties of the PLA-SMNC-ZnO NPs biocomposite films were evaluated using the agar diffusion method. A culture of Gram(− ve) *Escherichia coli* (ATCC8739) and Gram(+ ve) *Staphylococcus aureus* (ATCC6538) was prepared by inoculating a loop of bacterial culture into 250 ml of LB media and allowing it to grow overnight in a rotatory shaker at140 rpm. Subsequently, 10 µL of the bacterial dilution was evenly spread on agar petri plates, and wells were created in the solid culture medium. The biocomposite flms were then carefully placed in the wells, and Streptomycin discs were employed as the standard control. The plates were subsequently incubated for 24 h at 37 °C to facilitate bacterial growth and the development of inhibition zones around the flms. The diameter of the inhibition zone around each flm was measured, with a larger diameter indicating a greater extent of bacterial inhibition and stronger antibacterial activity [\[35](#page-15-7)]. Based on WCA measurement and antimicrobial assay, it was found that 7% flm exhibited the maximum water contact angle and a larger zone of inhibition. This flm was further used for various characterization and packaging applications.

### **Characterization of the Bio‑Composite**

# **Surface Morphological Study of PLA‑SMNC‑ZnO NPs Biocomposite**

The surface morphology of the nanocomposite flms was studied with the help of SEM, focusing on the binding status between ZnO NPs and the PLA-SMNC biocomposite [\[33](#page-15-5)]. The samples were examined to understand the biocomposite flm's surface comprehensively. Additionally, atomic force microscopy was utilized to analyze the 3D structure and surface roughness of the PLA-SMNC and PLA-SMNC-ZnO NPs flm samples, each measuring 2 \* 2cm [\[33](#page-15-5), [36](#page-15-8)].

### **Structural Analysis of the Bio‑Composite**

The crystalline or amorphous nature of the PLA and their composites was ascertained by XRD, performed at room temperature using a Cu-Kα source and a generator set at 40 kV and 15mA [[33\]](#page-15-5). As described previously, the functional groups in the PLA-SMNC-ZnO NPs flm samples were detected through FTIR analysis.

# **Mechanical Properties**

A testometric machine with grips having a cross-head speed of 50 mm/min and separated at 50 mm was used to assess the mechanical qualities of the flm samples. Established procedures were followed to ascertain the tensile strength, elastic modulus, and elongation at break. Tensile strength was computed by dividing the specimen's initial cross-sectional area by the maximum load the flm could bear before failing. The linear stress/strain curve's slope was used to calculate the elastic modulus, and the specimen's elongation at break was determined by calculating the percentage change in length between the grips at the point of failure [\[32](#page-15-4), [37,](#page-15-9) [38\]](#page-15-10).

### **Thermal Analysis**

Thermal gravimetric analysis was performed for the PLA, PLA-SMNC, and PLA-SMNC-ZnO NPs flms using a thermogravimetric analyzer. The heating process was conducted in a nitrogen atmosphere, starting from 30 °C and gradually increasing to  $600 \degree C$  at  $10 \degree C/\text{min}$  [\[37\]](#page-15-9).

### **Water Absorption**

Samples measuring 2 \* 2 cm were submerged in 250 ml of water for 3 days to examine the water absorption capabilities of the PLA, PLA-SMNC, and PLA-SMNC-ZnO NPs flms. The change in the weight of the flm was measured every 24 h. The composite flms were dried at 37 °C before measuring the water absorption percentage. The amount of water absorbed by the flm was ascertained by measuring each sample's weight before and after immersion and calculating the diference between the initial and fnal weights [\[39](#page-15-11)].

### **UV Barrier Analysis**

UV−Vis Spectrophotometer analyses the absorbance at 280–400 nm wavelength for PLA, PLA-SMNC, and PLA-SMNC-ZnO NPs flms. The samples were cut into regular rectangle shapes (1\*4 cm) and placed into the cuvette. The transmittance for both UVA (310–400 nm) and UVB (280–315 nm) were recorded, and the calculation was worked out using the following equations [[39\]](#page-15-11):

$$
UVA = \frac{ Transmission(315nm + 320nm + \dots + 400nm)}{18}
$$
 (2)

$$
UVB = \frac{ Transmission(280nm + 285nm + \dots + 315nm)}{8}
$$
 (3)

# **Results and Discussion**

# **Characterization of Synthesized ZnO NPs**

### **UV–Vis Spectroscopy**

UV−Vis spectrometry analysis was performed to validate the synthesis of ZnO NPs. UV−Vis spectrometry analysis, ranging from 300 to 400 nm, revealed a distinctive peak at 355 nm, which is the characteristic absorbance peak of ZnO NPs (Fig. [1a](#page-5-0)). This observation aligns with prior research indicating that the absorbance band of ZnO NPs typically ranges from 315 to 360 nm [[40](#page-15-12)]. Similarly, comparable results were noted during the production of ZnO NPs using zinc nitrate and *B. fabellifer* fruit pulp [\[29](#page-15-1)]. Particularly in the UV region, these nanoparticles displayed a distinctive absorbance peak linked to stimulating their surface plasmons (SP). This further emphasizes the infuence of nanoparticle size on this phenomenon. Interestingly, the SP resonance bands were observed to shift, either towards the red or blue end, as a result of quantum size efects.

#### **SEM and Zeta Potential Analysis**

The surface morphology of the synthesized ZnO NPs was examined using SEM analysis. Fig. [1](#page-5-0)b captures an image of the ZnO NPs, revealing the arrangement of irregularly shaped particles that are agglomerated and unevenly distributed. The particle size was determined to be approximately 20–30 nm, based on the SEM image analysis using ImageJ software [[28\]](#page-15-0). This aggregation was observed due to the



<span id="page-5-0"></span>**Fig. 1 a** UV Spectrophotometric analysis of ZnO NPs, **b** SEM analysis of ZnO NPs, **c** Zeta potential of ZnO NPs

interface between the NPs and bio-organic capping particles, aided by hydrogen bonding and electrostatic forces [[41\]](#page-15-13). The zeta potential value of the synthesized ZnO NPs was measured to be − 6.29 mV, indicating the relatively stable nature of the ZnO NPs (Fig. [1c](#page-5-0)).

# **FTIR Analysis**

The FTIR spectrum depicted in Fig. [2a](#page-6-0) represents the analysis of the ZnO NPs synthesized by utilizing *B. fabellifer* leaf extract. The main objective of this spectroscopic investigation was to detect the functional groups attributed to the biological molecules accountable for efectively capping and stabilizing the ZnO NPs. The observed peaks within the 3200–3600 cm−1 range in the FTIR spectrum correspond to the hydroxyl group stretching of intramolecular hydrogen bonds and the stretching of 'CH' bonds in alkanes [[28](#page-15-0)]. Moreover, the discernible peaks at 1489 were Associated with the absorption band of C–C bonds, while the bands within the 1100–1300  $cm^{-1}$  range indicated the presence of C–N bonds of amines and C–OH groups. Additionally, the vibrations that afect the bond length found between 400 and 750 cm−1 indicated the presence of ZnO, specifcally equivalent to its stretching mode and a prominent peak at  $514 \text{ cm}^{-1}$ , corresponding to ZnO [[42](#page-15-14)]. Previous studies have confrmed the existence of a diverse range of compounds within *B. fabellifer*, such as polyphenols, alkaloids, aromatic hydrocarbons, terpenoids, and steroids [[29](#page-15-1)]. As a result, our FTIR analysis revealed the existence of functional groups in the extract's biomolecules, including all the abovelisted phytochemicals. These compounds play an important role in the reduction and capping processes of ZnO NPs.



<span id="page-6-0"></span>**Fig. 2 a** FTIR analysis of *B. fabellifer* leaves and ZnO NPs, **b** XRD analysis of ZnO NPs

They also prevent the NPs from clumping together in the extract solution. This not only led to the formation of ZnO particles but also highlighted the stabilizing capabilities of the plant extracts, indicating their dual role as both reducing agents and stabilizers.

### **XRD Analysis**

The crystalline structure of the nanoparticles was comprehensively analyzed via XRD. The peaks, observed at specifc 2θ values of 47.21°, 56.27°, 62.52°, 67.61°, 68.76°, and 69.02°, were determined to correspond to lattice planes (102), (110), (103), (200), (112), and (201) respectively, as per the JCPDS Card Number (36-1451). By analyzing specific positions of the peaks, such as  $31.46^{\circ}$ ,  $34.12^{\circ}$ , and 35.94°, the lattice planes (100), (002), and (101) were also identifed among others and when compared to the JCPDS card they were found to correspond precisely, confrming the presence of a hexagonal wurtzite crystal structure (Fig. [2](#page-6-0)b). Signifcantly, the existence of pristine NPs is indicated by the planes  $(100)$ ,  $(002)$ , and  $(101)$   $[24]$  $[24]$ . Moreover, the size of the NPs was calculated using the Debye–Scherrer equation (Eq. [1](#page-3-0)). The determined average crystal size of the nanoparticle was 10.67 nm.

### **Optimization of PLA‑SMNC‑ZnO NPs Composite**

The PLA-SMNC-ZnO NPs blending ratio was optimized by the appearance of the casted flm, water contact angle, and antimicrobial properties. The flm cast with 1–3% of ZnO NPs were observed to be transparent. Whereas the flms with 4–7% were opaque. However, a further increase in the proportion of ZnO NPs resulted in the formation of clusters. Hence, the bio-composite flms with 1–7% ZnO NPs were chosen for further optimization.

### **WCA**

WCA is employed to assess the water absorption characteristics of the flm. The WCA was found to be increasing as the amount of ZnO NPs increased, which exemplifes the improved flm's water-repellent characteristics. An angle exceeding 90° indicates the flm's water-repelling nature. Table [1](#page-6-1) shows the result of WCA for diferent compositions of the PLA-SMNC-ZnO NPs composite flm. Film compositions with 1 to 3% of ZnO NPs were found to be hydrophilic, as the WCA was measured to be less than 90°. Whereas the flms with 4–7% ZnO NPs exhibited WCA of more than 90°, which signifies a hydrophobic nature [\[43](#page-15-15)]. The highest water contact angle of 92.5° was measured for the flm with 7%

<span id="page-6-1"></span>**Table 1** Optimization of PLA-SMNC-ZnO NPs ratio based on water contact angle

PLA $(wt\%)$	SMNC (wt% of PLA)	$ZnO$ NPs (wt% of PLA)	WCA <sup>o</sup>
95.5	3.5	1	48
94.5	3.5	2	51.5
93.5	3.5	3	75.5
92.5	3.5	4	80.5
91.5	3.5	5	83
90.5	3.5	6	86.4
89.5	3.5	7	92.5

ZnO NPs. The fndings were similar to the results reported for PLA-ZnO NPs composite [[44\]](#page-15-16).

# **Antimicrobial Assay**

The antimicrobial activity of the PLA-SMNC-ZnO NPs  $(3–7%)$  was evaluated against both the Gram( $-$  ve) bacteria *E. coli* and Gram(+ve) bacteria *S. aureus*. The zone of inhibition was observed by measuring the dimensions of the clear areas encircling the biocomposite flm (Fig. [3a](#page-7-0) and b). The flm with 3% Zno NPs did not show any signifcant antibacterial activity. Figure [4](#page-7-1) shows the measured zone of inhibition values of the PLA-SMNC-ZnO NPs flms for *E. coli* and *S. aureus*. This study observed that the inhibitory efects of the flm on both microbes were enhanced as the concentration of ZnO NPs increased from 4 to 7%. Specifcally, the inhibition zone for *E. coli* showed a signifcant increase from 25 to 32 mm, while for *S. aureus*, it increased from 21 to 35 mm. These results are in accordance with a previous study conducted by Ghozali et al. [\[35\]](#page-15-7), where the highest concentration of ZnO additives also leads to the widest clear zone. This suggests that the direct interaction between the metal oxide and the bacterial cell wall may play a crucial role in the observed antibacterial efects as it leads to damage to the cell. Additionally, the reactive sites of the metal oxide likely facilitate the antibacterial activities observed in this study [[45](#page-15-17)].

The antibacterial mechanism of PLA-SMNC-ZnO NPs flms on the studied bacteria is mainly attributed to the release of  $\text{Zn}^{2+}$  ions and the generation of reactive oxygen species (ROS). The ROS generated include hydroxyl radicals (·OH) and ions (OH−), hydrogen peroxide  $(H_2O_2)$ , superoxide anions  $(O_2^-)$  and hydroperoxyl radicals (HOO·). Nanoparticles gather on the surface of bacterial cell membranes and destroy them. Furthermore, the cytotoxic action of ZnO nanomaterials is believed to be caused by physical electrostatic interactions between surfaces of bacterial cells with diferent charges, mechanical damage to cell envelopes, particle penetration, internalization, and accumulation within the cells. These factors are



 $(a)$ 

<span id="page-7-0"></span>**Fig. 3** Antimicrobial assay of PLA-SMNC-ZnO NPs against **a** *E. coli*, and **b** *S. aureus* 

<span id="page-7-1"></span>**Fig. 4** Zone of inhibition of PLA-SMNC-ZnO NPs flms for *E. coli* and *S. aureus*



all proposed to contribute to the destruction of bacterial cells by ZnO nanomaterials [[46](#page-15-18), [47\]](#page-15-19).

The incorporation of ZnO NPs into the PLA-SMNC flm has also been shown to enhance surface roughness, a crucial factor in strengthening the antimicrobial properties of ZnO NPs in the flm. This increase in roughness results in a greater efective surface area compared to the previously geometrically fat region. This expansion is expected to lead to a higher production of active species, such as ions and ROS, which are responsible for the antimicrobial effects. Additionally, the added roughness provides a larger surface area for bacterial attachment. This aligns with recent research by Valireini et al. [[48](#page-15-20)], which demonstrated that rougher surfaces promote increased bacterial adhesion through physical interactions with bacterial cell walls. With the augmentation in adhesion between materials and bacteria, a nuanced structural metamorphosis ensues. This transformative process fortifes the defence against microbes, encompassing heightened abrasive impacts and the conveyance of active agents from the material's surface to bacterial cells.

# **Characterization of the PLA‑SMNC‑ZnO NPs Composite**

### **Surface Morphological Study**

The surface level morphology of the PLA and its composite flms were investigated using SEM (Fig. [5\)](#page-8-0). Smooth and fat surface of the neat PLA was free of noticeable cracks, voids, or air pockets (Fig. [5](#page-8-0)a). Signifcant changes in the microstructures of the flm were detected after introducing 3.5 wt% SMNC into the PLA matrix. The PLA-SMNC flm, in particular, showed the development of pores inside the matrix (Fig. [5](#page-8-0)b), with diameters ranging from 3 to 8 nm. Although porous flms are frequently used in biomedical applications, the PLA-SMNC composite alone may not be ideal for any application due to its porosity structure, which



 $(a)$ 

 $(b)$ 

<span id="page-8-0"></span>

**Fig. 5** SEM analysis of **a** PLA, **b** PLA-SMNC, and **c** PLA-SMNC-ZnO NPs flms

may compromise the flm's migratory qualities [[49\]](#page-15-21). As ZnO NPs were further integrated into the matrix, the SEM analysis exposed a seamless distribution of ZnO NPs throughout the PLA-SMNC matrix (Fig. [5](#page-8-0)c). The addition of ZnO NPs notably decreased the flm's porosity due to the favorable integration of ZnO NPs with both PLA and SMNC.

AFM analysis, a highly advanced imaging technique for scanning and measuring minuscule features on surfaces, was adopted to explore the surface morphology and roughness of PLA, PLA-SMNC, and PLA-SMNC-ZnO NPs flms (Fig. [6](#page-9-0)). Upon performing AFM analysis on the PLA flm, it was evident that the surface was sleek, potentially raising concerns for its use in packaging (Fig. [6](#page-9-0)a). However, the PLA-SMNC flm had discernible pores. Due to these pores, this composite was still unsuitable for many applications (Fig. [6](#page-9-0)b). Conversely, the AFM image of the PLA-SMNC-ZnO NPs flm showed an absence of pores and small peaklike protrusions due to the clustering of ZnO NPs within the film (Fig. [6c](#page-9-0)). Additionally, there was an increase in surface roughness compared to the original PLA flm. Surfaces with higher levels of roughness can enhance their hydrophobic properties. This is because of the air trapped by hydrophobic surfaces, which becomes even more efective with increased roughness, according to Wan et al. [[50\]](#page-15-22). As a result, it can be concluded that incorporating ZnO NPs into the PLA-SMNC matrix has led to a noticeable improvement in hydrophobicity and a reduction in pore formation in the flm, making it



<span id="page-9-1"></span>**Fig. 7** XRD analysis of **a** PLA, **b** PLA-SMNC, and **c** PLA-SMNC-ZnO NPs flms

suitable for many applications. Similar research by Nonato et al. [[51\]](#page-15-23) has also emphasized the benefcial impact of ZnO NPs on improving the surface roughness of PLA flms.

### **Structural Analysis**

The crystallographic characteristics of PLA and different PLA biocomposites were investigated using XRD and depicted in Fig. [7.](#page-9-1) While the neat PLA exhibited largely

<span id="page-9-0"></span>

amorphous structures with no discernible crystallization, the PLA-SMNC composite exhibited prominent peaks at approximately 15.90° and 18.27° of 2θ, indicating the presence of (200) and (110) planes, respectively. These fndings suggest that PLA follows an orthorhombic structure with three unequal axes at right angles to each other within its crystalline system [[52](#page-15-24)]. After incorporating ZnO NPs, the intensities of the two peaks associated with PLA's crystallinity decreased, but this did not diminish the ZnO NPs' ability to promote PLA crystallization. Moreover, distinct difraction peaks at 31.31°, 33.92°, and 35.68° were identifed in the XRD patterns of the PLA-SMNC-ZnO NPs composite flms. These peaks corresponded to the (100), (002), and (101) crystal planes of ZnO, respectively [[53\]](#page-15-25).

From the FTIR graph, the presence of various functional groups was explored (Fig. [8\)](#page-10-0). The vibration of the hydroxyl bonds in the PLA molecules was thought to be the cause of the unique band at  $3653 \text{ cm}^{-1}$  that was seen in the pure PLA film. It was determined that the stretching of the  $-CH<sub>3</sub>$ group was responsible for another peak at 2935 cm<sup>-1</sup>. Notably, the sharp peak at 1739 cm−1 corresponds to the vibrations that afect the bond length of carbonyl bonds in the ester groups of the PLA structure. Furthermore, the peaks at 1444 cm−1 and 1352 cm−1 were suggestive of symmetrical and asymmetrical bend vibrations of C-H bonds, revealing more about the flm's molecular structure. The strength of the carbonyl group peak was reduced in the PLA-SMNC flm. This is believed to be due to hydrogen bond formation



<span id="page-10-0"></span>**Fig. 8** FTIR analysis of **a** PLA, **b** PLA-SMNC, and **c** PLA-SMNC-ZnO NPs flms

between the PLA matrix's carbonyl groups and the SMNC's hydroxyl groups. Interestingly, the PLA-SMNC composite exhibited identical peaks to the pure PLA spectra but at differing intensities. This implies that the presence of SMNC afects the features of these peaks. By including ZnO NPs, the carbonyl group stretching vibration peak was efectively moved to a lower wavelength, demonstrating the possibility of intermolecular interactions between PLA, SMNC, and ZnO NPs. Additionally, the emergence of a novel band at 421 cm−1 substantiated the successful inclusion and integration of ZnO NPs into the PLA-SMNC composite, reinforcing the incorporation procedure [[28,](#page-15-0) [30,](#page-15-2) [42\]](#page-15-14).

### **Mechanical Properties**

The mechanical characteristics of PLA biocomposite flms were meticulously assessed in Universal Testing Systems for tensile, compression, and fexure testing equipped with a 0.5 kN frame capacity. The tensile properties were thoroughly evaluated by subjecting samples with a width of 20 and 70 mm long to a cross-head velocity of 2 mm/min. Table [2](#page-10-1) displays the varying thicknesses of PLA-based biocomposite flms. Initially, the PLA flm had a thickness of 0.220 mm, but with the addition of SMNC, the thickness was reduced to 0.120 mm. It is worth noting that the PLA-SMNC-ZnO NPs biocomposite showed a signifcant decrease in thickness compared to the neat PLA flm. However, the thickness of PLA-SMNC-ZnO NPs was more than that of the PLA-SMNC composite flm. Table [1](#page-6-1) provides a comprehensive analysis of several mechanical parameters, including tensile strength (TS), Young's modulus (EM), ultimate force, break distance, and total elongation break (EB) of diferent PLA biocomposite flms and the neat PLA flm. The TS evaluates the utmost stress the flm can endure before fracturing. For the pure PLA flm, this was found to be 12.1 MPa. Nevertheless, the inclusion of SMNC led to a reduction in TS, presumably owing to insufficient interfacial bonding between the SMNC and the PLA matrix. Conversely, the addition of 7 wt% of ZnO NPs to the PLA-SMNC composite, resulted in a signifcant increase in the TS of 14.87% over the plain

<span id="page-10-1"></span>**Table 2** Mechanical property analysis: (a)PLA, (b)PLA-SMNC, (c) PLA-SMNC-ZnO NPs flms

	PLA	PLA-SMNC	PLA- SMNC- $ZnO$ NPs
Tensile strength (MPa)	12.1	10.1	13.9
Young's modulus (GPa)	0.00307	0.00303	0.00689
Ultimate force $(N)$	46.8	19.8	40.4
Break distance (mm)	26.9	17.9	6.60
Total elongation break $(\%)$	67.3	44.8	16.5
Thickness (mm)	0.22	0.12	0.16

PLA flm and 32.62% over the PLA-SMNC flm. This is likely due to the efective dispersion of ZnO NPs in the PLA matrix, promoting the establishment of robust intermolecular hydrogen bonds between the constituents of the flm. Alterations in tensile strength and elongation values might be attributed to the creation of hydrogen bonds between the  $(-OH)$  groups in SMNC and the  $(O<sub>2</sub>)$  atoms within the ZnO-NPs [[54\]](#page-15-26).

However, the fexibility of the PLA flm was negatively impacted by the addition of SMNC and ZnO, as evidenced by a decrease in its overall EB. The EB of the PLA- SMNC-ZnO NPs composite flm notably decreased by 75.48% compared to the neat PLA flm and 63.17% compared to the PLA-SMNC flm, signifying reduced fexibility in the composite flm. This phenomenon was ascribed to the restrictive infuence of stif SMNC and ZnO NPs on the motion of PLA molecular structures, consequently limiting the EB. The decreased EB in the PLA-SMNC-ZnO NPs flm is likely due to the clustering and uneven distribution of ZnO-NPs, and the recrystallization of the ZnO-NPs in the polymer matrix [[54\]](#page-15-26). Similarly, fndings by Yu et al. [\[30](#page-15-2)] align with this trend, confrming the enhancement of various mechanical properties while negatively impacting a few through the addition of ZnO NPs to both neat PLA and PLA-SMNC composite.

### **Thermal Properties**

Thermal assessments of PLA, PLA-SMNC, and PLA-SMNC-ZnO NP films were conducted using TGA and DTG techniques, as shown in Fig. [9](#page-11-0)a and b respectively. The analysis indicated signifcant disparities in the thermal characteristics among the various flm types. TGA outcomes unveiled a two-stage degradation mechanism for pure PLA and its composite forms. For PLA, the frst stage of degradation is initiated at 84 °C, which resulted in an almost 8% reduction in weight and complete degradation observed at 384. During the frst degradation stage of PLA-SMNC, the weight decreased by 7% at 170 °C, with a complete loss occurring at 381 °C. This weight loss might be ascribed to moisture evaporation inside the coatings. Similarly, in the PLA-SMNC-ZnO NPs composite, there was a 4% reduction in weight at 147 °C during the initial degradation stage, and complete weight loss was attained at 320 °C. However, the inclusion of ZnO NPs in the nanocomposite flms noticeably decreased their thermal resistance compared to the pure PLA film. This may be due to the degrading effect of ZnO NPs on PLA at higher temperatures. These fndings corroborated with the conclusions of previous studies by Suryanegara et al. [[55](#page-15-27)] and Shankar et al. [[56](#page-15-28)], which also reported a decrease in PLA's thermal stability with the incorporation of ZnO NPs. Murariu and colleagues proposed a theory suggesting that zinc compounds initiate a chemical process where transesterifcation reactions between the molecules are catalyzed, leading to the creation of PLA with reduced molecular weight. Additionally, depolymerization occurs, resulting in the formation of lactic acid at higher temperatures [[57\]](#page-16-0).

### **Water Absorption**

For evaluating the water absorption properties of PLA biocomposite flms, a comprehensive water absorption test was conducted, and the fndings are graphically displayed



<span id="page-11-0"></span>**Fig. 9 a** TGA anlaysis of PLA, PLA-SMC, and PLA-SMNC-ZnO NPs flms, and **b** DTG analysis of PLA, PLA-SMNC, and PLA-SMNC-ZnO NPs flms

<span id="page-12-0"></span>**Fig. 10** Water absorption analysis for PLA, PLA-SMNC, and PLA-SMNC-ZnO NPs flms



in Fig. [10.](#page-12-0) Water absorption in flms is afected by several parameters, including exposed surface area, fuid temperature, difusivity, fbre content, the span of exposure, and film orientation. In this study, a  $2 * 2$  cm film sample was submerged in water for 3 days, and the percentage of weight loss was used as an indicator for water absorption. The fndings unmistakably demonstrate that the pure PLA flm exhibits a signifcantly greater water absorption rate in comparison to both the PLA-SMNC and PLA-SMNC-ZnO NPs flms, underscoring the enhanced water resistance of the latter. Notably, there was a substantial difference of approx. 10.97% in water absorption between the PLA and PLA-SMNC-ZnO NPs flms. After 24 h of immersion, an 8.3% difference was observed between the PLA-SMNC and PLA-SMNC-ZnO NPs flms. Even after 72 h, this trend persisted, showcasing the sustained water-resistant capabilities of the PLA-SMNC-ZnO NPs flm. The fndings strongly indicate that the PLA-SMNC-ZnO NPs flm is highly suitable for packaging usage, as it outperforms both PLA and PLA-SMNC flms regarding water absorption. Moreover, numerous studies have highlighted the signifcant advantage of using this flm in packaging food products, as it efectively prevents spoilage owing to its lower water absorption properties. The observed results were found to be analogous to the previous studies by Drelich et al. [[58](#page-16-1)] and Rahman et al. [[59](#page-16-2)], both concluded that neat PLA has a greater susceptibility to water absorption than the PLA-SMNC composite and that PLA-ZnO composites possess superior barrier properties compared to neat PLA. The fndings further emphasize the improved barrier properties of the PLA-SMNC flm due to the integration of ZnO NPs. The water absorption of PLA-SMNC-ZnO NPs flms is lower than PLA-SMNC flms due to the hydrophobic nature of ZnO NPs and the strong interfacial adhesion between the ZnO NPs and the SMNC matrix. The hydrophobicity of ZnO NP**s** reduces



<span id="page-12-1"></span>**Fig. 11** UV stability test for **a** PLA, **b** PLA-SMNC, and **c** PLA-SMNC-ZnO NPs flms

the water uptake of the PLA-SMNC-ZnO NPs flms, which in turn improves their water resistance [[48](#page-15-20)].

# **UV Barrier Analysis**

UV rays, spanning a range of 100 to 400 nm, comprise UVA, UVB, and UVC types. A protective phenomenon occurs upon encountering the ozone layer, preventing UVC rays from reaching the Earth's surface. Despite this shielding, UVA and UVB rays successfully penetrate the Earth's atmosphere. UVA rays possess wavelengths within 315 to 400 nm, while UVB rays range from 280 to 315 nm. When these rays interact with UV-sensitive materials and biopolymers, they can cause material degradation and initiate chemical processes [[37](#page-15-9)]. To better understand this phenomenon,

an investigation was conducted to determine the transmission percentage of UV rays through PLA-SMNC and PLA-SMNC-ZnO NPs flms.

The PLA flm exhibited a UV ray transmission of nearly 45%, compared to only about 16% for the PLA-SMNC flm and approximately 10% for the PLA-SMNC-ZnO NPs flm, as illustrated in Fig. [11](#page-12-1). This indicates a signifcant 35% decrease in UV ray transmission when comparing the PLA-SMNC-ZnO NPs flm to the standard PLA flm. Additionally, there was a slight reduction of 6% when comparing the PLA-SMNC and PLA-SMNC-ZnO NPs flms. It is noteworthy that none of the flms showed any transmission in the UVC range of 100–280 nm. In terms of UVA rays, the PLA flm had a transmission of 41.232%, while the PLA-SMNC flm had a lower transmission of 10.768%, and the PLA-SMNC-ZnO NPs flm showed the lowest transmission of 5.247%. By adding 7% ZnO NPs to the PLA matrix, a notable decrease from  $41.232$  to  $5.247\%$ , underscoring the efficacy of SMNC in bolstering the UV-blocking capabilities of PLA.

The observed decrease in UV transmission of PLA-SMNC-ZnO NPs films is due to the synergistic effect of SMNC and ZnO NPs. SMNC is a biodegradable and renewable material that is used as a matrix in the production of biocomposite flms. The precise composition of nanostructures plays a vital role in manipulating the energy gap of materials, ultimately leading to targeted absorption of UV bands while maintaining low absorbance in the visible range. As previously investigated, the energy gap determines the range of light wavelengths a material can capture. Inorganic materials such as ZnO are renowned for their wide energy gaps and offer excellent potential in designing UV-protective coatings [\[60](#page-16-3)]. Because of this, it is known for its UV-blocking properties. Whereas SMNC has been shown to have UVabsorbing properties that can reduce the transmission of UV radiation. The incorporation of ZnO NPs into the SMNC matrix leads to the formation of a network structure that enhances the interfacial adhesion between the ZnO NPs and the SMNC matrix  $[61-63]$  $[61-63]$ . Previous studies, such as those conducted by Shankar et al. [[53](#page-15-25)], have also illustrated the UV light-blocking properties of ZnO NPs when integrated into a PLA matrix. In conclusion, including SMNC and ZnO NPs in the PLA matrix enhances its ability to withstand UV radiation, thereby improving its overall durability. Evidently, the signifcant changes in the transmission percentage were due to the integration of SMNC and ZnO NPs obtained from *B. fabellifer* leaves into the PLA matrix.

# **Conclusion**

The study on PLA-based composites, enriched with SMNC and ZnO NPs derived from *B. fabellifer* leaves, has unveiled signifcant advancements that render these composite materials highly versatile for a multitude of applications. Inextricably tied to human necessities throughout history, the *B. fabellifer* tree bears both economic and social signifcance. Its leaves, known for their high cellulose content and natural fbers, lend exceptional mechanical strength and UV stability to materials. Additionally, ZnO NPs derived from the same source alleviate compatibility issues with PLA in the PLA-SMNC-ZnO NPs composite. Furthermore, by incorporating cellulose and ZnO NPs obtained from repurposed waste tree leaves, the approach exemplifes sustainable resource utilization without necessitating dedicated tree harvesting. This unique amalgamation of utilizing discarded materials and unconventional sources for cellulose and ZnO NPs underscores the cost-efectiveness and eco-friendliness inherent in the composition of PLA-SMNC-ZnO NPs flms. The successful integration of ZnO NPs within the PLA-SMNC matrix has notably improved the flm's robustness, making it well-suited for various industrial uses. Notably, the PLA-SMNC-ZnO NPs composite exhibited enhanced mechanical properties, including increased TS and EM compared to the pure PLA and PLA-SMNC flms. This improvement can be attributed to the effective integration of ZnO NPs, strengthening intermolecular bonds and reinforcing structural stability. Additionally, the flms demonstrated reduced water absorption and increased hydrophobicity, showcasing their potential for moistureresistant and barrier applications. Moreover, the antibacterial efficacy against *E. coli* and *S. aureus* bacteria makes these flms particularly promising for safeguarding packaged goods from potential microbial threats. Furthermore, the signifcant enhancement in UV stability broadens their practical applications, particularly in settings where prolonged exposure to sunlight is expected. Furthermore, conducting thorough evaluations of these biocomposite flms' biodegradability and environmental impact in different scenarios could yield vital insights for their widespread use in various industries, such as food packaging, medicine, and sustainable consumer goods.

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#### **Declarations**

**Competing Interests** The authors declare no competing interests.

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