**ORIGINAL PAPER**



# **An Experimental and Metamodeling Approach to Tensile Properties of Natural Fibers Composites**

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Accepted: 14 June 2022 / Published online: 11 July 2022 © The Author(s), under exclusive licence to Springer Science+Business Media, LLC, part of Springer Nature 2022

#### **Abstract**

The present work presents an analysis of the tensile properties of Palm as well as Lufa natural fber composites (NFC) in high density polyethylene (HDPE), polypropylene (PP), Epoxy, and Ecopoxy (BioPoxy 36) matrixes, taking into consideration the efect of fbers volume fraction variation. Finite element analysis i.e. representative volume element (RVE) model with chopped random fber orientation was utilized for predicting the elastic properties. Tensile test following ASTM D3039 standard was conducted. Artifcial neural network, multiple linear regression, adaptive neuro-fuzzy inference system, and support vector machine were implemented for defning the design space upon the considered parameters and evaluating the reliability of these machine learning approaches in predicting the tensile strength of natural fbers composites. Furthermore, BioPoxy 36 with 0.3 lufa fbers exhibited the highest tensile strength. Finite element analysis (FEA) fndings profusely agreed with the experimental results. ANFIS Machine Learning (ML) tool showed least prediction error in predicting tensile strength of natural fbers composites.

**Keywords** Palm fbers · Lufa fbers · Tensile properties · Finite element analysis · Machine learning

## **Introduction**

Natural fbers' composite material attained a notable attention in materials' science area due to their profuse advantages i.e. light weight, low density, high strength to weight ratio, environmental-friendly, low cost, and their availability in the nature. Moreover, natural fbers like sisal, coir, kenaf, palm, bamboo, jute, hemp, and lufa were vastly considered in engineering and science researches. In consequence of the environmental awareness throughout the last two decades, development of a recyclable and environmental-friendly

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composite material has drastically increased [\[1\]](#page-15-0). Discovering a new alternative material to the widely utilized materials like metals, synthetic fber composites, and alloys has become the prevalent research area in academia as well as in the industry  $[2, 3]$  $[2, 3]$  $[2, 3]$  $[2, 3]$ . Furthermore, studies on the application of natural fbers i.e. wood, pineapple, lufa, feather, palm, jute, animal silk, and so on became widespread among scientists and engineers, due to their merits, such as; low electricity usage for producing these natural fbers [\[4\]](#page-15-3), less tool wear comparing with that involved in processing synthetic fbers' composites [[5–](#page-15-4)[7\]](#page-15-5), less harmful gases emission when burned at end of life or exposed to high-heat [\[8](#page-15-6)], lower hazards throughout the production process, cheaper than synthetic fbers [[9](#page-15-7)], notable strength, low density, and high stifness [\[10](#page-15-8)]. Recently, natural fibers composites made of kenaf, jute, sisal, hemp, fax [[10](#page-15-8), [11](#page-15-9)] have been increasingly involved in various engineering felds i.e. circuit boards, building materials, automotive, etc. [[12–](#page-15-10)[14\]](#page-15-11). Short service life, high water and moisture absorption, micro-organisms and sunrays degradation are main constraints of expanding the production of natural fber composites (NFCs) [\[15](#page-15-12)], yet a suitable development of NFCs is able to let these green composite to emerge into new markets and attain bigger demand [\[16](#page-15-13)[–19](#page-15-14)].

World widely, date palm trees produce more than 8 million tons of date fruits every year, thereby keeping tones of fbrous wastes. A natural woven mat surrounds the trunk of date palm, the aforementioned is generally involved in ropes and baskets production. Cucurbitaceae family includes a subcategory called lufa, its unripe fruit is used in Chinese, Indian, and Vietnamese dishes. While its mature fruit is widely used as a shower sponge and further household utilizations. The three-dimensional network structure of lufa leads to its high strength, toughness and stifness [[20,](#page-15-15) [21](#page-15-16)]. Taban et al. [\[19\]](#page-15-14) proposed replacing synthetic fbers by palm fbers in acoustic isolation applications. Shalwan et al. [[22\]](#page-15-17) mentioned an increase in the tensile strength of epoxy from 58 to 68 MPa by adding date palm fbers. Ibra-him et al. [[23\]](#page-15-18) observed that increasing the date palm fibers' volume fraction up to 0.5 increases the tensile strength and young's modulus. Shen et al. [[24](#page-15-19)] highlighted the notable behavior of lufa natural fbers composites in acoustic and vibrations applications. While Mani et al. [[25\]](#page-15-20) observed an increase in the tensile strength of lufa NFC by increasing fbers' content up to 40% in epoxy matrix. The signifcant growth in NFCs utilization evidences the requirement of an efficient design and development of the composites in order to achieve optimal characteristics. Researchers in natural fbers composites area applied computational techniques i.e. numerical and analytical, in order to simulate thermal, physical, and mechanical properties while developing a new NFC [\[26](#page-15-21)[–28](#page-15-22)]. Moreover, studies mainly focused on predicting the micromechanical properties of NFCs, yet, simulation fndings exhibited signifcant prediction accuracy and profusely agreed with the experimental results. For instance, Parsad et al. [[29](#page-15-23)] concluded that the fnite element analysis results of lufa NFC agreed with the experimental fndings. Similarly, Sowmya et al. [[30](#page-15-24)] spotted the light on the strong capability of Finite element analysis in predicting the mechanical characteristics of hemp natural fbers' composite, thus, fnite element analysis (FEA) fndings displayed signifcant agreement with experimental results. However, representative volume element estimates the characteristics of a composite material unit cells at macro-, nano-, and micro-scale. Representative volume element is the major efective homogenized multiscale FEA, therefore it has to be frstly applied for the analysis of composite materials with complex structures like NFC which contain diverse length scales [[31](#page-15-25)[–36](#page-15-26)].

Previously machine learning was utilized for detecting C60 solubility, however it is currently involved in predicting molecular characteristics of designed materials. Despite the fact that experimental testing is signifcantly essential for developing a new material, machine learning contributes in decreasing the cost as well as the computational time throughout an experiment, as the required tools for running the machine learning algorithms are free to access and easily

available [[12](#page-15-10), [37](#page-15-27)[–40](#page-15-28)]. Recently, artifcial intelligence was applied by several researchers in composite materials and natural fbers composites. Antil et al. [\[41\]](#page-15-29) utilized artifcial neural network (ANN) and RSM to study the erosion behavior of S Glass composites, inputs included nozzle diameter, impingement angle, and slurry pressure. Pati et al. [\[42\]](#page-15-30) applied ANN for predicting the wear behavior of glass/ epoxy composites, input parameters consisted of; erodent temperature, erodent size, RBD content, impingement angle, and impact velocity. While Baseer et al. [\[43](#page-15-31)] assigned shear strength, failure stress and strain, tensile modulus, and tensile strength as input parameters for evaluating the interfacial and tensile properties of hybrid composite material. Atuanya et al. [\[44](#page-15-32)] emphasized the reliability of using ANN to predict the mechanical behavior of NFCs, authors implemented artificial neural networks for predicting the mechanical properties of date fbers' reinforced low-density polyethylene (recycled), input data consisted of fbers' weight percentages, while the output was tensile strength, young's modulus, elongation, fexural modulus, and hardness. Daghigh et al. [[45\]](#page-15-33) utilized K-Nearest Neighbor Regressor for predicting the heat defection temperature of latania NFCs, pistachio shell NFCs, and date seed NFCs. Also, Daghigh et al. [[46\]](#page-15-34) applied decision tree regressor and adaptive boosting regressor for studying the fracture toughness of the aforementioned natural fbers composites. Garg et al. [\[47\]](#page-15-35) implemented extreme machine learning to investigate the mechanical factor of Jute as well as Coir natural fbers composites. Wang et al. [[48\]](#page-15-36) used random forest machine learning approach for analyzing the acoustic emission of Flax NFCs.

The aim of this study was to investigate the elastic properties of palm and luffa NFCs in BioPoxy 36, Epoxy, Polypropylene, and High-Density Polyethylene matrixes by modeling and simulating the micro-mechanical properties of NFC using FEA representative volume element (RVE) chopped fbers' orientation, validating the optimal confguration by experimentally testing it and validating its mechanical properties (Tensile ASTM D3039), and developing an Artifcial Neural Networks, Multiple Linear Regression, Adaptive Neuro-Fuzzy Inference System, and Support Vector Machine based Metamodel. Impact of increasing fber content was identifed through assigning multiple fbers volume fractions i.e. 0.1, 0.2, 0.3.

## **Experimental Procedure**

In this research, date palm meshes and lufa were considered as reinforcements, while BioPoxy 36, epoxy, polypropylene, high density polyethylene were selected as a matrix. This section describes main stages of tensile testing samples' preparation, which include; fbers' extraction, matrixes materials' supply, molds' preparation, and natural fbers

composite specimens' preparation. Figure [1](#page-2-0) describes the main stages of tensile testing procedure.

## **Materials and Fibers' Preparation**

Date palm meshes that surround the stem were extracted from a palm tree located in north Lebanon, the fbers were kept to dry for 72 h and then washed with cold water in order to remove all the dust and impurities. Next, the cleaned palm fbers were dried through placing them under the sunlight on a mosquito net for 23 days [[49](#page-15-37)]. The aim of using a mosquito net was to let the air ventilate the bottom of palm fbers, and avoid any water drop from staying below the fbers, which thereby could harm the fbers by creating moisture. Lufa sponge was supplied from a local store, it was dry and peeled, its length was 46 cm and its average diameter was 16 cm. Regarding the matrixes, BioPoxy 36 was ofered by

the manufacturer of this green resin i.e. EcoPoxy, Canada. Aquaglass epoxy resin and its hardener were supplied from Colortek, Lebanon. Moreover, polypropylene 528 k and high-density polyethylene F00952 were supplied from Sabic.

#### **Molds Development**

Since both thermoplastic and thermoset matrixes are considered in this research, and each of the aforementioned has its specifc preparation technique, therefore it was compulsory to utilize two diferent mold's types. For natural fbers' reinforced thermosets, 36 silicon molds were developed by the following steps: (1) silicon sealant was added into water/dish soap mixture, (2) putty was then mixed well till it reached an unsticky dough structure, (3) next, a wooden pattern that has same dimensions of the specimens was inserted in the putty and pressed well around the corners, (4) lastly, the putty



<span id="page-2-0"></span>**Fig. 1** Palm and lufa NFCs specimens' preparation and testing

was rested on a plastic tray for 15 min to dry before removing the pattern. Meanwhile for the thermoplastic NFCs, the mold required a female mold along with its male part that contributes in compressing the molten NFC till it solidifes. Hence, a plywood board was trimmed into small parts that included the inner and outer walls of the mold, and their base. Thereby, these parts were assembled using wood screws to create 15 female molds that consist of 3 cavities each.

### **Samples Preparation**

Luffa as well as date palm fibers were chopped using a waring blender in order to have a homogeneous mixture with the considered matrixes, the approximate fbers' dimensions were 1 cm length and 0.5 mm diameter, and the desired fbers' orientation was random (Fig. [1](#page-2-0)). In terms of NFCs with thermoset matrixes, the resin was mixed with its hardener for 5 min, the resin to hardener mixing ratio was 4:1 for BioPoxy 36 and 1.8:1 for epoxy. 0.1, 0.2, 0.3 fbers' volume fractions were considered in for the experimental specimens. Since the molds may have a small variety in their heights, the specimens' thickness was ensured using a tiny stick marked on 5 mm from its tip. First, coat of silicone spray was applied to the bottom of the mold and a small layer of resin/hardener mixture was poured, then the fbers were added upon a specifc volume fraction i.e. 0.1, 0.2, or 0.3, next, the remaining quantity of the resin was added and the fbers were pushed downward in order to release any available air bubbles. The specimens were prepared at a room temperature of 21 ℃ and humidity of 66%, all 36 specimens were kept for 1 week to fully cure. Regarding the specimens with PP matrix, the granules were put in an aluminum foil sprayed with silicone realizing agent, and then placed in a toaster at 230 ℃ for 5 min to melt, then the fbers were added into the molten plastic and kept in the toaster for 3 more minutes for getting the most soft structure that helps in taking the mold's shape, next, the molten NFC was placed in a preheated female mold at 80 ℃ and pressed using a screw clamp on the male part of the mold. Thus, the aforementioned NFC was cooled down inside the mold for 15 min at a room temperature of 18 ℃. Meanwhile, specimens with HDPE matrix were prepared through similar process, yet the frst melting stage took 3 min, and 2 min after adding the fbers, which was due to the low melting temperature of HDPE (190 ℃). Furthermore, specimens' dimensions were  $120\times20\times5$  mm following the ASTM 3039 standard. A total of 72 samples were prepared for the tensile test by considering 8 diferent NFCs i.e. palm/epoxy, palm/BioPoxy, palm/ PP, palm/HDPE, luffa/epoxy, luffa/BioPoxy, luffa/PP, and lufa/HDPE with fbers volume fraction of 0.1, 0.2, and 0.3. Thus, each NFC combination had 3 replicated samples.

## **Tensile Test**

The tensile test was conducted using a Hounsfeld universal machine and a laser extensometer. Two refective tapes were taped to each specimen in order to test strain variations through the extensometer. The considered gage length was 60 mm, and the speed rate was 5 mm/min [\[50](#page-16-0)]. Following ASTM D3039 standard regarding chopped and randomly oriented composite materials, both sides of all 72 specimens were covered with emery cloth (grade 100 sand papers), which contributed in increasing the grip and preventing the samples from slipping out of the machine clamps. After tightening the machine clamps on the specimens' edges, the applied force as well as the strain rate were adjusted to be zero. Hence, tension load was applied upon the specifed speed rate and the specimens extended till failure. Results were revealed through stress–strain graphs as well as excel fles that included whole details of force, break distance, ultimate tensile strength, and strain.

## **Finite Element Analysis**

Selected materials in this research in the numerical analysis are: BioPoxy 36, epoxy, high density polyethylene and polypropylene as a matrix and date palm and lufa fbers as a reinforcement. The matrixes and fbers were assumed to be homogenized and isotropic. Different fiber volume fractions were considered (0.1 to 0.3) in order to evaluate the effect of fber content on NFC elastic properties. RVE with random chopped orientations was implemented for predicting the elastic properties of BioPoxy/palm, BioPoxy/luffa, Epoxy/palm, Epoxy/lufa, PP/palm, PP/lufa, HDPE/palm, and HDPE/lufa. Figure [2](#page-4-0) shows the utilized representative volume element unit cell.

ANSYS "Materials Designer" tool was utilized, which automatically applies the approach of representative volume element homogenization method. Fibers' diameter was considered to be 5 μm and the RVE geometry was assigned to be square. Fibers' to matrix bonding was considered to be free of faws, and the natural fbers composites were considered to be free of voids. Meshing type utilized for the representative volume element was conformal. Furthermore, Orthotropic output of RVE chopped was assigned into a  $120 \times 20 \times 5$  mm beam (following ASTM D3039), thereby a tensile load was applied on the beam till its failure in order to measure its tensile strength. Materials properties considered for the simulation are displayed in Table [1](#page-4-1).



**Fig. 2** RVE with randomly oriented chopped fbers

<span id="page-4-1"></span><span id="page-4-0"></span>**Table 1** Input properties of the selected materials for simulation

Materials	Young's modulus	Poisson's ratio		
Date Palm	700 MPa	0.19		
Luffa	80 MPa	0.3		
BioPoxy 36	1850 MPa	0.3		
Epoxy	23 MPa	0.3		
PP	630 MPa	0.3		
<b>HDPE</b>	150 MPa	0.28		

## **Machine Learning Models**

Machine learning is a subcategory of artifcial intelligence, it is a technique where the computers learn the way of doing something that is generally particular to human and gained through experience. Usually, the efficiency of the algorithm increases by increasing the quantity of learning samples [\[51](#page-16-1)]. Deep learning became popular in many research areas since 2006, where it was implemented for determining the performance in felds like speech recognition, object recognition, image segmentation, and machine translation. Majority of deep learning approaches are usually presented as deep neural networks as they involve neural network architecture. There are two types of machine learning algorithms, supervised and unsupervised. Supervised machine learning proved its convenience in most manufacturing applications as the aforementioned provide labeled data [[52\]](#page-16-2).

$$
prediction error % = \frac{|Expt.Value - Pred.Value|}{Expt.Value} \times 100
$$
\n(1)

Prediction error is a simple method that evaluates the reliability of a training model, where the prediction model is

validated through new input data that were unconsidered previously in testing the model. Therefore, the error percentage of a training model can be defned using this tool. Moreover, a common technique for defning the error of a model is Root Mean Square Error (RMSE).

$$
RMSE = \sqrt{\frac{1}{N} \sum_{i=1}^{N} (pi - qi)^2}
$$
 (2)

where qi is the actual value, pi is the prediction of the deliberate information, and N is the complete training data.

Artifcial neural network, Response Surface metamodel, adaptive neuro-fuzzy inference system, and support vector machine were implemented in this research to defne the design space upon the considered parameters, and determine most convenient approach in predicting tensile strength values of input parameters that were unconsidered in the experimental tensile test of natural fbers composites. Inputs of the proposed model are matrix types, fbers' types, and fbers' volume fraction. Results of all experimentally tested specimens were applied for training, testing, and validation. Reliability of TS prediction model was evaluated using the mean absolute percentage error.

# **Results and Discussion**

This section presents results of tensile properties obtained from tensile test experiment, fnite element analysis, and machine learning of palm and luffa NFCs in Epoxy, BioPoxy 36, HDPE and PP matrices.

#### **Tensile Test Results**

Tensile test fndings are listed in this section, which includes tensile strength, strain, and young's modulus of lufa as well as palm natural fbers composites in BioPoxy, epoxy, HDPE and PP matrices. First, some stress and strain charts are displayed to show the tensile behavior of these NFCs, then the efect of increasing the natural fbers volume fraction in the considered matrices. Figure [3](#page-5-0) shows the stress–strain behavior of BioPoxy NFC with 0.1 lufa fbers.

As shown in Fig. [3](#page-5-0), the stress increased gradually to reach a yield strength of 3.77 MPa at 0.262%, then it followed a continuous increase to attain an ultimate tensile strength of 35.3 MPa at 4.31% straight before its brittle failure. The stress and strain behavior of palm/biopoxy NFC at 0.3 is shown in Fig. [4](#page-5-1).

As shown in Fig. [4](#page-5-1), ecopoxy with 0.3 palm NFC recorded a yield strength of 2.89 MPa at 0.135%, thereby the stress increased notably to reach an ultimate tensile strength of 20.2 MPa at 2.51%, thus, a brittle failure was observed at a



<span id="page-5-0"></span>**Fig. 3** Stress and strain curve of lufa/biopoxy "0.1"



<span id="page-5-1"></span>**Fig. 4** Stress and strain curve of palm/biopoxy "0.3"



<span id="page-5-2"></span>**Fig. 5** Stress and strain curve of palm/epoxy "0.3"

strain of 2.78%. Figure [5](#page-5-2) displays the stress–strain behavior of epoxy NFC with 0.3 palm.

As exhibited in Fig. [5,](#page-5-2) a yield strength of 1.67 MPa was observed at 1.8%, then the stress increased to reach a 4.13 MPa ultimate tensile strength at 14.1%, which emphasizes the notable ductility of this material, thereby the Palm/ Epoxy NFC performed a plastic failure at 17.5%. Figure [6](#page-5-3) shows the stress and strain behavior of epoxy natural fbers composite with 0.3 luffa fibers.



<span id="page-5-3"></span>**Fig. 6** Stress and strain curve of lufa/epoxy "0.3"



<span id="page-5-4"></span>**Fig. 7** Stress and strain curve of palm/PP "0.1"

As Fig. [6](#page-5-3) shows, the stress increased gradually and a yield strength of 1.5 MPa was exhibited at 1.22%, then the stress continued increasing along with a notable increment in the strain, hence an ultimate tensile strength of 3.43 MPa was observed at 16.1%. Figure [7](#page-5-4) displays the stress–strain curve of polypropylene NFC with 0.1 palm fbers.

As exhibited in Fig. [7](#page-5-4), polypropylene with 0.1 pam fibers revealed a yield strength of 3.07 at 0.143%, then the stress drastically increased to attain an ultimate tensile strength of 24.7 MPa at 5.52%, next it decreased to 23 MPa at 5.58% right before its brittle failure at 5.61%. The stress and strain behavior of lufa/PP NFC at 0.1 is illustrated in Fig. [8](#page-6-0).

As shown in Fig.  $8$ , polypropylene with 0.1 luffa fibers revealed a yield strength of 3.01 MPa at 0.139%, followed by a signifcant stress increment where a 22.6 MPa ultimate tensile strength was exhibited at 2.87%, right before its failure at 2.91%. Figure [9](#page-6-1) displays the stress and strain behavior of palm/HDPE NFC at 0.3.

As displayed in Fig. [9,](#page-6-1) HDPE NFC with 0.3 palm fibers exhibited a yield strength of 1.47 MPa at 0.053%, thereby the stress increased to reach an ultimate tensile strength of 11.1 MPa at 6.09%, followed by a notable decrease throughout the necking phase, hence a ductile failure was exhibited



<span id="page-6-0"></span>**Fig. 8** stress and strain curve of lufa/PP "0.1"



<span id="page-6-1"></span>**Fig. 9** Stress and strain curve of palm/HDPE "0.3"



<span id="page-6-2"></span>**Fig. 10** Stress and strain curve of lufa/HDPE "0.1"

at 6.7%. Figure [10](#page-6-2) shows the stress–strain behavior of highdensity polyethylene reinforced with 0.1 luffa fibers.

As observed in Fig. [10](#page-6-2), a yield strength of 1.28 MPa was observed at 0.027%, then a gradual increase in the stress was observed along with an increase in the strain to reach an ultimate tensile strength of 17.7 at 7.15%. Hence, the stress followed a descending trend till its ductile failure at 21.7%. It is worthy to mention that natural fbers composites with HDPE matrix revealed the highest ductility.

<span id="page-6-3"></span>

As shown in Table [2](#page-6-3), natural fbers' reinforced BioPoxy displayed the highest tensile strengths along with the least strain values. While palm as well as luffa reinforced epoxy exhibited the lowest TS compared to the selected matrixes, yet it provided the highest strain record in this study. Moreover, PP matrix revealed a notable TS (between 18 and 25 MPa), while the tensile strength of HDPE samples ranged between 10 and 17 MPa.

As illustrated in Fig. [11,](#page-7-0) In terms of BioPoxy/palm NFC, the tensile strength increased from 20.8 MPa to 23.6 by increasing the fbers volume fraction from 0.1 to 0.2, thus it decreased back to 21.05 MPa by reaching 0.3. Whereas reinforcing BioPoxy with Lufa fbers exhibited the highest TS outcome in this research, increasing the fbers volume fraction of luffa from 0.1 to 0.3 increased the tensile strength from 33.16 to 35.5 MPa respectively. It worthy to mention that TS of pure BioPoxy 36 is 57.9 MPa. However, highest TS observed in BioPoxy/palm was 23.6 MPa at 0.2, and in BioPoxy/lufa was 35.3 MPa at 0.3. Addition of lufa fbers into BioPoxy resin contributed in higher tensile strengths compared to palm fbers.

As Exhibited in Fig. [12,](#page-7-1) Epoxy with 0.1 palm exhibited a tensile strength of 2.37 MPa, by growing the fbers' content, TS increased to 3.32 MPa at 0.2 and 4.34 MPa at

<span id="page-7-1"></span><span id="page-7-0"></span>

<span id="page-7-2"></span>0.3. While lufa reinforced epoxy displayed a TS 2.85 MPa at 0.1 fbers' volume fraction, then reached 3.69 MPa by growing the fbers content up to 0.3. Adding lufa as well as palm fbers into epoxy matrix improves the tensile strength, yet, palm had a better impact than luffa fibers. Highest TS observed in NF reinforced epoxy was 4.34 MPa in epoxy with 0.3 palm fbers.

Regarding palm fibers reinforced polypropylene, as shown in Fig. [13](#page-7-2) increasing the fbers' content from 0.1 to 0.3 decreased the tensile strength from 25.23 to 21.1 MPa respectively. Similarly, increasing  $V_f$  of luffa fibers in PP reduced TS from 22.57 MPa at 0.1 to 18.33 MPa at 0.3 respectively. Furthermore, addition of palm fbers in PP matrix resulted greater TS values than that of PP/luffa, thus peak TS was 25.23 MPa observed in PP with 0.1 palm fbers. It is worthy to mention that the tensile test fndings of PP/ lufa were in accordance with Demir et al. [[53\]](#page-16-3) at 0.1 fbers'

volume fraction, whereas the results of PP/palm agreed with the fndings of Otaibi et al. [\[54](#page-16-4)] at 10 wt%.

As clearly shown in Fig. [14](#page-8-0), loading palm fbers into HDPE matrix reduced the tensile strength to 12.57 MPa at a volume fraction of 0.3, yet it was 16.97 MPa at 0.1 and 15.4 MPa at 0.2. Reinforcing HDPE with lufa fbers reduced the tensile strength from 16.6 MPa at 0.1 to 10.12 MPa at 0.3 respectively. HDPE/palm and HDPE/lufa exhibited identical TS values at 0.1 and 0.2 fbers' volume fraction, whilst palm fbers had a better efect at 0.3. Results of HDPE/palm agreed with mulinari et al. [\[55](#page-16-5)] at 10 wt% and Mahdavi et al. [[56\]](#page-16-6) at 20 wt%. However, the observed results highlight the potential of palm/biopoxy to be used in industrial applications, lufa/biopoxy for aircraft minor components, palm/ epoxy and lufa/epoxy for appliances coating applications, palm/PP and lufa/PP for automotive parts, and palm/HDPE and lufa/HDPE for bio-packaging.

<span id="page-8-0"></span>

### **FEA and Experimental Results**

Since RVE chopped was observed to be the most accurate model due to its non-linear trends and its agreement with the literature, this model was involved in analyzing the elastic properties of the materials utilized for conducting the tensile test. In ANSYS Explicit Dynamics space, orthotropic output of RVE chopped was assigned as an input into a  $120 \times 20 \times 5$  mm beam following ASTM D3039 (Fig. [15](#page-9-0)). After meshing the sample with linear elements order, the total number of nodes was 693 and the number of elements was 384. Moreover, bottom of the sample was fxed through applying a nodal displacement of 0 mm, and a nodal force was applied on the top of the sample similar to the real tensile test. Furthermore, all models were analyzed by frstly applying a low nodal force on the beam, thereby increasing it till its failure in order to measure its tensile strength. However, each natural fber composite required diferent nodal force to break the sample. Figure [15](#page-9-0) shows FEA beam model of (a) palm/biopoxy at 0.2, (b) palm/Epoxy at 0.3, (c) palm/ PP at  $0.1$ , (d) palm/HDPE at  $0.1$ , (e) luffa/biopoxy at  $0.1$ , (f) luffa/epoxy at  $0.3$ , (g) luffa/PP at  $0.2$ , and (h) luffa/HDPE at 0.1.

This section compares the tensile strength obtained through RVE chopped model followed by FEA simulation, and the conducted experiment. As shown in Fig. [16,](#page-10-0) experimental tensile strength of BioPoxy/palm increased from 20.8 MPa to 23.6 MPa by increasing the fbers' volume fraction from 0.1 to 0.2, then it dropped to reach a TS of 21.050 MPa at 0.3. A common behavior was displayed by FEA model, TS increased from 20.617 to 22.726 MPa at 0.2, thereby decreased to 22.28 MPa at 0.3. Whereas experimental TS of biopoxy/lufa increased from 33.17 to 35.3 MPa through increasing  $V_f$  from 0.1 to 0.3, similarly FEA tensile strength followed an ascending trend to reach a TS value of 33.011 MPa at 0.3. Tensile strength results obtained from FEA showed a good agreement with the experimental findings.

As shown in Fig. [17](#page-10-1), both methods showed increasing trends while increasing the fibers' content of luffa as well as palm. However, tensile strength of epoxy/palm observed in FEA model increased from 2.45 to 3.31 MPa by increasing  $V_f$  from 0.1 to 0.2, and to 3.79 MPa by increasing  $V_f$  up to 0.3, whilst the experimental TS of epoxy/palm increased to 3.32 MPa and 4.34 MPa respectively. In terms of epoxy/ lufa, the numerically observed tensile strength increased to 3.17 MPa at 0.2, and 3.47 MPa from 0.2 to 0.3, thus the corresponding experimental results rose to 3.1 MPa by increasing the fbers' volume fraction from 0.1 to 0.2, and 3.69 MPa from 0.2 to 0.3. FEA fndings signifcantly agreed with the tensile test results.

As seen in Fig. [18](#page-10-2), considering both FEA and experimental results, increasing the fbers' content to 0.3 decreased TS of both PP/palm and PP/luffa, respectively. Experimental TS of PP/palm NFC, increasing  $V_f$  to 0.3 reduced the tensile strength to 22.47 MPa and 21.1 MPa respectively, similarly FEA TS followed a descending trend, TS decreased from 24.274 to 22.3 MPa by increasing the fbers' content to 0.2, and to 15.423 MPa at 0.3. While Regarding TS of PP/lufa, both methods displayed a continuous decline in tensile strength while increasing the fbers volume fraction. Moreover, the simulation results notably agreed with the experimental fndings.

As shown in Fig. [19,](#page-10-3) increasing  $V_f$  of palm from 0.1 to 0.2 in HDPE matrix decreased the tensile strength from 16.97 to 15.622 MPa, consequently it decreased to 12.57 MPa at 0.3, similarly TS of HDPE/palm observed in FEA model exhibited a continuous decline to reach a TS value of 11.375 MPa at 0.3. Furthermore, addition of lufa fbers in HDPE reduced the tensile strength, that was observed through FEA simulation as well as FEA model. It is worthy to mention that the overall agreement between FEA simulation and tensile test results is quite acceptable [[57](#page-16-7), [58](#page-16-8)]. Although FEA and experimental TS were following same trends in PP/lufa and HDPE/lufa, the predicted values were slightly lower than the experimental.

#### **Machine Learning Findings**

Artificial neural network, multiple linear regression, support vector machine, and adaptive neuro-fuzzy inference system were adapted in this research to determine <span id="page-9-0"></span>**Fig. 15** FEA beam model following ASTM D3039 of **a** 0.2 Palm/biopoxy, **b** 0.3 Palm/ epoxy, **c** 0.1 Palm/PP, **d** 0.1 palm/HDPE, **e** 0.1 lufa/biopoxy, **f** 0.3 lufa/Epoxy, **g** 0.2 lufa/PP, and **h** 0.1 lufa/HDPE



the design space as well as to specify most convenient Machine Learning (ML) approach in predicting TS of NFCs. This section introduces the outcome of the aforementioned ML tools in predicting the tensile strength of palm NFCs as well as luffa NFCs. Levenberg-Marquardt algorithm was utilized in this research for training all components of ANN prediction model, which exhibited a swift and stable convergence. The design of the ANN model is

<span id="page-10-1"></span><span id="page-10-0"></span>

<span id="page-10-3"></span><span id="page-10-2"></span>shown in Fig. [20](#page-11-0) The model includes 3 inputs, 8 hidden layers, and 1 output.

The model was generated using neural network ftting tool in MATLAB. Input data consisted of: matrix type, fbers' type, and fbers' volume fraction, while the output was tensile strength. Data of all experimentally tested NFC specimens were considered, 70% of the data were used for training the model, 15% for testing, and 15% for validation. Figure [21](#page-11-1) displays the schemes of tensile strength regressions for all considered NFCs throughout three diferent

#### <span id="page-11-0"></span>**Fig. 20** Artifcial neural network model structure



<span id="page-11-1"></span>



fibers' volume fractions i.e. 0.1, 0.2, and 0.3. This figure highlights the correlation between the ANN output and the experimental data (target).

The correlation between the output values and target values were represented through the solid line, while best correlation that can be generated is represented through the dotted line. The overall regression coefficient of ANN model was observed to be 0.989, which means it is satisfactory as it is close to 1. In this research, regression model was implemented using curve ftting tool in MATLAB. In order to reach a better curve ftting results, two diferent response surface models were generated using cubic polynomial approximation functions, one for palm NFCs and the other for lufa NFCs. Therefore, input data included matrix type and fbers volume fraction, whereas the output was the corresponding TS values. Moreover, response surface metamodel provides a surface ftting that covers the design in order to predict responses for inputs not considered throughout the



<span id="page-12-0"></span>**Fig. 22** Response surface ftting of **a** Palm NFCs, and **b** Lufa NFCs

experiment. A cubic polynomial approximation function is involved in study to develop the response surfaces shown in Fig. [22a](#page-12-0), b.

MATLAB cftool was utilized for developing the RSM for TS of palm NFCs as well as lufa NFCs. Goodness of ft were evaluated using sum of square error (SSE), R-square adjusted, and root mean square error (RMSE). The values of R-square adjusted ranges between 0 and 1, where a good ft would get a value close 1. In contrast, RMSE and SSE values should be close to zero in order to have a good surface ft. The goodness of ft included an SSE of 44.32, R-square of 0.9781, Adjusted R-square of 0.9705 and RMSE of 1.388. While regarding the RSM model of lufa NFCs, the SSE: 104.6, R-square: 0.977, Adjusted R-square: 0.9699, and RMSE: 2.006. The considered ANFIS model consists of; 3 inputs (matrix type, fibers' type, and  $V_f$ ), 4 membership function for the frst input, 2 for the second input, and 3 membership functions for the third input. Structure of the ANFIS model is illustrated in Fig. [23](#page-12-1).

Neuro fuzzy designer tool in MATLAB was utilized for applying ANFIS model. 80% of the experimental results were utilized for training the model and the remaining 20% were used for the testing. All fbers, matrixes, and fbers' volume fractions were considered. FIS model was generated using Gaussmf membership function with a constant output. Training was completed at epoch 2, and average testing error was observed to be 1.4876. ANFIS model plot that displays the training data as well as the FIS output data is shown in Fig. [24.](#page-13-0)

The support vector machine model in this research was generated through regression learner tool in MATLAB. 4 Matrix types, 2 fibers' type, and 3 fibers volume fraction were considered as input parameters. Figure [25](#page-13-1) shows the training and testing output datasets developed by the SVM model. Blue dotes illustrates the true data, while the yellow dots are the predicted data.



Goodness of fit: SSE: 104.6 R-square: 0.977 Adjusted R-square: 0.9699



<span id="page-12-1"></span>**Fig. 23** ANFIS model structure

Hence, experimental results of all NFCs were utilized for training the SVM model, the kernel function used was the Gaussian, and the involved preset was the fne Gaussian SVM. Moreover, the obtained RMSE was 2.7296, R-squared was 0.93, and MSE was 7.4509. Table [3](#page-14-0) shows the predicted data using machine learning methods as well as real tensile strength results of palm and luffa NFCs.

As clearly shown in Table [3](#page-14-0) the predicted TS using ANN, and ANFIS, displayed a strong agreement with experimental results with a prediction error of 3.21% and 2.17%, respectively, whilst TS values obtained through MLR and SVM showed a decent agreement with an error of 6.86% and 12.65%. That emphasizes that these models are able to be trained and involved in assuming the tensile strength of natural fbers composite materials. Furthermore, lufa/Bio-Poxy at 0.3  $V_f$  revealed the highest tensile strength values throughout all the considered ML approaches. Most suitable

<span id="page-13-0"></span>



<span id="page-13-1"></span>



machine learning approach for predicting tensile strength of natural fbers composites is ANFIS as it showed least prediction error "2.17%".

# **Conclusion**

This research presents an investigation on the mechanical properties of lufa and palm NFCs in Epoxy, Ecopoxy "BioPoxy 36", PP and HDPE matrixes. RVE chopped model was considered for analyzing the orthotropic properties of palm and luffa NFCs, thereby, the output of RVE chopped was assigned into a  $120 \times 20 \times 5$  mm beam as for ASTM D3039, and loaded under tensile load till failure. Moreover, experimental tensile test was conducted for testing the tensile properties of the considered NFCs as well as validating the simulation results. Thus, diferent machine learning techniques were implemented in order to identify the design space i.e. Artifcial neural network, multiple linear regression, adaptive neuro-fuzzy inference system, and support vector machine. Effect of increasing fiber content in the matrix was taken into account by considering several fbers' volume fractions, from 0.1 to 0.5 in model validation and from 0.1 to 0.3  $V_f$  for the simulation, experiment <span id="page-14-0"></span>ANFIS, and SVM

Matrix	Fibers	$V_f$	Experimental TS (MPa)	<b>ANN</b> TS (MPa)	MLR TS (MPa)	<b>ANFIS</b> TS (MPa)	<b>SVM</b> TS (MPa)
BioPoxy 36	Palm	0.1	20.800	20.550	20.810	21.100	19.835
BioPoxy 36	Palm	0.2	23.600	23.250	22.851	23.800	22.256
BioPoxy 36	Palm	0.3	21.050	20.150	21.943	21.900	20.646
Epoxy	Palm	0.1	2.370	2.423	3.090	2.070	3.928
Epoxy	Palm	0.2	3.323	3.275	3.906	3.260	4.411
Epoxy	Palm	0.3	4.343	4.235	3.027	4.343	5.489
PP	Palm	0.1	25.233	24.850	23.887	25.100	24.242
PP	Palm	0.2	22.467	22.150	23.055	22.467	22.043
PP	Palm	0.3	21.100	20.950	21.783	20.700	20.245
<b>HDPE</b>	Palm	0.1	16.967	17.110	17.502	16.750	16.828
<b>HDPE</b>	Palm	0.2	15.400	14.900	14.600	15.400	16.459
<b>HDPE</b>	Palm	0.3	12.567	12.320	12.513	12.567	12.642
BioPoxy 36	Luffa	0.1	33.167	33.150	33.188	33.167	31.041
BioPoxy 36	Luffa	0.2	35.133	34.980	34.674	34.550	30.356
BioPoxy 36	Luffa	0.3	35.300	36.090	35.828	34.900	33.637
Epoxy	Luffa	0.1	2.853	2.419	3.040	3.130	3.852
Epoxy	Luffa	0.2	3.097	3.282	3.624	3.097	4.370
Epoxy	Luffa	0.3	3.687	4.266	2.912	3.765	4.730
PP	Luffa	0.1	22.567	22.440	22.042	22.500	22.060
PP	Luffa	0.2	21.300	21.210	21.535	21.800	20.835
PP	Luffa	0.3	18.333	17.370	18.768	18.333	17.460
<b>HDPE</b>	Luffa	0.1	16.600	17.630	17.116	16.600	16.661
<b>HDPE</b>	Luffa	0.2	15.433	15.280	15.328	15.150	16.255
<b>HDPE</b>	Luffa	0.3	10.123	10.230	10.318	9.285	10.546
Prediction error %				3.21%	6.86%	2.17%	12.65%

and machine learning models. Regarding the experimental tensile test, NFCs with biopoxy matrix showed the greatest tensile strength values and lowest strain, whereas NFCs with epoxy matrix displayed least tensile strengths in this research with highest strain values.

Furthermore, NFCs with PP matrix exhibited signifcant tensile strength values ranging between 18 and 25 MPa. TS of NFCs with HDPE matrix ranged from 10 to 17 MPa. Peak tensile strength observed in lufa/BioPoxy at 0.3 with a value of 35.5 MPa. Increasing the fbers' volume fraction from 0.1 to 0.2 of natural fbers' reinforced biopoxy increased TS then decreased at 0.3. Reinforcing BioPoxy with luffa fibers showed better properties compared to date palm fbers. Growing the content of palm as well as luffa fibers' up to 0.3 increased the tensile strength of epoxy. Peak TS in natural fbers' reinforced epoxy was 4.34 MPa in palm/epoxy at 0.3. While addition of luffa and palm fibers decreases the tensile strength of HDPE and PP matrixes, however, palm revealed higher TS values in PP (25.23 MPa at 0.1) and in HDPE (16.97 MPa at 0.1). Based on the revealed results, palm/ BioPoxy can be utilized for industrial applications, lufa/ BioPoxy can be used for producing aircraft components, lufa/epoxy and palm/epoxy can be involved in appliances'

coating, while lufa/PP and palm/PP have the ability to emerge in automotive parts' production. It is worthy to mention that simulation results signifcantly agreed with the tensile test fndings, where these FEA results followed the exact same trends observed in the experimental fndings, as well as majority of the TS values were in accordance. Furthermore, in terms of machine learning application, predicted tensile strengths through ANN, and ANFIS drastically agreed with the experimental outcome, having a prediction error of 3.21% and 2.17%. Whereas tensile strength predicted using MLR and SVM displayed an acceptable agreement with a prediction error of 6.86% and 12.65%. Which therefore provides evidences the capability of these machine learning models of being trained and utilized for predicting TS of natural fbers composites. Moreover, throughout all the considered ML approaches, highest tensile strength values were revealed in luffa/BioPoxy at  $0.3 V_f$ . ANFIS can be considered as suitable machine learning approach for predicting tensile strength of natural fbers composites as it exhibited lowest prediction error.

**Funding** The authors have not disclosed any funding.

#### **Declarations**

**Conflict of interest** The authors declare that they have no confict of interest.

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