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Characterization of Solution Blow Spun Poly(Lactic) Acid Based Nanofbers Containing Sucuk Spice Mix Essential Oils

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Abstract

In this study, essential oil (EO) from sucuk spice mix was incorporated into poly(lactic) acid (PLA) and nanofbers were produced via solution blow spinning (SBS). EO incorporation was performed at 0, 5, 10, 15, 20, 25, and 30% (v/w) of PLA. Mechanical, barrier, microstructural, topographical, contact angle, chemical, thermal, antioxidant, and antibacterial properties against *E. coli* and *S. aureus* were characterized. Nanofbers were not soluble in water due to the hydrophobic nature of PLA. Microstructural images revealed that beadles and uniform nanofbers can be produced with SBS, and EO addition decreased the average fber diameter. The surface topography of nanofbers was improved by increasing EO concentration. Melting and glass transition temperatures decreased and the thermal stability of nanofbers increased with EO addition. Nanofbers containing EO at 30% had the highest antioxidant capacity (40.02%). Nanofbers containing 20, 25, and 30% of EO had greater growth inhibition against *Staphylococcus aureus* and *Escherichia coli*. As a result, PLA-based nanofbers containing sucuk spice mix EO can be produced with SBS, and nanofibers containing 20, 25, and 30% EO could be a promising system not only in drug delivery systems but also in food packaging.

Keywords Nanofber · Food packaging · Solution blow spinning · Poly(lactic) acid · Antibacterial activity

Introduction

Nano-technological developments in recent years have led to produce new materials for biomedical applications such as tissue engineering, wound dressing and controlled release of drugs, and active food packaging applications [[1,](#page-11-0) [2](#page-11-1)]. One of the most widely investigated Nanomaterial is nanofber because it has unique properties such as a large surface area

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to volume ratio, high encapsulation efficiency, low specific gravity, and controlled-release properties [\[3\]](#page-11-2). The production of nanofbers can be achieved by several methods including melt spinning, melt blowing, and electrospinning (ES) from which ES is the most widely studied method due to its adaptability to a wide range of polymeric materials and consistency in producing very fne fbers [[4\]](#page-11-3). In the ES process, a polymeric solution is pumped towards a large electrical feld to overcome surface tension, and hence, a fuid jet occurs and travels through the grounded collector. The use of a large electrical feld and conductive collector, and a relatively low nanofber production rate limit the commercialization and the use for an immediate in-situ application of electrospun nanofbers [[5\]](#page-11-4).

Recently, a novel alternative nanofiber production technique is developed by Medeiros et al. [[3\]](#page-11-2) as solution blow spinning (SBS). In this technique, a concentric nozzle is used through which polymer solution and pressurized gas are simultaneously passed. The aerodynamic drag and shear forces caused by the pressurized gas exiting the nozzle are combined to form a Taylor cone-like shape at the nozzle tip, which is formed in the electrospinning process.

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During the fight of polymeric solution towards to collector placed at a fxed distance, the solvent evaporates and the polymer is formed a reticulated yarn structure at the micro and nanoscale, called nanofber. SBS can be successfully used for the production of nanofbers from poly-lactic acid (PLA) [\[6](#page-11-5)[–8](#page-11-6)], poly-lactic co-glycolic acid (PLGA) [[5\]](#page-11-4), fsh skin gelatin [[9\]](#page-11-7), corn zein [[10](#page-11-8)] and poly-methyl methacrylate (PMMA) [\[3](#page-11-2)].

Biomedical applications such as wound dressing, tissue scaffolding, drug delivery, and food packaging applications of nanofbers are focused on biodegradable polymers. In this regard, PLA is one of the most promising polymers due to its biodegradability, nontoxicity, well processes ability, and good mechanical properties [[11](#page-11-9)]. In addition, PLA can be processed into fbers that are ideal for the controlled release of encapsulated volatile compounds such as EOs due to their extremely high surface area. PLA-based nanofibers containing linalool $[12]$, copaiba oil $[13]$ $[13]$, and terpinene 4-ol $[14]$ $[14]$ were produced with SBS.

Sucuk is the most widely consumed fermented meat product in Turkey, and it consists of diferent spices mainly cumin, red pepper, and black pepper. This spice mix is responsible for the main aroma of sucuk after the aromatic compounds are formed by fermentation. It is well known that EOs from spices are natural preservatives that have high antioxidant capacities and antimicrobial activities against most food-pathogenic bacteria. The addition of EOs into biodegradable nanofbers for food packaging applications is a very trendy topic in the literature. Aytac et al. [[1\]](#page-11-0) reported that thymol-enriched zein nanofbers could be used for the enhancing shelf life of chicken meat. In another study, PLA nanofbers were enriched with guar gum and thyme EO, and it was stated that these nanofbers could be a promising system in the safety and shelf life of foods [[15\]](#page-11-13).

To the best of our knowledge, there is no study focusing on the production of solution blow spun PLA nanofbers incorporated with sucuk spice mix EO. This study mainly aimed to develop sucuk spice mix EO-loaded PLA-based nanofbers with SBS and to characterize the packaging properties of nanofbers in terms of mechanical, barrier, microstructural (SEM), topographical (AFM), chemical (XRD and FTIR), thermal (DSC and TGA), antioxidant, and antibacterial properties.

Materials and Methods

Materials

Polylactic acid (PLA) pellets $(M_w = 75.000 \text{ g/mol},$ $T_g = 65-75$ °C, Density@25 °C = 1.24 g/cm³) were purchased from Natureworks (Minnetonka, MN, USA). Before dissolving PLA pellets, residual moisture was

removed by holding pellets for 24 h in an air-circulating oven at 40 °C. The spice mix of sucuk containing 46.88% cumin, 46.87% red pepper, and 6.25% black pepper was kindly provided from Köfteci Yusuf (Bursa, Turkey). The extraction of EOs from the spice mix was carried out with a Clevenger-type (Sesim Chemical Laboratory, Ankara, Turkey) hydro-distillation method. For this purpose, 50 g of spice mix was mixed with 500 g of distilled water in a distillation balloon and boiled for 5 h under refux. The extracted EOs were dehydrated with sodium sulfate anhydrous and stored in a dark bottle at 4 °C. Chloroform used for the dissolving of pellets, 2,2-Diphenyl-1-picrylhydrazyl (DPPH), and methanol for antioxidant activity were purchased from Sigma-Aldrich (Germany) at analytical grade.

Preparation of Polymer Solutions

The polymer solutions were prepared by combining 5 gr of PLA with chloroform (100 mL) in a screw cap glass bottle and the mixture was held on a magnetic stirrer until completely dissolved $(-6 h)$. After the dissolution of PLA, spice mix EO was added at 0 (control), 5, 10, 15, 20, 25, and 30% depending on the PLA concentration, and the solution was mixed additional 30 min. The polymer solution was transferred into a 20 mL syringe and connected to the syringe pump (New Era, NE-4000 model, USA) of the SBS system.

Solution Blow Spinning (SBS)

The nanofber production system with the SBS technique consists of four main parts: a gas unit as compressed air, a syringe pump to control the injection rate of polymer solutions, the spinning apparatus consisting of a setup of concentric nozzles, and the static plate collector coated with aluminum foil. The feeding rate, air pressure, and collector distance were set based on preliminary experiments (data not shown) as 120 µL/min, 0.15 MPa (air velocity 7.3 m/s), and 30 cm, respectively. Nanofber production equipment was settled in a fume hood to prevent the nanofbers from dispersing in the air at which temperature and humidity were 23 °C and 52%, respectively. The air coming out of the compressor was used by passing through a dehumidifer and thus the humidity of the air was minimized. The schematic illustration of SBS and concentric nozzles were reported earlier by Medeiros et al. [[3\]](#page-11-2). The inner nozzle was positioned 2 mm beyond the outer nozzle and the distance between the nozzles was 0.5 mm. The obtained nanofbers were wrapped with aluminum foil and stored in a Ziploc freezer bag in a desiccator.

Characterization of Nanofbers

Water Solubility

The water solubility of nanofbers was determined at room temperature according to Pelissari et al. [[16](#page-11-14)] with slight modifcations. For this purpose, 20 mg of nanofber was mixed with 20 mL of distilled water at 25 °C on a magnetic stirrer for 24 h. Then, the mixture was fltered and the permeant was dried in an oven at 105 °C up to constant weight. The water solubility of nanofibers was determined from the residue after evaporation.

Mechanical Properties

The tensile properties (tensile strength and elongation at break) of nanofbers were determined with a TA-HD Plus Texture analyzer (Stable Micro Systems Co, Ltd., Godalm-ing, UK) based on ASTM standard method [\[17\]](#page-11-15). Nanofiber sheets conditioned at 53% relative humidity were cut into strips (1 cm width \times 8 cm length) and fixed to mini tensile grips. The initial gap between the tensile grips was fxed to 6 cm, and the probe speed was arranged as 2 mm/s until the break of strips. The stress–strain curves of nanofbers were used for the calculation of tensile strength (TS) and elongation at break (EAB), and at least eight nanofber strips were tested for all samples.

The burst strength (BS) and deformation (BD) of nanofbers were also investigated with a Texture Analyzer equipped with a film support ring with a 10 mm diameter and nanofibers were cut into a rectangular shape as $3 \text{ cm} \times 3 \text{ cm}$. The ball probe at 6.25 mm diameter (SMS P/0.25S) was moved through the center point of the ring at 0.2 mm/s test speed. The maximum force to rupture the flm sample was recorded, as well as the maximum distance before rupture, and all the nanofbers were tested at least in triplicate [\[18\]](#page-11-16).

Water Vapor Permeability (WVP)

The water vapor permeability (WVP) of nanofbers was determined using the gravimetric modified cup method as described by McHugh et al. [\[19\]](#page-11-17) based on the ASTM E96–80 method. Nanofber sheets were cut into a circular shape in 80 mm diameter and a digital micrometer (Mitutoyo Manufacturing) was used for determining the thickness at five different locations of nanofibers. The nanofibers were attached to poly (methyl methacrylate) cups with a 50 mm diameter opening flled with 6 mL of distilled water. Three diferent cups for each type of nanofber were placed in a cabinet containing anhydrous silica (0% relative humidity). The water loss of cups was recorded every 2 h intervals with an analytical balance, and the WVP of nanofbers was calculated as $g.mm/m² h.kPa$.

Scanning Electron Microscopy and Fiber Diameter

Nanofbers were cut into small pieces and glued onto aluminum stubs with a double-sided adhesive carbon band. Then, samples were coated by a sputter coating (Leica, EM ACE600 model, USA) with gold–palladium at 25 mA. Nanofibers were visualized by field emission scanning electron microscopy (SEM) (Carl Zeiss, Gemini SEM360 model, Germany) at an accelerated voltage of 10 kV. The SEM micrographs of nanofbers were further analyzed by ImageJ software to calculate fber diameter distribution from at least 100 random measurements and the average fber diameter was also determined with a Gaussian curve to ft by OriginLab software (OriginPro 8.5 version, USA).

Atomic Force Microscopy (AFM)

The surface topography of nanofibers was visualized with atomic force microscopy (Nanosurf, Flex AFM model, Switzerland). A rectangular cantilever with a spring constant of 20–60 N/m was positioned over the sample, and $15 \mu m \times 15 \mu m$ images were captured from the samples. The roughness of samples was statistically calculated by device software based on the average of the absolute value of the height deviations from a mean surface, R_a and the rootmean-square average of height deviations taken from the mean data plane, R_q.

Contact Angle Measurement

The wettability of nanofbers enriched with EO was determined by a sessile-drop method using contact angle measurement (Athension Theta, Biolin Scientifc, Sweeden). Distilled water $(5 \mu L)$ was dropped onto nanofibers, and then the contact angle between the drop and surface of nanofbers was measured and the related images were captured.

Thermal Properties

The thermal properties of nanofibers were characterized using diferential scanning calorimeter (DSC) and thermogravimetric analyses (TGA). Before DSC and TGA analyses, nanofbers were conditioned for 3 weeks in a desiccator containing silica. Nanofbers (4–5 mg) were analyzed under a nitrogen atmosphere at a fow capacity of 20 mL/min with a DSC device (TA Instruments, Discovery DSC251 model, USA). Temperature scanning ranged from − 40 to 200 °C at a heating rate of 5 °C/min. For TGA, approximately 10 mg of nanofber was weighed and scanned from 20 to 600 °C at a heating rate of 10 °C/min with a TGA device (TA Instruments, SDT650 model, USA) under nitrogen as purge gas at 20 mL/min fow rate.

X‑ray Difraction (XRD)

The effect of EO incorporation on the crystallinity of PLA nanofbers was determined with a wide-angle XRD device (Bruker, AXS Discovery D8, MA, USA) equipped with Cu-Kα radiation source (λ = 1.546 Å). The incidence angle as 2 θ ranged from 4 to 40° at a scanning rate of 2° min⁻¹. All samples were held in a desiccator containing silica for 2 weeks to remove moisture.

FTIR Spectrum

The FTIR spectrum of nanofibers containing EOs was determined using an FTIR device (TA, Nicolet iS50, CA, USA). The samples were directly placed in the sample chamber of the device and the spectrums were taken directly with the help of an ATR diamond crystal at a scan resolution of 2 cm−1 between the wavenumber range from 4000 to 400 cm−1. All samples were stored in a desiccator with silica for minimizing the moisture content of the samples.

Antioxidant Capacity

The antioxidant capacity of nanofbers enriched with spice mix EO was determined with the 2,2-diphenyl-2-picrylhydrazyl (DPPH) scavenging activity. For this purpose, 100 mg of nanofber was mixed with 5 mL of methanol in a centrifuge tube and held in a shaking water bath at 25 °C for 2 h. Samples were then centrifuged at 5000×*g* for 10 min and 0.5 mL of supernatant was vigorously mixed with 3.5 mL of 0.06 mM DPPH and held for 30 min in a dark environment. The absorbance of the solution was measured at 517 nm with a spectrophotometer (Rigol, Ultra 3660 model, India). The DPPH scavenging activity was calculated with the following equation:

$$
AC(\%) = \frac{Abs_{Control} - Abs_{Sample}}{Abs_{Control}} \times 100
$$
 (1)

where *AC* is the antioxidant capacity, $Abs_{Control}$ is the absorbance of the DPPH solution and Abs_{Sample} is the absorbance of test samples.

Antibacterial Activity

The antibacterial activity of nanofibers against *E. coli* ATTC35218 and *S. aureus* ATCC6538 was evaluated for the control and samples containing 25 and 30% spice mix EOs which had the lowest fber diameters, better surface topography, and the highest antioxidant capacity. The Agar disc difusion method did not show inhibition zones probably due to the hydrophobic properties of PLA-based nanofbers. Therefore, the bacterial colony counting method was carried out as described earlier by Lin et al. [\[20](#page-11-18)] for nanofbers. For this purpose, 400 mg of nanofbers were cut with a sterile razor blade and each surface of nanofbers was held in a UV cabin for 30 min to sterilize the surface. *S. aureus* and *E. coli* were frstly incubated in Mueller Hinton broth at 37 °C for 24 h and inoculum was re-suspended to provide a final density of 5.0×10^5 CFU/mL in phosphate-buffered saline according to 0.5 McFarland turbidity standard. The sterilized nanofbers (~400 mg) were immersed in bacterial suspension in 20 mL of conical falcon tubes, and tubes were incubated in a shaker at 37 °C for 24 h. Dilutions between 101 and 109 were prepared by adding 1 mL culture into 9 mL of phosphate bufer saline solution, and then, 0.1 mL of the diluted culture were spread onto a Mueller Hinton agar plate and incubated at 37 °C for 24 h. The colony-forming unit (CFU) were counted and the antibacterial activity of nanofbers was defned as the following equation:

$$
Anibacterial activity(\%) = \frac{A - B}{A} \times 100\tag{2}
$$

where *A* and *B* are the number of colonies before and after nanofbers are added, respectively [[1\]](#page-11-0).

Statistical Analysis

The production process of nanofbers was repeated three times and all analyses were performed at least in duplicate. The results were analyzed by one-way analysis of variance (ANOVA) using the SPSS 21.0 package program (IBM, Chicago, IL, USA) and results were presented as mean \pm standard deviation. Signifcant diferences between the means were evaluated with Duncan's multiple range test at a significance value of $p < 0.05$.

Results and Discussions

Water Solubility

The water solubility of nanofibers is crucial in pharmacy, whereas insoluble nanofibers are desired for packaging applications. The efect of EO incorporation on the water solubility of PLA-based solution blow-spun nanofibers is summarized in Table [1.](#page-4-0) PLA-based nanofibers displayed low water solubility between 3.79 and 17.78 mg/L, probably due to the hydrophobic nature of PLA and EO. However, the increase of EO concentration signifcantly increased the water solubility of nanofibers, except for 25% EO concentration, which showed a similar value with 10% EO addition, and the highest water solubility was observed from 30% **Table 1** Water solubility and mechanical properties of solution blow spun nanofbers enriched with sucuk spice mix essential oil

Values are mean + standard deviation

WS: Water solubility; *TS* Tensile strength; *EAB* Elongation at break; *BS* Burst strength; *BD* Burst deformation; *PLA* Polylactic acid; *EO* Essential oil obtained from sucuk spice mix

^{a–e}Means within the same column with different letters are different ($p < 0.05$)

EO added nanofbers. The low water solubility of solution blow spun nanofbers could be attributed to the hydrophobic nature of PLA. EOs are also hydrophobic, but the increasing concentration of EOs caused the migration to water during the solubility test, and hence increased the water solubility results, except for 25% EO added nanofbers.

Mechanical Properties

The mechanical properties of polymer materials, either flm or nanofber sheet, are considered a key factor in packaging applications, and the mechanical behavior of nanofber sheets depends highly on processing conditions, composition, fber structure, interactions between fbers, and individual properties of fber constituents [[21\]](#page-11-19). Tensile strength (TS), elongation at break (EAB), burst strength (BS) and burst deformation (BD) results of solution blow spun nanofbers enriched with sucuk spice mix EO are summarized in Table [1.](#page-4-0) The TS values of solution blow spun nanofbers signifcantly decreased with increasing EO addition, and the neat PLA nanofbers showed the highest TS value, while the nanofbers enriched with 30% EO had the lowest TS value. The decreasing trend of TS values could be attributed to the average fber diameter, which decreased with increasing EO incorporation. The addition of EO improved the EAB values probably due to the plasticizer efect of EO, and a similar situation was also observed from the BD values. The BS values of nanofbers display the strength of nanofber sheets against vertical forces, and the increasing EO content signifcantly decreased the BS values, except for the 15% EO added sample which showed similar BS value with neat PLA nanofber sheet, probably due to similar fber diameters. The obtained mechanical results were also in accordance with the atomic force microscopy results, in which the addition of EO resulted in a higher surface roughness, and therefore mechanical properties decreased with increasing EO. In addition, the crystallinity of nanofbers generally increased when compared to neat PLA nanofber, except for 10% EO addition, and the highest crystallinity was determined from a 15% EO incorporated sample (Table [3\)](#page-9-0), which supported to higher BS value. Our results were also in accordance with the results of Promnil et al. [\[22](#page-11-20)] who determined the mechanical properties of electrospun PLA nanofbers.

Water Vapor Permeability

The water vapor permeability (WVP) of a polymer flm or nanofber sheet is an important property when choosing the packaging material to control moisture transfer between the food and the environment. In general, packaging materials with low WVP are preferred for food packaging applications. However, although nanofber sheets are produced like polymer flms, they cannot be expected to have low WVP. Because nanofbers have a porous structure on a nanoscale and water vapor can move through these pores. The WVP of neat PLA and EO-incorporated PLA nanofber sheets is summarized in Table [2](#page-5-0). The neat PLA nanofibers showed the highest WVP, and the incorporation of EO into nanofbers signifcantly decreased the WVP, and the lowest value was observed from the 30% EO incorporated nanofber sheet $(p<0.05)$. The surface hydrophobicity and morphology of nanofbers could directly be attributed to WVP, as well as average fber diameter and crystallinity. The hydrophobic structure of the PLA and hydrophobicity of EO, and surface roughness of nanofbers were in accordance with the barrier properties against water vapor. Moreover, the increasing crystallinity with increasing EO addition also improved the barrier properties of nanofbers via behaving as a nucleating element and occurring in impermeable regions onto nanofber sheets [[23](#page-11-21)]. In a study conducted by Yang et al. [[24\]](#page-11-22) the WVP of solution blow spun gelatin nanofibers was determined higher than our results, due to the hydrophilic structure of gelatin. In another study, the incorporation of EO into electrospun poly-vinyl alcohol (PVA) nanofbers increased the WVP, and this increase was related to the

Table 2 Water vapor permeability, antioxidant capacity, average fber diameter and surface roughness factor of solution blow spun nanofbers enriched with sucuk spice mix essential oil

Values are mean \pm standard deviation

WVP Water vapor permeability (g.mm/m².h.kPa); *AC* Antioxidant capacity; *AFD* Average fiber diameter;

Ra Surface roughness factor; *PLA* Polylactic acid; *EO* Essential oil obtained from sucuk spice mix

a^{-g}Means within the same column with different letters are different ($p < 0.05$)

Fig. 1 Scanning electron microscope image of neat PLA nanofbers obtained with solution blow spinning

increasing hydrophilicity of nanofibers due to the emulsi-fication process [[25](#page-11-23)].

SEM Analysis

The production of PLA-based nanofibers with the SBS process was successfully achieved on a nanometer scale. The microstructure of neat (control) and EO-enriched PLA nanofibers, as well as fiber diameter distribution, are depicted in Figs. [1](#page-5-1) and [2](#page-6-0), respectively. The neat PLA nanofbers showed a rough surface when compared with EO-enriched nanofbers. As reported earlier by Oliveira et al. [\[6](#page-11-5)], the viscosity of polymer solutions is responsible for smooth fber morphology. SEM micrographs revealed that beads and flms as an indicator of poor nanofber spinning conditions were not observed for all samples. This result indicates that the chosen polymer concentration and spinning variables such as air pressure, working distance, and feed rate, were suitable for the production of nanofbers without beads and flms. However, nanofbers enriched with EO presented a homogeneous aspect (Fig. [2\)](#page-6-0), except for the neat PLA nanofbers (Fig. [1\)](#page-5-1), which displayed a rough nanofber surface. This could be attributed to the plasticizing efect of EO [\[14\]](#page-11-12) and the evaporation rate of polymer solution, in which EO addition changed the evaporation rate of solutions [\[26](#page-11-24)]. Similar results were also reported by Bonan et al. [\[13\]](#page-11-11) who produced PLA-based SBS nanofbers containing Copaiba oil, and by Miranda et al. [\[26](#page-11-24)] for polystyrene-based SBS nanofbers with orange oil. The average fber diameter was signifcantly afected by the EO incorporation ($p < 0.05$). EO added up to 20% significantly increased the fber diameter when compared with neat PLA nanofber. However, nanofbers containing 25 and 30% EO showed the lowest fber diameter. The fnal morphology of nanofbers obtained by SBS can mainly be changed by the feed rate, polymer concentration, air pressure, and distance between the nozzle and collector [\[6](#page-11-5)]. All these parameters were kept constant, and only EO concentration varied which resulted in the variation of average fber diameter. The variation in fber diameter could probably be related with the change in viscosity and evaporation rate of polymer solution. The production of nanofbers from viscous polymer solutions is more difficult, less effective and stable than the viscous solutions which produced from polymers with poor compatibility [[7,](#page-11-25) [8\]](#page-11-6).

Surface Topography

The surface topography of solution-blown spun nanofbers are shown in Figs. [3](#page-6-1) and [4](#page-7-0) for control and EO enriched, respectively. The surface of nanofbers was the most uneven, but EO incorporation caused a slightly smoother surface, except for 30% addition which resulted in an uneven surface. The surface roughness factor (R_a) of nanofibers without EO was determined as 331 nm, and 5, 10, 15, 20, 25, and 30% EO-incorporated nanofbers showed 399, 381, 356, 256, 295, and 590 nm R_a values, respectively (Table [1\)](#page-4-0). In general, the surface roughness of nanofbers

Fig. 2 The effect of essential oil addition on the morphological structure of PLA nanofbers obtained with solution blow spinning

Fig. 3 The surface topography of neat PLA nanofbers obtained with solution blow spinning

is closely related to the average fber diameter. Although the lowest fber diameter was observed from the sample containing 30% EO, the highest surface roughness was obtained. Our results showed that the R_a values of nanofibers were similar to values obtained from electrospun nanofbers [[27,](#page-12-0) [28](#page-12-1)] indicating that SBS is a successful technique for the production of nanofbers in nanoscale. Agrawal and Pramanik [\[28](#page-12-1)] stated that the suitable surface roughness of nanofbers for cell attachments and growth in tissue engineering applications should be in the range of 262–435 nm. In another study, the incorporation of chitosan in electrospun PVA nanofbers decreased the surface roughness probably due to lower fber diameter [[29](#page-12-2)]. However, Milleret et al. [\[30\]](#page-12-3) reported that the increasing starch ratio in the nanofber mats increased the fber diameter and hence surface roughness.

Contact Angle

The hydrophilicity or hydrophobicity of a surface can be determined with the contact angle of a water drop, and this analysis gives signifcant information about the wettability of the surface. If the contact angle is lower than 90°, the surface is wettable, whereas if a higher angle than 90° is observed, the surface is non-wettable and the material is completely hydrophobic. The higher contact angle is generally accepted for food packaging applications due to low water sensitivity and good barrier against water vapor [\[31](#page-12-4)]. The surface wettability and hydrophobicity of EO-incorporated PLA nanofbers are illustrated in Fig. [5](#page-8-0)a. It is well known that PLA contains a large number of hydrophobic ester groups, and therefore all the nanofbers had a hydrophobic surface structure showing a higher contact angle than 100° [[32](#page-12-5)]. Moreover, the increasing ratio of EO from the sucuk spice mix caused an increase in contact angle results and the highest contact angle was observed from the sample containing 30% EO. The addition of hydrophobic compounds such as EOs in the PLA-based nanofbers improved the hydrophobicity and less wettable surfaces were obtained. Furthermore, the crystallinity of solution blow spun PLA

nanofbers was increased based on XRD and DSC results, and the higher the crystallinity, the more difficult the diffusion and transformation of water [[33\]](#page-12-6). Our results were in accordance with the WVP and AFM images, in which WVP decreased and nanofber sheets became more smooth depending on the EO concentration.

Thermal Properties of Nanofbers

DSC curves of nanofibers with and without EO are illustrated in Fig. [6](#page-9-1) and related results are summarized in Table [3](#page-9-0). The neat PLA as control displayed glass transition (T_p) , crystallization (T_c) , and melting (T_m) temperatures of 70.79, 99.66, and 174.16 °C, respectively, which are similar values reported in the literature [\[13](#page-11-11), [14](#page-11-12), [34\]](#page-12-7). These temperatures showed small shifts with the addition of EO at diferent ratios. In general, T_g and T_c showed a higher decrease with increasing EO concentration when compared to T_m which was similar to the control. The decreasing tendency of T_g and T_c could be explained by the plasticizing effect of EO $[12, 12]$ $[12, 12]$ $[12, 12]$ [14](#page-11-12)]. EO incorporation into the PLA-based nanofbers may improve the fexibility of PLA, and also afect the crystallinity compared to neat PLA. Souza et al. [\[12](#page-11-10)] reported that

Fig. 5 Contact angle (**a**), XRD (**b**), FTIR (**c**) and antimicrobial properties (**d**) of solution blow spun PLA-based nanofbers containing various amounts of essential oils

nanofbers produced with SBS showed a higher degree of crystallinity than electrospun nanofbers due to better orientation during the SBS process, and also they stated that this was because solvent evaporation was faster in nanofber production with SBS.

Another thermal analysis, thermo-gravimetric analysis (TGA) is performed to indicate the thermal decomposition of nanofbers. TGA curves and related results are presented in Fig. [6](#page-9-1) and Table [4,](#page-10-0) respectively. It can be observed that there were two main degradation temperatures for neat PLA and EO-added nanofbers. The frst degradation was due to the evaporation of water and EOs, whereas the second degradation was due to the degradation of PLA. The onset temperature of frst degradation and related weight loss for control nanofber was recorded as 46.98 °C and 2.34%, respectively. The addition of EO up to 15% did not afect the onset temperature of the frst degradation, but higher concentrations of EO such as 20, 25, and 30% displayed 57.02, 51.16, and 73.77 °C of onset temperatures for the first degradation, respectively. The weight loss of samples at the frst decomposition temperature ranged between 0.51 and 6.77%. EO incorporation caused higher weight loss than control, except for the 5% EO added sample, due to evaporation of water and EO. The main thermal degradation occurred in the second stage, at which the onset temperatures ranged from 195.04 to 246.82 °C, and the weight loss was between 90.31 and 97.46%. The highest weight loss in the second stage was observed from the control, and it decreased with increasing EO incorporation, probably due to the loss of EO in the frst stage. The thermal degradation behavior of nanofbers is mainly dependent on the mobility and compactness of polymer chains [[15](#page-11-13), [35](#page-12-8)]. The interactions between EO and PLA, the most probable hydrogen bonds, caused the improvement of the regularity and compactness of polymer chains, and hence thermal stability increased [[15\]](#page-11-13). In addition, the increased thermal stability with EO concentration could cause an increasing residue after thermal treatments.

XRD

The effect of EO incorporation on the crystallinity of solution blow spun PLA nanofbers was illustrated in Fig. [5b](#page-8-0). The XRD pattern of neat PLA nanofiber exhibited typical broad peaks at 2θ angles around 15° and 30° indicating an amorphous structure. The addition of EO into PLA nanofbers did not cause shifting or occurring new peaks in the

Fig. 6 DSC and TGA curves of solution blow spun PLA-based nanofbers containing various amounts of essential oils

pattern. However, the intensity of the peaks was increased depending on the EO concentration indicating that the crystallinity of the nanofber sheets obtained by solution blowing decreased probably due to the rapid evaporation of solvent during air pressurizing [[36\]](#page-12-9). A similar observation was reported by Wang et al. [\[37](#page-12-10)] for *Perilla* EO-loaded electrospun PLA nanofbers showing that the crystallinity strength of EO-loaded PLA nanofiber sheets was higher than that of neat PLA nanofbers. This meant that PLA and EOs had good compatibility and did not greatly alter the structural organization of the polymer molecule. These results were well supported by the mechanical and barrier properties of solution blow-spun PLA nanofbers, as well as surface hydrophobicity. Mori et al. [\[38](#page-12-11)] found that the XRD patterns of EO-loaded PLA nanofbers did not signifcantly change and only peak intensity variations were observed.

FTIR Spectrum

FTIR analysis of solution blow spun PLA nanofbers loaded with sucuk spice mix EO was performed for the investigation of intermolecular interactions between PLA and EO, and all the nanofber sheets displayed a similar spectrum indicating no obvious changes in the molecular and chemical structure after the EO addition (Fig. [5](#page-8-0)c). The FTIR spectrum of EO from sucuk spice mix revealed multi peaks at the wavenumbers of 2962.28, 2927.42, and 2870.87 cm−1 relating with the symmetric and asymmetric –C–H– stretching vibrations of hydrocarbons ($-CH_3$ – and $-CH_2$) attached to oxygen (ethers) [[39](#page-12-12)]. The peaks located at 1704.12, 1607.83, 1463.71, 1212.56, and 840.06 cm−1 were relevant to aliphatic esters, aromatic cyclohexenes, aromatic cycles, carbonyls, and sulfate esters, respectively [\[40](#page-12-13)]. The neat PLA nanofber displayed characteristic peaks corresponding to –C=O– ester carbonyl (1754 cm−1), –C–H– deformation (1452 and 1382 cm⁻¹), and -C–O– stretching (1128 and 1085 cm^{-1}) [[41\]](#page-12-14). The backbone structure of PLA nanofibers remained constant, but the intensities of some peaks increased depending on the EO addition. For instance, the intensity of the peak at 2994 cm^{-1} increased with EO incorporation, and the main peak of EO at 1704 cm−1 shifted to 1754 cm−1 for nanofbers, as well as increasing peak intensity. This could be attributed to hydrogen bonding between the amide I bands of PLA and EO. The increasing peak intensities and occurring new peaks for PLA nanofbers

T_g Glass transition temperature (°C); ΔH_g Heat of glass transition (J/g); *T_c* Crystallization temperature (°C); *ΔHc* Heat of crystallization (J/g); *Tm* Melting temperature (°C); *ΔHm* Heat of melting (J/g). *PLA* Polylactic acid; *EO* Essential oil obtained from sucuk spice mix

Table 4 Thermo-gravimetric analysis results of PLA based SBS nanofibers containing sucuk spice mix essential oil

 $T_{d1 onset}$ Onset temperature of the first thermal degradation (°C); $T_{d2 onset}$ Onset temperature of the second thermal degradation (°C); ΔW_1 Mass loss of the first thermal degradation (%); ΔW_2 Mass loss of the second thermal degradation (%); *PLA* Polylactic acid; *EO* Essential oil obtained from sucuk spice mix

PLA+%30 EY 73.77 5.80 220.85 91.76 2.44

incorporated with sucuk spice mix EO indicated that the sucuk spice mix EO could be successfully loaded in solution blow spun PLA nanofibers.

Antioxidant Activity

The oxidation reactions occurring in the food which contact with oxygen cause the deterioration of lipids and the production of free radicals in foods. The scavenging of these free radicals is very important and in this regard, the antioxidant activity of EO-incorporated nanofbers was measured by the DPPH scavenging activity. The EO obtained from the spice mix containing cumin, red and black pepper contains high content of phenolic compounds and therefore displays antioxidant activity. As summarized in Table [2](#page-5-0), the neat PLA nanofber did not have antioxidant activity, whereas the EO incorporation, as expected, signifcantly increased the antioxidant activity of solution blow spun nanofibers $(p < 0.05)$. The increment of EO at 5% caused at least a 4.57% increase in the antioxidant activity of nanofibers. However, the highest amount of EO addition (30%) caused a 50.68% increase in antioxidant activity when compared with the lowest EO addition (5%). Our results were in accordance with the results of Marouf et al. [\[15](#page-11-13)] for electrospun PLA nanofbers enriched with thyme EO.

Antibacterial Activity

The EOs of sucuk spice mix contains various compounds, such as thymol, carvacrol, cumin aldehyde, terpinene, and limonene [[42\]](#page-12-15) and it is also well known that these compounds impart antimicrobial activity to EOs against various microorganisms. Due to the hydrophobic nature of PLA and EOs, the agar disc difusion method did not give results for solution blow spun PLA nanofbers, and antibacterial activity against Gram-negative *E. coli* and Gram-positive *S. aureus* bacteria was evaluated via colony counting method (Fig. [5](#page-8-0)d). The antibacterial activity of nanofbers containing the highest amount of EO (20, 25, and 30%) was determined because these nanofbers displayed the highest antioxidant activity and the lowest average fber diameter (25 and 30%). It is clear in Fig. [5](#page-8-0)d that the increasing EO concentration in nanofbers signifcantly increased the growth inhibition of *E. coli*, whereas 25 and 30% EO-incorporated nanofbers showed similar antibacterial activity against *S. aureus*. The increasing concentration of EO caused higher difusivity in the broth medium, which is consistent with the water solubility results of nanofbers. Moreover, the growth inhibition of nanofbers against *S. aureus* was higher than in *E. coli*. This might be related to the cell wall composition of Gram-negative and Gram-positive bacteria. Thus, *E. coli* has a thin peptidoglycan layer and an outer layer of lipoproteins, lipopolysaccharides, and phospholipids, while the cell wall of *S. aureus* comprises a peptidoglycan layer with lots of pores. The porous cell wall structure of *S. aureus* makes it easy to be permeated by active compounds of EO [\[43](#page-12-16), [44](#page-12-17)].

Conclusions

In this study, novel solution-blown spun nanofbers containing various amounts of EO were successfully fabricated and the mechanical, water barrier, microstructural, topographical, chemical, thermal, antioxidant, and antimicrobial characterization of nanofbers were performed. The hydrophobic nature of PLA caused fewer water-soluble nanofbers, but higher EO concentration increased the water solubility. Uniform and bead-free nanofbers were obtained by the SBS technique and the addition of EO decreased the average fber diameter. EO incorporation decreased the surface roughness and caused a smoother surface due to plasticizing efect of EO. The XRD results were well ftted to the mechanical and barrier properties of nanofbers, as well as DSC results. FTIR spectra of neat and EO-loaded PLA nanofbers revealed that PLA nanofbers could be successfully loaded with sucuk spice mix EO. DSC and TGA curves of nanofbers revealed that the melting and glass transition temperatures decreased, whereas the thermal stability of nanofbers increased with EO addition. Nanofbers containing EO displayed antioxidant activity and increasing EO concentration signifcantly improved the antioxidant activity. Moreover, nanofbers containing 20, 25, and 30% EO had good antibacterial activity against *E. coli* and *S. aureus*. It can be concluded from these results that the SBS technique can be successfully used for the production of PLA-based nanofber enriched with EOs, and these nanofbers can be applied not only for drug delivery systems but also for food packaging applications.

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